

High-Efficiency Photocatalytic Degradation of Tannic Acid Using TiO₂ Heterojunction Catalysts

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photocatalytic activity toward the degradation of tannic acid, which is a typical organic in nuclear wastewater. At an optimal loading of 16% 1T-MoS₂, the 1T-MoS₂@TiO₂ shows the highest degradation capacity of 98%, which is 3.2 times higher than that of pure TiO₂. The degradation rate of 16% 1T-MoS₂@TiO₂ is much



higher than that of 13% 2H-MoS₂@TiO₂. The enhanced photocatalytic activity might be attributed to the improved charge transfer according to the mechanism investigation, supported by the X-ray photoelectron spectroscopy (XPS) and electrochemical impedance spectroscopy (EIS) analyses. This work provides new opportunities for constructing highly efficient catalysts for nuclear waste disposal.

1. INTRODUCTION

Nuclear energy is a nonrenewable alternative energy source with high energy density, environmental protection, and a small footprint.¹⁻³ However, radioactive organic wastewater is produced in nuclear fuel production, operation of nuclear power plants, and decommissioning of nuclear facilities, which causes long-term harm to the ecological environment. Organic nuclear waste liquid mainly consists of the radioactive element uranium, organic matter tannic acid (TA), and tributyl phosphate. It has the characteristics of complex composition, intense radioactivity, poor safety and stability, and a high amount of heat generation.⁴ The presence of organic matter not only easily causes the blockage and regeneration of the ion exchange resin channel but also has potential safety hazards. Therefore, it is required to pretreat the organic matter before solidifying the radioactive waste liquid containing organic matter.

Traditional methods for removing organic matter include high-temperature oxidation, wet chemical reaction, physical adsorption, photocatalysis, etc. Among them, the semiconductor photocatalysis technology displays many advantages, for example, strong oxidizing ability, effective absorption of light, environment friendliness, mild reaction conditions, and stable chemical properties without secondary pollutants. It has become a research focus in energy and the environment.^{5,6} The photocatalytic efficiency depends on the light absorption intensity and carrier transfer efficiency of photocatalysts. To date, excellent materials such as graphene,⁷⁻¹² molybdenum

disulfide,¹³⁻¹⁹ and titanium dioxide (TiO₂) have been developed for the photocatalytic processes.²⁰⁻²⁹

Anatase TiO₂ material used as a photocatalyst demonstrates high efficiency and excellent chemical stability. The main characteristics of anatase TiO₂ are its nontoxicity, low cost, and strong corrosion resistance. Therefore, it is widely used in the photocatalytic field.^{20,30-35} Solar energy, generated through photoelectricity and water photocatalysis over photocatalysts (e.g. TiO₂), can alleviate the environmental and energy crisis.³⁶⁻³⁸ TiO₂ wide band gap, high electron-hole recombination rate, and few active catalytic sites are the main obstacles hindering its wide application.^{39,40} It is therefore interesting to develop new photocatalytic materials with good photocatalytic activities or tailor the structure and particle size. It is found that nanoflower TiO₂ has better degradation ability toward methylene blue than pure TiO₂⁴¹

Developing heterojunction materials is another effective method to accelerate charge separation, activate the catalyst surface, and enhance photocatalytic active sites.⁴² New composite materials synthesized with metals or semiconductors contain the Schottky barrier heterojunctions. Therefore,

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Figure 1. Schematic diagram of the procedure of preparing a composite material (1T/2H) MoS2@TiO2.

they are often used for photocatalytic hydrogen evolution.⁴³ However, some metals are high-cost, and hence it is necessary to seek alternatives. The CdSe/graphene/TiO₂ composite prepared shows high photocatalytic activity for methylene blue under visible light.⁷ type-II heterojunctions have been proven to be promising excellent photocatalytic materials.^{44,45} The large exposed edge of MoS₂ originates from the rich catalytic active sites on the sulfur edge of the crystal layer, and the narrow band gap has good complementary characteristics with ${\rm TiO_2}^{26,46-49}$ Yuan et al. loaded ${\rm MoS_2}$ on ${\rm TiO_2}$ to construct a type-II heterostructure material for efficient solar hydrogen production.⁵⁰ Tang et al. synthesized a series of anatase $TiO_2@$ MIL-101(Cr) by the solvothermal method for photocatalytic degradation of bisphenol A (BPA) and 59% TiO2@MIL-101(Cr) could achieve 99.4% degradation rate within 240 min under ultraviolet light irradiation.⁵¹ Therefore, MoS₂@TiO₂ heterojunction materials have the advantages of large specific surface area, excellent electron transfer ability, and enhanced active sites in theory. However, there is still less information for applying these kinds of composite materials in the degradation of pollutants in nuclear wastewater.

