

Article

A Novel Treatment: Effects of Nano-Sized and Micro-Sized Al₂O₃ on Steel Surface for the Shear Strength of Epoxy–Steel Single-Lap Joints

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Abstract: Traditional steel surface treatment (e.g., sand blasting, or silane treatment) was regarded as an effective method to improve the bonding strength of steel–epoxy single-lap joints. In the present study, a new steel surface treatment method was developed. With this method, the steel surfaces were treated with suspensions of nano-sized and micro-sized Al₂O₃ particles in ethanol/water mixture using the dip-coating method. Both Al₂O₃ particle sizes were previously treated or not treated with silane. Single-lap shear tests of the steel–epoxy bonds were conducted to compare the effects of the treating methods. According to the testing results, the highest increase in the bonding strength (by 51.8%) was found for the steel coated with the suspension of silane treated nano-Al₂O₃ particles. Scanning electron microscopy (SEM) analysis and energy dispersive spectrometer (EDS) analysis indicates that the nano-Al₂O₃ particles were clearly attached to the treated steel surfaces. Moreover, the steel surface with the silane-treated nano-Al₂O₃ particles was found to clearly enhance the contact angle between the steel and epoxy resin. The fracture morphology analysis of the single-lap shear testing specimen shows that the bonding between the steel and adhesive changed from steel–epoxy interfacial failure to cohesive failure when the steel surfaces were treated with the nano-Al₂O₃ particles suspension. The developed steel surface treatment method with the suspension of nano-particles proves to be effective and reliable in enhancing the bonding strength of the steel-to-epoxy adhesives.

Keywords: adhesion; interface mechanical properties; nanostructures; shear strength



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1. Introduction

Adhesive-bonded joints provide many advantages compared to traditional joints such as welding, bolting, and riveting. However, these joints suffer from a poor performance of the steel–epoxy interface [1–6]. One of the most potential treatments is adding nano-materials to the weak layer. Recent studies showed that different kinds of nano-particles can be coated onto the steel surface, such as fullerene, carbon nanotubes (CNTs), and graphene [7–10]. In this study, nano/micro aluminum trioxide (Al₂O₃) particles were both used in the experiments. Different contents of nano/micro particles were compared. For the reinforcement effect to be guaranteed, the main challenge regarding the incorporation of nanomaterials is the ability to achieve their uniform dispersion by disaggregation of the micrometric agglomerates formed as a consequence of attractive van de Waals interactions among the particles [11–15].

Some researchers have revealed that the enhancement efficiency of the covalent bond is more than five times than the Van der Waals force interface. In fact, the force between nano-materials and substrate was the Van der Waals force, resulting in the cracking of composites' interface and lower properties [3]. Silane was regarded as a kind of effective

chemical bond in the solution, and it can provide a strong covalent bond in the interfacial area. Although silane has been widely used in the engineering field, the degree of coverage and functionalization of surfaces upon silanization depends on the chemical functionality of the surface and silane. An optimum concentration of silane in the solution exists and yields an adequate molecular coverage on the surface of the functionalized filler. The optimum concentration is known to be in the range of 0.03~5 wt.%.

In the application area, the main problems are that silane-modified nano-particles can easily agglomerate in the solution. In particular, different sizes of particles will seriously determine the effect of the application. For example, in the field of interface bonding between steel and adhesives, nanoparticles act as the links between the steel/resin. The well-dispersed particles can be designed as the reinforced particles in the interfacial area. The size of the nanoparticles is the key factor affecting the dispersion and mechanical properties [16–18]. Al_2O_3 particles were regarded as the particle with the most potential in the nano-material application area. The sol-gel method was regarded as an efficient method to modify the particles in the solution [19–21]. Physical and chemical methods are the common methods used for nanoparticles. For smaller particles, chemical dispersants are more effective. Polyethyleneimine (PEI) and polystyrene sulphonic acid (PSS) are two strong acids that provide an effective function for the dispersion of Al_2O_3 particles [22].

The interface of the composite is one of the key area influencing the performance of the joints. İsmail Saraç et al. investigated the effects of Al_2O_3 , TiO_2 , and nano- SiO_2 of nano-particles in resin to enhance the properties of the single-lap joint. The results showed that the shear strength was improved up to 97% in 4 wt.% nano- Al_2O_3 reinforced specimens [23]. E. Marin et al. analyzed the intrinsic corrosion resistance of a $\text{TiO}_2/\text{Al}_2\text{O}_3$ atomic layer on a grinded steel surface. The polarization curves showed that atomic layer coatings improved the corrosion resistance of stainless steel [14]. İclal Avinc Akpınar et al. obtained the failure mode of adhesively bonded joints by using Graphene-COOH, Carbon Nanotube-COOH, and Fullerene C60 as the nanostructure in resin. 0.25%, 0.5%, 1%, 2%, and 3% nano-particles were examined in the experiments, and the results showed that 1% was the best nanostructure reinforcement ratio [24].

In this study, a suspension of silanized nano/micro-sized Al_2O_3 particles in a water/ethanol mixture was developed as the surface treatment for steel surfaces to improve the bonding performance between steel and epoxy adhesive. SEM, EDS, and contact angle tests were conducted to analyze the morphology and properties of the steel surface after treatment. The interfacial bonding properties were determined by the single-lap joint test method. The study aims to provide an effective nano-particle-treated method on steel surfaces to enhance the bonding performances of steel-epoxy joints in the civil engineering field.

2. Experimental

2.1. Raw Materials

Q235 steel plate was used in the experiment. The size of the steel plate is 102 mm × 26 mm × 2 mm. Room temperature curable epoxy resin (bisphenol-A, (DGEBA)) and curing agent with a weight ratio of 100:31 was provided by Shandong Dagong Composite Materials Co. Ltd., Linyi, China. Nano $\alpha\text{-Al}_2\text{O}_3$ (the diameter is about 30 nm) and micron $\alpha\text{-Al}_2\text{O}_3$ (the diameter is about 1 μm) was bought from Maoguo nano-technology company, Shanghai, China. The KH550 silane coupling agent (γ -amino-propyltriethoxysilane) was used (molecular formula is $\text{H}_2\text{NCH}_2\text{CH}_2\text{CH}_2\text{Si}(\text{OC}_2\text{H}_5)_3$).

2.2. Steel Surface Treatment

In the present study, steel was treated with grinding treatment, sand blasting, silane, nano- Al_2O_3 coating, and micron- Al_2O_3 coating. Each treatment was done as follows (Table 1).

Table 1. Preparation procedures of different samples.

	Grind	Sand Blast	Silane	Nano/Micron Al ₂ O ₃ -Silane	Nano/Micron Al ₂ O ₃
Grind	✓				
Sand blast	✓	✓			
Silane	✓	✓	✓		
Al ₂ O ₃ -silane	✓	✓		✓	
Al ₂ O ₃	✓	✓			✓

Grinding treatment: One end of steel plate (26 mm × 12.7 mm) was cross-grounded by sandpaper (120 mesh). The samples were considered as control sample and designated as “C”.

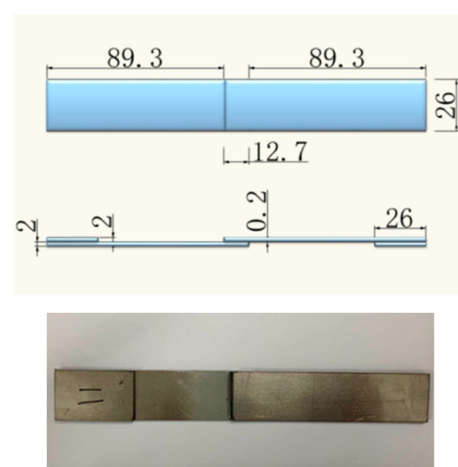
Sand blasting treatment: After grinding treatment, steel plates were sandblasted using a sand blasting machine. The nozzle of the spray gun is 10 cm from the steel plates in 30 s; the particle of sand is about 100 μm, which was designated as “S”.

Silane treatment: 2 wt.% silane agent was mixed into ethanol–water solution (weight ratio of 9:1) and magnetically stirred for 20 min at room temperature, and the solution was designated as “silane solution”. “S” sample was immersed by silane solution, and then the steel was dried in an oven at 110 °C for 1 h; the steel was designated as “Si”.

Al₂O₃ treatment: For the steel coated with Al₂O₃ suspensions, the first step was to prepare appropriate Al₂O₃ suspensions. Nano and micron Al₂O₃ of 1 wt.%, 2 wt.%, 3 wt.%, and 4 wt.% were added into the 9:1 mixture of ethanol and water, respectively. Nano and micron Al₂O₃ of 1 wt.%, 2 wt.%, 3 wt.%, and 4 wt.% were slowly added into the prepared silane coupling solution mentioned above at room temperature and magnetically stirred for 10 min, respectively. Next, the Al₂O₃ suspension was sonicated for 30 min at room temperature. The Al₂O₃ suspensions were then dip-coated onto the steel surface.

