

# Theoretical Investigation of the BCN Monolayer and Their Derivatives for Metal-free CO<sub>2</sub> Photocatalysis, Capture, and Utilization

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Cite This: *ACS Omega* 2024, 9, 3772–3780

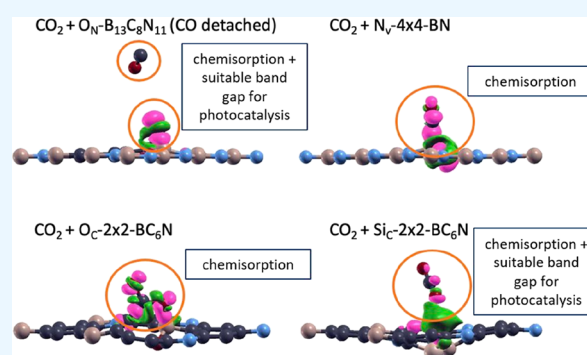
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**ABSTRACT:** In recent years, carbon capture and utilization (CCU) has been explored as an attractive solution to global warming, which is mainly caused by increasing CO<sub>2</sub> emission levels. Many functional materials have been developed for removing atmospheric CO<sub>2</sub> and converting it to more useful forms of carbon. Traditional metallic photocatalytic species have drawbacks—photocorrosion, low visible-light absorbance, and environmental damage; therefore, metal-free materials have attracted considerable research attention. In particular, boron nitride (BN) possesses unique B–N bonds, characterized by a large difference in the electronegativity of atoms that facilitates CO<sub>2</sub> reduction, and catalytic CO<sub>2</sub> reduction by boron carbon nitride (BCN) has been demonstrated under visible light; hence, these two materials can be considered potential CO<sub>2</sub> reduction photocatalysts. However, further modification of the materials and their applicability to other CCU applications have not been extensively explored. Therefore, we decided to investigate the modification of BCN monolayers, with the aim of ensuring that the properties of the materials are better suited, first, to the requirements of CO<sub>2</sub> photocatalysis, and second, to those of carbon capture or other optoelectronic applications. In this study, we considered various novel BCN monolayers, based on modification via metal-free substitutional doping and nitrogen vacancy creation, and performed first-principles density functional theory calculations. The effects of the modifications on band gap tuning, charge transfer, and the CO<sub>2</sub> adsorption ability were all studied. Specifically, O<sub>N</sub>-B<sub>13</sub>C<sub>8</sub>N<sub>11</sub> and Si<sub>C</sub>-2 × 2-BC<sub>6</sub>N were shown to possess excellent properties for photocatalytic CO<sub>2</sub> reduction, and O<sub>C</sub>-2 × 2-BC<sub>6</sub>N and N<sub>V</sub>-4 × 4-BN can be considered for future CO<sub>2</sub> capture materials. These results contribute to existing CCU approaches, suggesting that BCN monolayer modification merits further investigation, and offering insights relevant to other photocatalytic applications.



## 1. INTRODUCTION

Global warming affects multiple aspects of the environment and threatens human health. The average global temperature in 2022 was 1.15 °C above that in 1850–1900.<sup>1</sup> The increase in global temperature has led to the melting of ice caps, rise in sea levels, loss of coastal land, reductions in biodiversity, and increased risk of extreme weather conditions.<sup>2</sup> According to the World Health Organization, climate change is expected to cause approximately 250,000 additional deaths annually after 2030.<sup>3</sup> Despite global efforts to address this impending crisis, fossil fuels remain the primary source of energy, highlighting the urgent need to control the levels of produced atmospheric carbon dioxide.<sup>4,5</sup> Scientists have made many attempts to respond to the increasing CO<sub>2</sub> level problem. Existing approaches can be generally categorized as either carbon capture and storage (CCS) approaches or carbon capture and utilization (CCU) approaches.<sup>6</sup> Both approaches include the selective removal of CO<sub>2</sub> gas from the atmosphere. Current carbon capture routes include absorption-based methods, membrane separation, adsorption-based methods,

chemical looping, and direct air capture in postcombustion, precombustion, or with the use of oxyfuel combustion. In CCS, after CO<sub>2</sub> is captured, it is compressed and sequestered in geological formations.<sup>7</sup> However, the high cost and uncertain long-term effects of CCS make the CCU strategy a more attractive and feasible alternative.<sup>6</sup> CCU aims to transform the captured CO<sub>2</sub> into other valuable products, such as carboxylic acids, alcohols, aldehydes, etc., or to use the gas to replace conventional raw materials in industrial processes.<sup>8</sup> Various materials that can adsorb CO<sub>2</sub> and catalyze the most crucial step in this approach, that is, the CO<sub>2</sub> reduction reaction (CRR), with high selectivity and efficiency, have been investigated.<sup>4,9</sup>

Received: October 7, 2023

Revised: December 15, 2023

Accepted: December 15, 2023

Published: January 5, 2024



Methods such as CO<sub>2</sub> electrocatalysis, photocatalysis, and CO<sub>2</sub> biotransformation have demonstrated promise for transforming greenhouse gases into value-added compounds or photoelectric materials.<sup>10,11</sup>

Among these methods, the photocatalyzed CRR stands out because it can utilize sunlight, a widely available and sustainable energy source, to produce CCU products.<sup>4,10</sup> In photocatalyzed CRR, photons are absorbed by the photocatalyst to create electrons that participate in the reduction reaction of adsorbed CO<sub>2</sub>, and the final products will be desorbed from the material.<sup>12</sup> Chemisorption is a necessary condition required to transfer electrons from the photocatalyst to the adsorbed gas and activate it.<sup>13</sup> Recent research focusing on finding suitable photocatalysts and understanding the CRR mechanism indicates that photocatalytic CRR is one of the most effective methods for managing the CO<sub>2</sub> levels. Similarly, photocatalysts were designed to degrade other harmful atmospheric gases, such as NO<sub>x</sub> and organic pollutants, under accessible conditions.<sup>14–16</sup> Thus, there is a need for further studies in this field.<sup>17</sup>