Tannic acid is one of the important components of organic acids in nuclear waste liquid, and it is prone to the complex reaction of metals. If tannic acid is discharged into water, it will cause serious pollution of the water body, reduce the content of organic matter in the water, and be toxic to organisms in the water. Therefore, green and safe degradation of tannic acid has important practical significance. A 98% tannic acid was successfully degraded by titanium oxide-coated magnetic activated carbon under ultraviolet light for 150 min with improved stability.⁵² Qiu et al.⁵³ prepared nano-TiO₂-coated tiles for photocatalytic degradation experiments. The photocatalytic device was selected as a UV lamp; the results showed that the degradation rate after 6 h reached 76% and Fe–N codoped SiO₂@TiO₂ could degrade 99% of tannic acid in 80 min.⁵⁴

In this work, two new composites, $1T-MoS_2@TiO_2$ with a rich $1T-MoS_2$ heterojunction and $2H-MoS_2@TiO_2$ with type-II heterojunction, were synthesized.⁵⁵ The photocatalytic performance of using these two types of materials for the photocatalytic degradation of tannic acid in nuclear wastewater have been investigated. The effect of the compositions of MoS_2 on the photocatalytic activity has been studied and the optimized MoS_2 content has been evaluated.

2. EXPERIMENTAL SECTION

2.1. Materials. Raw chemicals, molybdenum trioxide $(MoO_3, 99.9\%)$ Shanghai Aladdin Biochemical Technology Co., Ltd.), urea $(CH_4N_2O, 99\%)$, Shanghai Aladdin Biochemical Technology Co., Ltd.), thioacetamide $(CH_3CSNH_2, 99\%)$, Chengdu Cologne Chemical Co., Ltd.), and nano titanium dioxide (P25, 99.9\%) Shanghai King Chemical Co., Ltd.), were used without any further purification.

2.2. Synthesis of 1T-MoS₂, 2H-MoS₂, and MoS₂@TiO₂. Two-dimensional (2D) composite materials of 1T-MoS₂@ TiO₂ and 2H-MoS₂@TiO₂ were synthesized using the hydrothermal method. For the preparation of 1T-MoS₂@ TiO_2 , 96 mg of MoO₃, 112 mg of thioacetamide, and 0.96 g of urea were dissolved in 58.8 mL of deionized water and then 5.3 g of TiO₂ was added. After stirring for 2 h, the solution was transferred to a Teflon-lined autoclave reactor, which was kept in an oven at 200 °C for 12 h. Thereafter, it was cooled to room temperature and then the autoclave reactor was taken out. Finally, the as-prepared 1T-MoS₂@TiO₂ was collected and washed with deionized water and ethanol and then stored in deionized water before use. In theory, 2% 1T-MoS₂@TiO₂ was obtained based on the above statement. Composites with different compositions of MoS₂ were prepared by changing the MoO_3 /urea ratio using the same procedure listed above.

2H-MoS₂@TiO₂ was synthesized using a similar method to the preparation of 1T-MoS2@TiO₂, and the only difference is that the reaction temperature was changed from 200 to 240 °C. The flow chart of the specific process is shown in Figure 1.

2.3. Photocatalytic Degradation of Tannic Acid. The photocatalytic degradation of tannic acid was carried out in a reactor (model BL-GHX-V, Shanghai Bilang Instrument Co., Ltd.). Because the light source is a 1000 W mercury lamp, which has a high heat output, the equipment is equipped with a cold trap, and water is continuously supplied to protect the equipment during the reaction. The experiment was divided into the dark response stage and the light reaction stage. The dark reaction stage was to ensure that the material and the tannic acid in the solution reach a physical adsorptiondesorption equilibrium within 120 min. After the material reached the physical adsorption-desorption equilibrium, the photocatalysis reaction was carried out (m/v = 1.8 mg/mL, $C_{\text{TA}} = 40 \text{ ppm}, T = 300 \text{ K}, \text{UV intensity} = 1000 \text{ W}$). During the photocatalytic reaction (dark reaction stage and light reaction stage), the tannic acid solution was sampled and