2.3. Preparation of Single Lap Joints

The prepared steel plates end (26 mm × 12.7 mm) were coated with epoxy resin. A prepared steel plate and a resin covered steel plate were overlapped. After that, the overlapping region was pressed by a weight of 400 g, as shown in Figure 1. A standard curing cycle involved being at room temperature for 1 day, followed by a post curing process at 60 °C for another day. The average thickness of the cured resin between steels was consistent and within the range of 0.15–0.25 mm.

**Figure 1.** Size of single-lap samples.

2.4. Scanning Electron Microscopy (SEM) Analysis

The morphology of the steel surface was characterized by SEM, and the test voltage was 20 kV. Then, the steel was coated with a layer of gold on the surface by the precision etching and plating instrument (Gatan682, American Gatan Company, Pleasanton, CA, USA).

2.5. FTIR Analysis

The covalent bond of solution was tested by FTIR analysis (Spectrum 100, Perkin Elmer Instruments, Waltham, MA, USA) from wavenumbers of 4000 cm^{-1} to 400 cm^{-1} .

2.6. Contact Angle Tests

The contact angle machine was provided by Daheng (Group) Co., Ltd., Beijing Image Vision Technology Branch, Beijing, China. The size of sample was $40 \times 26 \times t\text{ mm}^3$. The samples were tested by the contact angle machine. For each group, five points were tested in order to obtain a reliable and repeatable result.

3. Results and Discussion

3.1. Al_2O_3 Suspension of Tests

Nano Al_2O_3 suspensions were well dispersed in alcohol in 2mins, as shown in Figure 2a. After 6 h, the nano solution remained in a well-dispersed state (Figure 2b). This kind of phenomenon can be attributed to the size of the particles and the dispersion processes, which mean that the nano- Al_2O_3 can easily disperse in the ethanol solution.

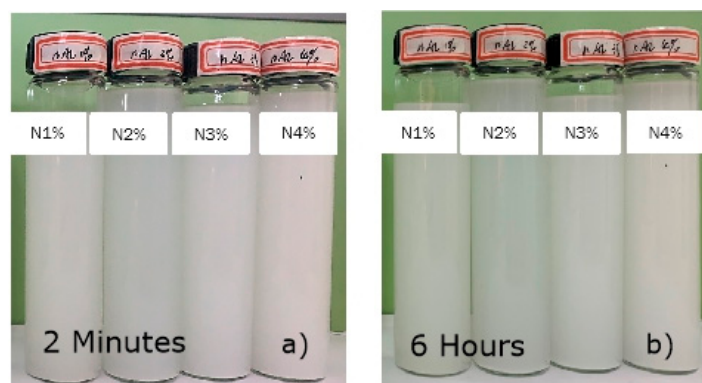


Figure 2. Photos of micro/nano-particles suspension solution in (a) 2 min. (b) 6 h.

The size of particles plays an important part in the solution. Because the colloid is in constant motion, the molecules around it gain kinetic energy and can counteract gravity without settling. Because the specific surface area of the nano-particle is large, the particle in the solution is more susceptible to the electrostatic action of the surrounding liquid on the particle, thus, the dispersion occurs [25].

The dispersion process of nano-particles can be divided into four steps. The first step is incorporation, then infiltration, disintegration of particle groups, and agglomeration of dispersed particles. Nano-particles tend to diffuse in the ethanol solution, just like the solvent in a solution with a disordered state. The smaller the size of the nano-particles, the stronger the motion pattern will be. When the nano-particles are dispersed in the solution for a period of time, the particles in the solution easily reach equilibrium, and after several hours, the effect of agglomeration and dispersion will reach a balance. According to the DLVO theory, there are mainly two kinds of forces between colloid; one is inter-molecular force, which is mainly manifested as mutual attraction between colloid, and the other is electrostatic force, which is mainly manifested as mutual repulsion between colloid [26].

In the solution, these two forces work together, and when the attractive force is greater than the repulsive force, the solution will aggregate and deposit. When the repulsive force

is greater than the attractive force, the colloid particles in the solution will maintain a higher energy barrier under the action of the repulsive force. When the Van der Waals's force and inter-molecular force keep in equilibrium, the energy barrier in the solution is zero. Then, the solution system is extremely unstable. Polymerization occurs when nanoparticles collide [27].

The sol–gel system is a kind of metastable system. The higher the energy barrier is, the longer the stable time is. When the height of the energy barrier is zero, the system is extremely unstable and the particles meeting will aggregate and deposit [28].

In the process of ultrasonic, the nano-particles absorb energy. Then, the nano-particles are influenced by Brownian motion, but the system is very unstable. As time goes by, the number of nanoparticles influenced by Brownian motion is reduced and slows the movement of the particles. In order to reduce the high surface energy, particles will reunite through molecular inter-atomic forces and deposit together [9].

The diameter of the nano-particles was about 100 nano-meters; the mixture is called the colloid, with a relatively stable state. Brownian motion is an important reason for the stable dispersion of colloids. In the solution, the colloidal particles do not stay in a fixed position, so the colloidal particles do not deposit due to gravity. In colloid, colloidal particles agglomerate by colliding with each other. The particle dispersion system is a multi-phase dispersion system, and the surface free energy of the dispersion system increases when the size of particles decreases.

After a certain period of time, the colloidal particles will collide with each other and reunite. The colloidal particles will reunite from small to large, and this reunion will reach a state of equilibrium due to Brownian motion. Gravity cannot be ignored for particles with a large mass or of a particle size, such as micron particles on the micron scale. After a period of time, in the ethanol solution, the effect of gravity is greater than the dispersion of particles in the solution, so deposition occurs. These two kinds of nano-particles are both used for the application of steel treatment.

FTIR was conducted to test the covalent bond between Al_2O_3 and silane in solution. According to the FTIR analysis of the two kinds of solutions (Figure 3), large amounts of hydroxyl groups can be detected in the solution at about 3300 cm^{-1} . This is due to the abundance of hydrolyzed silane molecules in the solution. Nano- Al_2O_3 formed the covalent bonding in the mixed solution. The covalent bond is located near the peak position of 878 cm^{-1} . At about 1040 cm^{-1} , the Al_2O_3 formed the covalent bonding with silane (Table 2).

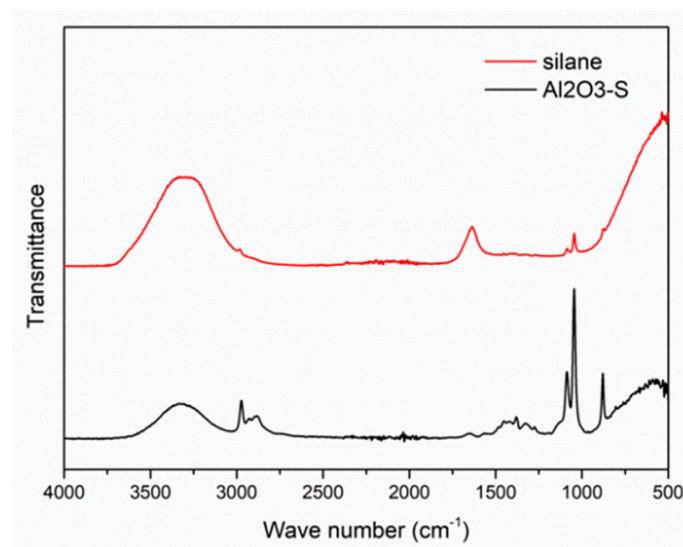


Figure 3. FTIR spectrum of silane solution and silanized Al_2O_3 suspension.

Table 2. FTIR analysis of silane and silanized Al₂O₃ solution.

Different Solution		Reference Range (cm ⁻¹)	Bond Characteristic
Silane	Silanized-CNT		
3300	3300	3570~3050	OH stretching vibration
-	2965	2960~2875	CH stretching vibration
1636	-	1650~1560	-NH ₂
-	1100	1150~1040	C-OH
1048	1040	1100~1000	R-O-Si
	878	955~830	Si-OH

3.2. Steel Surface Treatment

3.2.1. Surface Topography of Steel by Different Treatments

After the procedure of sand blasting, the surface roughness of steel was enhanced. The morphology of the steel surface is shown in Figure 4a. However, different sizes of particles on the steel surface showed a variety of properties. As shown in Figure 4b–e, micro Al₂O₃ particles were coated on the steel surface, but due to its larger size, the particles cannot form stable nano/micro structures on the steel surface [29]. Thin silane molecules film could form strong covalent bonding between steel/particles. Nano-Al₂O₃ particles grew on the steel surface, which could form a good film structure [1].

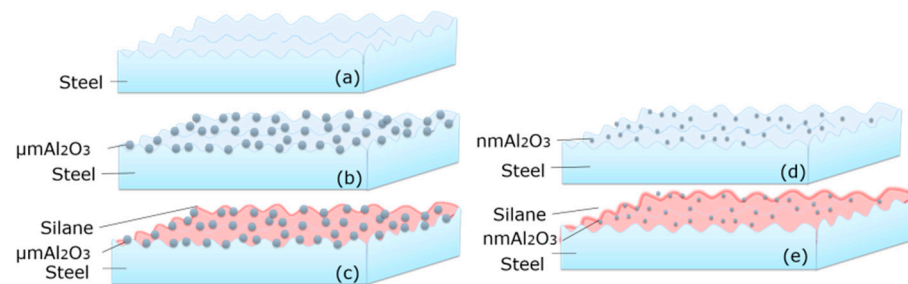


Figure 4. Schematic diagrams of steel surface morphology after (a) sand blasting; (b) Micro Al₂O₃; (c) nano Al₂O₃; (d) micro-silanized Al₂O₃; and (e) nano-silanized Al₂O₃.

