

# Metal Exchange Boosts the CO<sub>2</sub> Selectivity of Metal Organic Frameworks Having Zn-Oxide Nodes

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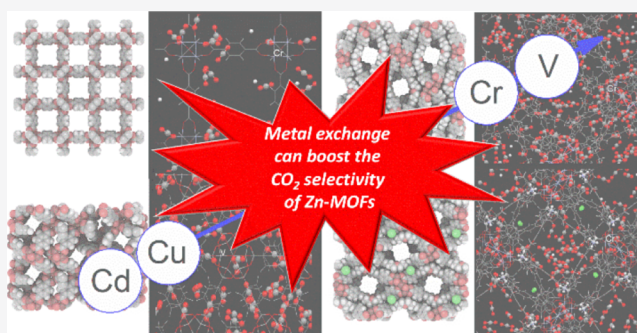
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**ABSTRACT:** A large number of metal organic frameworks (MOFs) synthesized to date have nodes with a Zn metal, and a detailed understanding of their gas separation efficiency upon metal exchange is needed to pave the way for designing the next generation of MOFs. In this work, we implemented a protocol to identify MOFs with Zn nodes out of 10,221 MOFs and classified them into two main groups. Depending on the pore properties and adsorption selectivities, two MOFs from IRMOFs and two MOFs from ZnO-MOFs were selected. The metal atom (Zn) of the selected four MOFs was exchanged with eight different metals (Cd, Co, Cr, Cu, Mn, Ni, Ti, and V), and 32 different metal-exchanged MOFs (M-MOFs) were obtained. By performing grand canonical Monte Carlo simulations, we investigated the influence of the metal type on the CO<sub>2</sub>/H<sub>2</sub> and CO<sub>2</sub>/CH<sub>4</sub> separation performances of these 32 M-MOFs. Physical properties of the MOFs such as the pore size and surface area, and chemical properties such as the partial charges of the atoms in the framework were investigated to understand the effect of metal exchange on the gas adsorption and separation performances of materials. Exchange of Zn with V and Cr led to a remarkable increase in the CO<sub>2</sub> uptakes of selected MOFs and these increases were reflected on the adsorption selectivity, working capacity, and the adsorbent performance score of MOFs. The exchange of Zn with V increased the selectivity of one of the MOFs from 119 to 355 and the adsorbent performance score from 70 to 444 mol/kg, while for another MOF, exchange of Zn with Cr increased the selectivity from 161 to 921 and the adsorbent performance score from 162 to 1233 mol/kg under the condition of vacuum swing adsorption. The molecular level insights we provided to explain the improvement in the gas separation performances of M-MOFs will serve as a guide to design materials with exceptional CO<sub>2</sub> separation performances.



## 1. INTRODUCTION

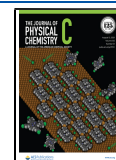
Metal organic frameworks (MOFs) are promising adsorbents with high CO<sub>2</sub> capture capacity due to their unique crystal structures and great degree of tunability.<sup>1,2</sup> MOFs can be easily modified to achieve high porosities,<sup>3</sup> record surface areas,<sup>4</sup> and a wide range of pore sizes and shapes.<sup>5</sup> Several experimental and computational studies have shown that MOFs outperform many traditional zeolites in CO<sub>2</sub> capture from the CO<sub>2</sub>/H<sub>2</sub> mixture.<sup>3,6,7</sup> To further harness the tremendous potential of MOFs, a recent direction has been changing the metal type in the structure to generate a new MOF. The strong influence of the metal choice on the CO<sub>2</sub> uptake has long been known in the field especially with the MOF-74 series,<sup>8</sup> as the CO<sub>2</sub> uptake of a Zn-DOBDC analog, Mg-DOBDC, was experimentally measured as 23.6 wt % at 0.1 bar, 296 K, almost four times higher than that of Zn-DOBDC and two times higher than those of Ni- and Co-DOBDC. Therefore, changing the metal type in a structure to develop a new MOF with an improved gas uptake is quite a useful approach to further harness the immense potential of MOFs. Various experimental techniques have been employed for metal substitution such as direct synthesis,<sup>9</sup> postsynthetic metal exchange,<sup>10</sup> and cluster metal-

ation through stepwise synthesis,<sup>11</sup> which allowed the investigation of metal-exchanged MOFs, M-MOFs, where M corresponds to the metal identity. Metal exchange techniques have been widely applied on commonly studied MOFs such as HKUST-1,<sup>12</sup> IRMOF-10,<sup>13</sup> and UiO-66,<sup>14</sup> and the impact of metal exchange on the adsorption of gases such as CO<sub>2</sub>, CH<sub>4</sub>, N<sub>2</sub>, and H<sub>2</sub> was investigated. In direct synthesis methods, it is not possible to foresee whether the new M-MOF will retain the original crystal structure and maintain its permanent porosity,<sup>15</sup> and in the postsynthetic metal exchange method, partial metal exchange may hinder the gas adsorption potential of MOFs.<sup>16</sup> Therefore, it is not experimentally viable to exercise all possible metal combinations in a MOF using a trial-and-error approach. Computational studies are very useful to

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alter the type of metal in a MOF, which allows in silico discovery of new MOFs with exceptional CO<sub>2</sub> adsorption and separation performances.

Cu–Cu paddlewheel MOFs are one of the most commonly examined M-MOFs in the literature.<sup>17</sup> H<sub>2</sub> uptake of M-BTC, where M = Co, Ni, Fe, and Zn, was experimentally compared with that of Cu-BTC and gravimetric uptakes were found to increase by 10, 58, 59, and 60% for Co, Ni, Fe, and Zn-BTC, respectively.<sup>18</sup> By postsynthetic partial metal substitution, O<sub>2</sub>/N<sub>2</sub> selectivities of Mn-, Fe-, and Co-exchanged Cu-BTC were experimentally found to be higher than the O<sub>2</sub>/N<sub>2</sub> selectivity of Cu-BTC in the order of Mn-BTC > Fe-BTC ≈ Co-BTC ≫ Cu-BTC at 77 K, which was supported with the difference in binding energies for O<sub>2</sub> and N<sub>2</sub> in M-MOFs calculated with density functional theory.<sup>19</sup> However the metal centers were not attractive as at cryogenic temperatures (77 K) for O<sub>2</sub> and N<sub>2</sub> molecules at ambient temperatures (273, 283, and 298 K). Isoreticular MOFs (IRMOFs) have been widely studied due to their high chemical stabilities and wide range of surface areas (500 to 5500 m<sup>2</sup>/g).<sup>20</sup> Partial Co<sup>2+</sup> ion exchange was experimentally studied on Zn-based IRMOF-1 and increased H<sub>2</sub>, CH<sub>4</sub>, and CO<sub>2</sub> uptakes of Co-IRMOF-1 with increasing Co<sup>2+</sup> concentration were reported.<sup>21</sup> Zn-oxide-based MOFs are investigated due to their low toxicity and high gas storage capacities. Zhou et al.<sup>22</sup> synthesized a Zn-BTC type MOF, which could not adsorb any CO<sub>2</sub> and N<sub>2</sub> due to pore shrinkage upon activation, whereas 96% metal ion exchange of Zn<sup>2+</sup> with Cu<sup>2+</sup> led to N<sub>2</sub> (CO<sub>2</sub>) adsorption capacity up to 325 (~75) cm<sup>3</sup>/g at 77 K (273 K).