Article

Figure 2. Microscopies of the as-synthesized samples. (a) SEM image of 16% $1T-MoS_2@TiO_2$. (b) SEM image of 13% $2H-MoS_2@TiO_2$. (c) TEM image of 16% $1T-MoS_2@TiO_2$. (d) High-resolution TEM of 16% $1T-MoS_2@TiO_2$, which shows the lattice spacing of TiO₂ and $1T-MoS_2$. (e) High-resolution TEM of 13% $2H-MoS_2@TiO_2$, which shows the lattice spacing of TiO₂ and $2H-MoS_2$. (f) Element mapping of Mo, S, Ti, and O of $1T-MoS_2@TiO_2$.



Figure 3. (a) XRD patterns of anatase TiO_2 , 2H-MoS₂, 1T-MoS₂, and composite materials 1T-MoS₂@ TiO_2 and 2H-MoS₂@ TiO_2 . (b) Raman shift of 1T-MoS₂@ TiO_2 and 2H-MoS₂@ TiO_2 .

filtered every 20 min, and then the absorbance of tannic acid in the solution was tested with a UV2600 ultraviolet-visible spectrophotometer. Through the formula $C/C_0 = A/A_0$, the tannin concentration value was obtained.

2.4. Active Species Capture Experiment. The experiment was divided into four groups, and the amount of compound sample and tannic acid remained unchanged (m/v = 1.8, 5 mg; 40 mL, $C_{TA} = 40$ ppm). Isopropyl alcohol (IPA) (2 mL), ethylenediaminetetraacetic acid (EDTA) (2 mg), and Na₂SO₄ (2 mg) were added to the first three groups and the last group was a blank control. The experiment was divided into dark reaction and light reaction for 2 h each and the light source was a mercury lamp. The samples are taken for testing at the end of the dark reaction and light reaction.

2.5. Characterization. The morphology of $MoS_2@TiO_2$ was investigated by scanning electron microscopy (SEM, Carl Zeiss, Ultra 55) and transmission electron microscopy (TEM, Carl Zeiss, Libra 200). The elemental mapping was recorded on a TEM equipped with an energy dispersive spectrometer. Powder X-ray diffraction (XRD) patterns were performed on a Cu K α radiation PAN analytical X-ray diffractometer. A laser Raman spectrometer (inVia) was used to analyze the samples by Raman spectroscopy. X-ray photoelectron spectroscopy (XPS, Thermo Fisher Scientific) was used to analyze the elemental composition and valence of the materials. The light

absorption characteristics of the samples were carried out on a solid ultraviolet absorption spectrometer (UV–vis, Shimadzu Enterprise Management (China) Co., Ltd.) with the equipment model Solidspec-3700. UV–vis absorbance spectra were detected on a UV2600 UV–vis spectrophotometer (Unico Instruments Co., Ltd.). The Mott–Schottky curve (Mott– Schottky) and AC impedance (EIS) of the cell with the asprepared sample as an electrolyte are examined on an electrochemical analyzer of CHI instruments (Shanghai Chenhua Instrument Company, CHI660C) and the details are shown in the Supporting information.

3. RESULTS AND DISCUSSION

3.1. Composite Materials Characterization Results. Figure 2a,b shows the scanning electron microscopy images of $1T-MoS_2@TiO_2$ and $2H-MoS_2@TiO_2$, respectively. It can be seen that $(1T/2H) MoS_2$ and TiO_2 grow together to present a sheet-like package structure, connecting into one piece and piling up. It can be seen from Figure S1a,b in the Supporting Information that $1T-MoS_2$ and $2H-MoS_2$ have similar particle sizes and morphology. Figure 2c,d shows the transmission microscope images of $1T-MoS_2@TiO_2$. It can be seen that the gray part is TiO_2 nanoparticles, and the black banded part is $1T-MoS_2$ distributed uniformly on the TiO_2 surface. The images in Figure S2a,b in the Supporting Material further



Figure 4. Optical properties of the as-prepared materials. (a) UV-vis diffuse reflectance spectra. (b) Plots of $(ah\nu)^{1/2}$ as a function of E_g photon energy.