As shown in Figure 5a, there was an obvious gully on the grinding steel surface. After sand blasting (Figure 5b), the steel surface roughness was highly improved and the regular gully disappeared. In Figure 6a–f, 1–3% silanized Al₂O₃ was shown in SEM photos. Compared with 1 wt.% Al₂O₃, in 3 wt.% nano-Al₂O₃ was an obvious reunion. Differently sized particles form different morphologies on the steel surface.

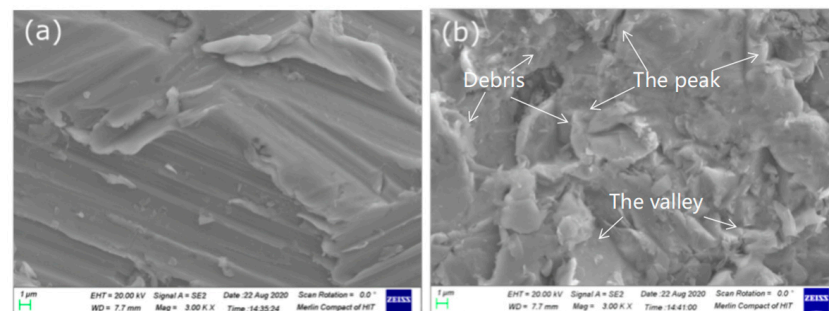


Figure 5. SEM photos of steel surfaces after (a) grinding and (b) sand blasting.

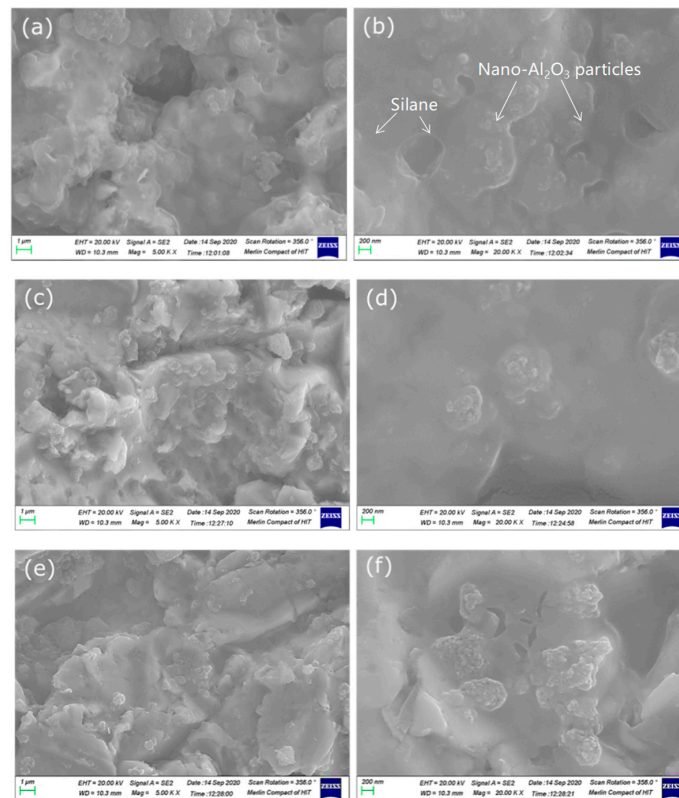


Figure 6. SEM photos of 1 wt.% (a,b), 2% (c,d), and 3% (e,f) silanized Al₂O₃ on steel surface.

The uniform dispersion of nanoparticles in the resin was clearly observed. In Figure 6a,b, it is obvious that the amount of nano-Al₂O₃ was lower than that in Figure 6c,d.

The EDS photos (Figure 7) show that silanized Al₂O₃ particles were observed in elemental analysis. In addition to the obvious aluminum element, silicon can also be clearly observed in EDS analysis. It shows that the silanized nanoparticles were adsorbed on the surface of steel. It is further proved that the treatment of Al₂O₃ particles on the surface of steel is effective.

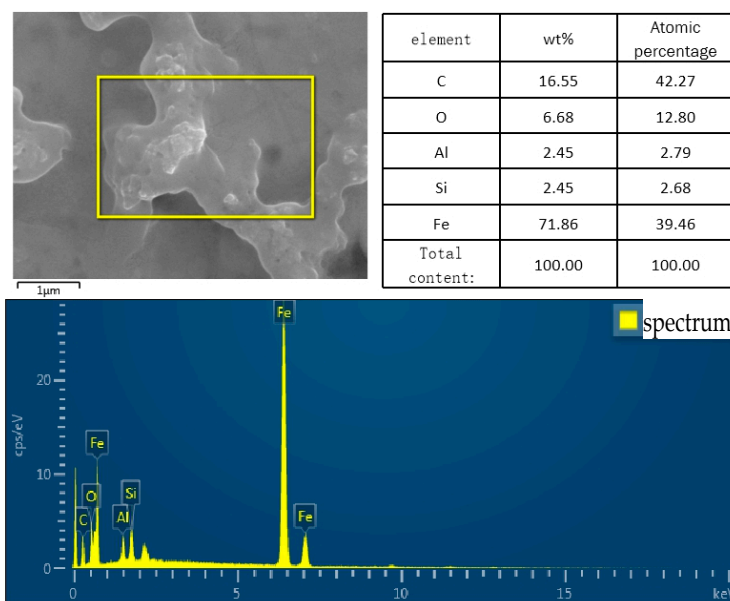


Figure 7. EDS photos of Al₂O₃ particles of steel surface.

3.2.2. Contact Angle Tests

On steel surface, contact angle is related to the roughness of the steel surface and the chemical composition. The greater the roughness is, the better the wettability of the water molecules at the interface is. Chemical treatment also affects the wettability of the liquid on the steel surface to some extent. This is because when different chemical components are grafted on the surface of steel, different functional groups and polarity of water molecules will produce different forces, which will lead to the formation of water molecules of different angles at the interface of micro droplets (in Figure 8).

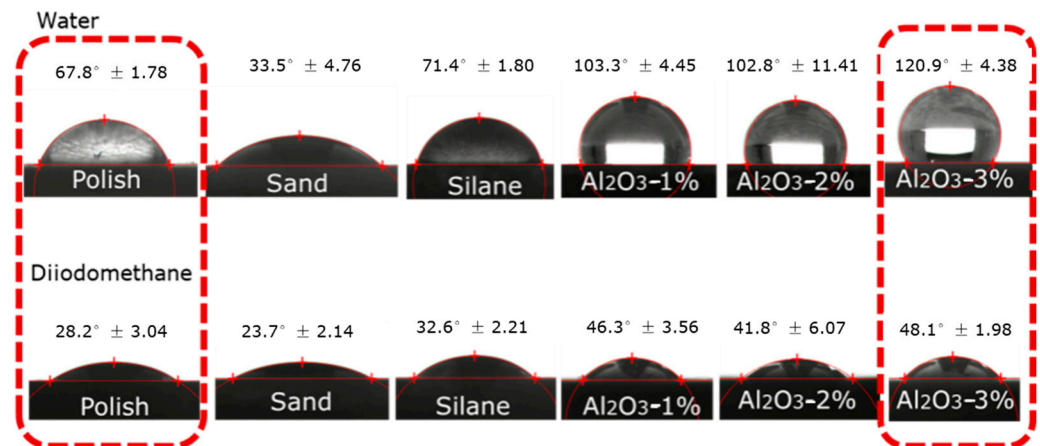


Figure 8. Photos of contact angle tests.

For grinded steel, the contact angle between water and steel was only $67.8^\circ \pm 1.78$, whereas the contact angle of diiodomethane and steel was $28.2^\circ \pm 3.04$. The nano-Al₂O₃ particle-treated steel surface was $120.9^\circ \pm 4.38$, whereas the contact angle of diiodomethane and steel was $48.1^\circ \pm 1.98$. The results of the contact angle tests show that the hydrophobic properties were improved after the nano-grafting procedure. The micro-structure was formed on the steel surface by the sand blasting procedure and the gully of steel was only several micrometers. Moreover, after the nano-grafting procedure, the steel surface was covered by nano-particles. The hydrophobic properties are due to the micro/nano-structure of steel.

In the SEM photos (Figure 9a,b), it can be observed that the bottom surface of the triangular structure formed by sandblasting and the length of the peak is $12\ \mu\text{m}$. During sandblasting, the arrangement of the surface of the matrix is regular; the smallest circulating elements can be simplified into the structure shown in Figure 10b. Light green represents the lower surface of the droplets of distilled water.