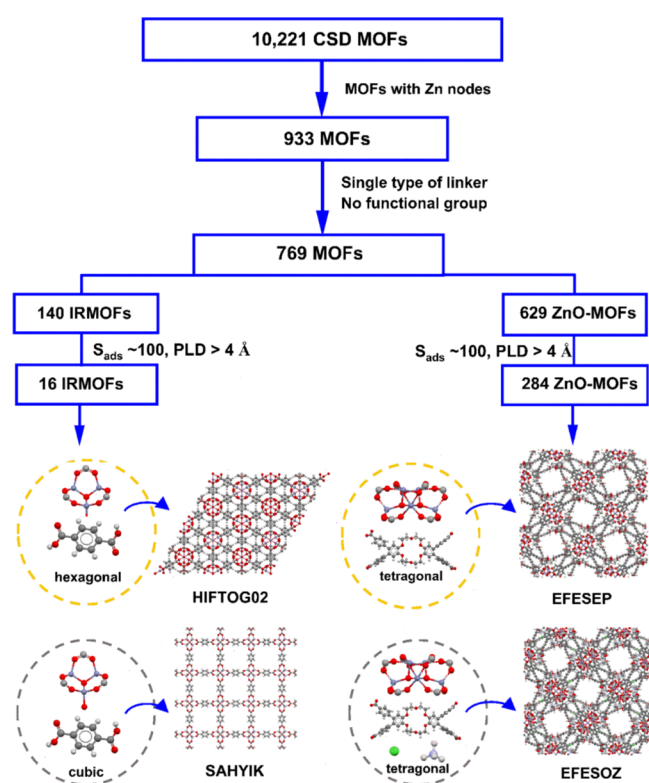
Although experimental measurements have been performed for commonly studied MOFs, our knowledge on the gas adsorption mechanisms of MOFs upon metal exchange is still limited. To understand the molecular origins of gas adsorption mechanisms on M-MOFs, molecular simulations have been performed but only for a few materials. Yuan et al.<sup>23</sup> performed grand canonical Monte Carlo (GCMC) simulations for adsorption and separation of an equimolar CO<sub>2</sub>/H<sub>2</sub> mixture on M-SIFSIX-3, where M = Cu, Zn, and Fe, and found that CO<sub>2</sub> molecules adsorb preferentially closer to the SIF<sup>2-</sup> anion, whereas H<sub>2</sub> molecules prefer to adsorb at the pore center. CO<sub>2</sub>/H<sub>2</sub> selectivities of Cu, Zn, and Fe-SIFSIX-3 were computed as 300, 400, and 500, respectively, at 10 bar, 298 K. Borycz et al.<sup>13</sup> investigated the change in the CO<sub>2</sub> affinity of Mg-, Ca-, Fe-, Cu-, Ge-, Sr-, Cd-, Sn-, and Ba-exchanged IRMOF-10 using *ab initio* calculations and molecular modeling. Due to the larger ionic radii of Ba<sup>2+</sup>, the metal center had stronger interactions with the CO<sub>2</sub> compared to other metals, leading to high binding energies, and three times more CO<sub>2</sub> molecules near the metal center compared to original IRMOF-10. Recently, our group performed GCMC simulations on M-HKUST-1 and M-HATGUF (M = Cd, Co, Cr, Cu, Fe, Mn, Mo, Ni, Ru, and Zn) and showed that Cr-HKUST-1 and Cd-HKUST-1 have 11 and 38% enhanced CO<sub>2</sub>/H<sub>2</sub> selectivities in addition to 27 and 60% enhanced adsorbent performance scores (APS) compared to the original Cu-HKUST-1.<sup>24</sup> As this literature review shows, studies focusing on the mixture gas adsorption and/or separation performance of M-MOFs are very scarce. Existing studies have focused only on commercial MOFs due to the fact that choosing the ideal MOF that will have a much-enhanced gas separation performance upon the metal exchange is challenging and the large diversity of MOF structures necessitates an initial screening of thousands of available materials.

Coordination geometry of the metal nodes, size and shape of the linkers, and topology of the framework play an important role in the extent of metal exchange.<sup>25</sup> Considering the fact that most of the synthesized MOFs to date have nodes with the Zn metal,<sup>26,27</sup> understanding their gas separation performance upon metal exchange is required to pave the way for achieving better performing materials. Therefore, in this work, we proposed a methodology to select the most appropriate material candidates for metal exchange among the MOFs having Zn-oxide nodes and investigated the influence of metal exchange on their CO<sub>2</sub>/H<sub>2</sub> and CO<sub>2</sub>/CH<sub>4</sub> mixture separation performances. We systematically categorized 10,221 MOFs according to their metal nodes, coordination of organic linkers, and the shape of the metal cluster. After this categorization, we focused on (i) two IRMOFs that have the same chemical building blocks, but the change of synthesis conditions results in MOFs with different crystal geometries (cubic vs hexagonal) due to the presence/absence of catenation, and (ii) two ZnO-MOFs with and without the guest ions incorporated to the MOF structure. Metal atoms (Zn) of these four MOFs were exchanged with M = Cd, Co, Cr, Cu, Mn, Ni, Ti, and V to examine 32 unique M-MOFs both for CO<sub>2</sub>/H<sub>2</sub> and CO<sub>2</sub>/CH<sub>4</sub> mixture separations. Gas separation performances of M-MOFs computed under pressure swing adsorption (PSA) and vacuum swing adsorption (VSA) conditions were then compared with those of the most recently reported IRMOFs and ZnO-MOFs. The relationship between the calculated performance metrics of *in-silico*-designed M-MOFs and their physical (pore size, surface area, and porosity) and chemical (partial charge of the metal atom) properties were examined. The investigation of gas–MOF and gas–gas interaction energies, as well as the preferential adsorption sites of gas molecules in the framework provided a molecular level understanding of the impact of metal exchange on gas separation performances of MOFs. These results will guide the future computational and experimental studies on M-MOFs to generate materials with superior CO<sub>2</sub>/H<sub>2</sub> and CO<sub>2</sub>/CH<sub>4</sub> separation performances.

## 2. COMPUTATIONAL METHODS

**2.1. Classification and Selection of MOFs for *In Silico* Metal Exchange.** We recently performed high-throughput computational screening of 10,221 MOFs retrieved from the Cambridge Structural Database (CSD)<sup>28</sup> for CO<sub>2</sub>/H<sub>2</sub> separation and showed that MOFs perform well as CO<sub>2</sub> adsorbents.<sup>29</sup> In this work, we first utilized CSD ConQuest software<sup>30</sup> and conducted a connectivity search on these 10,221 MOFs. The guidelines described in the literature<sup>27</sup> were followed for targeted classification of MOFs, and 933 MOFs with Zn-oxide nodes were identified. These 933 MOFs were further narrowed down to 769 MOFs after excluding the MOFs that have functional groups and/or more than one type of organic linker. Complete metal-exchange of MOFs severely depends on factors such as the pore size distribution and flexibility of the framework, framework stability, coordination geometry of the metal node, and steric hindrance caused by the linkers.<sup>25</sup> We aimed to work on relatively simple MOF structures and eliminated the MOFs consisting of mixed linkers and/or functional groups due to the following reasons: (i) MOFs designed with mixed linker strategies can have defects,<sup>31,32</sup> or differing topologies and crystal structures,<sup>33</sup> but we aimed to work on MOFs with similar topologies after metal exchange and with a defect-free, homogeneous adsorption surface. (ii) MOFs with functional groups can have strong

electrostatic interactions with CO<sub>2</sub> as shown in our previous works<sup>29,34</sup> but we aimed to isolate the influence of metal exchange on CO<sub>2</sub>/H<sub>2</sub> and CO<sub>2</sub>/CH<sub>4</sub> selectivity by omitting the MOFs with functional groups and mixed linkers. 769 MOFs with Zn-oxide nodes, a single type of linker, and no functional groups were divided into two final subgroups depending on the shape of the metal cluster (coordination of the node): 140 IRMOF-like MOFs (IRMOFs) and remaining (629) MOFs with Zn-oxide nodes (ZnO-MOFs). The main difference between IRMOF-like and ZnO-MOFs is the shape of the metal cluster.<sup>27</sup> MOFs with IRMOF-like structures have a 4-connected configuration with oxygen atoms connected to each metal site as shown in Figure S1a, while MOFs with ZnO-MOF type structures have two metal atoms connected with four oxygen and one Zn atom bonded with six oxygen as shown in Figure S1b. This classification methodology is shown in Figure 1 together with the representative MOFs selected for



**Figure 1.** Targeted MOF categorization according to the type of node and building blocks to select the representative MOFs for in silico metal exchange. Carbon, oxygen, hydrogen, zinc, chlorine, and nitrogen are represented with gray, red, white, purple, green, and light blue colors, respectively.

metal exchange from IRMOFs and ZnO-MOFs. Representative MOFs from IRMOFs and ZnO-MOFs, which satisfy the following two criteria, were selected for metal exchange: (i) MOFs with pore limiting diameter (PLD)  $> 4 \text{ \AA}$  to avoid pore sizes smaller than or very close to the size of the gas molecules after geometry optimization, (ii) MOFs with CO<sub>2</sub>/H<sub>2</sub> mixture selectivity ( $S_{ads}$ )  $\sim 100$  at 1 bar, because these materials can have outstanding separation performances upon an increase in their CO<sub>2</sub> uptakes with metal exchange. We also identified a peer-MOF for each selected MOF to compare the change in their gas separation performances upon the metal exchange. 16

IRMOFs and 284 ZnO-MOFs had pore sizes  $> 4 \text{ \AA}$  and  $S_{ads} \sim 100$ .

Among 16 IRMOFs, HIFTOG02 (PLD: 4.15  $\text{\AA}$ ,  $S_{ads}$ : 119) was selected because it has the same building blocks as SAHYIK (known as IRMOF-1 or MOF-5)<sup>35</sup> but arranged in a hexagonal crystal geometry unlike the cubic geometry of SAHYIK.<sup>36</sup> Although SAHYIK (PLD: 7.8  $\text{\AA}$ ) has a low  $S_{ads}$ , 9.5 at 1 bar, it was selected as the peer for HIFTOG02 to compare the influence of catenation on MOF performance upon the metal exchange. Among 629 ZnO-MOFs, EFESOP (SNU-200) with a PLD of 6.13  $\text{\AA}$  and a  $S_{ads}$  of 160 was selected with its peer EFESOP (PLD: 6.2  $\text{\AA}$ ,  $S_{ads}$ : 160), which has the same crystal structure as EFESOP but with the guest ions Cl<sup>-</sup> and NH<sub>4</sub><sup>+</sup>,<sup>37</sup> to investigate the influence of the guest ions on the gas separation performances of the metal exchanged ZnO-MOFs.

