

Ion-Triggered Hydrogels Self-Assembled from Statistical Copolypeptides

Bing Wu,* Saltuk B. Hanay, Scott D. Kimmins, Sally-Ann Cryan, Daniel Hermida Merino, and Andreas Heise*



Cite This: *ACS Macro Lett.* 2022, 11, 323–328



Read Online

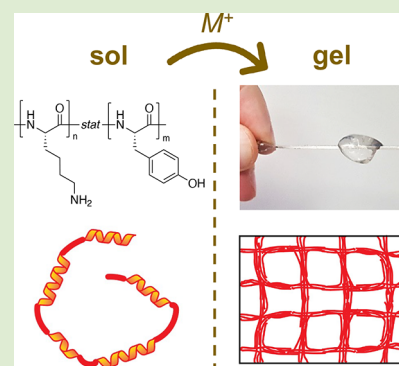
ACCESS |

Metrics & More

Article Recommendations

Supporting Information

ABSTRACT: Statistical copolypeptides comprising lysine and tyrosine with unprecedented ion-induced gelation behavior are reported. Copolypeptides are obtained by one-step *N*-carboxyanhydride (NCA) ring-opening polymerization. The gelation mechanism is studied by in situ SAXS analyses, in addition to optical spectroscopy and transmission electron microscopy (TEM). It is found that the gelation of these statistically polymerized polypeptides is due to the formation of stable intermolecular β -sheet secondary structures induced by the presence of salt ions as well as the aggregation of an α -helix between the copolypeptides. This behavior is unique to the statistical lysine/tyrosine copolypeptides and was not observed in any other amino acid combination or arrangement. Furthermore, the diffusion and mechanical properties of these hydrogels can be tuned through tailoring the polypeptide chain length and ion strength.



Hydrogels comprise three-dimensional cross-linked networks, allowing them to hold significant amounts of water, which makes them attractive materials for biomedical applications such as drug delivery,¹ wound dressing,² antimicrobial coatings,³ and regenerative medicine.⁴ Natural hydrogels based on proteins (e.g., collagen, gelatin, and fibroin) are often the materials of choice due to their inherent biocompatibility. They form physical hydrogels through peptide-based biomacromolecule self-assembly.^{5–8} Although these natural polymers have drawn large attention from the biomaterials research community, the difficulty in their purification motivated researchers to explore synthetic peptide analogues. Two main synthetic approaches exist to design peptides capable of forming physical hydrogels.⁹ The first is based on the synthesis of peptides with precisely controlled amino acid sequences, of which several examples have been reported to form physical hydrogels either spontaneously or upon a trigger.^{10–13} While predominantly oligomeric, these peptides vary in length and share the common feature of forming self-assembled nanostructures through hydrophobic, ionic, hydrogen bonding, or secondary structure interaction.^{14–16} These typical higher-order motives include nanotubes, nanotapes, or nanospheres, which then aggregate to form hydrogel networks.

The second approach uses polymerization-derived polypeptides. The dominant design motive in these hydrogel-forming polypeptides is based on block structures.^{17,18} This includes hybrid block and graft copolymers containing a polypeptide and a synthetic block or pure polypeptide block copolymers. The latter are readily accessible by sequential amino acid *N*-

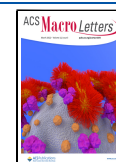
carboxyanhydride (NCA) polymerization.^{19,20} Similar to oligopeptides, three-dimensional networks in aqueous media are formed through hydrophobic, ionic, as well as secondary structure interactions such as β -sheet- or α -helix-driven molecular arrangements.²¹ For example, Deming reported diblock copolypeptides incorporating oppositely charged ionic blocks that form β -sheet-structured hydrogel assemblies via polyion complexation.²² We and others have reported hydrogels through hydrophobic interaction from amphiphilic linear as well as branched block copolypeptides comprising lysine or glutamic acid in their hydrophilic blocks and phenylalanine, isoleucine, leucine, or alanine in their hydrophobic blocks.^{23–25}

Statistical copolypeptides, which are copolymers obtained in a one-step binary copolymerization of NCAs, have not been proposed as hydrogelators to date, as it is indeed counter-intuitive to assume their gelation due to the absence of defined blocks suitable for self-assembly. Here we disclose the first example of a statistical copolypeptide comprising lysine (Lys) and tyrosine (Tyr) that can form hydrogels upon addition of buffer salt solution. While for some natural proteins and sequence-defined oligopeptides salt-triggered gelation is known,^{26,27} salt usually compromises the stability of block

Received: December 10, 2021

Accepted: February 10, 2022

Published: February 16, 2022



copolypeptide hydrogels due to the destructive impact salts have on amphiphilic interaction.²⁸ The statistical Lys/Tyr copolypeptides thus display properties otherwise only found in sequence-controlled oligopeptides, making them materials with an unprecedented structure/property profile. Through detailed characterization, we propose a unique mechanism explaining the salt-triggered transition from single copolypeptides to porous hydrogel networks. Moreover, preliminary physico-chemical properties of the porous hydrogels (e.g., mechanical property and diffusion property) are discussed.

