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## Tuning the electronic properties of hexanuclear cobalt sulfide superatoms *via* ligand substitution†

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Molecular clusters are attractive superatomic building blocks for creating materials with tailored properties due to their unique combination of atomic precision, tunability and functionality. The ligands passivating these superatomic clusters offer an exciting opportunity to control their electronic properties while preserving their closed shells and electron counts, which is not achievable in conventional atoms. Here we demonstrate this concept by measuring the anion photoelectron spectra of a series of hexanuclear cobalt sulfide superatomic clusters with different ratios of electron-donating and electron-withdrawing ligands,  $\text{Co}_6\text{S}_8(\text{PET}_3)_{6-x}(\text{CO})_x$  ( $x = 0-3$ ). We find that  $\text{Co}_6\text{S}_8(\text{PET}_3)_6$  has a low electron affinity (EA) of 1.1 eV, and that the successive replacement of  $\text{PET}_3$  ligands with CO gradually shifts its electronic spectrum to lower energy and increases its EA to 1.8 eV. Density functional theory calculations reveal that the increase of EA results from a monotonic lowering of the cluster highest occupied and lowest unoccupied molecular orbitals (HOMO and LUMO). Our work provides unique insights into the electronic structure and tunability of superatomic building blocks.

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### Introduction

Electron affinity and ionization energy are fundamental properties of the elements. Together they govern the interactions and bonding of close-contacting atoms and control the collective properties of solids. Because they are intrinsic to each element, however, the electron affinity and ionization energy of a given atom cannot be altered. This presents an immense synthetic challenge for the design of tunable materials as substituting atoms often leads to entirely new structures, interactions and collective behaviors.

By analogy to “atomic” building blocks, certain clusters can be used as “superatomic” building blocks for the assembly of novel materials.<sup>1-22</sup> Within this context, the family of metal chalcogenide molecular clusters has recently received renewed attention for the creation of functional solids with tunable properties, including ferromagnetism, electrical conductivity, tunable optical gaps and thermal switching.<sup>1,7,23-32</sup> One of the key advantages of this approach over traditional atomic solids is

that the characteristics of the building blocks can be tuned pre-assembly without changing the total electron count of the superatom. Using the molecular cluster  $\text{Co}_6\text{Te}_8\text{L}_6$  ( $L =$  passivating ligand) as a model system, Khanna *et al.* recently predicted theoretically that changing  $L$  from  $\text{PET}_3$  to CO would increase its electron affinity, in effect transforming the cluster from a superatomic alkali metal toward a superatomic halogen.<sup>33,34</sup> The potential to alter the donor/acceptor property of these superatomic clusters sheds light on assembling superatomic materials with fine-tuned properties. Experimentally demonstrating this concept by measuring the electron affinity and/or ionization energy of metal chalcogenide clusters is challenging, however, because it requires bringing the charged clusters into the gas phase without damaging the inorganic core or dissociating the ligands.

In this work, we use anion photoelectron spectroscopy to probe the electron affinity (EA) and electronic structure of a series of cobalt sulfide clusters, whose ligand shells consist of differing combinations of  $\text{PET}_3$  and CO. The clusters  $\text{Co}_6\text{S}_8(\text{PET}_3)_{6-x}(\text{CO})_x$  are synthesized from the parent compound  $\text{Co}_6\text{S}_8(\text{PET}_3)_6$  by ligand substitution with CO. These clusters are then brought into the gas phase, where electrons are attached to form anions, using a unique infrared desorption/photoelectron-emission/supersonic helium expansion source. Mass spectrometry confirms the existence of carbonylated products with  $x$  up to 3 (*i.e.*  $\text{Co}_6\text{S}_8(\text{PET}_3)_5(\text{CO})^-$ ,  $\text{Co}_6\text{S}_8(\text{PET}_3)_4(\text{CO})_2^-$ , and  $\text{Co}_6\text{S}_8(\text{PET}_3)_3(\text{CO})_3^-$ ). We find that the electron affinity and vertical detachment energy increase with the number of CO ligands,

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demonstrating the electronic spectral tunability of this family of superatoms. These results are further examples of the superatom concept, and an important step towards designing materials from programmable building blocks.

## Methods

### Synthesis

Dicobalt octacarbonyl and sulfur were purchased from Strem Chemicals. Triethylphosphine was purchased from Alfa Aesar. All other reagents and solvents were purchased from Sigma Aldrich. Dry and deoxygenated solvents were prepared by elution through a dual-column solvent system. Unless otherwise stated, all reactions and sample preparations were carried out under inert atmosphere using standard Schlenk techniques or in a N<sub>2</sub>-filled glovebox. While a multi-step synthesis of Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6</sub> has been previously reported,<sup>35</sup> we have developed an alternative one-pot approach detailed below.

**Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6</sub>.** Sulfur (1.16 g, 36.25 mmol) was suspended in ~30 mL of toluene in a 200 mL Schlenk flask under N<sub>2</sub> atmosphere. In two separate flasks, Co<sub>2</sub>(CO)<sub>8</sub> (4.12 g, 12.05 mmol) and PET<sub>3</sub> (4.27 g, 36.14 mmol) were dissolved in ~20 mL of toluene. The Co<sub>2</sub>(CO)<sub>8</sub> solution was added to the S suspension, followed by quick addition of the PET<sub>3</sub> solution. The reaction mixture was refluxed under N<sub>2</sub> for 2 days. The reaction mixture was then opened to air, and hot filtered through Celite. The filtrate was cooled to room temperature and left to stand for ~3 h. Black crystals formed during that period; the resulting suspension was filtered through a fine frit and the solid was washed with toluene, and diethyl ether. The dark, black crystals were collected, dried *in vacuo*, and stored under N<sub>2</sub>. Yield: 2.2 g (42%). MS-MALDI *m/z*<sup>+</sup> calculated 1317.92; found, 1317.95.

**Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub>.** Safety note: CO is a toxic gas; this reaction can only be performed in a well-ventilated fumehood. Sulfur (0.15 g, 4.68 mmol) and Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6</sub> (1.00 g, 0.76 mmol) were combined in ~75 mL of toluene in a 200 mL Schlenk flask. An external bubbler was attached to the system, and the mixture was sparged with CO for 30 min. The mixture was heated to 100 °C and stirred under a CO atmosphere for 16 h. The reaction mixture was cooled to room temperature, sparged with N<sub>2</sub>, and the solvent was removed *in vacuo*. The resulting solid was used directly without further purification.

### Anion photoelectron spectroscopy

Anion photoelectron spectroscopy was conducted by crossing a mass-selected negative ion beam with a fixed energy photon beam and analyzing the energies of the resultant photo-detached electrons. This technique is governed by the well-known energy-conserving relationship,  $h\nu = \text{EBE} + \text{EKE}$ , where  $h\nu$ , EBE, and EKE are the photon energy, electron binding energy (photodetachment transition energy), and the electron kinetic energy, respectively. The details of our apparatus have been described elsewhere.<sup>36</sup> Briefly, the photoelectron spectra were collected on an apparatus consisting of an ion source, a linear time-of-flight mass spectrometer for mass analysis and selection, and a magnetic-bottle photoelectron spectrometer for

electron energy analysis (resolution ~35 meV at 1 eV EKE). The third harmonic (355 nm, 3.49 eV per photon) of a Nd:YAG was used to photodetach electrons from the cluster anion of interest. Photoelectron spectra were calibrated against the well-known atomic lines of the copper anion, Cu<sup>-</sup>.

To make the parent anions Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub><sup>-</sup> in the gas phase, a specialized infrared desorption/laser photoemission (IR/PE) supersonic helium expansion source is employed.<sup>37</sup> Briefly, a low power IR pulse (1064 nm) from a Nd:YAG laser hits a translating graphite bar thinly coated with the Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub> sample. Because the graphite absorbs most of the energy, a localized thermal shock lasting a few nanoseconds propels the clusters into the gas phase. Almost simultaneously, a high power pulse of 532 nm light from a second Nd:YAG laser strikes a close-by photoemitter (Hf wire), creating a shower of electrons that attach to the evaporated neutral clusters. Also, almost simultaneously, a plume of ultrahigh purity helium gas rapidly expands from a pulsed valve located slightly upstream, cooling the nascent anions and directing them into the mass spectrometer, where they are analyzed. Because the Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub> clusters are slightly sensitive to air and moisture, they are coated onto the graphite bar in a N<sub>2</sub>-filled glove box. The graphite bar is then enclosed inside an air-tight “suitcase” container, which is only opened under high vacuum after being transferred to the vacuum chamber.

### Theoretical calculations

First-principles electronic structure calculations on the anion and neutral forms of Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub> ( $x = 0-3$ ) clusters were carried out within density functional theory formalism. The ADF set of codes were used to perform the calculations where a PBE0 hybrid functional comprised of the PBE generalized gradient functional and 25% Hartree-Fock exchange, as proposed by Ernzerhof *et al.*, is used to incorporate exchange and correlation effects.<sup>38-40</sup> The atomic wave functions are expressed in terms of Slater-type orbitals (STO) located at the atomic sites and the cluster wave functions are constructed from a linear combination of these atomic orbitals.<sup>41</sup> A TZ2P basis set and a large frozen electron core was used. The zero-order regular approximation (ZORA) is used to include scalar-relativistic effect.<sup>42,43</sup> The trial structures of the clusters were taken from the previously optimized structure of the Co<sub>6</sub>Te<sub>8</sub>(PET<sub>3</sub>)<sub>6</sub> where the Te sites were replaced by S atoms.<sup>33</sup> The quasi-Newton method without any symmetry restriction is used to determine the lowest energy arrangement for each trial structure of the clusters. The calculations also covered the several possible spin states for all the clusters during optimization. The adiabatic electron affinity (AEA) is calculated as the total energy difference between the anion in its ground state geometry and the neutral cluster in its ground state geometry, while the vertical detachment energy (VDE) is given by the total energy difference between the anion in its ground state and the neutral cluster in the geometry of the anion.

