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# A new form of $Cd_3TeO_6$ revealing dimorphism

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Phase-formation studies in the system  $CdO$ – $TeO_3$  using a  $CsCl/NaCl$  melt at comparatively low temperatures revealed that tricadmium orthotellurate(VI),  $Cd_3TeO_6$ , is dimorphic. The new modification of  $Cd_3TeO_6$  is denoted as the  $\beta$ -form and adopts the rhombohedral  $Mg_3TeO_6$  structure type with one Cd and two O sites in general positions, and two Te sites with site symmetry  $\bar{3}$  each. In comparison with the previously reported monoclinic cryolite-type  $\alpha$ -form that was prepared at higher temperatures,  $\beta$ - $Cd_3TeO_6$  has a much lower density and most likely represents a metastable modification. Whereas the  $[TeO_6]$  octahedra in both polymorphs are very similar and show only minor deviations from ideal values, the polyhedra around the Cd<sup>II</sup> sites are different, with a distorted  $[CdO_6]$  octahedron in both modifications but an additional  $[CdO_8]$  polyhedron with a [4 + 4] coordination in the  $\alpha$ -form.

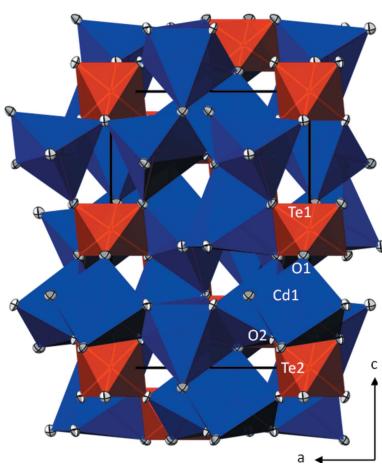
## 1. Chemical context

Various salts of *meta*-telluric acid,  $H_2TeO_4$ , have been reported as a result of high-pressure and high-temperature experiments (3000 atm; 973 K) aiming at various  $M^{II}TeO_4$  phases, where  $M = Mg, Ca, Sr, Ba, Cd$  or  $Pb$  (Sleight *et al.*, 1972). Meanwhile, the crystal structures of the Ca, Sr and Ba salts were determined (Hottentot & Loopstra, 1979; Weil *et al.*, 2016) whereas those of the other phases remain unknown to date. In a recent project on single-crystal growth of the Cd salt of *meta*-telluric acid, we used a  $CsCl/NaCl$  salt mixture (Žemcžužny & Rambach, 1909) at temperatures  $< 800$  K as a flux. Instead of the target phase  $CdTeO_4$ , we obtained a new form of  $Cd_3TeO_6$ . The previously reported  $Cd_3TeO_6$  polymorph crystallizes as a monoclinically distorted cryolite-type material in space-group type  $P2_1/n$  (Burckhardt *et al.*, 1982) while the new form adopts the rhombohedral  $Mg_3TeO_6$  structure type.

Prior to the current study, solid solutions  $Cd_{3-x}Mn_xTeO_6$  with  $x = 3, 2, 1.5$  and 1 were prepared in polycrystalline form (Ivanov *et al.*, 2012), but not the cadmium end member, *i.e.* where  $x = 0$ . We report here the crystal structure of the new polymorph of  $Cd_3TeO_6$ , together with a comparative discussion of isostructural solid solutions  $Cd_{3-x}Mn_xTeO_6$ . In the following, we refer to the previously reported monoclinic polymorph of  $Cd_3TeO_6$  (Burckhardt *et al.*, 1982) as the  $\alpha$ -form, and the new rhombohedral polymorph as the  $\beta$ -form of  $Cd_3TeO_6$ .

## 2. Structural commentary

The crystal structure of  $\beta$ - $Cd_3TeO_6$  (Fig. 1) is made up from a distorted close packing of hexagonal oxygen layers extending



**Table 1**

Selected bond lengths ( $\text{\AA}$ ) in rhombohedral  $\beta\text{-Cd}_3\text{TeO}_6$  and in isotopic ( $\text{Cd}_{1.5}\text{Mn}_{1.5}\text{TeO}_6$  and  $\text{Mn}_3\text{TeO}_6$ ).

|                      | $\beta\text{-Cd}_3\text{TeO}_6^a$ | $\text{Cd}_{1.5}\text{Mn}_{1.5}\text{TeO}_6^b$ | $\text{Mn}_3\text{TeO}_6^c$ |
|----------------------|-----------------------------------|--|-----------------------------|
| M1—O1                | 2.2348 (17)                       | 2.147  | 2.1055 (14)                 |
| M1—O2 <sup>i</sup>   | 2.2455 (17)                       | 2.150  | 2.1275 (13)                 |
| M1—O1 <sup>ii</sup>  | 2.2907 (19)                       | 2.240  | 2.2009 (13)                 |
| M1—O2 <sup>iii</sup> | 2.3051 (18)                       | 2.260  | 2.2311 (12)                 |
| M1—O2                | 2.3370 (18)                       | 2.273  | 2.2313 (13)                 |
| M1—O1 <sup>iv</sup>  | 2.4658 (19)                       | 2.412  | 2.3841 (13)                 |
| Te1—O1               | 1.9339 (17)                       | 1.955  | 1.9247 (13)                 |
| Te2—O2               | 1.9290 (17)                       | 1.959  | 1.9214 (12)                 |

Notes: (a) This study; (b) Ivanov *et al.* (2012) on the basis of X-ray powder diffraction data at room temperature (no s.u. given in original publication); (c) Weil (2006) on the basis of single-crystal X-ray data at room temperature. [Symmetry codes: (i)  $y - \frac{1}{3}, -x + y + \frac{1}{3}, -z + \frac{2}{3}$ ; (ii)  $-x + \frac{1}{3}, y + \frac{2}{3}, -z + \frac{2}{3}$ ; (iii)  $-y, x - y, z$ ; (iv)  $-y + \frac{1}{3}, x - y + \frac{2}{3}, z - \frac{1}{3}$ ]