Many materials have been developed to adsorb CO<sub>2</sub> and act as a photocatalyst, thereby enabling the chemically inert gas to undergo chemical reactions.<sup>18–20</sup> Since Geim et al. successfully isolated the monolayer graphene in 2004 and won the Nobel Prize in Physics in 2010 for this discovery, two-dimensional (2D) materials have been widely researched both theoretically and experimentally, including their usage as CRR photocatalysts.<sup>21</sup> 2D materials exhibit remarkable chemical and physical properties, including a large surface area, which provides accessible active sites, thinness, which minimizes the migration distance of photogenerated carriers, and a 2D planar structure, which facilitates material modification to improve their properties further.<sup>21</sup> These properties align well with the requirements of CRR photocatalysts. Notably, density functional theory (DFT) has been employed to study various 2D photocatalysts, aiming to aid experimental research.<sup>22</sup> Existing CRR 2D photocatalysts include metal-containing catalysts, such as metal oxides, metal sulfides, bismuth oxyhalides, MXenes, 2D metal–organic frameworks, and metal-free catalysts, such as graphene and its derivatives, carbon nitride, and boron nitride (BN) monolayers.<sup>4,23–27</sup> Although the catalytic performance of metal-free 2D photocatalysts is not as good as that of metal-containing 2D photocatalysts in terms of their production rate, metallic species are prone to photocorrosion and have poor absorption of visible light. Moreover, because metal-free monolayer materials use lightweight and abundant elements, which aligns with the concept of green chemistry, research interest has shifted toward improving metal-free photocatalysts through surface modification.<sup>4,28</sup> For example, graphene and its derivatives exhibit tunable band gaps and enhanced catalytic activity by doping with heteroatoms (O, B, N, P, etc.).<sup>29</sup> Therefore, metal-free monolayer photocatalysts are particularly interesting and worthy of a detailed investigation.

BN monolayers have a structure similar to that of graphene and possess a wide indirect band gap, which is unsuitable for photocatalysis.<sup>30</sup> However, BN has a unique B–N channel characterized by a high electronegativity difference that promotes the photoreduction reaction.<sup>13</sup> Thus, to take advantage of this high electronegativity difference, modifications are needed to address the large indirect band gap (approximately 5.5 eV) and weak CO<sub>2</sub> adsorption due to unreactive surface chemistry.<sup>4,13</sup> When BN is doped with O, N, or P, its visible-light absorption and photoactivity are enhanced.<sup>13</sup> Doping BN with Fe or Ni dimers and CuMn heteronuclear dimers improves CO<sub>2</sub>

to CH<sub>4</sub> conversion.<sup>31</sup> When doped with C, h-BCN nanosheets can catalyze the H<sub>2</sub> and O<sub>2</sub> evolution reactions.<sup>32</sup> Doping the lattice structure with aromatic carbon induces visible-light absorption and catalysis.<sup>26</sup> Novel BCN monolayers may exhibit diverse analogous material properties by tuning the B, C, and atomic arrangements and concentrations.<sup>30</sup> Given the variable properties of such materials, BCN monolayers have been used in a wide range of applications, including gas sensors, catalysis, and energy storage.<sup>32</sup>

Many BCN studies prioritized photocatalytic water splitting and other applications over developing and characterizing CO<sub>2</sub> reduction photocatalysis.<sup>32</sup> However, some researchers found that BCN monolayers showed a greater ability to absorb CO<sub>2</sub> than graphene.<sup>33</sup> Other groups calculated CO<sub>2</sub> adsorption energies ( $E_{\text{ad}}$ ) on BC<sub>6</sub>N and BC<sub>3</sub>N<sub>2</sub> monolayers, both showing weak interactions.<sup>34,35</sup> Previous studies also experimentally confirmed that BCN can reduce CO<sub>2</sub> to CO under visible-light illumination.<sup>30</sup> However, metal-free modifications of BCN for photocatalysis need to be better understood.

We aim to investigate new BCN monolayer structures suitable for CO<sub>2</sub> reduction photocatalysts while considering other CCU applications. Various 2D BCN configurations have been proposed and can be grouped based on their B:N molar ratio, which changes their electronic structure and material properties. Here, we investigated B<sub>13</sub>C<sub>8</sub>N<sub>11</sub> and B<sub>11</sub>C<sub>12</sub>N<sub>9</sub> as representative BCN structures with more B atoms than N atoms. BC<sub>6</sub>N is a structure with equal numbers of B and N atoms, and B<sub>11</sub>C<sub>8</sub>N<sub>13</sub> and B<sub>9</sub>C<sub>12</sub>N<sub>11</sub> are structures with more N atoms than B atoms. These structures were chosen and proposed based on relevant structures reported in prior studies.<sup>30,34</sup> Oxygen substitutional doping was conducted to achieve metal-free modification of the materials. Because O-doping on the BN monolayer induces chemisorption, a similar effect is expected on the BCN monolayers.<sup>36</sup> The target band gap of an ideal photocatalyst that absorbs visible light is 1.8–2.5 eV.<sup>13</sup> Moreover, the target  $E_{\text{ad}}$  for chemisorption is  $\geq 0.3$  eV.<sup>37</sup> We used DFT to calculate the most stable atomic structures of the BN, BC<sub>6</sub>N, B<sub>13</sub>C<sub>8</sub>N<sub>11</sub>, B<sub>11</sub>C<sub>12</sub>N<sub>9</sub>, B<sub>11</sub>C<sub>8</sub>N<sub>13</sub>, and B<sub>9</sub>C<sub>12</sub>N<sub>11</sub> monolayers as well as C-doped, O-doped, Si-doped, and N-vacancy-modified monolayers. We then calculated the binding energies of CO<sub>2</sub> on these monolayer surfaces to determine whether chemisorption is possible. Band structures and charge transfers were also calculated to determine the monolayer with the best potential for use as a CO<sub>2</sub> reduction photocatalyst and other capture and utilization applications. As such, this novel research contributes to the further development of new and existing CCU approaches.

## 2. EXPERIMENTAL SECTION

A suitable band gap for visible-light absorption (i.e., in the range of 1.8–2.5 eV) and sufficient chemisorption of CO<sub>2</sub> (i.e.,  $E_{\text{ad}} \geq 0.3$  eV) are the two main parameters used to assess potential CO<sub>2</sub> photocatalysts, which is the primary aim of this research.<sup>13,34</sup> Hereafter, substitutional doping is denoted as X<sub>Y</sub>, where X represents the dopant and Y is the substituted atom. Nitrogen vacancies are denoted as N<sub>v</sub>.

**2.1. Computational Details.** We performed first-principles calculations based on DFT within the commonly used Perdew–Burke–Ernzerhof (PBE) generalized gradient approximation (GGA) implemented in the ABINIT code.<sup>38</sup> The projected augmented wave (PAW) method was used to generate pseudopotentials with the ATOMPAW code.<sup>39,40</sup> The electron configurations and radius cutoffs used for B, C, N, and Si were

[He]2s<sup>2</sup>2p<sup>1</sup> and 0.90 Å, [He]2s<sup>2</sup>2p<sup>2</sup> and 0.79 Å, [He]2s<sup>2</sup>2p<sup>3</sup> and 0.64 Å, [He]2s<sup>2</sup>2p<sup>4</sup> and 0.74 Å, and [Ne]3s<sup>2</sup>3p<sup>2</sup> and 1.01 Å, respectively.