We note that the coordination geometry of the node, steric hindrance exposed by the linkers, pore size, and the stability of the framework are the preliminary factors that we considered before selecting the candidate MOFs for metal exchange.<sup>25</sup> The four MOFs we considered in our work for metal exchange (SAHYIK, HIFTOG02, EFESOP, and EFESOP) have been experimentally reported to be stable with mediocre pore sizes.<sup>36,37</sup> We considered pore sizes between 3.3 and 4  $\text{\AA}$  as small, 4 and 20  $\text{\AA}$  as mediocre, and 20 and 45  $\text{\AA}$  as large by following the convention used in our previous study.<sup>29</sup> We selected SAHYIK as a benchmark MOF because several previous works have both experimentally and computationally performed metal exchange on IRMOF series.<sup>15,38–40</sup> For example, Ti, V, Cr, Mn, Fe, and Ni variants of MOF-5 were experimentally reported via cation exchange.<sup>38,39</sup> HIFTOG02<sup>36</sup> is the catenated form of MOF-5, thus we assumed that a metal exchange procedure similar to that of MOF-5 can be easily implemented on this material. We selected SNU-200 variants, EFESOP and EFESOP, because SNU-200<sup>37</sup> has Zn-oxide nodes in frequently observed coordination geometries (octahedral and tetrahedral), and in a previous work,<sup>41</sup> exchange of Zn with Pd, Cd, Cu, Ni, Co, Mn, and Cr has been experimentally reported for a structure, which has a similar type of node geometry.

To examine the impact of metal type on the CO<sub>2</sub>/H<sub>2</sub> and CO<sub>2</sub>/CH<sub>4</sub> separation performances of SAHYIK, HIFTOG02, EFESOP, and EFESOP, we studied eight different metals, Cd, Cu, Co, Cr, Mn, Ni, Ti, and V, which are commonly employed in both computational and experimental metal exchange studies on MOFs,<sup>12,13,39</sup> and obtained 32 unique M-MOFs. Prior to metal exchange, bound and unbound solvent molecules were cleaned from structures and metal exchange and geometry optimization were performed with Materials Studio 2020.<sup>42</sup> Original metal (Zn) of the structures was modified using the “modify element” tool implemented in the Materials Studio, and energy of the structure was minimized with an ultrafine convergence tolerance of 10<sup>-5</sup> kcal/mol. A smart algorithm was employed for geometry optimization with a maximum iteration of 1000 steps. Convergence tolerances for geometry optimization were set as 10<sup>-6</sup>  $\text{\AA}$  for the distance, 10<sup>-5</sup> kcal/mol for energy, and 10<sup>-4</sup> kcal/mol/ $\text{\AA}$  for force. Unit cell parameters were allowed to relax during geometry optimization, which enabled the bonds between atoms of the MOF relax after the metal exchange step. Geometry optimized version of the original MOF is labeled as “opt” throughout the manuscript. More details of this method were described in our previous study.<sup>24</sup>

Structural properties of MOFs such as accessible surface area ( $S_{\text{acc}}$ ), the largest cavity diameter (LCD), PLD, and porosity ( $\phi$ ) were calculated using the Zeo++ software version 0.3.0.<sup>43</sup> To calculate the accessible surface area, we used a  $N_2$ -sized (3.70 Å) probe, whereas a probe size of zero was used for the porosity calculations. The PLD and LCD of the MOFs were calculated via percolating a probe through the simulation cell,<sup>43</sup> and PLD is defined as the size of the largest probe molecule that can pass the simulation cell freely at least through a channel in the MOF without overlapping any atoms (largest free sphere), while LCD is the largest included sphere in the unit cell.

**2.2. Calculation of Gas Uptakes and Separation Performances of MOFs.** Following the geometry optimization of M-MOF series, GCMC simulations were performed using RASPA simulation software<sup>44</sup> (version 2.0.36) to obtain  $CO_2/H_2$  and  $CO_2/CH_4$  mixture uptakes. Universal force field was used for the framework atoms.<sup>45</sup>  $CO_2$  was modeled as a three-site molecule with partial charges on each site,<sup>46</sup>  $H_2$  was modeled as a single sphere,<sup>47</sup> and  $CH_4$  was modeled using the TraPPE model as a unified sphere with van der Waals interactions.<sup>48</sup> Lorentz–Berthelot mixing rules were applied to define pairwise interactions between unlike atoms. Partial charges to framework atoms were assigned with the  $Q_{\text{eq}}$  (charge equilibration) method implemented in RASPA<sup>49,50</sup> because simulations performed with  $Q_{\text{eq}}$  charges reproduced the available experimental gas adsorption data for M-HKUST-1 (M = Cr, Co, Fe, Ru, and Ni Zn) as we previously reported.<sup>24</sup> We used the formal oxidation state for each metal as implemented in the charge equilibration module of the RASPA simulation code.<sup>44,50</sup> These correspond to a +2 oxidation state for Cd, Co, Cr, Cu, Mn, Ni, and Zn; and a +3 oxidation state for Ti and V. For GCMC simulations, 10,000 cycles for the initialization run and 10,000 cycles for the production run were used, and the average values of gas adsorption were taken from the production step. The isosteric heat of adsorption ( $Q_{\text{st}}$ ) values for gas molecules were computed during GCMC simulations using the fluctuation method in RASPA.<sup>44</sup> Gas adsorption snapshots were taken from the last simulation cycle to represent an image of the equilibrium state of adsorption.

To represent industrial precombustion gas mixture, the mole fractions of  $CO_2$  and  $H_2$  were set as 0.15 and 0.85, respectively,<sup>51</sup> while for the  $CO_2/CH_4$  mixture, an equimolar composition was considered representing the landfill gas.<sup>52</sup> Adsorption and desorption conditions were set as 10 (1) and 1 (0.1) bar to mimic the PSA (VSA) process, respectively, at 298 K. We examined adsorption-based gas separation performances of MOFs under PSA and VSA conditions using five different metrics, adsorption selectivity ( $S_{\text{ads}}$ ),  $CO_2$  working capacity ( $\Delta N_{CO_2}$ ), adsorbent regenerability ( $R$  %), and adsorbent performance score (APS).  $S_{\text{ads}}$  reflects the affinity of a material toward the more strongly adsorbed gas, which is  $CO_2$  for both separations.  $\Delta N_{CO_2}$  represents the amount of  $CO_2$  that can be captured throughout each adsorbent regeneration cycle. APS merges  $S_{\text{ads}}$  and  $\Delta N_{CO_2}$  to determine the best performing MOFs.  $R$  % is used to determine how much  $CO_2$  can be regenerated at each adsorbent regeneration cycle. Details for the calculation of these metrics are given in Table 1. For an efficient adsorption-based separation process, adsorbent materials offering high selectivity and high working capacity

**Table 1. Calculation of Adsorbent Performance Evaluation Metrics Used in This Work<sup>a</sup>**

metric	formula
mixture adsorption selectivity	$S_{\text{ads}}^{\text{mix}} = \frac{N_{\text{ads},CO_2} / y_{CO_2}}{N_{\text{ads},j} / y_j}$
working capacity	$\Delta N_{CO_2} = N_{\text{ads},CO_2} - N_{\text{des},CO_2}$
APS	$APS = S_{\text{ads}}^{\text{mix}} \times \Delta N_{CO_2}$
percent regenerability	$R \% = \Delta N_{CO_2} / N_{\text{ads},CO_2} \times 100\%$

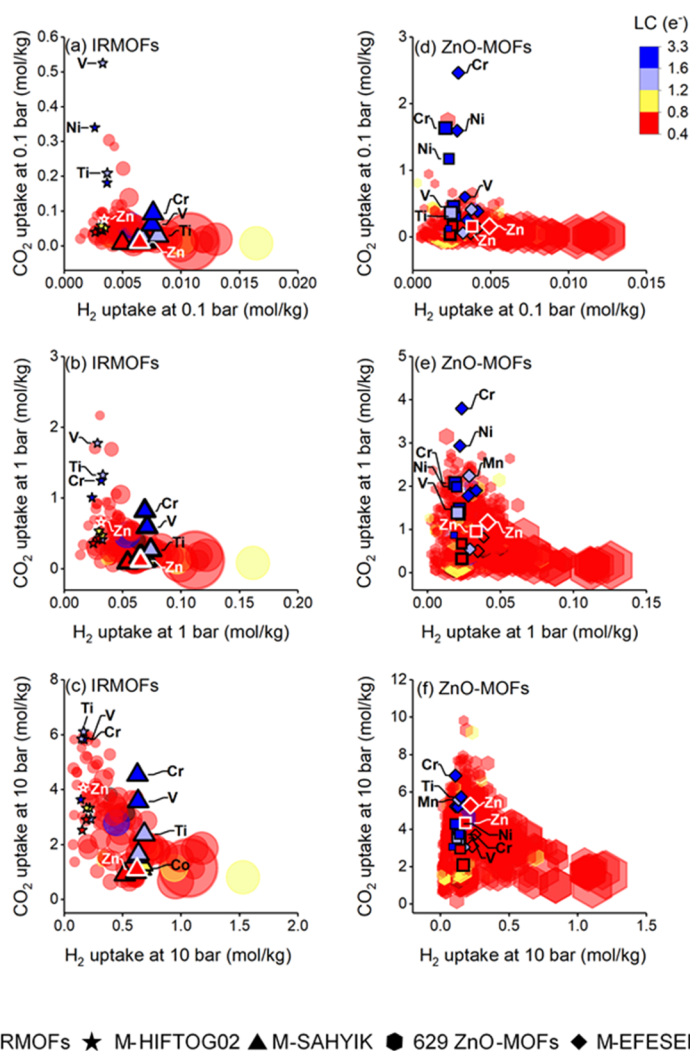
<sup>a</sup> $N_{\text{ads},i}$ : gas uptake under adsorption conditions (mol/kg),  $N_{\text{des},i}$ : gas uptake under desorption conditions (mol/kg),  $y$ : bulk composition of the gas mixture, and  $j$ :  $H_2$  or  $CH_4$ .

that lead to high APS together with a good value of  $R$  % (generally >85%) are desired.

