

Article

Tuning of Magnetic Hyperthermia Response in the Systems Containing Magnetosomes

Matus Molcan ¹, Andrzej Skumiel ², Milan Timko ¹, Ivo Safarik ^{3,4}, Kristina Zolochovska ¹
and Peter Kopcansky ^{1,*}

- ¹ Institute of Experimental Physics, Slovak Academy of Sciences, Watsonova 47, 04001 Kosice, Slovakia
² Faculty of Physics, Adam Mickiewicz University, Uniwersytetu Poznańskiego 2, 61-614 Poznań, Poland
³ Department of Nanobiotechnology, Biology Centre, ISBB, CAS, Na Sadkach 7, 370 05 Ceske Budejovice, Czech Republic
⁴ Regional Centre of Advanced Technologies and Materials, Czech Advanced Technology and Research Institute, Palacky University, Slechtitelu 27, 783 71 Olomouc, Czech Republic
* Correspondence: kopcans@saske.sk

Abstract: A number of materials are studied in the field of magnetic hyperthermia. In general, the most promising ones appear to be iron oxide particle nanosystems. This is also indicated in some clinical trial studies where iron-based oxides were used. On the other hand, the type of material itself provides a number of variations on how to tune hyperthermia indicators. In this paper, magnetite nanoparticles in various forms were analyzed. The nanoparticles differed in the core size as well as in the form of their arrangement. The arrangement was determined by the nature of the surfactant. The individual particles were covered chemically by dextran; in the case of chain-like particles, they were encapsulated naturally in a lipid bilayer. It was shown that in the case of chain-like nanoparticles, except for relaxation, a contribution from magnetic hysteresis to the heating process also appears. The influence of the chosen methodology of magnetic field generation was also analyzed. In addition, the influence of the chosen methodology of magnetic field generation was analyzed. The application of a rotating magnetic field was shown to be more efficient in generating heat than the application of an alternating magnetic field. However, the degree of efficiency depended on the arrangement of the magnetite nanoparticles. The difference in the efficiency of the rotating magnetic field versus the alternating magnetic field was much more pronounced for individual nanoparticles (in the form of a magnetic fluid) than for systems containing chain nanoparticles (magnetosomes and a mix of magnetic fluid with magnetosomes in a ratio 1:1).

Keywords: alternating magnetic field; rotating magnetic field; magnetic nanoparticles; magnetic hyperthermia; heat evolution



Citation: Molcan, M.; Skumiel, A.; Timko, M.; Safarik, I.; Zolochovska, K.; Kopcansky, P. Tuning of Magnetic Hyperthermia Response in the Systems Containing Magnetosomes. *Molecules* **2022**, *27*, 5605. <https://doi.org/10.3390/molecules27175605>

Academic Editor: Vasyly M. Haramus

Received: 26 July 2022

Accepted: 29 August 2022

Published: 31 August 2022

Publisher's Note: MDPI stays neutral with regard to jurisdictional claims in published maps and institutional affiliations.



Copyright: © 2022 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (<https://creativecommons.org/licenses/by/4.0/>).

1. Introduction

Iron-based magnetic nanoparticles are often investigated especially in the field of bioresearch. In particular, increasing attention is being paid to research on magnetic hyperthermia. The principle follows from the magnetic nanoparticles' ability to induce local heating under the influence of various types of applied alternating magnetic fields (AMF) [1]. It is extremely difficult to determine the ideal procedures for the preparation of a sample for hyperthermia as well as the method of its characterization. It is necessary to take into account a number of parameters, such as material, equipment, and the selection of a suitable measuring or application methodology. Hyperthermic indicators such as the heating rate or maximum power output are affected by particle concentration, particle size and shape, type of surfactant, viscosity of the surrounding medium, etc. [2]. The concentration of magnetic nanoparticles (NPs) has a great effect on hyperthermia properties, as shown by Fe₃O₄ NPs at various concentrations when treated in an alternative magnetic field; it was observed that the ΔT sharply increases with increasing the NPs concentration

