One-pot synthesis of liquid photocrosslinkable poly(L-lactide) with terminal triacrylate

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We synthesized a poly(L-lactide)–pentaerythritol triacrylate (PETA) polymer modified with acrylic trifunctional groups using a one-pot method based on ring-opening polymerization of L-lactide and PETA. We calculated the molecular weight and structure of PLA–PETA using gel permeation chromatography (GPC) and nuclear magnetic resonance (NMR) (1H, 13C, heteronuclear multiple bond correlation [HMBC]) spectroscopy. Photocrosslinking PLA–PETA using the Omnirad 1173 photoinitiator yielded a transparent sample with 91% crosslinkage. The crosslinked sample was analyzed using differential scanning calorimetry (DSC), thermogravimetric analysis (TGA), and thermomechanical analysis (TMA) to determine its thermal properties and thermal expansion coefficient. *In vitro* cell toxicity tests showed an average cell viability >90%, indicating that the PLA–PETA polymer had good biocompatibility with cells after photocrosslinking.

1. Introduction

Polylactic acid (PLA) is a widely used aliphatic polyester material with excellent mechanical properties, high biocompatibility, and good biodegradability. Carothers first synthesized low-molecular-weight PLA in 1932. In 1954, the DuPont company obtained the patent for producing high-molecular-weight polymeric PLA. Since PLA can be produced using renewable resources, it is considered to be environmentally friendly. It can be mixed into various resins using numerous techniques to produce materials that can be used in industrial packaging, agriculture, and medical treatments.[1–3]

PLA has the widest biomedical application among all synthetic aliphatic polyesters, and the US Food and Drug Administration (FDA) approved it for contact with human cells in 1970. In 1974, the DuPont company introduced PLA sutures. Since then, much research and development has been conducted on using methods such as chemical modification, copolymerization, and blending to produce forms of PLA that can be used in biomedical applications, such as bone fixation, stent screw implanting, tissue engineering, and drug delivery.[3–8]

PLA application is usually limited by its shortcomings of brittleness, low toughness, slow crystallization, poor heat resistance, insufficient mechanical strength, and poor shape stability at high temperatures. Scientific efforts have been made to improve the crystallinity, thermal stability, and mechanical

properties of PLA using methods such as chemical modification, copolymerization, or hybrid processing.[9–14] A commonly used chemical method for polymer modification involves incorporating appropriate end groups. End groups are introduced into the PLA backbone through polymerization or the post-polymerization end group modification of compounds that already have the desired functionality while retaining the properties of the PLA matrix. Although molecules can be polymerized with desired end groups or introduce various chain-end molecules in polymerization followed by post-polymerization functionalization, the reactions may not be complete. In addition, the end-group functionalization of PLA generally requires multiple reaction steps.[15–22]

Crosslinking allows PLA chains to chemically bond into a network structure, thereby improving the toughness and thermal properties of the polymer and changing its solubility and gas permeability. Unfortunately, crosslinking reduces degradability; however, the degradation process of PLA can be regulated by controlling process conditions and parameters. The crosslinking methods of PLA consist of chemical, ionizing radiation, and ultraviolet (UV)/visible light crosslinking. The chemical crosslinking of PLA usually requires using crosslinkers or initiators, such as polyisocyanates, polyfunctional anhydrides, and peroxides (for generation of free radicals), to trigger the formation of crosslinked structures.[23] Gamma rays or electron beams have been used for crosslinking and grafting polymers, while the structure and length of the PLA chain and radiation dose affect the crosslinking density. Polymer crosslinking by radiation does not require functionalization, considering free radicals can be generated directly in the polymer chains. However, PLA crosslinked by radiation in the

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absence of a crosslinker will undergo molecular-chain breaking.[24,25]

Photocrosslinking has biomedical applications as it can be performed under low-viscosity, solvent-free, and low-temperature conditions. The crosslinking of PLA involves double-bond functionalization at chain ends to form linear, branched, or star copolymers, followed by UV irradiation to initiate radical polymerization.[16,18,21] Three-dimensional chemical bonding between PLA chains can be formed via photocrosslinkable groups, which enhance the hydrolysis resistance, thermal stability, and mechanical properties of PLA.[25–31]