### 3. RESULTS AND DISCUSSION

**3.1. Effect of Metal Exchange on  $CO_2/H_2$  Uptakes of MOFs.** We first compared the effect of metal exchange on  $CO_2/H_2$  mixture uptakes of IRMOFs and ZnO-MOFs. We discussed the performances of MOFs with respect to their structural properties and the largest charge (LC) assigned to the metal atom.  $CO_2$  and  $H_2$  uptakes of 140 IRMOFs and 629 ZnO-MOFs computed at 0.1, 1, and 10 bar are shown in Figure 2. We color-coded and scaled the size of the data points according to the LC assigned to the framework atoms and PLD of the MOFs, respectively. Most of the IRMOFs and ZnO-MOFs had LC between 0.4 and 0.8 e<sup>-</sup>, and IRMOFs had relatively larger pores (PLD range: 3.9–25.1 Å, LCD range: 5.5–26.1 Å) compared to ZnO-MOFs (PLD range: 3.7–19.2 Å, LCD range: 4.2–21.6 Å).  $CO_2$  uptakes of ZnO-MOFs were higher than those of IRMOFs at 0.1, 1, and 10 bar, while IRMOFs had slightly higher  $H_2$  uptakes than ZnO-MOFs as pressure increased. Therefore, without metal exchange, ZnO-MOFs were potentially more promising in terms of their  $CO_2$  capture performance from the  $CO_2/H_2$  mixture compared to IRMOFs. IRMOFs and ZnO-MOFs with small pore sizes (shown with the smallest symbols) generally had the highest  $CO_2$  uptakes at all pressures as shown in Figure 2a–f. In MOFs having small pores,  $CO_2$  molecules have increased interactions with the framework atoms due to the strong confinement.<sup>53</sup> We note that most of the IRMOFs and ZnO-MOFs had small pore sizes and low LCs, and metal exchange can intensify the  $CO_2$ -MOF interactions, making these materials promising with high  $CO_2$  uptakes.

The  $CO_2$  and  $H_2$  uptakes of M-MOFs that we focused on are also shown in Figure 2. For the group of IRMOFs, we compared the gas adsorption performances of M-HIFTOG02 (stars) and M-SAHYIK (triangles) with 140 IRMOFs (circles) in Figure 2a–c. The ranges of PLD for M-HIFTOG02 and M-SAHYIK were computed as 4.1–4.6 and 7.8–8.2 Å, respectively. M-HIFTOG02 is the catenated form of M-SAHYIK,<sup>36</sup> therefore the former series had narrower pore sizes, smaller  $S_{\text{acc}}$ , and lower porosity than the latter (given in Table S1). As shown in Figure 2a–c, the metal exchange in M-HIFTOG02 had a strong influence on the  $CO_2$  uptakes, while  $H_2$  uptakes remained almost the same. We observed the greatest improvement in the  $CO_2$  uptake of M-HIFTOG02 at 0.1 bar (Figure 2a), where  $CO_2$  uptake of the original Zn-HIFTOG02 increased from 0.08 mol/kg to 0.18, 0.21, 0.34, and 0.52 mol/kg for Cr, Ti-, Ni-, and V-HIFTOG02, respectively. Ti, Ni, Cr, and V had the highest LC (1.40,



● 140 IRMOFs ★ M-HIFTOG02 ▲ M-SAHYIK ● 629 ZnO-MOFs ◆ M-EFESSEP ■ M-EFESZO

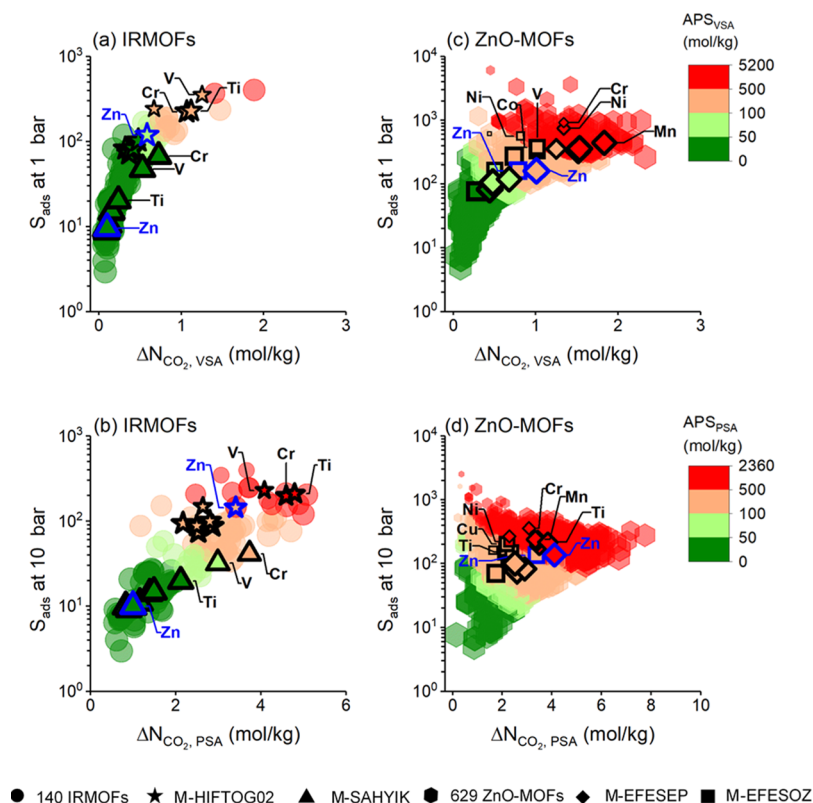
**Figure 2.** (a–c)  $\text{CO}_2$  and  $\text{H}_2$  uptakes of 140 IRMOFs (circles), M-HIFTOG02 (stars), and M-SAHYIK (triangles) at 0.1, 1, and 10 bar. (d–f)  $\text{CO}_2$  and  $\text{H}_2$  uptakes of 629 ZnO-MOFs (hexagonals), M-EFESSEP (diamonds), and M-EFESZO (squares) at 0.1, 1, and 10 bar. Color and size of symbols represent the LC and PLD of MOFs, respectively.

1.58, 2.07, and 2.33  $e^-$  for Ti, V, Cr, and Ni, respectively) among the M-HIFTOG02 series. Because  $\text{CO}_2$ -MOF electrostatic interactions are important at low pressures,<sup>53</sup> high LCs of these metals led to enhanced  $\text{CO}_2$  uptakes in these MOFs. Thus, at 0.1 bar,  $\text{CO}_2$  uptakes of V-HIFTOG02 and Ni-HIFTOG02 even exceeded those of all IRMOFs. The effect of metal exchange on the  $\text{CO}_2$  uptakes of M-IRMOFs was weaker but still observable at 10 bar compared to 0.1 bar. At 10 bar, Ti-HIFTOG02 with an exceptional  $\text{CO}_2$  uptake of 6.12 mol/kg surpassed the  $\text{CO}_2$  uptakes of all IRMOFs. This is a significant finding because the IRMOF with the highest  $\text{CO}_2$  uptake in our subset had a  $\text{CO}_2$  uptake value of 5.94 mol/kg and Ti-HIFTOG02 surpassed even this limit. This demonstrates that by exchanging Zn with V and Ti, we can build MOFs with exceptional  $\text{CO}_2$  uptakes that exceed the upper limits of  $\text{CO}_2$  uptake for all the existing IRMOFs.

Zn-SAHYIK had higher  $\text{H}_2$  uptakes and lower  $\text{CO}_2$  uptakes compared to Zn-HIFTOG02.  $\text{CO}_2$  uptakes of SAHYIK increased up to nine times, from 0.01 mol/kg to 0.09 mol/kg at 0.1 bar, when Zn is exchanged with Cr and the  $\text{CO}_2$  uptake of Cr-SAHYIK even overcame the  $\text{CO}_2$  uptake of Zn-HIFTOG02. However, the upper limit for  $\text{CO}_2$  uptakes of M-SAHYIK was lower than that of M-HIFTOG02, as catenated

MOFs (M-HIFTOG02 in this work) have been shown to have higher  $\text{CO}_2$  adsorption due to increased framework densities and lower porosities when compared to their noncatenated counterparts (M-SAHYIK in this work).<sup>29,54</sup> This can be supported with the higher  $Q_{\text{st},\text{CO}_2}$  values that we calculated for M-HIFTOG02 (ranging from  $-23$  to  $-37$  kJ/mol) than those for M-SAHYIK (ranging from  $-13$  to  $-29$  kJ/mol) at 0.1 bar. At 1 bar and 10 bar (Figure 2b,c),  $\text{CO}_2$  uptakes of M-SAHYIK almost reached those of M-HIFTOG02. At high pressures, the impact of physical properties such as the surface area and porosity of MOFs on gas uptakes is more pronounced.<sup>55</sup> M-HIFTOG02 series were easily saturated with  $\text{CO}_2$  molecules due to their lower porosities and surface areas (as shown in Table S1) compared to M-SAHYIK series, which have larger pores and a higher surface area providing more space for gas adsorption.

