# Heliyon 7 (2021) e07917

Contents lists available at ScienceDirect

# Heliyon

journal homepage: www.cell.com/heliyon

# **Research article**

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# The effects of different activating agents on the physical and electrochemical properties of activated carbon electrodes fabricated from wood-dust of *Shorea robusta*

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#### ARTICLE INFO

Keywords: Shorea robusta Specific capacitance Energy density Power density Cyclic stability

### ABSTRACT

This study focuses on the effects of activating agents on the physical and electrochemical properties of activated carbon (AC) electrodes, fabricated from wood dust of Shorea robusta. Three different activating agents namely H<sub>3</sub>PO<sub>4</sub> KOH and Na<sub>2</sub>CO<sub>3</sub> have been used to prepare ACs, which were named as: Sr-H<sub>3</sub>PO<sub>4</sub>, Sr-KOH and Sr-Na2CO3. The ACs were characterized by TGA/DSC, XRD, Raman, SEM, FTIR and BET. All the as prepared ACs were found to be amorphous in nature. The oxygen surface functionality was developed at the surface. The surface area of Sr-H<sub>3</sub>PO<sub>4</sub>, Sr-KOH and Sr-Na<sub>2</sub>CO<sub>3</sub> were found to be 1269.5 m<sup>2</sup>/g, 280.6 m<sup>2</sup>/g and 58.9 m<sup>2</sup>/g respectively. The activated carbon-electrodes were then fabricated and supercapacitive performances were evaluated by "three electrode system" in aqueous 6M KOH using cyclic voltammetry (CV), galvanostatic charge discharge (GCD) and electrochemical impedance spectroscopy (EIS).The GCD performed at 1A/g revealed the specific capacitance values were 136.3 F/g, 42.2 F/g and 59.1 F/g for Sr-H<sub>3</sub>PO<sub>4</sub>, Sr-KOH and Sr-Na<sub>2</sub>CO<sub>3</sub>-electrodes, respectively. Energy density for Sr-H<sub>3</sub>PO<sub>4</sub> electrode was found to be 3.0 Wh/kg at 99.6 W/kg power densities. Moreover, it also displayed imposing cyclic stability of about 96.9 %, 89.5 % and 78.5 % after 1000 cycles of charge/discharge respectively. The overall electrochemical performance of Sr-H<sub>3</sub>PO<sub>4</sub> showed outstanding supercapacitive performances demonstrating the high possibility of this material to be used for the EDLC application in supercapacitive energy storage. The Nyquist plot also showed the lowest internal resistance of about 0.4  $\Omega$  for Sr-H<sub>3</sub>PO<sub>4</sub> electrode.

# 1. Introduction

The storage of energy shows a key role in achieving global energy demand and sustainability. To till date, various energy storage devices have been established, like solar cell, fly wheel, compressed air, fuel cell, supercapacitor and battery [1, 2]. Among them, supercapacitors and batteries have been proven to be the most effective and promising electrochemical energy storage devices for practical application [3] due to their excellent power density, outstanding pulse charge-discharge performance, superior lifespan and low maintenance cost [4]. Based on charge storage mechanisms, supercapacitors may be (i) electrical double layer capacitors (EDLC) or (ii) pseudo-capacitors [5]. For super-capacitors, conducting polymers and carbon-based materials have been regarded as the most promising materials [6, 7, 8, 9]. They are generally prepared from lignocellulosic materials that consisted of high percent of

carbon and low inorganic content as well [10, 11, 12, 13] and are considered to have a huge potential in improving the electrode performance in supercapacitors [14, 15, 16]. Various biomass precursors have been chemically treated and carbonized into activated carbons for supercapacitors such as cherry stone [17], fish scale [18], waste paper [19], water bamboo [20], yeast cells [21], pine-cone [22], willow catkins [23], celtics leaves [24], waste tea-leaves [25], sunflower seed shell [26], ginkgo shells [27], cow dung [28], human hair [29] and sewage sludge [30]. The use of these materials in electrochemical performances are due to large specific surface area, pore size distribution and pore volume [31]. Such type of characteristic properties could be obtained by activation of the precursor. Two types of activation techniques are generally employed: physical activation and chemical activation.

In physical activation, precursor is carbonized at high temperatures by passing oxidizing gases like CO<sub>2</sub>, H<sub>2</sub>O vapors whereas in chemical

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https://doi.org/10.1016/j.heliyon.2021.e07917

Received 30 June 2021; Received in revised form 11 August 2021; Accepted 1 September 2021

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activation, chemicals like  $ZnCl_2$ ,  $H_3PO_4$ , KOH,  $K_2CO_3$ ,  $Na_2CO_3$  were used as activating agent during carbonization step [15, 16]. Chemical activation process exhibited superior to physical activation as it doesn't require high heating during carbonization. In these days, attention has been given for chemical activation to have better electrochemical performances.