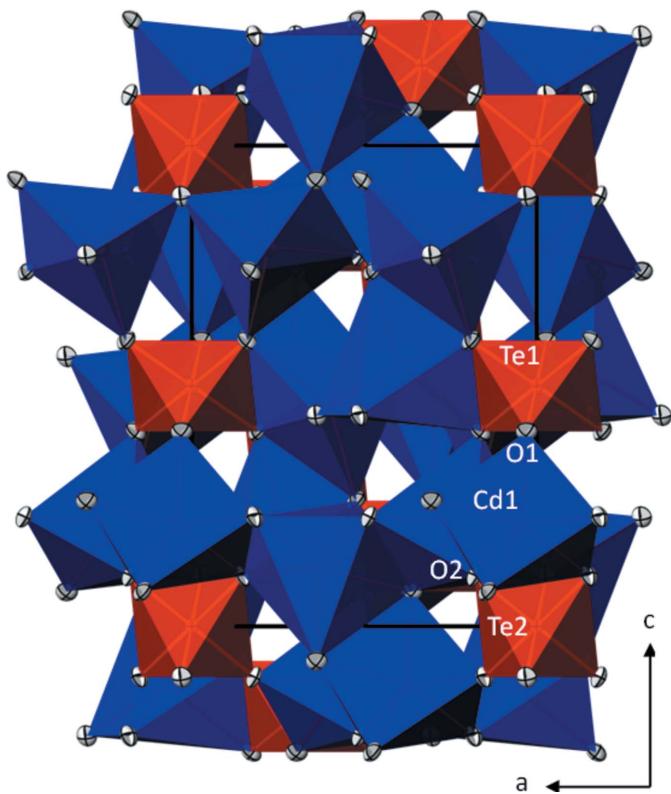
parallel to (001). The Cd site (site symmetry 1) and the two unique Te sites (each with site symmetry  $\bar{3}$ ) are situated in the octahedral interstices of this arrangement. The distorted  $[\text{CdO}_6]$  octahedron has Cd—O distances ranging from 2.2348 (17)–2.4658 (19)  $\text{\AA}$  (Table 1) and shares one edge with a  $[\text{Te}1\text{O}_6]$  octahedron, another edge with a  $[\text{Te}2\text{O}_6]$  octahedron, and four edges with neighbouring  $[\text{CdO}_6]$  octahedra. Both  $[\text{TeO}_6]$  octahedra show only minute deviations from the ideal octahedral symmetry. They are isolated from each other and are connected to six  $[\text{CdO}_6]$  octahedra by sharing edges. The average Te—O bond length in  $\beta\text{-Cd}_3\text{TeO}_6$  (1.931  $\text{\AA}$ ) is in

very good agreement with the mean Te—O bond length of 1.923  $\text{\AA}$  calculated for numerous ( $> 100$ ) oxotellurates with octahedrally coordinated Te<sup>VI</sup> (Christy *et al.*, 2016; Gagné & Hawthorne, 2018). Both unique O atoms are bonded to one Te and three Cd atoms in the form of a distorted tetrahedron.

Like  $\beta\text{-Cd}_3\text{TeO}_6$ ,  $\text{Mn}_3\text{TeO}_6$  (Weil, 2006) as well as phases with  $x = 2, 1.5$  and 1 of the  $\text{Cd}_{3-x}\text{Mn}_x\text{TeO}_6$  solid-solution series (Ivanov *et al.*, 2012) adopt the rhombohedral  $\text{Mg}_3\text{TeO}_6$  structure type. A comparison of the bond lengths of the  $[\text{MO}_6]$  ( $M = \text{Cd}, \text{Mn}$ ) octahedra in the end members  $\beta\text{-Cd}_3\text{TeO}_6$  and  $\text{Mn}_3\text{TeO}_6$  and the solid solution  $\text{Cd}_{1.5}\text{Mn}_{1.5}\text{TeO}_6$  (mixed occupancy for the  $M$  site) shows intermediate values for the solid solution, consistent with the different ionic radii for six-coordinate  $\text{Cd}^{II}$  and  $\text{Mn}^{II}$  of 0.95 and 0.83 (high-spin)  $\text{\AA}$ , respectively (Shannon, 1976). For a quantitative structural comparison of the end members  $\beta\text{-Cd}_3\text{TeO}_6$  and  $\text{Mn}_3\text{TeO}_6$  the program *compstru* (de la Flor *et al.*, 2016) available at the Bilbao Crystallographic Server (Aroyo *et al.*, 2006) was used. The degree of lattice distortion is 0.0204, the maximum distance between the atomic positions of paired atoms is 0.0680  $\text{\AA}$  for pair O2, the arithmetic mean of all distances is 0.0417  $\text{\AA}$ , and the measure of similarity is 0.011. All these values show a high similarity between the two crystal structures.

The structure of the monoclinic  $\alpha$ -form of  $\text{Cd}_3\text{TeO}_6$  (Burckhardt *et al.*, 1982) comprises of two cadmium sites (one on a general position and one on an inversion centre), one tellurium site on an inversion centre and three oxygen sites in general positions. While the  $[\text{TeO}_6]$  octahedra in both  $\text{Cd}_3\text{TeO}_6$  polymorphs have nearly the same bond length distribution [ $2 \times 1.904 (4)$ ,  $2 \times 1.924 (5)$ ,  $2 \times 1.948 (4)$   $\text{\AA}$  in the  $\alpha$ -form; for the  $\beta$ -form, see: Table 1], the set of coordination polyhedra around the two  $\text{Cd}^{II}$  cations in the two structures is different. In  $\beta\text{-Cd}_3\text{TeO}_6$ , the cadmium site has a coordination number (CN) of six with an octahedral oxygen environment whereas in  $\alpha\text{-Cd}_3\text{TeO}_6$ , only one site is octahedrally surrounded [range of Cd—O bond lengths: 2.211 (5)–2.350 (4)  $\text{\AA}$ ] and the other site exhibits a distorted [4 + 4] coordination [range of Cd—O bond lengths: 2.237 (5)–3.010 (5)  $\text{\AA}$ ].

As noted above, the end members  $\beta\text{-Cd}_3\text{TeO}_6$  and  $\text{Mn}_3\text{TeO}_6$  crystallize in the same structure type, suggesting a full miscibility over the complete range of  $x$  for the solid-solution series  $\text{Cd}_{3-x}\text{Mn}_x\text{TeO}_6$ . However, the adopted structure type for the complete range of  $x$  appears to be dependent on the reaction temperature. Single crystals of  $\alpha\text{-Cd}_3\text{TeO}_6$  for structure analysis were grown from a 9 CdO: 11  $\text{TeO}_2$  mixture that was heated in air at 1350 K for three h (Burckhardt *et al.*, 1982) while single crystals of  $\beta\text{-Cd}_3\text{TeO}_6$  were obtained at much lower temperatures (793 K) using a flux method. This suggests that the high-temperature synthesis yields the thermodynamically stable modification. The rule of thumb that in the majority of cases the denser polymorph represents also the thermodynamically stable modification supports this assumption because  $\alpha\text{-Cd}_3\text{TeO}_6$  [ $D_x = 7.490 (2) \text{ g cm}^{-3}$ ; Burckhardt *et al.*, 1982] is much denser than  $\beta\text{-Cd}_3\text{TeO}_6$  [ $D_x = 6.941 \text{ g cm}^{-3}$ ]. Under consideration of the similar reaction conditions for

**Figure 1**

The crystal structure of  $\beta\text{-Cd}_3\text{TeO}_6$  in polyhedral view in a projection along [010].  $[\text{CdO}_6]$  octahedra are blue and  $[\text{TeO}_6]$  octahedra are red. Displacement ellipsoids are drawn at the 90% probability level.