Figure 5. High-resolution XPS spectra of the as-prepared samples. (a) Mo 3d of $1T-MoS_2$ and $1T-MoS_2@TiO_2$. (b) S 2p of $1T-MoS_2$ and $1T-MoS_2@TiO_2$. (c) Ti 2p of TiO_2 and $1T-MoS_2@TiO_2$. (d) Mo 3d of $2H-MoS_2$ and $2H-MoS_2@TiO_2$. (e) S 2p of $2H-MoS_2$ and $2H-MoS_2@TiO_2$. (f) O 1s of TiO_2 and $1T-MoS_2@TiO_2$.

prove the uniformity of MoS₂ distributed on the TiO₂ surface. The inset in the upper right corner shows that $1T-MoS_2$ has a lattice spacing of 0.89 nm corresponding to its (002) plane, and TiO₂ has a lattice spacing of 0.30 nm corresponding to its (101) plane. Figure 2e shows a high-resolution TEM image of 2H-MoS₂@TiO₂. It can be seen that the black band-like 2H- MoS_2 is also supported on the surface of TiO_2 . The upper right part of Figure 2e shows that the lattice spacing of 2H-MoS₂ is 0.61 nm corresponding to its (002) plane. The lattice spacing of TiO₂ is 0.30 nm corresponding to its (101) plane. Figure 2f shows the energy-dispersive spectrometry (EDS) energy spectrum analysis of the four elements Mo, S, Ti, and O in the composite material, which indicates that MoS₂ is distributed evenly on the surface of TiO2. The EDS results show that Mo and S are distributed in the same area. SEM and TEM analyses have proved that two components (MoS₂ and TiO_2) form a heterojunction material.

Figure 3a shows the XRD patterns of anatase TiO₂, 2H-MoS₂, 1T-MoS₂, and composite materials 1T-MoS₂@TiO₂ and

2H-MoS₂@TiO₂. The diffraction peaks of TiO₂ at 25.27, 37.77, and 47.99° correspond to (101), (004), (200) of anatase TiO₂. The peaks of 53.88, 55.05, 62.83, 68.72, 70.26, and 75.04° correspond to (211), (105), (204), (116), (220), (215) crystal planes. These results indicate that the anatase phase of TiO_2 is obtained in the current catalysts. The (002) crystal face of 2H-MoS₂ is detected at 14.37° in 2H-MoS₂@ TiO_{2i} and the (002) crystal face unique to $1T-MoS_2$ is detected at 9.12° in 1T-MoS₂@TiO₂. However, the remaining crystal planes of 2H-MoS₂ and 1T-MoS₂ are not detected, which might be due to the small proportions of these two materials in the composite samples. The XRD results indicate that the TiO_2 is distributed evenly in the composite materials. Due to the difference in the structure, their Raman characteristics are also different. Figure 3b shows the Raman spectra of the two composite materials. It can be seen that 1T- $MoS_2@TiO_2$ has a stronger Raman spectrum at 146 cm⁻¹ than 2H-MoS₂@TiO₂, which is attributed to 1T-Mo-Mo stretching vibration in MoS₂. The Raman shift (326, 512 cm⁻¹) is related

Article



Figure 6. Photocatalytic degradation of tannic acid. (a) Photocatalytic degradation tests in the presence of the composite material loaded with different amounts of 1T-MoS₂. (b) Photocatalytic degradation stage of the composite material containing 1T-MoS₂. (c) Stability of 16% 1T-MoS₂@TiO₂. (d) Photocatalytic degradation of tannic acid in the presence of the composite material loaded with different amounts of 2H-MoS₂. (e) Photocatalytic degradation stage of the composite material containing 2H-MoS₂. (f) Stability of 13% 2H-MoS₂@TiO₂. (g) UV-vis spectrum of 16% 1T-MoS₂@TiO₂ degradation of tannic acid. (h) UV-vis spectrum of 13% 2H-MoS₂@TiO₂ degradation of tannic acid. (i) Comparison of the degradation rate with different catalysts.

to the phonon mode in 1T-MoS₂. 2H-MoS₂ exhibits typical Raman shifts of ~378 and 404 cm⁻¹ for E_{2g}^{-1} and A_{1g} , respectively, which are substantially different from those of 1T-MoS₂.