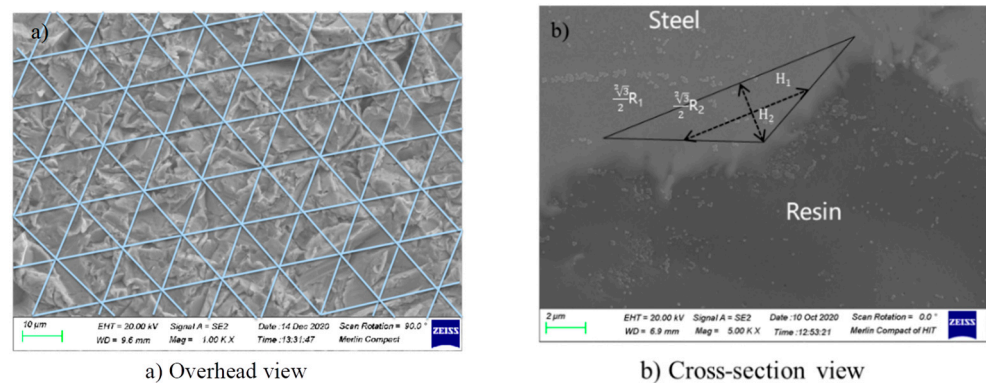


Figure 9. Triangular mesh division of matrix surface after sandblasting (a): overhead view and (b) cross-cutting perspective.

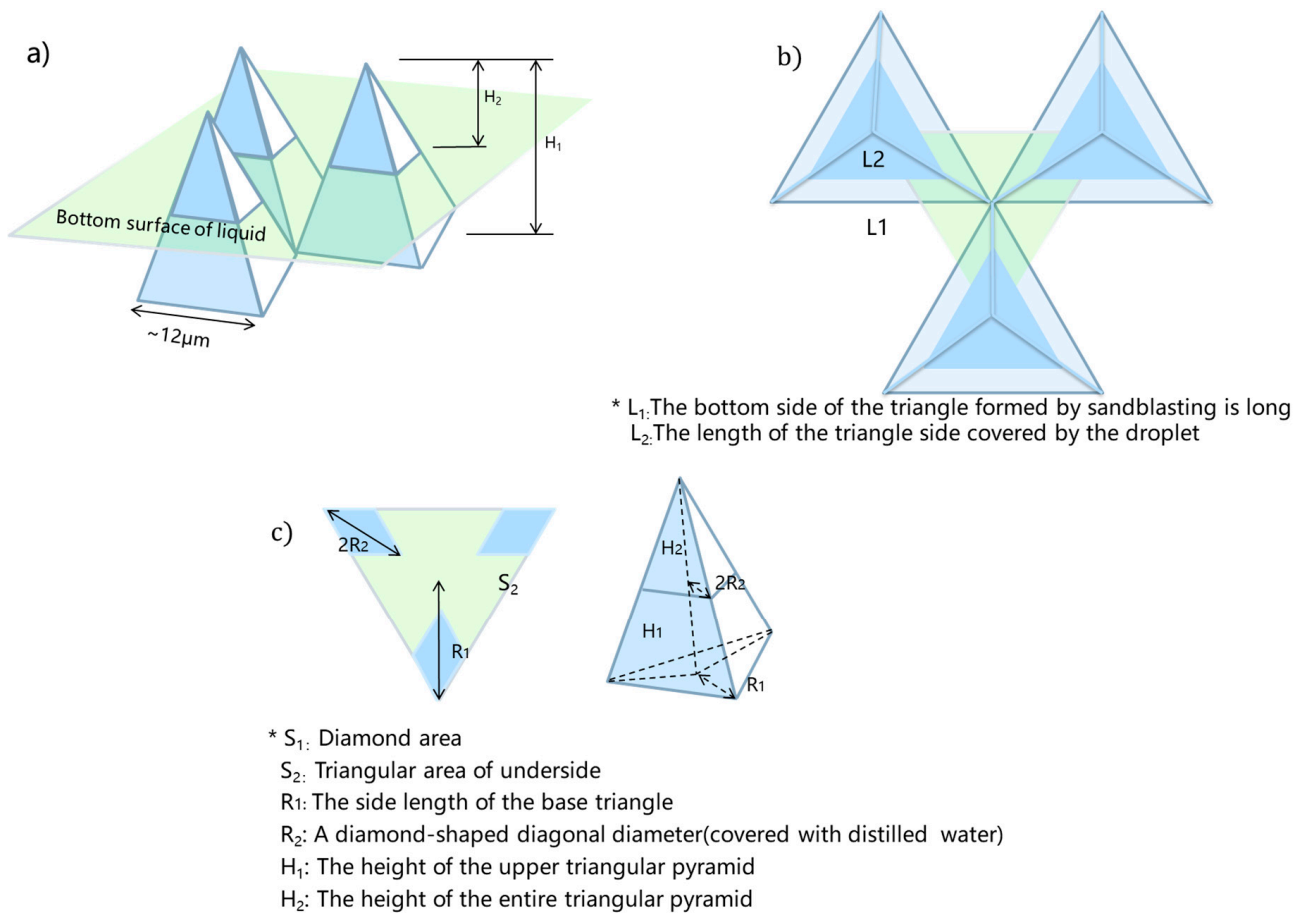


Figure 10. Triangular pyramid model formed by sandblasting: (a) Three-dimensional view of the steel surface formed by sandblasting; (b) overhead view of steel surface; and (c) a triangular pyramid formed by sandblasting (* refers to the parameter of peaks and valley of steel after sandblasting).

$$\cos\theta_r = f_1\cos\theta_1 + f_2\cos\theta_2 \tag{1}$$

According to Equation (1), θ_r is the intrinsic contact angle between a solid and a liquid, and θ_1 is the solid-liquid contact angle. The grinded samples followed Young’s equation and $\theta_r = 67.80 \pm 1.7^\circ$, $\theta_1 = 120.9 \pm 4.38^\circ$ and $f_1 + f_2 = 1$. According to Equation (2), f_1 can be obtained as 35.36%. The Cassie–Baxter model and the triangular size formed by sandblasting can infer that the droplet depth of the distilled water covering the steel substrate is 1.46 μm .

$$f_1 = \frac{\cos\theta_r + 1}{\cos\theta_1 + 1} \tag{2}$$

In Figure 11, Young’s equation is applicable to the solid–gas, solid–liquid, liquid–gas three-phase interface, namely, the ideal solid surface with a smooth surface and uniform composition, and the tension between the contact angles accords with the equation. However, the equation only applies to the ideal smooth surface; to investigate the actual production of the roughness of the surface, Wenzel et al. carried out thorough research and point out that when the roughness is more, the solid and liquid in the actual contact area is greater than the ideal smooth surface, formed in the substrate surface hydrophobic or hydrophilic surface; thus, the Wenzel equation was proposed.

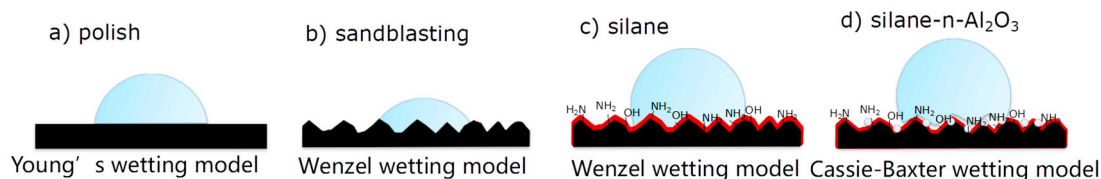


Figure 11. The matrix surface formed by different treatment methods and related models (a) polish followed Young's wetting model (b) sandblasting followed Wenzel wetting model (c) silane followed Wenzel wetting model (d) silane-n- Al_2O_3 followed Cassie-Baxter wetting model.

When the composition of a solid surface becomes more complex, the Wenzel equation is no longer applicable. On this basis, Cassie et al. proposed the air rough cushion model. At the same time, in the actual micro–nano structure surface, sometimes the surface is metastable because it is still in an unstable state.

In other words, the liquid does not completely infiltrate into the microstructure on the solid surface, and there is a certain gap between the solid–liquid materials, which makes the three phases form a new composite model. The Cassie–Baxter state refers to the model in which the liquid droplets do not completely immerse into the microstructure, and there is a small amount of air residual, which is also the micro/nano structure model applicable to this paper.

As an important indicator to measure the hydrophobic performance of solid surfaces, the contact angle of the steel surface after grinding was $67.8^\circ \pm 1.78$ in this paper. As the substrate surface was relatively flat, the contact area between the steel surface and the substrate is close to the actual contact area when water droplets drop on the substrate surface, and Young's model was adopted.

After the sandblasting treatment, the contact angle between the sample and the substrate surface was reduced to $33.5^\circ \pm 4.76$. After spraying, silanized nano-coating is coated on the surface of the sample, and an air transition layer can be added between the silanized nano-coating and the droplet, so that the droplet will not make full contact with the steel matrix, thus improving the hydrophobic performance of the steel surface to a large extent. For example, the steel matrix treated by 2 wt.% Al_2O_3 nano-particle suspension.

