

Fluorine-Rich Zinc Oxoclusters as Extreme Ultraviolet Photoresists: Chemical Reactions and Lithography Performance

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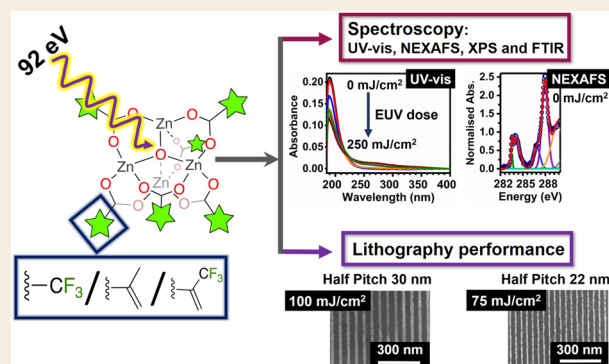
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ABSTRACT: The absorption of extreme ultraviolet (EUV) radiation by a photoresist strongly depends on its atomic composition. Consequently, elements with a high EUV absorption cross section can assist in meeting the demand for higher photon absorbance by the photoresist to improve the sensitivity and reduce the photon shot noise induced roughness. In this work, we enhanced the EUV absorption of the methacrylic acid ligands of Zn oxoclusters by introducing fluorine atoms. We evaluated the lithography performance of this fluorine-rich material as a negative tone EUV photoresist along with extensive spectroscopic and microscopic studies, providing deep insights into the underlying mechanism. UV–vis spectroscopy studies demonstrate that the presence of fluorine in the oxocluster enhances its stability in the thin films to the ambient atmosphere. However, the EUV photoresist sensitivity (D_{50}) of the fluorine-rich oxocluster is decreased compared to its previously studied methacrylic acid analogue. Scanning transmission X-ray microscopy and in situ X-ray photoelectron spectroscopy in combination with FTIR and UV–vis spectroscopy were used to gain insights into the chemical changes in the material responsible for the solubility switch. The results support decarboxylation of the ligands and subsequent radical-induced polymerization reactions in the thin film upon EUV irradiation. The rupture of carbon–fluorine bonds via dissociative electron attachment offers a parallel way of generating radicals. The mechanistic insights obtained here will be applicable to other hybrid materials and potentially pave the way for the development of EUV materials with better performance.

KEYWORDS: EUV photoresists, EUV lithography, STXM, NEXAFS, in situ XPS, infrared spectroscopy, UV spectroscopy



1. INTRODUCTION

The dimensions of the smallest components that can be fabricated using photolithography are directly proportional to the wavelength of the radiation used in the process. Over decades, the wavelength of the imaging light used in the semiconductor industry for photolithography has decreased from G-line (436 nm) to I-line (365 nm) to KrF (248 nm) to ArF (193 nm) immersion lithography (ArFi).^{1,2} Presently, extreme ultraviolet (EUV) at 13.5 nm has emerged as the successor of ArFi as next-generation lithography to reach the industry's objective of patterning feature sizes of 10 nm and below in a cost-effective manner.^{3,4}

The steep jump in the wavelength from 193 to 13.5 nm has posed numerous challenges, and a key factor that will determine the future improvements of EUV lithography (EUVL) is the choice of the photoresist and its solubility-switching chemistry.^{1,5,6} For UV lithography, organic materials are used as photoresists and their reactions are based on photochemical acid generation via electronically excited states. The photon energy of EUVL (13.5 nm, ~92 eV), however, is ~14× higher than that of its predecessor, ArF (193 nm, 6.4

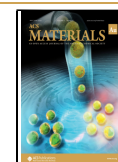
eV), and the EUV absorption mechanism and resulting chemical reactions are different. EUV photons are highly energetic and surpass the ionization potential of the material. Thus, EUV absorption by the material results in the ejection of photoelectron(s). These electrons have sufficiently high energy to further interact with the neighboring molecules in the thin film, giving rise to secondary electron processes.^{7–10} Thus, EUV photon absorption by the resist initiates a cascade of reactions in the thin films,^{6,8,11} and because of this complexity, understanding the fundamentals is challenging. Moreover, the deconvolution of these events is not straightforward, especially in the thin film samples. Henceforth, a fundamental understanding of the induced chemical changes or reaction path

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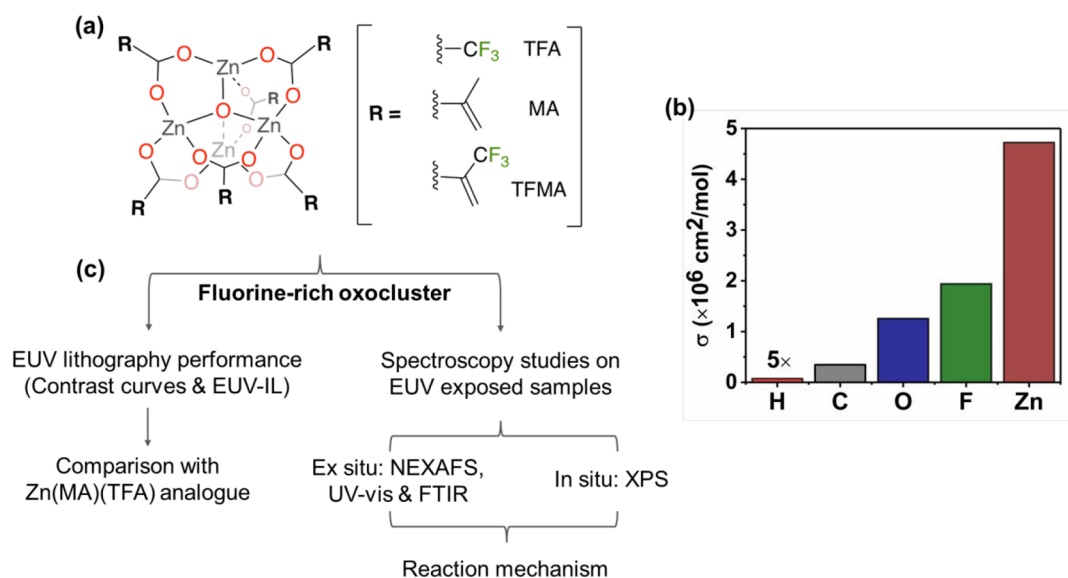


Figure 1. (a) Schematic representation of Zn oxoclusters: fluorine-rich oxocluster Zn(TFMA) (average composition Zn(TFMA)_{5.9}(TFA)_{0.1}) and Zn(TFMA)(MA)(TFA) (average composition Zn(TFMA)_{2.4}(MA)_{2.9}(TFA)_{0.8}) and the previously studied analogue, Zn(MA)₅(TFA)₁. (b) EUV atomic absorption cross section at 92 eV of elements H(×5), C, O, F, and Zn obtained from the CXRO database.²⁴ (c) Schematic representation of the study flow for fluorine-rich Zn oxocluster for EUVL applications.

mechanisms upon irradiating the photoresist material with ionizing EUV radiation is crucial.¹²

The primary challenge posed by the conventional organic resists (mainly constituting C, H, and O) is their low EUV absorption, and this drew the attention of researchers toward incorporating elements (especially metals) having a high photon absorption cross section at 92 eV to potentially increase sensitivity and reduce stochastic effects.^{1,11,13–17} Our previous studies on a hybrid zinc-based oxocluster, Zn(MA)₅(TFA)₁ (MA is methacrylate ligand; TFA is trifluoroacetate ligand, Figure 1a), demonstrated an EUV linear absorption coefficient of 12 μm^{-1} , much higher than that of traditional organic photoresists ($\sim 5 \mu\text{m}^{-1}$).¹⁸ Further, mechanistic insights for the solubility switch in Zn(MA)(TFA) photoresist upon EUV exposure using spectroscopic studies demonstrated polymerization of the terminal double bonds of the MA ligands and formation of fluoride species in the thin films from TFA via dissociative electron attachment (DEA).¹⁹ The same reaction paths were found when the material was irradiated with low energy electrons of 20 and 80 eV.²⁰

In this work, we study fluorine-rich Zn-based oxoclusters, designed using the 2-(trifluoromethyl)acrylate (TFMA) ligand instead of MA. The terminal double bond of the ligand was kept intact to allow polymerization upon EUV irradiation, yet at the same time more fluorine was introduced. The molar absorption cross section of fluorine at 92 eV is $\sim 26\times$ larger than that of H and $\sim 5.6\times$ that of C (Figure 1b). Hence, its incorporation, along with the metals, in the resist material has the potential to increase the EUV absorbance. Moreover, some studies have demonstrated an overall enhanced EUV lithography performance upon the incorporation of fluorine in the photoresist material.^{9,21–23}

This work focuses on the detailed study of fluorine-rich Zn oxocluster for EUVL application (Figure 1c) mainly in two ways:

- (i) investigating the changes in processability, stability of oxoclusters in thin films, and EUV lithography perform-

ance upon fluorine incorporation in comparison with its previously studied Zn(MA)(TFA) analogue;¹⁹ and

- (ii) monitoring chemical changes on the fluorine-rich zinc oxocluster thin films upon EUV irradiation using near-edge X-ray absorption fine structure (NEXAFS), FTIR and UV–vis spectroscopy, and X-ray photoelectron spectroscopy (XPS) studies.

For comparison, we include a mixed-ligand compound Zn(TFMA)(MA)(TFA) with average composition Zn(TFMA)_{2.4}(MA)_{2.9}(TFA)_{0.8} in the study. The present studies reveal that the incorporation of more fluorine in methacrylate-Zn oxoclusters is not a magic bullet. Replacing MA with TFMA did aid in increasing the thin film stability over time, but the formation of smooth thin films on substrates proved to be difficult. Although the EUV absorption of the fluorine-rich oxocluster Zn(TFMA) (Figure 1a) must be theoretically higher than that of its previously studied analogue Zn(MA)(TFA), the sensitivity (D_{50}) for EUV lithography turns out to be unexpectedly lower. Spectroscopy studies show loss of CO₂ from carboxylate ligands, and a strong decrease of C=C/C=O ratio as a function of EUV dose supports radical initiated polymerization in the thin film. Furthermore, fluorine migration from the CF₃ groups toward Zn was observed, forming new Zn–F species. This likely occurs via a dissociative electron attachment (DEA) path (low-energy electrons). We believe that the present results enhance our understanding of Zn based fluorine-rich photoresists and similar systems for lithography application and will propel advance focused investigations for fluorine-rich photoresist materials.