Figure 4 shows the UV-vis diffuse reflectance spectra of TiO₂, 2H-MoS₂, 1T-MoS₂, and their composite materials 1T-MoS₂@TiO₂ and 1T-MoS₂@TiO₂. The adsorption band below 400 nm is the intrinsic band gap of TiO₂, which is in the ultraviolet light range. 2H-MoS₂ and 1T-MoS₂ present total absorption in the tested wavelength, with an apparent absorption in the visible light range. The spectra of 1T/2H- $MoS_2@TiO_2$ display combined features of MoS_2 and TiO_2 having the absorption wavelength range of 200-400 nm. Figure 4b shows that the curve of $(ah\nu)^{1/2}$ changes with the photon energy. The tangent intercept is the band gap value of the material, where the band gap value of TiO₂ and 2H-MoS₂ is 3.2 and 1.9 eV, respectively. Because MoS₂ has a narrow band gap value, it can change its band gap value when combined with TiO₂. The absorption wavelength range of the two composite materials, $2H-MoS_2@TiO_2$ and $1T-MoS_2@$

 TiO_2 , is between 200 and 400 nm, which is in the ultraviolet wavelength range.

XPS is used to analyze the chemical composition of the product (Figure S3 in the Supporting Material). The XPS spectra show the existence of Mo, S, Ti, and O, and the consistency of the valence state with pure MoS₂ and TiO₂. The Mo 3d spectrum of the 1T-MoS₂@TiO₂ composite material and the spectrum of pure 1T-MoS₂ are shown in Figure 5a, where the characteristic peaks of pure 1T-MoS₂ are at around 231.88 and 228.58 eV generated by Mo $3d_{3/2}$ and Mo $3d_{5/2}$ orbitals corresponding to Mo⁴⁺. The Mo 3d_{3/2} and Mo 3d_{5/2} orbitals of the 1T-MoS2@TiO2 material have characteristic peak positions at around 231.18 and 227.98 eV, which means that the Mo⁴⁺ of the composite material shifts to the direction of low binding energy. Similarly, the weaker peaks produced by pure 1T-MoS₂ at 229.38 and 233.28 eV also move to the lower binding energy direction, which decreased to 229.05 and 232.72 eV, respectively. Therefore, it is proved that composite materials are more conducive to the transfer of electrons. Figure 5b shows the location of the characteristic peaks of the

Article

S $2p_{1/2}$ and S $2p_{3/2}$ orbitals of pure 1T-MoS₂ at around 162.68 and 161.68 eV. The locations of the characteristic peaks of the S $2p_{1/2}$ and S $2p_{3/2}$ orbitals in the 1T-MoS₂@TiO₂ material are at around 161.98 and 160.98 eV, demonstrating that S^{2-} in the 1T-MoS₂@TiO₂ material also shifts to the low binding energy. Figure 5c,f shows the comparison spectra of Ti 2p and O 1s of 1T-MoS₂@TiO₂ and pure TiO₂, respectively. The characteristic peaks of pure TiO₂ are at around 464.38 and 458.68 eV, which correspond to Ti^{4+} by Ti $2p_{1/2}$ and Ti $2p_{3/2}$ orbitals, respectively. The position of the characteristic peaks generated by the Ti $2p_{1/2}$ and Ti $2p_{3/2}$ orbitals of 1T-MoS₂@TiO₂ is about 464.68 and 458.88 eV, the position of the characteristic peaks generated by the O 1s orbital of pure TiO₂ is about 529.98 eV, and the O 1s orbital of 1T-MoS₂@TiO₂ is at around 530.08 eV. These results show that the characteristic peaks of Ti^{4+} and O^{2-} in 1T-MoS₂@TiO₂ are shifted to the high binding energy, while the characteristic peaks of Mo⁴⁺ and S^{2-} are shifted to the low binding energy. This shifting demonstrates electron transfer from TiO₂ to MoS₂ and improved the successful recombination of MoS₂ and TiO₂. In Figure 5d, it can be seen that the characteristic peaks of the Mo $3d_{3/2}$ and Mo $3d_{5/2}$ orbitals of pure 2H-MoS₂ are at around 232.18 and 228.78 eV, and the Mo $3d_{3/2}$ and Mo $3d_{5/2}$ characteristic peaks of the 2H-MoS2@TiO2 material are located near 231.28 and 227.98 eV, so they are also shifted to the direction of low binding energy. The weaker peaks of pure 2H-MoS₂ at 229.48 and 232.98 eV also shift to the low binding energy direction, shifting to 229.35 and 232.48 eV, respectively. The shift peaks show that composite materials can accelerate electron transfer. It can be found in Figure 5e that the characteristic peaks of the S $2p_{1/2}$ and S $2p_{3/2}$ orbitals of pure 2H-MoS₂ are around 162.88 and 161.78 eV, and the characteristic peaks of the S $2p_{1/2}$ and S $2p_{3/2}$ orbitals of the 2H-MoS₂@TiO₂ material are located at the position of around 161.98 and 160.98 eV. According to the spectrum analysis, it can be seen that the comparison spectrum of Ti 2p and O 1s of 2H-MoS₂@TiO₂ and pure TiO₂ also displays the shift of the characteristic peaks of Ti⁴⁺ and O²⁻ of the 2H-MoS₂@TiO₂ to the direction of high binding energy (Figure S4 in the Supporting Material).

