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OPEN Molecular dynamics simulations for mechanical properties of borophene: parameterization of valence force field model and Stillinger-Weber potential

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While most existing theoretical studies on the borophene are based on first-principles calculations, the present work presents molecular dynamics simulations for the lattice dynamical and mechanical properties in borophene. The obtained mechanical quantities are in good agreement with previous first-principles calculations. The key ingredients for these molecular dynamics simulations are the two efficient empirical potentials developed in the present work for the interaction of borophene with lowenergy triangular structure. The first one is the valence force field model, which is developed with the assistance of the phonon dispersion of borophene. The valence force field model is a linear potential, so it is rather efficient for the calculation of linear quantities in borophene. The second one is the Stillinger-Weber potential, whose parameters are derived based on the valence force field model. The Stillinger-Weber potential is applicable in molecular dynamics simulations of nonlinear physical or mechanical quantities in borophene.

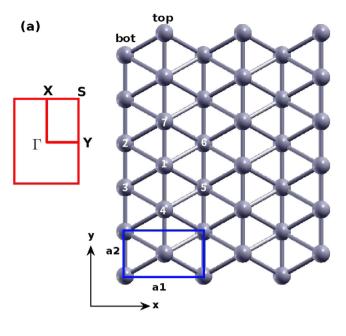
Boron can be formed into a number of finite clusters due to plenty of chemical bonding for this element¹. Some of these planar boron clusters were proposed as potential basis for the formation of single-layer boron (i.e. borophene)²⁻⁴. The structure of borophene on different substrates was predicted theoretically^{5,6}, and was produced in $recent\ experiments^{7-10}.$

Since then, intensive theoretical studies have been performed for various properties of the borophene. The stabilized structures of the borophene were investigated by first-principles calculations¹¹⁻¹⁷. The interaction between borophene and various types of substrates plays an important role for the stability and physical properties of the borophene^{18–25}. The electronic, optical and thermodynamic properties were investigated for the borophene²⁶. It was demonstrated that there is no Schottky barrier between the metallic borophene and other two-dimensional semiconductors, which is useful for the construction of two-dimensional electronic devices with enhanced performance²⁷. Borophene has negative Poisson's ratio in the out-of-plane direction and the in-plane directions resulting from its puckered configuration^{7,28–30}.

Besides the pure borophene, few approaches have been proposed to modify the electronic or phonon properties in borophene. For instance, hydrogenation can effectively tune the electronic current or other mechanical properties for the borophene 31-33. Free edges in the borophene nanoribbon were found to be important for mechanical, electronic, magnetic, and thermal transport properties^{34–36}. The strain effect has been studied for mechanical properties^{28,37,38}, or magnetic properties in borophene³⁹. Several defects were predicted to cause considerable effects on the anisotropic mechanical properties of the borophene³⁰.

Along with the study of fundamental physical properties for borophene, there have also been some investigations on possible applications of borophene in the applied research fields. For example, the application of borophene as high capacity electrodes or anode materials was examined by several recent works^{40,41}. First-principles

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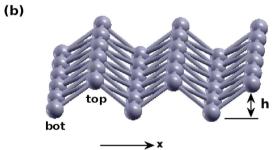


Figure 1. Structure for borophene. (a) Top view. Atoms are categorized into top chains and bottom chains. The top chain includes atoms like 1, 4, and 7. The bottom chain includes atoms like 2, 3, 5, and 6. The unit cell is shown by blue rectangle. The first Brillouin zone is shown by red rectangle on the left. (b) Perspective view illustrates the puckered configuration, with h as the distance between the top and bottom chains along the out-of-plane z-direction. The pucker is perpendicular to the x-axis and is parallel with the y-axis.

calculations predicted possible superconductivity phenomenon in borophene due to the phonon-electron interaction $^{42-44}$, which can be further manipulated by strain and carrier-doping 45 .

From the above literature survey, we find that most existing theoretical works are based on the first-principles calculations. These calculations are accurate, but are limited to small (or bulk) structures due to high computation requirements. As an alternative approach, the molecular dynamics simulation can be utilized to investigate very large systems typically containing more than 10^4 atoms. The key ingredient in the molecular dynamics simulation is the interatomic interaction. The only one interaction potential available for the borophene is the ReaxFF force field model⁴⁶. The present work aims to develop efficient linear and nonlinear empirical potentials for the borophene, which can assist further theoretical investigations for borophene of large size.

In this paper, we provide the valence force field (VFF) model and the Stillinger-Weber (SW) potential for the description of the interaction in borophene with low-energy triangular structure. The VFF model is a linear potential, which can be used to calculate elastic quantities such as the phonon dispersion. The SW potential is a nonlinear potential, which is derived based on the VFF model. The SW potential can be applied in the molecular dynamics simulation of nonlinear physical or mechanical properties for borophene. We demonstrate the usage of the SW potential with the publicly available LAMMPS package.

