

Article

Influence of Isostatic Pressure on the Elastic and Electronic Properties of $K_2SiF_6:Mn^{4+}$

Mekhrdod Subhoni^{1,2,3,4,*} , Umar Zafari³, Chong-Geng Ma¹, Alok M. Srivastava⁵, William W. Beers⁵, William E. Cohen⁵ , Mikhail G. Brik^{1,6,7,8,*}, Michal Piasecki^{7,9} and Tomoyuki Yamamoto^{2,10,11}

- ¹ College of Sciences & CQUP-T-BUL Innovation Institute, Chongqing University of Posts and Telecommunications, Chongqing 400065, China; cgma.ustc@gmail.com
- ² Kagami Memorial Research Institute for Materials Science and Technology, Waseda University, Tokyo 169-0051, Japan; tymmt@waseda.jp
- ³ Center of Innovative Development of Science and New Technologies, National Academy of Sciences of Tajikistan, Dushanbe 734025, Tajikistan; zafari_umar@mail.ru
- ⁴ Physical Technical Institute, National Academy of Sciences of Tajikistan, Dushanbe 734063, Tajikistan
- ⁵ Current Lighting Solutions LLC, 1099 Ivanhoe Road, Cleveland, OH 44110, USA; srivastaam@outlook.com (A.M.S.); william.beers@gecurrent.com (W.W.B.); bill.cohen@gecurrent.com (W.E.C.)
- ⁶ Institute of Physics, University of Tartu, W. Ostwald Str. 1, 50411 Tartu, Estonia
- ⁷ Faculty of Science and Technology, Jan Długosz University, Armii Krajowej 13/15, PL-42200 Częstochowa, Poland; m.piasecki@ujd.edu.pl
- ⁸ Academy of Romanian Scientists, Ilfov Str. No. 3, 050044 Bucharest, Romania
- ⁹ Inorganic Chemistry Department, Uzhhorod National University, Pidhirna Str. 46, 88000 Uzhhorod, Ukraine
- ¹⁰ Faculty of Science and Engineering, Waseda University, Tokyo 169-8555, Japan
- ¹¹ Institute of Condensed-Matter Science, Waseda University, Tokyo 169-8555, Japan
- * Correspondence: mehrdodq@gmail.com (M.S.); mikhail.brik@ut.ee (M.G.B.)



Citation: Subhoni, M.; Zafari, U.; Ma, C.-G.; Srivastava, A.M.; Beers, W.W.; Cohen, W.E.; Brik, M.G.; Piasecki, M.; Yamamoto, T. Influence of Isostatic Pressure on the Elastic and Electronic Properties of $K_2SiF_6:Mn^{4+}$. *Materials* **2022**, *15*, 613. <https://doi.org/10.3390/ma15020613>

Academic Editor: Jonas Joos

Received: 10 December 2021

Accepted: 11 January 2022

Published: 14 January 2022

Publisher's Note: MDPI stays neutral with regard to jurisdictional claims in published maps and institutional affiliations.



Copyright: © 2022 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (<https://creativecommons.org/licenses/by/4.0/>).

Abstract: Isostatic pressure effects on the elastic and electronic properties of non-doped and Mn^{4+} -doped K_2SiF_6 (KSF) have been investigated by first-principles calculations within density functional theory (DFT). Bulk modulus was obtained by the Murnaghan's equation of states (EOS) using the relationship between volume and pressures at pressures between 0 and 40 GPa, and elastic constants were calculated by the stress-strain relationship giving small distortions at each pressure point. The other elastic parameters such as shear modulus, sound velocity and Debye temperature, which can be obtained from the elastic constants, were also estimated. The influence of external isostatic pressure on the electronic properties, such as crystal field strength $10Dq$ and emission energy of ${}^2E \rightarrow {}^4A^2$ transition (E_{em}), of $KSF:Mn^{4+}$ was also studied. The results suggest that $10Dq$ and E_{em} linearly increase and decrease, respectively, with increasing pressure.

Keywords: K_2SiF_6 ; Mn^{4+} ; isostatic pressure; interionic distances; bulk modulus; elastic constant; Debye temperature; crystal field strength $10Dq$; emission energy

1. Introduction

Mn^{4+} -doped phosphors as red photon generating systems have been extensively studied over the past decades due to their great promise for use in LED devices [1–7]. Among a large number of Mn^{4+} activated red-emitting phosphors, $K_2SiF_6:Mn^{4+}$ (KSF: Mn^{4+}) has been developed as generators of red photons in phosphor-converted white LEDs (pc-LEDs). The sharp-line emission of KSF: Mn^{4+} peaks at about 630 nm, where the human eye sensitivity to red light is still quite high. With minimum emission beyond 650 nm, the emission spectrum of KSF: Mn^{4+} is suitable for supplying the red photons necessary to produce white light with high efficacy (lumens per watt) and color rendering index (CRI) in pc-LEDs. Given its commercial importance, detailed analyses of the spectroscopic properties, such as absorption and emission spectra, electronic and geometric properties of KSF: Mn^{4+} were performed [8–10] and recently this red phosphor was commercialized [11].

The optical properties of the Mn^{4+} ions (with the $3d^3$ electron configuration) are strongly influenced by the composition and crystal structure of the host lattice. In the Tanabe–Sugano diagram for the d^3 electron configuration, the “ Mn^{4+} -ligand” bonding covalence defines the emission energy of the ${}^2E \rightarrow {}^4A_2$ spin-forbidden transition and the strength of the octahedral crystal field denoted as $10Dq$. It was determined by DFT calculations that weak Mn^{4+} -ligand hybridization generally leads to higher Mn^{4+} emission energies [12]. The hybridization between the Mn^{4+} ions and ligands is related to the Mn^{4+} -ligand distances. The application of pressure is one of the state parameters that can change the peak energy and $10Dq$ by reducing the inter-atomic distances. Studies of the Mn^{4+} optical properties under external pressure have been reported in the archival literature [13–15]. The goal of this study was to quantitatively explore by first-principles calculations within density functional theory, the electronic, elastic and optical properties of K_2SiF_6 in the absence and presence of the activator ion (Mn^{4+}).

2. Computational Method

All the density functional calculations in this paper were performed by the plane-wave basis projector augmented wave package, VASP [16], using the generalized gradient approximation proposed by Perdew, Burke and Ernzerhof (GGA-PBE) [17] to express electron–electron correlation. The space group of KSF is Fm-3m with a lattice constant of 8.134 Å [18], in which the Si^{4+} ions are surrounded by six F^- ions and the local site symmetry is described by the O_h point group. The Mn^{4+} -doped KSF models were constructed by replacing one Si^{4+} ion by one Mn^{4+} ion in the unit cell of KSF, which includes 36 atoms corresponding to four formula units. After careful convergence tests with respect to the plane wave cut-off energy, 900 eV was selected for the energy cut-off in all the calculations. The Monkhorst–Pack k-points grid sampling [19] was set as $4 \times 4 \times 4$. The structural parameters of non-doped KSF and $\text{KSF}:\text{Mn}^{4+}$ were optimized by allowing relaxations of the lattice constants and internal atomic positions under isostatic external pressures between 0 and 40 GPa by a 5 GPa step.