Pentaerythritol triacrylate (PETA), a monomer containing side hydroxyl and three acrylic functional groups, has high reactivity, high crosslinking density, high hardness, strong water resistance, and strong chemical resistance. It is mainly used in free radical polymerization and serves as a reactive diluent of polyfunctional acrylate widely used in UV curing. Currently, photocrosslinked PLA with a network structure can be formed in 2–3 steps. Here, we designed a modified-PLA liquid polymer with three acrylic functional groups, where L-lactide (LA) and pentaerythritol triacrylate (PETA) were used as the reaction monomers. To obtain PLA–PETA polymers with low molecular weight, we controlled the molar ratio of the reaction monomer (LA : PETA) and performed ring-opening polymerization (ROP) using a one-pot method. This study aimed to develop a photocrosslinkable and biodegradable material based on low-molecular-weight PLA and evaluate its chemical and thermal properties and biocompatibility.[28,31]

2. Experimental

2.1 Materials

L-Lactide 99.5% (Green Square Materials Inc), pentaerythritol tri and tetraacrylate (Photomer 4335), tin(II) 2-ethylhexanoate 96% (Sn(Oct)$_2$,Alfa Aesar®), 4-methoxyphenol (SHORA), 2-hydroxy-2-methyl-1-phenylpropanone (I.G.M Omnirad 1173), PBS Tablets (MP Biomedicals), tetrahydrofuran (TEDIA), ethyl acetate (TEDIA), acetone (TEDIA), high-purity grade silica gel, pore size 60 Å, 70–230 mesh (AACASH Exports).

2.2 Synthesis of PLA–PETA

In this study, we used pentaerythritol tri- and tetraacrylate (Photomer 4335), which contain three and four acrylic

functional groups, respectively, to prevent impurities interfering with the polymerization reaction. A glass cylinder was filled with silica gel and mobile-phase 70% ethyl acetate/30% N-hexane, to which we added the Photomer 4335, allowing us to separate out the PETA.

The monomer L-lactide and PETA in molar ratios of 3 : 1 (0.15 : 0.05), 5 : 1 (0.25 : 0.05), 7 : 1 (0.35 : 0.05) were respectively placed in 250 mL three-neck flasks and dried under vacuum for 1 h. To the flasks filled with dry nitrogen gas were added Sn(Oct)$_2$ (0.05 wt% of lactide) and 200 ppm of 4-methoxyphenol, and the mixtures were heated at 130 °C in an oil bath for 12 h with stirring (Scheme 1).

At the end of the reaction, the reactors were cooled to room temperature. The resulting polymers were dissolved in acetone and precipitated in an excess of deionized water while being agitated. The precipitated materials were washed with deionized water and dried under reduced pressure. These polymers were denoted as PLA–PETA, where PLA–PETA is the number-average molecular weight determined by 1H nuclear magnetic resonance (NMR) spectroscopy and gel permeation chromatography (GPC, Fig. 1, see Section 2.4 below and Table 1).

We found that the polymer with a 7 : 1 molar ratio was a solid, so it did not meet the requirements of being able to set in a liquid state. Further, the LA chain segment had a 3 : 1 molar ratio and its chain length was too short. We therefore set the photopolymerization conditions for PLA–PETA assuming a molar ratio of 5 : 1 and analyzed the properties of the PLA–PETA based on this molar ratio.
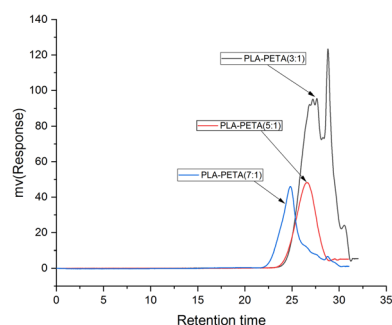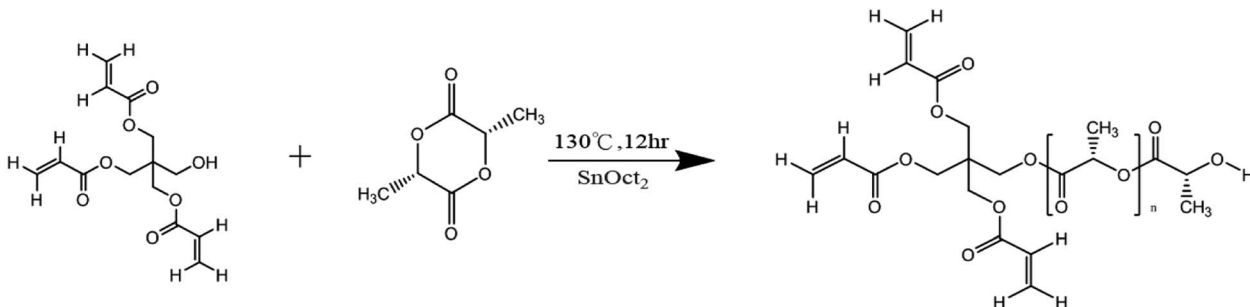

Fig. 1 GPC diagrams of PLA–PETA copolymers, mobile phase: THF, 1 mL min^{-1}, 35 °C.

