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Evaluation of the Corrosion Resistance Properties of Electroplated Chitosan-Zn_{1-x}Cu_xO Composite Thin Films

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Received: 4 November 2017; Accepted: 1 December 2017; Published: 6 December 2017

Abstract: Novel chitosan–zinc copper oxide (Zn_{1-x}Cu_xO) composites were electrochemically synthesized through galvanostatic deposition. The prepared chitosan-based composite thin films were elaborately investigated to determine their structural, morphological, compositional, impedance, and corrosion properties. X-ray diffraction analysis was performed to reveal their structural orientation of composite thin films. Energy dispersive analysis by X-ray evidently confirmed the existence of Zn, Cu, and O in the composite thin films. Nyquist plots revealed that the chitosan-Zn_{1-x}Cu_xO thin films had obvious semi-circular boundaries, and higher resistance was observed for chitosan-ZnO due to the grain boundary effect. Corrosion properties were evaluated using both an electrochemical method and the ASTM weight gain method, which revealed good corrosion rates of 34 and 35 × 10⁻³ mm/y, respectively, for chitosan-ZnO thin film.

Keywords: electrochemical; composite thin films; corrosion; morphology; impedance

1. Introduction

Organic-inorganic nanocomposite materials are attracting substantial attention because of their combination feasibility in the properties of organic and inorganic components [1–5]. Significant interest has been spawned in fabrication of nanocomposite films containing metal oxide nanoparticles in a polymer matrix through electrochemical routes [6,7]. Compared to other preparation routes such as layer-by-layer self-assembly, electrodeposition imparts benefits such as higher deposition rate, shorter processing time, and the possibility of depositing thicker films. Moreover, shape selective fabrication over a uniform film surface with controlled composition can be obtained on different forms of the conducting substrate [8,9]. The fabrication of composite films can be achieved by electrochemical co-deposition of organic and inorganic components [10]. Recent reports have revealed the electrochemical preparation of various natural biomacromolecules such as chitosan [11,12], alginate acid [13], and hyaluronic acid [14,15], and that electrodeposition is a feasible route to fabricate thin and porous structured films [16,17]. Chitosan is an important natural polymer for various applications including biomedical sensors, implants, and anti-microbial and microfluidic devices [12,18–20]. Nanocomposite materials based on chitosan with metal oxide materials have become attractive in recent years due to their synergistic behavior [21]. Recently, Sanmugam et al. [21] successfully demonstrated the solvent free synthesis of chitosan-zinc oxide (ZnO) nanocomposites using chitosan and ZnCl₂. Earlier, Li et al. [22] reported the synthesis of chitosan-ZnO thin films using an electrochemical route.

ZnO is a semiconductor material with a wide range of applications including biosensors, catalysts, and photovoltaic devices [23–27]. Various methods have been used to synthesize ZnO nanoparticles

such as sol-gel, radio-frequency (RF) sputtering, chemical vapor deposition, pulsed laser deposition, and spray pyrolysis [28–32]. In addition, ZnO lattices have been shown to contain various types of defect helping them to behave as n-type semi-conductors due to zinc and oxygen vacancies and interstitials, as well as more complex defects [33–38]. Some research groups have modified ZnO properties through adding dopant materials such as N, Al, Ga, Cu, Mg, and In, which can significantly enhance the electrical, dielectric, and optical properties of ZnO [30,32,38–40]. Copper oxide (CuO) is a p-type semiconductor material used in a broad range of applications including anti-corrosion properties, solar cells, biosensors, gas sensor, superconductor, lithium-ion electrode materials, magnetic storage media, and field effect transistors [41–44], and its composite materials have recently attracted attention due to promising features in electrochemical behavior [45,46]. Recently, Arena et al. [47] demonstrated electrochemically-derived chitosan-copper oxide nanocomposites for successful non-enzymatic sensing of hydrogen peroxide.

Corrosion is a common problem which significantly affects the properties of materials [48,49], and high resistance to corrosion is attributed to the spontaneous development of a chemically stable oxide film surface [49–51]. Transition metal oxides and conducting polymers are the promising candidates due to their strong bonding structure which can be withstanding in diverse applications including anti-corrosion applications [20,52–55]. From the detailed investigations of earlier literatures, we have prepared chitosan-zinc copper oxide (chitosan-Zn_{1-x}Cu_xO) composite thin films with various concentrations of zinc and copper chloride by electrosynthesis. To the best of our knowledge, there are no reports available on the electrosynthesis of chitosan-Zn_{1-x}Cu_xO composite thin films. The structural, compositional, morphological, and electrochemical properties of the prepared composite thin films were studied in detail. Hence, we explored the corrosion behavior of electrosynthesized chitosan-Zn_{1-x}Cu_xO composite thin films using an electrochemical and the American standard test method (ASTM) weight loss methods efficiently. The best corrosion resistance performance was exhibited in the chitosan-ZnO thin film, compared with other composite thin films, with the corrosion rates of 34 and 35 × 10⁻³ mm/y by an electrochemical method and the ASTM weight gain method, respectively.

2. Results and Discussion

The chitosan-Zn_{1-x}Cu_xO composite thin films were electrochemically synthesized on a mild steel substrate using galvanostatic mode. In order to adjust the Zn and Cu element composition, zinc chloride (ZnCl₂) and copper chloride (CuCl₂) concentrations were adjusted in an electrolyte bath. Figure 1a–c illustrates the electrochemically synthesized chitosan-Zn_{1-x}Cu_xO composite thin films on a mild steel substrate, the prepared film structure, and the corroded surface of a composite film post testing, respectively.

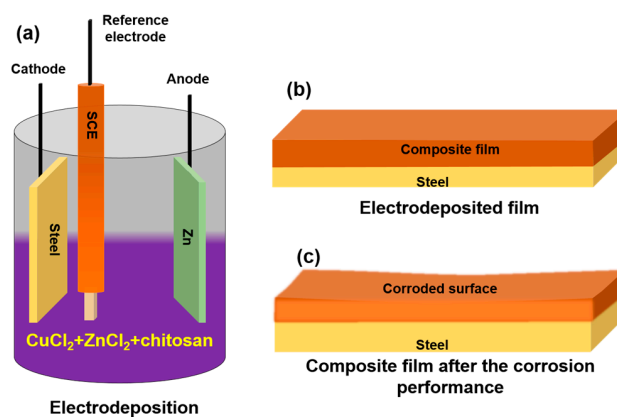


Figure 1. Schematic diagrams of the chitosan-Zn_{1-x}Cu_xO composite thin film. Electrosynthesis (a) and structure of the films before (b) and after (c) corrosion performance testing.