Wang et al. [32] has reported, KOH as activating agent that showed better performance to obtain honeycomb like carbon foam. It exhibited a large specific surface area of 1313 m<sup>2</sup>/g and a high specific capacitance of 473 F/g. In the same way, porous carbon was converted from natural silk via coupled activation and graphitization process, which showed a high specific surface area of 2494 m<sup>2</sup>/g and a large pore volume of 2.28 cm<sup>3</sup>/g. Further for supercapacitor application, the silk derived hierarchial porous carbon demonstrated a high specific capacitance of 242 F/g, a high-energy density of 102 Wh/kg, as well as excellent cycling life stability (9% loss after 10,000 cycles).

Similarly, activated carbon nanospheres was obtained after KOH activation and further post processing of coconut shells were found to have a surface area of 2000  $m^2/g$ , total pore volume of 1.2 cm<sup>3</sup>/g, specific capacitance of 250 F/g in 1M H<sub>2</sub>SO<sub>4</sub> electrolyte. 93% capacitance retention was observed after 2000 cycles [33]. Correspondingly, Molina-Sabio et al. has studied the carbon sample obtained after the processing of sunflower seed shell to have specific capacitance of 244 F/g in 0.1 M HCl electrolyte, surface area of 2585 m<sup>2</sup>/g pore volume of 1.4 cm<sup>3</sup>/g and good cycle stability [31]. Zhang et al. presented the activated carbon nanospheres obtained from carrageenan and possesses a surface area of 1704 m<sup>2</sup>/g, total pore volume of 0.9 cm<sup>3</sup>/g, specific capacitance of 343 F/g using KOH and CO<sub>2</sub> activation in 1 M H<sub>2</sub>SO<sub>4</sub>. And it has a cycle life of about 2500 cycles [34]. Here, KOH was applied to produce microporosity than mesoporosity [35]. In these days, attention has also given towards phosphoric acid as activating agent to produce mesopores along with micropores [31]. These highly porous carbon materials having high surface areas are considered as potential candidates for fast charge transfer and the double layer formation of electrolyte ions in the networks of the electrode [34, 36] and was reported for the preparation of electrode materials for electrical double layer capacitors (EDLC) [37, 38, 39].

In this study, it is a particular interest to select the wood dust of *Shorea robusta* (common name "Sal") as a precursor to make activated carbon (AC). *Shorea robusta* is a large tree reaching a height of 45m, belongs to the family- Dipterocarpaecea. This tree is native to the Indian subcontinent, ranging south of the Himalaya, from Myanmar in the east to Nepal, India and Bangladesh. *Shorea robusta* is distributed in the southern terrain of Nepal from east to west.

It is considered as the most valuable tree species in Nepal. Most of the rural communities of Nepal, which constitute 80% of the total population of the country depend on Sal forest for their survival [40]. The major part of *Shorea robusta* is used in carpentry and construction, and also as a main source of fuel wood. Carpentry work constitutes a major source of waste byproducts as sawdust and wood dust. These waste byproducts have high carbon content in lignocellulosic form having low content of inorganic materials and relatively high content of volatile matters. Hence, currently, attention has given to convert waste byproducts into renewable energy storage material in general and biomass energy in particular. Thus, one of the cost effective way is to convert the wood residue to activated carbon which in turn used for energy storage device.

There are tremendous efforts in finding new materials for supercapacitor application. One of the research areas is materials preparation from natural products where main focus is optimizing the conditions for obtaining high surface area porous carbonaceous material. In this study, wood dust of *Shorea robusta* was used as precursor, which was chemically activated with different activating agents namely; phosphoric acid (H<sub>3</sub>PO<sub>4</sub>), potassium hydroxide (KOH) and sodium carbonate (Na<sub>2</sub>CO<sub>3</sub>). In addition, a low carbonization temperature of 400 °C unlike higher temperature (600–800 °C) was used. Besides these, other new contribution of this study were the selection of an electrolyte to have wide voltage window, high electrochemical stability of the materials as supercapacitor electrode, high ionic conductivity as well as availability at high purity, which are most important parameters for electrode materials to be suitable for supercapacitor application. These are the originality/ novelty of this study.

# 2. Experimental procedures

### 2.1. Materials and methods

The analytical grade chemicals were used as procured. Phosphoric acid (H<sub>3</sub>PO<sub>4</sub>), having 85% purity, was obtained from Fischer Scientific, India. Potassium hydroxide (KOH) and anhydrous sodium carbonate (Na<sub>2</sub>CO<sub>3</sub>) were procured from Merck specialties Pvt. Ltd. Nanoporous carbon black, Polyvinylidine fluoride (PVDF), N-methyl pyroline (NMP) were obtained from Sigma-Aldrich (USA). Wood dust of *Shorea robusta* (Sal) has been obtained from local carpentry/saw mill of Kathmandu, Nepal.

# 2.1.1. Synthesis of activated carbon

Wood dust of *Shorea robusta (Sal)* collected from local saw mill was washed with distilled water and dried at 110  $^{\circ}$ C for 24h. It was crushed, grinded and sieved through 150  $\mu$ m sized sieve to obtain a fine powder, which was used as a precursor material for the preparation of ACs. It was followed by chemical activation. Three different activating agents, H<sub>3</sub>PO<sub>4</sub>, KOH and Na<sub>2</sub>CO<sub>3</sub> have been used.