Results

Figure 1 shows the structure of borophene, in which structural parameters are from the *ab initio* calculation²⁸. Figure 1(a) shows the top view of the borophene with the rectangular unit cell in the xy plane. The two lattice bases are $a_1 = 2.866$ Å and $a_2 = 1.614$ Å. The first Brillouin zone is shown by red rectangle on the left. The two basic vectors for the reciprocal lattice are $b_1 = 2.192$ Å⁻¹ and $b_1 = 3.893$ Å⁻¹. There are two inequivalent boron atoms in the unit cell. Boron atoms are categorized into the top chain and the bottom chain. The top chain includes atoms like 1, 4, and 7. The bottom chain includes atoms like 2, 3, 5, and 6. Figure 1(b) discloses the puckered configuration of the borophene. The pucker is

VFF type		Bond st	Angle bending				
expression	$\frac{1}{2}K_{14}(\Delta r_{14})^2$	$\frac{1}{2}K_{12}(\Delta r_{12})^2$	$\frac{1}{2}K_{35}(\Delta r_{35})^2$	$\frac{1}{2}K_{47}(\Delta r_{47})^2$	$\frac{1}{2}K_{143}(\Delta\theta_{143})^2$	$\frac{1}{2}K_{135}(\Delta\theta_{135})^2$	
parameter 15.873		4.025	1.828	1.411	2.523	1.150	
r_0 or θ_0	1.614 1.880		2.866 3.228		64.581	99.318	

Table 1. The VFF model for borophene. The second line gives an explicit expression for each VFF term, where atom indexes are from Fig. 1 (a). The third line is the force constant parameters. Parameters are in the unit of $\frac{eV}{\hat{A}^2}$ for the bond stretching interactions, and in the unit of eV for the angle bending interaction. The fourth line gives the initial bond length (in unit of \hat{A}) for the bond stretching interaction and the initial angle (in unit of degrees) for the angle bending interaction. The angle θ_{ijk} has atom i as the apex.

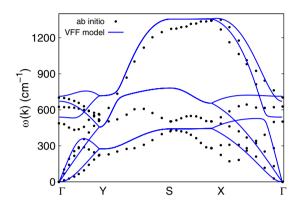


Figure 2. Phonon dispersion for borophene along Γ YSX Γ from the VFF model (blue lines) is compared to the data from the *ab initio* calculation (black dots)²⁸. Phonon branches are doubly degenerate along YX due to the inverse symmetry between the top and bottom boron chains.

perpendicular to the x-direction, while it is parallel with the y-direction. The height of the pucker is h = 0.911 Å, which is the distance between the top chain and the bottom chain along the out-of-plane z-direction.

Valence force field model. The VFF model is a useful linear model for the description of interatomic interactions in covalent materials, in which interactions are decomposed into some characteristic bond stretching and angle bending components⁴⁷. These interaction components are in close relation with the vibration morphology of phonon modes. It is thus a proper approach to determine the VFF model for a covalent material based on its phonon dispersion. There are two typical terms for the VFF model, i.e., the bond stretching V_r and the angle bending V_{θ} ,

$$V_r = \frac{1}{2} K_r (\Delta r)^2, \tag{1}$$

$$V_{\theta} = \frac{1}{2} K_{\theta} (\Delta \theta)^2, \tag{2}$$

where Δr and $\Delta \theta$ are the small variations for the bond length and the angle. The two force constant parameters are K_r and K_θ .

Table 1 shows five VFF terms for the borophene, four of which are the bond stretching interactions shown by Eq. (1) while the other two terms are the angle bending interactions shown by Eq. (2). The VFF model for the borophene is determined with the assistance of the phonon dispersion. The phonon dispersion for the borophene has been obtained by previous *ab initio* calculations. Parameters for the VFF model are determined by fitting the phonon dispersion to the *ab initio* calculations²⁸. GULP⁴⁸ is used for the computation of the phonon dispersion and the fitting of parameters in the VFF model.

Figure 2 shows that the phonon dispersion calculated from the VFF model is comparable with *ab initio* results. Good agreements are achieved for the three acoustic branches in the long wavelength limit. Similar phonon dispersion can be found from other *ab initio* calculations^{26,29,38}. There is no imaginary mode from the VFF model, which appears in the *ab initio* calculations for the flexure mode in the long wave limit^{26,28,29,38}. This imaginary mode is due to the long-range interaction in the borophene, which is not considered by the VFF model. Hence, there is no imaginary mode here. Figure 3 shows the vibration morphology of the three optical phonons at the Γ point. Each boron atom chain vibrates as a rigid chain in these phonons. The three phonon modes at the Y point are doubly degenerate, due to the inverse reflection symmetry between the top and bottom boron chains in the borophene. We have presented the vibration morphology for the three phonons at the Y point in Fig. 4, where only atoms from the top chains are involved in the vibration. For the other three phonons (not displayed