2. EXPERIMENTAL DETAILS

2.1. Synthesis of Fluorine-Rich Zn Oxoclusters: Zn(TFMA) and Zn(TFMA)(MA)(TFA)

In this study, fluorine-rich oxoclusters were synthesized from a commercially available precursor having a Zn tetranuclear oxo-core and an organic shell of 6 trifluoroacetate (TFA) ligands, following the same exchange procedure as described in our previous work.¹⁸ The effect of fluorine incorporation in the reactive organic ligands of the

Zn-oxocluster was studied by investigating Zn(TFMA) (still containing a minor fraction of TFA, average composition Zn-(TFMA)_{5.9}(TFA)_{0.1}) and a similar cluster with a mixed ligand shell, Zn(TFMA)(MA)(TFA) (average composition Zn-(TFMA)_{2.4}(MA)_{2.9}(TFA)_{0.8}); see Figure 1a. The EUVL performance of the materials was compared to the previously studied analogue, Zn(MA)(TFA).^{18,19}

2.1.1. Synthesis of Zn(TFMA). Oxo[hexa(trifluoroacetato)] tetrazinc trifluoroacetic acid adduct (Zn(TFA); 500 mg, 0.52 mmol, 1 equiv) and 2-(trifluoromethyl)acrylic acid (TFMAA; 870 mg, 6.2 mmol, 12 equiv) were dissolved in acetonitrile, and the solution was stirred for 5 h at 45 °C. The remaining solvent was then evaporated using a rotary evaporator, and an oily residue was obtained. The residue was reprecipitated by adding toluene to it and evaporating the solvent on a rotary evaporator. The process was repeated four to five times to remove excess of acid and obtain a white solid product. This was dried in a vacuum oven at 40 °C and stored under nitrogen. The reaction scheme is shown in SI, Figure S1.

¹H NMR (300 MHz, DMSO-*d*₆) δ: 6.52 (1H, =CH₂) and 6.30 (1H, =CH₂), Figure S2.

¹⁹F NMR (300 MHz, DMSO-*d*₆) δ: -73.69 (-CF₃, TFA) and -63.92 (-CF₃, TFMA); integration of 46.3 and 1.0 of peaks, respectively, showing that most of the TFA ligand in the precursor has been exchanged by TFMA and on average 5.9 TFMA ligand and 0.1 TFA ligand is present in the organic shell of Zn(TFMA) oxocluster, Figure S2.

FTIR: 695 (COO angle bending), 828 (s, CH₂ bending mode), 993 and 912 (bending out of plane, γ (CH)C=CH₂), 1130 and 1167 (ν C-F), 1380–1487 (CH₂ + ν_s COO), 1583 (ν_{as} COO (TFMA)), 1631 (ν C=C (TFMA) monodentate), 1660 (ν C=C) and 1678 cm⁻¹ (ν_{as} COO (TFMA) monodentate); spectra shown in SI (Figure S3).

Mass spectrometry: the observed isotopic distribution corresponds to the presence of the tetranuclear Zn core of the oxocluster (Figure S4). The Zn oxocluster is susceptible to form complexes with acetonitrile (electrospray solvent), also observed in our previous studies.^{18,25} The isotopic distribution observed in the mass spectra was assigned to the oxocluster complex: Zn₄O with 5 TFMA ligands and 2 CH₃CN molecules, which is in concordance with simulated spectra using ChemCalc.²⁶

The deposition of the thin films on substrates of Zn(TFMA) was done by spin coating 2 w/v % of resist solution from CHCl₃:PGMEA (7:3, v/v). All thin films were spin-coated at an acceleration of 3000 rpm/s and 2100 rpm speed for 30 s immediately followed by postapplication bake (PAB) at 90 °C/30 s. The thickness of the thin films obtained using the above parameters on Si substrates is 14–20 nm, determined by atomic force microscopy (AFM), using a Bruker ScanAsyst-air probe.

2.1.2. Synthesis of Zn(TFMA)(MA)(TFA). Zn(TFA) (500 mg, 0.52 mmol, 1 equiv), 2-(trifluoromethyl)acrylic acid (TFMAA; 270 mg, 3.1 mmol, 6 equiv), and methacrylic acid (MAA; 440 mg, 3.1 mmol, 6 equiv) were used as precursors for the reaction, and the same synthesis procedure as described above for the Zn(TFMA) synthesis was used with acetonitrile as the solvent.

¹H NMR (300 MHz, DMSO-*d*₆) δ: 6.52 (1H, =CH₂, TFMA), 6.30 (1H, =CH₂, TFMA), 5.85 (1H, =CH₂, MA), 5.36 (1H, =CH₂, MA) and 1.85 (-CH₃, MA); integration of 2.4 (2H, =CH₂, TFMA), 2.0 (2H, =CH₂, MA) and 3.1 (-CH₃, MA) was obtained for peaks, Figure S5.

¹⁹F NMR (300 MHz, DMSO-*d*₆) δ: -73.58 (-CF₃, TFA) and -63.85 (-CF₃, TFMA); integration of 0.30 and 1.0 was obtained for peaks, respectively, Figure S5.

From the ¹H and ¹⁹F NMR analysis of Zn(TFMA)(MA)(TFA) oxocluster, the composition of the organic shell was determined as an average of 2.4 TFMA, 2.9 MA, and 0.8 TFA. The deposition of the thin films of the Zn(TFMA)(MA)(TFA) oxocluster was done by spin coating CHCl₃:PGMEA (7:3, v/v) as the casting solvent for 2 w/v % of resist solution. All thin films were spin-coated at an acceleration of 3000 rpm/s and 2100 rpm speed for 30 s followed by immediate postapplication bake (PAB) at 90 °C/30 s.

2.2. Silanization Process

Clean Si substrates were treated with an ozone photoreactor to maximize the reactive terminal hydroxyl groups on the Si substrate surface, immediately followed by a silanization process. Silanes used in this study were: 3-(chloropropyl)triethoxysilane, 3-(mercaptopropyl)-trimethoxysilane, 3-(triethoxysilyl)propyl methacrylate, 3-(aminopropyl)triethoxysilane, 1H,1H,2H,2H-perfluorooctyltriethoxysilane, and bis(trimethylsilyl)amine (HMDS). Silanization was carried out in the liquid phase except for HMDS which was deposited in the vapor phase. For liquid phase silanization, a reaction mixture was prepared in 80 mL of 96% ethanol to which a catalytic amount of 99% acetic acid was added to adjust the pH of the solution at ~6. To this solution, 2 mL of silane was added, and the clean treated Si substrates were placed in the reaction solution for ~40 min at room temperature with constant stirring, washed with ethanol (96%), and dried with nitrogen. The silanized substrates were annealed in an oven at ambient pressure, 130 °C/24 h.

2.3. Sample Preparation

Two types of substrates were used for near edge X-ray absorption fine structure (NEXAFS) measurements on Zn(TFMA) samples: (1) for reference (unexposed) spectra, a 5 × 5 mm² substrate with an array of 5 × 5 free-standing SiN_x windows (0.15 × 0.15 mm²) was used; and (2) for EUV exposure 7.5 × 7.5 mm² substrates with a SiN_x membrane window (30 nm thick) of 3 × 3 mm² was used. Zn(TFMA) was spin coated on the membrane and exposed to EUV photons at different doses (exposed area of 0.5 × 0.5 mm² per dose) on a single SiN_x membrane window (3 × 3 mm²). The measurements were performed in transmission mode.

For FTIR spectroscopy, thin films were spin coated on double side polished Si substrates (200 μm thick) while for UV-vis analysis the thin films were spin coated on quartz substrates (500 μm thick). For the XPS study, oxocluster was spin coated on the Si substrate sputter coated with Cr/Au (5 nm/40 nm). Thin films for the contrast curves and EUV-IL experiments were spin coated on Si substrates (with native oxide layer) without any prior treatment. The thickness of the thin films spin coated using parameters mentioned above (section 2.1) on Si/SiO₂ (for EUV-IL, FTIR and XPS) and quartz substrates (UV-vis) are in the range of 14–20 nm. Remaining thin film thicknesses after development on the SiN_x substrate for STXM studies are shown in the Supporting Information (SI), Figure S16.

2.4. EUV Exposure

EUV (13.5 nm) exposures were performed using synchrotron radiation at the Paul Scherrer Institute (PSI), Switzerland at the Swiss Light Source XIL-II beamline for contrast curves, L/S features (EUV-interference lithography, EUV-IL),^{27,28} and ex situ spectroscopy measurements. Open frame exposures for contrast curves and NEXAFS measurements were performed over a range of doses, using a 0.5 × 0.5 mm² aperture mask. For FTIR and UV-vis spectroscopy studies exposures were done using a 1.7 × 1.7 mm² aperture mask. EUV-IL was used to test nanopatterning performance using a mask with periodic line/space (L/S) half-pitch (HP) of 50 nm, 40 nm, 30 nm and 22 nm.

2.5. Postexposure Analysis and Ex Situ Spectroscopy Study

A scanning transmission X-ray microscope (STXM) (PoLux beamline²⁹ at PSI) was used to perform NEXAFS spectroscopy measurements as illustrated in Figure S6. The measurements were done under normal incidence using a line scan over a length of 25 μm. The beam was defocused to a spot size of ~1 μm to avoid radiation damage. For each energy, a sample scan was made on the sample, exposed to EUV, and developed using diluted 0.1% propionic acid in 2-heptanone for 10 s and a reference scan was made on a nearby area on the SiN_x membrane that was cleaned (unexposed photoresist, AFM images shown in Figure S17) by the development. Carbon K-edge spectra were recorded for an energy range of 270–350 eV with an energy resolution of 0.35 eV. For F K-edge measurements, the energy was scanned from 675 to 710 eV. By repeating scans at the

same sample area, we could show that radiation damage is negligible under these conditions. ATHENA software was used for spectra normalization and fitting of the C K-edge and F K-edge.

UV–vis spectroscopy was performed on thin films deposited on quartz using a Shimadzu UV2600 spectrophotometer, and FTIR was performed in transmission mode under vacuum using a Bruker Vertex 80v spectrometer. The samples for UV–vis and FTIR spectroscopy were not developed after exposure.