Scheme 1 Synthesis of PLA–PETA.

Table 1 PLA–PETA samples in terms of their molecular weight and state of matter[a]

Polymer	LA : PETA[a]	Mn_{MNR}[b] (g mol^{-1})	Mw_{GPC}[c] (g mol^{-1})	Mn_{GPC}[c] (g mol^{-1})	Mw/Mn_{GPC}[c]	State of matter
PLA–PETA-3 : 1	0.15 : 0.05	519	985	684	1.44	Liquid
PLA–PETA-5 : 1	0.25 : 0.05	801	1907	1544	1.23	Liquid
PLA–PETA-7 : 1	0.35 : 0.05	1371	3644	2731	1.33	Solid

[a] LA : PETA = mol : mol. [b] $Mn = ((H_{Area}{}^{8}/H_{Area}{}^{9}) \times 72) + 298 + 72$. [c] Molecular weights determined by GPC calibrated with PS standards.

2.3 Photopolymerization conditions[21,32]

The photocrosslinking process was carried out in a UV cabinet (Philips, HILIPS HPA 400/30S, intensity 10–13 mW cm^{-2}) using 0.5 wt% Omnirad 1173 as a photoinitiator. The following conditions were used: time of irradiation – maximum 5 min, air atmosphere and room temperature. We obtained the transparent PLA–PETA polymer liquid. This PLA–PETA was subjected to photocrosslinking reaction to become transparent solid polymer indicated as PLA–PETA–PO (Fig. 2).

2.4 Polymer characterization

2.4.1 NMR spectroscopy. The chemical structures and compositions of the obtained polymers were characterized using a Bruker Avance III HD-600 MHz NMR spectrometer at ambient temperature using CDCl$_3$ as the corresponding solvent with chemical shifts referenced to residual CHCl$_3$. The number-average molecular weights of PLA–PETA were calculated from NMR spectral analysis.[28,33,34]

2.4.2 Gel permeation chromatography (GPC). The molecular weight and molecular weight distribution were determined by GPC (Jasco,RI-2031 Plus) at 35 °C, with tetrahydrofuran was used as the eluent at a flow rate of 1.0 mL min^{-1}, and the sample concentration was 5 mg mL^{-1}. Molecular weights were calculated against polystyrene (PS) standards.

2.4.3 Fourier transform infrared spectroscopy analysis (FTIR). FTIR spectroscopy analysis was conducted with a PerkinElmer Spectrum 100 apparatus equipped with a germanium attenuated total reflectance (ATR) crystal. Measurements were performed at a resolution of 4 cm^{-1} and the accumulation of 16 scans in the wavelength range of 550–4000 cm^{-1} at room temperature. For reproducible test results, at least two measurements of each sample were taken.

2.4.4 Differential scanning calorimetry (DSC). DSC measurements were performed using a PerkinElmer Pyris 6 differential scanning calorimeter operated under a N$_2$ atmosphere. The samples (∼5 mg) were placed in a DSC pan and heated from 5 to 200 °C at a heating rate of 10 °C min^{-1}. The midpoint of the change in the slope of the plot of the heat capacity during the heating scan was taken to be the glass transition temperature.

2.4.5 Thermogravimetric analysis (TGA). The thermal stability of the polymers was characterized using TGA employing a PerkinElmer Pyris 1 TGA instrument. Samples of 3–6 mg were heated from 30 °C to 830 °C at a rate of 10 °Cmin^{-1} under nitrogen atmosphere at a flow rate of 20 cm^3 min^{-1}.

2.4.6 Thermomechanical analysis (TMA). TMA measurements were performed using TA Instruments TMA Q400 apparatus. The force applied was 0.05 N and the sample was heated from 15 to 180 °C at a heating rate of 3 °C min^{-1}.

2.4.7 Derivation of gel content. After irradiation, the gel fractions of the samples were measured by first dissolving 200 mg of the polymers in acetone at room temperature for 48 h. Thereafter, the solutions were filtered and the insoluble portions were vacuum dried at 50 °C for 8 h to ensure that no residual acetone remained.[20,32,35] The final gel fractions were calculated using the following equation:

$$\text{Gel fraction (\%)} = (W_B/W_A) \times 100\%$$

where W_A is the initial dry weight of the crosslinked polymer and W_B is the remaining weight (dry gel component) of the crosslinked polymer after dissolving it in 50 °C at room temperature for 48 h.