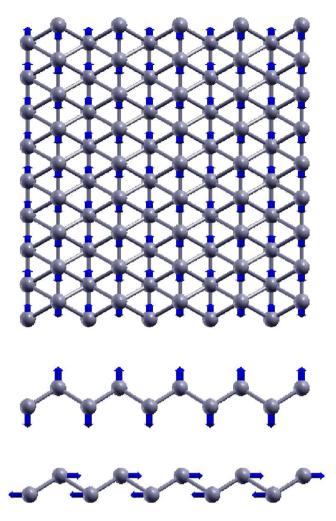


Figure 3. Three optical phonons at Γ point. From top to bottom, the frequencies are 539.2, 632.3, and 702.4 cm⁻¹, respectively. Each atom chain vibrates as a rigid chain in these phonons. Arrow attached to the atom represents the vibration component of this atom.

here), only atoms from the bottom chains are involved in the vibration. Actually, the phonon branches are doubly degenerate along the whole YX line, due to the inverse reflection symmetry in the borophene.

An obvious feature in the phonon dispersion of the borophene shown in Fig. 2 is that the highest-frequency branch (around $1300\,\mathrm{cm^{-1}}$) along SX has much higher frequency than the other branches. Figure 5 shows the vibration morphology of these phonons at the X point. The highest-frequency phonon is the intra-chain optical vibration; i.e., neighboring boron atoms within the same chain are vibrating in an out-of-phase manner. It means that the intra-chain bonds like r_{14} in Fig. 1(a) are much stronger than the inter-chain bonds like r_{12} . This anisotropy is reflected in the force constant parameters for the VFF model listed in Table 1. The intra-chain parameter K_{14} is larger than the inter-chain parameter K_{12} by a factor of four, so the resultant frequency for the intra-chain optical phonons (around $1300\,\mathrm{cm^{-1}}$) is about twice of the frequency for the inter-chain optical phonons (around $700\,\mathrm{cm^{-1}}$) shown in Fig. 2.

Parameterization of the SW potential. The VFF model presented in the previous section describes the linear component of the interaction within borophene, but it does not provide any information for the nonlinear interaction. The SW potential is one of the most efficient nonlinear potentials. The SW potential includes both linear and nonlinear interactions. One of the present authors (J.W.J.) has shown that the linear component of the SW potential can be derived analytically based on the VFF model⁴⁹. A distinct feature of this analytic approach is that the parameters of the SW potential are governed by a constraint, which guarantees that all bonds and angles are fully relaxed in the initial configuration. The constraint is shown as follows,

$$\rho = \frac{-4B(d - r_{\text{max}})^2}{(Bd - d^5)},\tag{3}$$

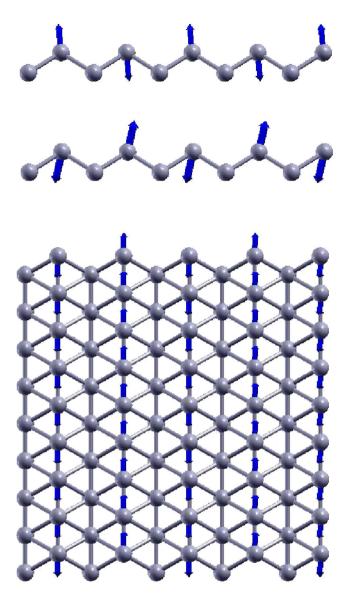


Figure 4. Three optical phonons at Y point. From top to bottom, the frequencies are 246.5, 386.1, and $589.4 \,\mathrm{cm}^{-1}$, respectively. Each atom chain vibrates as a rigid chain in these phonons. The other three phonons at Y point have similar vibration morphology, which involves only atoms from the bottom chains.

where *d* is the equilibrium bond length, and other quantities are potential parameters. The nonlinear component of the SW potential is determined by the parameter *B*, which can be determined by the third-order nonlinear constant.

The parameters for the two-body SW potential can be derived from the bond stretching terms in the VFF model. The obtained parameters are shown in Table 2. The parameters for the three-body SW potential are obtained from the angle bending terms in the VFF model. The parameters are shown in Table 3. Using this SW potential, we calculate the phonon dispersion for the borophene, which is the same as the phonon dispersion calculated using the VFF model as shown in Fig. 6. It indicates that the SW potential has inherited the linear interaction of the VFF model.

The publicly available package LAMMPS is widely used for MD simulations. The SW potential has a slightly different form in LAMMPS. The parameters for the SW potential can be determined by comparing the potential forms in GULP and LAMMPS. Table 4 shows some representative parameters for the SW potential used by LAMMPS. It should be noted that we have introduced eight atom types for boron atoms in the borophene as shown in Fig. 7. It is because the cut off (4.0 Å) for the two-body SW potential between atoms like 3–5 is larger than the distance between atoms 3–6 (3.289 Å). However, there is no interaction between atoms like 3–6. It is thus necessary to differentiate bonds like 3–5 and 3–6 with atoms 3 and 6 denoted by different atom types, so that the two-body SW potential is not imposed on the bonds like 3–6. There are 512 lines (categorized into six classes) in the SW potential script used by LAMMPS. We have shown in Table 4 one representative term for each of these six classes. These files for LAMMPS can be found in the Supplemental materials, including the SW potential script (borophene.sw), an input file for LAMMPS (in.tension), and a structure generation code (xyz.f90).