In all calculations, the self-consistent field (SCF) was terminated when the total energy difference was less than  $2.7 \times 10^{-9}$  eV twice consecutively. The converged values for the kinetic energy cutoff, vacuum height of the unit cells, and Monkhorst–Pack  $k$ -point grids were considered to be reached when the total energy difference was smaller than  $2.7 \times 10^{-3}$  eV twice consecutively.

The Broyden–Fletcher–Goldfarb–Shanno (BFGS) minimization algorithm was used in the structural optimization calculations. The maximum force tolerance used was  $2.6 \times 10^{-3}$  eV/Å ( $5.0 \times 10^{-5}$  Ha/Bohr).

**2.2. Atomic Structures.** Convergence calculations for the vacuum height, kinetic energy cutoff, and  $k$ -mesh were conducted using the above-mentioned criteria before the structural optimization calculations. To converge the cell size and relax the bond length of CO<sub>2</sub>, we used the molecule-in-a-big-box method. For all pure and modified monolayers, from the primitive cells to all  $4 \times 4$  cells shown in Table 1, the slab model was used. The vacuum height, kinetic energy cutoff, and  $k$ -mesh were converged according to the above-mentioned criterion.

**Table 1. Calculated Lattice Parameters  $a$  (Å) for the Fully Optimized Monolayers used**

monolayer	$a$ (Å)
4 × 4-BN	10.05
N <sub>v</sub> -4 × 4-BN	9.95
C <sub>2</sub> -doped-4 × 4-BN	10.05
C <sub>4</sub> -doped-4 × 4-BN	10.04
B <sub>13</sub> C <sub>8</sub> N <sub>11</sub>	10.12
O <sub>C</sub> -B <sub>13</sub> C <sub>8</sub> N <sub>11</sub>	10.08
O <sub>N</sub> -B <sub>13</sub> C <sub>8</sub> N <sub>11</sub>	10.10
N <sub>13</sub> C <sub>8</sub> B <sub>11</sub>	9.96
B <sub>11</sub> C <sub>12</sub> N <sub>9</sub>	10.53
N <sub>11</sub> C <sub>12</sub> B <sub>9</sub>	9.94
2 × 2-BC <sub>6</sub> N	9.96
O <sub>C</sub> -2 × 2-BC <sub>6</sub> N	9.94
O <sub>N</sub> -2 × 2-BC <sub>6</sub> N	9.98
Si <sub>C</sub> -2 × 2-BC <sub>6</sub> N	10.13

The modified monolayers used the same vacuum height, energy cutoff (i.e., the highest value obtained), and  $k$ -mesh as the  $4 \times 4$ -BN and  $2 \times 2$ -BC<sub>6</sub>N monolayers. All structures were relaxed by using these values via the BFGS method to obtain the optimized coordinates of the atoms. The relaxed coordinates were used to calculate the electronic structures and total energies of all monolayers.

For the CO<sub>2</sub> monolayer complexes, we placed CO<sub>2</sub> in the middle of the bridge site between the B–C bond parallel to the plane in the pristine BCN structures.<sup>34</sup> After testing with different CO<sub>2</sub> placements, we used orientations that maximized absorption. In the O- and Si-doped BCN structures, CO<sub>2</sub> was placed vertically in the hexagonal centers. In N<sub>v</sub>-4 × 4-BN, CO<sub>2</sub> was inserted vertically into the vacant hole. Cell sizes of the modified  $4 \times 4$  monolayers were used as the values are greater than that of CO<sub>2</sub> to avoid CO<sub>2</sub> interaction with its images. Full relaxations of the complex interfaces were conducted to obtain the optimized structures and lattice parameters used for the electronic structure and total energy calculations.

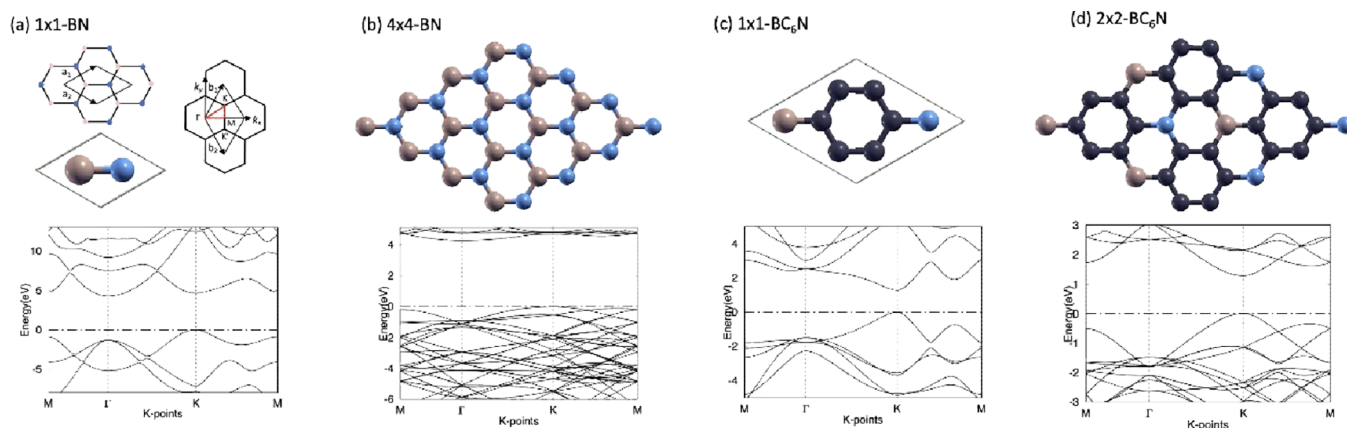
**2.3. Adsorption Energy.** The adsorption energy ( $E_{ad}$ ) between CO<sub>2</sub> and the monolayer is needed to assess whether the molecule is chemisorbed or physisorbed on the monolayer surface and whether the reduction reaction can occur. Thus,  $E_{ad}$  is crucial for determining the effectiveness of a photocatalyst.  $E_{ad}$  can be found with the following equation:

$$E_{ad} = E_{CO_2+ML} - E_{ML} - E_{CO_2} \quad (1)$$

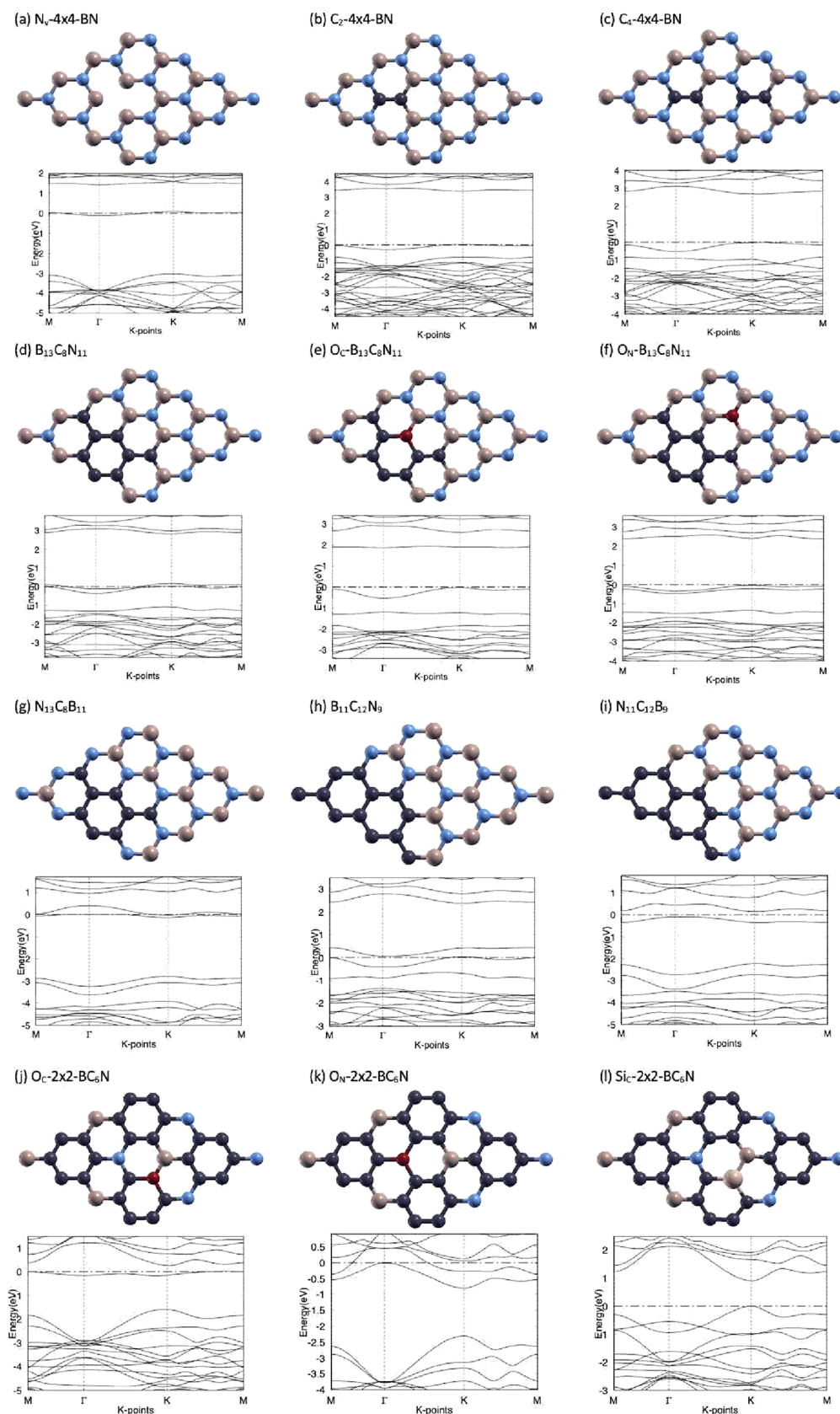
where  $E_{CO_2+ML}$  is the total energy of the complex interface,  $E_{ML}$  is the total energy of the monolayers, and  $E_{CO_2}$  is the total energy of CO<sub>2</sub>. A negative  $E_{ad}$  indicates that CO<sub>2</sub> is adsorbed on the monolayer, and the more negative the value of  $E_{ad}$ , the stronger the adsorption. When  $E_{ad} \leq -0.3$  eV, chemisorption occurs.

**2.4. Electronic Structures.** **2.4.1. Band Structure.** The band structures of the monolayers and complexes formed after the CO<sub>2</sub> adsorption were calculated. The high-symmetry  $k$ -point circuit was  $M(1/2, 0, 0)$ ,  $\Gamma(0, 0, 0)$ ,  $K(1/3, 2/3, 0)$ , and  $M(1/2, 0, 0)$ , as shown in Figure 1. The direct band gap is preferred for efficient photocatalysis.

**2.4.2. Charge Transfer.** Monolayers that demonstrated CO<sub>2</sub> chemisorption were selected for charge-transfer calculations. The charge transfer that occurred is indicated by a difference in



**Figure 1.** (a) Real and reciprocal space lattice vectors of the BN monolayers. The high-symmetry  $k$ -points in the first Brillouin zone are  $M(1/2, 0, 0)$ ,  $\Gamma(0, 0, 0)$ ,  $K(1/3, 2/3, 0)$ , and  $M(1/2, 0, 0)$ , and the  $k$ -point circuit is shown in red. The primitive cell and band structure plot of the BN monolayer are also shown. (b)  $4 \times 4$ -BN monolayer atomic structure and band structure plot. (c) BC<sub>6</sub>N primitive cell and band structure plot. (d)  $2 \times 2$ -BC<sub>6</sub>N atomic structure and band structure plot. The Fermi level was set to 0 eV. B, C, and N atoms are colored pink, black, and blue, respectively.



**Figure 2.** Relaxed atomic and band structure plots of the studied monolayers: (a)  $N_v$ - $4 \times 4$ -BN, (b)  $C_2$ - $4 \times 4$ -BN, (c)  $C_4$ - $4 \times 4$ -BN, (d)  $B_{13}C_8N_{11}$ , (e)  $O_C$ - $B_{13}C_8N_{11}$ , (f)  $O_N$ - $B_{13}C_8N_{11}$ , (g)  $N_{13}C_8B_{11}$ , (h)  $B_{11}C_{12}N_9$ , (i)  $N_{11}C_{12}B_9$ , (j)  $O_C$ - $2 \times 2$ - $BC_6N$ , (k)  $O_N$ - $2 \times 2$ - $BC_6N$ , and (l)  $Si_C$ - $2 \times 2$ - $BC_6N$ .

charge density between the isolated monolayer and  $CO_2$  and the combined complex and is calculated as follows:

$$\Delta\rho(r) = \rho_{CO_2+ML}(r) - \rho_{ML}(r) - \rho_{CO_2}(r) \quad (2)$$

where  $\rho_{\text{CO}_2+\text{ML}}(r)$ ,  $\rho_{\text{ML}}(r)$ , and  $\rho_{\text{CO}_2}(r)$  represent the charge densities of the monolayer and  $\text{CO}_2$  complex, monolayer, and  $\text{CO}_2$ , respectively.