3.2. Photocatalytic Degradation of Tannic Acid. The photocatalytic activities of $1T-MoS_2@TiO_2$ and $2H-MoS_2@$ TiO₂ were evaluated by testing the degradation rate of tannic acid (TA) in water with a concentration of 40 mg/L under ultraviolet light (a mercury lamp with a light source of 1000 W). A commercial TiO₂ was used for comparison.

Figure 6a shows the photocatalytic degradation capacity of tannic acid over blank sample, TiO₂, 1T-MoS₂, and composite materials of x% 1T-MoS₂@TiO₂ (x = 2, 6, 10, 16, 20). The data in Figure 6a shows that the composite materials are far more active than the blank, TiO₂, and 1T-MoS₂ materials with an order of x% 1T-MoS₂@TiO₂ > 1T-MoS₂ > TiO₂ > blank.

The adsorption capability of pure TiO_2 is stronger than that of the composite $1\text{T-MoS}_2(@\text{TiO}_2 \text{ and } 2\text{H-MoS}_2(@\text{TiO}_2)$ materials. We can see significant desorption in the beginning of the photoreaction stage for the pure TiO_2 . That is, the plot of the photodegradation of tannic acid in the first 40 min (Figure 6a,b,d,e) of the reaction has a slight increasing trend, which is due to the remarkable desorption of tannic acid adsorbed by the material in the dark reaction stage. As the photocatalytic reaction gradually increases and then exceeds the analytical amount, so the graph shows a downward trend. Figure 6b shows the degradation plot of tannic acid regarding the typical amount of $1T-MoS_2$. Significant enhancement in the degradation of 63% is achieved with 2% $1T-MoS_2$ in 1 h as compared to that of 31% using the TiO₂ sample without MoS₂. A further degradation improvement is found when $1T-MoS_2$ increases to 6 and 10%. The optimal amount of $1T-MoS_2$ is found to be 16% with 98% degradation. The degradation efficiency is not improved by increasing the $1T-MoS_2$ content to 20% or more (Figure S5).

We guess that there may be two reasons for this phenomenon. First, MoS₂ is combined with TiO₂ to form a heterojunction material, and photogenerated electrons are formed on the active sites on the surface of TiO₂ and transferred to MoS₂. Therefore, it promotes the separation of electron-hole pairs and improves the photocatalytic reaction efficiency. A negative effect was found with further increase in the MoS₂ loading. Part of the active sites on the TiO₂ surface is covered and the places where photoelectrons are generated on the TiO₂ surface reduce, resulting in a decrease in the photocatalytic degradation efficiency. Second, as the load of 1T-MoS₂ increases, the overall color of the composite material becomes darker. After adding the tannic acid solution, the whole solution becomes more turbid. During the photoreaction, part of the light source cannot reach the surface of the catalyst. Therefore, the photogenerated excitons cannot diffuse on the surface of MoS₂ and TiO₂ and form photogenerated electrons and holes. The materials cannot undergo redox reactions with tannic acid, thereby reducing the photocatalytic degradation efficiency.