## Results and discussions

Fig. 1a depicts the structure of Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub><sup>-</sup>. Our strategy to synthesize these clusters begins by preparing the parent



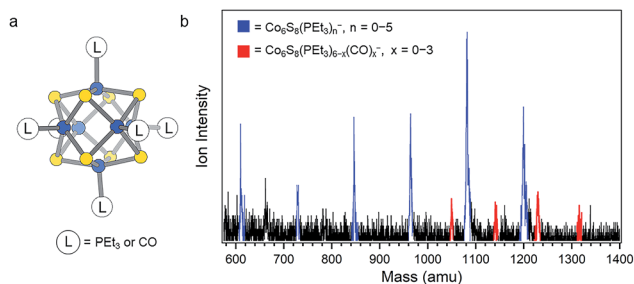


Fig. 1 (a) Structure of Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub>. (b) Anion mass spectrum of Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub><sup>-</sup> generated using IR/PE anion source. This work focuses on the red peaks in the mass spectrum.

compound Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6</sub> on a multigram scale. To achieve this, we have developed a new approach detailed in the Methods section. Briefly, under a CO atmosphere, Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6</sub> is reacted with six equivalents of S in toluene at ~100 °C to partially substitute CO for PET<sub>3</sub>, which is trapped as S=PET<sub>3</sub>. Using the mass spectrometry technique described below, we observe a mixture of clusters in which up to three CO ligands have been substituted for PET<sub>3</sub>. It is possible to separate these different species using column chromatography,<sup>30</sup> but for this study we can use the mixture of clusters without further purification.

Measuring the electron affinity and electronic structure of Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub> requires a negative ion photoelectron spectroscopic study on their parent anions, Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub><sup>-</sup>, in the gas phase. Common ionization methods (*e.g.* electrospray ionization and matrix-assisted laser desorption/ionization), however, fail to generate the parent anions Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub><sup>-</sup> in the gas phase due to the effect of the solvent/matrix on neutral Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub> clusters: the obtained anions are either dissociated or tagged by the solvent/matrix molecules. To make the parent anions in the gas phase, we instead employ a unique infrared desorption/photoemission (IR/PE) source as described in the Methods section. A typical anion mass spectrum of Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub><sup>-</sup> obtained with the IR/PE source is shown in Fig. 1b. The spectrum contains two major series of anions: Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>n</sub><sup>-</sup> (*n* = 0–5) and Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub><sup>-</sup> (*x* = 0–3). The Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>n</sub><sup>-</sup> clusters are from the sequential dissociation of PET<sub>3</sub> from Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6</sub> during infrared desorption, while the Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub><sup>-</sup> clusters are the anions of interest in this study. Though the Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub><sup>-</sup> signals are weaker, their high photodetachment cross-sections allow us to collect their anion photoelectron spectra.

Fig. 2 presents the anion photoelectron spectra of Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub><sup>-</sup> (*x* = 0–3) from which the adiabatic electron affinity (AEA) and vertical detachment energy (VDE) of each cluster are determined. The value of the AEA is taken to be the onset of the lowest electron binding energy (EBE) peak in the photoelectron spectrum. The VDE is the vertical transition energy from the ground state of the anion to the neutral state at the anion geometry. It is determined as the EBE value at the intensity maximum of the peak of interest, *i.e.* typically the first EBE peak. Based on the anion photoelectron spectra, the AEA values of Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub> are 1.1, 1.3, 1.7 and 1.8 eV for *x* = 0, 1, 2, 3, respectively, while the corresponding anion VDE

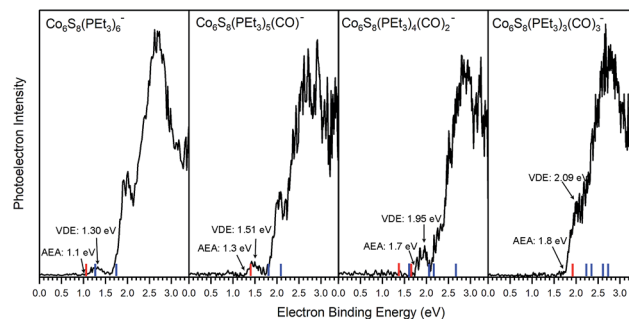


Fig. 2 Negative ion photoelectron spectra of Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub><sup>-</sup> (*x* = 0–3) anions collected using 355 nm (3.49 eV) photons from a Nd:YAG laser. The arrows point to the experimental values of AEA and VDE; the red lines correspond to theoretical AEA, and the blue lines correspond to vertical transitions from the anion to the singlet and triplet neutral states.

values are 1.30, 1.51, 1.95 and 2.09 eV, respectively. By sequentially substituting half of the PET<sub>3</sub> ligands with CO, the AEA of this superatom increases from 1.1 eV to 1.8 eV, all while maintaining the same oxidation state for the Co<sub>6</sub>S<sub>8</sub> core. As a reference, the reducing agent SO<sub>2</sub> and oxidizing agent SO<sub>3</sub> have AEA values of 1.11 eV and 1.90 eV,<sup>44,45</sup> respectively, where the increase in AEA is the result of the change in the S atom oxidation state. Thus, the observed changes in AEA values for the ligated superatoms are significant, given that the oxidation state of the Co<sub>6</sub>S<sub>8</sub> core does not change across the ligation series: these results underscore the remarkable characteristics of the superatoms. The observed unambiguous trend of increasing AEA and VDE values with ligand substitution is a direct evidence that the electronic properties of this superatom can be tuned effectively in this way.