2.4.8 Hydrolytic degradation tests.[20,36] PLA–PETA–PO crosslinked lumps were chosen for *in vitro* hydrolytic degradation testing. After crushing the photocrosslinked lumps of PLA–PETA–PO, the fine powder obtained was pressed through a standard sieve (#35). In a shaking incubator chamber, the particles were placed in 15 mL screw-top bottles filled with phosphate-buffered saline (PBS) solution (pH 7.4) and incubated at 37 °C. Samples were removed every 10 days and the PBS was replaced for all the remaining vials before being rinsed with distilled water and surface dried using water-absorbent paper. The samples were then dried in a vacuum oven at 45 °C for 2 days, after which the final dry mass (W_2) was recorded. Mass loss was calculated as the difference between the dry mass (W_2) and the initial mass (W_1) of each sample. Results for mass loss

Fig. 2 Photocrosslinking of PLA–PETA–PO.

were normalized by dividing the mass loss over the initial mass (W_1), in terms of percentage, calculated using the following equation:

$$\text{Mass loss } \% = \left(\frac{W_1 - W_2}{W_1} \right) \times 100\%$$

2.4.9 Cytotoxicity tests. Cytotoxicity testing of the test substances was conducted in a sterile environment according to the method described in the ISO 10993-12 specification. The experimental substance was extracted in proportion to 0.2 g ± 10% ml^{-1}, and 0.2 g of the crosslinked test substance was added to 1 mL of extraction buffer at 37 ± 1 °C for an extraction period of 24 ± 2 h. *In vitro* cytotoxicity test assessment with reference to ISO 10993-5, take 100 μL, L929 cell suspension with a concentration of 1×10^5 cells per mL, add the preparation completed test substance dilution, cultured 24 ± 2 hours. Three repeated tests were carried out in both the positive and negative control groups and the dilution of the test material.

3. Results

3.1 1H NMR spectroscopy characterization

Fig. 3 shows the NMR spectrum of PLA–PETA. In the 1H NMR spectrum, the –CH absorption peaks of PLA can be assigned as $\delta H^4 = 5.10$–5.17 ppm [1H, CH], the methyl-CH$_3$ absorption peaks of PLA can be assigned as δH^8, $\delta H^9 = 1.45$–1.56 ppm [3H, CH$_3$], the absorption peaks of the CH$_3$ at one end of the PLA–PETA polymer can be assigned as $\delta H^5 = 4.32$–4.34 ppm [1H, –CH], the CH$_2$ absorption peak of PETA can be assigned as δH^6, $\delta H^7 = 4.22$–4.27 ppm [2H, CH$_2$], and the vinyl hydrogen (–CH = CH$_2$) absorption peak of PETA can be assigned as δH^1, δH^2, $\delta H^3 = 5.84$–6.42 ppm [3H, CH = CH$_2$]. The molecular weight of lactide monomer (C$_6$H$_8$O$_4$) is 144. After the ring-opening polymerization reaction of lactide monomers, a polymer with a repeat unit of (COCHCH$_3$O)n is formed. The molecular weight of this repeat unit is calculated to be 72. We calculated the number-average molecular weight (Mn) based on that of the repeated unit CH$_3$ (δH^8) in the PLA chain segment and the peak integrated area of
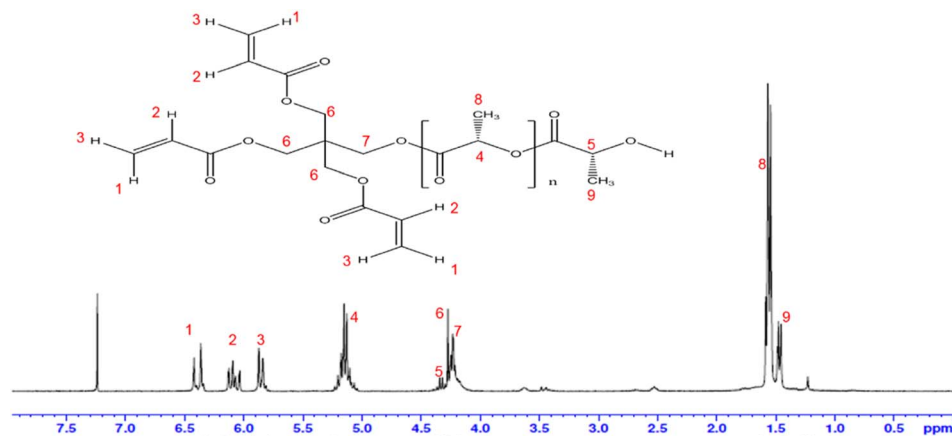

Fig. 3 ^{1}H NMR (300 MHz, CDCl$_3$, 290 K) spectrum of PLA–PETA.