Their structural properties were studied using X-ray diffraction (XRD) analyses. Figure 2a–d shows typical X-ray diffraction patterns of chitosan-ZnO, chitosan-Zn_{0.6}Cu_{0.4}O, chitosan-Zn_{0.3}Cu_{0.7}O, and chitosan-CuO composite thin films, respectively. The observed XRD patterns were indexed with joint committee of powder diffraction standard (JCPDS) patterns of CuO (#89-5898 & #78-2076) and ZnO (#89-0511). The XRD pattern in Figure 2a reveals that the electrosynthesized chitosan-ZnO composite thin film exhibited a polycrystalline hexagonal structure. In this XRD pattern, chitosan-related conventional diffraction lines CS1 (13.2°), and CS3 (17.3°) were observed, as has previously been reported [22]. In addition to this, a conventional ZnO (002) lattice orientation peak was present at $2\theta = 33.9^\circ$, which revealed that chitosan was more dominant than ZnO in the chitosan-ZnO complex matrix. In the XRD pattern of chitosan-Zn_{0.6}Cu_{0.4}O composite thin film (Figure 2b), strong chitosan-related CS1 (13.1°) and CS3 (17.3°) peaks as well as a low intensity CS4 peak (18.8°) were observed with CS1 being the predominant peak orientation. Moreover, ZnO lattice planes (100), (002), (101), (220), (110), (103), and (004) along with CuO lattice planes (002), (−111), (200), (−112), (112), (202), (−113), and (022), and CuO-related lattice planes (110), (−111), (−112), (112), (020), (022), (221), and (004) with (200) as the predominant orientation were present in the XRD pattern. The ZnO related (002), (101), (103), and (201) lattice orientations were observed for the chitosan-Zn_{0.3}Cu_{0.7}O thin film (Figure 2c). In addition, conventional chitosan-based CS1 (@13.2°), CS2 (@15.9°), CS3 (@17.3°), and CS4 (@18.7°) peaks were present, which clearly infers that the Zn_{1-x}Cu_xO alloy was fully incorporated into the chitosan polymer. From the XRD pattern of chitosan-CuO composite thin film (Figure 2d), (110), (200), (−112), (112), and (022) lattice planes were predicted for CuO in addition to the chitosan conventional peaks.

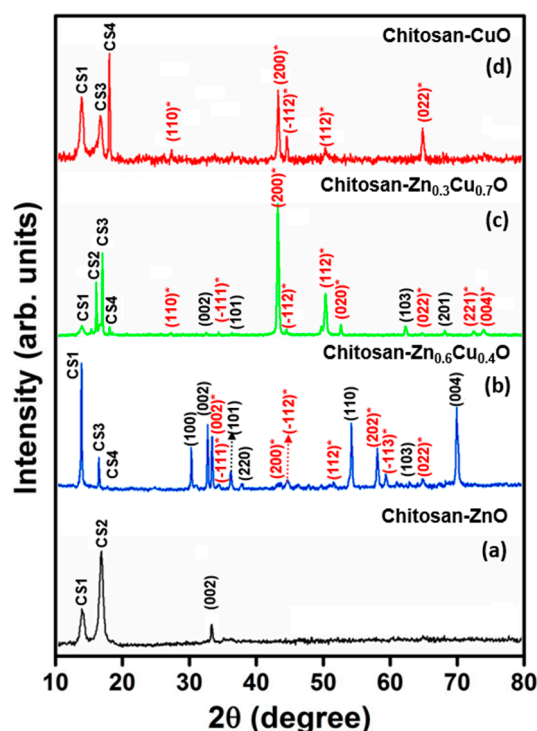


Figure 2. XRD patterns of electro-synthesized composite thin films: (a) chitosan-ZnO; (b) chitosan-Zn_{0.6}Cu_{0.4}O; (c) chitosan-Zn_{0.3}Cu_{0.7}O; and (d) chitosan-CuO (red color with * indexed peaks are CuO-based lattice planes, black color indexed peaks are ZnO-based lattice planes, and chitosan-based peaks are indexed with prefix CS).

The average crystallite size of chitosan-Zn_{1-x}Cu_xO composite thin films were calculated using the Debye–Scherrer’s equation [56]:

$$D = K\lambda / \beta \cos\theta, \quad (1)$$

where D is the crystallite size, K is the Scherrer constant, λ is the X-ray wavelength, β is the full-width at half-maximum, and θ is the diffraction angle. The crystallite size was found to be 25, 31, 41, and 38 nm for chitosan-ZnO, chitosan-Zn_{0.6}Cu_{0.4}O, chitosan-Zn_{0.3}Cu_{0.7}O, and chitosan-CuO composite thin films, respectively, revealing a decrement in crystallite size in the absence of Zn in the chitosan-CuO composite.

The morphological properties of the composites were studied using scanning electron microscopy (SEM), which was a convenient method for studying the films' surfaces. The micrographs revealed morphological differences with various combinations of precursor solution with chitosan. Nano-slab-like morphology was observed in the chitosan-ZnO composite thin film, as shown in Figure 3a, in which some discontinuities and overlapping were evident. An inhomogeneous surface with voids and hillocks was observed in the chitosan-Zn_{0.6}Cu_{0.4}O composite thin film SEM image (Figure 3b). The surface image of chitosan-Zn_{0.3}Cu_{0.7}O composite thin film is presented in Figure 3c, in which spherical-shaped fine grains covered the entire surface of the film, resulting in smooth surface morphology from the lower grain size. Smooth and uniform surface morphological properties are evident on the surface of the chitosan-CuO composite thin film (Figure 3d). The observed results indicate that the film surface was altered by adjusting the precursor combination with chitosan for electrosynthesis of the films.