The tuning of the electronic properties can be further demonstrated through the spectral features located at higher electron binding energies than that of the first EBE peak. These higher EBE peaks arise from transitions from the anion ground state to various excited electronic states of the neutral cluster. The shape of these features relates to the electronic structure of the neutral clusters. While the photoelectron spectra of Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub><sup>-</sup> shift to higher electron binding energy as *x* increases, the overall spectral shapes are remarkably similar, confirming that the sequential exchange of PET<sub>3</sub> with CO ligand leaves the essential electronic structure of Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub> superatoms largely unchanged. What is changing with sequential ligation, however, is a monotonic decrease in the energies of the highest occupied and lowest unoccupied molecular orbital (HOMO and LUMO) levels, even though the gap between them remains relatively constant. This is discussed further below.

To support these experimental observations, we modeled the ground state structures of Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub><sup>-</sup>, and calculated their AEA and VDE using density functional theory formalism (Fig. 3). In agreement with previous calculations of the Co<sub>6</sub>Te<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub> system,<sup>33</sup> we find that the anionic clusters Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>6-x</sub>(CO)<sub>x</sub><sup>-</sup> have doublet spin ground states, while the neutral species have singlet spin states. For Co<sub>6</sub>S<sub>8</sub>(PET<sub>3</sub>)<sub>4</sub>(CO)<sub>2</sub><sup>-</sup>,



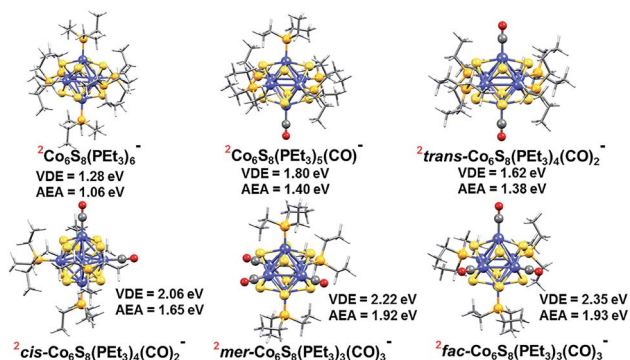


Fig. 3 Optimized ground state structure of anionic  $\text{Co}_6\text{S}_8(\text{PET}_3)_{6-x}(\text{CO})_x^-$  ( $x = 0-3$ ) clusters. The red superscript indicates the spin multiplicity ( $2S + 1$ ) of each cluster.

there are two possible structures: *trans*- $\text{Co}_6\text{S}_8(\text{PET}_3)_4(\text{CO})_2^-$  and *cis*- $\text{Co}_6\text{S}_8(\text{PET}_3)_4(\text{CO})_2^-$ , with the latter lower in energy by 0.20 eV. The replacement of three  $\text{PET}_3$  ligands with CO leads to *fac*- $\text{Co}_6\text{S}_8(\text{PET}_3)_3(\text{CO})_3^-$  and *mer*- $\text{Co}_6\text{S}_8(\text{PET}_3)_3(\text{CO})_3^-$  isomers, with the latter isomer lower in energy by 0.07 eV. The small energy difference between the  $\text{Co}_6\text{S}_8(\text{PET}_3)_3(\text{CO})_3^-$  isomers suggests their co-existence in the experimental beam. Table S1† contains more calculation results on neutral and anionic species.

We modeled the electronic structure of the superatoms using the anion photoelectron spectra as fingerprints of the ground state geometry and electronic states of both the neutral and anionic clusters.<sup>46</sup> Three points of comparison are used to establish the theoretical ground state of the system: AEA, VDE and  $\text{VDE}^*$ . VDE and  $\text{VDE}^*$  are the vertical transitions from the anion ground state to the neutral with singlet and triplet spin multiplicity, respectively. Experimentally, VDE and  $\text{VDE}^*$  correspond to the electron binding energies of the first (lowest EBE) and second (next highest EBE) peaks in the photoelectron spectra.

Table 1 presents both theoretical and experimental values of AEA, VDE and  $\text{VDE}^*$  for all the superatoms considered here. Using AEA, VDE and  $\text{VDE}^*$  as references to validate the computation, we observe close agreement between experimental and calculated values. The agreement between experiments and calculations is further demonstrated by including the computed values in the anion photoelectron spectra in