**Table 2**  
Experimental details.

|   |   |
|---|---|
| Crystal data  |   |
| Chemical formula  | Cd <sub>3</sub> TeO <sub>6</sub>                          |
| M <sub>r</sub>  | 560.80  |
| Crystal system, space group                                       | Trigonal, R <bar{3}>.H</bar{3}>                           |
| Temperature (K)   | 296   |
| a, c (Å)  | 9.1620 (2), 11.0736 (3)                                   |
| V (Å <sup>3</sup> )   | 805.01 (4)  |
| Z   | 6   |
| Radiation type  | Mo K $\alpha$   |
| $\mu$ (mm <sup>-1</sup> )   | 17.06   |
| Crystal size (mm)   | 0.08 (radius)   |
| Data collection   |   |
| Diffractometer  | Bruker APEXII CCD   |
| Absorption correction   | Multi-scan ( <i>SADABS</i> ; Krause <i>et al.</i> , 2015) |
| T <sub>min</sub> , T <sub>max</sub>                               | 0.527, 0.749  |
| No. of measured, independent and observed [I > 2σ(I)] reflections | 11351, 1623, 1526   |
| R <sub>int</sub>  | 0.033   |
| (sin θ/λ) <sub>max</sub> (Å <sup>-1</sup> )                       | 1.025   |
| Refinement  |   |
| R[F <sup>2</sup> > 2σ(F <sup>2</sup> )], wR(F <sup>2</sup> ), S   | 0.023, 0.046, 1.29  |
| No. of reflections  | 1623  |
| No. of parameters   | 33  |
| Δρ <sub>max</sub> , Δρ <sub>min</sub> (e Å <sup>-3</sup> )        | 2.57, -1.53   |

Computer programs: *APEX3* and *SAINT* (Bruker, 2015), *SHELXL2017/1* (Sheldrick, 2015), *ATOMS for Windows* (Dowty, 2006) and *publCIF* (Westrip, 2010).

preparation of monoclinic α-Cd<sub>3</sub>TeO<sub>6</sub> and the given solid solutions Cd<sub>3-x</sub>Mn<sub>x</sub>TeO<sub>6</sub> (1270 K following a ceramic route; Ivanov *et al.*, 2012), it appears likely that the rhombohedral β-Cd<sub>3</sub>TeO<sub>6</sub> end member can be prepared only at lower temperatures whereas certain amounts of manganese substituting cadmium in the Cd<sub>3-x</sub>Mn<sub>x</sub>TeO<sub>6</sub> solid-solution series stabilize the Mg<sub>3</sub>TeO<sub>6</sub> structure type at higher temperatures. Unfortunately, because of the scarcity of β-Cd<sub>3</sub>TeO<sub>6</sub> material, a detailed investigation of the thermal behaviour of this phase, *e.g.* in terms of stability and a possible phase transition to α-Cd<sub>3</sub>TeO<sub>6</sub>, could not be undertaken.

### 3. Database survey

According to a search of the Inorganic Crystal Structure Database (ICSD; Belsky *et al.*, 2002), the Mg<sub>3</sub>TeO<sub>6</sub> structure type is realized for eponymous Mg<sub>3</sub>TeO<sub>6</sub> (Schulz & Bayer, 1971), Ca<sub>3</sub>UO<sub>6</sub> (Holc & Golic, 1983), Mn<sub>3</sub>WO<sub>6</sub> (Klüver & Müller-Buschbaum, 1994), Li<sub>3</sub>AlD<sub>6</sub> (Brinks & Hauback, 2003; Løvvik *et al.*, 2004), Mn<sub>3</sub>TeO<sub>6</sub> (Weil, 2006), selected solid solutions Cd<sub>3-x</sub>Mn<sub>x</sub>TeO<sub>6</sub> (Ivanov *et al.*, 2012), Mn<sub>3-x</sub>Co<sub>x</sub>TeO<sub>6</sub> (Singh *et al.*, 2014; Ivanov *et al.*, 2014), Mn<sub>2.4</sub>Cu<sub>0.6</sub>TeO<sub>6</sub> (Wulff *et al.*, 1998), (Ca<sub>0.2667</sub>Y<sub>0.7333</sub>)<sub>3</sub>(Y<sub>0.2</sub>Sn<sub>0.3</sub>)Sn<sub>0.5</sub>O<sub>6</sub> (Kaminaga *et al.*, 2006), Mn<sub>2</sub>InSbO<sub>6</sub> and Mn<sub>2</sub>ScSbO<sub>6</sub> (Ivanov *et al.*, 2011), Sc<sub>3</sub>(Sc<sub>0.295</sub>Al<sub>0.705</sub>)O<sub>6</sub> (Müller *et al.*, 2004) and Ho<sub>3</sub>ScO<sub>6</sub> (Badie, 1973).

### 4. Synthesis and crystallization

The rhombohedral β-form of Cd<sub>3</sub>TeO<sub>3</sub> was obtained as one of the products from a flux synthesis using a CsCl/NaCl salt

mixture (molar ratio 0.65/0.35). To 1.7 g of the salt mixture were added CdO (0.13 g) and TeO<sub>3</sub> (0.18 g). TeO<sub>3</sub> had previously been prepared by heating H<sub>6</sub>TeO<sub>6</sub> at 573 K for 8 h. The reaction mixture was evacuated and sealed in a silica ampoule, heated from room temperature within 3 h to 793 K, kept at that temperature for 90 h and cooled within 10 h back to room temperature. The silica ampoule was subsequently broken and the solidified melt leached out with water for 2 h. The off-white product was filtered off, washed with water and was air-dried. The title compound was present in the form of a few nearly spherical colourless crystals. Other phases identified by single-crystal X-ray diffraction measurements of selected crystals were α-Cd<sub>3</sub>TeO<sub>6</sub> (Burckhardt *et al.*, 1982), the mixed-valent Te<sup>IV/VI</sup> compound Cd<sub>2</sub>Te<sub>2</sub>O<sub>7</sub> (Weil, 2004) and a new form of incommensurately modulated CdTe<sub>2</sub>O<sub>5</sub> (Weil & Stöger, 2018). Estimated on optical inspection with a microscope, all these phases represent minor by-products. Powder X-ray diffraction measurements of the bulk additionally revealed triple-perovskite-type CsCdCl<sub>3</sub> (Siegel & Gebert, 1964) as the main phase and the Te<sup>IV</sup> compound CdTeO<sub>3</sub> (Krämer & Brandt, 1985) as a minority phase. Some additional reflections in the X-ray powder diffraction pattern of the bulk could not be assigned to the phases mentioned above or to any other known phase(s).