### 3. RESULTS AND DISCUSSION

To study whether the monolayers can serve as a photocatalyst or be applied for other purposes, we calculated the atomic and electronic structures,  $E_{\text{ad}}$ , and charge transfer of  $\text{CO}_2$  adsorbed on the pure and modified monolayers.

**3.1. Pure Monolayer Calculations.** We first calculated the relaxed structures of the  $\text{CO}_2$ , BN primitive cell, and  $\text{BC}_6\text{N}$  primitive cell that are consistent with previous studies, and the values were used in later calculations. The  $\text{CO}_2$  molecule was relaxed to a linear structure and to a bond length of 1.17 Å, which agrees well with previous theoretical and experimental values of 1.16 Å.<sup>41,42</sup> The BN monolayer has a two-atom primitive cell and a large indirect band gap, as shown in Figure 1a. The length of the lattice parameter after relaxation is 2.51 Å, which is consistent with a previous theoretical of 2.49 Å and an experimentally obtained value of 2.54 Å.<sup>30,43</sup> The  $\text{BC}_6\text{N}$  primitive cell, as shown in Figure 1c, was relaxed to a lattice parameter of 4.98 Å, which is consistent with the previous theoretical value (5.01 Å).<sup>34</sup> The primitive cells were used for the  $4 \times 4$  cells shown in Figure 1b,d.

**3.2. Atomic and Electronic Structures of the Monolayers.** Using the values obtained from pure monolayer calculations, the optimized atomic structures of the modified monolayers were obtained after full relaxation. The lattice parameters of all monolayers are listed in Table 1, and their atomic and corresponding band structures are shown in Figure 2.

After relaxation, both  $\text{N}_{11}\text{C}_{12}\text{B}_9$  and  $\text{O}_\text{C}-2 \times 2\text{-BC}_6\text{N}$  have the smallest lattice constant, at 9.94 Å. As the atomic radii of N and O atoms are smaller than those of B and other atoms present, higher concentrations of N and O and a low concentration of B contribute to the decrease in the lattice size. Moreover, the largest monolayer is  $\text{B}_{11}\text{C}_{12}\text{N}_9$ , with a lattice constant of 10.53 Å, which is even greater than that of  $\text{Si}_\text{C}-2 \times 2\text{-BC}_6\text{N}$  (10.13 Å). This expansion in the lattice size can be attributed to the high B concentration and low N concentration of the monolayer. As B atoms have a relatively large atomic radius, a structure predominantly formed with B tends to be larger.

Additionally, as shown in Table 1,  $\text{BC}_6\text{N}$  and its derivatives tend to have lattice parameters smaller than those of the other studied monolayers. This is because the average C–C bond length of approximately 1.39 Å is longer than the B–N bond length of approximately 1.44–1.45 Å. Therefore, a high concentration of C–C pairs results in a smaller lattice parameter.

The results in Table 2 and Figure 2 show the various types of band structures observed for the studied monolayers. The current defect concentrations are relatively high, and the band structures reflect the interaction between the dopants or vacancies more compared to configurations where defects are located more sparsely. The structures we are most interested in are those with a direct band gap in the range of 0.875–2.55 eV after adjusting for the DFT calculation underestimations, i.e., those capable of visible-light absorption. Therefore, only  $\text{O}_\text{N}-\text{B}_{13}\text{C}_8\text{N}_{11}$  (2.44 eV),  $2 \times 2\text{-BC}_6\text{N}$  (1.27 eV), and  $\text{Si}_\text{C}-2 \times 2\text{-BC}_6\text{N}$  (0.90 eV) are suitable for  $\text{CO}_2$  photocatalysis. Moreover, for  $\text{B}_{13}\text{C}_8\text{N}_{11}$ ,  $\text{N}_{13}\text{C}_8\text{B}_{11}$ ,  $\text{B}_{11}\text{C}_{12}\text{N}_9$ , and  $\text{N}_{11}\text{C}_{12}\text{B}_9$ , which have similar carbon concentrations but reversed B:N molar ratios, we observe that the type of band gap remains the same.  $\text{B}_{13}\text{C}_8\text{N}_{11}$

**Table 2. Calculated Band Gaps for All Monolayers in the Present Study**

monolayer	band gap (eV)	band gap type
$4 \times 4\text{-BN}$	4.27	indirect
$\text{N}_\text{v}-4 \times 4\text{-BN}$	1.35	indirect
$\text{C}_2\text{-doped } 4 \times 4\text{-BN}$	3.37	direct
$\text{C}_4\text{-doped } 4 \times 4\text{-BN}$	2.71	direct
$\text{B}_{13}\text{C}_8\text{N}_{11}$	0	metallic
$\text{O}_\text{C}-\text{B}_{13}\text{C}_8\text{N}_{11}$	1.90	indirect
$\text{O}_\text{N}-\text{B}_{13}\text{C}_8\text{N}_{11}$	2.44	direct
$\text{N}_{13}\text{C}_8\text{B}_{11}$	0	metallic
$\text{B}_{11}\text{C}_{12}\text{N}_9$	0.03	indirect
$\text{N}_{11}\text{C}_{12}\text{B}_9$	0.25	indirect
$2 \times 2\text{-BC}_6\text{N}$	1.27	direct
$\text{O}_\text{C}-2 \times 2\text{-BC}_6\text{N}$	0.23	indirect
$\text{O}_\text{N}-2 \times 2\text{-BC}_6\text{N}$	0	metallic
$\text{Si}_\text{C}-2 \times 2\text{-BC}_6\text{N}$	0.90	direct

and  $\text{N}_{13}\text{C}_8\text{B}_{11}$  have metallic band gaps, whereas  $\text{B}_{11}\text{C}_{12}\text{N}_9$  and  $\text{N}_{11}\text{C}_{12}\text{B}_9$  have indirect band gaps.