Fig. 2. The key result is that the AEA and both VDE values increase as  $\text{PET}_3$  ligands are successively replaced with CO. Starting with  $\text{Co}_6\text{S}_8(\text{PET}_3)_6^-$ , the calculated AEA, VDE, and  $\text{VDE}^*$  values of 1.06, 1.28, and 1.74 eV are close to the experimental values of 1.1, 1.30 eV, and 1.91 eV, respectively. Experimentally, the replacement of one  $\text{PET}_3$  with a CO ligand results in a  $\sim 0.2$  eV increase in the AEA (1.3 eV) and VDE values (1.51 and 2.05 eV), whereas the calculated AEA (1.40 eV) and VDE values (1.80 and 2.08 eV) for  $\text{Co}_6\text{S}_8(\text{PET}_3)_5\text{CO}^-$  are only slightly higher. For *cis*- $\text{Co}_6\text{S}_8(\text{PET}_3)_4(\text{CO})_2^-$  and *trans*- $\text{Co}_6\text{S}_8(\text{PET}_3)_4(\text{CO})_2^-$ , the computed ground states have AEA values of 1.65 and 1.38 eV, VDE values of 2.06 and 1.62 eV, and  $\text{VDE}^*$  values of 2.66 and 2.16 eV, respectively. A comparison between the experimental and theoretical results for  $\text{Co}_6\text{S}_8(\text{PET}_3)_4(\text{CO})_2^-$  confirms that this cluster is primarily the *cis*-isomer: the experimental values AEA = 1.7 eV, VDE = 1.95 eV, and  $\text{VDE}^*$  = 2.51 eV all agree very well with the *cis*-isomer calculations. We note that the small shoulder at 2.19 eV in the spectrum (Fig. 2) matches the calculated  $\text{VDE}^*$  of the *trans*-isomer, raising the possibility that a small amount of the *trans*-isomer is also present. For  $n = 3$ , both the *mer*- $\text{Co}_6\text{S}_8(\text{PET}_3)_3(\text{CO})_3^-$  and *fac*- $\text{Co}_6\text{S}_8(\text{PET}_3)_3(\text{CO})_3^-$  have similar computed AEA and VDE values that are consistent with the experimental data. Because the *mer*- and *fac*-clusters are close in energy in their neutral and anionic forms, it is very likely that they both exist in the ion beam, and that the  $\text{Co}_6\text{S}_8(\text{PET}_3)_3(\text{CO})_3^-$  photoelectron spectrum contains transitions from both isomers.

The replacement of  $\text{PET}_3$  ligands by CO concurrently lowers the energies of the HOMO and LUMO, with the result that the HOMO–LUMO gap is essentially unchanged across the cluster series. Fig. 4a illustrates this trend for the neutral species while their absolute energy values are given in Table S1.† The lowering of the HOMO and LUMO can be in part understood in terms of the vastly different ligand field effect of the cluster surface passivation:  $\text{PET}_3$  is a strong  $\sigma$ -donor that increases the electron density in the core while CO is a strong  $\pi$ -acceptor that removes electron density from the core, thus lowering the energy of the electronic spectrum.<sup>33</sup> This behavior can be seen in the Fig. S1 and S2† which show the one-electron energy levels of  $\text{Co}_6\text{S}_8(\text{PET}_3)_{6-x}(\text{CO})_x^-$  ( $x = 0-3$ ). The change in the AEA closely tracks the change in the LUMO as the electron is attached to the LUMO of the neutral. This is illustrated in Fig. 4b, which shows the changes in the energy of the LUMO ( $\Delta\text{LUMO}$ ), theoretical AEA ( $\Delta\text{AEA}^T$ ) and experimental AEA ( $\Delta\text{AEA}^E$ ) of

Table 1 Theoretical and experimental adiabatic and vertical detachment energies (1<sup>st</sup> and 2<sup>nd</sup> peak) of  $\text{Co}_6\text{S}_8(\text{PET}_3)_{6-x}(\text{CO})_x^-$  ( $x = 0-3$ ) clusters<sup>a</sup>

Anionic clusters	AEA <sup>T</sup> (eV)	VDE <sup>T</sup> (eV)		AEA <sup>E</sup> (eV)	VDE <sup>E</sup> (eV)	
		VDE	VDE*		VDE	VDE*
$\text{Co}_6\text{S}_8(\text{PET}_3)_6^-$	1.06	1.28	1.74	1.1	1.30	1.91
$\text{Co}_6\text{S}_8(\text{PET}_3)_5(\text{CO})^-$	1.40	1.80	2.08	1.3	1.51	2.05
<i>cis</i> - $\text{Co}_6\text{S}_8(\text{PET}_3)_4(\text{CO})_2^-$	1.65	2.06	2.66	1.7	1.95	2.51
<i>trans</i> - $\text{Co}_6\text{S}_8(\text{PET}_3)_4(\text{CO})_2^-$	1.38	1.62	2.16			
<i>mer</i> - $\text{Co}_6\text{S}_8(\text{PET}_3)_3(\text{CO})_3^-$	1.92	2.22	2.71	1.8	2.09	2.67
<i>fac</i> - $\text{Co}_6\text{S}_8(\text{PET}_3)_3(\text{CO})_3^-$	1.93	2.35	2.60			

<sup>a</sup> The superscripts T and E indicate theoretical and experimental values, respectively.