A Bruker Dimension Icon atomic force microscope was used for thickness measurements to obtain contrast curves and inspect the thin film quality or line/space (L/S) pattern printed on Si substrates in Bruker ScanAsyst-air mode.

Top-down scanning electron microscopy (SEM) images of L/S patterns were recorded using a FEI Verios 460 scanning electron microscope operating at a current of 100 pA and a voltage of 5.0 kV.

2.6. In Situ XPS Study

EUV exposure and XPS measurements were performed at the BEAR beamline of the ELETTRA synchrotron radiation facility, at a base pressure of 10^{-9} mbar.^{30,31} A sample was exposed to 92 eV for a range of doses at different spots ($\sim 70 \times 300 \mu\text{m}^2$), and subsequently, XPS was measured at the exposed spots. The light at 13.5 nm was linearly polarized light, with the monochromator (vertical slit $50 \mu\text{m}$) set for maximum throughput and using a silicon filter for higher order rejection. High-resolution XPS was acquired with the full beam and no filter. The incident dose at 92 eV was determined in a separate scan by measuring the number of photons reaching the sample with the same beamline parameters used for the exposure using an absolute photodiode (AUXUV100 by OptoDiode corp.). The binding energy (BE) of the core levels were referenced to the C–F peak at 292.81 eV for C(1s) and 688.21 eV for F(1s) as an internal reference to calibrate the BE scale of each spectrum.¹⁹ A bias of -50 eV was applied to the sample. A pass energy of 10 eV for the electron analyzer and a step size of 0.2 eV were used. XPS was acquired for C (1s), O(1s), and F(1s) with photon energies of 400, 650, and 800 eV, respectively, and fresh spots on the thin films were used for each exposure followed by XPS measurements. The XPS spectra were fitted using UNIFIT 2008 software.

3. RESULTS AND DISCUSSION

3.1. Processability and Stability Studies

Compared to the previously studied Zn(MA)(TFA), the solubility of the new fluorine-rich oxocluster Zn(TFMA) was increased in most organic solvents, especially in CHCl_3 which was used for casting thin films. In the case of Zn(MA)(TFA), a saturated solution (2w/v% in CHCl_3 :PGMEA, 9:1 v/v) had to be used, which limited the control over the film thickness of the resist. The increased solubility of fluorine-rich Zn(TFMA) allows for varying the thickness of the thin resist films.

In addition, the stability of fluorine-rich Zn(TFMA) thin films to ambient conditions was also enhanced. Since some studies presented in our work are conducted ex situ, the material stability over time plays a vital role for analysis. To track the chemical changes in the terminal double bond (π to π^* transition, ~ 190 nm) of the TFMA ligand under ambient conditions, UV–vis spectroscopy was used. The absorption band of the double bond hardly changes for up to 5 h (shown in Figure S8) at ambient conditions. Some loss of absorption was observed after 7 h, possibly due to polymerization of the double bonds. Moreover, the absorption band shape was not severely affected as was observed in the case of its analogue Zn(MA)(TFA).¹⁸ This increase in the stability of Zn(TFMA) thin film over its analogue Zn(MA)(TFA) can be attributed to the more hydrophobic nature of the material due to the presence of fluorine and the high electronegativity of the

fluorine decreasing the effective electron density on the double bond, thus slowing the homopolymerization reaction.

3.2. Thin Films and EUV Contrast Curves

Thin films of the trifluoromethacrylate-containing Zn oxoclusters could be used for EUV exposure experiments, but they showed irregularities (Figure S11) that are probably related to poor adhesion to the substrate. Efforts were made to prevent this undesirable behavior by varying the surface energy of the substrate prior to spin coating by functionalization using silanes: 3-(mercaptopropyl)trimethoxysilane, 3-(triethoxysilyl)propyl methacrylate, 3-(aminopropyl)triethoxysilane, and 3-(chloropropyl)triethoxysilane. Water contact angles (Θ_c) of the substrates coated with these silanes were 26, 44, 46, and 65° , respectively. Unfortunately, none of these attempts led to the formation of high-quality thin films (Figure S9). In addition to the silanes mentioned above, Si surfaces were functionalized/modified using 1H,1H,2H,2H-perfluorooctyltriethoxysilane ($\Theta_c = 80^\circ$), HMDS, and hydrogen fluoride (HF, 2%; dip for 1 min), but no continuous thin film was obtained on the modified substrates.

The incorporation of fluorine in the resist material was anticipated to increase the EUV absorption of the material, which is expected to improve the EUVL performance of the photoresist. The EUV absorption coefficient of the Zn oxoclusters was estimated from the elemental absorption cross section at 92 eV using the CXRO database and the relative amount of each element in the oxocluster, and the estimated density.^{32,33} The density of the Zn(TFMA) and Zn(MA)(TFA) oxoclusters was estimated using the molecular weight of the respective oxocluster and its CPK molecular volume calculated using Spartan 18. A scaling factor of 2 was used, based on the comparison of calculated and experimental densities for the acetate and pivalate clusters.^{34,35} The estimated EUV absorption coefficient obtained through this for Zn(TFMA) and its analogue Zn(MA)(TFA) are 14.3 and $10.6 \mu\text{m}^{-1}$, respectively. The latter agrees well with the experimentally determined value of $12.4 \mu\text{m}^{-1}$.¹⁸

The sensitivities of the fluorine-rich oxoclusters for EUV photons were quantified experimentally using the contrast curves (exemplified in Figure 2) and were compared to their analogue Zn(MA)(TFA) which exhibited high sensitivity in our previous studies. The performance of the Zn(TFMA)-(MA)(TFA) oxocluster was also studied using contrast curves and is shown in SI Figure S10. Table 1 summarizes the results from contrast curves of three Zn-oxoclusters with increasing fluorine content: Zn(MA)(TFA),¹⁸ Zn(TFMA)(MA)(TFA) (Figure S10) and Zn(TFMA) (Figure 2). For direct comparison here the contrast curves were only compared for the samples developed by using diluted propionic acid in CHCl_3 . Unexpectedly, the studies using contrast curves revealed that the overall EUVL performance of the new fluorine-rich Zn oxoclusters was not enhanced. It was observed that the sensitivity (dose for 50% thickness loss, D_{50}) and the contrast (γ)³⁶ of the fluorine-rich oxocluster, Zn(TFMA), $D_{50} = 40 \text{ mJ}/\text{cm}^2$ and $\gamma = 1.71$, was decreased as compared to its analogue Zn(MA)(TFA), $D_{50} = 3.3\text{--}10.8 \text{ mJ}/\text{cm}^2$ and $\gamma = 2.95\text{--}0.95$.

This decrease in the sensitivity could arise from various factors. The electron density on the terminal double bond of TFMA ligand is relatively low because of the neighboring electron-withdrawing CF_3 group.³⁷ Previous studies have shown that introduction of a $-\text{CF}_3$ group at the α position

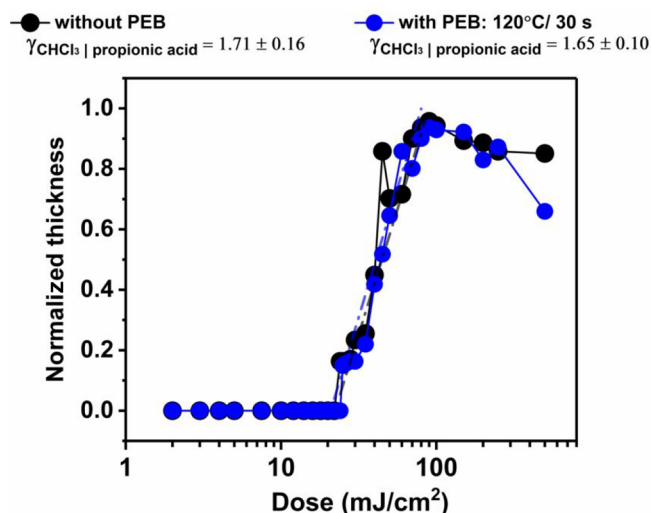


Figure 2. Contrast curves for Zn(TFMA) thin films using CHCl_3 :PGMEA as the casting solvent and developed using propionic acid (0.05%) in CHCl_3 for 8 s. The thicknesses are normalized relative to the thickness of the unexposed thin film (~ 14 nm). The contrast values γ are the slopes of the curves between D_0 and $\sim D_{90}$.

Table 1. Comparison of Sensitivity (D_{50}) and contrast (γ) of the Three Zn-Based Oxoclusters for EUV Lithography Performance without and with Postexposure Bake (PEB) and Developed Using Diluted Propionic Acid in CHCl_3 ^a

material	without PEB	
	D_{50} (mJ/cm ²)	contrast (γ)
Zn(MA)(TFA) ¹⁸	3.3–10.8	2.95–0.95
Zn(TFMA)(MA)(TFA)	40.6	3.10
Zn(TFMA)	40.0	1.71
material	with PEB	
	D_{50} (mJ/cm ²)	contrast (γ)
Zn(MA)(TFA) ¹⁸	100 °C/30 s	
	6.4	0.92
Zn(TFMA)(MA)(TFA)	120 °C/30 s	
	50.5	3.01
Zn(TFMA)	120 °C/30 s	
	45.0	1.65

^aThe concentration of the propionic acid used is mentioned in their respective contrast curve figures.

decreases the rate of radical homopolymerization of the TFMA unit.^{38,39} Different molecular packing of the oxoclusters in Zn(TFMA) compared to its analogue Zn(MA)(TFA) or a reduced ratio of the solubility rates of the unexposed and exposed regions in the developer of choice can also contribute toward this observed decrease in the sensitivity. Therefore, incorporation of more fluorine does not always directly translate into better EUVL performance of the photoresist, in contrast to the previous reported studies.^{23,40} Further, contrast curve studies demonstrated that postexposure bake (PEB) applied at 120 °C/30 s under ambient conditions led to a further slight decrease in the sensitivity (also see section 3.4). Spectroscopic studies to identify the chemical changes induced by PEB in the oxocluster were also carried out, and are discussed in section 3.7.