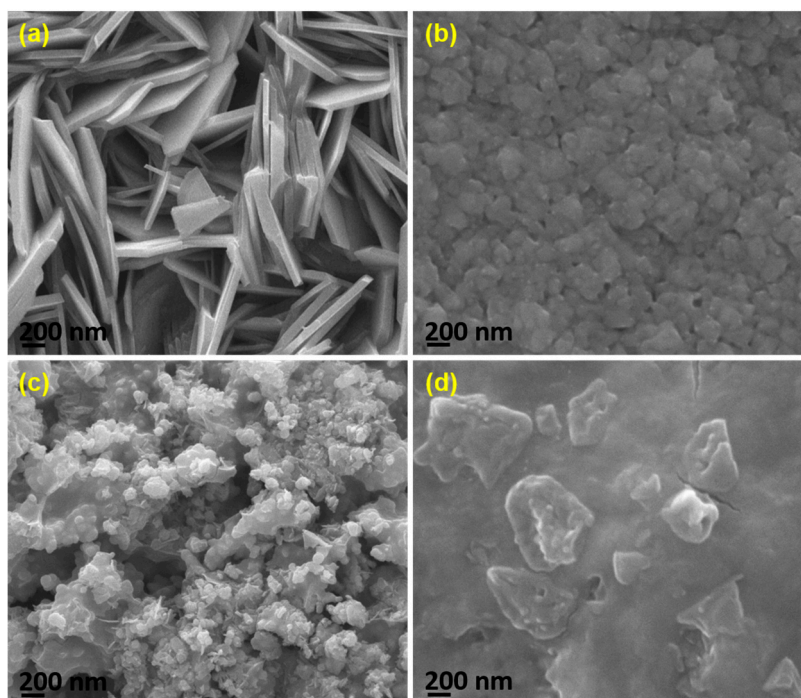


Figure 3. SEM Micrographs of electrosynthesized composite thin films (a) chitosan-ZnO; (b) chitosan-Zn_{0.6}Cu_{0.4}O; (c) chitosan-Zn_{0.3}Cu_{0.7}O; and (d) chitosan-CuO.

The nanostructure formation with stoichiometric composition were confirmed by energy dispersive analysis by X-rays (EDAX) studies [57]. The composition ratio of the metal oxides combined with chitosan in the composite thin films using EDAX are shown in Figure 4a–d. Chitosan-ZnO composite thin film had a combination mixture of 32.80 and 36.66 corresponding to Zn and O, respectively, as shown in Figure 4a. The EDAX spectrum of chitosan-Zn_{0.6}Cu_{0.4}O composite thin film (Figure 4b) revealed the atomic percentage of Zn, Cu, and O to be 22.58, 14.72, and 31.88, respectively. From Figure 4c, the observed atomic percentages of Cu, Zn and O were 26.80, 12.46, and 33.66, respectively, for the chitosan-Zn_{0.7}Cu_{0.3}O composite thin film. The compositional ratio of Cu and O were 29.97 and 34.91, respectively, in the chitosan-CuO thin film (Figure 4d).

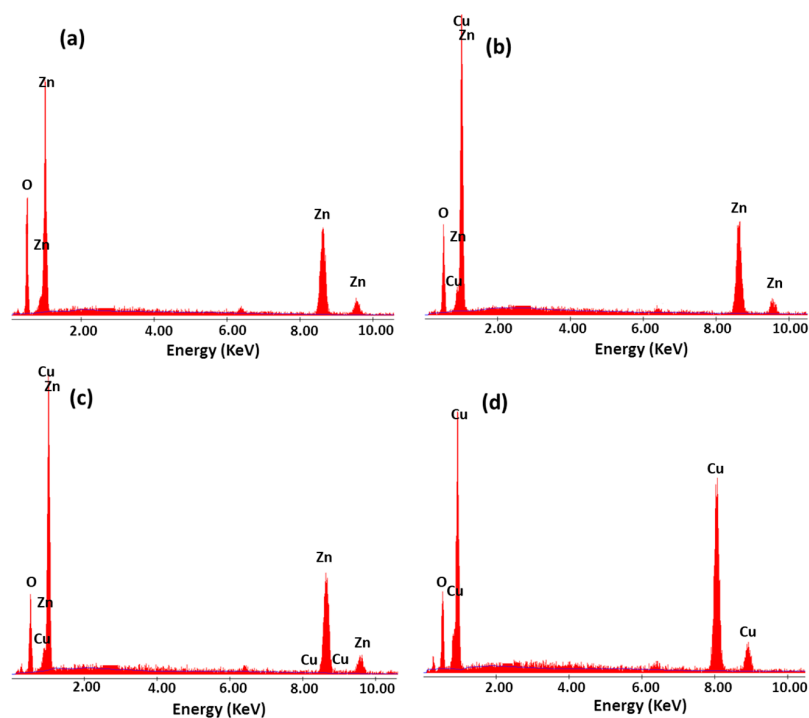


Figure 4. EDAX spectra of electro synthesized composite thin films: (a) chitosan-ZnO; (b) chitosan-Zn_{0.6}Cu_{0.4}O; (c) chitosan-Zn_{0.3}Cu_{0.7}O; and (d) chitosan-CuO.

Figure 5a shows the cyclic voltammograms (CVs) of the electro synthesized composite thin films recorded using the composite thin films as electrodes with an electrolyte solution of 0.1 M HCl at a scan rate of 20 mV s⁻¹. The electro synthesized composite thin films showed good redox electrochemical behavior in acidic solution with an anodic peak obtained at around 0.18 V vs. a saturated calomel electrode (SCE), while a cathodic peak observed at around 0.55 V vs. SCE for chitosan-ZnO. In addition, the cathodic peak shifted toward negative and the anodic reaction rate decreased with a decrease in Zn content in the chitosan-Zn_{1-x}Cu_xO composite films. The positive shift of chitosan-ZnO thin films depressed the anodic current as they offered greater resistance [58]. It is evident that the electro synthesized composite thin films obeyed the expected electron transfer in an acidic medium [59]. In this case, the chitosan-ZnO electro synthesized thin films acted as effective conductivity barriers [60].

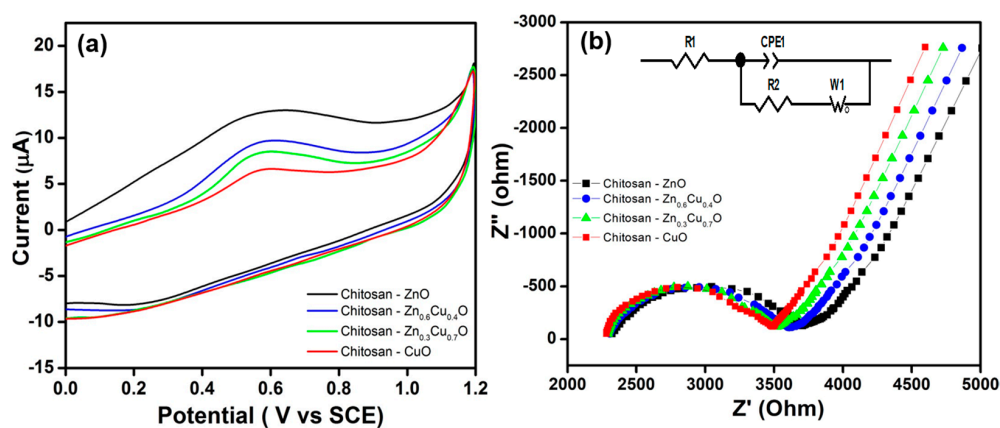


Figure 5. (a) CV curves and (b) impedance spectra for the electro synthesized composite thin films. Inset—the Randomize-Simplex model fitted circuit consisting of resistor R₁ in series with a parallel combination of R₂ C₂, which is in series with W₃.