The contact angle with water droplets was increased to $102.7^\circ \pm 1.89$. If the cause of a solid surface roughness microstructure size in the submicrometer or nanometer order of magnitude, and the solid material and liquid contact angle value is above 90° , at this point, the microstructure in the air will not squeeze out a liquid but will be surrounded by a liquid and trapped within the microstructure, and will fill the microstructure airspace. This creates a composite surface where air and solid surfaces intersect. The presence of air will further improve the value of the apparent contact angle. It can be said that any solid surface with an apparent contact angle greater than 120° with water has such a microstructure, which is a composite surface formed by air and solid materials, such as the contact angle of $120.9^\circ \pm 4.38$ after 3 wt.% Al_2O_3 particle treatment.

Silane coupling agent molecules with active groups on the surface of the steel adsorbed the substrate, so as to form a relatively dense structure on the side that is close to the substrate of the silane coupling agent molecular film, and in the process of adsorption, the formation of the structure became looser, and the formation of the silane coupling agent film leads to the relatively uniform dispersion of nanoparticles.

Due to the polymerization characteristics of the silane coupling agent itself, covalent bonds are easily formed between the active hydroxyl groups, whereas the unreacted amino group is exposed on the outside. Because the polarity of the amino group is weaker than that of the hydroxyl group, the contact angle formed between the amino group and the water drop is significantly higher than that of the steel matrix with only the active hydroxyl group in Figure 12.

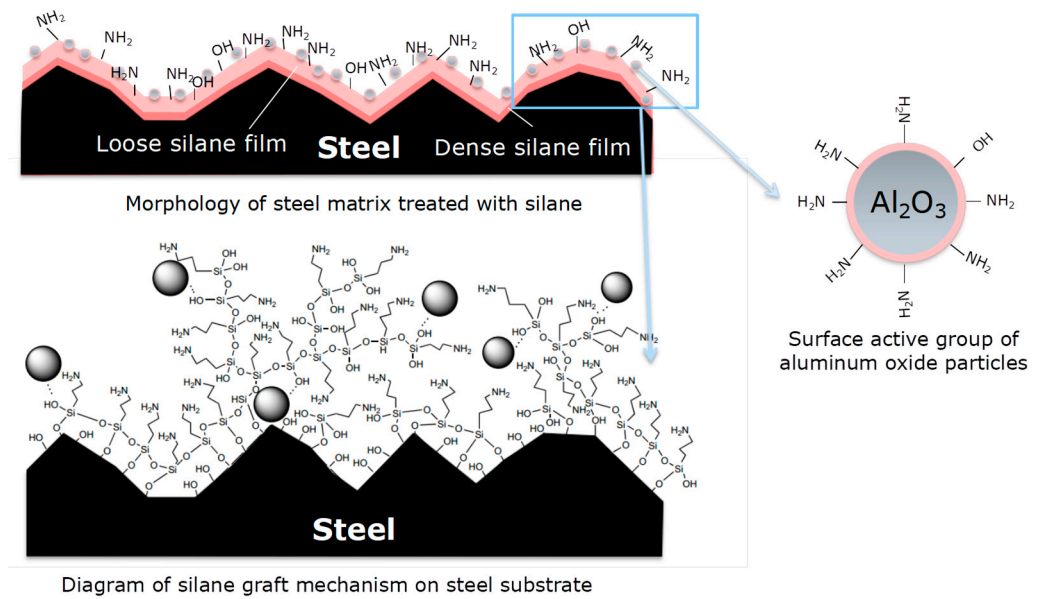


Figure 12. Wettability angle models of steel substrate surface with different treatments.

Using the XRD analysis, it can be concluded that, compared with the sandblasting treatment in Figure 13, the presence of α - Al_2O_3 can be obviously detected on the substrate surface of sandblasting treatment and surface grafted Al_2O_3 film at about 27° . All of the three samples shows peaks of Fe on steel surface at about 44° , 65° and 83° .

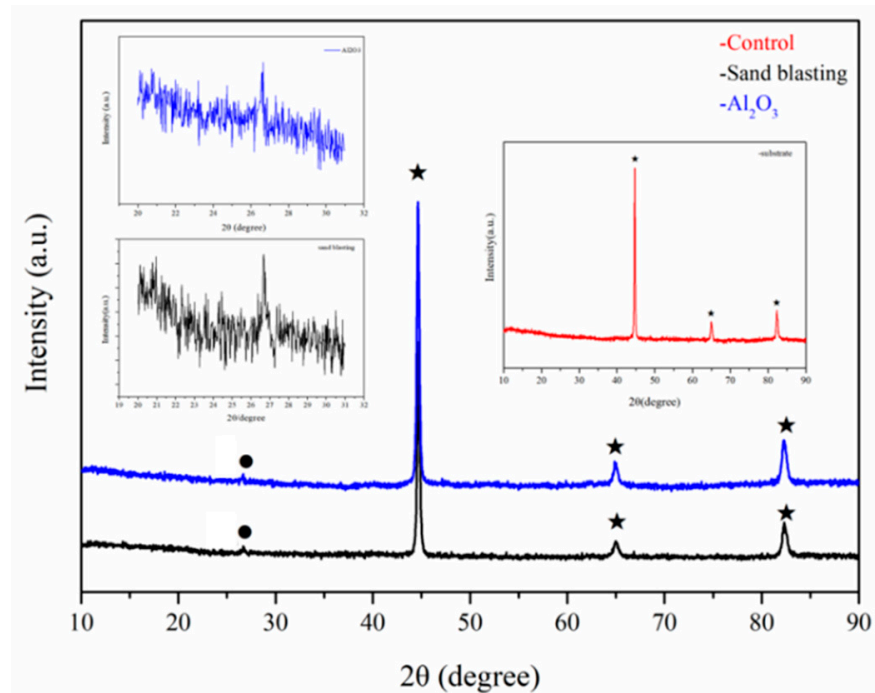


Figure 13. XRD analysis of steel substrate surface. (● refer to Al; ★ refer to Fe).

In Figure 14, after soaking in 60°C distilled water, floating rust occurs on the substrate surface of the sample after grinding, whereas only slight modifications occurs on the substrate treated with silanized nano- Al_2O_3 film. This corrosion extends from the edge of the sample to the center, indicating the good integrity of the silane film.

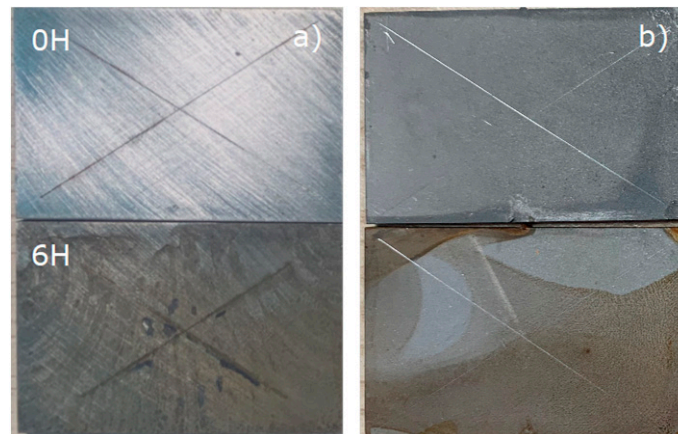


Figure 14. Durability test of steel surface with different treatments: (a) control; and (b) $n\text{-Al}_2\text{O}_3$.

3.3. Interfacial Tests between Steel and Resin

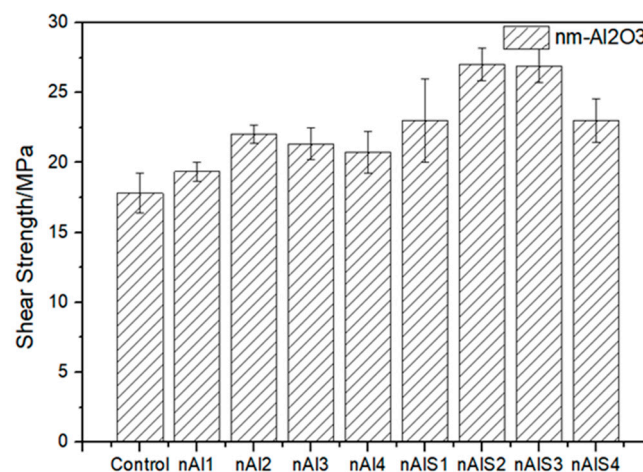
Single-Lap Joint Tests

The single-lap test is a commonly used mechanical method to test the interfacial properties of steel/resin. The interface bonding properties of steel surface and resin treated by different methods can be tested through single-lap joint tests.

In Figure 14, the overall distribution of strain of the sample is symmetric along the width of the sample and anti-symmetric along the axial direction of the sample. However, due to the uneven force of the sample in the loading process, the strain value is slightly asymmetrical.

During the measurement of interfacial shear force in the single-lap sample, it is obvious that the strain value on the surface of the sample is at its maximum at about 5 cm from the edge of the lap wire. The maximum strain value can reach 0.0615 for the control group sample and 0.00515 for the sample treated with aluminum oxide particles.