We compared the gas adsorption performances of M-EFESSEP (diamonds) and M-EFESZO (squares) with those of 629 ZnO-MOFs (hexagonals) in Figure 2d–f. Similar to M-IRMOFs, metal exchange did not remarkably change the  $\text{H}_2$  uptakes of M-ZnO-MOFs but altered their  $\text{CO}_2$  uptakes. As shown in Figure 2d, high LC assigned to Cr-EFESSEP and Cr-EFESZO provided the maximum  $\text{CO}_2$  uptakes observed for



**Figure 3.** (a,b)  $\Delta N_{\text{CO}_2}$  and  $S_{\text{ads}}$  of M-HIFTOG02 (stars), M-SAHYIK (triangles), and 140 IRMOFs (circles), (c,d)  $\Delta N_{\text{CO}_2}$  and  $S_{\text{ads}}$  of M-EFESOP (diamonds), M-EFESOPZ (squares), and 629 ZnO-MOFs (hexagonals) are shown for  $\text{CO}_2/\text{H}_2$  mixture separation under VSA and PSA conditions. Size of symbols represents  $R\%$ , color of symbols represents APS under the corresponding working conditions.

both the M-EFESOP and M-EFESOPZ series, followed by Ni and V. With Cr-EFESOP (Cr-EFESOPZ),  $\text{CO}_2$  uptakes increased by 15 (10)-fold compared to the original Zn-EFESOP (Zn-EFESOPZ) at 0.1 bar. Smaller PLDs of M-EFESOP (3.9–6.1 Å) lead to slightly higher  $\text{CO}_2$  uptakes compared to M-EFESOPZ (PLD: 3.7–6.8 Å). Due to the presence of guest ions in the pores of M-EFESOPZ, it had a smaller surface area compared to M-EFESOP (given in Table S2), therefore the available space for  $\text{CO}_2$  and  $\text{H}_2$  molecules was less in M-EFESOPZ compared to that in M-EFESOP. Moreover, the presence of anions ( $\text{Cl}^-$ ) has been shown to reduce  $\text{CO}_2$ – $\text{CO}_2$  interactions,<sup>56</sup> both Lennard Jones (LJ) and Coulomb, in the narrow pores of M-EFESOPZ, which lead to lower  $\text{CO}_2$  adsorption compared to M-EFESOP. For example, changing the metal from Zn to Cr increased the  $\text{CO}_2$  uptake of M-EFESOP from 0.17 to 2.47 mol/kg at 0.1 bar, while for M-EFESOPZ, the  $\text{CO}_2$  uptake increased from 0.16 to 1.63 mol/kg. The  $\text{CO}_2$  uptake of Cr-EFESOP is even higher than the  $\text{CO}_2$  uptakes of all ZnO-MOFs at 0.1 bar. At 1 bar (Figure 2e), Cr still had a positive impact on the  $\text{CO}_2$  uptakes, and  $\text{CO}_2$  uptake of Zn-EFESOP (Zn-EFESOPZ) increased from 1.17 to 3.80 mol/kg (from 0.95 to 2.07 mol/kg) when Cr was used. At 10 bar, exchanging Zn with Cr in EFESOP slightly increased  $\text{CO}_2$  uptake from 5.28 to 6.88 mol/kg (Figure 2f).

We compared the increase in  $\text{CO}_2$  uptakes of M-IRMOFs and M-ZnO-MOFs between 0.1 and 10 bar. At low pressure,  $\text{CO}_2$  uptakes of M-ZnO-MOFs were found to be more affected from the metal exchange compared to M-IRMOFs. For example, at 0.1 bar,  $\text{CO}_2$  uptake of M-HIFTOG02, increased up to 7-fold when Zn was exchanged with V (Figure 2a), whereas  $\text{CO}_2$  uptakes of M-EFESOP increased up to 15-fold

when Zn was exchanged with Cr (Figure 2d). The surface area and porosity of MOFs, and proximity of gas molecules to metal centers became important factors in determining the gas adsorption properties of M-MOFs with increased pressure.<sup>13</sup> At 10 bar, with the enhancement in the  $\text{CO}_2$  uptake, Ti-HIFTOG02 surpassed all IRMOFs (Figure 2c), while Cr-EFESOP and Cr-EFESOPZ surpassed most of the ZnO-MOFs (Figure 2f). Overall, we observed that exchange of Zn with Cr significantly increased the  $\text{CO}_2$  uptakes of both M-IRMOFs and M-ZnO-MOFs.

To explain the change in the gas uptakes of MOFs upon the metal exchange, we examined the contribution of LJ interaction energies ( $E_{\text{LJ}}$ ) for MOF- $\text{H}_2$ , MOF- $\text{CO}_2$ ,  $\text{CO}_2$ – $\text{CO}_2$ ,  $\text{H}_2$ – $\text{H}_2$ , and  $\text{CO}_2$ – $\text{H}_2$ , and the contribution of the Coulombic interaction energies ( $E_{\text{Coulomb}}$ ) for MOF- $\text{CO}_2$  and  $\text{CO}_2$ – $\text{CO}_2$  to the total energy. Figure S2 shows that the highest contributions to total energy were by  $E_{\text{LJ,MOF-CO}_2}$  and  $E_{\text{Coulomb,MOF-CO}_2}$  for M-SAHYIK and M-HIFTOG02, while the gas–gas interaction energies  $E_{\text{LJ,H}_2\text{-H}_2}$  and  $E_{\text{LJ,CO}_2\text{-H}_2}$  had almost no contribution at any pressure. MOF- $\text{H}_2$  LJ interaction energy,  $E_{\text{LJ,MOF-H}_2}$ , in M-SAHYIK was more effective compared to that in M-HIFTOG02 (Figure S2a–c), which can be explained by the size and geometry of the structures. Large rectangular pores of M-SAHYIK did not favor  $\text{CO}_2$  adsorption as much as small triangular pores of M-HIFTOG02, which resulted in lower MOF- $\text{CO}_2$  interaction energies and higher MOF- $\text{H}_2$  interaction energies. Thus, higher  $\text{H}_2$  adsorption was observed in M-SAHYIK (0.05–0.07 mol/kg) compared to M-HIFTOG02 (0.02–0.03 mol/kg) at 1 bar. As pressure increased from 0.1 to 10 bar, contribution of  $E_{\text{Coulomb,MOF-CO}_2}$