Electrochemical impedance spectroscopy (EIS) analysis was carried for the electro synthesized chitosan-Zn_{1-x}Cu_xO composite thin films at room temperature, as shown in Figure 5b. The semi-circle part of the curve shortened with an increase in Cu atomic percentage in the chitosan-Zn_{1-x}Cu_xO film [61]. High resistivity was observed for the chitosan-ZnO thin film, which might have been due to more defects caused by the larger grain size [62]. The semicircle was attributed to the grain boundary and the straight line indicated electron transport at the thin electrode/electrolyte interface. The resistance decreased obviously for chitosan-Zn_{0.6}Cu_{0.4}O, chitosan-Zn_{0.3}Cu_{0.7}O, and chitosan-CuO composite thin films compared to chitosan-ZnO due to the grain boundary effect. The estimated EIS parameter values are given in Table 1. We can see that the bulk resistance values decreased with respect to the percentage of Cu atoms incorporated into the chitosan-ZnO matrix, which increased the conductivity of the chitosan-Zn_{1-x}Cu_xO (x = 0.4, 0.7, and 1) matrix system. Similarly, Lee et al. [63] observed a decrease in resistance with an increase in Sn cation composition in ZnO.

Table 1. EIS and corrosion parameters of electro synthesized composite thin films.

Composite	EIS Parameters			Corrosion Parameters			
	R1 (Ohm)	R2 (Ohm)	CPE1 μF	Electrochemical Method		Weight Loss Method	
				Corrosion Current Density (cm ⁻²)	Corrosion Rate (mmpy) 10 ⁻³	Weight Loss (mg)	Corrosion Rate (mmpy) 10 ⁻³
Chitosan-ZnO	2321	3699	0.63	2.81 × 10 ⁻⁶	34	160	35
Chitosan-Zn _{0.6} Cu _{0.4} O	2308	3610	0.58	8.12 × 10 ⁻⁶	99	260	58
Chitosan-Zn _{0.3} Cu _{0.7} O	2290	3510	0.52	2.63 × 10 ⁻⁵	320	415	93
Chitosan-CuO	2282	3488	0.31	5.12 × 10 ⁻⁵	623	635	141

Anodic corrosion was recorded for the electrochemical route prepared for chitosan-ZnO, chitosan-Zn_{0.6}Cu_{0.4}O, chitosan-Zn_{0.3}Cu_{0.7}O and chitosan-CuO composite thin films on steel electrodes in 3% NaCl (*w/v*) medium. The coated steel surface was maintained under potentiodynamic conditions with a potential sweep between -0.5 and -0.6 V vs. SCE. The corrosion current densities (*j*_{corr}) were obtained by extrapolating the linear portions to zero in Tafel plots. Similar observations were reported for copper oxides with a corrosion mechanism by Wan et al. [64]. The corrosion current and corrosion potential were determined by extrapolating the linear portions of the anodic and cathodic Tafel curves from Figure 6, which clearly show the corrosion current density and potential of the various chitosan-based composites. The corrosion current density of the chitosan-ZnO thin film coated steel electrode was 2.81 × 10⁻⁶ A/cm², which was quite low compared to the values for the other composites. The polarization curves determined that the electrochemically synthesized chitosan-ZnO coating inhibited the anodic dissolution of steel in the corrosive solution. The corrosion rate was evaluated in accordance with the following equation [65]:

$$CR = 3272(j_{\text{corr}}EM)/Ad, \quad (2)$$

where *CR* is the corrosion rate in mm per year, *j*_{corr} is the corrosion current density in cm⁻², *EM* is the equivalent molar mass of the oxidized element in g/equiv molar mass, *A* is the surface area of the specimen in cm² and *d* is the density of the specimen in g/cm³.

The corrosion rate of chitosan-ZnO was 34 × 10⁻³ mm/y due to its low current density of 2.81 × 10⁻⁶ A/cm², indicating that it had higher corrosion resistance compared to the other composites. The chitosan-Zn_{0.6}Cu_{0.4}O composite thin film's corrosion resistance value was 99 × 10⁻³ mm/y with a slightly bowed polarization curve and its current density was 8.12 × 10⁻⁶ A/cm², as shown in Figure 6. Furthermore, the chitosan-Zn_{0.3}Cu_{0.7}O composite thin film corrosion resistance value was found to be 320 × 10⁻³ mm/y with a current density of 2.63 × 10⁻⁵ A/cm². Finally, the chitosan-CuO composite thin film corrosion resistance value was 623 × 10⁻³ mm/y with a semicircle of the polarization creating a semicircle on the plot, and its current density value was estimated at 5.12 × 10⁻⁵ A/cm². From the above results, we can confirm that the chitosan-ZnO composite thin film exhibited the best corrosion resistance.

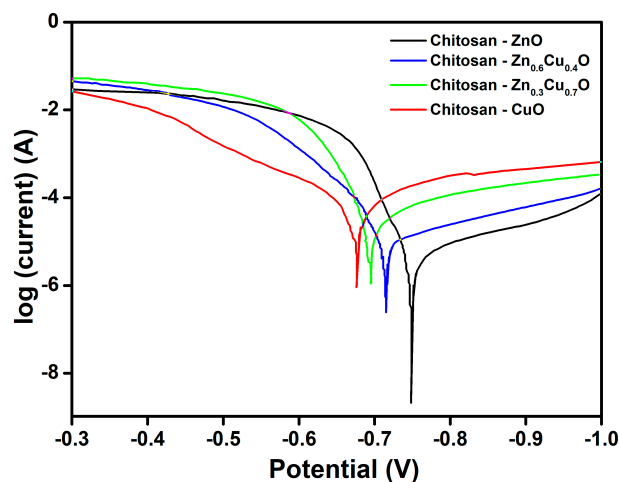


Figure 6. Tafel polarization curves for electro synthesized composite thin films.