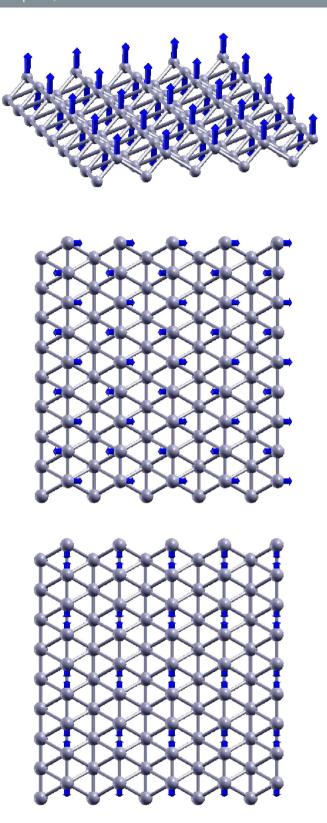


Figure 5. Three optical phonons at X point, involving vibration of atoms from the top chains. From top to bottom, the frequencies are 431.2,603.5, and $1323.9\,\mathrm{cm}^{-1}$, respectively. The frequency of the intra-chain longitudinal optical phonon ($1323.9\,\mathrm{cm}^{-1}$) is obviously higher than the two intra-chain transverse phonons. The other three phonons at X point have similar vibration morphology, which involves only atoms from the bottom chains.

Figure 8 shows the stress strain relations for the borophene of size 100×100 Å. The structure is uniaxially stretched in the x or y directions at 1 K and 300 K. The Young's modulus is $163 \, \mathrm{Nm^{-1}}$ and $394 \, \mathrm{Nm^{-1}}$ in the x and y directions respectively at 1 K, which are obtained by linear fitting of the stress strain relations in [0, 0.01]. These

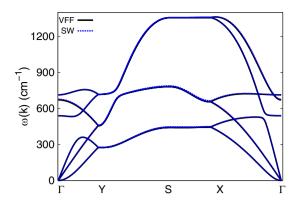


Figure 6. Phonon dispersion from the VFF model and the SW potential.

	A (eV)	ρ (Å)	B (Å4)	$r_{\min}(ext{Å})$	r _{max} (Å)
r_{14}	5.135	0.769	4.207	0.0	2.050
r_{12}	1.402	0.618	6.246	0.0	2.419
r ₃₅	2.957	2.065	36.096	0.0	4.000
r ₄₇	1.683	1.413	65.146	0.0	4.100

Table 2. Two-body SW potential parameters used by GULP⁴⁸. The expression is $V_2 = Ae^{[\rho/(r-r_{max})]}(B/r^4 - 1)$. The quantity (r_{ij}) in the first line lists one representative term for the two-body SW potential between atoms i and j. Atom indexes are from Fig. 1(a).

	K(eV)	θ_0 (degree)	ρ1 (Å)	ρ2 (Å)	$r_{\min 12}$ (Å)	r _{max12} (Å)	$r_{\min 13}$ (Å)	r _{max13} (Å)	$r_{\min 23}$ (Å)	r _{max23} (Å)
θ_{143}	28.382	64.581	0.769	0.618	0.0	2.050	0.0	2.419	0.0	2.419
θ_{135}	5.852	99.318	0.618	0.618	0.0	2.419	0.0	2.419	2.240	3.047

Table 3. Three-body SW potential parameters used by GULP⁴⁸. The expression is $V_3 = Ke^{[\rho_1/(r_{12} - r_{max12}) + \rho_2/(r_{13} - r_{max13})]}$ (cos θ – cos θ_0)². The first line (θ_{ijk}) presents one representative term for the three-body SW potential. The angle θ_{ijk} has the atom i as the apex. Atom indexes are from Fig. 1(a).

	ε (eV)	σ (Å)	а	λ	γ	$\cos \theta_0$	A	BL	p	q	tol
B ₁ -B ₅ -B ₅	1.000	0.769	2.666	0.000	1.000	0.000	5.135	12.030	4	0	0.0
B ₁ -B ₂ -B ₂	1.000	0.618	3.914	0.000	1.000	0.000	1.402	42.820	4	0	0.0
B ₁ -B ₃ -B ₃	1.000	2.065	1.937	0.000	1.000	0.000	2.957	1.985	4	0	0.0
$B_1-B_1-B_1$	1.000	1.413	2.902	0.000	1.000	0.000	1.683	16.343	4	0	0.0
B ₁ -B ₈ -B ₅	1.000	0.000	0.000	28.382	0.000	0.429	0.000	0.000	0	0	0.0
B ₁ -B ₈ -B ₆	1.000	0.000	0.000	5.852	0.000	-0.162	0.000	0.000	0	0	0.0