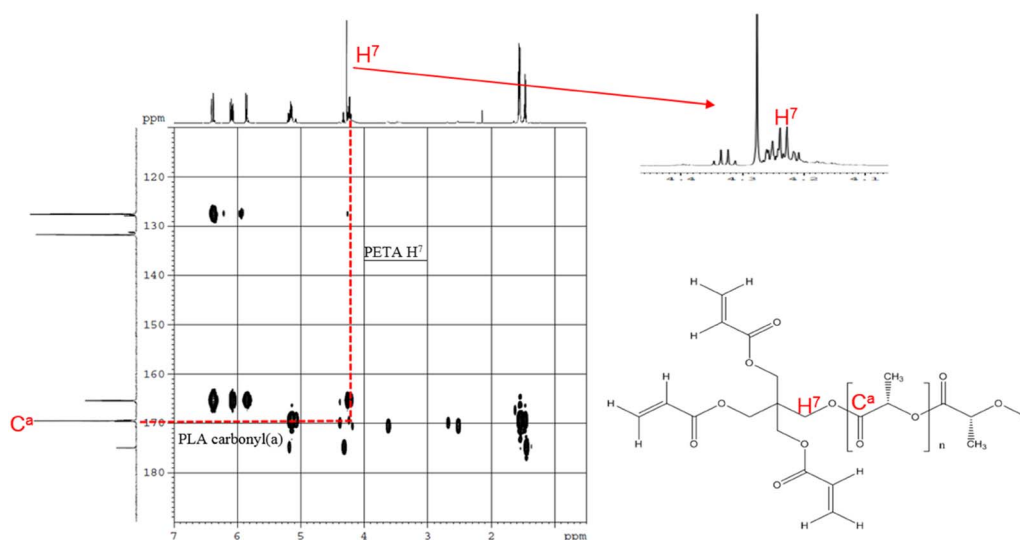**Fig. 4** ^{1}H–^{13}C HMBC NMR spectrum of PLA–PETA. NMR correlations are due to the covalent bonding between PLA and PETA.

the CH$_3$ (δH^9) at one end of the PLA–PETA polymer (Table 1). The Mn of the PLA–PETA copolymer measured *via* gel permeation chromatography (GPC) was higher than that calculated *via* 1H NMR spectroscopy because the hydrodynamic volume of standard polystyrene was lower than that of the aliphatic polyester with the same molar mass, so the measurement data differed.

The integral area of δH^{8}Area: CH$_3$.

The integral area of δH^{9}Area: CH$_3$.

Mn = ((δH^{8}Area/δH^{9}Area) × 72) + 298 + 72 (molecular weight of PETA: 298).

3.2 Spectral characterization of ^{1}H–^{13}C HMBC

A liquid-cooled superconducting NMR system (Bruker Avance III HD-600 MHz) was used to confirm the carbon–hydrogen linkages in the synthesis structure of PLA–PETA *via* ^{1}H–^{13}C heteronuclear multiple bond correlation (HMBC) spectroscopy. In the ^{1}H–^{13}C HMBC spectrum (Fig. 4), the carbon base unit, C=O (δC^a = 169 ppm), of the PLA chain segment has a distal correlation with the CH$_2$ (δH^7 = 4.22–4.24 ppm) of the PETA segment. This suggests that when PETA is mixed with the catalyst Sn(Oct)$_2$, the lactide uses the hydroxyl group of PETA as the initiator for ROP, forming a short-chain polymer containing three acrylic functional groups.[34]

3.3 Gel formation of PLA–PETA

According to the infrared spectra shown in Fig. 5, the peak at 1635 cm^{-1} is characteristic of the C=C double bond in PLA–
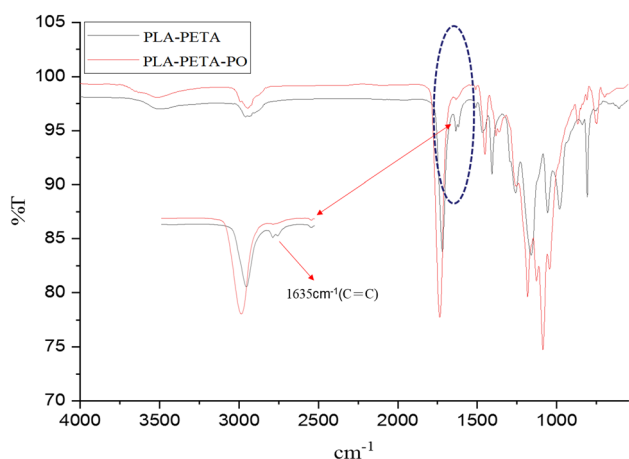

Fig. 5 FTIR spectra of PLA–PETA and PLA–PETA–PO, which PLA–PETA–PO is the polymer solid after photocrosslinking reaction of PLA–PETA polymer liquid.