3. Results and Discussion

The calculated and experimental lattice constants of non-doped KSF and $\text{KSF}:\text{Mn}^{4+}$ are summarized in Table 1 together with the calculated Si–F and Mn–F bond lengths. The experimental lattice constant of non-doped KSF, $a = 8.134$ Å [18], is reproduced well by the current calculations, $a = 8.336$ Å, within a typical overestimation due to the GGA-PBE functional use. It is noted here that a slight volume expansion occurs after inclusion of the Mn^{4+} ions into KSF host. The calculated lattice constant of $\text{KSF}:\text{Mn}^{4+}$ is 8.357 Å, which is larger than that of KSF 8.336 Å. This expansion can be simply explained by the difference in ionic radii of Si^{4+} (0.40 Å) and Mn^{4+} (0.53 Å). The calculated Mn^{4+} – F^- bond-length is also larger than the calculated Si^{4+} – F^- bond-length by 6.69%, which supports the earlier experimental result of +7.43% [20].

Table 1. Comparison of lattice constants, a , and bond lengths of Si–F and Mn–F in the non-doped and Mn^{4+} -doped K_2SiF_6 , respectively, between experiments and calculations.

System		Calc. (Å)	Exp. (Å)
K_2SiF_6	a	8.336	8.134 ^a
	Si–F	1.720	1.683 ^a
$\text{K}_2\text{SiF}_6:\text{Mn}^{4+}$	a	8.357	
	Mn–F	1.835	1.807 ^b

^a Ref. [18]. ^b Ref. [20].

The calculated pressure dependence of the relative volume change V/V_0 for non-doped KSF is plotted in Figure 1a, which is fitted to the Murnaghan's equation of state (EOS) [21] expressed by:

$$V/V_0 = \left(1 + P \frac{B'}{B}\right)^{-\frac{1}{B'}} \quad (1)$$

where V and V_0 are volumes at pressure P and ambient pressure, respectively, and B and B' are bulk modulus and its pressure derivative, respectively. This fitting yields that B and B' are 20.01 GPa and 4.68, respectively. Our results agree well with the earlier study [22] using GGA-PBE, $B = 21.79$ GPa and $B' = 4.47$, although a slight difference appears between these two from a difference in calculating methods, i.e., projector augmented wave and pseudopotential methods in the current and earlier [22] studies, respectively. To the best of our knowledge, no experimental bulk modulus for KSF has been reported yet. For Mn^{4+} -doped KSF, V/V_0 is also plotted in Figure 1b, which is also fitted to the Murnaghan's EOS. The fitting yields $B = 19.84$ GPa and $B' = 4.68$, which implies that inclusion of Mn^{4+} into KSF leads to a decrease of the bulk modulus.

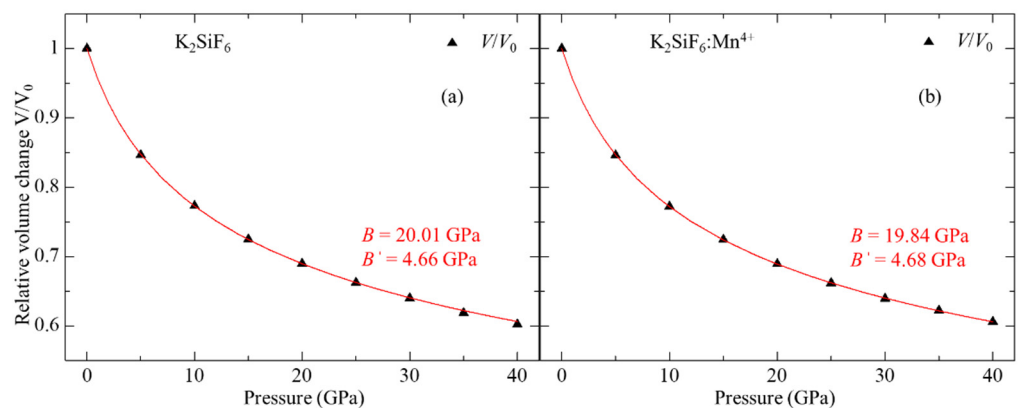
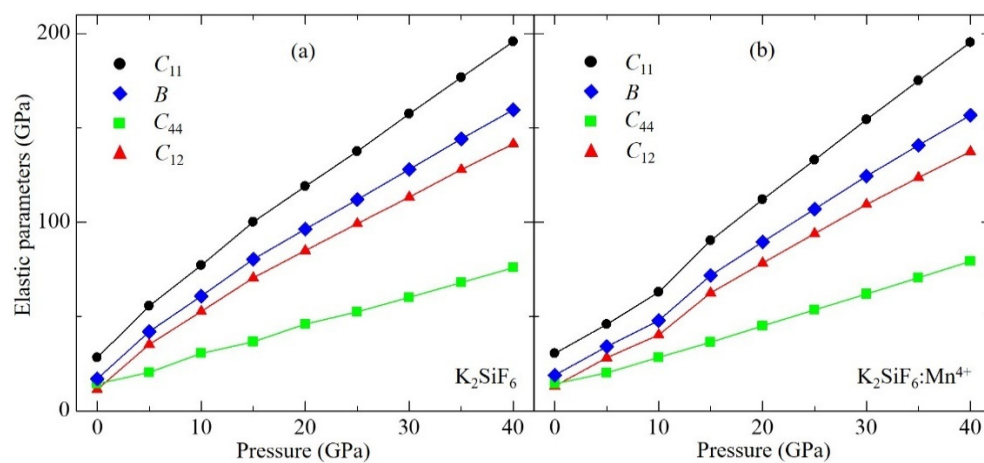


Figure 1. Calculated relative volume V/V_0 of (a) non-doped and (b) Mn^{4+} -doped K_2SiF_6 as a function of pressure. Red solid curves denote fitting results to the Murnaghan's equations of state.