First of all, the fine wood dust powder was divided into three parts. The first part was impregnated with H<sub>3</sub>PO<sub>4</sub>, the second part was impregnated with KOH and third part with Na<sub>2</sub>CO<sub>3</sub>. The precursor to activating agent was taken in the ratio 1:1 (w/w), mixed thoroughly and left for 24 h at room temperature for proper soaking. Then after, the soaked samples were evaporated at 110 °C in an oven to obtain dry samples. This method is in practice for preparation of the activated carbons [41, 42, 43]. The weighed amount of impregnated precursor was introduced into the horizontal tubular furnace for carbonization at 400 °C under the continuous flow of N<sub>2</sub> for 3h. The tube furnace was turned off to cool down the samples maintaining inert atmosphere of nitrogen. The hot distilled water was used for washing the cooled samples for several times followed by washing with cold distilled water. The process was continued until washed water became neutral and ions of activating agents also get removed. Finally, the obtained samples were dried at 110 °C, stored and used for advanced characterization and electrochemical application. Thus obtained three different as prepared ACs were then named on the basis of activating agent and carbonization condition (Table 1).

# 2.2. Characterization

The pyrolytic behaviors of the raw wood dust powder were investigated by TGA/DSC (SDT Q600 V20.9 Build 20). The morphology of the as prepared ACs was determined by Scanning Electron Microscope (SEM, Nanoeye, Korea). The samples was gold coated prior to SEM observation. Phase state was evaluated by using X-ray Diffraction (XRD, Rigaku RINT 2000 Diffractometer). Similarly, surface carbon-oxygen functional groups present in samples were identified performing FT-IR measurements using FT-IR instrument (Bruker, Vertex 70, Germany). The %

 Table 1. Name of as prepared ACs on the basis of activating agents and carbonization conditions.

Name of ACs	Activating agent	Carbonization temp (°C)	Carbonization time (h)	Atmosphere
Sr–H <sub>3</sub> PO <sub>4</sub>	H <sub>3</sub> PO <sub>4</sub>	400	3	N <sub>2</sub>
Sr–KOH	КОН	400	3	N <sub>2</sub>
Sr–Na <sub>2</sub> CO <sub>3</sub>	Na <sub>2</sub> CO <sub>3</sub>	400	3	N <sub>2</sub>

transmission of samples was recorded over  $4000-400 \text{ cm}^{-1}$ . The presence of the amorphous ACs was confirmed by Raman signal [labRAM HR800 (JOBIN YVON)]. The surface area of the ACs was measured by BET (Micromeritics ASAP 2020 system).

# 2.2.1. Fabrication of activated-carbon-electrodes

In order to fabricate activated-carbon-electrodes, 8 mg of each of the three as prepared ACs (Sr–H<sub>3</sub>PO<sub>4</sub>, Sr–KOH and Sr–Na<sub>2</sub>CO<sub>3</sub>) was mixed separately with 1 mg of carbon black and 1 mg of PVDF. The carbon black was used to provide sufficient conductivity whereas PVDF acted as a binder. To disperse PVDF, 200  $\mu$ L NMP solution was added to the mixture and it was grinded and mixed thoroughly. Then the mixture solution was dropped with the help of micro-pipette on 1 cm<sup>2</sup> Ni-foam, that acts as a current collector and dried thoroughly at 70 °C in an oven for overnight. Thus, activated-carbon-electrodes were fabricated which have been used as working electrodes.

# 2.2.2. Experimental set up for electrochemical characterization

The three-electrode electrochemical set up was established for electrochemical characterization in which as fabricated activated-carbonelectrodes was used as working electrode, platinum-plate was used as counter electrode and Ag/AgCl was used as a reference electrode. Here, 6M aqueous KOH was used as an electrolyte. Prior to measurement, activated-carbon-electrodes were soaked in aqueous KOH for overnight. A Metrohm Autolab (PGSTAT 302 N) potentiostat/galvanostat was used to perform electrochemical measurement. The supercapacitor performances were evaluated by (1) cyclic voltammetry (CV), (2) galvanostatic charge/discharge (GCD) and (3) electrochemical impedance spectroscopy (EIS).

For the CV measurement, a potential window of -1.0 to -0.2 V was used at the scan rates of 2, 5, 10, 20, 50 and 100 mV/s. The GCD measurement was performed at current densities of 1, 2, 3, 5, 10, 15 and 20 A/g over the same potential window as the CV measurements. The EIS was studied over the frequency range of 100 kHz to 0.1 Hz at the perturbation signal of 10 mV. The circuits of the cells were evaluated using Nova 1.1 software.