Two statistical copolypeptides, poly(L-lysine-*stat*-L-tyrosine), p(Lys_xTyr_y), with total degrees of polymerization of 100 and 150 were synthesized from the protected monomers maintaining a fixed *N*-ε-carbobenzyloxy-L-lysine (L-Lys(Z)) to *O*-benzyl-L-tyrosine (L-Tyr(Bzl)) ratio of 4:1 (Scheme S1).²⁹ ¹H NMR analysis confirmed the copolypeptide composition (Figures S1 and S2), while size exclusion chromatography (SEC) showed monomodal traces with dispersities (*D*) around 1.2 and relative shifts in agreement with the targeted degrees of polymerization (Figure S5). After deprotection, no gelation was observed in DI water for any of these two copolypeptides. After mixing with phosphate-buffered saline (PBS) solutions of different concentrations, both copolypeptides were found to form self-supporting hydrogels over time at a minimum copolypeptide concentration of 1.9 mM (Table 1, Figure 1). Furthermore, mixing of

Table 1. Dependence of p(Lys₈₀Tyr₂₀) Gelation (Vial Inversion) on Copolypeptide and PBS Buffer Concentration

M (copolypeptides)	PBS concentration			
	150 mM	100 mM	50 mM	25 mM
1 mM	no gel	no gel	no gel	no gel
1.9 mM	gel	gel	gel	no gel
3.8 mM	gel	gel	gel	no gel
7.6 mM	gel	gel	gel	no gel

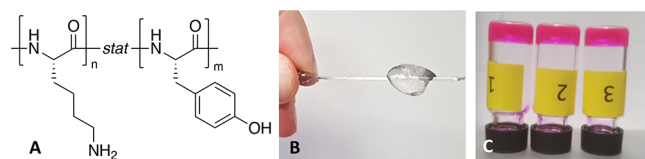


Figure 1. Structure of statistical copolypeptide poly(L-lysine-*stat*-L-tyrosine) (A), image of hydrogel in PBS buffer on a spatula (B), and examples of hydrogels dyed with Rhodamine B.

the copolypeptide DI water solution with CaCl₂ or MgCl₂ induced rapid yet heterogeneous gelation, suggesting that the gelation mechanism of these copolypeptides is induced by the presence of charged ions (e.g., Na⁺ in PBS buffer). For the following studies, PBS solution, due to its weak ion strength, was chosen for initiating the gelation to avoid the morphological inhomogeneity caused by rapid gelation (concentration: 2.5 and 3.8 mM for p(Lys₁₂₀Tyr₃₀) and p(Lys₈₀Tyr₂₀)).

To elucidate the mechanism of this unusual gelation, the hydrogel sample composed of p(Lys₈₀Tyr₂₀) formed in PBS D₂O was first analyzed by FTIR spectroscopy. As shown in Figure 2a, a predominant peak can be found at ca. 1624 cm⁻¹ in the polypeptide hydrogel's amide I band, indicating the formation of antiparallel β-sheet secondary structures in the copolypeptide hydrogel sample.^{30–32} An additional weak band

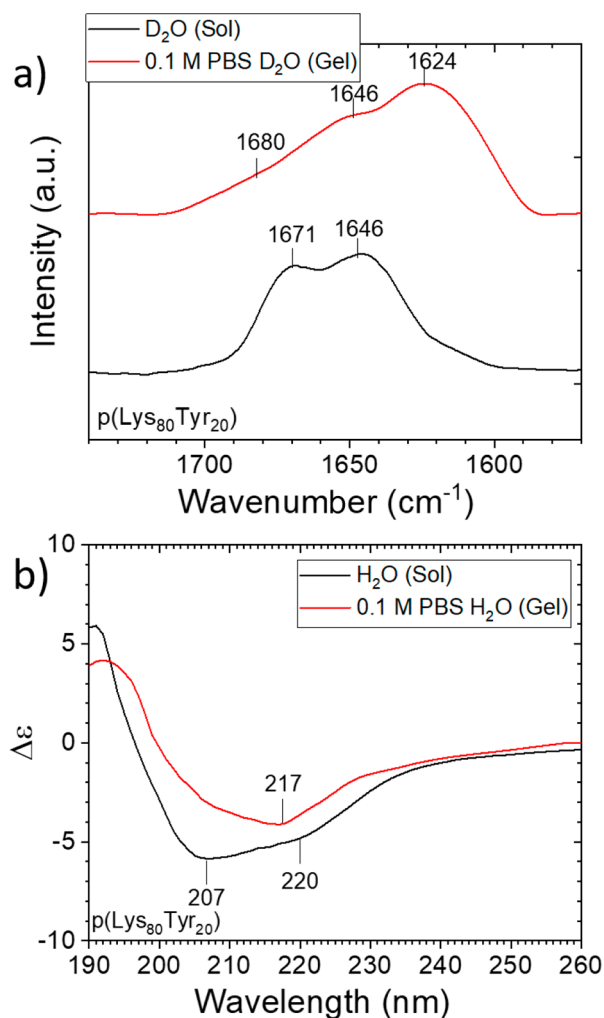


Figure 2. (a) FTIR analyses of poly(Lys₈₀Tyr₂₀) recorded in D₂O and 2 h after mixing with 0.1 M PBS buffer D₂O solution (for deconvolution, see Figure S9). (b) CD analyses of poly(Lys₈₀Tyr₂₀) recorded in H₂O and 2 h after mixing with 0.1 M PBS buffer H₂O solution.