The elastic constants of non-doped and Mn^{4+} -doped KSF were calculated using a stress-strain method [23] implemented in VASP, which are summarized in Table 2 and plotted in Figure 2 as a function of pressure. Here the optimized structure at each pressure was used to calculate the elastic constants. Three irreducible elastic constants for the cubic structure, i.e., C_{11} , C_{12} and C_{44} , at zero-pressure were reported in the earlier study [22]; the values are 31.90, 9.28 and 15.10 GPa, respectively. Those in the current study show similar values of 28.2, 11.3 and 14.3 GPa for C_{11} , C_{12} and C_{44} , respectively. The difference between these two studies is derived from the difference in calculating conditions as discussed above for the bulk modulus. The calculated C_{11} and C_{12} values for the non-doped KSF are larger than those of $\text{KSF}:\text{Mn}^{4+}$ at all pressures calculated in this work. On the other hand, the calculated C_{44} of non-doped KSF is larger than that of $\text{KSF}:\text{Mn}^{4+}$ at zero pressure, the difference between these two becoming smaller as pressure increases, and finally C_{44} of $\text{KSF}:\text{Mn}^{4+}$ becomes larger than that of non-doped KSF between 20 and 25 GPa. It can be noted that all the elastic constants of both non-doped and $\text{KSF}:\text{Mn}^{4+}$ increase with increase of pressure, in which C_{11} and C_{12} increase more rapidly than C_{44} as shown in Figure 3.

Table 2. Pressure dependence of the elastic constants C_{ij} (all in GPa) for the non-doped and Mn^{4+} -doped K_2SiF_6 .

System	Pressure	C_{11}	C_{12}	C_{44}
K_2SiF_6	0	28.2	11.3	14.3
	5	55.6	35.2	20.4
	10	77.1	52.7	30.5
	15	100.1	70.5	36.6
	20	119.0	84.9	46.0
	25	137.5	99.2	52.5
	30	157.4	113.3	60.1
	35	176.7	127.9	68.0
	40	195.7	141.5	75.8
$K_2SiF_6:Mn^{4+}$	0	23.7	9.0	11.0
	5	45.8	28.1	20.1
	10	62.9	40.3	28.4
	15	90.3	62.5	36.4
	20	112.0	78.3	45.0
	25	133.0	93.9	53.5
	30	154.4	109.4	61.9
	35	175.0	123.7	70.5
	40	195.3	137.4	79.2

**Figure 2.** Calculated elastic constants C_{ij} and bulk moduli B for (a) non-doped and (b) Mn^{4+} -doped K_2SiF_6 as a function of pressure.

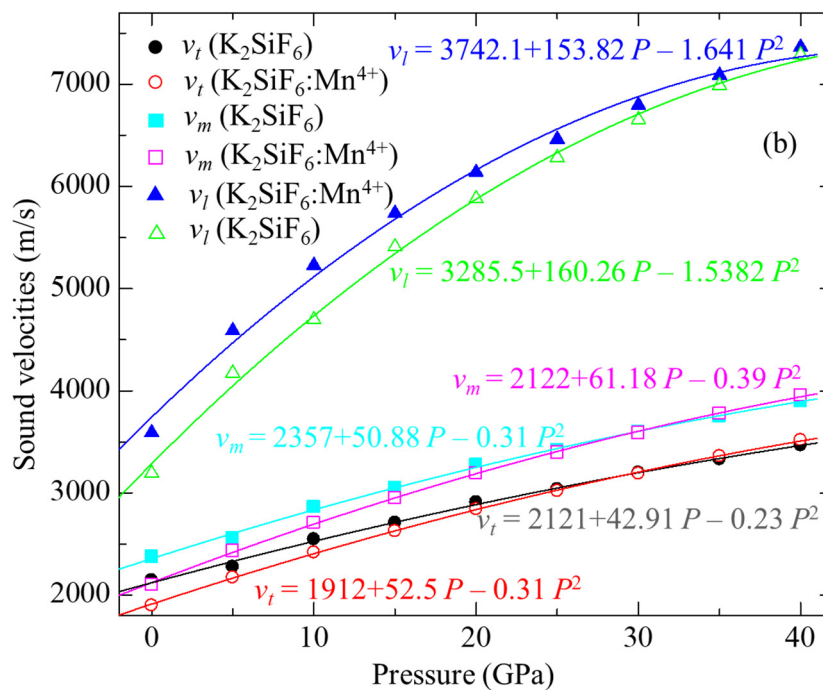
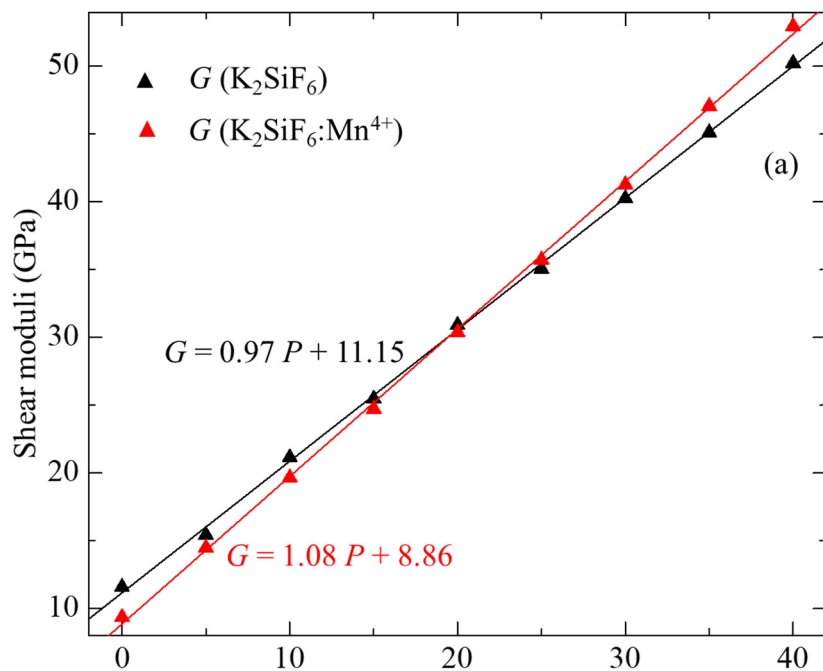


Figure 3. Cont.

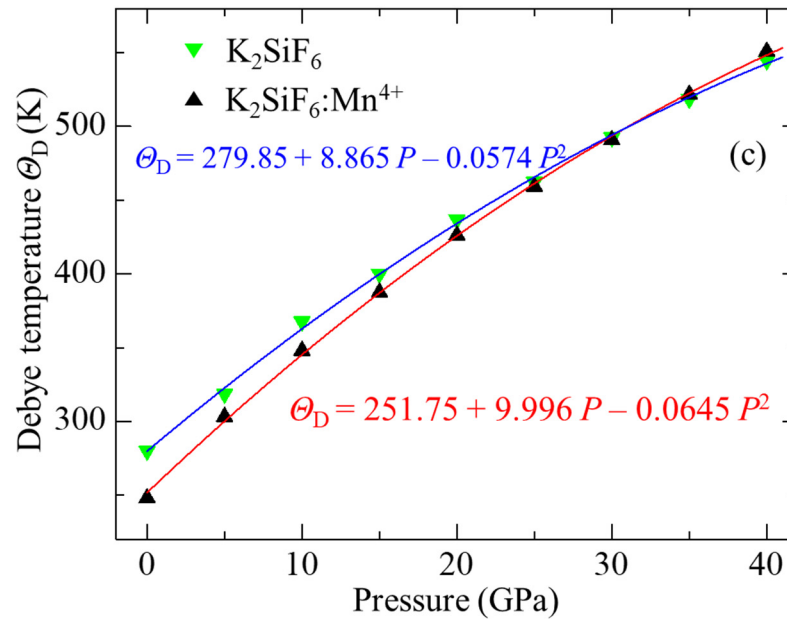


Figure 3. Calculated (a) shear moduli, (b) sound velocities and (c) Debye temperatures for the non-doped and Mn^{4+} -doped K_2SiF_6 as a function of pressure.