The single-lap specimens showed that the single shear strength of the specimens gradually increases by 16.5% after the treatment of the steel surface by nano- Al_2O_3 , and the interfacial bonding properties of the specimens can be further improved by 52.8% by silanized nano-particles treatment (Figure 15a). The detail are shown in Table 3.



(a)

Figure 15. Cont.

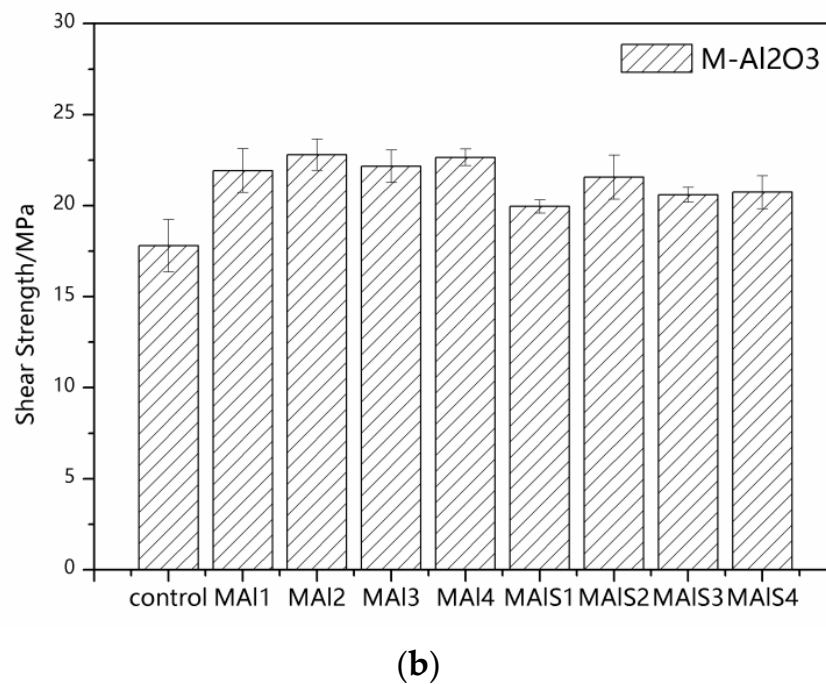


Figure 15. Effect of different contents of (a) nano-Al₂O₃ particles and (b) micro-Al₂O₃ particles on the interface properties of steel/resin.

Table 3. Effect of different content of nano-Al₂O₃ particles.

Sample	Control	nAl1	nAl2	nAl3	nAl4	nAIS1	nAIS2	nAIS3	nAIS4
Shear strength	17.8 ± 1.43	19.35 ± 0.66	22.04 ± 0.65	21.33 ± 1.13	20.74 ± 1.47	23.01 ± 2.96	27.03 ± 1.18	26.9 ± 1.21	22.99 ± 1.55

The micro-particles were also tested by universal tensile testing machine. The single lap specimens shown that the single shear strength of the specimens gradually increases by 27.2% after the treatment of the steel surface by micro-particles, and the interfacial bonding properties of the specimens can be further improved by 16.5% by silanized micro-particles treatment [11] (Figure 15b). The detail are shown in Table 4.

Table 4. Effect of different contents of micro-Al₂O₃ particles.

Sample	MAI1	MAI2	MAI3	MAI4	MAIS1	MAIS2	MAIS3	MAIS4	MAI1
Shear strength	21.92 ± 1.2	22.8 ± 0.86	22.16 ± 0.9	22.65 ± 0.45	19.95 ± 0.36	21.56 ± 1.21	20.59 ± 0.41	20.74 ± 0.9	21.92 ± 1.2

DIC morphology of the single-lap sample is shown in Figure 16. The green area is the test cloud area, and the line shown in the figure is the test line. There were 200 spaced points on the line to be tested (Figure 16). White paint and black paint speckle were sprayed on the surface of the sample by VIC-3D technology. Then, the deformation of speckle on the surface of the sample under different load levels was observed by camera and LED technology, so as to analyze and compare the influence of different treatment methods on the mechanical properties of the single-lap samples.

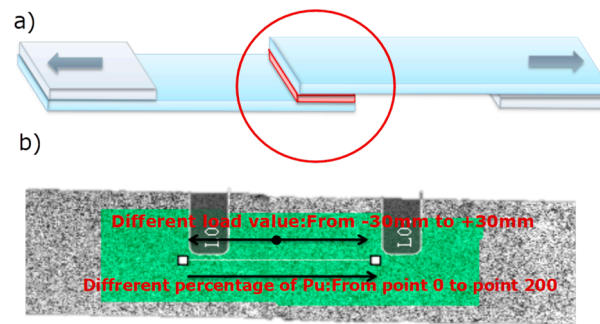


Figure 16. Single-lap sample and speckle on sample surface (a) graphic of single-lap joints; (b) illustration of single lap joints for DIC.

Figure 17 shows the 2D picture of the control sample and 2 wt.% Al_2O_3 sample. In Figure 18 (based on Figure 16), when the load level is 90% Pu, the strain value of the control sample is 0.004; when the load level is 100% Pu, the strain value is 0.014, which means that when the load level of the sample increases by 10%, the strain value increases by 3.5 times. As shown in Figure 19 (based on Figure 16), the maximum load value of the control group sample is 6 KN, and the maximum load value of the nano-treated sample is 8 KN.

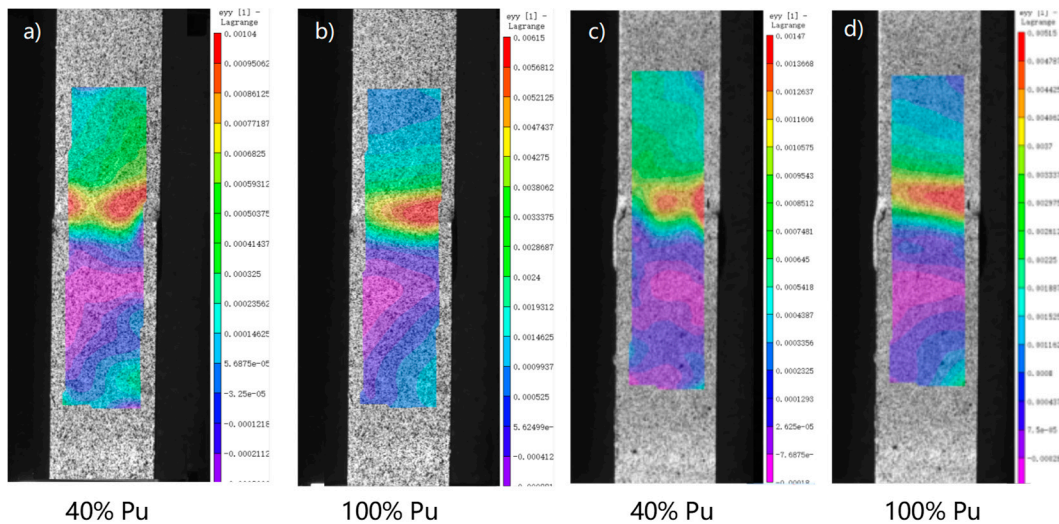


Figure 17. Mechanical properties of single-lap joints with different treatment methods by DIC analysis procedure: Control: (a) 40% Pu; (b) 100% Pu; 2 wt.% Al_2O_3 : (c) 40% Pu; and (d) 100% Pu.

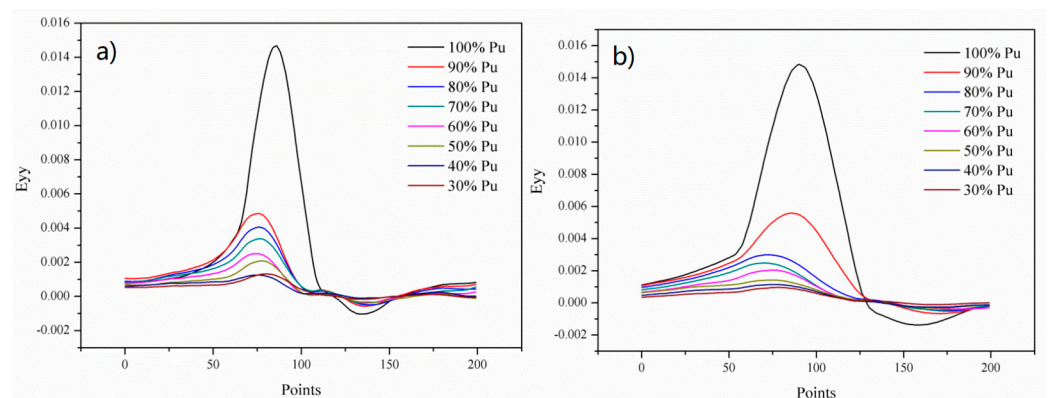


Figure 18. Mechanical properties of single-lap joints with different treatment methods by DIC analysis procedure. (a) Control; and (b) 2 wt.% Al_2O_3 .