3.3. EUV-IL Performance of Fluorine-Rich Zn Oxoclusters

EUV-IL experiments were performed on Zn(TFMA) and Zn(TFMA)(MA)(TFA) oxoclusters to print L/S patterns of HP 50 nm, 40 nm, 30 and 22 nm. Selected SEM images of L/S patterns printed in Zn(TFMA) using EUV-IL and developed using diluted propionic acid in CHCl_3 and with 2-heptanone are shown in Figures 3 and S15, respectively. Irregularities in the films can be seen along with scumming (residual resist between lines) and bridging between the printed lines. The decrease in the sensitivity demonstrated via contrast curve measurements is also observed here. The L/S patterns printed with the fluorine-rich oxocluster are only observed at a relatively higher dose than required for its analogue Zn(MA)(TFA).¹⁹

The Zn(TFMA)(MA)(TFA) oxocluster was studied with EUV-IL in the same way as Zn(TFMA). Figures S12–S14 show selected SEM and AFM images of L/S features of HP 50, 40, 30, and 22 nm developed using both 2-heptanone and diluted propionic acid in CHCl_3 . The SEM and AFM images show considerable scumming after development also in this case.

3.4. Effect of PEB on EUV-IL Performance

Application of postexposure bake (PEB) did not lead to improvement in the EUV-IL performance for both Zn(TFMA)(MA)(TFA) and Zn(TFMA) oxocluster (120 °C/30 s). On the contrary, the application of PEB enhanced crystallization, bridging and scumming in L/S patterns. Figure S16 compares the performance of Zn(TFMA) oxocluster with and without PEB at the same dose of 75 mJ/cm².

Although the thin films show irregularities upon spin coating, the chemical changes induced by exposure to EUV irradiation can be studied using different spectroscopic techniques. Ex situ STXM-NEXAFS, FTIR, and UV–vis spectroscopy and in situ XPS studies were used to gain insights into the chemical changes in Zn oxoclusters as a function of the EUV dose. To monitor the chemical changes induced by EUV irradiation, the spectroscopic studies were focused on fluorine-rich Zn(TFMA) oxocluster.

3.5. STXM Measurements

To gain insights into the chemical changes upon EUV exposure in fluorine-rich Zn(TFMA) oxocluster, ex situ X-ray absorption spectroscopy (XAS) studies were performed using the STXM at the PoLLux beamline at PSI.^{29,41,42} The changes in the material as a function of EUV dose were observed especially in the near-edge region of the C K-edge spectra as it provides quantitative information on the types and amount of covalent bonding present in organic components. The spectra were measured in transmission mode, and thus, the whole thin film thickness was probed. Chemical changes in the material upon applied doses of 25, 50, and 250 mJ/cm² were monitored. The remaining thickness after development and quality of the exposed area on the SiN_x membrane after development was inspected using AFM, shown in SI Figure S17.

All the spectra were normalized to the continuum above ~ 310 eV. The interpretation of the spectra was based on the “building-block model”, and hence, the molecular oxocluster was considered as being composed of the individual local units.^{43–45} In the fitting procedure, the step function representing the ionization to the continuum^{46,47} was represented by three steps corresponding to the C–C/C–H (292.0 eV), –COO (294.9 eV), and –C–F (298.3 eV) units

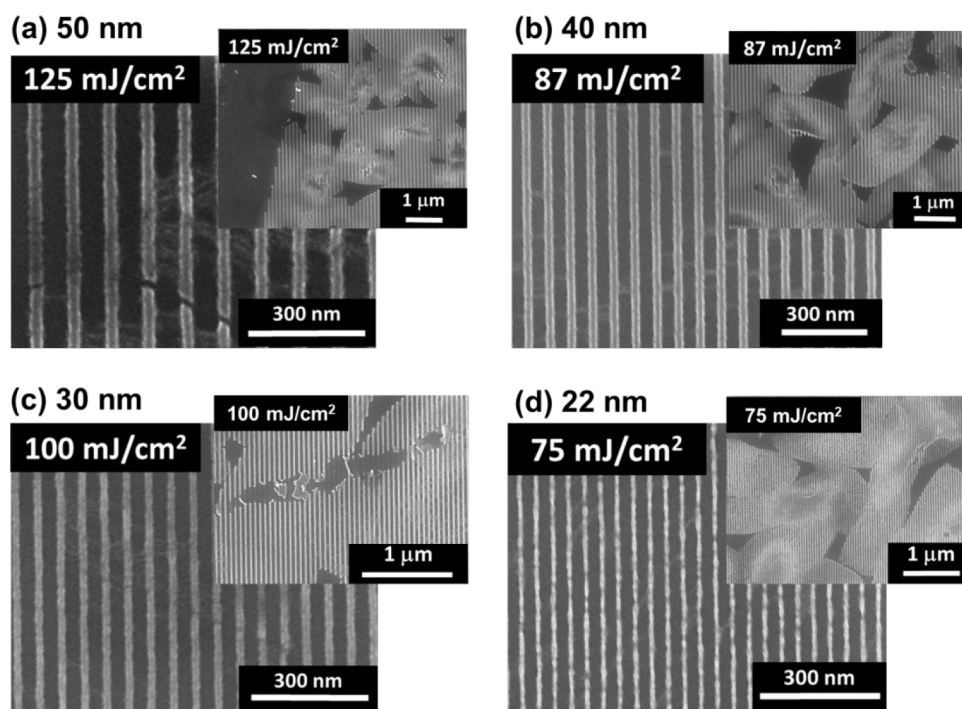


Figure 3. SEM images of L/S patterns of half-pitch: (a) 50 nm, (b) 40 nm, (c) 30 nm, and (d) 22 nm, printed using EUV-IL exposures on Zn(TFMA) oxocluster and developed using diluted propionic acid (0.1%) in 2-heptanone (10 s). The inset images show the irregularities of the thin film (zoom out).

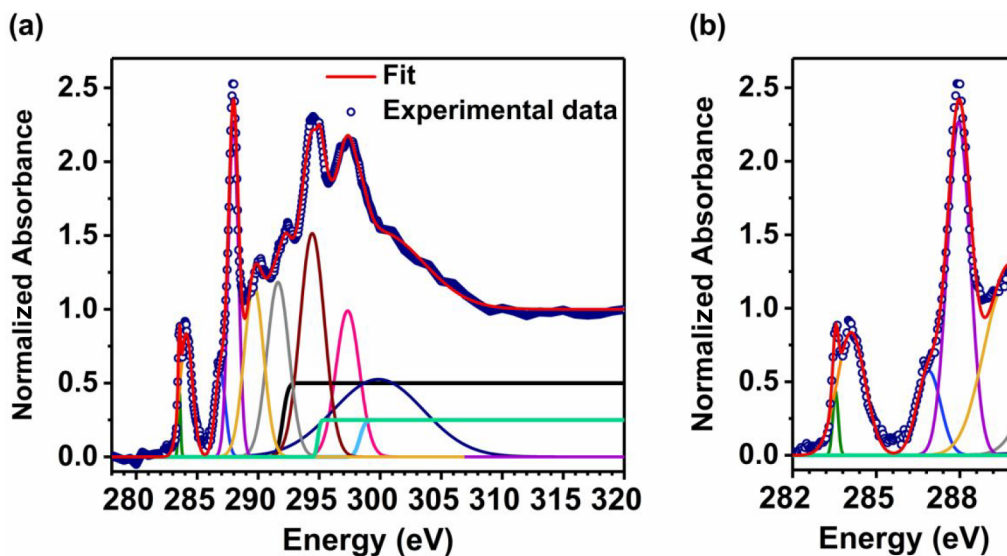


Figure 4. STXM-XAS C K-edge spectra of an unexposed Zn(TFMA) thin film: (a) complete spectrum and (b) zoom-in on the low-energy features.

of the Zn-oxocluster, which have characteristically different binding energies, as discussed below.

The fitted spectra of the reference (unexposed) sample of Zn(TFMA), shown in Figure 4, presents clear, sharp peaks at 284.2 and 284.7 eV. Quantum chemical calculations (see the SI for details) indicate that these correspond to the electronic transitions from C(1s) orbitals on C1 and C2 to the lowest unoccupied MO, which is a π^* orbital delocalized over the C=C–C=O unit ($C(1s) \rightarrow \pi^*_{C=C-C=O}$) (Figure S20). The strong absorption at 287.9 eV and the shoulder at 286.8 eV are mainly due to transitions from C(1s) of C1, C2, and C3 to the next higher π^* MO (Figure S20). The peaks at higher energies

are not assigned here; they represent a complex mix of transitions involving σ^* and Rydberg orbitals.^{48–51}

Upon conversion due to EUV irradiation, the peaks at 284.2 and 284.7 eV decrease in intensity relative to the main peak at 286.8/287.9 eV. This is consistent with the disappearance of the C=C bonds due to cross-linking reactions. In the reaction product, the C=O bond remains and the LUMO becomes the localized $\pi^*_{C=O}$, which has nearly the same energy as the LUMO+1 orbital in the intact TFMA ligand. The ratio $A_{C=C}/A_{C=O}$ (where A is the fitted area under the peaks) decreased as a function of EUV dose as shown in Figure 5. Radical polymerization of C=C is one of the major channels leading

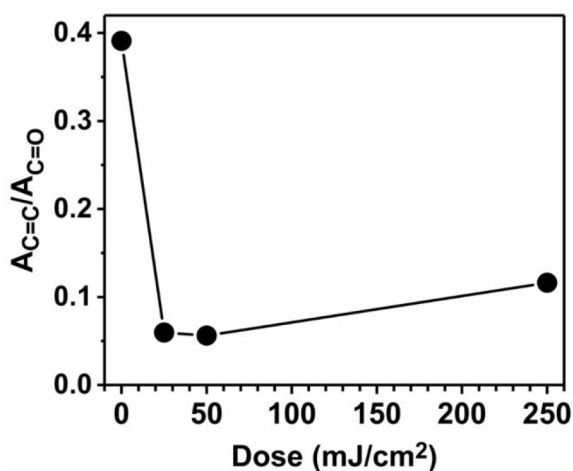


Figure 5. $A_{C=C}/A_{C=O}$ vs EUV dose for the area under the peaks, C=C units (284.2 and 284.7 eV), C=O units (286.8 and 287.9 eV) in C(1s) K-edge NEXAFS.

to a significant decrease in the component as a function of dose after decarboxylation reaction. The C K-edge fitted spectra measured for the exposed area on Zn(TFMA) thin film are shown in Figure S18. No significant changes were observed in the shape of the recorded F K-edge spectra as a function of EUV dose (Figure S19).