Using these calculated elastic constants, sound velocities can be obtained. The mean sound velocity v_m is expressed [24] in terms of the longitudinal sound velocity v_l and the transverse one v_t as:

$$v_m = \left[\frac{1}{3} \left(\frac{2}{v_t^3} + \frac{1}{v_l^3} \right) \right]^{-1/3} \quad (2)$$

in which v_l and v_t are calculated [25] by:

$$v_l = \sqrt{\frac{3B + 4G}{3\rho}} \quad \text{and} \quad v_t = \sqrt{\frac{G}{\rho}} \quad (3)$$

Here $G = (G_V + G_R)/2$ is the isotropic shear modulus, in which $G_V = (C_{11} - C_{12} + 3C_{44})/5$ is the Voigt's shear modulus (an upper limit for G), and $G_R = 5(C_{11} - C_{12})C_{44}/[4C_{44} + 3(C_{11} - C_{12})]$ is the Reuss's shear modulus (a lower limit for G). Debye temperature, Θ_D , is calculated using the following equation [25]:

$$\Theta_D = \frac{h}{k} \left[\frac{3n}{4\pi} \left(\frac{N_A \rho}{M} \right) \right]^{1/3} v_m \quad (4)$$

where h and k are the Planck's and Boltzmann's constants, respectively, N_A is the Avogadro's number, ρ is the density, M is the molecular weight, and n denotes the number of atoms per formula unit (here nine for KSF). The calculated shear modulus, G , sound velocities, v_m , v_t and v_l , and Debye temperature, Θ_D , as a function of pressure for both non-doped and Mn^{4+} -doped KSF are summarized in Table 3 and plotted in Figure 3a–c for shear moduli, sound velocities, and Debye temperature, respectively. The G , v_m , v_t , v_l and Θ_D values for both systems increase with pressure. Inclusion of Mn^{4+} into KSF leads to a decrease of G , v_m , v_t , v_l and Θ_D at zero pressure. Each of the differences in those parameters between non-doped and Mn^{4+} -doped KSF becomes smaller with increase of pressure. Calculated G of $\text{KSF}:\text{Mn}^{4+}$ becomes larger than that of non-doped KSF at pressures between 20 and 25 GPa as in the case of C_{44} , while v_m , v_l and Θ_D of $\text{KSF}:\text{Mn}^{4+}$ becomes larger than that of the non-doped one between 30 and 35 GPa.

Table 3. Calculated shear moduli, sound velocities and Debye temperatures for the non-doped and Mn⁴⁺-doped K₂SiF₆.

System	<i>P</i> , GPa	<i>G</i> , GPa	<i>v_t</i> , m/s	<i>v_l</i> , m/s	<i>v_m</i> , m/s	Θ _D , K
K ₂ SiF ₆	0	11.59	2146.70	3590.80	2375.61	280
	5	15.44	2278.49	4589.10	2557.08	319
	10	21.15	2547.46	5227.87	2861.95	368
	15	25.48	2708.88	5738.69	3048.37	400
	20	30.91	2909.98	6138.43	3273.96	437
	25	35.06	3037.20	6463.19	3418.59	462
	30	40.26	3199.14	6796.39	3600.57	493
	35	45.13	3331.08	7089.15	3749.39	519
	40	50.24	3467.45	7364.17	3902.49	545
K ₂ SiF ₆ :Mn ⁴⁺	0	9.36	1900.48	3193.65	2104.06	248
	5	14.49	2174.34	4174.21	2432.98	303
	10	19.65	2419.06	4696.07	2708.78	348
	15	24.72	2628.23	5410.22	2953.19	388
	20	30.38	2842.32	5881.25	3194.64	426
	25	35.74	3020.83	6283.09	3396.21	459
	30	41.28	3191.09	6653.71	3588.09	491
	35	47.08	3361.47	6990.49	3779.15	522
	40	52.97	3518.48	7289.33	3954.87	551

Calculated electronic densities of states (DOSs) of non-doped and Mn⁴⁺-doped KSF at 0 GPa are shown in Figure 4. As illustrated in this figure, new orbitals associated with the Mn⁴⁺ 3d orbitals appear in the band gap due to a doping of Mn⁴⁺ ion. Current results are consistent with our previous report [26], although the previous one was undertaken with larger super cells expanded by 2 × 1 × 1. The Mn⁴⁺ ion in KSF is surrounded by six F[−] ions in a crystal environment of cubic symmetry. Hence, the 3d ground-state splits into the triply and doubly degenerated t_{2g} and e_g orbitals, respectively. The 3d states of Mn⁴⁺ hybridize with the F[−] 2p states in both t_{2g} and e_g orbitals as shown in Figure 4b. The Mn⁴⁺ ion has three 3d electrons, which fully occupy t_{2g} up-spin state.

To discuss the influence of pressure on the electronic structure, calculated DOSs of KSF:Mn⁴⁺ are compared among those at different pressures between 0 and 40 GPa, which are shown in Figure 5. The energies of the top of each state originating from 3d orbital of Mn⁴⁺ ion, i.e., up-spin and down-spin states of t_{2g} and e_g, are plotted in Figure 6. As the top of the occupied band was set to zero in Figure 5, the tops of all t_{2g} up-spin states were located at zero. On the other hand, down-spin t_{2g} states shift to the lower energy side, while up- and down-spin e_g states shift to the higher energy side with increasing pressure. Calculated 10Dq, which denotes the crystal field strength and is defined as a difference in energy between up-spin t_{2g} and e_g states, at different pressures is summarized in Table 4 and plotted in Figure 7. Calculated 10Dq value at zero pressure of 2.76 eV agrees well with the experimental value of 2.74 eV [27], although our current calculations were carried out by GGA-PBE without the effect of strong electron correlation for Mn 3d electrons. As shown in these tables and figures, 10Dq increases linearly as pressure increases. Least square fitting to the pressure dependence of 10Dq yields 10Dq = 0.00346 P + 2.734. In addition, from a geometrical point of view, 10Dq values are plotted in Figure 8 as a function of bond length between Mn⁴⁺ and surrounding F[−] ions. As shown in this figure, a very good linear relation can be found between 10Dq and Mn-F bond length. The least-square fitting result is 10Dq = −6.37 d_{Mn-F} + 14.46. This result implies that 10Dq can be estimated by this

equation for other Mn^{4+} -doped fluorides, in which the doped Mn^{4+} ion is located at the O_h symmetry point indicated by six F^- ions. This result is consistent with our previous study on A_2SiF_6 [26], where $\text{A} = \text{K}, \text{Rb}$ and Cs , in which $10Dq$ decreases with increasing $\text{Mn}^{4+}-\text{F}^-$ bond-length.