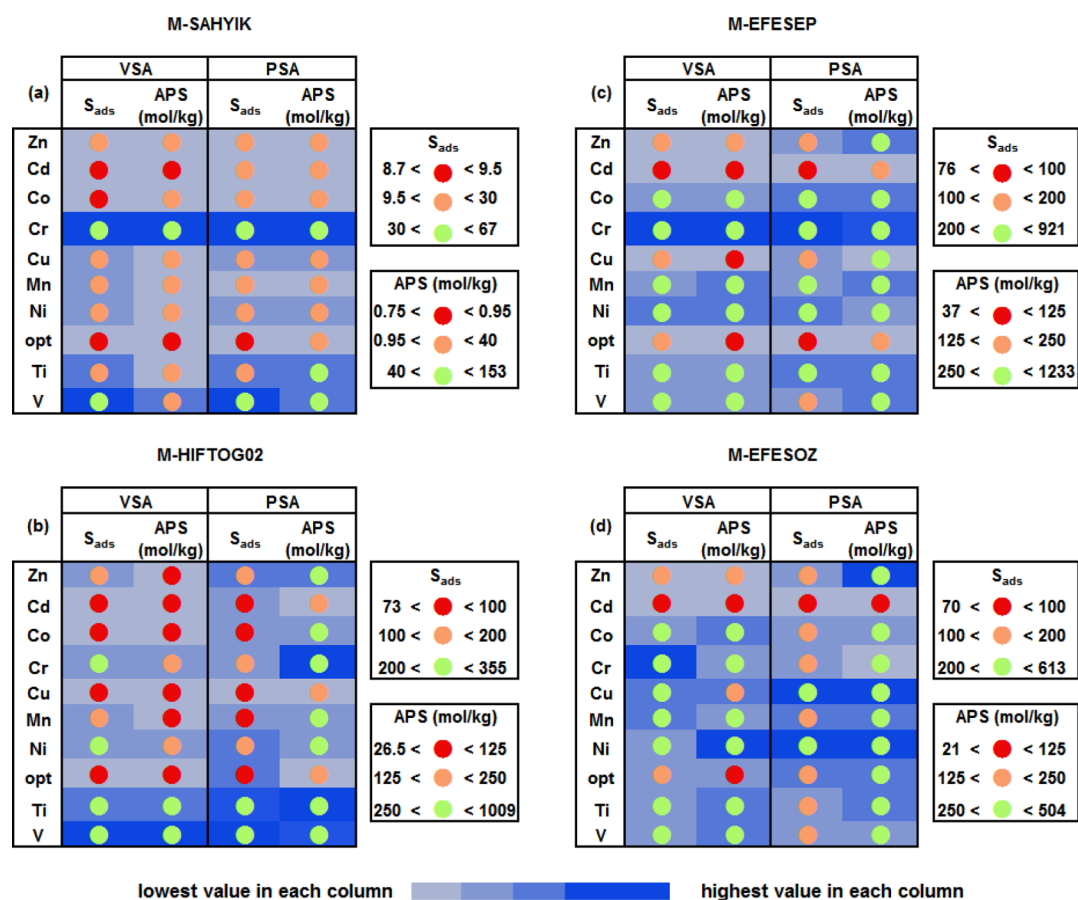
in M-HIFTOG02 decreased, whereas contributions of  $E_{LJ,MOF-CO_2}$  and  $E_{LJ,CO_2-CO_2}$  increased as shown in Figure S2d–f. This was due to the presence of a high  $CO_2$  loading in the pores of M-HIFTOG02 at 10 bar, which can be observed from the adsorption snapshots given in Figure S3. The contribution of  $E_{Coulomb,MOF-CO_2}$  to the total energy was higher for V, Ni, Ti, and Cr-exchanged versions of M-SAHYIK and M-HIFTOG02 as the higher LC enhanced the calculated  $CO_2$ -MOF electrostatic interactions compared to those in Zn-SAHYIK and Zn-HIFTOG02. In Figure S4, a slightly higher contribution of  $E_{LJ,CO_2-CO_2}$  and  $E_{Coulomb,CO_2-CO_2}$  to total interaction energy in M-EFESOP compared to M-EFESOPZ is shown. Simulation snapshots shown in Figure S5 also indicate that all pores provide adsorption sites in M-EFESOP, whereas in M-EFESOPZ mainly the large pore was preferred by both gases. As a result, MOFs with V, Ti, Ni, and Cr, which have LC > 1.4 e<sup>-</sup> and PLD < 5 Å led to a stronger confinement of  $CO_2$  at low pressure, as supported by a high contribution of  $E_{Coulomb,MOF-CO_2}$  to total interaction energy, and as a result these MOFs had the highest  $CO_2$  uptakes (the largest improvement in  $CO_2$  uptake was observed for Cr-EFESOP, from 0.17 to 2.47 mol/kg at 0.1 bar). As the pressure increased to 10 bar, Cr-EFESOP had the highest  $CO_2$  uptake with the improvement from 5.28 to 6.88 mol/kg, followed by Ti-HIFTOG02 with the improved  $CO_2$  uptake from 4.06 to 6.12 mol/kg. This increase was attributed to the tight packing of  $CO_2$  molecules inside the MOF pores as shown by decreased  $E_{Coulomb,MOF-CO_2}$  and increased  $E_{LJ,MOF-CO_2}$ ,  $E_{LJ,CO_2-CO_2}$ . Overall, MOFs with V, Ti, Ni, and Cr metals in M-HIFTOG02 series had higher  $CO_2$  uptakes than M-SAHYIK series at all pressures due to the catenated structure of M-HIFTOG02, and MOFs with V, Ti, Ni, and Cr in M-EFESOP series had higher  $CO_2$  uptakes than M-EFESOPZ series due to the empty space left by the absence of guest ions.

**3.2. Effect of Metal Exchange on  $CO_2/H_2$  Separation Performances of MOFs.** The ultimate goal of our work was to determine the effect of metal exchange on the adsorption-based gas separation performances of MOFs. In Figure 3,  $S_{ads}$  and  $\Delta N_{CO_2}$  of IRMOFs and ZnO-MOFs for  $CO_2/H_2$  mixture separation were shown under VSA and PSA conditions. Effects of metal exchange on APS were shown with the color of the symbols and adsorbent regenerability ( $R\%$ ) was shown with the size of the symbols in Figure 3. Under VSA and PSA conditions, ZnO-MOFs had higher adsorption selectivity ( $S_{ads}$ ) and  $CO_2$  working capacity ( $\Delta N_{CO_2}$ ), than IRMOFs, which was expected due to the higher  $CO_2$  uptakes of ZnO-MOFs. As Figure 3a,b shows, M-HIFTOG02 materials (stars) had higher  $S_{ads}$  and  $\Delta N_{CO_2}$  compared to M-SAHYIK series (triangles).  $S_{ads}$  and  $\Delta N_{CO_2}$  of original Zn-HIFTOG02 were computed as 119 and 0.58 mol/kg, respectively, whereas with changing the metal from Zn to V,  $S_{ads}$  and  $\Delta N_{CO_2}$  increased to 355 and 1.25 mol/kg, respectively, leading to a 6-fold increase in APS under VSA conditions. Under PSA conditions, V and Ti increased the  $S_{ads}$  of Zn-HIFTOG02 from 144 to 229 and 211, respectively, while its  $\Delta N_{CO_2}$  increased from 3.4 to 4 and 4.79 mol/kg, respectively. A trade-off between  $S_{ads}$  and  $\Delta N_{CO_2}$  has been commonly observed for MOFs with different physical properties,<sup>57</sup> whereas here we showed that  $S_{ads}$  and  $\Delta N_{CO_2}$  can be improved simultaneously with metal exchange. We also

considered the change in  $R\%$  values upon the metal exchange because  $R\%$  is a significant metric for the efficiency of adsorbents in cyclic processes. All M-HIFTOG02 except Ni-HIFTOG02 had  $R\%$  values equal to or larger than 70%, which is a good range of  $R\%$  values for an industrial gas separation process.

Exchange of Zn with Cr in M-SAHYIK (triangles) increased the  $S_{ads}$  of Zn-SAHYIK from 10 to 67 at 1 bar, and from 10 to 41 at 10 bar, in addition to an enhanced  $\Delta N_{CO_2}$  from 0.1 to 0.7 mol/kg under VSA conditions, and from 1 to 3.7 mol/kg under PSA conditions as shown in Figure 3. Therefore, upon the metal exchange, APS of Cr-SAHYIK increased 50 times under VSA conditions, and 15 times under PSA condition compared to the APS of Zn-SAHYIK. M-SAHYIK series also preserved their high  $R\%$  (represented by the size of the triangles) under VSA conditions upon the metal exchange, which made them highly promising materials. Overall, the metal exchange had a greater impact on the adsorbent performance evaluation metrics of M-SAHYIK than M-HIFTOG02, and the separation performance of Cr-SAHYIK even approached to that of Zn-HIFTOG02 in terms of both  $S_{ads}$  and  $\Delta N_{CO_2}$  under VSA conditions, and in terms of  $\Delta N_{CO_2}$  under PSA conditions. At this point, we compared the performances of M-SAHYIK and M-HIFTOG02, which had the highest APS and  $R\% > 70\%$  with the performances of all IRMOFs. Under VSA (PSA) conditions, APS of V-HIFTOG02 is higher than 98% (98%) of all IRMOFs, whereas APS of Cr-SAHYIK is higher than 85% (75%) of all IRMOFs, indicating that metal exchange significantly extended the current adsorbent performance limits of IRMOFs under VSA and PSA conditions.

In Figure 3c,d,  $S_{ads}$  and  $\Delta N_{CO_2}$  trends of M-EFESOP (diamonds) and M-EFESOPZ (squares) were compared with the results of ZnO-MOFs (hexagonals) under VSA and PSA conditions. Under VSA conditions, both  $S_{ads}$  and  $\Delta N_{CO_2}$  of Zn-EFESOP and Zn-EFESOPZ were improved with the metal exchange, while the improvement in  $S_{ads}$  was more pronounced compared to the improvement in  $\Delta N_{CO_2}$ . Changing the metal from Zn to Cr increased  $S_{ads}$  of EFESOP six times (from 161 to 921) and of EFESOPZ four times (from 160 to 613) under VSA conditions. Moreover, Zn-EFESOP exhibited  $\Delta N_{CO_2}$ ,  $S_{ads}$ , and  $R\%$  of 1.01 mol/kg, 160, and 86%, respectively, whereas Mn-EFESOP led to a more promising combination of  $\Delta N_{CO_2}$ ,  $S_{ads}$ , and  $R\%$  as 1.83 mol/kg, 446, and 81%, respectively, under VSA conditions. Similarly,  $\Delta N_{CO_2}$ ,  $S_{ads}$ , and  $R\%$  values of Zn-EFESOPZ were computed as 0.79 mol/kg, 160, and 83%, and these values were improved to 1.02 mol/kg, 378, and 74%, respectively, for Co-EFESOPZ. We note that the metal exchange with Cr increased the APS of Zn-EFESOP exceptionally from 162 to 1233 mol/kg, while Ni increased the APS of Zn-EFESOPZ from 126 to 462 mol/kg, under VSA conditions. Upon the exchange of Zn with V, Ni, and Cr, a decrease was observed in  $\Delta N_{CO_2}$  of M-EFESOP and M-EFESOPZ under PSA conditions, which was due to the more significant increase in their  $CO_2$  uptakes upon metal exchange at 1 bar compared to at 10 bar. However, with the increase in the  $S_{ads}$  (from 137 to 359), APS of Zn-EFESOP increased from 562 to 1104 mol/kg with Cr under PSA conditions. Overall, metal-exchange with Cr, Ni, and V led to a good combination of  $\Delta N_{CO_2}$  and  $S_{ads}$  for M-EFESOP and M-EFESOPZ under VSA conditions, while under PSA conditions an improvement in  $S_{ads}$  was achievable.