PETA. There were two obvious wavelet peaks before the crosslinking was done, and these disappeared afterward. This suggests that when the active radicals react with the monomer double bonds, the molecular chains grow and become highly crosslinked, forming a cross-network polymer. Due to the crosslinking and vitrifying relationship of the polymers, some free radicals and unreacted double bonds were restricted from participating in the crosslinking polymerization reaction, resulting in failure to reach 100% conversion. The gel content test showed that the gel content was 91% after crosslinking.[32,34]

3.4 Thermal analysis

Table 2 shows the thermal properties of PLA–PETA, measured *via* DSC, TGA, and TMA. The polymer chain structure, molecular weight and degree of crystallization affect the glass transition temperature (T_g) and the thermal degradation temperature. Before crosslinking, PLA–PETA is a short-chain polymer, and after photocrosslinking it forms a transparent network structure. The T_g of PLA–PETA after crosslinking is 47.7 °C, 55.5 °C higher than before crosslinking (−7.8 °C; Fig. 6). The crosslinking of PLA–PETA led to a high crosslinking density of the side acrylate in the copolymer, with the resulting high-density crosslinked network structure preventing the PLA chains from readily relaxing and rotating. Therefore, the T_g still reached 47.7 °C when the molecular chains did not form a high molecular weight structure.

It can be seen from Fig. 7 that PLA has only a single-stage thermal degradation curve, with a 5% weight loss temperature ($T_{5\%}$) of 280.78 °C, and its highest thermal decomposition temperature is 299.66 °C. PLA–PETA–PO has two thermal degradation stages: that of the PLA chain segment and that of the PETA chain segment. Specifically, the first stage is the decomposition of the oligomer ester bond in the PLA from 268.53 °C to 298.67 °C. The second stage is the decomposition of the PETA chain segment, starting from the turning point at 400 °C and ending at the maximum thermal decomposition temperature, 452.28 °C. In our experiment, at 500 °C all materials had almost entirely decomposed, leaving only ash.

Fig. 8 shows the coefficient of thermal expansion (CTE) and the T_g of the TMA test samples. The CTE indicates the degree of expansion and contraction of materials under temperature changes. According to the MatWeb materials properties datasheet for the PLA biopolymer, the CTE of PLA is 101–120 (μm m^{-1} °C^{-1}).[37] In this study, the CTE of PLA–PETA was 153.6 (μm m^{-1} °C^{-1}) as the high crosslinking density of the side acrylate in PLA–PETA made the movement of molecular chains difficult.

Table 2 Summary of the DSC, TGA and TMA results[a]

Polymer	$T_{g,DSC}$[a] (°C)	$T_{g,DSC}$[b] (°C)	$T_{g,TMA}$ (°C)	$T_{5\%}$[c] (°C)	T_{max1}[d]	T_{max2}[d]	CTE (μm m^{-1} °C^{-1})
PLA–PETA–PO	−7.8	47.7	45.8	268	297	452	153.6