# 3. Results and discussion

# 3.1. Thermogravimetric analysis (TGA)/Differential scanning calorimetry (DSC)

TGA/DSC measurements were carried out to understand the pyrolytic behavior of the wood dust powder/precursor. Figure 1 illustrates the TG and DSC curves of raw wood dust powder of *Shorea robusta* (Sal). A slight weight loss was observed from 60-100  $^{\circ}$ C under the N<sub>2</sub> flow of 60 mL/min in TG curve due to dehydration. This was also obvious from a sharp



Figure 1. TGA/DSC curves of raw sawdust power (Sr-Raw).

endothermic peak at 100 °C from DSC. A shallow peak in TG plot at around 200–300 °C might be due to decomposition of hemicellulose which was substantially completed at 300–310 °C [23] is confirmed from DSC curve, where a slight weight loss was clearly observed in between 200-300 °C. A significant weight loss was observed in 300–400 °C from TG, which was confirmed by a sharp endothermic peak at around 390 °C of DSC. This weight loss indicates the breakdown of cellulose into components like carboxyl, lactone, lacto and carbonyl. The conversion of cellulose into organic volatile matters leading to desorption of CO<sub>2</sub> and CO might also be the reason for the weight loss. It exhibited that about 60% of weight loss occurs around 400 °C [35]. Samples are more stable beyond 400 °C. Therefore, 400 °C was found to be suitable for carbonization process [42].

# 3.2. X-ray diffraction (XRD) measurement

Figure 2 shows the XRD pattern of the as prepared ACs. The XRD pattern of raw precursor (Sr-Raw) showed peaks at around 15 and 22, 20 degree (Bragg's angle), whereas in chemically activated carbonized samples; Sr-H<sub>3</sub>PO<sub>4</sub>, Sr-KOH and Sr-Na<sub>2</sub>CO<sub>3</sub>, those two sharp peaks were found to be disappeared completely. But broad peak was seen at around 26, 20 degrees corresponding to (002) diffraction of the disordered stacking of the microstructures. Since no higher and sharper peak at (002) plane was seen, which confirms the absence of degree of graphitization. The broad peaks at around 20 to 30, 20 degree indicates the presence of amorphous nature of carbon. Besides this, there was no other peaks visualized in XRD pattern which revealed that there is no any other effect of activating agents like H<sub>3</sub>PO<sub>4</sub>, KOH and Na<sub>2</sub>CO<sub>3</sub>. However, some peaks could be seen in Sr-H<sub>3</sub>PO<sub>4</sub> even after careful washing, which might be due to some impurities or moisture. It is common to have some shoot off peak due to impurities from activating agent. The effect of such impurities on the performance of materials has not been substantiated yet [43].

# 3.3. Raman scattering analysis

Figure 3 presents the Raman spectra of as prepared ACs, which demonstrated strong D and G bands approximately at 1355 cm<sup>-1</sup> and 1598 cm<sup>-1</sup> respectively. Here D band at 1355 cm<sup>-1</sup> indicated for disordered carbon in all as prepared ACs (Sr–H<sub>3</sub>PO<sub>4</sub>, Sr–KOH and Sr–Na<sub>2</sub>CO<sub>3</sub>). Similarly, G band was obvious in sample Sr–H<sub>3</sub>PO<sub>4</sub> and Sr–Na<sub>2</sub>CO<sub>3</sub> while in Sr–KOH, it was found to be small. The peak intensity ratio of G band to D band [I(G)/I(D)] provides semi quantitative information of crystallization degree of graphitic carbon in all as prepared samples. The ratio I(G)/I(D) for Sr–H<sub>3</sub>PO<sub>4</sub>, Sr–KOH and Sr–Na<sub>2</sub>CO<sub>3</sub> was found to be 1.1, 1.1



Figure 2. XRD pattern of Sr– $H_3PO_4$ , Sr–KOH and Sr– $Na_2CO_3$  along with Sr-Raw.



Figure 3. Raman spectra of Sr-H<sub>3</sub>PO<sub>4</sub>, Sr-KOH and Sr-Na<sub>2</sub>CO<sub>3</sub>.

and 1.2 respectively. The ratio I(G)/I(D) was found to be approximately 1 in all the ACs showing the formation of disordered carbon (amorphous carbon) [44]. The similar results have been confirmed by XRD pattern.

# 3.4. Fourier transforms infrared (FTIR) analysis

Figure 4 shows the FTIR spectra of Sr-H<sub>3</sub>PO<sub>4</sub>, Sr-KOH and Sr-Na2CO3. FT-IR spectra of "raw sawdust powder" of Shorea rabusta (Sr-Raw) showed the adsorption band at around 3331  $\text{cm}^{-1}$  which was assigned to -OH stretching of carboxyl, phenol and alcohol vibration and adsorbed water. The presence of this band in Sr-H<sub>3</sub>PO<sub>4</sub> is indicative of better functionalization of carbon. Though it looks to be disappeared from other samples but, in fact it has not disappeared rather decreased. This is due to presence of -OH group. It is also supported from thermal analysis (TGA/DSC). The TGA/DSC curve (Figure 1) clearly shows mass loss at around 100 °C, indicating evaporation of water molecules. The band at around 2942 cm<sup>-1</sup> is assigned to aliphatic C-H stretching adsorption. The band at around 2333 cm<sup>-1</sup> (2330-2359 cm<sup>-1</sup>) is assigned for  $CH_3$  of alkane. The broad band at 1580 cm<sup>-1</sup> (in the region 1500–1900 cm<sup>-1</sup>) is due to C–C vibrations in aromatic rings. Similarly, the broad band observed in the region 1000-1300 cm<sup>-1</sup> is usually focused with oxidized carbon and has been assigned to C-O stretching in acids, alcohols, phenols, ethers and/or esters groups [45]. This point is also supported by TGA/DSC curve (Figure 1), which clearly shows significant mass loss between 300 to 400 °C. This is due to the breakdown of



cellulose into carboxyl, lactone and lacto groups, which is due to desorption of  $CO_2$  and CO, converting cellulose into organic volatile matters present in the sawdust [46]. These results are in good agreement with the findings of many investigators [45, 46]. The FTIR spectra show that the effective functionalization of precursor by the activating agents used in this study.