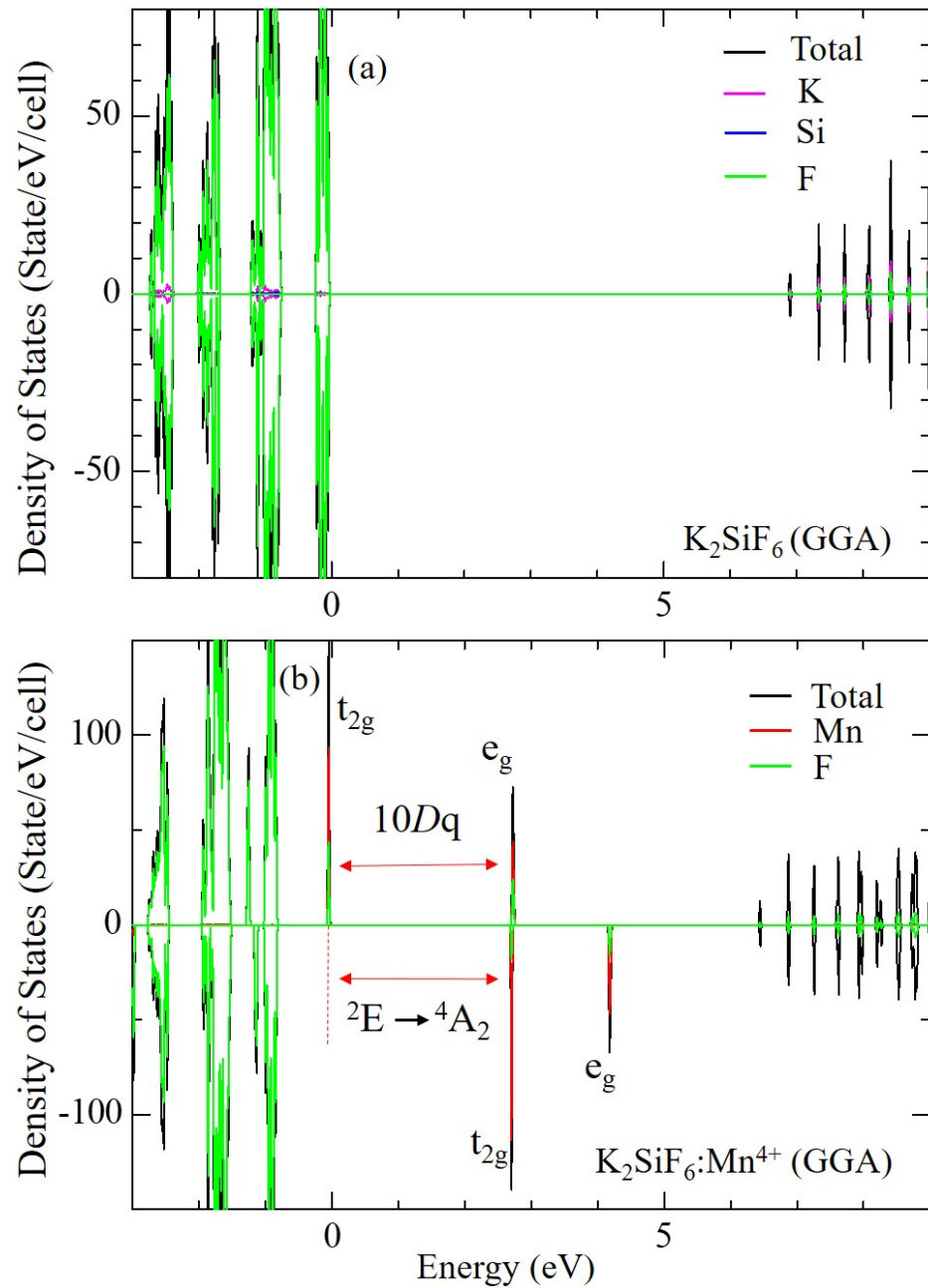


Figure 4. Calculated electronic density of states of (a) non-doped and (b) Mn^{4+} -doped K_2SiF_6 .