while the specific absorption rate (SAR) remains almost constant [3,4]. Moreover, the presence of interparticle interactions can influence the hyperthermia properties of magnetic NPs in an AC magnetic field [5]. Surface coating of magnetic nanoparticles can have a substantial effect on the magnetothermal properties, as shown in the case of CoFe_2O_4 magnetic nanoparticles modified by triethylene glycol (TEG) coating, which caused increased NPs saturation magnetization as measured by SQUID. It was shown that TEG coating increases the heating efficiency of the CoFe_2O_4 magnetic NPs due to an increase in saturation magnetization and a decrease in the strength of the magnetic interactions between the coated nanoparticles [6]. In addition, the bimagnetic NPs can exhibit more interesting properties than single-phase NPs, as shown in the case of cube-like bimagnetic hard/soft ($\text{CoFe}_2\text{O}_4/\text{Fe}_3\text{O}_4$) and soft/hard ($\text{Fe}_3\text{O}_4/\text{CoFe}_2\text{O}_4$) nanocomposites (core/coating) with an average dimension of 20 nm. The $\text{CoFe}_2\text{O}_4/\text{Fe}_3\text{O}_4$ nanocomposites presented a larger saturation magnetization than the CoFe_2O_4 NPs, which is effective for their potential use in magnetic hyperthermia [7]. In general, the basic requirement is to achieve maximum heating of a unit amount of magnetic substance per unit time under the conditions of the applied magnetic field at a given frequency. This efficiency is known as the Specific Absorption Rate (SAR) [8]. Of course, input parameters such as frequency and intensity of the applied magnetic field have a significant effect on the heating effect [9]. On the other hand, the limitations/limits of magnetic hyperthermia must not be forgotten in order to avoid unwanted and dangerous patient discomfort [10]. The most well-known criterion is the so-called Brezovich's criterion [11], stating that the AMF is harmless to the human body if its amplitude H_0 and frequency f satisfy the condition $f \cdot H_0 < 4.85 \times 10^8 \text{ A} \cdot \text{m}^{-1} \cdot \text{s}^{-1}$. Apart from Brezovich's criterion, others can also be found in the literature. Hergt et al. [12] suggested a less rigid criterion where $f \cdot H = 5 \times 10^9 \text{ A} \cdot \text{m}^{-1} \cdot \text{s}^{-1}$. When examining the heating effect, we must also take into account the method of generating the magnetic field, which can significantly affect the efficiency [13]. The most commonly used method in the field of magnetic hyperthermia is to generate an AMF. The mechanisms underlying the production of heat for nanoparticles include the Néel mode (rotation of the magnetic moment of the magnetic nanoparticles) and rotational friction provided by the viscous drag of the suspending medium as the nanoparticles align with the magnetic field [14]. However, studies appear where the possibility of increasing efficiency by means of a so-called rotating magnetic field (RMF) exists [15–19]. The RMF is AMF, which creates a resultant field during the superposition of two or more AMFs of identical frequency but spatially displaced in phase with respect to each other [19]. The motivation of this work was to analyze the heating effect of magnetite nanoparticles of different configurations by two approaches. Both AMF and RMP were used to heat them. The response of particles in the form of stable colloid of individual particles (magnetic fluid), in the form of chains (magnetosomes), as well as a mix of colloidal systems of magnetic fluid and magnetosomes, was monitored.

2. Materials and Methods

The magnetic hyperthermia effect was studied at three types of magnetite colloids of different characters. The first sample represented the individual magnetite nanoparticles in the form of dextran stabilized magnetic fluid. The second sample was magnetite magnetosome nanoparticles in the chain-like structure form, and the last sample represented the mix of colloidal systems of magnetic fluid and magnetosome chains. For better illustration, magnetosomes are chain-like structures of a single domain, single crystal magnetic nanoparticles of magnetite, Fe_3O_4 or greigite, Fe_3S_4 (depending on the species of bacteria), enclosed and connected by a lipid bilayer membrane [20]. The magnetosome chains are formed by the biomineralization process [21] in magnetotactic bacteria.

In order to determine the hydrodynamic size of the prepared samples, the dynamic light scattering measurements (DLS) were carried out on a Malvern Zetasizer Nano ZS (Malvern, UK) as well as magnetization measurements (VSM magnetometer—Cryogenic Ltd., London, UK) to determine core size distribution function.

2.1. Samples Description

2.1.1. Individual NPs (Magnetic Fluid)

Dextran stabilized Fe_3O_4 magnetic nanoparticles were synthesized using the modified Molday procedure [22]. The dextran (70 kDa, Sigma Aldrich; 19 g) was dissolved in water (75 mL), while $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$ (5 g) and $\text{FeCl}_2 \cdot 4\text{H}_2\text{O}$ (2.1 g) were dissolved in 2 M HCl (13 mL). The solutions were mixed and placed in a water bath (60 °C). Subsequently, under mixing, 7.5% ammonium hydroxide (75 mL) was added dropwise, and the mixing continued for another 15 min at 60 °C. The next day the magnetic fluid was centrifuged (5000 rpm, 45 min; Universal 320, Hettich Zentrifugen, Germany) [23].