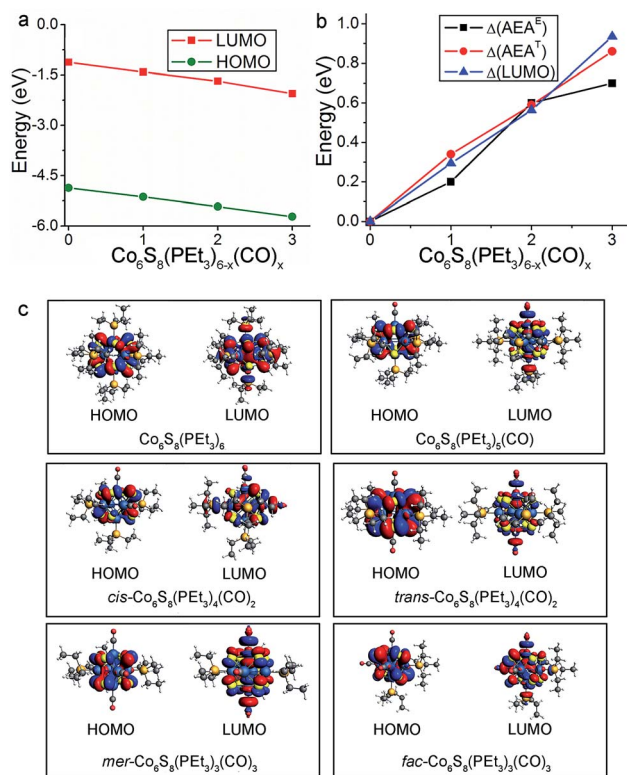


Fig. 4 (a) The absolute energy values of the HOMO and LUMO for neutral  $\text{Co}_6\text{S}_8(\text{PET}_3)_{6-x}(\text{CO})_x$  ( $x = 0-3$ ). (b) Incremental differences in the experimental AEA values, the theoretical AEA values, and the LUMO values of  $\text{Co}_6\text{S}_8(\text{PET}_3)_{6-x}(\text{CO})_x$ , each with respect to  $\text{Co}_6\text{S}_8(\text{PET}_3)_6$ . (c) Molecular orbital iso-surfaces of the HOMO and LUMO in  $\text{Co}_6\text{S}_8(\text{PET}_3)_{6-x}(\text{CO})_x$  ( $x = 0-3$ ) clusters.

$\text{Co}_6\text{S}_8(\text{PET}_3)_{6-x}(\text{CO})_x$  as  $x$  increases from 0 to 3. The molecular orbital iso-surfaces of the HOMO and LUMO in  $\text{Co}_6\text{S}_8(\text{PET}_3)_{6-x}(\text{CO})_x$  are shown in Fig. 4c. A Mulliken population analysis of the molecular orbitals reveals that hybridization between the p-states of S and the 3d-states of Co forms the HOMO in each cluster, while the LUMO is an antibonding state comprising the  $3d_z^2$ -orbital of Co and the lone pair of the ligand ( $\text{PET}_3/\text{CO}$ ). An additional phenomenon combines with the traditional ligand field effect to explain the changes in AEA and VDE:  $\text{PET}_3$  and CO form charge transfer complexes, and the induced dipoles at the cluster surface play a critical role in changing the electronic properties. As a reference, surface dipoles can either increase or decrease the work function of metals.<sup>47,48</sup> In this case, as  $\text{PET}_3$  ligands are replaced by CO, the sign of the dipole changes and hence AEA increases as more CO are bound to the core.

Lastly, though the ionization energy of  $\text{Co}_6\text{S}_8(\text{PET}_3)_{6-x}(\text{CO})_x$  is not directly measured, the successive rise in AEA and lowering of HOMO are evidence of a trend of increasing ionization energy as  $\text{PET}_3$  is replaced by CO.

## Conclusions

By combining cluster synthesis, gas-phase anion photoelectron spectroscopy, and electronic structure calculations, we have demonstrated that the electronic properties of metal

chalcogenide superatomic clusters can be tuned by varying their capping ligands. While the total electron count of the closed-shell core remains unchanged, substitution of the ligands leads to a simultaneous rise or fall of the HOMO and LUMO levels, resulting in a change of the cluster donor/acceptor behavior.

These results are significant because such drastic change in the cluster properties is rarely observed in other superatomic systems. Besides changes of electron affinity and ionization potential, ligand substitution of a ligated cluster almost always results in changes of other core chemical properties such as the HOMO–LUMO gap, electron configuration, and electronic structure. For example, in phosphine-passivated gold clusters approximated by the confined nearly free electron gas model, charge transfer from the phosphine ligands results in significant changes to the core electron count and HOMO–LUMO gap.<sup>49</sup> The unpredictable outcome of ligand substitution can impose important challenges in designing programmable cluster building blocks. In this context, the metal chalcogenide cluster presented in this work is unique: ligand substitution can be used to tune its electron affinity and ionization potential while maintaining its HOMO–LUMO gap, electronic structure and effective electron count, thus enabling us to control its oxidative and reductive properties. Such tunability may prove crucial for applications as catalysts, semiconductor dopants, and superatomic building blocks for cluster-assembled materials. In one example, we recently showed that related clusters can be used as electron-donating or electron-accepting surface dopants for two-dimensional semiconductors, where the band structures can be controlled by the choice of ligands.<sup>32,50</sup> Such dopants could also be made magnetic because of their finite spin moments.<sup>51</sup> More broadly, our results open the door to the design of tunable functional materials from superatomic building blocks.