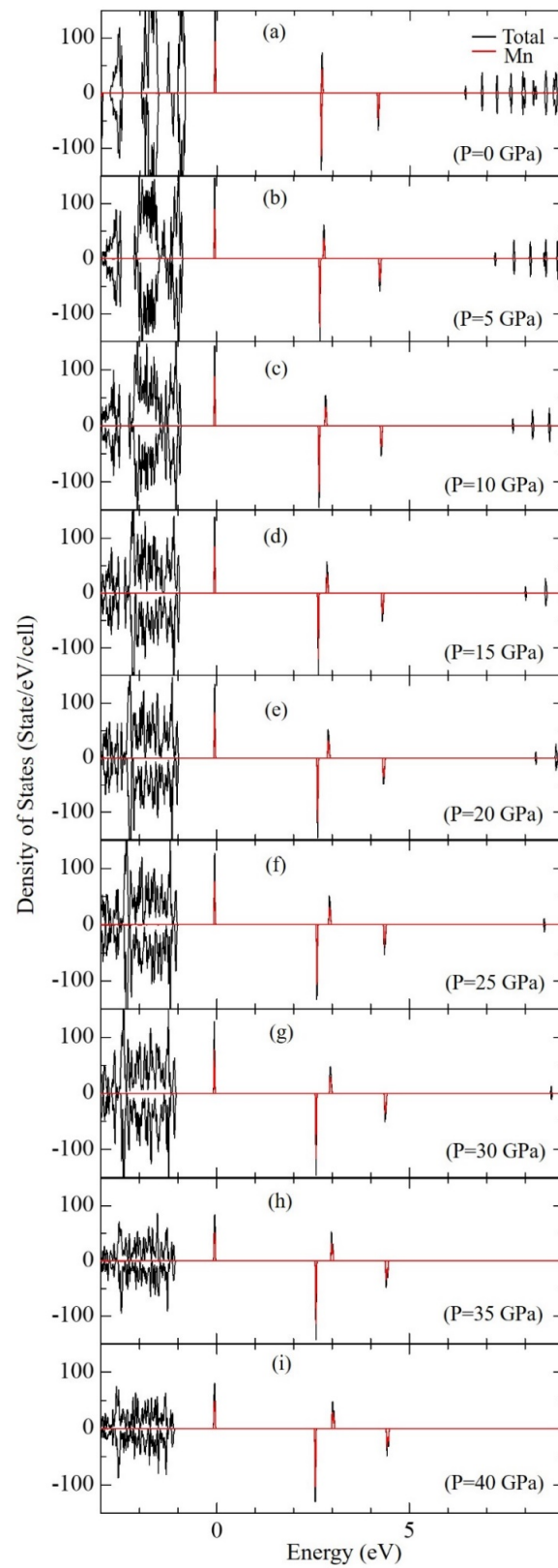


Figure 5. Calculated densities of states (DOSs) of Mn^{4+} -doped K_2SiF_6 at $P =$ (a) 0, (b) 5, (c) 10, (d) 15, (e) 20, (f) 25, (g) 30, (h) 35 and (i) 40 GPa, respectively.

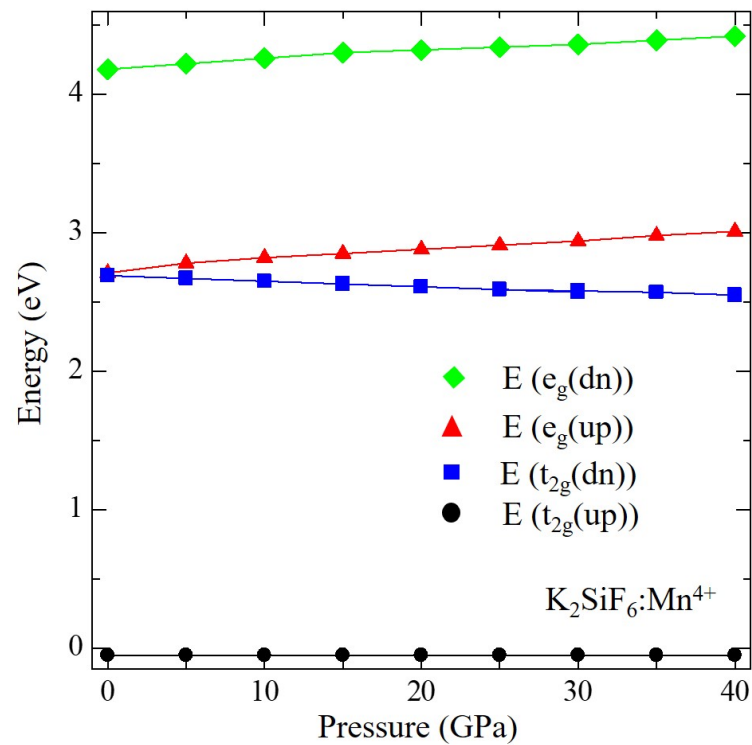


Figure 6. Calculated energy levels of the t_{2g} -up, t_{2g} -down, e_g -up and e_g -down states of Mn^{4+} -doped K_2SiF_6 as a function of pressure.

Table 4. Comparison of the experimental and calculated ${}^2E \rightarrow {}^4A_2$ emission transition energy and $10Dq$ for the Mn^{4+} -doped K_2SiF_6 at different pressures.

	Pressure (GPa)	Calc. (eV)	Exp. (eV)
$10Dq$	0	2.76	2.74 ^a
	5	2.83	
	10	2.87	
	15	2.90	
	20	2.93	
	25	2.96	
	30	3.00	
	35	3.03	
${}^2E \rightarrow {}^4A_2$	40	3.06	
	0	2.74	1.99 ^b
	5	2.72	
	10	2.70	
	15	2.68	
	20	2.66	
	25	2.64	
	30	2.63	
35	2.62		
40	2.60		

^a Ref. [27]. ^b Ref. [9].

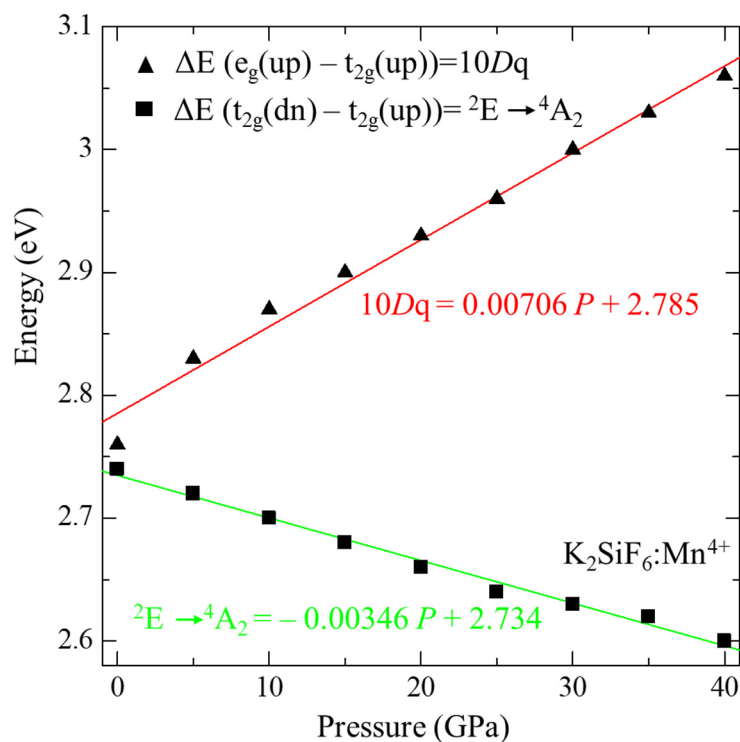


Figure 7. Calculated emission energy, E_{em} , and $10Dq$ for Mn^{4+} -doped K_2SiF_6 as a function of pressure. The solid lines are the least-square linear fittings.

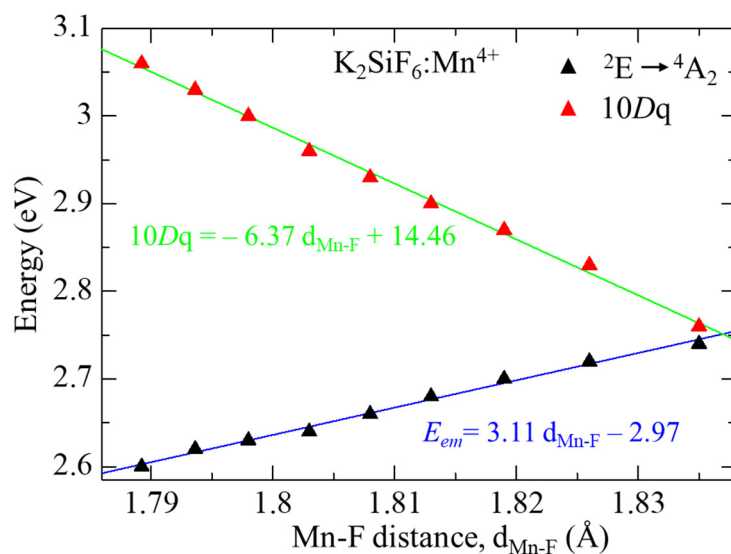


Figure 8. Calculated $10Dq$ parameter and E_{em} for $K_2SiF_6:Mn^{4+}$ as a function of Mn-F bond length.