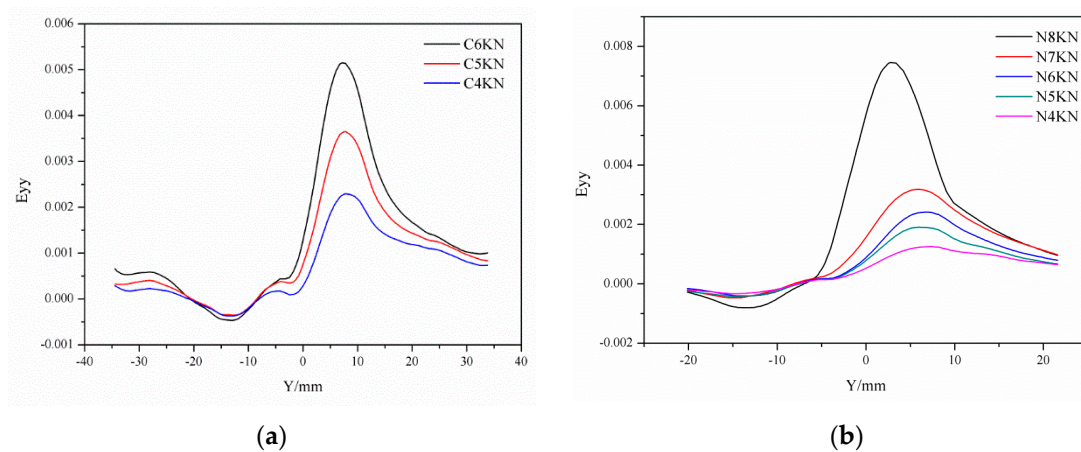


Figure 19. Mechanical properties of single-lap joints with different treatment methods by DIC analysis procedure: (a) Control; and (b) 2 wt.% Al_2O_3 .

Adding a certain amount of nano-particles into the resin can effectively improve the bonding property of the steel–resin interface. But in the resin system and steel used in this paper, the increase of nano-particle content does not significantly affect the interface properties (from 1–4%); it has little effect on the bonding property of the steel–resin interface [12].

In Figure 20a, the epoxy could directly bond with steel using nano- Al_2O_3 . In Figure 20b, the sketch of epoxy resin and steel bonding performance shows that silane could form a thin film between the steel and resin [12,13].

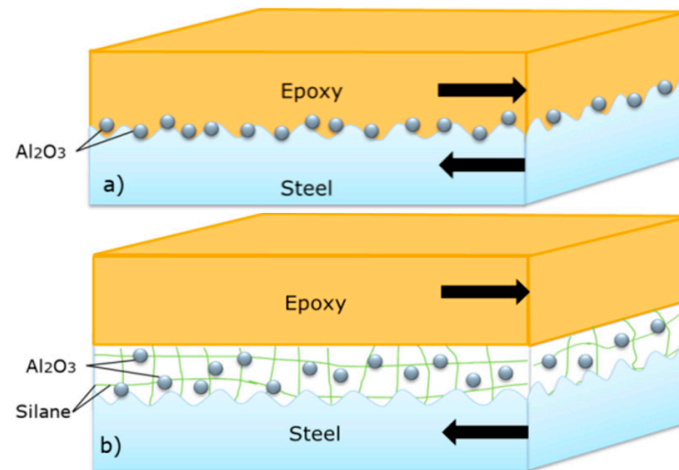


Figure 20. Sketch of epoxy resin and steel surface bonding: (a) Nano- Al_2O_3 particles on steel surface; and (b) silanized Al_2O_3 particles on steel surface.

In the optical photos shown in Figure 21, the fracture was made relatively smooth by grinding. After tensile treatment, the fracture surface of the steel treated with Al_2O_3 particles shows a different morphology. The resin morphology of the single-lap fracture can be observed by optical microscope, and it further confirmed that the failure mode of the interface changed after different treatment methods. The fracture morphology of the sample surface can be clearly observed by optical microscope. After nano- Al_2O_3 treatment, the weak layer of the single-lap joint sample was transferred from the steel resin interface to the inside of the adhesive layer [9].

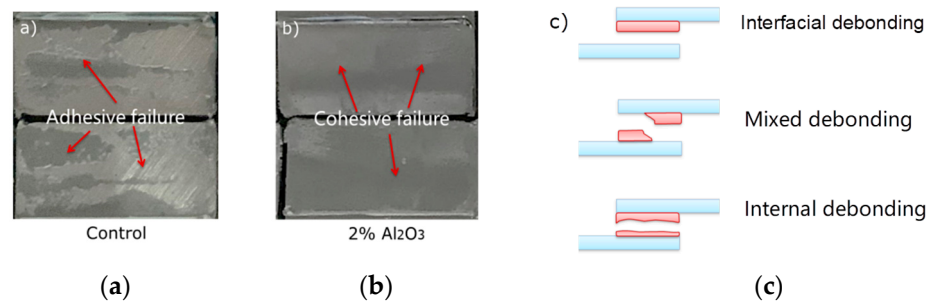


Figure 21. Digital photos of epoxy resin and steel surface: (a) Control sample; (b) 2% Al₂O₃; and (c) failure modes.

After the grind treatment, the fracture of the single-lap sample was relatively smooth, and the failure was mainly caused by the failure of the interface between the resin and steel. Among them, 2 wt.% nano-Al₂O₃ particles showed the most obvious effect on improving the interfacial bonding property, and obvious corrugated cracks can be observed at the fracture (in Figure 22). This phenomenon indicated that the fracture at the interface of the sample after nano-treatment was significantly improved. Adding nano-particles into resin can improve the bonding property of the interface to a certain extent, but the improvement effect is not as obvious as that of steel surface [14].

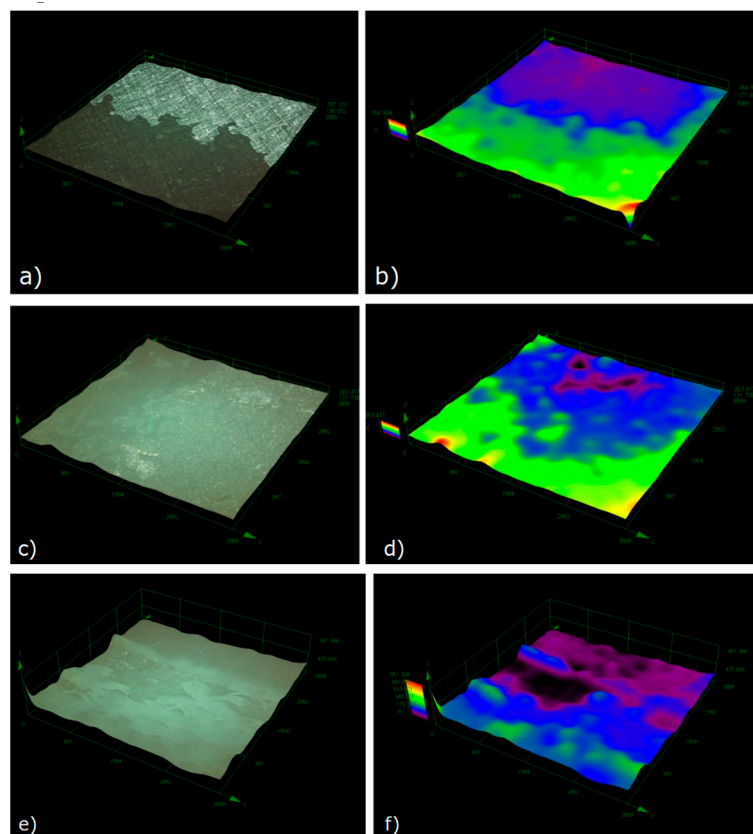


Figure 22. Optical micrographs and 3D analysis of single-lap cross-section fracture: (a,b) control sample; (c,d) 1 wt.% nAl₂O₃; and (e,f) 3 wt.% nAl₂O₃.

4. Conclusions

In the present paper, suspensions of nano/micro-Al₂O₃ particles in ethanol/water were used to treat the steel surface using the dip-coating method. By FTIR and SEM analysis, the silanized nano-Al₂O₃ can form a thin film by covalent bonding. The shear strength of nano-Al₂O₃-treated samples was improved by 51.8%. The shear strength of

steel/adhesive joints is influenced by steel surface conditions related to particle dispersion state, surface roughness, and chemical bonding.

1. A new steel surface treatment method was developed. The steel was treated with a suspension of nano- or micro- Al_2O_3 particles in a mixture of ethanol/water using the dip-coating process. The particles were previously treated, or not treated, with silane. Compared to the method in which the adhesive is blended with nano-particles, the present dip-coating method showed better results in the enhancement of the interfacial shear strength between the steel—epoxy adhesive.
2. Steel treated with the suspension including silane-treated nano- Al_2O_3 increases the bond strength between the steel—epoxy by 51.8%, much higher than those treated with the micro- Al_2O_3 . For the later case, the increase is about 28.1%. Nano- Al_2O_3 is more likely to form multi-scale microstructures on the treated steel surface.
3. Particles in suspension treated with a silane coupling agent was previously necessary to modify and functionalize the nano-particles and finally results in enhanced surface treatment effect. FTIR analysis shows that a silane coupling agent can form covalent bonds with Al_2O_3 particles. SEM analysis showed the silane coupling agent forming thin films with Al_2O_3 particles on the steel surface.