3.6. UV–Vis Spectroscopy

The chemical changes in the material upon EUV irradiation, especially in the C=C double bond of the ligand, can be effectively tracked using UV–vis spectroscopy. A decrease in the absorption band at ~ 190 nm (π to π^* transition)¹⁹ was observed as a function of EUV dose, shown in Figure 6a. This bleach can be mainly attributed to the polymerization, in agreement with the STXM results (Scheme 1b). In addition, a simultaneous increase in absorbance around ~ 275 nm was observed which can be due to the scattering of light or absorption by the network of oxoclusters/aggregates formed after EUV exposure (Scheme 1a).⁵² Figure 6b represents the decrease in A/A_0 at ~ 190 nm as a function of EUV dose. Considering that the absorbance reaches a steady value $A/A_0 \approx 0.55$ at high conversion, we can estimate that roughly 69% of

the C=C double bonds are converted at the D_{50} dose (Table 1).

3.7. Infrared Spectroscopy

Chemical changes in the organic part of the oxocluster (ligands) upon EUV exposure were further tracked using FTIR. Figure S3 shows the FTIR of Zn(TFMA) oxocluster in powder form and in thin film. The loss of two peaks (marked by asterisks) assigned to ν_{as} COO (TFMA) monodentate (1678 cm^{-1}) and ν C=C (TFMA) monodentate (1631 cm^{-1}) occurs upon thin film formation.¹⁸

Figure 7a represents the evolution of FTIR peaks of Zn(TFMA) thin films as a function of EUV dose. A decrease in the absorbance ratio A/A_0 (A_0 is the area under the peak of unexposed thin film) of ν C=C (1660 cm^{-1} , Figure 7b) was observed in agreement with NEXAFS and UV–vis spectroscopy. The conversion at D_{50} is approximately half of the maximum conversion achieved at high dose. Simultaneously, a decrease in the A/A_0 of $-\text{COO}$ (TFMA, 1678 cm^{-1}) was observed with increasing dose due to decarboxylation and outgassing of CO_2 upon EUV exposure. A/A_0 of C–F (1130 and 1167 cm^{-1}) also decreased, as in our previous studies. This is likely due to the formation of F^- species formed via dissociative electron attachment (DEA). More details about C–F bond cleavage in the organic shell of the oxocluster can be found in the XPS section below. The negative peak at 1100 cm^{-1} is assigned to Si–O bond stretching and is probably the result of a slightly thicker SiO_2 layer on the surface of Si substrate in the reference than in the sample.

In addition, the effect of PEB ($120\text{ }^\circ\text{C}/30\text{ s}$) on chemical changes was studied using ex situ UV–vis spectroscopy (Figure S21) and FTIR (Figure S22). The effect of PEB depends on the postexposure chemistry. In nonchemically amplified photoresists, PEB sometimes, but not always, results in the improvement of lithography performance.^{53–55} Contrast curve studies presented above revealed a slight decrease in the sensitivity (D_{50}) of Zn(TFMA)(MA)(TFA) and Zn(TFMA) oxoclusters, but this is probably within the margins of error. The UV–vis and FTIR results showed no significant extra decrease in the absorption intensity of $-\text{C}=\text{C}-$ upon PEB application. The lack of a strong effect of PEB in the present case is consistent with the proposed mechanisms, in which reactive intermediates that would survive ambient conditions do not play a role. For the overall patterning performance, PEB

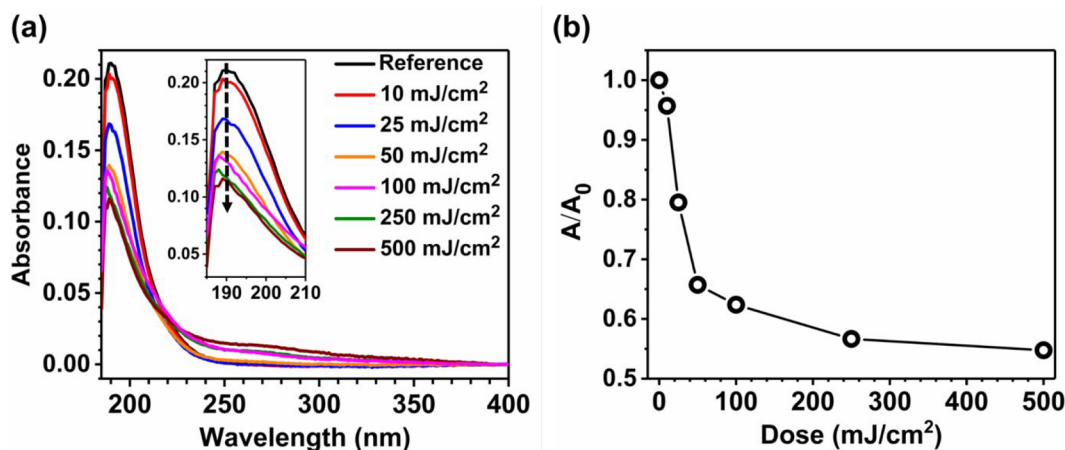
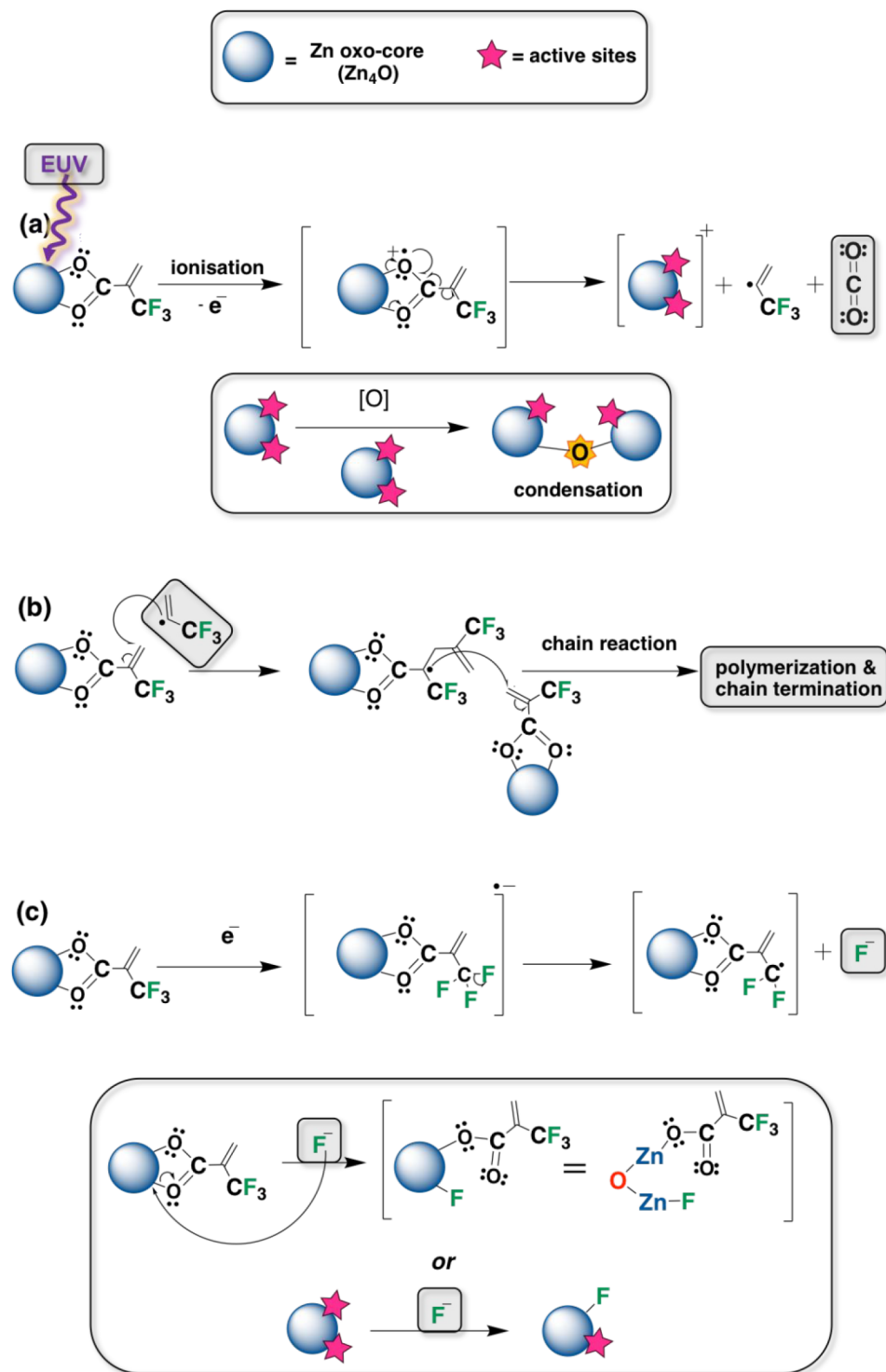


Figure 6. (a) UV–vis spectroscopy on Zn(TFMA) thin film as a function of EUV dose without application of PEB and no development. (b) A/A_0 at ~ 190 nm as a function of dose.

Scheme 1. Proposed Reaction Mechanisms for Zn(TFMA) Oxoclusters upon EUV Irradiation^a

^a(a) Ionization of oxocluster followed by decarboxylation reaction and generation of radical and active sites on oxo-core in the thin film, showing condensation reaction between two clusters; (b) radical initiated polymerization of the terminal double bond of TFMA ligand in the thin films; and (c) C-F bond cleavage via the DEA pathway, showing the formation of Zn-F species.

has a negative effect (increased scumming, Figure S16), probably because of thermal decomposition of the unexposed material.⁵⁶

3.8. Post-EUV Exposure In Situ XPS

To investigate the chemical changes induced by EUV irradiation of Zn(TFMA) in more depth, we combined ex

situ spectroscopy studies with in situ X-ray photoelectron spectroscopy. In situ spectroscopy allows avoidance of the chemical changes in the material due to the reaction between the sample and the ambient air, and carbon contamination of the surface. XPS spectra were recorded for C(1s), O(1s), and F(1s) as a function of EUV dose in a range of 5–500 mJ/cm².