^{a} Non-crosslinked PLA–PETA. ^{b} Crosslinked PLA–PETA–PO. ^{c} Decomposition temperature of polymers at a weight loss of 5%. ^{d} The maximum decomposition temperature of the polymer.
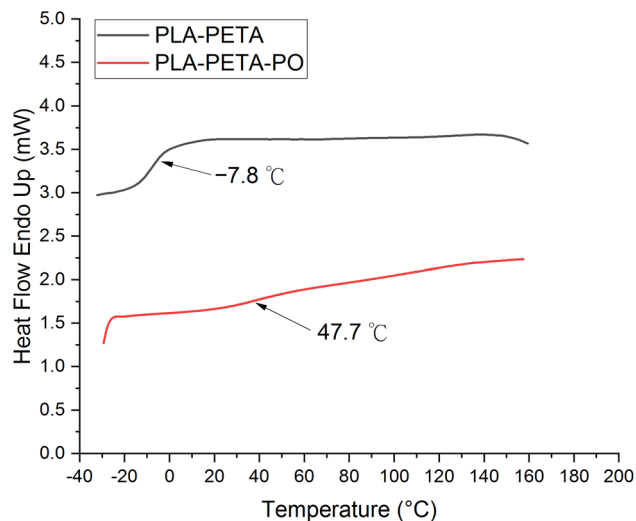
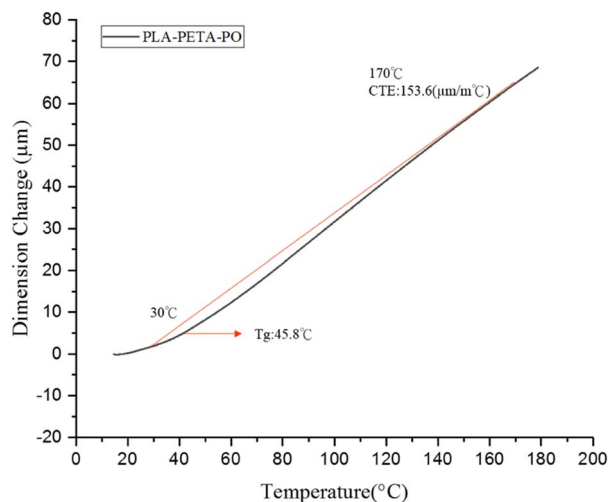Fig. 6 DSC thermograms of (A) non-crosslinked PLA−PETA and (B) crosslinked PLA−PETA−PO.
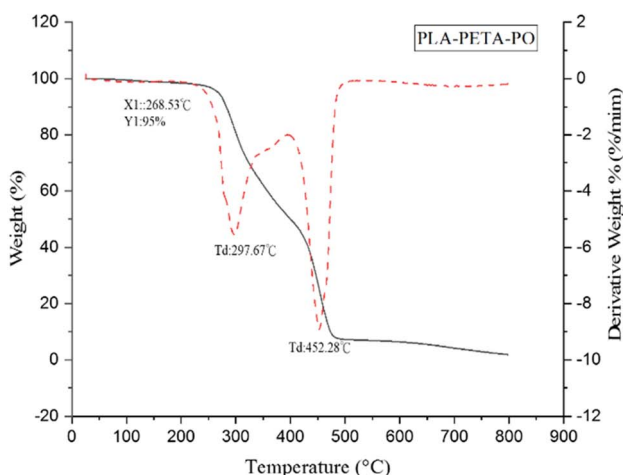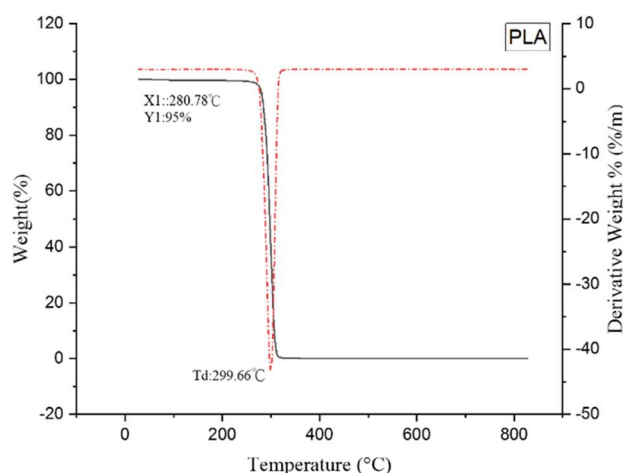

Fig. 8 TMA thermograms of PLA−PETA−PO.

However, the PLA chains that did not react could not form a crosslinked structure so were soft and easier to move. Moreover, the free volume in the network structure was higher than that in the unmodified PLA. Therefore, the CTE increased by 52.6–33.6 (μm m^{-1} $^\circ$C^{-1}). The T_g point on the line tangent to the turning point of the thermal expansion curve was 45.8 °C, which was relatively close to that measured *via* DSC.

3.5 Hydrolytic degradation and cytotoxicity testing

The hydrolysis of PLA is an autocatalytic reaction. In the presence of water, the ester bond segment of PLA is hydrolytically broken to generate oligomers. The carboxyl terminus of the oligomers can catalyze the cleavage of the ester bond to release H$^+$ ions. The ester bond is then broken into a carboxylic acid and alcohol *via* the chemical hydrolysis of H$^+$ ions.[38] According to Fig. 9, the PLA−PETA−PO sample lost