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References

1. Yuan, W.; Yang, T.; Yang, G.; Liu, S.; Du, Y.; Liu, C. Enhancing mechanical properties of adhesive laminates joints using ultrasonic vibration-assisted preprocessing. *Compos. Struct.* **2019**, *227*, 111325. [[CrossRef](#)]
2. Akpınar, S.; Ozel, A. Experimental and numerical determination of the thermal cycle performance of joints obtained with nanostructure-doped nanocomposite adhesives. *Compos. Part B Eng.* **2019**, *174*, 106959. [[CrossRef](#)]
3. Akpınar, I.A.; Gültekin, K.; Akpınar, S.; Akbulut, H.; Ozel, A. Experimental analysis on the single-lap joints bonded by a nanocomposite adhesives which obtained by adding nanostructures. *Compos. Part B Eng.* **2017**, *110*, 420–428. [[CrossRef](#)]
4. Budhe, S.; Banea, M.D.; de Barros, S.; da Silva, L.F.M. An updated review of adhesively bonded joints in composite materials. *Int. J. Adhes. Adhes.* **2017**, *72*, 30–42. [[CrossRef](#)]
5. van Dam, J.P.B.; Abrahams, S.T.; Yilmaz, A.; Gonzalez-Garcia, Y.; Terry, H.; Mol, J.M.C. Effect of surface roughness and chemistry on the adhesion and durability of a steel-epoxy adhesive interface. *Int. J. Adhes. Adhes.* **2020**, *96*, 102450. [[CrossRef](#)]
6. Zhou, H.; Liu, H.-Y.; Zhou, H.; Zhang, Y.; Gao, X.; Mai, Y.-W. On adhesive properties of nano-silica/epoxy bonded single-lap joints. *Mater. Des.* **2016**, *95*, 212–218. [[CrossRef](#)]
7. Avasthi, P.; Kumar, A.; Balakrishnan, V. Aligned CNT Forests on Stainless Steel Mesh for Flexible Supercapacitor Electrode with High Capacitance and Power Density. *ACS Appl. Nano Mater.* **2019**, *2*, 1484–1495. [[CrossRef](#)]
8. Sydlík, S.A.; Lee, J.-H.; Walsh, J.J.; Thomas, E.L.; Swager, T.M. Epoxy functionalized multi-walled carbon nanotubes for improved adhesives. *Carbon* **2013**, *59*, 109–120. [[CrossRef](#)]
9. Wan, Y.-J.; Gong, L.-X.; Tang, L.-C.; Wu, L.-B.; Jiang, J.-X. Mechanical properties of epoxy composites filled with silane-functionalized graphene oxide. *Compos. Part A Appl. Sci. Manuf.* **2014**, *64*, 79–89. [[CrossRef](#)]
10. Hussain, A.K.; Sudin, I.; Basheer, U.M.; Yusop, M.Z.M. A review on graphene-based polymer composite coatings for the corrosion protection of metals. *Corros. Rev.* **2019**, *37*, 343–363. [[CrossRef](#)]
11. Zhao, Z.; Chen, M.; You, C.; Li, W.; Tie, D.; Liu, H. Effect of $\alpha\text{-Al}_2\text{O}_3$ additive on the microstructure and properties of MAO coatings prepared on low carbon steel. *J. Mater. Res. Technol.* **2020**, *9*, 3875–3884. [[CrossRef](#)]

12. Zhang, W.; Au, P.-I.; Zhang, X.; Fan, H.; Sun, C.; Saunders, M.; Leong, Y.-K. Spherical α -Al₂O₃ suspensions layered sequentially with anionic and cationic polyelectrolytes: Chemistry, rheology and TEM images. *Powder Technol.* **2018**, *338*, 716–724. [[CrossRef](#)]
13. Wan, P.; Zhao, N.; Qi, F.; Zhang, B.; Xiong, H.; Yuan, H.; Liao, B.; Ouyang, X. Synthesis of PDA-BN@f-Al₂O₃ hybrid for nanocomposite epoxy coating with superior corrosion protective properties. *Prog. Org. Coat.* **2020**, *146*, 105713. [[CrossRef](#)]
14. Marin, E.; Lanzutti, A.; Lekka, M.; Guzman, L.; Ensinger, W.; Fedrizzi, L. Chemical and mechanical characterization of TiO₂/Al₂O₃ atomic layer depositions on AISI 316 L stainless steel. *Surf. Coat. Technol.* **2012**, *211*, 84–88. [[CrossRef](#)]
15. Wang, W.; Xian, G.; Li, H. Surface modification of ramie fibers with silanized CNTs through a simple spray-coating method. *Cellulose* **2019**, *26*, 8165–8178. [[CrossRef](#)]
16. Zheng, H.; Guo, M.; Shao, Y.; Wang, Y.; Liu, B.; Meng, G. Graphene oxide–poly(urea–formaldehyde) composites for corrosion protection of mild steel. *Corros. Sci.* **2018**, *139*, 1–12. [[CrossRef](#)]
17. Zheng, H.; Shao, Y.; Wang, Y.; Meng, G.; Liu, B. Reinforcing the corrosion protection property of epoxy coating by using graphene oxide–poly(urea–formaldehyde) composites. *Corros. Sci.* **2017**, *123*, 267–277. [[CrossRef](#)]
18. Ding, S.; Xiang, T.; Li, C.; Zheng, S.; Wang, J.; Zhang, M.; Dong, C.; Chan, W. Fabrication of self-cleaning super-hydrophobic nickel/graphene hybrid film with improved corrosion resistance on mild steel. *Mater. Des.* **2017**, *117*, 280–288. [[CrossRef](#)]
19. Avilés, F.; Sierra-Chi, C.A.; Nistal, A.; May-Pat, A.; Rubio, F.; Rubio, J. Influence of silane concentration on the silanization of multiwall carbon nanotubes. *Carbon* **2013**, *57*, 520–529. [[CrossRef](#)]
20. Parhizkar, N.; Ramezanzadeh, B.; Shahrabi, T. Corrosion protection and adhesion properties of the epoxy coating applied on the steel substrate pre-treated by a sol-gel based silane coating filled with amino and isocyanate silane functionalized graphene oxide nanosheets. *Appl. Surf. Sci.* **2018**, *439*, 45–59. [[CrossRef](#)]
21. Chang, J.-K.; Lin, C.-S.; Cheng, W.-J.; Lo, I.H.; Wang, W.-R. Oxidation resistant silane coating for hot-dip galvanized hot stamping steel. *Corros. Sci.* **2020**, *164*, 108307. [[CrossRef](#)]
22. Garcia-Ivars, J.; Alcaina-Miranda, M.-I.; Iborra-Clar, M.-I.; Mendoza-Roca, J.-A.; Pastor-Alcañiz, L. Enhancement in hydrophilicity of different polymer phase-inversion ultrafiltration membranes by introducing PEG/Al₂O₃ nanoparticles. *Sep. Purif. Technol.* **2014**, *128*, 45–57. [[CrossRef](#)]
23. Sarac, I.; Adin, H.; Temiz, S. Investigation of the effect of use of Nano-Al₂O₃, Nano-TiO₂ and Nano-SiO₂ powders on strength of single lap joints bonded with epoxy adhesive. *Compos. Part B Eng.* **2019**, *166*, 472–482. [[CrossRef](#)]
24. Akpınar, I.A.; Gültekin, K.; Akpınar, S.; Akbulut, H.; Ozel, A. Research on strength of nanocomposite adhesively bonded composite joints. *Compos. Part B Eng.* **2017**, *126*, 143–152. [[CrossRef](#)]
25. Peng, B.; Takai, C.; Razavi-khosroshahi, H.; Fuji, M. Effect of silane modification on CNTs/silica composites fabricated by a non-firing process to enhance interfacial property and dispersibility. *Adv. Powder Technol.* **2018**, *29*, 2091–2096. [[CrossRef](#)]
26. Adamczyk, Z.; Weronki, P. Application of the DLVO theory for particle deposition problems. *Adv. Colloid Interface Sci.* **1999**, *83*, 137–226. [[CrossRef](#)]
27. Hou, S.; Su, S.; Kasner, M.L.; Shah, P.; Patel, K.; Madarang, C.J. Formation of highly stable dispersions of silane-functionalized reduced graphene oxide. *Chem. Phys. Lett.* **2010**, *501*, 68–74. [[CrossRef](#)]
28. Orouji, S.M.; Naderi, R.; Mahdavian, M. Controlled oxidation of mild steel by potassium permanganate solution to enhance protective functioning of silane coatings. *Colloids Surf. A Physicochem. Eng. Asp.* **2020**, *603*, 125251. [[CrossRef](#)]
29. Zhang, L.; Ren, Y.; Luo, Q.; Ying, X.; Xu, H.; Xuan, J. A Novel Method to from Well-adhered γ -Al₂O₃ Coating in 316L Stainless Steel Microchannels. *Energy Procedia* **2015**, *75*, 2044–2048. [[CrossRef](#)]