### 5. Refinement

Crystal data, data collection and structure refinement details are summarized in Table 2. Standardized coordinates (Gelato & Parthé, 1987) from the isotopic phase Mn<sub>3</sub>TeO<sub>6</sub> (Weil, 2006) were taken as starting parameters for refinement. The highest and lowest remaining electron density peaks are located 1.56 and 1.53 Å from sites Te2 and O1, respectively.

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# supporting information

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## A new form of Cd<sub>3</sub>TeO<sub>6</sub> revealing dimorphism

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### Computing details

Data collection: *APEX3* (Bruker, 2015); cell refinement: *SAINT* (Bruker, 2015); data reduction: *SAINT* (Bruker, 2015); program(s) used to solve structure: coordinates from isotopic structure; program(s) used to refine structure: *SHELXL2017/1* (Sheldrick, 2015); molecular graphics: *ATOMS for Windows* (Dowty, 2006); software used to prepare material for publication: *publCIF* (Westrip, 2010).

### Tricadmium orthotellurate(VI)

#### Crystal data

|                                  |  |
|----------------------------------|--|
| Cd <sub>3</sub> TeO <sub>6</sub> | $D_x = 6.941 \text{ Mg m}^{-3}$                                |
| $M_r = 560.80$                   | Mo $K\alpha$ radiation, $\lambda = 0.71073 \text{ \AA}$        |
| Trigonal, $R\bar{3}:H$           | Cell parameters from 6637 reflections                          |
| $a = 9.1620 (2) \text{ \AA}$     | $\theta = 5.0\text{--}46.6^\circ$                              |
| $c = 11.0736 (3) \text{ \AA}$    | $\mu = 17.06 \text{ mm}^{-1}$                                  |
| $V = 805.01 (4) \text{ \AA}^3$   | $T = 296 \text{ K}$  |
| $Z = 6$                          | Spherical, colourless  |
| $F(000) = 1464$                  | $0.08 \times 0.08 \times 0.08 \times 0.08 \text{ (radius) mm}$ |

#### Data collection

|  |   |
|--|---|
| Bruker APEXII CCD                                | 1623 independent reflections  |
| diffractometer                                   | 1526 reflections with $I > 2\sigma(I)$                              |
| $\omega$ - and $\varphi$ -scans                  | $R_{\text{int}} = 0.033$  |
| Absorption correction: multi-scan                | $\theta_{\text{max}} = 46.8^\circ, \theta_{\text{min}} = 3.2^\circ$ |
| ( <i>SADABS</i> ; Krause <i>et al.</i> , 2015)   | $h = -18 \rightarrow 18$  |
| $T_{\text{min}} = 0.527, T_{\text{max}} = 0.749$ | $k = -18 \rightarrow 16$  |
| 11351 measured reflections                       | $l = -22 \rightarrow 22$  |

#### Refinement

|                                 |  |
|---------------------------------|--|
| Refinement on $F^2$             | $w = 1/[\sigma^2(F_o^2) + (0.0021P)^2 + 11.2674P]$                   |
| Least-squares matrix: full      | where $P = (F_o^2 + 2F_c^2)/3$                                       |
| $R[F^2 > 2\sigma(F^2)] = 0.023$ | $(\Delta/\sigma)_{\text{max}} < 0.001$                               |
| $wR(F^2) = 0.046$               | $\Delta\rho_{\text{max}} = 2.57 \text{ e \AA}^{-3}$                  |
| $S = 1.29$                      | $\Delta\rho_{\text{min}} = -1.53 \text{ e \AA}^{-3}$                 |
| 1623 reflections                | Extinction correction: <i>SHELXL-2017/1</i>                          |
| 33 parameters                   | (Sheldrick 2015),  |
| 0 restraints                    | $F_c^* = k F_c [1 + 0.001 x F_c^2 \lambda^3 / \sin(2\theta)]^{-1/4}$ |
|                                 | Extinction coefficient: 0.00434 (9)                                  |

*Special details*

**Geometry.** All esds (except the esd in the dihedral angle between two l.s. planes) are estimated using the full covariance matrix. The cell esds are taken into account individually in the estimation of esds in distances, angles and torsion angles; correlations between esds in cell parameters are only used when they are defined by crystal symmetry. An approximate (isotropic) treatment of cell esds is used for estimating esds involving l.s. planes.

*Fractional atomic coordinates and isotropic or equivalent isotropic displacement parameters ( $\text{\AA}^2$ )*

|     | <i>x</i>    | <i>y</i>    | <i>z</i>     | $U_{\text{iso}}^*/U_{\text{eq}}$ |
|-----|-------------|-------------|--------------|----------------------------------|
| Cd1 | 0.03947 (2) | 0.26424 (2) | 0.21210 (2)  | 0.00731 (4)                      |
| Te1 | 0.000000    | 0.000000    | 0.500000     | 0.00444 (5)                      |
| Te2 | 0.000000    | 0.000000    | 0.000000     | 0.00424 (5)                      |
| O1  | 0.0289 (2)  | 0.1903 (2)  | 0.40560 (16) | 0.0087 (2)                       |
| O2  | 0.1800 (2)  | 0.1509 (2)  | 0.10570 (16) | 0.0078 (2)                       |

*Atomic displacement parameters ( $\text{\AA}^2$ )*

|     | $U^{11}$    | $U^{22}$    | $U^{33}$     | $U^{12}$    | $U^{13}$     | $U^{23}$     |
|-----|-------------|-------------|--------------|-------------|--------------|--------------|
| Cd1 | 0.00657 (6) | 0.00745 (6) | 0.00789 (6)  | 0.00348 (5) | -0.00052 (4) | -0.00099 (4) |
| Te1 | 0.00419 (7) | 0.00419 (7) | 0.00492 (10) | 0.00210 (3) | 0.000        | 0.000        |
| Te2 | 0.00403 (7) | 0.00403 (7) | 0.00464 (10) | 0.00202 (3) | 0.000        | 0.000        |
| O1  | 0.0102 (6)  | 0.0074 (6)  | 0.0082 (5)   | 0.0043 (5)  | -0.0001 (5)  | 0.0024 (4)   |
| O2  | 0.0058 (5)  | 0.0069 (6)  | 0.0094 (6)   | 0.0021 (5)  | -0.0024 (4)  | -0.0017 (4)  |