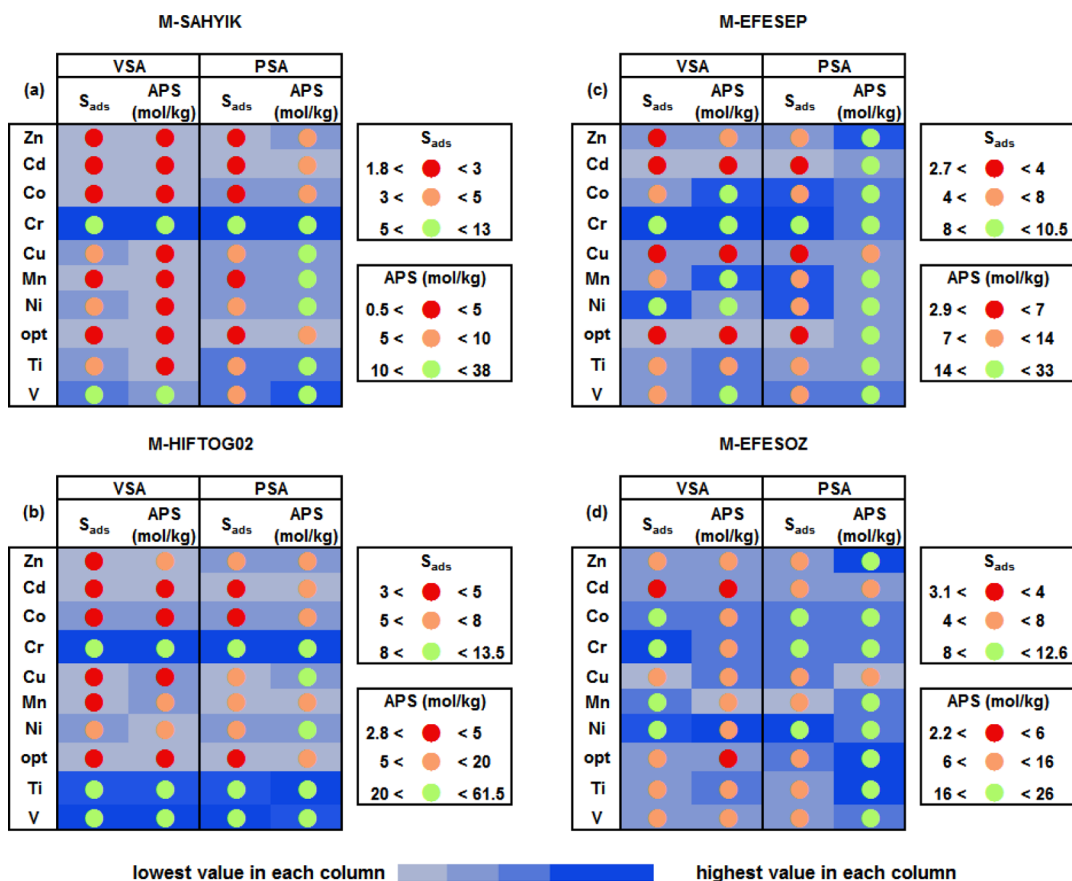
**Figure 4.** Infographic representing the effect of metal exchange on  $S_{\text{ads}}$  and APS of (a) M-SAHYIK, (b) M-HIFTOG02, (c) M-EFESEP, and (d) M-EFESOZ for  $\text{CO}_2/\text{H}_2$  mixture separation under VSA and PSA conditions. “opt” represents the “geometry optimized version of the original MOF”.

It is important to compare the calculated  $\text{CO}_2$  uptake and  $\text{CO}_2/\text{H}_2$  selectivity of the M-MOFs with those of existing MOFs constructed from the same metals to better situate the results of our work. Gagliardi and co-workers investigated single-component  $\text{CO}_2$  adsorption in M-IRMOF-10 ( $M = \text{Mg}, \text{Ca}, \text{Fe}, \text{Cu}, \text{Zn}, \text{Ge}, \text{Se}, \text{Cd}, \text{Sn}, \text{and Ba}$ ) and compared the  $\text{CO}_2$  adsorption isotherms of M-IRMOF-10 with IRMOF-1, IRMOF-7, IRMOF-16, and MOF-200.<sup>13</sup> At 1 bar, 298 K, single-component  $\text{CO}_2$  uptake of the best performing M-MOF (Ba-IRMOF-10) was computed as 1.9 mol/kg; while IRMOF-10, IRMOF-1, IRMOF-7, IRMOF-16, and MOF-200 had  $\text{CO}_2$  uptakes of 0.8, 1.82, 1.79, 0.56, and 0.59 mol/kg, respectively. In this study, we computed the highest  $\text{CO}_2$  uptake with Cr-SAHYIK (Cr-IRMOF-1) as 0.82 mol/kg. We note that this  $\text{CO}_2$  uptake was computed for a  $\text{CO}_2/\text{H}_2$ : 15/85 bulk mixture at 1 bar, 298 K, therefore the performance of Cr-SAHYIK can be considered as promising. We compared the  $\text{CO}_2$  uptake and  $\text{CO}_2/\text{H}_2$  selectivity of the top MOFs identified in this work with those of the top candidates of M-HKUST-1 series, where  $M = \text{Cd}, \text{Co}, \text{Cr}, \text{Cu}, \text{Fe}, \text{Mn}, \text{Mo}, \text{Ni}, \text{Ru}, \text{and Zn}$  in our previous study.<sup>24</sup> At 1 bar, 298 K, Cr-HKUST-1 had a  $\text{CO}_2$  uptake of 0.82 mol/kg and a selectivity of 97.63, while at 10 bar, 298 K, Cr-HKUST-1 was one of the top performers among other metal exchanged M-HKUST-1 with a  $\text{CO}_2$  uptake of 4.29 mol/kg and a selectivity of 67.75. In this work, Cr-EFESEP had the highest  $\text{CO}_2/\text{H}_2$  selectivity and  $\text{CO}_2$  uptake of 921.25 and 3.8 mol/kg, respectively, at 1 bar, 298 K, and 358.72 and 6.88 mol/kg, respectively, at 10 bar, 298 K.

These comparisons showed that metal exchange in Zn-MOFs led to superior performances compared to the performances of M-HKUST-1 for  $\text{CO}_2/\text{H}_2$  mixture separation.

Finally, in Figure 4, we summarized the overall effect of metal exchange on the  $\text{CO}_2/\text{H}_2$  separation performances of M-SAHYIK, M-HIFTOG02, M-EFESEP, and M-EFESOZ series under VSA and PSA conditions. The intensity of the blue color shifting from low to high represents the improvement in  $S_{\text{ads}}$  or APS, while the color of the circles reflects the most promising (green), the least promising (red), and in-between (yellow) performances ( $S_{\text{ads}}$  or APS) of M-MOFs. Figure 4a represents that exchange of Zn with Cr in M-SAHYIK increased both  $S_{\text{ads}}$  and APS under VSA and PSA conditions, and this was followed by V, uplifting the performance of Zn-SAHYIK to one of the highest performances. Similarly, changing the metal from Zn to Cr, Mn, or Ni enhanced the  $S_{\text{ads}}$  and APS of M-HIFTOG02 under VSA conditions, while Ti increased the APS of M-HIFTOG02 as much as Cr under PSA conditions as shown in Figure 4b. For M-EFESEP, significant improvements in the  $S_{\text{ads}}$  and APS under VSA and PSA conditions were observed when the original metal (Zn) was changed to V or Ti as shown in Figure 4c. As shown in Figure 4d, for M-EFESOZ exchange of Zn with a wide range of metals, Co, Cr, Mn, Ni, Ti, and V could double  $S_{\text{ads}}$  and APS of the MOF. These findings suggest that a major improvement in the gas separation efficiency of MOFs can be obtained by changing the metal from Zn to especially Cr or V, therefore, the





**Figure 5.** Infographic representing the effect of metal exchange on  $S_{ads}$  and APS of (a) M-SAHYIK, (b) M-HIFTOG02, (c) M-EFESEP, and (d) M-EFES0Z for  $\text{CO}_2/\text{CH}_4$  mixture separation under VSA and PSA conditions. “opt” represents the “geometry optimized version of the original MOF”.

synthesis and/or postsynthetic modification of MOFs with these metals warrants further studies.