near 1680 cm⁻¹ originating from this antiparallel β-sheet structure can also be observed.³³ On the other hand, a strong band near 1646 cm⁻¹ suggests the presence of α-helical secondary structures, which is quite common for polylysine in basic solution.³⁴ Notably, it has been reported that Tyr-Lys pairs can stabilize the helical structure.³⁵ Interestingly, no observable β-sheet secondary structure can be found in this copolypeptide's nongelling D₂O solution in the absence of salt ions. Instead, in addition to the absorption belonging to the α-helix structure, a strong IR band appeared near 1671 cm⁻¹, indicating a dominating secondary “β-turn” structure.³⁶ Circular dichroism analyses further supported the FTIR findings (Figure 2b). The minimum at 217 nm for the polypeptide gel formed 2 h after mixing with PBS suggests predominant antiparallel β-sheet structures,³⁷ while the two minima at 207 and 220 nm observed in the CD spectrum of the H₂O solution alone are indicative fingerprint minima of the α-helix secondary structure in polypeptides.³⁸

Next, the p(Lys₈₀Tyr₂₀) polypeptide gelation process was investigated by fluorescence spectroscopy. The spectrum of the polypeptide in 0.1 M PBS buffer solution taken after 2 h of mixing showed a significant increase in the intensity of the

peak around 305 nm compared to the same copolypeptide in water, while a broad peak around 400 nm remained constant after the mixing (Figure S6). The former fluorescence emission is characteristic of the tyrosine units of the polypeptides, while the latter belongs to the aggregated lysine residues.³⁹ Similar observations were made for both samples. Since the relative monomer composition of the two copolypeptides was kept constant during the measurement, this strengthening in intensity can only be caused by a change in the local environment of the lysine residues.

As suggested by previous studies,⁴⁰ when a stable localized intramolecular H bonding can be formed between tyrosine units (e.g., α -helix structure), the local environment for the tyrosine unit becomes more hydrophobic. This hydrophobicity may induce the formation of specific tertiary structures that result in colocalization of the tyrosine residues and prevent the observation of fluorescence emission of these tyrosine units.⁴¹ On the other hand, it has been reported that metal ions can effectively unfold a natural peptide molecule and promote the formation of tyrosine-based intermolecular β -sheet structure.^{42,43} Hence, one can speculate that in the presence of PBS buffer (namely, Na^+ and K^+) the tertiary structures of the copolypeptides are disturbed and subsequently unfold (reduction of the internal α -helix structure). This unfolding does result in the exposure of tyrosine units possible, while the subsequent formation of the intermolecular β -sheet induces the gelation of these polypeptides.

To obtain more insight into the morphological changes during the gelation process, a series of in situ SAXS analyses were carried out on the hydrogels. Here we used a flexible worm-like polymer chain model to fit our system as suggested by previous studies on peptide-based self-assembly systems.^{44–47} As shown in Figure 3a, for samples containing p(Lys₈₀-Tyr₂₀) mixed with 0.1 M PBS buffer, all scattering profiles can be fitted with a combination of two worm-like polymer chain form factor models with a structure factor of a mass fractal object (for a detailed fitting procedure description, see the SI). This type of form factor model has been frequently applied to fibrous hydrogel networks,^{47,48} while the fractal structure factor has also been used for branched fibrous hydrogel structures self-assembled from short polypeptides.⁴⁵ This model is supported by the observation of branched fibrous structures in the TEM analyses of polypeptide samples mixed with PBS buffer (Figure S7). Figure 3b shows the evolution of the fitted fractal dimension (D_f) over the gelation period. The whole gelation can be described as a process of a gradual branching of the network structure. Furthermore, it can be found that higher ion concentration does increase the branching rate and overall fractal dimension of the material. A similar effect of ion concentration on the nanofiber has also been observed in several short oligopeptide nanoassembly systems.⁴⁴ Additionally, the elevated PBS concentration can also increase the gelation speed.

Based on these results, we propose the following gelation model for this type of random copolypeptide. As shown in Figure 4a, due to the presence of charged ions, the α -helix structural domains of the copolypeptide formed by Lys and Tyr units is disrupted. Furthermore, as previously only observed in the micellar assemblies formed by those sequence-defined oligopeptides,²⁷ these charged salts can induce a conformational transition from the α -helix to intramolecular β -sheet. These β -sheet units can subsequently interact with β -sheet domains in adjacent copolypeptides to

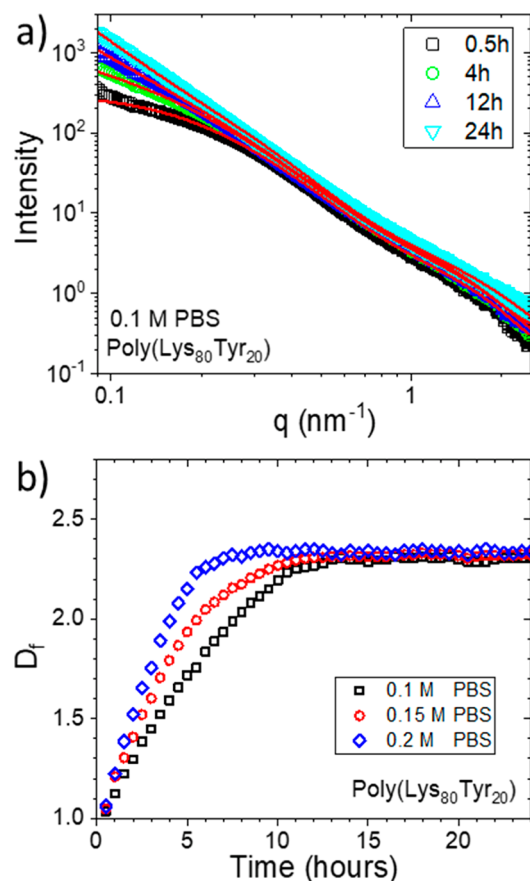


Figure 3. (a) Scattering profiles of the solution composed by 3.8 mM poly(Lys₈₀Tyr₂₀) and 0.1 M PBS buffer over a 24 h period. The model fitting is represented by the red line. (b) The fitted fractal dimension (D_f) of a 3.8 mM poly(Lys₈₀Tyr₂₀) sample mixed with different concentrations of PBS solution over a period of 24 h.