*Geometric parameters ( $\text{\AA}$ ,  $^\circ$ )*

|   |             |   |             |
|---|-------------|---|-------------|
| Cd1—O1                                  | 2.2348 (17) | Te1—O1 <sup>viii</sup>                      | 1.9339 (18) |
| Cd1—O2 <sup>i</sup>                     | 2.2455 (17) | Te1—O1 <sup>iii</sup>                       | 1.9339 (17) |
| Cd1—O1 <sup>ii</sup>                    | 2.2907 (19) | Te1—O1 <sup>ix</sup>                        | 1.9339 (17) |
| Cd1—O2 <sup>iii</sup>                   | 2.3051 (18) | Te1—O1 <sup>x</sup>                         | 1.9339 (17) |
| Cd1—O2                                  | 2.3370 (18) | Te1—O1 <sup>xi</sup>                        | 1.9339 (17) |
| Cd1—O1 <sup>iv</sup>                    | 2.4658 (19) | Te1—O1                                      | 1.9339 (17) |
| Cd1—Te2                                 | 3.2608 (2)  | Te2—O2                                      | 1.9290 (17) |
| Cd1—Te1 <sup>v</sup>                    | 3.3420 (2)  | Te2—O2 <sup>xii</sup>                       | 1.9290 (17) |
| Cd1—Cd1 <sup>ii</sup>                   | 3.3606 (3)  | Te2—O2 <sup>iii</sup>                       | 1.9290 (17) |
| Cd1—Cd1 <sup>vi</sup>                   | 3.4239 (3)  | Te2—O2 <sup>xiii</sup>                      | 1.9291 (17) |
| Cd1—Cd1 <sup>vii</sup>                  | 3.4537 (2)  | Te2—O2 <sup>xiv</sup>                       | 1.9291 (17) |
| Cd1—Cd1 <sup>i</sup>                    | 3.4538 (2)  | Te2—O2 <sup>xi</sup>                        | 1.9291 (17) |
|   |             |   |             |
| O1—Cd1—O2 <sup>i</sup>                  | 94.05 (7)   | O1 <sup>x</sup> —Te1—Cd1 <sup>xvii</sup>    | 79.65 (6)   |
| O1—Cd1—O1 <sup>ii</sup>                 | 84.10 (7)   | O1 <sup>xi</sup> —Te1—Cd1 <sup>xvii</sup>   | 100.35 (6)  |
| O2 <sup>i</sup> —Cd1—O1 <sup>ii</sup>   | 120.35 (7)  | O1—Te1—Cd1 <sup>xvii</sup>                  | 138.37 (6)  |
| O1—Cd1—O2 <sup>iii</sup>                | 107.98 (7)  | Cd1 <sup>xv</sup> —Te1—Cd1 <sup>xvii</sup>  | 117.776 (1) |
| O2 <sup>i</sup> —Cd1—O2 <sup>iii</sup>  | 82.41 (7)   | Cd1 <sup>xvi</sup> —Te1—Cd1 <sup>xvii</sup> | 62.224 (1)  |
| O1 <sup>ii</sup> —Cd1—O2 <sup>iii</sup> | 154.11 (6)  | Cd1 <sup>ii</sup> —Te1—Cd1 <sup>xvii</sup>  | 180.0       |
| O1—Cd1—O2                               | 107.37 (6)  | O1 <sup>viii</sup> —Te1—Cd1 <sup>xvii</sup> | 100.35 (6)  |
| O2 <sup>i</sup> —Cd1—O2                 | 148.88 (4)  | O1 <sup>iii</sup> —Te1—Cd1 <sup>xvii</sup>  | 79.65 (6)   |
| O1 <sup>ii</sup> —Cd1—O2                | 84.88 (6)   | O1 <sup>ix</sup> —Te1—Cd1 <sup>xvii</sup>   | 46.92 (5)   |