#### 3.5. $N_2$ adsorption/desorption isotherm

Figure 5 shows the N<sub>2</sub> adsorption/desorption isotherms at 77 K for all as prepared ACs. As can be seen in Figure 5, the isotherms showed a drastic uptake indicating the presence of micropores at lower relative pressure region ( $P/P^0 < 0.1$ ) in all the ACs. The uptake in lower relative pressure seems to decrease in KOH and Na<sub>2</sub>CO<sub>3</sub> activated carbon samples; however, at high relative pressure, the plateau was reached. It indicates that, these ACs namely Sr-KOH and Sr-Na<sub>2</sub>CO<sub>3</sub> consisted of micropores. The amount of nitrogen uptake in Sr-H<sub>3</sub>PO<sub>4</sub> has increased significantly at the pressure  $P/P^0 = 0.4$ , showing hysteresis loop during adsorption and desorption of nitrogen. Such hysteresis loop did not exist in the adsorption/desorption curve for Sr-KOH and Sr-Na<sub>2</sub>CO<sub>3</sub> samples. As the relative pressure increases up to  $P/P^0 \cong 0.5$ , the adsorption volume of Sr-H<sub>3</sub>PO<sub>4</sub> also increased. With further increase of pressure, widely opened knees and the clear hysteresis loop were observed. This was indicative of the presence of a considerable amount of small mesopores in Sr-H<sub>2</sub>PO<sub>4</sub>.

The BET specific surface areas of Sr–H<sub>3</sub>PO<sub>4</sub>, Sr–KOH and Sr–Na<sub>2</sub>CO<sub>3</sub> along with commercial carbon were calculated by BJH adsorption method which is given in Table 2. Among the three different samples, Sr–H<sub>3</sub>PO<sub>4</sub> showed highest BET surface area of 1269.5 m<sup>2</sup>/g followed by 280.6 m<sup>2</sup>/g of Sr–KOH and 58.9 m<sup>2</sup>/g of Sr–Na<sub>2</sub>CO<sub>3</sub>. It might be due to the presence of innumerable mesoporosity along with microporosity in the case of phosphoric acid activated carbon (Sr–H<sub>3</sub>PO<sub>4</sub>) [35]. The low surface area of Sr–KOH and Sr–Na<sub>2</sub>CO<sub>3</sub> may be due to low carbonization temperature of 400 °C for KOH and Na<sub>2</sub>CO<sub>3</sub> activation. As reported in literature [47, 48], the activation temperature should be > 700 °C which can react with carbon to generate pores. The SEM image of Sr–H<sub>3</sub>PO<sub>4</sub> sample (Figure 6 a.) also showed clear honey comb like structure with extensive porosity.

# 3.6. Scanning electron microscopy (SEM) analysis

Figure 6 a-c illustrate the SEM micrographs of Sr–H<sub>3</sub>PO<sub>4</sub>, Sr–KOH and Sr–Na<sub>2</sub>CO<sub>3</sub> at the magnification of x2.0k showing nice morphologies with pores. The evaporation of activating agents  $H_3PO_4$ , KOH and Na<sub>2</sub>CO<sub>3</sub> during carbonization process leads to the pore development on



Figure 5.  $N_2$  adsorption/desorption isotherms at 77 K for Sr-H<sub>3</sub>PO<sub>4</sub>, Sr-KOH and Sr-Na<sub>2</sub>CO<sub>3</sub>.

Table 2.	BET	Specific	surface	area	of	as	prepared	Activated	Carbo
samples.									

Name of Sample	Specific Surface Area (m <sup>2</sup> /g)
Sr–H <sub>3</sub> PO <sub>4</sub>	1269.5
Sr–KOH–4S	280.6
Sr–Na <sub>2</sub> CO <sub>3</sub>	58.9
Commercial carbon	876.02

the surfaces of the ACs [49]. Among them,  $H_3PO_4$  activated carbon sample i.e.  $Sr-H_3PO_4$ , exhibited honeycomb like internal structure with well-developed pores. Among the three activating agents,  $H_3PO_4$  is considered as the good dehydrating agent than KOH and  $Na_2CO_3$  as in Sr-KOH and Sr-Na<sub>2</sub>CO<sub>3</sub>, pores are not very obvious. It may be due to low carbonization temperature for Sr-KOH and Sr-Na<sub>2</sub>CO<sub>3</sub> [50]. The well-developed pores in Sr-H<sub>3</sub>PO<sub>4</sub> led to high BET surface area.