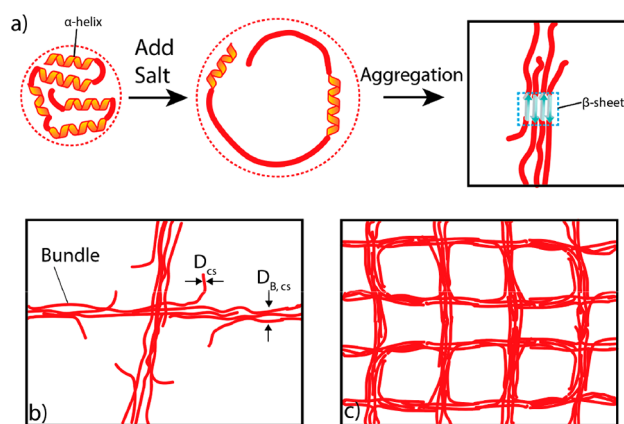


Figure 4. Illustration for the gelation process. (a) The introduction of salts disrupts the quaternary structure of the polymer and increases the hydrodynamic size of the polymer in the solution. When the concentration of the polymer is over a certain threshold, the intermolecular β -sheet starts forming. (b) A worm-like “bundle” structure was formed over time; D_{cs} is the diameter of the free polymer chain end; and $D_{cs,B}$ is the diameter of the bundle structure. (c) When a proper polypeptide and ion concentration are used, a superporous network structure can be finally produced.

form intermolecular β -sheets. As a result of the gradual increase in the number of the intermolecular connection

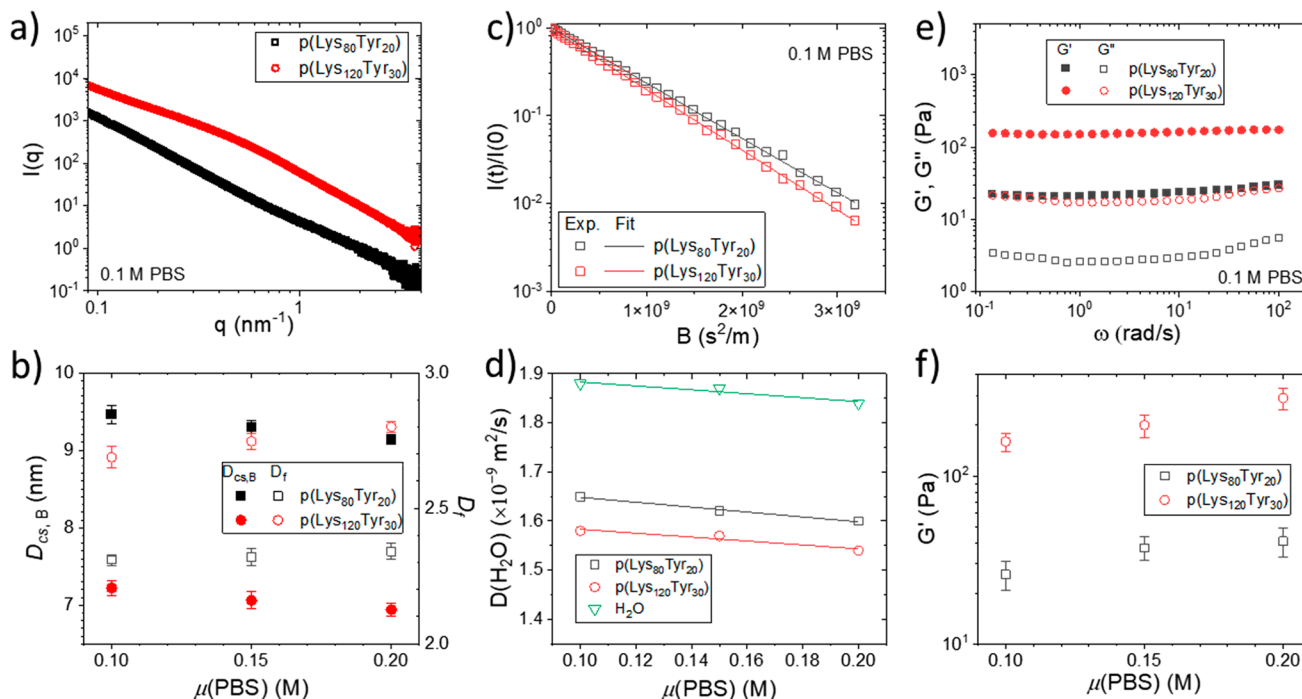


Figure 5. (a) Scattering profiles of the final gel composed by 0.1 M PBS buffer solution and 2.5 mM poly(Lys₁₂₀Tyr₃₀) and 3.8 mM poly(Lys₈₀Tyr₂₀). These scattering profiles are elevated proportionally for better display and comparison. (b) The extracted cross-section diameters (D_{cs}) and the fracture dimension (D_f) of the network structure in the final hydrogels with different compositions. All these profiles are recorded after 7 days of incubation to ensure the completion of the gelation. (c) ^1H NMR diffusometry analyses of H_2O diffusion in hydrogels made from different polypeptides mixed with 0.1 M PBS. (d) Diffusion coefficients of H_2O , $D_{\text{H}_2\text{O}}$, in solutions and different hydrogels. (e) Frequency sweep of hydrogels made from polypeptides mixed with 0.1 M PBS. (f) Plot of storage moduli (G') of the hydrogel samples versus their PBS concentrations.