Table 4. SW potential parameters used by LAMMPS⁵². The two-body potential expression is $V_2 = \epsilon A (B_L \sigma^P r_{ij}^{-P} - \sigma^q r_{ij}^{-q}) e^{[\sigma(r_{ij} - a\sigma)^{-1}]}$. The three-body potential expression is $V_3 = \epsilon \lambda e^{[\gamma \sigma (r_{ij} - a\sigma)^{-1} + \gamma \sigma (r_{jk} - a\sigma)^{-1}]}$ (cos θ_{jik} – cos θ_0)². Atom types in the first column are displayed in Fig. 7.

values are in good agreement with the *ab initio* results at 0 K temperature, eg. $170 \,\mathrm{Nm^{-1}}$ and $398 \,\mathrm{Nm^{-1}}$ in ref. 7, or $166 \,\mathrm{Nm^{-1}}$ and $389 \,\mathrm{Nm^{-1}}$ in ref. 28, or $163 \,\mathrm{Nm^{-1}}$ and $399 \,\mathrm{Nm^{-1}}$ in ref. 50. Figure 9 shows that the Young's modulus decreases with the increase of temperature. Previous *ab initio* calculations obtained negative Poisson's ratio for the uniaxial stretching of the borophene in the x and y directions, eg. $-0.02 \,\mathrm{and} -0.04 \,\mathrm{in}$ refs 7,28. The Poisson's ratio from the present SW potential are $-0.011 \,\mathrm{and} -0.029$, which are quite comparable with the *ab initio* results.

The yielding strain is about 0.15 and 0.1 in the x and y directions respectively at the low temperature of 1 K. These values agree with *ab initio* results at 0 K^{28,38,50}. The yielding strain decreases with increasing temperature. The nonlinear parameter $B=0.6d^4$, with d as the bond length, is determined by the third-order nonlinear constant (D) along the x-direction. The third-order nonlinear constant is obtained by fitting the stress strain relation to the function $\sigma=E\epsilon+\frac{1}{2}D\epsilon^2$, with E as the Young's modulus. The obtained value of D is $-944\,\mathrm{Nm}^{-1}$ along the x-direction, which is fitted to the *ab initio* result of ref. $28-924\,\mathrm{Nm}^{-1}$. The obtained value of D is $-2785\,\mathrm{Nm}^{-1}$ in the y direction.

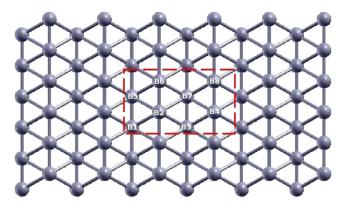


Figure 7. Eight atom types are introduced for the boron atoms in the borophene. Atoms B_1 , B_3 , B_5 , and B_7 are from the bottom chain. Atoms B_2 , B_4 , B_6 , and B_8 are from the top chain.

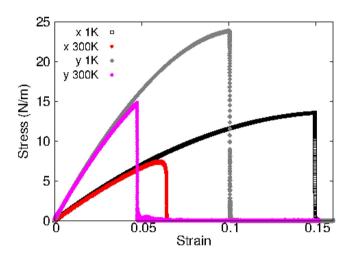


Figure 8. Stress-strain relations for the borophene of size 100×100 Å. The borophene is uniaxially stretched along the x or y directions at temperatures 1 K and 300 K.

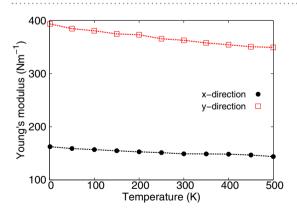


Figure 9. The temperature dependence for the Young's modulus in the borophene along the armchair and zigzag directions.

The SW potential is not able to describe accurately the complicate chemical bonding processes. However, it contains reasonably accurate nonlinear components, so it is able to provide some qualitative descriptions for the thermal-induced nonlinear phenomena. As an example, Fig. 10 shows the total potential energy in the borophene at varying temperatures. A sudden jump at the critical temperature $T_C = 550 \pm 50 \, \mathrm{K}$ reveals that the borophene is stable at temperatures bellow the critical temperature T_C . However, the structure becomes unstable for temperatures above the critical temperature. Figure 11 shows the destruction process for the borophene at 600 K. Some boron atoms have obviously larger thermal vibration amplitudes at $t = 290 \, \mathrm{ps}$. These boron atoms will be

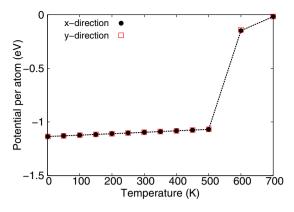


Figure 10. Potential energy per atom in borophene at different temperatures. A sudden jump at the critical temperature $T_C = 550 \pm 50$ indicates the instability of the borophene above T_C .