#### 3.7. Supercapacitive performances activated-carbon-electrodes

Though the surface area and porosity of Sr–KOH and Sr–Na<sub>2</sub>CO<sub>3</sub> were not very satisfactory as compared to Sr–H<sub>3</sub>PO<sub>4</sub>, yet their electrodes (activated-carbon-electrodes) were also used for supercapacitive performances and results were compared.

# 3.7.1. Cyclic voltammetry (CV) studies of activated carbon electrode

Comparative CV of Sr-H<sub>3</sub>PO<sub>4</sub>, Sr-KOH and Sr-Na<sub>2</sub>CO<sub>3</sub> electrodes and commercial carbon-electrode at 100 mV/s scan rate are shown in Figure 7. The CV at various scan rates of 2 mV/s, 5 mV/s, 10 mV/s, 20 mV/s and 50 mV/s were also obtained (not shown in figure). The CV were found to be approximately rectangular in shape which indicates the electrical double layer capacitive (EDLC) behavior. Among these three as prepared activated-carbon-electrodes, Sr-H<sub>3</sub>PO<sub>4</sub>-electrode, behaved well and rectangular shape (black line) was found to be comparable with commercial carbon electrode (pink line). The redox peaks were not observed in the CV however the larger loop area could be seen which is associated to high specific capacitance. Nature of such CV was evidenced for the EDLC processes which was reported in our previous study [51]. Hence, charge storage was found to be totally EDLC type. A maximum current of about 13 A/g was found which was significantly high, whereas in case of Sr-KOH-electrode and Sr-Na<sub>2</sub>CO<sub>3</sub>-electrode, CV were not very rectangular and current density was also found to be truncated. That's why their electrochemical capacitive behavior was deprived in comparison to Sr-H<sub>3</sub>PO<sub>4</sub>-electrode.

### 3.7.2. Specific capacitance studies of activated carbon-electrodes

The specific capacitance  $(C_{SP})$  of activated carbon-electrodes was calculated using Eq. (1)

$$C_{\rm SP} = \frac{I\Delta t}{m\Delta V} \tag{1}$$

where, discharge current is I (A), discharge time is  $\Delta t(s)$  in the potential window of  $\Delta V(V)$  and the mass of active material is m (g).

The specific capacitance value of all the three as prepared activated carbon-electrodes are shown in Table 3. As can be seen, the highest specific capacitance (136.3 F/g) has been obtained for Sr–H<sub>3</sub>PO<sub>4</sub> electrode, which is even higher than commercial-activated-carbon-electrode (91.6 F/g).

Figure 8 a. shows the plot of specific capacitances as a function of current density. The curve shows that there was a sharp decrease of specific capacitance with increase in current density up to 5 A/g in all three electrodes. Then after 5 A/g, specific capacitance was found to be approximately constant. The specific capacitance of  $Sr-H_3PO_4$  electrode yielded a value of 136.3 F/g at 1 A/g and kept a value of 126.5 F/g at 2 A/g, 121.9 F/g at 3 A/g, 115.9 F/g at 5 A/g, 107.2 F/g at 10 A/g, 101.5 F/g at 15 A/g and 97.3 F/g at 20 A/g, that appeared to be much higher than other two carbon electrodes namely Sr-KOH-electrode and Sr-Na<sub>2</sub>CO<sub>3</sub>-electrode, which are illustrated in Table 4. Figure 8 b shows the comparison of specific conductance of Sr-H<sub>3</sub>PO<sub>4</sub>-electrode, Sr-KOH-electrode and Sr-Na<sub>2</sub>CO<sub>3</sub>-electrode - with commercial-activated-carbon at 1 A/g current density. The highest specific capacitance of Sr-H<sub>3</sub>PO<sub>4</sub>-



**Figure 7.** Comparison of CV curves of Sr–H<sub>3</sub>PO<sub>4</sub>, Sr–KOH and Sr–Na<sub>2</sub>CO<sub>3</sub> electrodes along with commercial carbon electrode at 100 mV/s.

**Table 3.** Name of activated carbon-electrodes and their specific capacitance from GCD (at 1 A/g).

S.N.	Name of the activated-carbon- electrodes	Specific Capacitance (F/g)
1	Sr–Na <sub>2</sub> CO <sub>3</sub> -electrode	59.143
2	Sr–KOH-electrode	42.189
3	Sr–H <sub>3</sub> PO <sub>4</sub> -electrode	136.339
4	Commercial-activated-carbon-electrode	91.678



Sr-H<sub>3</sub>PO<sub>4</sub>

Sr-KOH

Sr-Na<sub>2</sub>CO<sub>3</sub>

Figure 6. SEM images of (a) Sr–H<sub>3</sub>PO<sub>4</sub> (b) Sr–KOH and (c) Sr–Na<sub>2</sub>CO<sub>3</sub>.



Figure 8. (a) Specific capacitance of three activated-carbon-electrodes as a function of current density.(b) Comparison of specific capacitance of three activated-carbon-electrodes along with commercial-activated-carbon electrode.

Table 4. Carbon-electrodes and their specific capacitance from GCD at different current densities.