between individual polypeptides, a worm-like bundle structure is produced over time (Figure 4b). Finally, when a critical concentration of copolypeptides and ions was used, a hyperbranched network was formed by the end of the gelation (Figure 4c). To the best of our knowledge, a gelation process through salt-triggered secondary structure transition has never been described for any copolypeptide. It appears to be a unique characteristic of the Lys/Tyr statistical copolypeptides, as neither the Lys/Tyr block copolypeptide nor statistical copolypeptides in which Lys was replaced by glutamic acid (Glu) or Tyr by phenylalanine (Phe) result in any gelation. Additional preliminary screening of different copolypeptide compositions revealed that a minimum of 9% Tyr (mol/mol) in the copolymer is needed to obtain hydrogels in PBS buffer (Table S1). All hydrogels were found to be stable against dilution with DI water as well as over two months at 37 °C. Addition of trifluoroacetic acid (TFA) causes the hydrogels to dissolve. Further systematic experiments will be necessary to fully understand the influence of the monomer arrangement in the copolypeptides. This includes the elucidation of preferential monomer addition, potentially resulting in segments rich in one monomer.

The fully hydrogelated samples were further investigated in a series of scattering, mechanical, and diffusion tests to understand their macroscopic physical properties in relation to their microscopic structure. Six polypeptide networks were produced from the two polypeptides p(Lys₁₂₀Tyr₃₀) and p(Lys₈₀Tyr₂₀) in combination with three different PBS concentrations. As shown in Figure 5a, the final gel state of these networks can all be well fitted with a worm-like polymer bundle model. It also can be found that the hydrogels made from the higher molecular weight copolypeptides have

generally smaller cross-section diameters ($D_{cs,B}$) for their bundle structure while obtaining a larger D_f (Figure 5b). Since these hydrogel network structures are induced by the intermolecular β -sheet formation (and aggregation of the α -helix to smaller extent), this correlation suggests that fewer copolypeptides are participating in the formation of the individual bundle structure for polypeptides with higher molecular weight. Furthermore, the ion concentration seems to have a negative impact on the cross-section's diameter of these network structures, as a smaller cross section was observed in copolypeptide networks formed at higher ion concentration. This may arise from the drop in the Debye screening length (L_κ) at higher ionic strength, which can result in a better bundle compactness. Similar observations were reported for short oligopeptide nanoassemblies.⁴⁴ Additional NMR diffusometry and mechanical tests also show that the physical properties of these hydrogels can be tuned through varying the salt concentration as well as the copolypeptide chain length. Pulsed field gradient NMR^{49,50} was used to analyze the self-diffusion coefficients of H_2O , $D_{\text{H}_2\text{O}}$, in these hydrogels. As shown in Figure 5c, due to the high water content and porous nature of these hydrogels, all the diffusion curves can be fitted with a single-component exponential decay function, and the extracted $D_{\text{H}_2\text{O}}$ values for all the hydrogels can be found with the same magnitude as the value of the free diffusing water in PBS buffer solution ($D_{\text{H}_2\text{O,PBS}}$). Further analyses on these $D_{\text{H}_2\text{O}}$ values across all the samples revealed that the ionic strength of the solution does not play a significant role in the diffusion of small solutes in these hydrogels since similar slopes can be found for the linear fittings of these $D_{\text{H}_2\text{O}}$ values upon increasing the PBS

concentration across the copolypeptide hydrogels with different chain length (Figure 5d). However, an observable change can be found for hydrogels made from copolypeptides with different chain length with slightly slower water diffusion in the hydrogel from a higher molecular weight, which may arise from the decrease in the mesh size of the networks as indicated from a larger D_f value for samples containing poly(Lys₁₂₀Tyr₃₀). While the materials are considered soft gels from the rheological analysis (Figure 5e), smaller mesh sizes also increase their mechanical properties as shown in their storage modulus analyses (Figure 5f).

In summary, we have demonstrated the first example of an ion-responsive hydrogellating statistical copolypeptide. A simple but unique gelation mechanism is proposed that relies on the ion-triggered structural rearrangement of the polypeptides to transition from intramolecular to intermolecular secondary structures. The fact that these porous hydrogels form in physiological buffer solution offers opportunities in biological application, for example, through incorporation of cells. Future work will focus on testing more variations of these statistical copolypeptides to better understand various factors' contribution to their gelation mechanism as well as their investigation as biofunctional materials.

■ ASSOCIATED CONTENT

SI Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acsmacrolett.1c00774>.