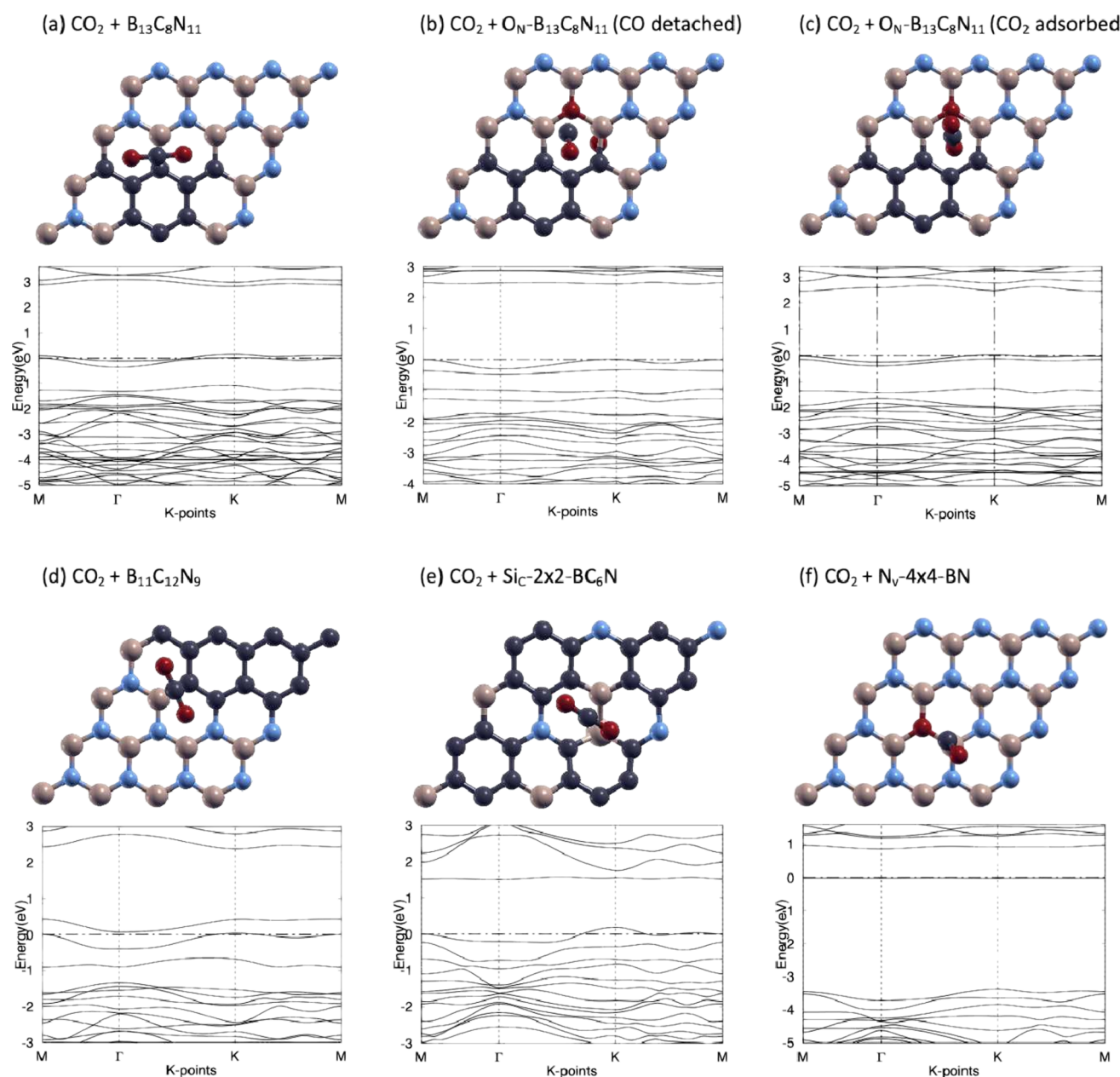
$\text{C}_2\text{-doped-}4 \times 4\text{-BN}$  and  $\text{C}_4\text{-doped-}4 \times 4\text{-BN}$  have direct band gaps of 3.37 and 2.71 eV, respectively (Table 2 and Figure 2b,c). The decrease in band gap between  $\text{C}_2\text{-doped-}4 \times 4\text{-BN}$  and  $\text{C}_4\text{-doped-}4 \times 4\text{-BN}$  indicates an inverse relationship between the C atom concentration and band gap. This trend matches the findings of Huang et al., who showed that the BN band gap can be tuned for specific functions by varying the C atom concentration.<sup>30</sup> This finding motivated us to further modify the materials by varying their C atom concentrations. We note that the band structures of the two monolayers ( $\text{B}_{13}\text{C}_8\text{N}_{11}$  and  $\text{B}_{11}\text{C}_{12}\text{N}_9$ ) were misinterpreted by Huang et al. Specifically, the band gaps determined in their study were large defect levels instead of direct band gaps. Therefore, these structures are incapable of  $\text{CO}_2$  photoredox catalysis, as previously proposed.

**3.3.  $\text{CO}_2$  and Monolayer Complexes.** Another important aspect of the  $\text{CO}_2$  catalysis is the adsorption of  $\text{CO}_2$  on the monolayer.  $\text{CO}_2$  chemisorption ( $E_{\text{ad}} \leq -0.3$  eV) is necessary for the reduction reaction to occur. We calculated the  $E_{\text{ad}}$  of  $\text{CO}_2$  on monolayers with a suitable band gap for photocatalysis and monolayers we expected to have good  $\text{CO}_2$  adsorption properties. The  $E_{\text{ad}}$ , adsorption type, and gas–monolayer distance ( $d$ ) of these monolayers are summarized in Table 3.

According to the results shown, modification of the pristine BN and BCN monolayers renders surface  $\text{CO}_2$  adsorption more energetically favorable. The unmodified  $4 \times 4\text{-BN}$  and  $2 \times 2\text{-BC}_6\text{N}$  monolayers have positive  $\text{CO}_2$   $E_{\text{ad}}$  values, indicating that  $\text{CO}_2$  cannot spontaneously adsorb onto these monolayers. These positive  $E_{\text{ad}}$  values agree with previous studies of these

**Table 3.  $\text{CO}_2$  Adsorption Energy  $E_{\text{ad}}$ , Adsorption Type, and Gas–Monolayer Distance ( $d$ ) of Selected Monolayers**

complex configuration	$E_{\text{ad}}$ (eV)	adsorption type	$d$ (Å)
$4 \times 4\text{-BN}$	0.004	no adsorption	3.71
$\text{N}_\text{v}-4 \times 4\text{-BN}$	−3.881	chemisorption	0.00
$\text{B}_{13}\text{C}_8\text{N}_{11}$	−0.037	physisorption	4.42
$\text{O}_\text{N}-\text{B}_{13}\text{C}_8\text{N}_{11}$ (CO detached)	−1.870	chemisorption	1.22
$\text{O}_\text{N}-\text{B}_{13}\text{C}_8\text{N}_{11}$ ( $\text{CO}_2$ adsorbed)	−0.041	physisorption	4.20
$\text{B}_{11}\text{C}_{12}\text{N}_9$	−0.007	physisorption	6.51
$2 \times 2\text{-BC}_6\text{N}$	0.061	no adsorption	3.21
$\text{O}_\text{C}-2 \times 2\text{-BC}_6\text{N}$	−1.801	chemisorption	1.44
$\text{Si}_\text{C}-2 \times 2\text{-BC}_6\text{N}$	−1.480	chemisorption	3.51