For comparison purposes, the corrosion rate (CR) estimated using the ASTM standard weight loss method [66,67] with the following equation:

$$CR = K T d W / A, \quad (3)$$

where CR is corrosion rate in mm/year (mmpy), K is a unit conversion constant, T is the period of immersion in hours, A is area of the specimen, W is weight loss in grams, and d is the metal density in g/cm^3 . 3 M NaCl was used as a corroding reagent. Before the analysis, the different composites coated films were weighed precisely to an accuracy of three decimal places. The specimens were immersed in the corrosive environment for 5 h, after which the corroded composite thin films were removed from the corroding reagent and then washed with distilled water. The weight of the corroded composite thin films was measured to an accuracy of three decimal places. The weight loss results presented in Table 1 revealed similar behavior of electrochemical corrosion pattern to the first corrosion rate experiment.

A schematic representation of the corroded surface of a composite film onto a steel substrate is presented in Figure 1c. The corroded structures created by the corrosion process were due to the flow of current from the anode to the cathode through ionic conductivity and from cathode to anode by the chitosan complex structure through electric conductivity [68,69]. Zn metal oxidation occurred at the anode whereas the chitosan complex hydrogen or oxygen reduction occurred at the cathode, which stimulated localized corrosion over the surface [69]. The following mechanism was derived from the observed trend of corrosion behavior of the chitosan- $Zn_{1-x}Cu_xO$ composite thin films: strong oxide bonding formation between Zn and chitosan; stable surface structure of the nano-slab-like morphology; and higher thickness, lower conductivity, and higher bulk resistance of the chitosan-ZnO complex compared to the other composites [48,69].

3. Materials and Methods

The electrochemical preparation route used to synthesize the chitosan- $Zn_{1-x}Cu_xO$ composite thin films by galvanostatic mode. In this work, we used 90% deacetylated chitosan (molecular weight 90 Da) for preparation of the composite thin films. The other precursors used to prepare the electrolytic bath solutions were $ZnCl_2$, $CuCl_2$, and acetic acid. In this film deposition method, mild steel, zinc, and SCE were used as the working, counter, and reference electrodes, respectively.

The cathodic substrate was etched by polishing mechanically to obtain a smooth surface, degreased with trichloroethylene and acid to remove impurities, and then cleaned using de-ionized water and acetone solvent. The anodic material was cleaned using nitric acid solution and acetone. The solution bath was adjusted from pH 1–3 using an acid, the deposition current value was fixed

at 2 mA/cm², and the deposition time was fixed at 30 min. All the compounds were mixed as a homogeneous solution in a 100 mL beaker using a mechanical shaker.

Chitosan (0.8 g) was dissolved in 1% acetic acid solution and then used as a bath precursor for electrosynthesis. For chitosan-ZnO nanocomposite preparation, 0.5 M ZnCl₂ solution was mixed with as-prepared chitosan solution in an electrolytic bath. The chitosan-Zn composite thin film was deposited using the aforementioned electrodeposition parameter values. For the chitosan-CuO composite, 0.5 M CuCl₂ solution was combined with as-prepared chitosan solution in an electrolytic bath and prepared using the same deposition parameters. Using these and the same as-prepared chitosan solution, the electroplated chitosan-Zn_{0.6}Cu_{0.4}O composite film was prepared with 0.5 M ZnCl₂ and 0.2 M CuCl₂, and the chitosan-Zn_{0.3}Cu_{0.7}O composite film was made using 0.1 M ZnCl₂ and 0.3 M CuCl₂.

The prepared composite thin films were characterized using the following instruments to analyze their properties. The structural properties of prepared composite thin films identified using an X-ray diffractometer (X'Pert PRO PANalytical diffractometer) with Cu K_α radiation ($\lambda = 0.15406$ nm) and a scanning rate of 0.01°/step in the 2 θ range of 10° to 80°. The surface morphology and compositional analysis of the composite films carried out using EDAX attached to a scanning electron microscope (Hitachi-S3000, Tokyo, Japan) to determine their size, shape, and composition. A three-electrode cell system consisting of a nanocomposite-coated mild steel specimen as a working electrode, SCE as a reference electrode, and platinum (Pt) as a counter electrode was used for the electrochemical measurements. The corrosion behavior of nanocomposite-coated mild steel specimens were evaluated in 3.0% NaCl solution with a potential sweep rate between -0.5 and -0.6 V vs. SCE. For comparison purposes, corrosion testing was performed using the ASTM weight loss method at room temperature (~ 27 °C). The impedance analysis was carried out using an Autolab BSTR 10A instrument (Metrohm Autolab B.V., Utrecht, The Netherlands). AC signal with amplitude of 50 mV and frequency range from 0.05 to 10⁵ HZ were used to study the performance of the thin films. CV measurements were carried out in 1M HCl using a CHI 1022 electrochemical analyzer/workstation (CH Instruments, Bee Cave, TX, USA).

4. Conclusions

Composite thin films of chitosan-ZnO, chitosan-Zn_{0.6}Cu_{0.4}O, chitosan-Zn_{0.3}Cu_{0.7}O, and chitosan-CuO were prepared by electrosynthesis in galvanostatic mode. The prepared composite thin films' structural, morphological, compositional, corrosion resistance, and impedance properties were plausibly studied using XRD, SEM, EDAX, Tafel polarization, and impedance spectroscopy, respectively. Dual-phase nature was observed for the chitosan-Zn_{1-x}Cu_xO composite thin films. The morphological properties of the chitosan-Zn_{1-x}Cu_xO thin films were enormously varied by precursor concentration in an electrolytic bath. Nano slabs and spherical-shaped grains were observed in the SEM micrographs of the composites. EDAX spectra revealed the atomic percentage values of electrosynthesized chitosan composite thin films. The best corrosion resistance performance was evident in the chitosan-ZnO thin film compared with other composite thin films, and it could be a potential material for applications requiring corrosion resistance.

Acknowledgments: This work was supported by the Korea Institute of Energy Technology Evaluation and Planning (KETEP) and the Ministry of Trade, Industry & Energy (MOTIE) of the Republic of Korea (No. 20174030201520), and the research program of Dongguk University in 2017.

Author Contributions: A.S. initiated the study, performed the extensive experiments related to the growth of the samples, and wrote the paper with the assistance of the co-authors. K.K. and J.Y.L. helped with the experimental work and data analyses. D.V. and H.-S.K.'s participation included planning and designing the experimental work and discussion. All of the authors read and approved the final manuscript.