The red emission from Mn^{4+} ion is assigned to the transition from the first excited state 2E to the ground state 4A_2 of the Mn^{4+} 3d state. Electron configuration of the 4A_2 state of the Mn^{4+} ion is all three electrons on up-spin t_{2g} while that of 2E is two electrons on up-spin t_{2g} and one on down-spin t_{2g} states. Here, the transition energy from 2E to 4A_2 , E_{em} , is calculated by an energy gap between the up-spin and down-spin t_{2g} states within a one electron approximation to avoid complicated calculations considering multielectron effects to obtain total electronic energy difference between the 2E and 4A_2 states. The calculated E_{em} value at zero pressure is 2.74 eV, which is larger than the experimental value of 1.99 eV [9]. This overestimation is derived from the method to obtain E_{em} . However, the change in E_{em} can be discussed qualitatively as in our earlier report on $KSF:Mn^{4+}$ [26]. As

the d and f electrons are localized, consideration of a strong electron–electron correlation effect, such as the DFT + U method we have undertaken previously [26], may lead to a better reproduction of E_{em} than the current calculations with GGA-PBE. In addition, the calculated E_{em} at zero pressure, i.e., 2.74 eV, is larger than that in the previous study on KSF:Mn⁴⁺ [26]. Here, a $1 \times 1 \times 1$ unit cell was used, whereas a $2 \times 1 \times 1$ super cell was employed in our previous study [26], which yield a difference in calculated E_{em} at zero pressure. E_{em} as a function of pressure is summarized in Table 4 and plotted in Figure 7. Contrary to the change in $10Dq$, E_{em} decreases almost linearly, $E_{em} = -0.00346 P + 2.734$, with increasing pressure. The calculated E_{em} as a function of Mn⁴⁺–F[−] bond length is also plotted in Figure 8 for KSF:Mn⁴⁺. It can be seen that the ${}^2E \rightarrow {}^4A_2$ transition energy has very good linear relation with Mn–F bond length, $E_{em} = 3.11 d_{Mn-F} - 2.97$.

4. Conclusions

First-principles calculations have been carried out for the non-doped and Mn⁴⁺-doped K₂SiF₆ to study the influence of isostatic pressure on the geometric structure, elastic and electronic properties of K₂SiF₆:Mn⁴⁺ red phosphor. In particular, the pressure effect on the elastic properties such as elastic constants, shear modulus, sound velocity and Debye temperature between non-doped and Mn⁴⁺-doped systems was discussed. Pressure dependence of two important electronic parameters, i.e., $10Dq$ and the emission energy of the ${}^2E \rightarrow {}^4A_2$ transition, of KSF:Mn⁴⁺ have been investigated, which yield good linear relationships both between $10Dq$ and external pressure and between emission energy of ${}^2E \rightarrow {}^4A_2$ and pressure. It can be noted that the current analysis of the pressure effect on $10Dq$ and emission energy of the ${}^2E \rightarrow {}^4A_2$ transition provided empirical formulae to obtain these values as a function of bond-length between Mn⁴⁺ and surrounding F[−] ions, which can be used for the estimation of these important parameters for other Mn-doped fluorides.

Author Contributions: Conceptualization, T.Y., M.G.B. and M.S.; methodology, M.S. and U.Z.; software, M.S., U.Z. and T.Y.; validation, T.Y., C.-G.M., M.P., M.G.B., W.W.B., W.E.C. and A.M.S.; formal analysis, M.S., U.Z., T.Y., M.G.B., M.P. and C.-G.M.; investigation, M.S., U.Z. and T.Y.; resources, M.G.B.; data curation, M.S., T.Y., C.-G.M. and M.P.; writing—original draft preparation, M.S., T.Y., M.G.B. and A.M.S.; writing—review and editing, T.Y., M.G.B., A.M.S., W.W.B. and W.E.C.; visualization, M.S. and U.Z.; supervision, T.Y., M.G.B. and A.M.S.; project administration, T.Y. and M.G.B.; funding acquisition, T.Y. and M.G.B. All authors have read and agreed to the published version of the manuscript.

Funding: This work was partly carried out at the Joint Research Center for Environmentally Conscious Technologies in Materials Science (project No. 30009, 30012, 31008, 31017, 02018, 02021, 02022, 02115, 02116 and 02117) at ZAIKEN, Waseda University, and supported by JSPS KAKENHI Grant number 20K05065. M.S. and U.Z. thank for support the IICN under the CIS Grant number 21-109 and 21-107, respectively. The authors are grateful to Kholmirzo Kholmurodov for his continuous encouragement and support. M.G.B. appreciates the supports from the Program for the Foreign Experts (Grant No. W2017011) offered by Chongqing University of Posts and Telecommunications and the National Foreign Experts Program for “Belt and Road Initiative” Innovative Talent Exchange (Grant No. DL2021035001L), Estonian Research Council grant PUT PRG111, European Regional Development Fund (TK141), and NCN project 2018/31/B/ST4/00924. C.-G. Ma acknowledges the support of the National Natural Science Foundation of China (Grant No. 52161135110) and China-Poland Intergovernmental Science and Technology Cooperation Program (Grant No. 2020[15]/10). M.S. appreciates the support from 2021 Chongqing Postdoctoral International Exchange Program of China Postdoctoral Science Foundation.

Institutional Review Board Statement: Not applicable.

Informed Consent Statement: Not applicable.

Data Availability Statement: The raw/processed data required to reproduce these findings cannot be shared at this time as the data also form a part of an ongoing study.

Conflicts of Interest: The authors declare no conflict of interest. The funders had no role in the design of the study; in the collection, analyses, or interpretation of data; in the writing of the manuscript, or in the decision to publish the results.