Experimental procedures and additional data (PDF)

■ AUTHOR INFORMATION

Corresponding Authors

Bing Wu – Department of Chemistry, RCSI University of Medicine and Health Sciences, Dublin 2, Ireland; Dutch-Belgian Beamline (DUBBLE), ESRF - The European Synchrotron Radiation Facility, Grenoble 38043 Cedex 9, France; orcid.org/0000-0002-2739-5124; Email: bing.wu@utoronto.ca

Andreas Heise – Department of Chemistry, RCSI University of Medicine and Health Sciences, Dublin 2, Ireland; Science Foundation Ireland (SFI) Centre for Research in Medical Devices (CURAM), RCSI, Dublin 2, Ireland; AMBER, The SFI Advanced Materials and Bioengineering Research Centre, RCSI, Dublin 2, Ireland; orcid.org/0000-0001-5916-8500; Email: andreasheise@rcsi.com

Authors

Saltuk B. Hanay – Department of Chemistry, RCSI University of Medicine and Health Sciences, Dublin 2, Ireland

Scott D. Kimmins – Instituto de Química, Pontificia Universidad Católica de Valparaíso, Placilla 2950 Valparaíso, Chile

Sally-Ann Cryan – School of Pharmacy and Biomolecular Sciences and Tissue Engineering Research Group, RCSI University of Medicine and Health Sciences, Dublin 2, Ireland; Science Foundation Ireland (SFI) Centre for Research in Medical Devices (CURAM), RCSI, Dublin 2, Ireland; AMBER, The SFI Advanced Materials and Bioengineering Research Centre, RCSI, Dublin 2, Ireland; orcid.org/0000-0002-3941-496X

Daniel Hermida Merino – Dutch-Belgian Beamline (DUBBLE), ESRF - The European Synchrotron Radiation Facility, Grenoble 38043 Cedex 9, France

Complete contact information is available at: <https://pubs.acs.org/10.1021/acsmacrolett.1c00774>

Author Contributions

The manuscript was written by B.W. and A.H. Edited by S.D.K. and S.B.H. Experiments were designed by B.W., S.B.H., D.H.M., S.-A.C., and A.H. Experiments were carried out by B.W., S.B.H., S.D.K., and D.H.M. All authors have given approval to the final version of the manuscript.

Notes

The authors declare no competing financial interest.

■ ACKNOWLEDGMENTS

The research received financial support from EU FP7 Marie Curie Actions under the NEOGEL project (Grant No. 316973) and the EU Horizon2020 Marie Curie Cofund project (Grant No. 713279). This project has received funding from the Translational Research in Nanomedical Devices (TREND) project, Science Foundation Ireland Investigators Program (Grant 13/IA/1840(T)).

■ REFERENCES

- (1) Hoare, T. R.; Kohane, D. S. Hydrogels in drug delivery: Progress and challenges. *Polymer* **2008**, *49* (8), 1993–2007.
- (2) Balakrishnan, B.; Mohanty, M.; Umashankar, P. R.; Jayakrishnan, A. Evaluation of an in situ forming hydrogel wound dressing based on oxidized alginate and gelatin. *Biomaterials* **2005**, *26* (32), 6335–6342.
- (3) Li, P.; Poon, Y. F.; Li, W. F.; Zhu, H. Y.; Yeap, S. H.; Cao, Y.; Qi, X. B.; Zhou, C. C.; Lamrani, M.; Beuerman, R. W.; Kang, E. T.; Mu, Y. G.; Li, C. M.; Chang, M. W.; Leong, S. S. J.; Chan-Park, M. B. A polycationic antimicrobial and biocompatible hydrogel with microbe membrane suctioning ability. *Nat. Mater.* **2011**, *10* (2), 149–156.
- (4) Slaughter, B. V.; Khurshid, S. S.; Fisher, O. Z.; Khademhosseini, A.; Peppas, N. A. Hydrogels in Regenerative Medicine. *Adv. Mater.* **2009**, *21* (32–33), 3307–3329.
- (5) Capozzi, F.; Laghi, L.; Belton, P. S. *Magnetic resonance in food science: defining food by magnetic resonance*; p 250.
- (6) Tan, W. H.; Takeuchi, S. Monodisperse alginate hydrogel microbeads for cell encapsulation. *Adv. Mater.* **2007**, *19* (18), 2696–2701.
- (7) Helary, C.; Bataille, I.; Abed, A.; Illoul, C.; Anglo, A.; Louedec, L.; Letourneur, D.; Meddahi-Pelle, A.; Giraud-Guille, M. M. Concentrated collagen hydrogels as dermal substitutes. *Biomaterials* **2010**, *31* (3), 481–490.
- (8) Yan, H.; Saiani, A.; Gough, J. E.; Miller, A. F. Thermoreversible protein hydrogel as cell scaffold. *Biomacromolecules* **2006**, *7* (10), 2776–2782.
- (9) Machado, C. A.; Smith, I. R.; Savin, D. A. Self-Assembly of Oligo- and Polypeptide-Based Amphiphiles: Recent Advances and Future Possibilities. *Macromolecules* **2019**, *52* (5), 1899–1911.
- (10) Sun, Y.; Kaplan, J. A.; Shieh, A.; Sun, H. L.; Croce, C. M.; Grinstaff, M. W.; Parquette, J. R. Self-assembly of a 5-fluorouracil-dipeptide hydrogel. *Chem. Commun.* **2016**, *52* (30), 5254–5257.
- (11) Rodon Fores, J.; Criado-Gonzalez, M.; Chaumont, A.; Carvalho, A.; Blanck, C.; Schmutz, M.; Boulmedais, F.; Schaaf, P.; Jierry, L. Autonomous Growth of a Spatially Localized Supramolecular Hydrogel with Autocatalytic Ability. *Angew. Chem. Int. Edit* **2020**, *59* (34), 14558–14563.
- (12) Gungormus, M.; Branco, M.; Fong, H.; Schneider, J. P.; Tamerler, C.; Sarikaya, M. Self assembled bi-functional peptide hydrogels with biomineralization-directing peptides. *Biomaterials* **2010**, *31* (28), 7266–7274.