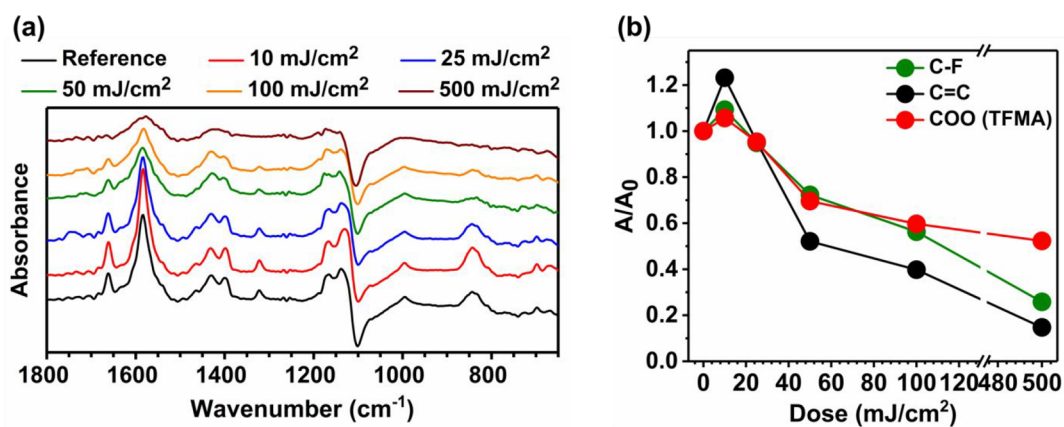


Figure 7. (a) FTIR spectra of the Zn(TFMA) thin film as a function of EUV dose (no PEB, no development). (b) A/A_0 as a function of dose for peaks in the spectra shown in (a); C–F ($1130\text{--}1167\text{ cm}^{-1}$), C=C (1631 cm^{-1}), and COO TFMA (1678 cm^{-1}).

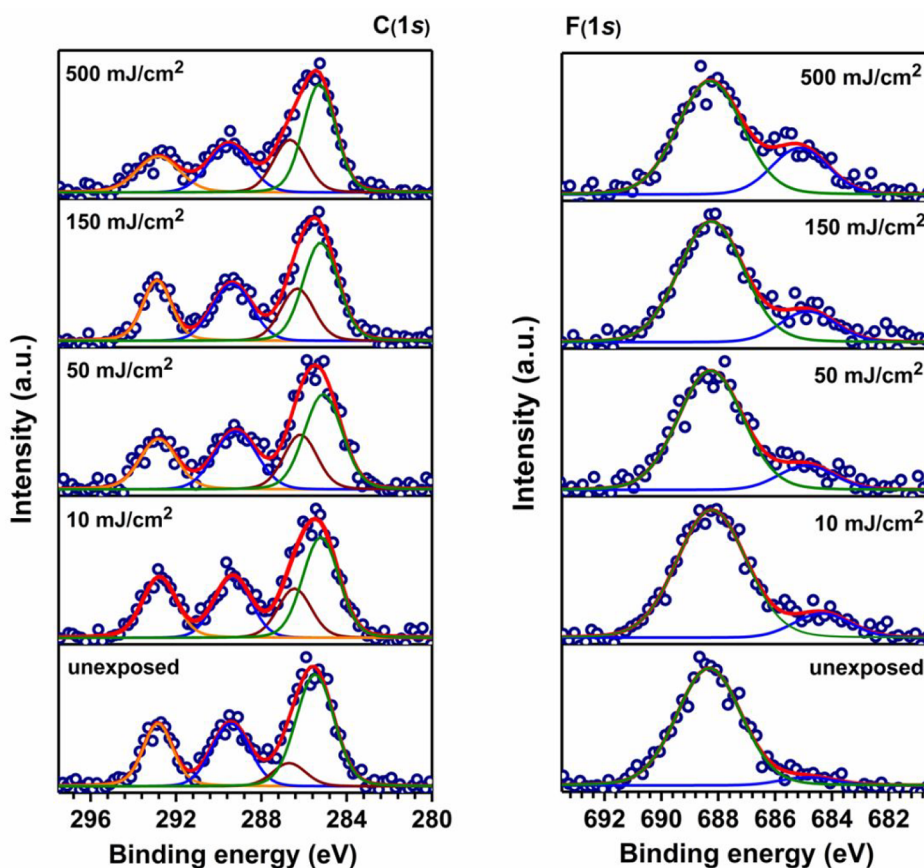


Figure 8. Selected C(1s) and F(1s) XPS spectra recorded as a function of EUV doses with photon energies of 400 and 800 eV, respectively. Fresh spots on the thin films were used for each EUV exposure ($\sim 92\text{ eV}$) followed by the XPS measurements.

A film thickness of 15 nm was measured using AFM on a reference sample spin coated on a Si substrate using the same conditions and same solution. Selected XPS spectra of the C(1s) and F(1s) ranges are shown in Figure 8. The C–F peaks at 292.81 eV for C(1s) and 688.21 eV for F(1s) were used as internal references to calibrate the binding energy scale of each spectrum.¹⁹ The C(1s) spectra can be fitted with four components assigned to sp^2 carbon ($285.58 \pm 0.25\text{ eV}$), sp^3 carbon ($286.41 \pm 0.21\text{ eV}$),⁵⁷ COO ($289.38 \pm 0.22\text{ eV}$), and C–F ($292.81 \pm 0.25\text{ eV}$).¹⁹ The fitted F(1s) spectra have two components assigned to C–F ($688.21 \pm 0.19\text{ eV}$) and Zn–F ($685.08 \pm 0.24\text{ eV}$).¹⁹

The fitted sp^2 carbon component is observed to decrease while the sp^3 component increases as a function of increasing EUV dose, in agreement with polymerization of the C=C double bonds of TFMA. The fractional area of the components assigned to –COO and C–F in the C(1s) spectra is observed to decrease as well, as a function of EUV dose, signifying the loss of –COO (outgassing as CO_2) and C–F from the organic shell of the ligand, as shown in Figure 9a.

The F(1s) XPS spectra also reveal a decrease of the C–F component (Figure 9b), and a new species assigned to a metal fluoride (Zn–F in this case) was observed. The fractional area of the Zn–F species is observed to increase as a function of

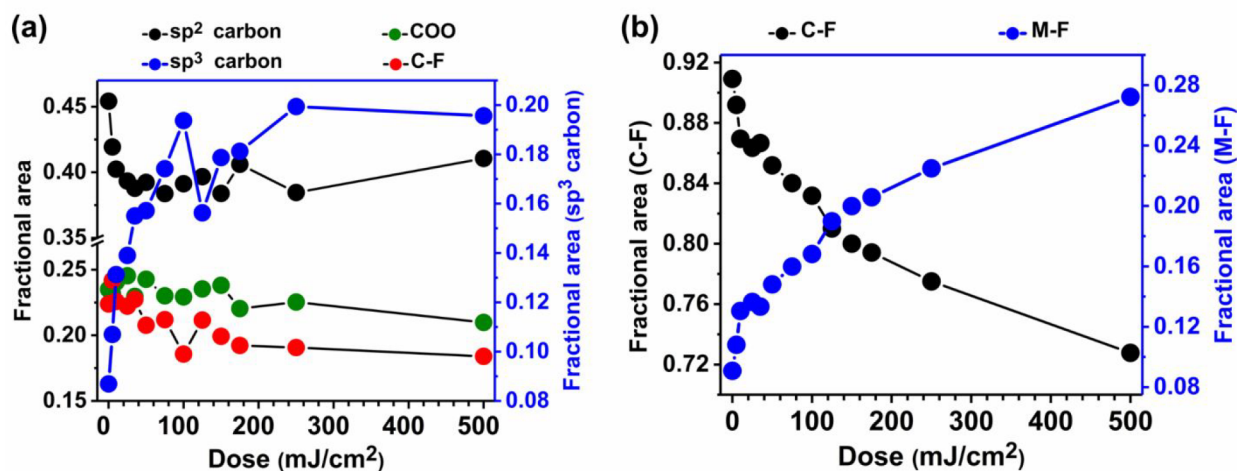


Figure 9. Fractional area of the fitted components as a function of EUV dose in XPS spectra of (a) C(1s), sp² carbon, -COO and C-F (left axis) and sp³ carbon (right axis); (b) F(1s), two components for C-F and M-F.

EUV dose, as also observed in our previous studies.¹⁹ The decrease in the C-F and simultaneous increase in the Zn-F is attributed to the migration of fluorine from the organic shell to the Zn-core, shown in Scheme 1c. In addition, the STXM studies also demonstrated no or very little loss of fluorine from the thin film after EUV exposure. The cleavage of the C-F bond is induced in the ligand via a well-known path: dissociative ion attachment (DEA)^{20,58} which is a common reaction pathway upon interaction with low energy electrons. The XPS spectra recorded for O(1s) were fitted with only one component (-COO) at a binding energy of 532.13 ± 0.17 eV. As an example, the fitted spectra for the unexposed thin film are shown in SI Figure S23 and the decrease in A/A_0 for the fitted -COO component in the O(1s) recorded spectra as a function of EUV dose also points toward decarboxylation, shown in SI Figure S23b.

3.9. Reaction Scheme

To summarize the results from all spectroscopy analysis and discussions above, the possible reaction pathways are proposed in Scheme 1. The ionization event of the Zn(TFMA) oxocluster after EUV photon absorption (Scheme 1a) will likely be followed by decarboxylation, consequently leaving active sites on the Zn-core of the oxocluster and formation of reactive radical species in the thin film. These radical species formed in step (a) can further initiate radical polymerization events in the thin film (cross-linking of the terminal double bond of TFMA) with the neighboring Zn oxocluster (Scheme 1b). Moreover, the low energy secondary electrons generated in the thin films after the ionization process (a) can induce bond cleavage of C-F via well-known DEA process⁵⁸ and parallel generation of radical species in thin films. The XPS studies (section 3.8) clearly indicate the formation of new Zn-F species after EUV exposure, shown in Scheme 1c.

The main contributor to the solubility switch is the cross-linking between ligands of neighboring clusters. The proposed radical mechanism allows propagation to form extended chains, in competition with termination reactions. Since reactive radicals do not survive long, especially under ambient conditions, the chemical reactions will be finished shortly after exposure, consistent with the lack of a PEB effect.

The different spectroscopic methods applied in the present work agree semiquantitatively on the extent of conversion that

is necessary to reach the solubility switch, which is the key property of a photoresist. Differences arise because different measurements were carried out on different samples and under different experimental conditions. Although the spin-coated films were shown to be fairly stable, some aging could not always be avoided, before and after exposure. Despite these quantitative limitations, it is evident that reaching the point of solubility switch, e.g., defined as the D_{50} value (Table 1) which is near 40 mJ/cm², requires a considerable conversion of the acrylate double bonds, of the order of 50%.