Conflicts of Interest: The authors declare no conflicts of interest.

References

1. Beecroft, L.L.; Ober, C.K. Nanocomposite materials for optical applications. *Chem. Mater.* **1997**, *9*, 1302–1317. [[CrossRef](#)]
2. Sanchez, C.; Soler-Illia, G.J.D.A.A.; Ribot, F.; Lalot, T.; Mayer, C.R.; Cabuil, V. Designed hybrid organic–inorganic nanocomposites from functional nanobuilding blocks. *Chem. Mater.* **2001**, *13*, 3061–3083. [[CrossRef](#)]
3. Gangopadhyay, R.; De, A. Conducting polymer nanocomposites: A brief overview. *Chem. Mater.* **2000**, *12*, 608–622. [[CrossRef](#)]
4. Mitzi, D.B. Thin-film deposition of organic-inorganic hybrid materials. *Chem. Mater.* **2001**, *13*, 3283–3298. [[CrossRef](#)]
5. Merachtsaki, D.; Xidas, P.; Giannakoudakis, P.; Triantafyllidis, K.; Spathis, P. Corrosion protection of steel by epoxy-organoclay nanocomposite coatings. *Coatings* **2017**, *7*, 84. [[CrossRef](#)]
6. Zhitomirsky, I. Electrophoretic deposition of organic–inorganic nanocomposites. *J. Mater. Sci.* **2006**, *41*, 8186–8195. [[CrossRef](#)]
7. Mirak, M.; Alizadeh, M.; Ghaffari, M.; Ashtiani, M.N. Characterization, mechanical properties and corrosion resistance of biocompatible Zn-HA/TiO₂ nanocomposite coatings. *J. Mech. Behav. Biomed. Mater.* **2016**, *62*, 282–290. [[CrossRef](#)] [[PubMed](#)]
8. Pang, X.; Zhitomirsky, I. Electrodeposition of composite hydroxyapatite–chitosan films. *Mater. Chem. Phys.* **2005**, *94*, 245–251. [[CrossRef](#)]
9. Dhanasekaran, V.; Mahalingam, T.; Chandramohan, R. Morphology selection for cupric oxide thin films by electrodeposition. *Microsc. Res. Tech.* **2011**, *74*, 980–983. [[CrossRef](#)] [[PubMed](#)]
10. Boccaccini, A.R.; Zhitomirsky, I. Application of electrophoretic and electrolytic deposition techniques in ceramics processing. *Curr. Opin. Solid State Mater. Sci.* **2002**, *6*, 251–260. [[CrossRef](#)]
11. Fernandes, R.; Wu, L.-Q.; Chen, T.; Yi, H.; Rubloff, G.W.; Ghodssi, R.; Bentley, W.E.; Payne, G.F. Electrochemically induced deposition of a polysaccharide hydrogel onto a patterned surface. *Langmuir* **2003**, *19*, 4058–4062. [[CrossRef](#)]
12. Yi, H.; Wu, L.-Q.; Bentley, W.E.; Ghodssi, R.; Rubloff, G.W.; Culver, J.N.; Payne, G.F. Biofabrication with chitosan. *Biomacromolecules* **2005**, *6*, 2881–2894. [[CrossRef](#)] [[PubMed](#)]
13. Cheong, M.; Zhitomirsky, I. Electrodeposition of alginic acid and composite films. *Colloids Surf. A* **2008**, *328*, 73–78. [[CrossRef](#)]
14. Gorelikov, I.; Kumacheva, E. Electrodeposition of polymer-semiconductor nanocomposite films. *Chem. Mater.* **2004**, *16*, 4122–4127. [[CrossRef](#)]
15. Grandfield, K.; Sun, F.; FitzPatrick, M.; Cheong, M.; Zhitomirsky, I. Electrophoretic deposition of polymer-carbon nanotube–hydroxyapatite composites. *Surf. Coat. Technol.* **2009**, *203*, 1481–1487. [[CrossRef](#)]
16. Lee, J.-H.; Leu, C.; Chung, Y.-W.; Hon, M.-H. Fabrication of ordered ZnO hierarchical structures controlled via surface charge in the electrophoretic deposition process. *Nanotechnology* **2006**, *17*, 4445. [[CrossRef](#)]
17. Park, J.J.; Luo, X.; Yi, H.; Valentine, T.M.; Payne, G.F.; Bentley, W.E.; Ghodssi, R.; Rubloff, G.W. Chitosan-mediated in situ biomolecule assembly in completely packaged microfluidic devices. *Lab Chip* **2006**, *6*, 1315–1321. [[CrossRef](#)] [[PubMed](#)]
18. Sun, F.; Zhitomirsky, I. Electrodeposition of hyaluronic acid and composite films. *Surf. Eng.* **2013**, *25*, 621–627. [[CrossRef](#)]
19. Wu, L.-Q.; Payne, G.F. Biofabrication: Using biological materials and biocatalysts to construct nanostructured assemblies. *Trends Biotechnol.* **2004**, *22*, 593–599. [[CrossRef](#)] [[PubMed](#)]
20. Anandhavelu, S.; Dhanasekaran, V.; Sethuraman, V.; Park, H.J. Chitin and chitosan based hybrid nanocomposites for super capacitor applications. *J. Nanosci. Nanotechnol.* **2017**, *17*, 1321–1328. [[CrossRef](#)]
21. Sanmugam, A.; Vikraman, D.; Venkatesan, S.; Park, H.J. Optical and structural properties of solvent free synthesized starch/chitosan-ZnO nanocomposites. *J. Nanomater.* **2017**, *2017*, 7536364. [[CrossRef](#)]
22. Li, Y.; Wu, K.; Zhitomirsky, I. Electrodeposition of composite zinc oxide–chitosan films. *Colloids Surf. A* **2010**, *356*, 63–70. [[CrossRef](#)]
23. Gu, B.; Xu, C.; Zhu, G.; Liu, S.; Chen, L.; Wang, M.; Zhu, J. Layer by layer immobilized horseradish peroxidase on zinc oxide nanorods for biosensing. *J. Phys. Chem. B* **2009**, *113*, 6553–6557. [[CrossRef](#)] [[PubMed](#)]