**3.3. Effect of Metal Exchange on  $\text{CO}_2/\text{CH}_4$  Separation Performances of MOFs.** Because we obtained improved  $\text{CO}_2/\text{H}_2$  separation performances with M-IRMOFs and M-ZnO-MOFs, we also tested their separation performances for an equimolar  $\text{CO}_2/\text{CH}_4$  mixture. In Figure 5, we represented the effect of exchanging Zn with different metals on  $S_{ads}$  and APS of M-HIFTOG02, M-SAHYIK, M-EFESEP, and M-EFES0Z for  $\text{CO}_2/\text{CH}_4$  separation under VSA and PSA conditions. As shown in Figure 5a,b, similar to  $\text{CO}_2/\text{H}_2$  separation, superior  $S_{ads}$  and APS values were obtained with V- and Ti-HIFTOG02 compared to Zn-HIFTOG02 for  $\text{CO}_2/\text{CH}_4$  separation under both conditions. With the improvement in its  $S_{ads}$ , Cr-SAHYIK had a similar  $S_{ads}$  compared to V-HIFTOG02 under VSA conditions. Zn-HIFTOG02 and Zn-SAHYIK had  $S_{ads}$  of 3.9 and 1.8, respectively, for  $\text{CO}_2/\text{CH}_4$  separation under VSA conditions, which increased to 13.5 with V-HIFTOG02 and to 13.2 with Cr-SAHYIK. This is due to the significantly higher  $\text{CO}_2$  uptakes and lower  $\text{CH}_4$  uptakes of V-HIFTOG02 and Cr-SAHYIK compared to those of their Zn-versions as shown in Figure S6. For example,  $\text{CO}_2$  uptake of Zn-HIFTOG02 increased from 0.25 to 1.06 mol/kg (~4 fold) with V and  $\text{CO}_2$  uptake of Zn-SAHYIK increased from 0.04 to 0.33 mol/kg (~9 fold) with Cr at 0.1 bar. At 1 bar,  $\text{CO}_2$  uptake of Zn-HIFTOG02 increased from 1.80 to 3.47 mol/kg (~2 fold) with V, and  $\text{CO}_2$  uptake of Zn-SAHYIK increased from 0.36 to 2.39 mol/kg (~6 fold) with Cr. Thus, V-HIFTOG02 and Cr-SAHYIK significantly outperformed all IRMOFs under

VSA conditions. This was followed by Cr- and Ni-HIFTOG02, and V-SAHYIK, which also had high  $\text{CO}_2$  uptakes close to the maximum of the  $\text{CO}_2$  uptakes of all IRMOFs. In Figure S7, we provided  $S_{ads}$  and  $\Delta N_{\text{CO}_2}$  of M-IRMOFs and M-ZnO-MOFs for  $\text{CO}_2/\text{CH}_4$  separation. As shown, both the selectivity and working capacity of MOFs increased with metal exchange. A remarkable increase in  $\Delta N_{\text{CO}_2}$  of Zn-SAHYIK, from 0.33 to 2.06 mol/kg under VSA conditions, was observed when Cr was used. Both V-HIFTOG02 and Cr-SAHYIK had the highest APS values and surpassed the performances of all IRMOFs under VSA conditions. Under PSA conditions, V-HIFTOG02 and Ti-HIFTOG02 had outstanding separation performances compared to all IRMOFs, while Cr-SAHYIK competed with the best performing IRMOFs due to its high  $S_{ads}$ ,  $\Delta N_{\text{CO}_2}$ , and high APS.

For M-ZnO-MOFs, as shown in Figure S6, exchanging Zn with Cr increased the  $\text{CO}_2$  uptake of Zn-EFESEP from 0.46 to 2.95 mol/kg, and of Zn-EFES0Z from 0.40 to 1.86 mol/kg, at 0.1 bar. Cr-EFESEP had a  $\text{CO}_2$  uptake higher than that of all ZnO-MOFs at 0.1 bar. At 1 and 10 bar, metal exchange contributed less to  $\text{CO}_2$  uptakes but was still distinctive, and Cr-EFESEP had as high  $\text{CO}_2$  uptakes as the best performing ZnO-MOFs at 1 bar. As a result, Cr significantly increased the  $S_{ads}$  of Zn-EFESEP from 3.8 to 10.5 for  $\text{CO}_2/\text{CH}_4$  separations and APS increased from 7.7 to 21.7 mol/kg under VSA conditions as shown in Figure 5c. Figure 5d shows that  $S_{ads}$  of Zn-EFES0Z increased from 4.1 to 12.6 when Zn was changed to Cr, and Cr-EFES0Z had a high  $S_{ads}$  close to the highest  $S_{ads}$

observed for ZnO-MOFs (shown in Figure S7). Ni followed Cr and increased  $S_{\text{ads}}$  of Zn-EFESOZ from 4.1 to 11.3 under VSA conditions, while its APS also increased from 6.7 to 15.7 mol/kg. Although we selected the MOFs for in silico metal exchange by considering their pore sizes and  $S_{\text{ads}}$  for  $\text{CO}_2/\text{H}_2$  separation, M-MOFs considered in this work performed well both for  $\text{CO}_2/\text{H}_2$  and  $\text{CO}_2/\text{CH}_4$  mixtures. We observed a significant increase in both  $S_{\text{ads}}$  and APS of MOFs especially when Zn was exchanged with V, Cr, and Ni. This is an important result signaling that the MOFs carefully selected for metal exchange for  $\text{CO}_2/\text{H}_2$  separation can have improved separation performances also for other gas mixtures such as  $\text{CO}_2/\text{CH}_4$ .

Finally, it is important to note that the M-MOFs we examined in this work are hypothetically produced with computational methods and with the assumption of successful exchange of all metals in the framework. Combination of several factors, such as the MOF topology, coordination geometry of the metal node, pore sizes, and size and shape of the linkers, determines the extent of metal exchange of MOFs in experimental studies.<sup>25</sup> These factors may lead to partial metal exchange in MOFs, which eventually influence the  $\text{CO}_2$  capture performance of materials. Our results can be considered as useful guidelines to achieve the upper limits of  $\text{CO}_2$  separation performances of metal-exchanged M-IRMOFs and M-ZnO-MOFs under industrially applicable conditions for  $\text{CO}_2/\text{H}_2$  and  $\text{CO}_2/\text{CH}_4$  mixture separations.

#### 4. CONCLUSIONS

In this study, we presented a methodology to identify the most promising MOFs, which have the potential to provide an outstanding  $\text{CO}_2$  capture performance after metal exchange and evaluated their  $\text{CO}_2/\text{H}_2$  and  $\text{CO}_2/\text{CH}_4$  mixture separation performances. Out of 10,221 MOFs, we first identified 769 MOFs containing Zn nodes and a single type of linker without any functional groups (140 IRMOFs and 629 ZnO-MOFs), then targeted the MOFs providing  $S_{\text{ads}} \sim 100$  and  $\text{PLD} > 4 \text{ \AA}$  (16 IRMOFs and 284 ZnO-MOFs) and selected the candidate MOFs and their peer MOFs for metal exchange (two IRMOFs, M-SAHYIK and M-HIFTOG02, and two ZnO-MOFs, M-EFESOP and M-EFESOZ). We exchanged the metal atom (Zn) of the selected four MOFs with eight different metals (Cd, Co, Cr, Cu, Mn, Ni, Ti, and V), and 32 unique M-MOFs (16 M-IRMOFs and 16 M-ZnO-MOFs) were obtained. Molecular-level insights obtained from atomically detailed simulations showed that higher contributions of MOF- $\text{CO}_2$  electrostatic interactions in MOFs with Cr, V, Ni, and Ti metals led to higher  $\text{CO}_2$  uptakes. We compared the  $\text{CO}_2$  separation performances of 32 in silico designed M-MOFs with those of 629 ZnO-MOFs and 140 IRMOFs. Our results showed that the most significant improvement in the  $\text{CO}_2/\text{H}_2$  and  $\text{CO}_2/\text{CH}_4$  selectivities of MOFs was obtained with Cr, V, Ni, and Ti metals. After the exchange of Zn with V,  $\text{CO}_2/\text{H}_2$  selectivity of HIFTOG02 (355 at 1 bar) exceeded the selectivity of 138 out of 140 IRMOFs. The exchange of Zn with Cr improved the selectivity of EFESOP (921 at 1 bar) and surpassed the selectivities of 602 ZnO-MOFs out of 629 ZnO-MOFs. These increases in selectivities were accompanied by an increase in working capacity of MOFs resulting in improved APSs. Thus, metal exchange was shown to significantly extend the current adsorbent performance limits of ZnO-MOFs. We hope these results will inspire computational and experimental studies that investigate the influence of metal exchange on  $\text{CO}_2$

separation performances of MOFs with different MOF topologies or a rich amount of guest ion pairs and will serve as a catalyst to identify novel M-MOFs with exceptional  $\text{CO}_2$  capture performances.

#### ■ ASSOCIATED CONTENT

##### Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acs.jpcc.1c03630>.

Classification of MOF node types according to the bonding scheme for IRMOFs and ZnO-MOFs; contribution of LJ interaction energies and Coulomb interaction energies to total energy in M-SAHYIK, M-HIFTOG02, M-EFESOP, and M-EFESOZ;  $\text{CO}_2$  and  $\text{H}_2$  adsorption snapshots of Zn-SAHYIK, Cr-SAHYIK, Zn-HIFTOG02, V-HIFTOG02, Zn-EFESOP, Cr-EFESOP, Zn-EFESOZ, and Cr-EFESOZ at 10 bar;  $\text{CO}_2$  and  $\text{CH}_4$  uptakes of IRMOFs, ZnO-MOFs, M-IRMOFs, and M-ZnO-MOFs at 0.1, 1, and 10 bar;  $\text{CO}_2/\text{CH}_4$  separation performances of IRMOFs, ZnO-MOFs, M-IRMOFs, and M-ZnO-MOFs under VSA and PSA conditions; and  $S_{\text{ads}}$  of M-IRMOFs and M-ZnO-MOFs at 0.1, 1, and 10 bar with respect to their physical properties (LCD, PLD,  $S_{\text{acc}}$ , and  $\phi$ ) and their chemical properties ( $\epsilon$ ,  $\sigma$ , and the LC assigned in the framework) (PDF)

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##### Notes

The authors declare no competing financial interest.

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