2.1.2. Chain-Like NPs (Magnetosomes)

Magnetosomes as a product of *AMB-1* magnetotactic bacteria were prepared under laboratory conditions. The culture conditions and magnetosome extraction recipe were described in detail recently [24]. Extracted magnetosome nanoparticles were dispersed in the HEPES buffer.

2.1.3. Colloidal Systems of Magnetic Fluid and Magnetosomes

The sample was prepared by mixing the magnetic fluid and the magnetosome suspension in a ratio of 1:1, (1 mL of magnetosomes suspension, $\varphi_v = 0.019\%$ + 1 mL of Dextran FF, $\varphi_v = 0.29\%$).

2.2. Apparatus for Generating a Magnetic Field

2.2.1. Rotating Magnetic Field

The test samples were exposed to both rotating and oscillating high-frequency magnetic fields. Due to the low concentration of nanoparticles in the samples (from $\varphi_v = 0.019\%$ for magnetosomes to 0.29% for dextran magnetic fluid) and thus due to the expected low efficiency of the calorimetric effect, systems were developed to record and measure temperature changes during operation magnetic field (Figure 1). In the case of RMF, a modernized version of the system described in more detail in our previous articles was used [17,18]. The main change to the circuit was the use of three transformers with ferrite cores to raise the voltage supplying parallel resonant circuits (L_P coils and C_P capacitors). The tested sample was located in the central part of the magnetic circuit, where three magnetic fluxes shifted in phase and space by 120 angular degrees and were flowing successively. Due to the superposition, a rotating magnetic field was created. In the case of AMF, the magnetic field was linearly polarized.

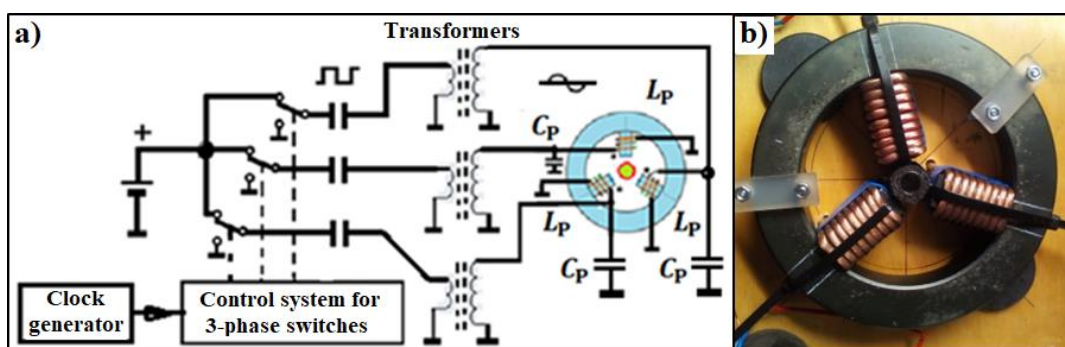


Figure 1. Diagram of a system for generating of rotating magnetic field of high frequency (a) and elements of the magnetic circuit: ferrite torus, 3 coils on ferrite cores (b).

2.2.2. Alternating Magnetic Field

For AMF generation, a double-layer coil (Figure 2) with self-inductance $L = 29.6 \mu\text{H}$ was used, which was connected in parallel with a polypropylene capacitor with an electric capacity of $C = 64$ or 54 nF to obtain the frequency $f = 115$ or 126 kHz . The first layer of

the coil was a thin-walled copper tube, and the second layer was a copper wire consisting of many wires with a total area of $S_{Cu} = 2 \text{ mm}^2$. The electric voltage across the resistor $R = 0.1 \Omega$ was used to measure the current flowing through the coil. During the calibration of the system (before the actual measurement of the calorimetric effect), a measuring coil (voltage probe) was inserted into the center of the coil. The measurement of the amplitude value of the magnetic field intensity H was based on the registration of the voltage induced in the probe's coil with the surface SC at the frequency f , which allowed one to calculate the value H using the basic laws of electromagnetic induction (i.e., Faraday's law). The parallel LC circuit was powered by the 300 W Power Amplifiers AL-300-HF. Its maximum values of current and voltage amplitudes are $I = 15.1 \text{ [A}_p]$ and $U = 39.6 \text{ [V}_p]$. Thanks to the use of a ferrite transformer on the secondary winding, the voltage amplitude $U = 252 \text{ [V}_p]$ and the corresponding amplitude of the current consumed by the parallel circuit $I = 1.68 \text{ [A}_p]$ were obtained. According to additional research carried out by the authors, in a parallel LC circuit, during resonance, the current of the coil and the capacitor is several dozen times greater than the current supplying this circuit. Both these currents are 180 degrees shifted, and this is the most important advantage of a parallel LC system compared to a series one. In both RMF and AMF systems, the sample temperature was measured with an optical fiber sensor. In experiments, the optical fiber temperature sensor [13] by FISO Technology Inc., model FOT-L-SD, was used with a temperature range of $-40 \text{ }^\circ\text{C}$ to $300 \text{ }^\circ\text{C}$, with a response time better than 1.5 s, accuracy of $0.1 \text{ }^\circ\text{C}$, and resolution of $0.01 \text{ }^\circ\text{C}$.