S.N.	Activated-Carbon- electrodes	Specific capacitance $C_{SP}$ (F/g) at different current densities						
		1 A/g	2 A/g	3 A/g	5 A/g	10 A/g	15 A/g	20 A/g
1	Sr-H <sub>3</sub> PO <sub>4</sub>	59.1	43.9	37.3	30.0	21.8	18.2	16.5
2	Sr–KOH	42.2	28.3	23.2	18.7	15.2	14.0	12.8
3	Sr-Na <sub>2</sub> CO <sub>3</sub>	136.3	126.5	121.9	115.9	107.2	101.5	97.3

electrode could be clearly perceived in Figure 8 b. and observed even higher than commercial-activated-carbon-electrode which was reported in our previous study [51].

# 3.7.3. Galvanostatic charge/discharge (GCD) studies of activated-carbonelectrodes

Galvanostatic charge/discharge (GCD) process is an analytical technique to calculate energy and power densities of the activated-carbonelectrodes. It provides the idea about the feasibility of applying these carbon for practical device fabrication.

Figure 9 shows the comparative charge discharge curves of Sr–H<sub>3</sub>PO<sub>4</sub>, Sr–KOH and Sr–Na<sub>2</sub>CO<sub>3</sub>-electrodes and commercial-activated-carbonelectrode at 1 A/g current density. One can see the equivalent and triangular shape charge discharge curves. Here the time taken for discharge was 60 s for Sr–KOH-electrode, 80 s for Sr–Na<sub>2</sub>CO<sub>3</sub>-electrode, 140 s for commercial carbon-electrode, whereas 210 s for Sr–H<sub>3</sub>PO<sub>4</sub>electrode which is comparatively quite higher than Sr–KOH, Sr–Na<sub>2</sub>CO<sub>3</sub> and commercial carbon electrodes respectively. The large discharge duration of Sr–H<sub>3</sub>PO<sub>4</sub> may be due to high specific capacitance 136.3 F/g. The straight lines of charge/discharge curve and triangular shape further confirms the presence of EDLC behavior. It was also an indication of having low impurities and proper insertion between ACs and electrolyte ions [50] and again could be interconnected with high surface area 1269.5 m<sup>2</sup>/g as well.

Here, we have calculated the energy and power densities of all three as prepared electrodes using Eqs. (2) and (3) [50]. A well-known "Ragone plot" is shown in Figure 10. In plot, one can clearly observe the outstripped energy density efficiency of the  $Sr-H_3PO_4$  electrode than Sr-KOH and  $Sr-Na_2CO_3$  electrodes which are clearly shown in Table 5. It was also obvious that  $Sr-H_3PO_4$  electrode has maximum energy storage capability of 3.0 Wh/kg at a power density of 99.6 W/kg, demonstrating the faster charging ability (Table 5). So one can say that  $Sr-H_3PO_4$ electrode is one of the potential candidates for EDLC application in supercapacitors. Energy and power densities are calculated by using Eqs. (2) and (3).

$$\mathbf{E} = \frac{1}{8} C_{SP} \Delta \boldsymbol{\nu}^2 \tag{2}$$



**Figure 9.** Comparison of GCD curves of Sr–H<sub>3</sub>PO<sub>4</sub>, Sr–KOH and Sr–Na<sub>2</sub>CO<sub>3</sub> electrodes at 1A/g along with commercial carbon electrode.

$$\mathbf{P} = \frac{E}{\Delta t} \tag{3}$$

where, E is the energy density in Wh/kg, P is the power density in W/kg, Csp is the specific capacitance in F/g,  $\Delta V$  is the potential window (V), whereas  $\Delta t$  (s) is the time of discharge.

# 3.7.4. Cyclic stability of activated carbon electrodes

Long life cycle is one of the vivacious characteristics that an ideal supercapacitor should have [50]. Figure 11 shows the life cycle assessment of three as prepared activated carbon electrodes. Here, 1000 cycles at a current densities of 3 A/g has been applied. It was found that the specific capacitance of the electrodes was slightly decreased at the beginning of cycle. However, it was found to be constant after 140 cycles. It was also observed that the specific capacitance was found to be dropped by 3.0 % for Sr–H<sub>3</sub>PO<sub>4</sub>-electrode, 10.6 % for Sr–KOH-electrode and 21.6% for Sr–Na<sub>2</sub>CO<sub>3</sub>-electrode after 1000 cycles (Table 5). This decay can be attributed to the decomposition of some electroactive



Figure 10. Ragone plots of Sr-H<sub>3</sub>PO<sub>4</sub>, Sr-KOH and Sr-Na<sub>2</sub>CO<sub>3</sub> electrodes.

Table 5. Activated-carbon-electrode and their specific capacitance, Energy density, Power density, Capacity retention and ESR values.