24. Khan, R.; Kaushik, A.; Solanki, P.R.; Ansari, A.A.; Pandey, M.K.; Malhotra, B. Zinc oxide nanoparticles-chitosan composite film for cholesterol biosensor. *Anal. Chim. Acta* **2008**, *616*, 207–213. [[CrossRef](#)] [[PubMed](#)]
25. Solanki, P.R.; Kaushik, A.; Ansari, A.A.; Sumana, G.; Malhotra, B. Zinc oxide-chitosan nanobiocomposite for urea sensor. *Appl. Phys. Lett.* **2008**, *93*, 163903. [[CrossRef](#)]
26. Wang, L.; Li, D.-B.; Li, K.; Chen, C.; Deng, H.-X.; Gao, L.; Zhao, Y.; Jiang, F.; Li, L.; Huang, F.; et al. Stable 6%-efficient Sb₂Se₃ solar cells with a ZnO buffer layer. *Nat. Energy* **2017**, *2*, 17046. [[CrossRef](#)]
27. Gholizadeh, A.; Reyhani, A.; Parvin, P.; Mortazavi, S.Z. Efficiency enhancement of ZnO nanostructure assisted Si solar cell based on fill factor enlargement and UV-blue spectral down-shifting. *J. Phys. D Appl. Phys.* **2017**, *50*, 185501. [[CrossRef](#)]
28. Yu, X.; Ma, J.; Ji, F.; Wang, Y.; Zhang, X.; Ma, H. Influence of annealing on the properties of ZnO:Ga films prepared by radio frequency magnetron sputtering. *Thin Solid Films* **2005**, *483*, 296–300. [[CrossRef](#)]
29. Hu, J.; Gordon, R.G. Atmospheric pressure chemical vapor deposition of gallium doped zinc oxide thin films from diethyl zinc, water, and triethyl gallium. *J. Appl. Phys.* **1992**, *72*, 5381–5392. [[CrossRef](#)]
30. Kotlyarchuk, B.; Savchuk, V.; Oszwaldowski, M. Preparation of undoped and indium doped ZnO thin films by pulsed laser deposition method. *Cryst. Res. Technol.* **2005**, *40*, 1118–1123. [[CrossRef](#)]
31. Omri, K.; Najeh, I.; Dhahri, R.; El Ghoul, J.; El Mir, L. Effects of temperature on the optical and electrical properties of ZnO nanoparticles synthesized by sol-gel method. *Microelectron. Eng.* **2014**, *128*, 53–58. [[CrossRef](#)]
32. Chandramohan, R.; Vijayan, T.A.; Arumugam, S.; Ramalingam, H.B.; Dhanasekaran, V.; Sundaram, K.; Mahalingam, T. Effect of heat treatment on microstructural and optical properties of CBD grown Al-doped ZnO thin films. *Mater. Sci. Eng. B Solid* **2011**, *176*, 152–156. [[CrossRef](#)]
33. Ben Belgacem, R.; Chaari, M.; Braña, A.F.; Garcia, B.J.; Matoussi, A. Structural, electric modulus and complex impedance analysis of ZnO/TiO₂ composite ceramics. *J. Am. Chem. Soc.* **2017**, *100*, 2045–2058. [[CrossRef](#)]
34. Dai, L.; Chen, X.L.; Wang, W.J.; Zhou, T.; Hu, B.Q. Growth and luminescence characterization of large-scale zinc oxide nanowires. *J. Phys. Condens. Matter.* **2003**, *15*, 2221–2226. [[CrossRef](#)]
35. Tuomisto, F.; Saarinen, K.; Grasza, K.; Mycielski, A. Observation of Zn vacancies in ZnO grown by chemical vapor transport. *Phys. Status Solidi B* **2006**, *243*, 794–798. [[CrossRef](#)]
36. Janotti, A.; Walle, C.G.V.D. Oxygen vacancies in ZnO. *Appl. Phys. Lett.* **2005**, *87*, 122102. [[CrossRef](#)]
37. Look, D.C.; Hemsley, J.W.; Sizelove, J.R. Residual native shallow donor in ZnO. *Phys. Rev. Lett.* **1999**, *82*, 2552–2555. [[CrossRef](#)]
38. Wang, M.; Ren, F.; Zhou, J.; Cai, G.; Cai, L.; Hu, Y.; Wang, D.; Liu, Y.; Guo, L.; Shen, S. N doping to ZnO nanorods for photoelectrochemical water splitting under visible light: Engineered impurity distribution and terraced band structure. *Sci. Rep.* **2015**, *5*, 12925. [[CrossRef](#)] [[PubMed](#)]
39. Fathollahi, V.; Amini, M.M. Sol-gel preparation of highly oriented gallium-doped zinc oxide thin films. *Mater. Lett.* **2001**, *50*, 235–239. [[CrossRef](#)]
40. Kılınc, N.; Arda, L.; Öztürk, S.; Öztürk, Z.Z. Structure and electrical properties of mg-doped ZnO nanoparticles. *Cryst. Res. Technol.* **2010**, *45*, 529–538. [[CrossRef](#)]
41. Pawar, S.M.; Kim, J.; Inamdar, A.I.; Woo, H.; Jo, Y.; Pawar, B.S.; Cho, S.; Kim, H.; Im, H. Multi-functional reactively-sputtered copper oxide electrodes for supercapacitor and electro-catalyst in direct methanol fuel cell applications. *Sci. Rep.* **2016**, *6*, 21310. [[CrossRef](#)] [[PubMed](#)]
42. Vikraman, D.; Park, H.J.; Kim, S.-I.; Thaiyan, M. Magnetic, structural and optical behavior of cupric oxide layers for solar cells. *J. Alloys Compd.* **2016**, *686*, 616–627. [[CrossRef](#)]
43. Chandrappa, K.G.; Venkatesha, T.V. Generation of nanostructured CuO by electrochemical method and its Zn-Ni-CuO composite thin films for corrosion protection. *Mater. Corros.* **2013**, *64*, 831–839. [[CrossRef](#)]
44. Momeni, M.M.; Nazari, Z.; Kazempour, A.; Hakimiyan, M.; Mirhoseini, S.M. Preparation of CuO nanostructures coating on copper as supercapacitor materials. *Surf. Eng.* **2014**, *30*, 775–778. [[CrossRef](#)]
45. Bu, I.Y.Y.; Huang, R. Fabrication of CuO-decorated reduced graphene oxide nanosheets for supercapacitor applications. *Ceram. Int.* **2017**, *43*, 45–50. [[CrossRef](#)]
46. Gholivand, M.B.; Heydari, H.; Abdolmaleki, A.; Hosseini, H. Nanostructured CuO/PANI composite as supercapacitor electrode material. *Mater. Sci. Semicond. Process.* **2015**, *30*, 157–161. [[CrossRef](#)]
47. Arena, A.; Scandurra, G.; Ciofi, C. Copper oxide chitosan nanocomposite: Characterization and application in non-enzymatic hydrogen peroxide sensing. *Sensors* **2017**, *17*, 2198. [[CrossRef](#)] [[PubMed](#)]