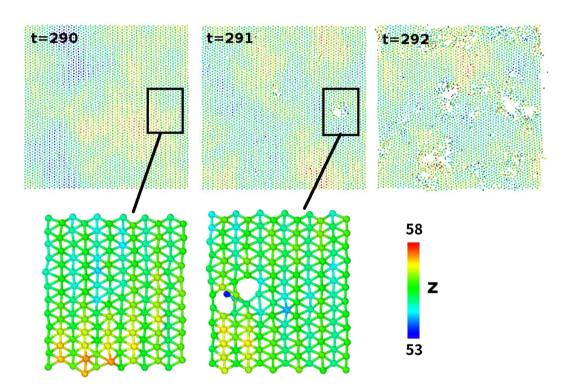


Figure 11. The destruction process for the borophene at 600 K. The colorbar is with respective to the atomic z-coordinate.

evaporated at t = 291 ps, which results in a vacancy defect in the borophene. The whole borophene is destroyed when there are too many vacancy defects at t = 292 ps.

We note that the VFF model and the SW potential parameterized in the present work are only applicable to describe the interaction for the triangular phase of the borophene, but can not be applied to other phases for the borophene.

Discussion

To summarize, we have developed two empirical potentials to describe the interaction between boron atoms within borophene. The first one is the linear VFF model, which is determined based on the phonon dispersion of the borophene. The VFF model is suitable for the calculation of linear quantities in borophene. Starting from the VFF model, we derive the second empirical potential, the SW potential, to describe the interaction for borophene. The SW potential includes both linear and nonlinear couplings, so it is applicable for the simulation of some nonlinear quantities. We also provide some necessary files for the application of the SW potential by using LAMMPS.

Method

One minor improvement for the implementation of three-body SW potential in LAMMPS. We discuss a practical improvement for the three-body SW potential implemented in LAMMPS. Let's take angles θ_{143} and θ_{137} in Fig. 1 as an explicit example to demonstrate this improvement.

It is obvious that these two angles should be quite different. As shown in Table 3, there is a three-body SW potential for θ_{143} , but there is no interaction for θ_{137} . The values of these two angles are quite different in the equilibrium configuration; i.e., $\theta_{143}^0 = 64.581^\circ$ and $\theta_{137}^0 = 115.419^\circ$. However, their arm lengths are the same; i.e., $r_{13} = r_{13}$ and $r_{14} = r_{17}$. In the three-body SW potential implemented in LAMMPS, angles are distinguished by their two arm lengths, so angles θ_{143} and θ_{137} become indistinguishable. Consequently, the three-body SW potential (intent solely) for θ_{143} will also be applied to θ_{137} , which causes a technical issue.

In general, it will become important to distinguish two angles constructed by the same type of atoms, when there are more than two different types angles around each atom. Similar situation also occurs in the simulation of MoS_2 using the three-body SW potential implemented in LAMMPS. We have proposed to distinguish each angle by its opposite arm length^{49,51}. That is, the bonds r_{34} and r_{37} are quite different in length, so they can be used to distinguish angles θ_{143} and θ_{137} . However, this bond length based criterion is an ad hoc technique, as the bond length is a structural dependent parameter. We note that this criterion is also adopted by GULP⁴⁸.

Here, we provide a more universal criterion to distinguish two angles. It is based on the input initial value θ_0 for the angle of the three-body SW potential. For example, in Table 3, the input initial value for the three-body SW potential regarding θ_{143} is $\theta_{143}^0 = 64.581^\circ$. For a given angle θ , if the following condition is satisfied, then the three-body SW potential will be applied to this angle,

$$\left|\cos\theta - \cos\theta_{143}^{\,0}\right| < 0.3. \tag{4}$$

It can be checked that θ_{137} does not obey the criterion in Eq. (4) and it is thus excluded from the three-body SW potential. The angle θ_{143} obeys the criterion in Eq. (4) under moderate deformations. Hence, the three-body SW potential is applied to the angle θ_{143} , while this three-body SW potential is not applied to the other angle θ_{137} .

The criterion in Eq. (4) can be implemented into LAMMPS by modifying its pair_sw.cpp source file in the following two steps. A continuous cut-off function in the range of [0.25, 0.35] has been introduced to avoid possible boundary effects in molecular dynamics simulations.

(1). First, find the following line in the pair_sw.cpp source file,

```
delcs = cs - paramijk - > costheta;
```

(2). Second, insert the following new lines after the above line,

```
 if(fabs(delcs) <= 0.25) \\ \{ \\ delcs = delcs; \\ \} \\ else if(fabs(delcs) < 0.35) \\ \{ \\ delcs = delcs * (0.5 + 0.5*sin(3.142*(delcs - 0.25)/(0.35 - 0.25) + 0.5*3.142)); \\ \} \\ else \\ \{ \\ delcs = 0.0; \\ \}
```

Then recompile the LAMMPS package. The recompiled LAMMPS executable file can be used to simulate borophene with the SW potential parameterized in the present work. It can also be used to simulate MoS_2 correctly using its SW potential. We note that this criterion does not affect other normal simulations using LAMMPS, where it is not necessary to distinguish angles. We expect this criterion to be implemented in LAMMPS in its future versions.