Figure 6d,e shows the photocatalytic degradation of tannins using TiO₂ and the composite MoS₂@TiO₂ with different content of MoS_2 . In comparison with the blank sample, TiO_2 and the composite display enhanced photocatalytic activity. In this case, we compare the degradation rate of various amounts of 2H-MoS₂ within 2 h with less activity of 2H-MoS₂@TiO₂ as compared to 1T-MoS₂@TiO₂. A distinguished enhanced photocatalytic activity was demonstrated after adding 3% 2H-MoS₂ to TiO₂. Among the content of 3, 6, 9, 13, and 16%, the best degradation of 98% is obtained on the sample of 2H-MoS₂@TiO₂ with 13% 2H-MoS₂. Figure 6c,f shows photocatalytic degradation stability experiments of tannic acid in the presence of $1T-MoS_2@TiO_2$ (16%) and $2H-MoS_2@TiO_2$ (13%). In the four cycles of experiments, the photocatalytic degradation efficiency gradually decreases with the number of cycles, but all exceeded 80%, indicating that the two composite materials have good stability. Figure 6g,h shows the UV-vis curves of 1T-MoS₂@TiO₂ (16%) and 2H-MoS₂@TiO₂ (13%) photocatalytic degradation of tannic acid. The peak change of tannic acid at a wavelength of 275 nm within 120 min of photoreaction was analyzed. It can be seen that the degradation rate of tannic acid in the presence of 16% 1T- MoS_2 ($aTiO_2$ reaches 98% when the photoreaction is 60 min, while 13% 2H-MoS₂@TiO₂ (13%) achieves the same degradation rate when the photoreaction is 120 min. Figure 6i compares the highest photocatalytic degradation rate of tannic acid in the presence of various materials in this study in 1 h, demonstrating that the composite material of $1T - MoS_2@$ TiO_2 is a highly efficient catalyst.

No matter what phase MoS_2 is, it shows excellent photocatalytic degradation performance for tannic acid. A 98% degradation rate is achieved with 16% 1T-MoS₂@TiO₂ after 60 min and the same degradation is reached with 13% 2H-MoS₂@TiO₂ after 120 min. In 120 min, the degradation



Figure 7. Degradation efficiency of the composite materials under different quenchers. (a) 1T-MoS₂@TiO₂. (b) 2H-MoS₂@TiO₂.



Figure 8. (a) Electrochemical impedance diagram. (b) Mott–Schottky curve of TiO_2 . (c) Mott–Schottky curve of $1T-MoS_2$. (d) Mott–Schottky curve of $2H-MoS_2$.

rate of commercial TiO₂ is 38%. The as-prepared MoS₂@TiO₂ in this study shows much better photocatalytic activity than nano-TiO₂-coated tiles with 76% tannic acid degraded in 6 h.⁵³ This heterojunction material also has a comparable ability on the degradation of tannic acid using titanium oxide-coated magnetic activated carbon (m/v = 2:1, $C_{TA} = 100$ ppm) or Fe– N codoped SiO₂@TiO₂ (m/v = 1:1 mg/mL, $C_{TA} = 30$ ppm) with 98% degradation rate in 150 min and 99% degradation rate in 80 min.^{52,54} It is interesting that we use less catalysts (m/v = 1:8, $C_{TA} = 40$ ppm) and achieve a similar performance. The significantly improved photocatalytic activity might be contributed to the large surface area, excellent electron transfer ability, and enhanced active sites of the 1T-MoS₂@TiO₂ heterojunction material. Therefore, we can see that our results are promising in terms of materials selection and degradation activity toward tannic acid.

To determine the active species involved in the degradation of tannic acid, Figure 7 shows the effect of photocatalytic degradation of tannic acid by the two composite materials after adding different quenchers. The quencher for •OH is IPA, the quencher for holes (h^+) is EDTA, and the quencher for e^- is Na₂SO₄. It can be seen from Figure 7a,b that in the dark reaction stage, the effect of the quencher on its adsorption performance is almost negligible. For the two materials in the light reaction stage, the inhibitory effect after adding IPA is significantly higher than EDTA and Na₂SO₄. That is, •OH suppression has the most significant impact on the photocatalytic performance of this material. The photogenerated

28544



Figure 9. Schematic diagram of the photocatalytic degradation mechanism of 1T-MoS₂@TiO₂ and 2H-MoS₂@TiO₂.

electrons are absorbed by the adsorbed oxygen molecules to generate ${}^{\circ}O_{2}^{-}$, the photogenerated holes generate ${}^{\circ}OH$, and ${}^{\circ}O_{2}^{-}$ is generated in the aqueous solution. The presence of EDTA and Na₂SO₄ hindered the formation of photogenerated ${}^{\circ}O_{2}^{-}$ and holes, respectively, and affected the production of ${}^{\circ}OH$ in the catalytic reaction, thereby reducing the photocatalytic performance of the composite photocatalyst.