|   |              |   |             |
|---|--------------|---|-------------|
| O2 <sup>iii</sup> —Cd1—O2                 | 69.79 (8)    | O1 <sup>x</sup> —Te1—Cd1 <sup>xviii</sup>     | 138.37 (6)  |
| O1—Cd1—O1 <sup>iv</sup>                   | 144.36 (6)   | O1 <sup>xi</sup> —Te1—Cd1 <sup>xviii</sup>    | 41.63 (6)   |
| O2 <sup>i</sup> —Cd1—O1 <sup>iv</sup>     | 82.90 (6)    | O1—Te1—Cd1 <sup>xviii</sup>                   | 133.08 (5)  |
| O1 <sup>ii</sup> —Cd1—O1 <sup>iv</sup>    | 67.58 (8)    | Cd1 <sup>xv</sup> —Te1—Cd1 <sup>xviii</sup>   | 62.225 (1)  |
| O2 <sup>iii</sup> —Cd1—O1 <sup>iv</sup>   | 106.78 (6)   | Cd1 <sup>xvi</sup> —Te1—Cd1 <sup>xviii</sup>  | 117.775 (1) |
| O2—Cd1—O1 <sup>iv</sup>                   | 91.75 (6)    | Cd1 <sup>ii</sup> —Te1—Cd1 <sup>xviii</sup>   | 117.775 (1) |
| O1—Cd1—Te2                                | 119.57 (5)   | Cd1 <sup>xvii</sup> —Te1—Cd1 <sup>xviii</sup> | 62.225 (1)  |
| O2 <sup>i</sup> —Cd1—Te2                  | 113.76 (5)   | O1 <sup>viii</sup> —Te1—Cd1 <sup>xix</sup>    | 79.65 (6)   |
| O1 <sup>ii</sup> —Cd1—Te2                 | 118.53 (5)   | O1 <sup>iii</sup> —Te1—Cd1 <sup>xix</sup>     | 100.35 (6)  |
| O2 <sup>iii</sup> —Cd1—Te2                | 35.59 (4)    | O1 <sup>ix</sup> —Te1—Cd1 <sup>xix</sup>      | 133.08 (5)  |
| O2—Cd1—Te2                                | 35.72 (4)    | O1 <sup>x</sup> —Te1—Cd1 <sup>xix</sup>       | 41.63 (6)   |
| O1 <sup>iv</sup> —Cd1—Te2                 | 93.60 (4)    | O1 <sup>xi</sup> —Te1—Cd1 <sup>xix</sup>      | 138.37 (6)  |
| O1—Cd1—Te1 <sup>v</sup>                   | 111.58 (5)   | O1—Te1—Cd1 <sup>xix</sup>                     | 46.92 (5)   |
| O2 <sup>i</sup> —Cd1—Te1 <sup>v</sup>     | 97.31 (5)    | Cd1 <sup>xv</sup> —Te1—Cd1 <sup>xix</sup>     | 117.775 (1) |
| O1 <sup>ii</sup> —Cd1—Te1 <sup>v</sup>    | 34.11 (4)    | Cd1 <sup>xvi</sup> —Te1—Cd1 <sup>xix</sup>    | 62.225 (1)  |
| O2 <sup>iii</sup> —Cd1—Te1 <sup>v</sup>   | 140.34 (4)   | Cd1 <sup>ii</sup> —Te1—Cd1 <sup>xix</sup>     | 62.225 (1)  |
| O2—Cd1—Te1 <sup>v</sup>                   | 95.47 (4)    | Cd1 <sup>xvii</sup> —Te1—Cd1 <sup>xix</sup>   | 117.775 (1) |
| O1 <sup>iv</sup> —Cd1—Te1 <sup>v</sup>    | 34.95 (4)    | Cd1 <sup>xviii</sup> —Te1—Cd1 <sup>xix</sup>  | 180.0       |
| Te2—Cd1—Te1 <sup>v</sup>                  | 116.088 (5)  | O2—Te2—O2 <sup>xii</sup>                      | 93.00 (8)   |
| O1—Cd1—Cd1 <sup>ii</sup>                  | 42.69 (5)    | O2—Te2—O2 <sup>iii</sup>                      | 87.00 (8)   |
| O2 <sup>i</sup> —Cd1—Cd1 <sup>ii</sup>    | 113.04 (5)   | O2 <sup>xii</sup> —Te2—O2 <sup>iii</sup>      | 180.00 (11) |
| O1 <sup>ii</sup> —Cd1—Cd1 <sup>ii</sup>   | 41.41 (4)    | O2—Te2—O2 <sup>xiii</sup>                     | 180.0       |
| O2 <sup>iii</sup> —Cd1—Cd1 <sup>ii</sup>  | 144.94 (5)   | O2 <sup>xii</sup> —Te2—O2 <sup>xiii</sup>     | 87.00 (8)   |
| O2—Cd1—Cd1 <sup>ii</sup>                  | 97.92 (4)    | O2 <sup>iii</sup> —Te2—O2 <sup>xiii</sup>     | 93.00 (8)   |
| O1 <sup>iv</sup> —Cd1—Cd1 <sup>ii</sup>   | 106.29 (4)   | O2—Te2—O2 <sup>xiv</sup>                      | 93.00 (8)   |
| Te2—Cd1—Cd1 <sup>ii</sup>                 | 130.823 (8)  | O2 <sup>xii</sup> —Te2—O2 <sup>xiv</sup>      | 87.00 (8)   |
| Te1 <sup>v</sup> —Cd1—Cd1 <sup>ii</sup>   | 71.352 (5)   | O2 <sup>iii</sup> —Te2—O2 <sup>xiv</sup>      | 93.00 (8)   |
| O1—Cd1—Cd1 <sup>vi</sup>                  | 104.72 (5)   | O2 <sup>xiii</sup> —Te2—O2 <sup>xiv</sup>     | 87.00 (8)   |
| O2 <sup>i</sup> —Cd1—Cd1 <sup>vi</sup>    | 41.86 (5)    | O2—Te2—O2 <sup>xi</sup>                       | 87.00 (8)   |
| O1 <sup>ii</sup> —Cd1—Cd1 <sup>vi</sup>   | 159.57 (5)   | O2 <sup>xii</sup> —Te2—O2 <sup>xi</sup>       | 93.00 (8)   |
| O2 <sup>iii</sup> —Cd1—Cd1 <sup>vi</sup>  | 40.55 (4)    | O2 <sup>iii</sup> —Te2—O2 <sup>xi</sup>       | 87.00 (8)   |
| O2—Cd1—Cd1 <sup>vi</sup>                  | 109.20 (4)   | O2 <sup>xiii</sup> —Te2—O2 <sup>xi</sup>      | 93.00 (8)   |
| O1 <sup>iv</sup> —Cd1—Cd1 <sup>vi</sup>   | 96.51 (4)    | O2 <sup>xiv</sup> —Te2—O2 <sup>xi</sup>       | 180.00 (13) |
| Te2—Cd1—Cd1 <sup>vi</sup>                 | 73.547 (5)   | O2—Te2—Cd1 <sup>xiii</sup>                    | 134.98 (5)  |
| Te1 <sup>v</sup> —Cd1—Cd1 <sup>vi</sup>   | 126.995 (8)  | O2 <sup>xii</sup> —Te2—Cd1 <sup>xiii</sup>    | 44.06 (5)   |
| Cd1 <sup>ii</sup> —Cd1—Cd1 <sup>vi</sup>  | 143.877 (10) | O2 <sup>iii</sup> —Te2—Cd1 <sup>xiii</sup>    | 135.94 (5)  |
| O1—Cd1—Cd1 <sup>vii</sup>                 | 104.17 (5)   | O2 <sup>xiii</sup> —Te2—Cd1 <sup>xiii</sup>   | 45.02 (5)   |
| O2 <sup>i</sup> —Cd1—Cd1 <sup>vii</sup>   | 153.98 (5)   | O2 <sup>xiv</sup> —Te2—Cd1 <sup>xiii</sup>    | 96.56 (5)   |
| O1 <sup>ii</sup> —Cd1—Cd1 <sup>vii</sup>  | 45.47 (5)    | O2 <sup>xi</sup> —Te2—Cd1 <sup>xiii</sup>     | 83.44 (5)   |
| O2 <sup>iii</sup> —Cd1—Cd1 <sup>vii</sup> | 108.67 (4)   | O2—Te2—Cd1 <sup>xi</sup>                      | 44.06 (5)   |
| O2—Cd1—Cd1 <sup>vii</sup>                 | 40.10 (4)    | O2 <sup>xii</sup> —Te2—Cd1 <sup>xi</sup>      | 83.44 (5)   |
| O1 <sup>iv</sup> —Cd1—Cd1 <sup>vii</sup>  | 71.44 (4)    | O2 <sup>iii</sup> —Te2—Cd1 <sup>xi</sup>      | 96.56 (5)   |
| Te2—Cd1—Cd1 <sup>vii</sup>                | 73.148 (6)   | O2 <sup>xiii</sup> —Te2—Cd1 <sup>xi</sup>     | 135.94 (5)  |
| Te1 <sup>v</sup> —Cd1—Cd1 <sup>vii</sup>  | 58.887 (1)   | O2 <sup>xiv</sup> —Te2—Cd1 <sup>xi</sup>      | 134.98 (5)  |
| Cd1 <sup>ii</sup> —Cd1—Cd1 <sup>vii</sup> | 71.628 (5)   | O2 <sup>xi</sup> —Te2—Cd1 <sup>xi</sup>       | 45.02 (5)   |
| Cd1 <sup>vi</sup> —Cd1—Cd1 <sup>vii</sup> | 143.600 (10) | Cd1 <sup>xiii</sup> —Te2—Cd1 <sup>xi</sup>    | 106.152 (5) |
| O1—Cd1—Cd1 <sup>i</sup>                   | 122.01 (5)   | O2—Te2—Cd1                                    | 45.02 (5)   |
| O2 <sup>i</sup> —Cd1—Cd1 <sup>i</sup>     | 42.10 (5)    | O2 <sup>xii</sup> —Te2—Cd1                    | 135.94 (5)  |