The proposed mechanism allows for multiple reactions per absorbed photon, in other words a quantum yield of reaction exceeding 1: multiple low energy electrons are expected to be generated per photon, each charge separation event can give rise to two reactive species, and the cross-linking reaction proceeds through a radical chain mechanism. A high quantum yield, however, is not guaranteed, because holes and electrons may recombine and the chain propagation may be interrupted by termination reactions. A challenge for future research is to quantify the processes in more detail. In particular, it is important to know how long chains are formed in the cross-coupling, what the termination steps are, and how the length of the chains is related to solubility change. Here, we can make an estimate of the average reaction quantum yield for conversion of the C=C bonds. Based on the linear EUV absorption coefficient (α) of $14.3 \mu\text{m}^{-1}$ for Zn(TFMA), and considering the thickness of the exposed thin film (d) of ~ 20 nm, the EUV transmittance, $T = e^{-\alpha d}$ through the resist is $\sim 75\%$. Consequently, $\sim 25\%$ of the photons are absorbed. At the D_{50} value of 40 mJ/cm² dose, at least $\sim 50\%$ of the acrylate double bonds has been converted, based on UV, IR, and XPS measurements. Using the estimated density of ~ 2 g/cm³, we find that the ratio of the number of C=C bonds converted to the number of photons absorbed is $\sim 10:1$, which suggests an average quantum yield of ~ 10 , in agreement with the mechanism.

4. CONCLUSIONS

In the present study, we have investigated fluorine-rich Zn oxoclusters for EUV lithography applications. The results have provided us with the following insights:

Stability and EUV lithography performance: Thin film stability of the fluorine-rich Zn(TFMA) oxocluster was increased

compared to its previously studied analogue Zn(MA)(TFA), while the sensitivity (D_{50}) for EUV lithography was decreased. The increased stability of the Zn(TFMA) oxocluster can be mainly attributed to the hydrophobic nature and high electronegativity of the fluorine. The decreased EUV sensitivity of fluorine-rich oxoclusters may arise from several factors: high electronegativity of fluorine slowing down the radical homopolymerization reaction rate; different molecular packing of the oxocluster in the thin-film; and different developer-resist interactions. This study shows that although the incorporation of more fluorine in such systems increases the EUV absorbance, this does not directly translate into higher EUVL sensitivity of the photoresist.

Chemical changes after EUV exposure: A combination of detailed spectroscopic studies shows decarboxylation of the carboxylate ligand upon EUV absorption in the Zn(TFMA) oxocluster. In addition, a strong decrease of the C=C/C=O area in STXM-NEXAFS studies as a function of EUV dose supports radical initiated polymerization in the thin film, which is also supported by UV-vis, FTIR, and in situ XPS studies. Furthermore, in situ XPS studies show the formation of new zinc fluoride species in the material upon EUV irradiation via the dissociative electron attachment pathway.

We believe that the present results enhance our understanding of Zn-based fluorine-rich photoresists and similar systems for lithography application and will advance focused investigations for fluorine-rich photoresist materials.

■ ASSOCIATED CONTENT

SI Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acsmaterialsau.1c00059>.

Synthesis reaction scheme for Zn(TFMA), synthesis of Zn(TFMA)(MA)(TFA) and its thin film deposition, brief description of scanning transmission X-ray microscope; stability study of the thin films of Zn(TFMA): UV-vis spectroscopy, silanization of the Si substrates, contrast curves for Zn(TFMA) and Zn(TFMA)(MA)(TFA) oxoclusters, spin coated thin films of Zn(TFMA) and Zn(TFMA)(MA)(TFA) oxoclusters, SEM and AFM images of L/S patterns, STXM-XAS on Zn(TFMA) oxocluster thin films, UV-vis spectroscopy, FTIR spectroscopy, XPS measurement (PDF)

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Notes

The authors declare no competing financial interest.

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■ REFERENCES

- (1) Li, L.; Liu, X.; Pal, S.; Wang, S.; Ober, C. K.; Giannelis, E. P. Extreme Ultraviolet Resist Materials for Sub-7 nm Patterning. *Chem. Soc. Rev.* **2017**, *46* (16), 4855–4866.
- (2) Ito, T.; Okazaki, S. Pushing the Limits of Lithography. *Nature* **2000**, *406* (6799), 1027–1031.
- (3) Wagner, C.; Harned, N. EUV Lithography: Lithography Gets Extreme. *Nat. Photonics* **2010**, *4* (1), 24–26.
- (4) Benschop, J. P. EUV Lithography: Past, Present and Future. *Proc. SPIE* **2021**, *11609*, 1160903.
- (5) De Simone, D.; Vesters, Y.; Vandenberghe, G. Photoresists in Extreme Ultraviolet Lithography (EUVL). *Adv. Opt. Technol.* **2017**, *6* (3–4), 163–172.
- (6) Narasimhan, A.; Wisehart, L.; Grzeskowiak, S.; Ocola, L. E.; Denbeaux, G.; Brainard, R. L. What We Don't Know About EUV Exposure Mechanisms. *J. Photopolym. Sci. Technol.* **2017**, *30* (1), 113–120.
- (7) Kostko, O.; Xu, B.; Ahmed, M.; Slaughter, D. S.; Ogletree, D. F.; Closser, K. D.; Prendergast, D. G.; Naulleau, P.; Olynick, D. L.; Ashby, P. D.; Liu, Y.; Hinsberg, W. D.; Wallraff, G. M. Fundamental Understanding of Chemical Processes in Extreme Ultraviolet Resist Materials. *J. Chem. Phys.* **2018**, *149* (15), 154305.
- (8) Ogletree, D. F. Molecular Excitation and Relaxation of Extreme Ultraviolet Lithography Photoresists. *Frontiers of Nanoscience* **2016**, *11*, 91–113.
- (9) Gallatin, G. M.; Narasimhan, A. K.; Brainard, R. L.; Neisser, M. Chapter 8: Photoresists for EUV Lithography. *EUV Lithography* **2018**, 493–592.