3.3. Electrochemical Performance of MoS₂@TiO₂. The fluorine-doped tin oxide (FTO) electrode is used in photocatalytic electrochemical performance. The preparation method is shown in the Supporting Material. Figure 8a shows the electrochemical impedance of several materials. From the Nyquist diagram, it can be seen that the arc radius of $1T-MoS_2$ is the smallest, the arc radius of TiO_2 is the largest, and the arc radius of 1T-MoS2@TiO2 is smaller than that of 2H-MoS₂@TiO₂. The high-frequency response of the Nyquist diagram is characteristic of the charge transfer process, and the arc radius of the AC impedance spectrum curve represents the resistance between the surface of the electrode material and the interface layer. The smaller the arc radius, the faster the charge transfer speed. Therefore, the surface charge transfer resistance of 1T-MoS₂@TiO₂ is less than that of 2H-MoS₂@TiO₂, which is more conducive to the separation of electrons and holes to improve the efficiency of the photocatalytic reaction. Compared with TiO₂ alone, the composite materials of 1T-MoS₂@TiO₂ and 2H-MoS₂@TiO₂ are significantly more effective in photocatalytic degradation of tannic acid. To study the conductivity and valence state of (1T/2H)-MoS₂ and TiO₂, we tested the energy band position by the Mott-Schottky (MS) method and observed from the MS curve that TiO_2 is an n-type semiconductor body. To clarify the energy band position of the material, the Mott-Schottky curves of TiO₂, 1T-MoS₂, and 2H-MoS₂ are shown in Figure 8b-d, respectively. Figure 8c shows that the 1T-MoS₂ Fermi level of the metal phase is -0.02 eV. The $E_{\rm VB}$ values of TiO₂ and 2H- MoS_2 are -0.07 and -0.11 V, respectively.

3.4. Photocatalytic Degradation Mechanism. Here, we try to explain the charge transfer mechanism of the two heterojunction materials. For $1T-MoS_2(@TiO_2)$, when light is irradiated on the surface of the composite material, the MoS_2 and TiO_2 surfaces form a tight surface because $1T-MoS_2$ belongs to the metal phase conductor. The large heterojunction interface can accelerate the transfer of electrons generated on the surface of TiO_2 , greatly improving the

efficiency of photocatalytic degradation. For 2H-MoS₂@TiO₂, when light hits the surface of the material, the photogenerated excitons diffuse rapidly on the surface of the material and dissociate into free electrons and holes at the conduction and valence bands of 2H-MoS₂ and TiO₂, respectively. The photogenerated electrons rapidly transfer from the conduction band of TiO₂ to the conduction band of MoS₂. An electron accumulation layer is formed in the conduction band of molybdenum sulfide, and a hole accumulation layer is formed in the valence band of titanium dioxide. The heterojunction enables the spatial separation of the electron-hole pairs to reduce the recombination rate effectively. The presence of MoS₂ in the two composite materials relatively shortens the original band gap value of TiO2, enables rapid electron migration, accelerates the separation of electron and hole pairs, and improves the efficiency of the photocatalytic reaction. Figure 9 schematically depicts the two possible reaction mechanisms behind the enhanced photocatalytic performance of these materials.

4. CONCLUSIONS

In summary, the composite materials 1T-MoS₂@TiO₂ and 2H- $MoS_2@TiO_2$ have been successfully synthesized. The results of the photocatalytic degradation of tannic acid show that these materials have significant effects on photocatalytic activity to degrade tannic acid. Loading 16% 1T-MoS₂ and 13% 2H-MoS₂ in the composite exhibits high photocatalytic degradation capacity of 98 and 75% in 60 min, much higher than that obtained by a commercial TiO_2 . The degradation speed of 16% $1T-MoS_2@TiO_2$ is much higher than that of $13\% 2H-MoS_2@$ TiO_2 . We believe that the tight interface between MoS_2/TiO_2 can effectively promote the transfer of light-induced electrons and provide many active sites for photocatalytic reactions. These features are key to the high-efficiency photocatalytic activity of MoS₂@TiO₂ photocatalysts. This study demonstrates the importance of MoS2@TiO2, a two-dimensional material, in improving the efficiency of photocatalytic activity. It provides a new idea for photocatalytic degradation of organic acids similar to tannic acid.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acsomega.1c02500.

Additional FTO electrode preparation and cell configuration, SEM images and high-resolution TEM images of $MoS_2@TiO_2$, and XPS spectra of the as-prepared materials (PDF)

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Notes

The authors declare no competing financial interest.

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