## Conflicts of interest

The authors declare no competing financial interest.

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## Notes and references

- 1 A. Pinkard, A. M. Champsaur and X. Roy, *Acc. Chem. Res.*, 2018, **51**, 919–929.
- 2 S. A. Claridge, A. W. Castleman, S. N. Khanna, C. B. Murray, A. Sen and P. S. Weiss, *ACS Nano*, 2009, **3**, 244–255.
- 3 E. G. Tulsy and J. R. Long, *Chem. Mater.*, 2001, **13**, 1149–1166.
- 4 S. A. Baudron, P. Batail, C. Coulon, R. Clerac, E. Canadell, V. Laukhin, R. Melzi, P. Wzietek, D. Jerome, P. Auban-Senzier and S. Ravy, *J. Am. Chem. Soc.*, 2005, **127**, 11785–11797.
- 5 B. Yoon, W. D. Luedtke, R. N. Barnett, J. Gao, A. Desireddy, B. E. Conn, T. Bigioni and U. Landman, *Nat. Mater.*, 2014, **13**, 807–811.
- 6 J. F. Corrigan, O. Fuhr and D. Fenske, *Adv. Mater.*, 2009, **21**, 1867–1871.
- 7 X. Roy, C.-H. Lee, A. C. Crowther, C. L. Schenck, T. Besara, R. A. Lalancette, T. Siegrist, P. W. Stephens, L. E. Brus, P. Kim, M. L. Steigerwald and C. Nuckolls, *Science*, 2013, **341**, 157–160.
- 8 S. N. Khanna and P. Jena, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1995, **51**, 13705–13716.
- 9 A. W. Castleman and S. N. Khanna, Clusters, *J. Phys. Chem. C*, 2009, **113**, 2664–2675.
- 10 A. C. Reber and S. N. Khanna, *Acc. Chem. Res.*, 2017, **50**, 255–263.
- 11 C.-H. Lee, L. Liu, C. Bejger, A. Turkiewicz, T. Goko, C. J. Arguello, B. A. Frandsen, S. C. Cheung, T. Medina, T. J. S. Munsie, R. D'Ortenzio, G. M. Luke, T. Besara, R. A. Lalancette, T. Siegrist, P. W. Stephens, A. C. Crowther, L. E. Brus, Y. Matsuo, E. Nakamura, Y. J. Uemura, P. Kim, C. Nuckolls, M. L. Steigerwald and X. Roy, *J. Am. Chem. Soc.*, 2014, **136**, 16926–16931.
- 12 X. Roy, C. L. Schenck, S. Ahn, R. A. Lalancette, L. Venkataraman, C. Nuckolls and M. L. Steigerwald, *Angew. Chem., Int. Ed.*, 2012, **51**, 12473–12476.
- 13 M. T. Trinh, A. Pinkard, A. B. Pun, S. N. Sanders, E. Kumarasamy, M. Y. Sfeir, L. M. Campos, X. Roy and X.-Y. Zhu, *Sci. Adv.*, 2017, **3**, e1700241.
- 14 V. Chauhan, S. Sahoo and S. N. Khanna, *J. Am. Chem. Soc.*, 2016, **138**, 1916–1921.
- 15 S. N. Khanna and A. C. Reber, *Nat. Chem.*, 2017, **9**, 1151–1152.
- 16 D. A. Tomalia and S. N. Khanna, *Chem. Rev.*, 2016, **116**, 2705–2774.
- 17 Z. Zheng, J. R. Long and R. H. Holm, *J. Am. Chem. Soc.*, 1997, **119**, 2163–2171.
- 18 M. Cargnello, A. C. Johnston-Peck, B. T. Diroll, E. Wong, B. Datta, D. Damodhar, V. V. T. Doan-Nguyen, A. A. Herzing, C. R. Kagan and C. B. Murray, *Nature*, 2015, **524**, 450–453.
- 19 M. N. O'Brien, M. R. Jones, B. Lee and C. A. Mirkin, *Nat. Mater.*, 2015, **14**, 833–839.
- 20 M. A. Boles and D. V. Talapin, *J. Am. Chem. Soc.*, 2015, **137**, 4494.
- 21 C. L. Poyser, T. Czerniuk, A. Akimov, B. T. Diroll, E. A. Gaudling, A. S. Salasyuk, A. J. Kent, D. R. Yakovlev, M. Bayer and C. B. Murray, *ACS Nano*, 2016, **10**, 1163–1169.
- 22 W.-L. Ong, S. M. Rupich, D. V. Talapin, A. J. H. McGaughey and J. A. Malen, *Nat. Mater.*, 2013, **12**, 410–415.
- 23 A. Voevodin, L. M. Campos and X. Roy, *J. Am. Chem. Soc.*, 2018, **140**, 5607–5611.
- 24 A. M. Champsaur, J. Yu, X. Roy, D. W. Paley, M. L. Steigerwald, C. Nuckolls and C. M. Bejger, *ACS Cent. Sci.*, 2017, **3**, 1050–1055.
- 25 A. M. Champsaur, C. Mézière, M. Allain, D. W. Paley, M. L. Steigerwald, C. Nuckolls and P. Batail, *J. Am. Chem. Soc.*, 2017, **139**, 11718–11721.
- 26 E. S. O'Brien, M. T. Trinh, R. L. Kann, J. Chen, G. A. Elbaz, A. Masurkar, T. L. Atallah, M. V. Paley, N. Patel, D. W. Paley, I. Kymissis, A. C. Crowther, A. J. Millis, D. R. Reichman, X.-Y. Zhu and X. Roy, *Nat. Chem.*, 2017, **9**, 1170–1174.
- 27 G. Lovat, B. Choi, D. W. Paley, M. L. Steigerwald, L. Venkataraman and X. Roy, *Nat. Nanotechnol.*, 2017, **12**, 1050–1054.
- 28 W.-L. Ong, E. S. O'Brien, P. S. M. Dougherty, D. W. Paley, C. Fred Higgs Iii, A. J. H. McGaughey, J. A. Malen and X. Roy, *Nat. Mater.*, 2017, **16**, 83–88.
- 29 B. Choi, J. Yu, D. W. Paley, M. Tuan Trinh, M. V. Paley, J. M. Karch, A. C. Crowther, C. H. Lee, R. A. Lalancette, X. Zhu, K. Kim, M. L. Steigerwald, C. Nuckolls and X. Roy, *Nano Lett.*, 2016, **16**, 1445–1449.
- 30 A. M. Champsaur, A. Velian, D. W. Paley, B. Choi, X. Roy, M. L. Steigerwald and C. Nuckolls, *Nano Lett.*, 2016, **16**, 5273–5277.
- 31 A. Turkiewicz, D. W. Paley, T. Besara, G. Elbaz, A. Pinkard, T. Siegrist and X. Roy, *J. Am. Chem. Soc.*, 2014, **136**, 15873–15876.
- 32 J. Yu, C. H. Lee, D. Bouilly, M. Han, P. Kim, M. L. Steigerwald, X. Roy and C. Nuckolls, *Nano Lett.*, 2016, **16**, 3385–3389.
- 33 V. Chauhan, A. C. Reber and S. N. Khanna, *J. Am. Chem. Soc.*, 2017, **139**, 1871–1877.
- 34 V. Chauhan and S. N. Khanna, *J. Phys. Chem. A*, 2018, **122**(28), 6014–6020.
- 35 F. Cecconi, C. A. Ghilardi, S. Midollini and A. Orlandini, *Inorg. Chim. Acta*, 1983, **76**, 183–184.
- 36 X. Zhang, G. Liu, G. Gantfoer, K. H. Bowen and A. N. Alexandrova, *J. Phys. Chem. Lett.*, 2014, **5**, 1596–1601.
- 37 A. Grubisic, H. Wang, X. Li, Y. J. Ko, S. Kocak, M. R. Pederson, K. H. Bowen and B. W. Eichhorn, *PNAS*, 2011, **108**, 14757–14762.
- 38 J. P. Perdew and W. Yue, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1986, **33**, 8800–8802.
- 39 M. Ernzerhof and G. E. Scuseria, *J. Chem. Phys.*, 1999, **110**, 5029.
- 40 C. Adamo, *J. Chem. Phys.*, 1999, **110**, 6158.
- 41 E. van Lenthe and E. J. Baerends, *J. Comput. Chem.*, 2003, **24**, 1142–1156.
- 42 E. van Lenthe, R. van Leeuwen, E. J. Baerends and J. G. Snijders, *Int. J. Quantum Chem.*, 1996, **57**, 281–293.