|   |             |   |              |
|---|-------------|---|--------------|
| O1 <sup>ii</sup> —Cd1—Cd1 <sup>i</sup>      | 90.20 (4)   | O2 <sup>iii</sup> —Te2—Cd1                  | 44.06 (5)    |
| O2 <sup>iii</sup> —Cd1—Cd1 <sup>i</sup>     | 101.62 (5)  | O2 <sup>xiii</sup> —Te2—Cd1                 | 134.98 (5)   |
| O2—Cd1—Cd1 <sup>i</sup>                     | 129.56 (4)  | O2 <sup>xiv</sup> —Te2—Cd1                  | 83.44 (5)    |
| O1 <sup>iv</sup> —Cd1—Cd1 <sup>i</sup>      | 41.48 (4)   | O2 <sup>xi</sup> —Te2—Cd1                   | 96.56 (5)    |
| Te2—Cd1—Cd1 <sup>i</sup>                    | 113.622 (7) | Cd1 <sup>xiii</sup> —Te2—Cd1                | 180.0        |
| Te1 <sup>v</sup> —Cd1—Cd1 <sup>i</sup>      | 58.888 (1)  | Cd1 <sup>xi</sup> —Te2—Cd1                  | 73.847 (5)   |
| Cd1 <sup>ii</sup> —Cd1—Cd1 <sup>i</sup>     | 110.788 (6) | O2—Te2—Cd1 <sup>xiv</sup>                   | 135.94 (5)   |
| Cd1 <sup>vi</sup> —Cd1—Cd1 <sup>i</sup>     | 69.450 (8)  | O2 <sup>xii</sup> —Te2—Cd1 <sup>xiv</sup>   | 96.56 (5)    |
| Cd1 <sup>vii</sup> —Cd1—Cd1 <sup>i</sup>    | 111.882 (5) | O2 <sup>iii</sup> —Te2—Cd1 <sup>xiv</sup>   | 83.44 (5)    |
| O1 <sup>viii</sup> —Te1—O1 <sup>iii</sup>   | 180.0       | O2 <sup>xiii</sup> —Te2—Cd1 <sup>xiv</sup>  | 44.06 (5)    |
| O1 <sup>viii</sup> —Te1—O1 <sup>ix</sup>    | 93.54 (7)   | O2 <sup>xiv</sup> —Te2—Cd1 <sup>xiv</sup>   | 45.02 (5)    |
| O1 <sup>iii</sup> —Te1—O1 <sup>ix</sup>     | 86.46 (7)   | O2 <sup>xi</sup> —Te2—Cd1 <sup>xiv</sup>    | 134.98 (5)   |
| O1 <sup>viii</sup> —Te1—O1 <sup>x</sup>     | 93.54 (7)   | Cd1 <sup>xiii</sup> —Te2—Cd1 <sup>xiv</sup> | 73.848 (5)   |
| O1 <sup>iii</sup> —Te1—O1 <sup>x</sup>      | 86.46 (7)   | Cd1 <sup>xi</sup> —Te2—Cd1 <sup>xiv</sup>   | 180.0        |
| O1 <sup>ix</sup> —Te1—O1 <sup>x</sup>       | 93.54 (7)   | Cd1—Te2—Cd1 <sup>xiv</sup>                  | 106.153 (5)  |
| O1 <sup>viii</sup> —Te1—O1 <sup>xi</sup>    | 86.46 (7)   | O2—Te2—Cd1 <sup>xii</sup>                   | 83.44 (5)    |
| O1 <sup>iii</sup> —Te1—O1 <sup>xi</sup>     | 93.54 (7)   | O2 <sup>xii</sup> —Te2—Cd1 <sup>xii</sup>   | 45.02 (5)    |
| O1 <sup>ix</sup> —Te1—O1 <sup>xi</sup>      | 86.46 (7)   | O2 <sup>iii</sup> —Te2—Cd1 <sup>xii</sup>   | 134.98 (5)   |
| O1 <sup>x</sup> —Te1—O1 <sup>xi</sup>       | 180.0       | O2 <sup>xiii</sup> —Te2—Cd1 <sup>xii</sup>  | 96.56 (5)    |
| O1 <sup>viii</sup> —Te1—O1                  | 86.46 (7)   | O2 <sup>xiv</sup> —Te2—Cd1 <sup>xii</sup>   | 44.06 (5)    |
| O1 <sup>iii</sup> —Te1—O1                   | 93.54 (7)   | O2 <sup>xi</sup> —Te2—Cd1 <sup>xii</sup>    | 135.94 (5)   |
| O1 <sup>ix</sup> —Te1—O1                    | 180.0       | Cd1 <sup>xiii</sup> —Te2—Cd1 <sup>xii</sup> | 73.848 (5)   |
| O1 <sup>x</sup> —Te1—O1                     | 86.46 (7)   | Cd1 <sup>xi</sup> —Te2—Cd1 <sup>xii</sup>   | 106.152 (5)  |
| O1 <sup>xi</sup> —Te1—O1                    | 93.54 (7)   | Cd1—Te2—Cd1 <sup>xii</sup>                  | 106.153 (5)  |
| O1 <sup>viii</sup> —Te1—Cd1 <sup>xv</sup>   | 41.63 (6)   | Cd1 <sup>xiv</sup> —Te2—Cd1 <sup>xii</sup>  | 73.848 (5)   |
| O1 <sup>iii</sup> —Te1—Cd1 <sup>xv</sup>    | 138.37 (6)  | O2—Te2—Cd1 <sup>iii</sup>                   | 96.56 (5)    |
| O1 <sup>ix</sup> —Te1—Cd1 <sup>xv</sup>     | 79.65 (6)   | O2 <sup>xii</sup> —Te2—Cd1 <sup>iii</sup>   | 134.98 (5)   |
| O1 <sup>x</sup> —Te1—Cd1 <sup>xv</sup>      | 133.08 (6)  | O2 <sup>iii</sup> —Te2—Cd1 <sup>iii</sup>   | 45.02 (5)    |
| O1 <sup>xi</sup> —Te1—Cd1 <sup>xv</sup>     | 46.92 (6)   | O2 <sup>xiii</sup> —Te2—Cd1 <sup>iii</sup>  | 83.44 (5)    |
| O1—Te1—Cd1 <sup>xv</sup>                    | 100.35 (6)  | O2 <sup>xiv</sup> —Te2—Cd1 <sup>iii</sup>   | 135.94 (5)   |
| O1 <sup>viii</sup> —Te1—Cd1 <sup>xvi</sup>  | 138.37 (6)  | O2 <sup>xi</sup> —Te2—Cd1 <sup>iii</sup>    | 44.06 (5)    |
| O1 <sup>iii</sup> —Te1—Cd1 <sup>xvi</sup>   | 41.63 (6)   | Cd1 <sup>xiii</sup> —Te2—Cd1 <sup>iii</sup> | 106.152 (5)  |
| O1 <sup>ix</sup> —Te1—Cd1 <sup>xvi</sup>    | 100.35 (6)  | Cd1 <sup>xi</sup> —Te2—Cd1 <sup>iii</sup>   | 73.848 (5)   |
| O1 <sup>x</sup> —Te1—Cd1 <sup>xvi</sup>     | 46.92 (6)   | Cd1—Te2—Cd1 <sup>iii</sup>                  | 73.847 (5)   |
| O1 <sup>xi</sup> —Te1—Cd1 <sup>xvi</sup>    | 133.08 (6)  | Cd1 <sup>xiv</sup> —Te2—Cd1 <sup>iii</sup>  | 106.152 (5)  |
| O1—Te1—Cd1 <sup>xvi</sup>                   | 79.65 (6)   | Cd1 <sup>xii</sup> —Te2—Cd1 <sup>iii</sup>  | 180.000 (11) |
| Cd1 <sup>xv</sup> —Te1—Cd1 <sup>xvi</sup>   | 180.0       | Te1—O1—Cd1                                  | 139.23 (10)  |
| O1 <sup>viii</sup> —Te1—Cd1 <sup>ii</sup>   | 46.92 (5)   | Te1—O1—Cd1 <sup>ii</sup>                    | 104.25 (8)   |
| O1 <sup>iii</sup> —Te1—Cd1 <sup>ii</sup>    | 133.08 (5)  | Cd1—O1—Cd1 <sup>ii</sup>                    | 95.90 (7)    |
| O1 <sup>ix</sup> —Te1—Cd1 <sup>ii</sup>     | 138.37 (6)  | Te1—O1—Cd1 <sup>xix</sup>                   | 98.13 (7)    |
| O1 <sup>x</sup> —Te1—Cd1 <sup>ii</sup>      | 100.35 (6)  | Cd1—O1—Cd1 <sup>xix</sup>                   | 116.00 (7)   |
| O1 <sup>xi</sup> —Te1—Cd1 <sup>ii</sup>     | 79.65 (6)   | Cd1 <sup>ii</sup> —O1—Cd1 <sup>xix</sup>    | 93.05 (7)    |
| O1—Te1—Cd1 <sup>ii</sup>                    | 41.63 (6)   | Te2—O2—Cd1 <sup>vii</sup>                   | 147.03 (10)  |
| Cd1 <sup>xv</sup> —Te1—Cd1 <sup>ii</sup>    | 62.223 (1)  | Te2—O2—Cd1 <sup>xi</sup>                    | 100.35 (7)   |
| Cd1 <sup>xvi</sup> —Te1—Cd1 <sup>ii</sup>   | 117.776 (1) | Cd1 <sup>vii</sup> —O2—Cd1 <sup>xi</sup>    | 97.59 (7)    |
| O1 <sup>viii</sup> —Te1—Cd1 <sup>xvii</sup> | 133.08 (5)  | Te2—O2—Cd1                                  | 99.25 (7)    |