S.N	Activated- carbon- electrodes	Specific Capacitance (F/g)	Energy density (Wh/Kg) at (1 A/g)	Power density (W/Kg) at (1 A/g)	Capacity retention %	ESR (Ω)
1	Sr-Na <sub>2</sub> CO <sub>3</sub>	59.1	1.8	95.1	78.4	0.6
2	Sr–KOH	42.1	0.8	95.2	89.4	0.5
3	Sr-H <sub>3</sub> PO <sub>4</sub>	136.3	3.0	99.6	96.9	0.4



Figure 11. Long term cycling performance of Sr-H<sub>3</sub>PO<sub>4</sub>, Sr-KOH and Sr-Na<sub>2</sub>CO<sub>3</sub> electrodes.

materials into the electrolyte during continuous charge/discharge assessment [52]. It revealed that  $Sr-H_3PO_4$ -electrode has excellent cycle stability than other 2 activated-carbon-electrodes. It may be due to high value of specific capacitance, high energy density and high % capacity retention of about 96.9 %. These aforesaid values might be due to enormous mesopores, along with micropores. These porosity not only improves the capacitance value but also increases the capacity retention/cyclic stability. However, Sarkar et al. has mentioned that some phosphorous group may remain on the surface which may contribute to enhance the capacitance by decreasing the surface hydrophobicity, and the metaphosphate which contributes to the stability of the capacitance [50]. Furthermore, its higher mesoporosity, facilitates the diffusion at higher current densities. Therefore, it could be resolved that both the

specific capacitance and capacity retention can be improved by controlling the pore structure and phosphorus environment as well.

# 3.7.5. Electrochemical impedance spectroscopy (EIS) studies of activated carbon materials

Electrochemical impedance spectroscopy (EIS) is an important analytical technique to get information about the characteristic frequency response of supercapacitors and the capacitive phenomena occurring in the electrodes. The Nyquist impedance plots of the three different activated carbon electrodes are reported in Figure 12.

Figure 12 exhibited enlarged spectrum of Nyquist impedance plot. High frequency region of the plot displayed an incomplete semi circle whereas, low frequency region exhibited straight line curve. The semicircle curve of Nyquist impedance plot were associated with polarization resistance which reflects the diffusion or transport of electrolyte into the porous electrode material, while the straight verticle line was related to the capacitive behavior as described by Lu et al. [52] and Toupin et al [53].

The result implies that Sr–H<sub>3</sub>PO<sub>4</sub>-electrode, exhibited the highest conductivity or lowest internal resistance of about 0.4  $\Omega$  [53] than other two carbon electrodes namely Sr–KOH-electrode and Sr–Na<sub>2</sub>CO<sub>3</sub>-electrode (Table 5). This point attributed to high specific capacitance of Sr–H<sub>3</sub>PO<sub>4</sub> electrode which significantly improves the ion diffusion and electron conduction procedure which has also been reflected by the higher power density of the Sr–H<sub>3</sub>PO<sub>4</sub>-electrode.

# 4. Conclusion

On the basis of results, following conclusions can be drawn.

The agro-waste, sawdust of *Shorea robusta* (Sal), can be used as good precursor materials for the production of nanoporous activated carbons. The carbonization temperature of 400  $^{\circ}$ C was found to be adequate to prepare phosphoric acid activated carbon (Sr–H<sub>3</sub>PO<sub>4</sub>), while this temperature was found to be insufficient for KOH and Na<sub>2</sub>CO<sub>3</sub> activated carbon (Sr–KOH and Sr–Na<sub>2</sub>CO<sub>3</sub>).

Among the three ACs,  $H_3PO_4$  activated carbon (Sr– $H_3PO_4$ ) showed best results in terms of porosity, surface area and electrochemical performances.

The overall electrochemical performance of Sr–H<sub>3</sub>PO<sub>4</sub>-electrode showed excellent capacitive behavior. The specific capacitance was found to be 136.3 F/g. The energy density was found to be 3.0 Wh/kg at power density of 99.6 W/kg, and low internal resistance of about 0.4  $\Omega$ . In the same way, cyclic stability was significantly high, 96.9 % after 1000 cycles of charge/discharge. Such type of behavior clearly indicates the



Figure 12. Nyquist plot of Sr-H<sub>3</sub>PO<sub>4</sub>, Sr-KOH and Sr-Na<sub>2</sub>CO<sub>3</sub> electrodes at high frequency.

potentiality of the electrode for EDLC application in supercapacitive energy storage.

# Declarations

# Author contribution statement

Dibyashree shrestha: Performed experiments; Analysed the data; wrote the manuscript.

Armila Rajbhandari: Conceived and designed the experiments.

#### Funding statement

This research did not receive any specific grant from funding agencies in the public, commercial, or not-for-profit sectors.

### Data availability statement

Data associated with this study has been deposited at Patan Multiple Campus, Tribhuvan University, Patan dhoka, Lalitpur, Nepal.

#### Declaration of interests statement

The authors declare no conflict of interest.

### Additional information

No additional information is available for this paper.

#### Acknowledgements

Dibyashree Shrestha is thankful to "Institute of Science and Technology" (IOST), Tribhuvan University, Kirtipur, Kathmandu, Nepal for the Ph.D. study leave. The author is very grateful to Prof. Dr. Soo Wohn Lee of "Research Centre for Eco Multi-Functional Nano Material, Global Research Laboratory (GRL), Sun Moon University (SMU), South Korea" for providing laboratory facilities to carry out all advanced instrumental characterizations. Similarly, the author is thankful to Prof. Dr. Santi Maensiri of "School of Physics, Institute of Science, Suranaree University of Technology (SUT), Thailand for providing laboratory facilities to carry out electrochemical characterizations.

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