- (10) Neisser, M.; Cummings, K.; Valente, S.; Montgomery, C.; Fan, Y.-J.; Matthews, K.; Chun, J.; Ashby, P. D. Novel Resist Approaches to Enable EUV Lithography in High Volume Manufacturing and Extensions to Future Nodes. *Proc. SPIE* **2015**, *9422*, 94220L.
- (11) Manouras, T.; Argitis, P. High Sensitivity Resists for EUV Lithography: A Review of Material Design Strategies and Performance Results. *Nanomaterials* **2020**, *10* (8), 1593.
- (12) Lawson, R. A.; Robinson, A. P. G. Overview of Materials and Processes for Lithography. In *Frontiers of Nanoscience*; Elsevier, 2016; Vol. 11, pp 1–90.
- (13) Luo, C.; Xu, C.; Lv, L.; Li, H.; Huang, X.; Liu, W. Review of Recent Advances in Inorganic Photoresists. *RSC Adv.* **2020**, *10* (14), 8385–8395.
- (14) Sekiguchi, A.; Harada, T.; Watanabe, T. A Study on Enhancing EUV Resist Sensitivity. *Proc. SPIE* **2017**, *10143*, 1014322.
- (15) De Silva, A. Progress and Challenges of EUV Patterning Material Design. *Proc. SPIE* **2021**, *11609*, 116090G.
- (16) Neisser, M.; Cho, K.; Petrillo, K. The Physics of EUV Photoresist and How It Drives Strategies for Improvement. *J. Photopolym. Sci. Technol.* **2012**, *25* (1), 87–94.
- (17) Trikeriotis, M.; Krysak, M.; Chung, Y. S.; Ouyang, C.; Cardineau, B.; Brainard, R.; Ober, C. K.; Giannelis, E. P.; Cho, K. Nanoparticle Photoresists from HfO_2 and ZrO_2 for EUV Patterning. *J. Photopolym. Sci. Technol.* **2012**, *25* (5), 583–586.
- (18) Thakur, N.; Tseng, L.-T.; Vockenhuber, M.; Ekinici, Y.; Castellanos, S. Stability Studies on a Sensitive EUV Photoresist Based on Zinc Metal Oxoclusters. *J. Micro/Nanolith. MEMS MOEMS* **2019**, *18* (4), 043504.
- (19) Thakur, N.; Bliem, R.; Mochi, I.; Vockenhuber, M.; Ekinici, Y.; Castellanos, S. Mixed-Ligand Zinc-Oxoclusters: Efficient Chemistry for High Resolution Nanolithography. *J. Mater. Chem. C* **2020**, *8* (41), 14499–14506.
- (20) Rohdenburg, M.; Thakur, N.; Cartaya, R.; Castellanos, S.; Swiderek, P. Role of Low-Energy Electrons in the Solubility Switch of Zn-Based Oxocluster Photoresist for Extreme Ultraviolet Lithography. *Phys. Chem. Chem. Phys.* **2021**, *23* (31), 16646–16657.
- (21) Christianson, M. D.; Meyer, M. M.; Ongayi, O.; Valeri, D.; Wagner, M. High Absorbing Resists Based on Trifluoromethacrylate-Vinyl Ether Copolymers for EUV Lithography. *Proc. SPIE* **2013**, *8682*, 868216.
- (22) Sasaki, T.; Yokokoji, O.; Watanabe, T.; Kinoshita, H. Development of Partially Fluorinated EUV-Resist Polymers for LER and Sensitivity Improvement. *Proc. SPIE* **2008**, *6923*, 692347.
- (23) Wu, L.; Vockenhuber, M.; Ekinici, Y.; Castellanos Ortega, S. The Role of the Organic Shell in Hybrid Molecular Materials for EUV Lithography. *Proc. SPIE* **2019**, *10957*, 109570B.
- (24) Henke, B. L.; Gullikson, E. M.; Davis, J. C. X-Ray Interactions: Photoabsorption, Scattering, Transmission, and Reflection at $E = 50$ – $30,000$ eV, $Z = 1$ – 92 . *At. Data Nucl. Data Tables* **1993**, *54* (2), 181–342.
- (25) Thakur, N.; Giuliani, A.; Nahon, L.; Castellanos, S. Photon-Induced Fragmentation of Zinc-Based Oxoclusters for EUV Lithography Applications. *J. Photopolym. Sci. Technol.* **2020**, *33* (2), 153–158.
- (26) Patiny, L.; Borel, A. ChemCalc: A Building Block for Tomorrow's Chemical Infrastructure. *J. Chem. Inf. Model.* **2013**, *53* (5), 1223–1228.
- (27) Mojarad, N.; Fan, D.; Gobrecht, J.; Ekinici, Y. Broadband Interference Lithography at Extreme Ultraviolet and Soft X-Ray Wavelengths. *Opt. Lett.* **2014**, *39* (8), 2286–2289.
- (28) Mojarad, N.; Gobrecht, J.; Ekinici, Y. Interference Lithography at EUV and Soft X-Ray Wavelengths: Principles, Methods, and Applications. *Microelectron. Eng.* **2015**, *143*, 55–63.
- (29) Raabe, J.; Tzvetkov, G.; Flechsig, U.; Böge, M.; Jaggi, A.; Sarafimov, B.; Vernooij, M. G. C.; Huthwelker, T.; Ade, H.; Kilcoyne, D.; Tyliszczak, T.; Fink, R. H.; Quitmann, C. PoLux: A New Facility for Soft x-Ray Spectromicroscopy at the Swiss Light Source. *Rev. Sci. Instrum.* **2008**, *79* (11), 113704.
- (30) Nannarone, S.; Borgatti, F.; Deluisa, A.; Doyle, B. P.; Gazzadi, G. C.; Giglia, A.; Finetti, P.; Mahne, N.; Pasquali, L.; Pedio, M.; Selvaggi, G.; Naletto, G.; Pelizzo, M. G.; Tondello, G. The BEAR Beamline at Elettra. *AIP Conf. Proc.* **2003**, *705* (1), 450–453.
- (31) Pasquali, L.; De Luisa, A.; Nannarone, S. The UHV Experimental Chamber for Optical Measurements (Reflectivity and Absorption) and Angle Resolved Photoemission of the BEAR Beamline at ELETTRA. *AIP Conf. Proc.* **2003**, *705* (1), 1142–1145.
- (32) CXRO X-Ray Interactions With Matter. http://henke.lbl.gov/optical_constants/ (accessed 2021-04-16).
- (33) Fallica, R.; Haitjema, J.; Wu, L.; Castellanos, S.; Brouwer, A. M.; Ekinici, Y. Absorption Coefficient of Metal-Containing Photoresists in the Extreme Ultraviolet. *J. Micro/Nanolithography, MEMS, MOEMS* **2018**, *17* (2), 023505.
- (34) Clegg, W.; Harbron, D. R.; Homan, C. D.; Hunt, P. A.; Little, I. R.; Straughan, B. P. Crystal Structures of Three Basic Zinc Carboxylates Together with Infrared and FAB Mass Spectrometry Studies in Solution. *Inorg. Chim. Acta* **1991**, *186* (1), 51–60.
- (35) Hiltunen, L.; Leskelä, M.; Mäkelä, M.; Niinistö, L. Crystal Structure of Mu4-Oxo-Hexakis(Mu-Acetato)Tetrazinc and Thermal Studies of Its Precursor, Zinc Acetate Dihydrate. *Acta Chem. Scand.* **1987**, *41a*, 548–555.
- (36) Jones, R. G.; Ober, C. K.; Hayakawa, T.; Luscombe, C. K.; Stingelin, N. Terminology of Polymers in Advanced Lithography (IUPAC Recommendations 2020). *Pure Appl. Chem.* **2020**, *92* (11), 1861–1891.
- (37) Li, R. Q.; Wang, M. X.; Zhang, Q. Y.; Chen, J. G.; Wang, K.; Zhang, X. Y.; Shen, S.; Liu, Z. T.; Liu, Z. W.; Jiang, J. Insight into the Intermolecular Interaction and Free Radical Polymerizability of Methacrylates in Supercritical Carbon Dioxide. *Polymers.* **2020**, *12* (1), 78.
- (38) Ito, H.; Miller, D. C. Radical Copolymerization of 2-Trifluoromethylacrylic Monomers. I. Kinetics of Their Copolymerization with Norbornenes and Vinyl Ethers as Studied by in Situ ^1H NMR Analysis. *J. Polym. Sci. Part A Polym. Chem.* **2004**, *42* (6), 1468–1477.
- (39) Ito, H.; Miller, D. C.; Willson, C. G. Polymerization of Methyl A-(Trifluoromethyl)Acrylate and a-(Trifluoromethyl) Acrylonitrile and Copolymerization of These Monomers with Methyl Methacrylate. *Macromolecules* **1982**, *15* (3), 915–920.
- (40) Wu, L.; Bepalov, I.; Witte, K.; Lugier, O.; Haitjema, J.; Vockenhuber, M.; Ekinici, Y.; Watts, B.; Brouwer, A. M.; Castellanos, S. Unravelling the Effect of Fluorinated Ligands in Hybrid EUV Photoresists by X-Ray Spectroscopy. *J. Mater. Chem. C* **2020**, *8* (42), 14757–14765.
- (41) Fallica, R.; Watts, B.; Rösner, B.; Della Giustina, G.; Brigo, L.; Brusatin, G.; Ekinici, Y. Changes in the near Edge X-Ray Absorption Fine Structure of Hybrid Organic-Inorganic Resists upon Exposure. *Nanotechnology* **2018**, *29* (36), 36LT03.
- (42) Watts, B.; Ade, H. NEXAFS Imaging of Synthetic Organic Materials. *Mater. Today* **2012**, *15* (4), 148–157.
- (43) Watts, B.; Swaraj, S.; Nordlund, D.; Lüning, J.; Ade, H. Calibrated NEXAFS Spectra of Common Conjugated Polymers. *J. Chem. Phys.* **2011**, *134* (2), 024702.
- (44) Su, G. M.; Patel, S. N.; Pemmaraju, C. D.; Prendergast, D.; Chabinc, M. L. First-Principles Predictions of Near-Edge X-Ray Absorption Fine Structure Spectra of Semiconducting Polymers. *J. Phys. Chem. C* **2017**, *121* (17), 9142–9152.
- (45) Outka, D. A.; Stöhr, J.; Rabe, J. P.; Swalen, J. D. The Orientation of Langmuir-Blodgett Monolayers Using NEXAFS. *J. Chem. Phys.* **1988**, *88* (6), 4076–4087.
- (46) Outka, D. A.; Stöhr, J. Curve Fitting Analysis of Near-Edge Core Excitation Spectra of Free, Adsorbed, and Polymeric Molecules. *J. Chem. Phys.* **1988**, *88*, 3539.
- (47) Dhez, O.; Ade, H.; Urquhart, S. G. Calibrated NEXAFS Spectra of Some Common Polymers. *J. Electron Spectrosc. Relat. Phenom.* **2003**, *128* (1), 85–96.

- (48) Koningsberger, D. C.; Mojet, B. L.; Van Dorssen, G. E.; Ramaker, D. E. XAFS Spectroscopy; Fundamental Principles and Data Analysis. *Top. Catal.* **2000**, *10* (3), 143–155.
- (49) Solomon, D.; Lehmann, J.; Kinyangi, J.; Liang, B.; Heymann, K.; Dathe, L.; Hanley, K.; Wirick, S.; Jacobsen, C. Carbon (1s) NEXAFS Spectroscopy of Biogeochemically Relevant Reference Organic Compounds. *Soil Sci. Soc. Am. J.* **2009**, *73* (6), 1817–1830.
- (50) Bagus, P. S.; Weiss, K.; Schertel, A.; Wöll, C.; Braun, W.; Hellwig, C.; Jung, C. Identification of Transitions into Rydberg States in the X-Ray Absorption Spectra of Condensed Long-Chain Alkanes. *Chem. Phys. Lett.* **1996**, *248* (3–4), 129–135.
- (51) Sedlmair, J.; Gleber, S. C.; Peth, C.; Mann, K.; Niemeyer, J.; Thieme, J. Characterization of Refractory Organic Substances by NEXAFS Using a Compact X-Ray Source. *J. Soils Sediments* **2012**, *12* (1), 24–34.
- (52) Tiwale, N.; Subramanian, A.; Freychet, G.; Gann, E.; Kisslinger, K.; Lu, M.; Stein, A.; Kim, J.; Nam, C.-Y. Hybrid Resist Synthesis by Ex-Situ Vapor-Phase Infiltration of Metal Oxides into Conventional Organic Resists. *Proc. SPIE* **2021**, *11612*, 116120A.
- (53) Haitjema, J.; Zhang, Y.; Vockenhuber, M.; Kazazis, D.; Ekinci, Y.; Brouwer, A. M. Extreme Ultraviolet Patterning of Tin-Oxo Cages. *J. Micro/Nanolith. MEMS MOEMS* **2017**, *16* (03), 033510.
- (54) Ekinci, Y.; Vockenhuber, M.; Terhalle, B.; Hojeij, M.; Wang, L.; Younkin, T. R. Evaluation of Resist Performance with EUV Interference Lithography for Sub-22-Nm Patterning. *Proc. SPIE* **2012**, *8322*, 83220W.
- (55) Kenane, N.; Keszler, D. A. High-Resolution Lithographic Patterning with Organotin Films: Role of CO₂ in Differential Dissolution Rates. *ACS Appl. Mater. Interfaces* **2021**, *13* (16), 18974–18983.
- (56) Mattson, E. C.; Cabrera, Y.; Rupich, S. M.; Wang, Y.; Oyekan, K. A.; Mustard, T. J.; Halls, M. D.; Bechtel, H. A.; Martin, M. C.; Chabal, Y. J. Chemical Modification Mechanisms in Hybrid Hafnium Oxo-Methacrylate Nanocluster Photoresists for Extreme Ultraviolet Patterning. *Chem. Mater.* **2018**, *30* (17), 6192–6206.
- (57) Drewniak, S.; Muzyka, R.; Stolarczyk, A.; Pustelny, T.; Kotyczka-Morańska, M.; Setkiewicz, M. Studies of Reduced Graphene Oxide and Graphite Oxide in the Aspect of Their Possible Application in Gas Sensors. *Sensors* **2016**, *16* (1), 103.
- (58) Langer, J.; Stano, M.; Gohlke, S.; Foltin, V.; Matejcik, S.; Illenberger, E. Reactions in Trifluoroacetic Acid (CF₃COOH) Induced by Low Energy Electron Attachment. *Chem. Phys. Lett.* **2006**, *419* (1–3), 228–232.