- 43 E. van Lenthe, A. Ehlers and E. J. Baerends, *J. Chem. Phys.*, 1999, **110**, 8943.
- 44 M. R. Nimlos and G. B. Ellison, *J. Phys. Chem.*, 1986, **90**, 2574–2580.
- 45 E. W. Rothe, S. Y. Tang and G. P. Reck, *J. Chem. Phys.*, 1975, **62**, 3829.
- 46 P. J. Robinson, G. Liu, S. Ciborowski, C. Martinez-Martinez, J. R. Chamorro, X. Zhang, T. M. McQueen, K. H. Bowen and A. N. Alexandrova, *Chem. Mater.*, 2017, **29**, 9892–9896.
- 47 P. C. Rusu and G. Brocks, *J. Phys. Chem. B*, 2006, **110**, 22628–22634.
- 48 V. Chauhan, A. C. Reber and S. N. Khanna, *Nat. Commun.*, 2018, **9**, 2357.
- 49 D. Mollenhauer and N. Gaston, *Phys. Chem. Chem. Phys.*, 2016, **18**, 29686.
- 50 A. C. Reber and S. N. Khanna, *npj Comput. Mater.*, 2018, **4**, 33.
- 51 V. Chauhan, A. C. Reber and S. N. Khanna, *Phys. Chem. Chem. Phys.*, 2017, **19**, 31940–31948.