|  |           |                            |            |
|--|-----------|----------------------------|------------|
| O1 <sup>iii</sup> —Te1—Cd1 <sup>xvii</sup> | 46.92 (5) | Cd1 <sup>vii</sup> —O2—Cd1 | 97.80 (7)  |
| O1 <sup>ix</sup> —Te1—Cd1 <sup>xvii</sup>  | 41.63 (6) | Cd1 <sup>xi</sup> —O2—Cd1  | 115.12 (8) |

Symmetry codes: (i)  $y-1/3, -x+y+1/3, -z+1/3$ ; (ii)  $-x+1/3, -y+2/3, -z+2/3$ ; (iii)  $-y, x-y, z$ ; (iv)  $-y+1/3, x-y+2/3, z-1/3$ ; (v)  $x+1/3, y+2/3, z-1/3$ ; (vi)  $-x-1/3, -y+1/3, -z+1/3$ ; (vii)  $x-y+2/3, x+1/3, -z+1/3$ ; (viii)  $y, -x+y, -z+1$ ; (ix)  $-x, -y, -z+1$ ; (x)  $x-y, x, -z+1$ ; (xi)  $-x+y, -x, z$ ; (xii)  $y, -x+y, -z$ ; (xiii)  $-x, -y, -z$ ; (xiv)  $x-y, x, -z$ ; (xv)  $-y+2/3, x-y+1/3, z+1/3$ ; (xvi)  $y-2/3, -x+y-1/3, -z+2/3$ ; (xvii)  $x-1/3, y-2/3, z+1/3$ ; (xviii)  $x-y+1/3, x-1/3, -z+2/3$ ; (xix)  $-x+y-1/3, -x+1/3, z+1/3$ .