**Figure 3.** Relaxed atomic and band structure plots of (a)  $B_{13}C_8N_{11}$  with top-adsorbed  $CO_2$ , (b)  $O_N-B_{13}C_8N_{11}$  with detached CO, (c)  $O_N-B_{13}C_8N_{11}$  with adsorbed  $CO_2$ , (d)  $B_{11}C_{12}N_9$  with top-adsorbed  $CO_2$ , (e)  $Si_C-2 \times 2-BC_6N$  with top-adsorbed  $CO_2$ , and (f)  $N_V-4 \times 4-BN$  with  $CO_2$  adsorbed in site of vacancy. The Fermi level was set to 0 eV. B and Si atoms are colored pink, C atoms black, N atoms blue, and O atoms are shown in red.

two materials, which indicates that modification of these materials to facilitate adsorption could be a worthwhile strategy.<sup>34,44</sup>

The rest of the relaxed monolayers demonstrated adsorption (i.e.,  $E_{ad} < 0$ ) to  $CO_2$ . Their relaxed structures are shown in Figure 3, and Table 4 lists the band gap values and types of these complex structures. Unmodified  $B_{13}C_8N_{11}$  and  $B_{11}C_{12}N_9$  demonstrate low  $E_{ad}$  values of  $-0.037$  and  $-0.007$  eV, respectively. Thus, both monolayers can only physically adsorb  $CO_2$ , indicating that London dispersion forces may be the dominant interactions. Because physisorbent materials tend to show low selectivity for  $CO_2$ , they are unsuitable for carbon capture.

To study the adsorption strength of the  $O_N-B_{13}C_8N_{11}$  monolayer in more detail, we investigated two  $CO_2$  adsorption locations. Placing  $CO_2$  closer to the hexagonal center leads to a

**Table 4.** Band Gaps of  $CO_2$  Adsorbed on Selected Monolayers

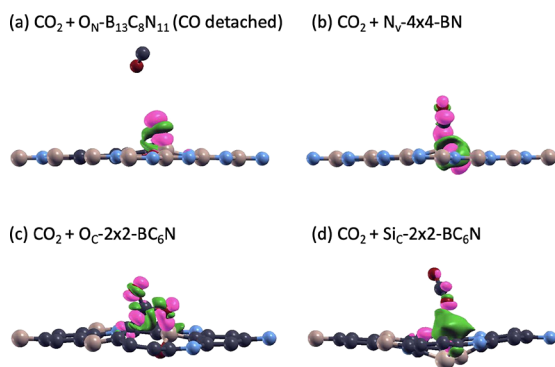
selected monolayer	band gap (eV)	band gap type
$N_V-4 \times 4-BN$	0.87	indirect
$B_{13}C_8N_{11}$	0	metallic
$O_N-B_{13}C_8N_{11}$ (CO detached)	2.42	direct
$O_N-B_{13}C_8N_{11}$ ( $CO_2$ adsorbed)	2.44	direct
$B_{11}C_{12}N_9$	0.04	indirect
$O_C-2 \times 2-BC_6N$	1.06	indirect
$Si_C-2 \times 2-BC_6N$	1.33	Indirect

strong attraction between the O atom and the C and B atoms ( $E_{ad} = -1.870$  eV), forming chemical bonds to the monolayer. This interaction yields a CO molecule that is only slightly

attracted to the monolayer, indicating that CO<sub>2</sub> was reduced to CO.

Next, we placed CO<sub>2</sub> farther from the monolayer in the vertical direction to study whether the entire molecule can be chemisorbed on the monolayer without reduction and, hence, serve as a CO<sub>2</sub> capture material. In this case, CO<sub>2</sub> exhibits only a weak physisorption of  $-0.041$  eV with the monolayer after relaxation, meaning that reduction is more energetically favorable than adsorption, given the ideal initial position. Among all selected monolayers, O<sub>N</sub>-B<sub>13</sub>C<sub>8</sub>N<sub>11</sub> (CO detached), O<sub>C</sub>-2 × 2-BC<sub>6</sub>N, Si<sub>C</sub>-2 × 2-BC<sub>6</sub>N, and N<sub>V</sub>-4 × 4-BN demonstrate chemisorption for CO<sub>2</sub>, with  $E_{ad} \leq -0.3$  eV.

Therefore, charge transfer in these cases demonstrating chemisorption was plotted using eq 1 to observe the change in electron density distribution caused by interactions between CO<sub>2</sub> and the monolayers. As shown in Figure 4, the pink and green regions represent charge accumulation and charge depletion, respectively. Charge is transferred from the green regions to the pink regions.



**Figure 4.** Charge transfer between CO<sub>2</sub> and selected monolayers demonstrated chemisorption with CO<sub>2</sub>. (a) O<sub>N</sub>-B<sub>13</sub>C<sub>8</sub>N<sub>11</sub> with detached CO (isovalue =  $6.75 \times 10^{-2}$  electron/Å<sup>3</sup>), (b) N<sub>V</sub>-4 × 4-BN (isovalue =  $6.75 \times 10^{-2}$  electron/Å<sup>3</sup>), (c) O<sub>C</sub>-2 × 2-BC<sub>6</sub>N (isovalue =  $6.75 \times 10^{-2}$  electron/Å<sup>3</sup>), and (d) Si<sub>C</sub>-2 × 2-BC<sub>6</sub>N (isovalue =  $1.35 \times 10^{-3}$  electron/Å<sup>3</sup>). B and Si atoms are shown in pink, C atoms are shown in black, N atoms are shown in blue, and O atoms are shown in red.

For the O<sub>N</sub>-B<sub>13</sub>C<sub>8</sub>N<sub>11</sub> with detached CO, as shown in Figure 4a, electrons were transferred from the O<sub>N</sub>-B<sub>13</sub>C<sub>8</sub>N<sub>11</sub> monolayer to the chemically bonded O atom. The charge accumulates in the region of newly formed B–O and C–O bonds, which cover the O atom, indicating strong chemisorption between the O atom and the monolayer. The CO molecule is located far from the monolayer, meaning that it exhibits a low attraction to the monolayer. This finding is consistent with the  $E_{ad}$  calculations and confirms the strong chemisorption between the O atom and the monolayer.