References

- 1. Zubarev, D. Y. & Boldyrev, A. I. Comprehensive analysis of chemical bonding in boron clusters. *Journal of computational chemistry* **28,** 251–268 (2007).
- 2. Piazza, Z. A. et al. Planar hexagonal b36 as a potential basis for extended single-atom layer boron sheets. Nature communications 5, 3113 (2014)
- 3. Li, W.-L. et al. The b35 cluster with a double-hexagonal vacancy: a new and more flexible structural motif for borophene. Journal of the American Chemical Society 136, 12257–12260 (2014).
- 4. Chen, Q. et al. Experimental and theoretical evidence of an axially chiral borospherene. ACS nano 9, 754-760 (2014).
- 5. Liu, Y., Penev, E. S. & Yakobson, B. I. Probing the synthesis of twodimensional boron by firstprinciples computations. *Angewandte Chemie* **52**, 3156–3159 (2013).
- 6. Zhang, Z., Yang, Y., Gao, G. & Yakobson, B. I. Twodimensional boron monolayers mediated by metal substrates. *Angewandte Chemie* 127, 13214–13218 (2015).
- 7. Mannix, A. J. et al. Synthesis of borophenes: Anisotropic, two-dimensional boron polymorphs. Science 350, 1513-1516 (2015).

- 8. Tai, G. et al. Synthesis of atomically thin boron films on copper foils. Angewandte Chemie 127, 15693-15697 (2015).
- 9. Feng, B. et al. Experimental realization of two-dimensional boron sheets. Nature Chemistry 8, 563-568 (2016).
- 10. Feng, B. et al. Direct evidence of metallic bands in a monolayer boron sheet. Physical Review B 94, 041408 (2016).
- 11. Lau, K. C., Pati, R., Pandey, R. & Pined, A. C. First-principles study of the stability and electronic properties of sheets and nanotubes of elemental boron. *Chemical Physics Letters* **418**, 549–554 (2006).
- 12. Kunstmann, J. & Quandt, A. Broad boron sheets and boron nanotubes: An ab initio study of structural, electronic, and mechanical properties. *Physical Review B* 74, 035413 (2006).
- Lau, K. C. & Pandey, R. Stability and electronic properties of atomistically-engineered 2d boron sheets. J. Phys. Chem. C 111, 2906–2912 (2007).
- 14. Yang, X., Ding, Y. & Ni, J. Ab initio prediction of stable boron sheets and boron nanotubes: Structure, stability, and electronic properties. *Physical Review B* 77, 041402 (2008).
- Zope, R. R. & Baruah, T. Snub boron nanostructures: Chiral fullerenes, nanotubes and planar sheet. Chemical Physics Letters 501, 193–196 (2011).
- 16. Yu, X., Li, L., Xu, X.-W. & Tang, C.-C. Prediction of two-dimensional boron sheets by particle swarm optimization algorithm. *J. Phys. Chem. C* 116, 20075–20079 (2012).
- 17. Wu, X. et al. Two-dimensional boron monolayer sheets. ACS Nano 6, 7443-7453 (2012).
- 18. Zhang, L. Z., Yan, Q. B., Du, S. X., Su, G. & Gao, H.-J. Boron sheet adsorbed on metal surfaces: Structures and electronic properties. J. Phys. Chem. C 116, 18202–18206 (2012).
- 19. Edgington, R. et al. Growth and electrical characterisation of doped boron layers on (111) diamond surfaces. *Journal of Applied Physics* 111, 033710 (2012).
- 20. Liu, H., Gao, J. & Zhao, J. From boron cluster to two-dimensional boron sheet on cu(111) surface: Growth mechanism and hole formation. Sci. Rep. 3, 3238 (2013).
- 21. Shu, H., Li, F., Liang, P. & Chen, X. Nveiling the atomic structure and electronic properties of atomically thin boron sheets on an ag (111) surface. *Nanoscale* 8, 16284–16291 (2016).
- 22. Xu, S., Zhao, Y., Liao, J., Yang, X. & Xu, H. The nucleation and growth of borophene on the ag (111) surface. Nano Research 9, 2616 (2016).
- 23. Zhang, Z. et al. Substrate-induced nanoscale undulations of borophene on silver. Nano Letters 16, 6622-6627 (2016).
- 24. Tsai, H.-S. et al. Fabrication of multilayer borophene on insulator structure. Small 12, 5251-5255 (2016).
- 25. Xu, S.-G., Zhao, Y.-J., Liao, J.-H., Yang, X.-B. & Xu, H. The formation of boron sheet at the ag(111) surface: From clusters, ribbons, to monolayers. *Preprint at* http://arxiv.org/abs/1601.01393 (2016).
- Peng, B. et al. The electronic, optical, and thermodynamic properties of borophene from first-principles calculations. Journal of Materials Chemistry C 4, 3592–3598 (2016).
- Liu, L. Z., Xiong, S. J. & Wu, X. L. Monolayer borophene electrode for effective elimination of both the schottky barrier and strong electric field effect. Applied Physics Letters 109, 061601 (2016).