- (13) Kretsinger, J. K.; Haines, L. A.; Ozbas, B.; Pochan, D. J.; Schneider, J. P. Cytocompatibility of self-assembled β -hairpin peptide hydrogel surfaces. *Biomaterials* **2005**, *26* (25), 5177–5186.
- (14) Kopecek, J.; Yang, J. Y. Peptide-directed self-assembly of hydrogels. *Acta Biomater* **2009**, *5* (3), 805–816.
- (15) Kopecek, J.; Yang, J. Y. Smart Self-Assembled Hybrid Hydrogel Biomaterials. *Angew. Chem. Int. Edit* **2012**, *51* (30), 7396–7417.
- (16) Glassman, M. J.; Chan, J.; Olsen, B. D. Reinforcement of Shear Thinning Protein Hydrogels by Responsive Block Copolymer Self-Assembly. *Adv. Funct. Mater.* **2013**, *23* (9), 1182–1193.
- (17) Zhou, X. F.; Li, Z. B. Advances and Biomedical Applications of Polypeptide Hydrogels Derived from alpha-Amino Acid N-Carboxyanhydride (NCA) Polymerizations. *Adv. Healthc. Mater.* **2018**, *7* (15), 1800020.
- (18) Keerthi, A.; Hou, I. C. Y.; Marszalek, T.; Pisula, W.; Baumgarten, M.; Narita, A. Hexa-peri-hexabenzocoronene with Different Acceptor Units for Tuning Optoelectronic Properties. *Chem-Asian J.* **2016**, *11* (19), 2710–2714.
- (19) Hadjichristidis, N.; Iatrou, H.; Pitsikalis, M.; Sakellariou, G. Synthesis of Well-Defined Polypeptide-Based Materials via the Ring-Opening Polymerization of alpha-Amino Acid N-Carboxyanhydrides. *Chem. Rev.* **2009**, *109* (11), 5528–5578.
- (20) Song, Z. Y.; Han, Z. Y.; Lv, S. X.; Chen, C. Y.; Chen, L.; Yin, L. C.; Cheng, J. J. Synthetic polypeptides: from polymer design to supramolecular assembly and biomedical application. *Chem. Soc. Rev.* **2017**, *46* (21), 6570–6599.
- (21) Bonduelle, C. Secondary structures of synthetic polypeptide polymers. *Polym. Chem-Uk* **2018**, *9* (13), 1517–1529.
- (22) Sun, Y. T.; Wollenberg, A. L.; O'Shea, T. M.; Cui, Y. X.; Zhou, Z. H.; Sofroniew, M. V.; Deming, T. J. Conformation-Directed Formation of Self-Healing Diblock Copolypeptide Hydrogels via Polyion Complexation. *J. Am. Chem. Soc.* **2017**, *139* (42), 15114–15121.
- (23) Murphy, R.; Walsh, D. P.; Hamilton, C. A.; Cryan, S. A.; Panhuis, M. I. H.; Heise, A. Degradable 3D-Printed Hydrogels Based on Star-Shaped Copolypeptides. *Biomacromolecules* **2018**, *19* (7), 2691–2699.
- (24) Nowak, A. P.; Breedveld, V.; Pakstis, L.; Ozbas, B.; Pine, D. J.; Pochan, D.; Deming, T. J. Rapidly recovering hydrogel scaffolds from self-assembling diblock copolypeptide amphiphiles. *Nature* **2002**, *417* (6887), 424–428.
- (25) Murphy, R.; Kordbacheh, S.; Skoulas, D.; Ng, S.; Suthiwanich, K.; Kasko, A. M.; Cryan, S. A.; Fitzgerald-Hughes, D.; Khademhosseini, A.; Sheikhi, A.; Heise, A. Three-dimensionally printable shear-thinning triblock copolypeptide hydrogels with antimicrobial potency. *Biomaterials Science* **2021**, *9*, 5144.
- (26) Baldwin, R. L. How Hofmeister ion interactions affect protein stability. *Biophys. J.* **1996**, *71* (4), 2056–2063.
- (27) Otsuka, T.; Maeda, T.; Hotta, A. Effects of Salt Concentrations of the Aqueous Peptide-Amphiphile Solutions on the Sol-Gel Transitions, the Gelation Speed, and the Gel Characteristics. *J. Phys. Chem. B* **2014**, *118* (39), 11537–11545.
- (28) Nowak, A. P.; Breedveld, V.; Pine, D. J.; Deming, T. J. Unusual salt stability in highly charged diblock co-polypeptide hydrogels. *J. Am. Chem. Soc.* **2003**, *125* (50), 15666–15670.
- (29) Hanay, S. B.; Ritzen, B.; Brougham, D.; Dias, A. A.; Heise, A. Exploring Tyrosine-Triazolinedione (TAD) Reactions for the Selective Conjugation and Cross-Linking of N-Carboxyanhydride (NCA) Derived Synthetic Copolypeptides. *Macromol. Biosci* **2017**, *17* (7), 1700016.
- (30) Krimm, S.; Abe, Y. Intermolecular Interaction Effects in the Amide I Vibrations of β Polypeptides. *Proc. Natl. Acad. Sci. U. S. A.* **1972**, *69* (10), 2788–2792.
- (31) Loksztajn, A.; Dzwolak, W.; Krysinski, P. Tyrosine side chains as an electrochemical probe of stacked beta-sheet protein conformations. *Bioelectrochemistry* **2008**, *72* (1), 34–40.