48. Dai, N.; Zhang, L.-C.; Zhang, J.; Chen, Q.; Wu, M. Corrosion behavior of selective laser melted Ti-6Al-4V alloy in NaCl solution. *Corros. Sci.* **2016**, *102*, 484–489. [[CrossRef](#)]
49. Dai, N.; Zhang, L.-C.; Zhang, J.; Zhang, X.; Ni, Q.; Chen, Y.; Wu, M.; Yang, C. Distinction in corrosion resistance of selective laser melted Ti-6Al-4V alloy on different planes. *Corros. Sci.* **2016**, *111*, 703–710. [[CrossRef](#)]
50. Chen, Y.; Zhang, J.; Dai, N.; Qin, P.; Attar, H.; Zhang, L.-C. Corrosion behaviour of selective laser melted Ti-TiB biocomposite in simulated body fluid. *Electrochim. Acta* **2017**, *232*, 89–97. [[CrossRef](#)]
51. Lu, H.-B.; Li, Y.; Wang, F.-H. Dealloying behaviour of Cu-20Zr alloy in hydrochloric acid solution. *Corros. Sci.* **2006**, *48*, 2106–2119. [[CrossRef](#)]
52. Lin, S.; Shih, H.; Mansfeld, F. Corrosion protection of aluminum alloys and metal matrix composites by polymer coatings. *Corros. Sci.* **1992**, *33*, 1331–1349. [[CrossRef](#)]
53. Singh, B.P.; Jena, B.K.; Bhattacharjee, S.; Besra, L. Development of oxidation and corrosion resistance hydrophobic graphene oxide-polymer composite coating on copper. *Surf. Coat. Technol.* **2013**, *232*, 475–481. [[CrossRef](#)]
54. Miao, J.; Zhang, L.-C.; Lin, H. A novel kind of thin film composite nanofiltration membrane with sulfated chitosan as the active layer material. *Chem. Eng. Sci.* **2013**, *87*, 152–159. [[CrossRef](#)]
55. Miao, J.; Lin, H.; Wang, W.; Zhang, L.-C. Amphoteric composite membranes for nanofiltration prepared from sulfated chitosan crosslinked with hexamethylene diisocyanate. *Chem. Eng. J.* **2013**, *234*, 132–139. [[CrossRef](#)]
56. Mahalingam, T.; Dhanasekaran, V.; Chandramohan, R.; Rhee, J.-K. Microstructural properties of electrochemically synthesized ZnSe thin films. *J. Mater. Sci.* **2012**, *47*, 1950–1957. [[CrossRef](#)]
57. Anandhavelu, S.; Thambidurai, S. Effect of zinc chloride and sodium hydroxide concentration on the optical property of chitosan-ZnO nanostructure prepared in chitin deacetylation. *Mater. Chem. Phys.* **2011**, *131*, 449–454. [[CrossRef](#)]
58. Kiriakidou, F.; Kondarides, D.I.; Verykios, X.E. The effect of operational parameters and TiO₂-doping on the photocatalytic degradation of azo-dyes. *Catal. Today* **1999**, *54*, 119–130. [[CrossRef](#)]
59. Poullos, I.; Tsachpinis, I. Photodegradation of the textile dye reactive black 5 in the presence of semiconducting oxides. *J. Chem. Technol. Biotechnol.* **1999**, *74*, 349–357. [[CrossRef](#)]
60. Liu, G.; Wu, T.; Zhao, J.; Hidaka, H.; Serpone, N. Photoassisted degradation of dye pollutants. 8. Irreversible degradation of alizarin red under visible light radiation in air-equilibrated aqueous TiO₂ dispersions. *Environ. Sci. Technol.* **1999**, *33*, 2081–2087. [[CrossRef](#)]
61. Egelhaaf, H.-J.; Oelkrug, D. Luminescence and nonradiative deactivation of excited states involving oxygen defect centers in polycrystalline ZnO. *J. Cryst. Growth* **1996**, *161*, 190–194. [[CrossRef](#)]
62. Chaari, M.; Ben Belgacem, R.; Matoussi, A. Impedance analysis, dielectric relaxation and modulus behaviour of ZnO-Sn₂O₃ ceramics. *J. Alloys Compd.* **2017**, *726*, 49–56. [[CrossRef](#)]
63. Lee, L.; Kao, R.-H.; Yang, C.-S.; Ku, C.-S.; Lee, H.-Y. An insulating-conductive transition driven by partial crystallization of amorphous Zn-Sn-O alloy. *J. Alloys Compd.* **2016**, *672*, 636–642. [[CrossRef](#)]
64. Wan, Y.; Wang, X.; Sun, H.; Li, Y.; Zhang, K.; Wu, Y. Corrosion behavior of copper at elevated temperature. *Int. J. Electrochem. Sci.* **2012**, *7*, 7902–7914.
65. Hosseini, M.; Bagheri, R.; Najjar, R. Electropolymerization of polypyrrole and polypyrrole-ZnO nanocomposites on mild steel and its corrosion protection performance. *J. Appl. Polym. Sci.* **2011**, *121*, 3159–3166. [[CrossRef](#)]
66. Aramaki, K.; Hagiwara, M.; Nishihara, H. The synergistic effect of anions and the ammonium cation on the inhibition of iron corrosion in acid solution. *Corros. Sci.* **1987**, *27*, 487–497. [[CrossRef](#)]
67. Hihara, L.; Latanision, R. Suppressing galvanic corrosion in graphite/aluminum metal-matrix composites. *Corros. Sci.* **1993**, *34*, 655–665. [[CrossRef](#)]
68. Brichau, F.; Deconinck, J. A numerical model for cathodic protection of buried pipes. *Corrosion* **1994**, *50*, 39–49. [[CrossRef](#)]
69. Metwally, I.; Al-Mandhari, H.; Gastli, A.; Nadir, Z. Factors affecting cathodic-protection interference. *Eng. Anal. Bound. Elem.* **2007**, *31*, 485–493. [[CrossRef](#)]