For N<sub>V</sub>-4 × 4-BN, shown in Figure 4b, there is considerable charge accumulation on the attached O atom of CO<sub>2</sub>, rendering the surrounding B atoms electron deficient. This observation is in accordance with the difference in electronegativity between B and O and confirms the chemical interaction between the bonded CO<sub>2</sub> gas and the N<sub>V</sub>-4 × 4-BN monolayer.

For O<sub>C</sub>-2 × 2-BC<sub>6</sub>N and Si<sub>C</sub>-2 × 2-BC<sub>6</sub>N, as shown in Figure 4c,d, O atoms tend to attract more charge than the other elements because of their high electronegativity. Thus, charge density accumulates more at the O atom that is closer to the

monolayer, leaving the corresponding monolayer region electron deficient.

Therefore, the charge transfers confirm the chemisorption between the monolayers and CO<sub>2</sub> and provide insights into the CO<sub>2</sub> reduction process.

**3.4. Applications.** Considering both the bandgap and  $E_{ad}$  of the monolayers, O<sub>N</sub>-B<sub>13</sub>C<sub>8</sub>N<sub>11</sub> and Si<sub>C</sub>-2 × 2-BC<sub>6</sub>N are suitable potential CO<sub>2</sub> reduction photocatalysts, adding to the relatively less explored nonmetallic photocatalysts. The reduction potential of O<sub>N</sub>-B<sub>13</sub>C<sub>8</sub>N<sub>11</sub> is suggested by the dissociation of the C–O bond in CO<sub>2</sub> during adsorption. However, bond dissociation occurs without visible light, indicating that this type of catalysis may not be a photodriven process. Nonetheless, detached CO can be further utilized. While the electrochemical conversion of CO<sub>2</sub> into single-carbon products can be controlled relatively easily, its direct conversion into C<sub>2+</sub> products is more environmentally friendly and commercially favorable.<sup>45</sup> As the direct conversion of CO<sub>2</sub> into C<sub>2+</sub> products currently suffers from unwanted side reactions and relatively low selectivity, this conversion can be achieved more effectively with CO as the key reaction intermediate.<sup>46</sup> Studies on electrochemical CO reduction demonstrate promising potential for producing vital short-chain C<sub>2+</sub> products, including ethylene, ethanol, acetic acid, and *n*-propanol, which can be converted to plastics, fuels, and feedstocks.

Although the other studied monolayers cannot serve as CO<sub>2</sub> photocatalysts, their band structures and adsorption abilities allow them to be used in other applications. For instance, the strong CO<sub>2</sub> chemisorption exhibited by O<sub>C</sub>-2 × 2-BC<sub>6</sub>N and N<sub>V</sub>-4 × 4-BN renders them good CO<sub>2</sub> capture materials. Monolayers with metallic characteristics and demonstrating CO<sub>2</sub> physisorption, such as B<sub>13</sub>C<sub>8</sub>N<sub>11</sub>, can be further developed into charge-regulated materials for CO<sub>2</sub> capture, as their properties are similar to those of other charge-regulated CO<sub>2</sub> capture materials.<sup>47</sup>

Many of these complex materials exhibited indirect band gaps after interacting with CO<sub>2</sub>, as seen in Table 4. Notably, the change in band gap will not hinder their previously discussed visible-light absorption properties, as the influence of CO<sub>2</sub> adsorption on the global band gap increases because CO<sub>2</sub> is arranged in a periodic manner in the conducted calculations. In reality, the concentration of CO<sub>2</sub> molecules adsorbed on these materials is much lower than that used in calculations, inducing a local change in the band structure rather than a global change. Therefore, the band gaps of these monolayers are indicative of their visible-light adsorption ability.

The band structures of the selected monolayer complexes are valuable for further studies. For instance, the 2.44 eV direct band gap of O<sub>N</sub>-B<sub>13</sub>C<sub>8</sub>N<sub>11</sub> (with adsorbed CO<sub>2</sub>) can be useful in fabricating optical devices, such as LEDs and semiconductor lasers. Complexes with small indirect band gaps may have potential in optoelectronic thin-film technologies.<sup>48</sup>

Although our research outlined the potential applications of the selected monolayers in these fields, further systematic theoretical studies should be implemented to determine the most ideal structural design for practical applications. Also, the specific placement of CO<sub>2</sub> on the monolayers used in this research might restrict the applicability of the materials when CO<sub>2</sub> is not ideally positioned, suggesting further investigation.

## 4. CONCLUSIONS

We used DFT calculations to investigate the atomic and electronic structures of BN- and BCN-based monolayers, as well

as their interactions with CO<sub>2</sub> gas. By modifying the structural configuration of the monolayers through doping and vacancy creation methods, we successfully tuned the band gaps of these monolayers and improved the adsorption of CO<sub>2</sub>. The calculated results indicated that O<sub>N</sub>-B<sub>13</sub>C<sub>8</sub>N<sub>11</sub> and Si<sub>C</sub>-2 × 2-BC<sub>6</sub>N are suitable CO<sub>2</sub> photocatalysts, whereas O<sub>C</sub>-2 × 2-BC<sub>6</sub>N and N<sub>V</sub>-4 × 4-BN are capable of CO<sub>2</sub> capture. Other monolayers capable of CO<sub>2</sub> adsorption possess favorable properties for utilization in optoelectronic materials. In summary, the novel metal-free materials proposed in this research lay a strong theoretical basis that facilitates the development of existing CCU approaches and inspires prospective work in this field.

As doping with O and Si atoms can increase the monolayers' attraction from no attraction or physisorption to chemisorption for CO<sub>2</sub>, future studies could explore the impact of other nonmetallic dopants or the concentration of dopants on the materials' electronic structure and interaction with CO<sub>2</sub>. The effect of stacking on CO<sub>2</sub> photoreduction can also be further investigated, as the current research only focused on monolayer materials, and real materials typically have stacking of layers that influence properties. Projected density of states (PDOS) calculations will also be helpful in studying the more detailed mechanism of the interaction between the monolayer and CO<sub>2</sub>. Finally, machine-learning interatomic potentials may be utilized in future calculations to better evaluate the materials' phononic properties, lattice thermal conductivity, and other electronic and mechanical features.<sup>49</sup>

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### Notes

The authors declare no competing financial interest.

## ACKNOWLEDGMENTS

We thank Dr. Qian for the computational support provided.

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