- (32) Chiou, J. S.; Tataru, T.; Sawamura, S.; Kaminoh, Y.; Kamaya, H.; Shibata, A.; Ueda, I. The Alpha-Helix to Beta-Sheet Transition in Poly(L-Lysine) - Effects of Anesthetics and High-Pressure. *Biochim. Biophys. Acta* **1992**, *1119* (2), 211–217.
- (33) Chirgadze, Y. N.; Nevskaya, N. A. Infrared spectra and resonance interaction of amide-I vibration of the antiparallel-chain pleated sheet. *Biopolymers* **1976**, *15* (4), 607–625.
- (34) Nevskaya, N. A.; Chirgadze, Y. N. Infrared spectra and resonance interactions of amide-I and II vibrations of α -helix. *Biopolymers* **1976**, *15* (4), 637–648.
- (35) Andrew, C. D.; Bhattacharjee, S.; Kokkoni, N.; Hirst, J. D.; Jones, G. R.; Doig, A. J. Stabilizing interactions between aromatic and basic side chains in alpha-helical peptides and proteins. Tyrosine effects on helix circular dichroism. *J. Am. Chem. Soc.* **2002**, *124* (43), 12706–12714.
- (36) Barth, A. Infrared spectroscopy of proteins. *Bba-Bioenergetics* **2007**, *1767* (9), 1073–1101.
- (37) Micsonai, A.; Wien, F.; Kernya, L.; Lee, Y. H.; Goto, Y.; Refregiers, M.; Kardos, J. Accurate secondary structure prediction and fold recognition for circular dichroism spectroscopy. *P Natl. Acad. Sci. USA* **2015**, *112* (24), E3095–E3103.
- (38) Greenfield, N. J. Using circular dichroism spectra to estimate protein secondary structure. *Nat. Protoc* **2006**, *1* (6), 2876–2890.
- (39) Homchadhuri, L.; Swaminathan, R. Novel absorption and fluorescence characteristics of L-lysine. *Chem. Lett.* **2001**, *30* (8), 844–845.
- (40) Partlow, B. P.; Bagheri, M.; Harden, J. L.; Kaplan, D. L. Tyrosine Templating in the Self-Assembly and Crystallization of Silk Fibroin. *Biomacromolecules* **2016**, *17* (11), 3570–3579.
- (41) Martel, A.; Burghammer, M.; Davies, R. J.; Di Cola, E.; Vendrely, C.; Riekel, C. Silk Fiber Assembly Studied by Synchrotron Radiation SAXS/WAXS and Raman Spectroscopy. *J. Am. Chem. Soc.* **2008**, *130* (50), 17070–17074.
- (42) Lee, M.; Kim, J. I.; Na, S.; Eom, K. Metal ions affect the formation and stability of amyloid aggregates at multiple length scales. *Phys. Chem. Chem. Phys.* **2018**, *20* (13), 8951–8961.
- (43) Ozbas, B.; Kretsinger, J.; Rajagopal, K.; Schneider, J. P.; Pochan, D. J. Salt-triggered peptide folding and consequent self-assembly into hydrogels with tunable modulus. *Macromolecules* **2004**, *37* (19), 7331–7337.
- (44) Feng, Y.; Taraban, M.; Yu, Y. B. The effect of ionic strength on the mechanical, structural and transport properties of peptide hydrogels. *Soft Matter* **2012**, *8* (46), 11723–11731.
- (45) Cheng, G.; Castelletto, V.; Moulton, C. M.; Newby, G. E.; Hamley, I. W. Hydrogelation and Self-Assembly of Fmoc-Tripeptides: Unexpected Influence of Sequence on Self-Assembled Fibril Structure, and Hydrogel Modulus and Anisotropy. *Langmuir* **2010**, *26* (7), 4990–4998.
- (46) Szymusiak, M.; Kalkowski, J.; Luo, H. Y.; Donovan, A. J.; Zhang, P.; Liu, C.; Shang, W. F.; Irving, T.; Herrera-Alonso, M.; Liu, Y. Core-Shell Structure and Aggregation Number of Micelles Composed of Amphiphilic Block Copolymers and Amphiphilic Heterografted Polymer Brushes Determined by Small-Angle X-ray Scattering. *ACS Macro Lett.* **2017**, *6* (9), 1005–1012.
- (47) Schoenmakers, D. C.; Rowan, A. E.; Kouwer, P. H. J. Crosslinking of fibrous hydrogels. *Nat. Commun.* **2018**, DOI: 10.1038/s41467-018-04508-x.
- (48) Jaspers, M.; Pape, A. C. H.; Voets, I. K.; Rowan, A. E.; Portale, G.; Kouwer, P. H. J. Bundle Formation in Biomimetic Hydrogels. *Biomacromolecules* **2016**, *17* (8), 2642–2649.
- (49) Wu, B.; Wiseman, M. E.; Seitz, M. E.; Tomić, K.; Heise, A.; Brougham, D. F.; Litvinov, V. M. Impact of morphology on O₂ permeability in silicone hydrogel membranes: new insights into domain percolation from experiments and simulations. *J. Membr. Sci.* **2021**, *621*, 118970.
- (50) Wu, B.; Chasse, W.; Zick, K.; Mantle, M. D.; Heise, A.; Brougham, D. F.; Litvinov, V. M. The effect of hydrogen bonding on diffusion and permeability in UV-cured Polyacrylate-based networks for controlled release. *J. Controlled Release* **2020**, *327*, 150–160.