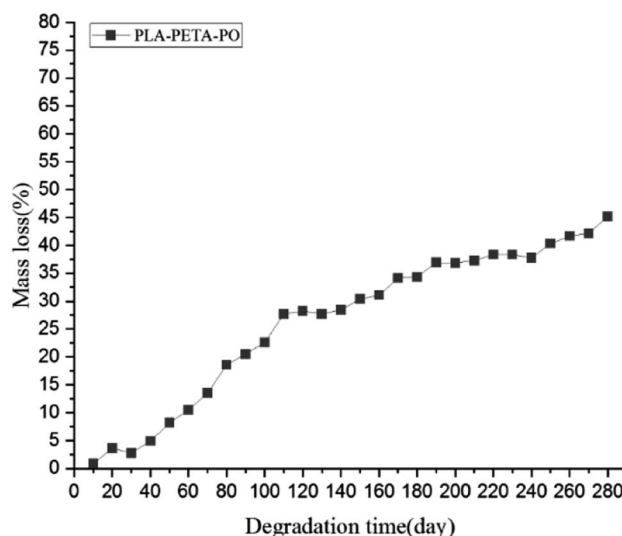Fig. 7 TGA thermograms of PLA and PLA−PETA−PO.

Fig. 9 Mass loss of crosslinked PLA−PETA−PO with degradation time.

Fig. 10 Physical changes in PLA–PETA–PO resulting from hydrolytic degradation after 280 days.
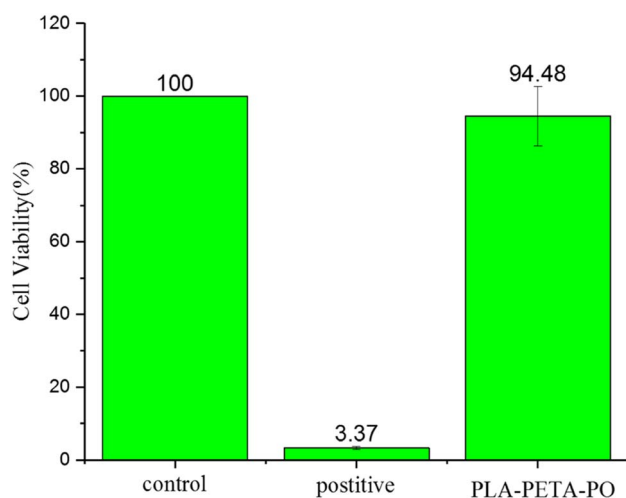

Fig. 11 Bar graph of PLA–PETA–PO cytotoxicity testing, where statistical significance was analyzed using Tukey's test.

4. Conclusions

Photocrosslinking by UV or visible light has been extensively explored in the biomedical field.[24,25,40] In this study, we modified PLA functional groups with L-lactide and PETA *via* one-pot ring-opening polymerization. NMR and HMBC spectroscopy confirmed that photocrosslinked, low-molecular-weight PLA–PETA polymers with a network structure are obtainable *via* one-step synthesis.

After the UV photocrosslinking of PLA–PETA, the degree of gelation rose to 91%; however, the thermal cracking temperature did not increase because the PLA chains did not form a high crosslinking density network in the acrylate chains to increase the heat resistance. After crosslinking, the T_g increased by 56.9 °C owing to an increase in crosslinking density, and the thermal expansion coefficient increased by 52.6–33.6 (μm m^{-1} °C^{-1}).

A degradation experiment was performed at a pH of 7.4 in PBS buffer, with the results showing that it takes 280 days to degrade 45% of the photocrosslinked PLA–PETA. Given that the crosslinking of PLA–PETA molecular chains affected the degradability of the material, it is necessary to further investigate these influencing factors in the future. However, cytotoxicity tests proved that the photocrosslinked polymer showed good biocompatibility.

Conflicts of interest

The authors declare that there is no conflict of interest regarding the publication of this article.

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26–28% of its weight after 120 days of hydrolysis, indicating that the PLA chain segment in the crosslinked product was hydrolyzed by water infiltration and diffusion, which then produced water-soluble oligomer products. After 120 days, the rate of weight loss decreased, leaving only the long broken chain segment of the PLA and the PLA chain segment that was coiled within the crosslinked polymer molecular cluster. After 280 days of continuous degradation, the weight loss was 45% (Fig. 10).

The decomposition products of the polymers and the residues of unreactive groups and crosslinking agents all exhibited good biocompatibility, *i.e.*, they exhibited no toxicity toward cells and had almost no impact on the phenotypic characteristics of cells.[39] According to the ISO 10993-5 specification *in vitro* cytotoxicity testing, the average cell viability of cells treated with PLA–PETA after crosslinking was >90% (Fig. 11), which shows that the polymer exhibits good biocompatibility and can be used in the field of biomedical materials.

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