- 28. Wang, H. et al. Strain effects on borophene: ideal strength, negative possion's ratio and phonon instability. New Journal of Physics 18, 73016–73022 (2016).
- 29. Sun, H., Li, Q. & Wan, X. G. First-principles study of thermal properties of borophene. *Physical Chemistry Chemical Physics* 18, 14927–14932 (2016).
- 30. Wang, V. & Geng, W. T. Lattice defects and the mechanical anisotropy of borophene. *Preprint at* http://arxiv.org/abs/1607.00642 (2016).
- 31. Xu, L., Du, A. & Kou, L. Hydrogenated borophene as a stable two-dimensional dirac material with ultrahigh fermi velocity. *Physical Chemistry Chemical Physics*, doi: 10.1039/C6CP05405F (2016).
- 32. Padilha, J. E., Miwab, R. H. & Fazziocd, A. Directional dependence of the electronic and transport properties of 2d borophene and borophane. *Phys. Chem. Chem. Phys.* **18**, 25491–25496 (2016).
- 33. Wang, Z.-Q., Lu, T.-Y., Wang, H.-Q., Feng, Y.-P. & Zheng, J.-C. High anisotropy of fully hydrogenated borophene. *Preprint at http://arxiv.org/abs/1608.04788* (2016).
- 34. Liu, Y. et al. Stable and metallic borophene nanoribbons from first-principles calculations. *Journal of Materials Chemistry C* 4, 6380–6385 (2016).
- 35. Zhang, Z., Xie, Y., Peng, Q. & Chen, Y. Phonon transport in single-layer boron nanoribbons. Nanotechnology 27, 445703 (2016).
- 36. Meng, F., Chen, X. & He, J. Electronic and magnetic properties of borophene nanoribbons. *Preprint at http://arxiv.org/abs/1601.05338* (2016).
- 37. Mortazavi, B., Rahaman, O., Dianat, A. & Rabczuk, T. Mechanical responses of borophene sheets: A first-principles study. *Physical Chemistry Chemical Physics*, doi: 10.1039/C6CP03828J (2016).
- 38. Pang, Z., Qian, X., Yang, R. & Wei, Y. Super-stretchable borophene and its stability under straining. *Preprint at http://arxiv.org/abs/1602.05370 (2016)*.
- 39. Zabolotskiy, A. D. & Lozovik, Y. E. Strain-induced pseudomagnetic field in dirac semimetal borophene. *Preprint at* http://arxiv.org/abs/1607.02530 (2016).
- 40. Shi, L., Zhao, T., Xu, A. & Xu, J. Ab initio prediction of borophene as an extraordinary anode material exhibiting ultrafast directional sodium diffusion for sodium-based batteries. *Science Bulletin* 16, 1138–1144 (2016).
- 41. Zhang, X. et al. Borophene as an extremely high capacity electrode material for li-ion and na-ion batteries. *Preprint at* http://arxiv.org/abs/1608.01892 (2016).
- 42. Gao, M., Li, Q.-Z., Yan, X.-W. & Wang, J. Prediction of phonon-mediated superconductivity in borophene. *Preprint at http://arxiv.org/abs/1602.02930 (2016)*.
- 43. Penev, E. S., Kutana, A. & Yakobson, B. I. Can two-dimensional boron superconduct? Nano letters 16, 2522-2526 (2016).
- 44. Zhao, Y., Zeng, S. & Ni, J. Phonon-mediated superconductivity in borophenes. Applied Physics Letters 108, 242601 (2016).
- 45. Xiao, R. C. et al. Preprint at http://arxiv.org/abs/1604.06519 (2016).
- 46. Le, M. Q., Mortazavi, B. & Rabczuk, T. Mechanical properties of borophene films: a reactive molecular dynamics investigation. *Nanotechnology* 27, 445709 (2016).
- 47. Yu, P. Y. Fundamentals of Semiconductors (Springer, New York, 2010).
- 48. Gale, J. D. Gulp: A computer program for the symmetry-adapted simulation of solids. *J. Chem. Soc., Faraday Trans.* **93**, 629–637. Code available from https://projects.ivec.org/gulp/ (1997).
- 49. Jiang, J.-W. Parametrization of stillinger-weber potential based on valence force field model: Application to single-layer mos₂ and black phosphorus. *Nanotechnology* **26**, 315706 (2015).
- 50. Zhang, Z., Yang, Y., Penev, E. S. & Yakobson, B. I. Elasticity, flexibility and ideal strength of borophenes. *Preprint at* http://arxiv.org/abs/1609.07533 (2016).
- 51. Jiang, J.-W., Park, H. S. & Rabczuk, T. Molecular dynamics simulations of single-layer molybdenum disulphide (mos₂): Stillinger-weber parametrization, mechanical properties, and thermal conductivity. *Journal of Applied Physics* **114**, 064307 (2013).
- 52. lammps. http://www.cs.sandia.gov/~sjplimp/lammps.html (2012).

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Author Contributions

Y.P.Z. performed some calculations. J.W.J. designed the project and wrote the paper. All authors discussed the results and commented on the manuscript.

Additional Information

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