References

1. Brik, M.; Srivastava, A. On the optical properties of the Mn⁴⁺ ion in solids. *J. Lumin.* **2013**, *133*, 69–72. [[CrossRef](#)]
2. Arai, Y.; Adachi, S. Optical properties of Mn⁴⁺-activated Na₂SnF₆ and Cs₂SnF₆ red phosphors. *J. Lumin.* **2011**, *131*, 2652–2660. [[CrossRef](#)]
3. Srivastava, A.; Brik, M. Ab initio and crystal field studies of the Mn⁴⁺-doped Ba₂LaNbO₆ double-perovskite. *J. Lumin.* **2012**, *132*, 579–584. [[CrossRef](#)]
4. Nguyen, H.-D.; Liu, R.-S. Narrow-band red-emitting Mn⁴⁺-doped hexafluoride phosphors: Synthesis, optoelectronic properties, and applications in white light-emitting diodes. *J. Mater. Chem. C* **2016**, *4*, 10759–10775. [[CrossRef](#)]
5. Adachi, S. Photoluminescence properties of Mn⁴⁺-activated oxide phosphors for use in white-LED applications: A review. *J. Lumin.* **2018**, *202*, 263–281. [[CrossRef](#)]
6. Adachi, S. Crystal-field and Racah parameters of Mn⁴⁺ ion in red and deep red-emitting phosphors: Fluoride versus oxide phosphor. *J. Lumin.* **2020**, *218*, 116829. [[CrossRef](#)]
7. Senden, T.; Van Dijk-Moes, R.J.A.; Meijerink, A. Quenching of the red Mn⁴⁺ luminescence in Mn⁴⁺-doped fluoride LED phosphors. *Light Sci. Appl.* **2018**, *7*, 8. [[CrossRef](#)]
8. Adachi, S. Photoluminescence spectra and modeling analyses of Mn⁴⁺-activated fluoride phosphors: A review. *J. Lumin.* **2018**, *197*, 119–130. [[CrossRef](#)]
9. Arai, T.; Adachi, S. Excited States of 3d³ Electrons in K₂₂SiF₆₆:Mn⁴⁺⁴⁺ Red Phosphor Studied by Photoluminescence Excitation Spectroscopy. *Jpn. J. Appl. Phys.* **2011**, *50*, 092401. [[CrossRef](#)]
10. Sijbom, H.F.; Joos, J.; Martin, L.; Eeckhout, K.V.D.; Poelman, D.; Smet, P. Luminescent Behavior of the K₂SiF₆:Mn⁴⁺ Red Phosphor at High Fluxes and at the Microscopic Level. *ECS J. Solid State Sci. Technol.* **2015**, *5*, R3040–R3048. [[CrossRef](#)]
11. Setlur, A.A.; Radkov, E.V.; Henderson, C.S.; Her, J.-H.; Srivastava, A.M.; Karkada, N.; Kishore, M.S.; Kumar, N.P.; Aesram, D.; Deshpande, A.; et al. Energy-Efficient, High-Color-Rendering LED Lamps Using Oxyfluoride and Fluoride Phosphors. *Chem. Mater.* **2010**, *22*, 4076–4082. [[CrossRef](#)]
12. Du, M.H. Chemical trends of Mn⁴⁺ emission in solids. *J. Mater. Chem. C* **2014**, *2*, 2475–2481. [[CrossRef](#)]
13. Wang, Y.; Wen, T.; Tang, L.; Yang, L.; Yang, W.; Zhao, Y. Impact of hydrostatic pressure on the crystal structure and photoluminescence properties of Mn⁴⁺-doped BaTiF₆ red phosphor. *Dalton Trans.* **2015**, *44*, 7578–7585. [[CrossRef](#)] [[PubMed](#)]
14. Zhang, N.; Tsai, Y.-T.; Fang, M.-H.; Ma, C.-G.; Lazarowska, A.; Mahlik, S.; Grinberg, M.; Chiang, C.-Y.; Zhou, W.; Lin, J.G.; et al. Aluminate Red Phosphor in Light-Emitting Diodes: Theoretical Calculations, Charge Varieties, and High-Pressure Luminescence Analysis. *ACS Appl. Mater. Interfaces* **2017**, *9*, 23995–24004. [[CrossRef](#)]
15. Lazarowska, A.; Mahlik, S.; Grinberg, M.; Lin, C.C.; Liu, R.-S. Pressure effect on the zero-phonon line emission of Mn⁴⁺ in K₂SiF₆. *J. Chem. Phys.* **2015**, *143*, 134704. [[CrossRef](#)]
16. Kresse, G.; Furthmüller, J. Efficiency of ab-initio total energy calculations for metals and semiconductors using a plane-wave basis set. *Comput. Mater. Sci.* **1996**, *6*, 15–50. [[CrossRef](#)]
17. Perdew, J.P.; Burke, K.; Wang, Y. Generalized gradient approximation for the exchange-correlation hole of a many-electron system. *Phys. Rev. B* **2014**, *54*, 16533–16539. [[CrossRef](#)]
18. Loehlin, J.H. Redetermination of the Structure of Potassium Hexafluorosilicate, K₂SiF₆. *Acta Cryst. C* **1984**, *40*, 570. [[CrossRef](#)]
19. Monkhorst, H.J.; Pack, J.D. Special points for Brillouin-zone integrations. *Phys. Rev. B* **1976**, *13*, 5188. [[CrossRef](#)]
20. Novita, M.; Honma, T.; Hong, B.; Ohishi, A.; Ogasawara, K. Study of multiplet structures of Mn⁴⁺ activated in fluoride crystals. *J. Lumin.* **2016**, *169*, 594–600. [[CrossRef](#)]
21. Murnaghan, F.D. The Compressibility of Media under Extreme Pressures. *Proc. Natl. Acad. Sci. USA* **1944**, *30*, 244–247. [[CrossRef](#)]
22. Brik, M.G.; Srivastava, A.M. Ab Initio Studies of the Structural, Electronic, and Optical Properties of K₂SiF₆ Single Crystals at Ambient and Elevated Hydrostatic Pressure. *J. Electrochem. Soc.* **2012**, *159*, J212–J216. [[CrossRef](#)]
23. Nielsen, O.H.; Martin, R.M. First-Principles Calculation of Stress. *Phys. Rev. Lett.* **1983**, *50*, 697–700. [[CrossRef](#)]
24. Poirier, J.; Brown, J.M. Introduction to the Physics of the Earth's Deep Interior. *Phys. Today* **1992**, *45*, 66–67. [[CrossRef](#)]
25. Anderson, O.L. A simplified method for calculating the debye temperature from elastic constants. *J. Phys. Chem. Solids* **1963**, *24*, 909–917. [[CrossRef](#)]
26. Subhoni, M.; Zafari, U.; Srivastava, A.M.; Beers, W.W.; Cohen, W.; Brik, M.G.; Yamamoto, T. First-principles investigations of geometrical and electronic structures of Mn⁴⁺ doped A₂SiF₆ (A = K, Rb, Cs) red phosphors. *Opt. Mater.* **2021**, *115*, 110986. [[CrossRef](#)]
27. Paulusz, A.G. Efficient Mn(IV) Emission in Fluorine Coordination. *J. Electrochem. Soc.* **1973**, *120*, 942. [[CrossRef](#)]