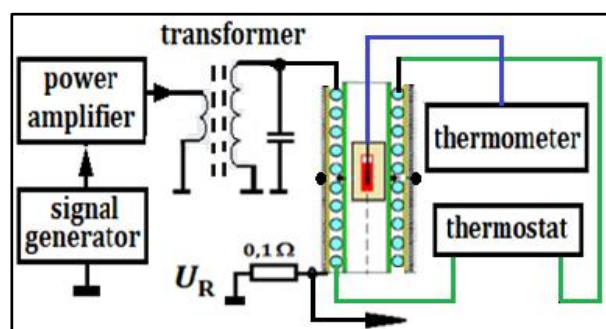


Figure 2. System for generating the alternating magnetic field.

3. Experimental Part

3.1. Nanoparticle Systems Characterization

The hydrodynamic size distribution performed by DLS is shown in Figure 3. In the case of the magnetic fluid sample (black line), we can see a uniform peak. Relatively width peak is a result of dextran coating. The magnetosome sample (green line) exhibits bimodal size distribution (observed before [25]) because of the polydispersity of magnetosome chains. The polydispersity is caused by various numbers of particles per chain [26] and various shapes of isolated chains as a result of their natural ability loops creation after isolation [27]. The mixture of magnetic fluid and magnetosomes (red line) represents a dominant peak (170 nm), which is attributed to signal from the dextran magnetic fluid particles, and a peak with a lower intensity (970 nm) attributed to magnetosomes. These size results point to different sample characteristics that may lead to different responses to the applied magnetic fields.

Magnetization curves under static DC field conditions (presented in Figure 4a) show superparamagnetic behavior. The differences in magnetization saturation are caused by different concentrations of the magnetic component (magnetite) in the analyzed sample. When the scale is changed (Figure 4b), a small coercivity (and hysteresis) in the case of a magnetosome sample can be visible. In order to prove it more clearly, it should be measured with a more sensitive device (for example, SQUID) and with a higher density of measured points. Moreover, such a kind of magnetization is a result of the high shape anisotropy of magnetosomes, and it was experimentally observed also in other works [28,29]. The

dynamic of magnetization increase (magnetic susceptibility slope), which leads to fast saturation, depends on the magnetite concentration but also can be attributed to size distribution. In the case of smaller and more uniform particles (magnetic fluid sample), we can see a faster response on the applied field compared with the mixed sample and magnetosome sample (Figure 4b).

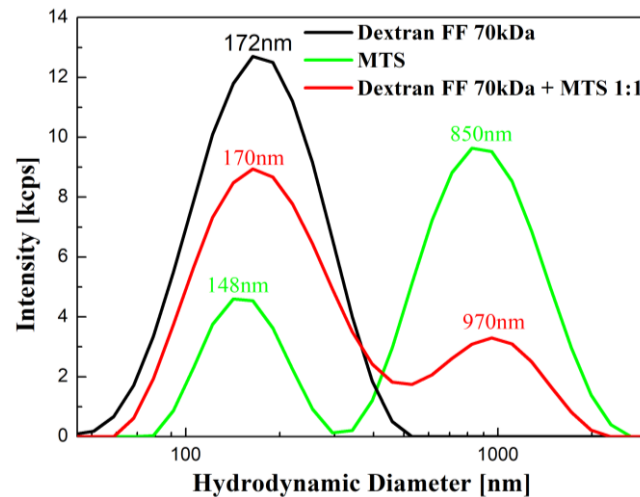


Figure 3. Hydrodynamic diameter distribution of magnetic fluid (dextran-coated Fe_3O_4 nanoparticles), magnetosome chains, and the mixture of magnetic fluid and magnetosomes (ratio 1:1).

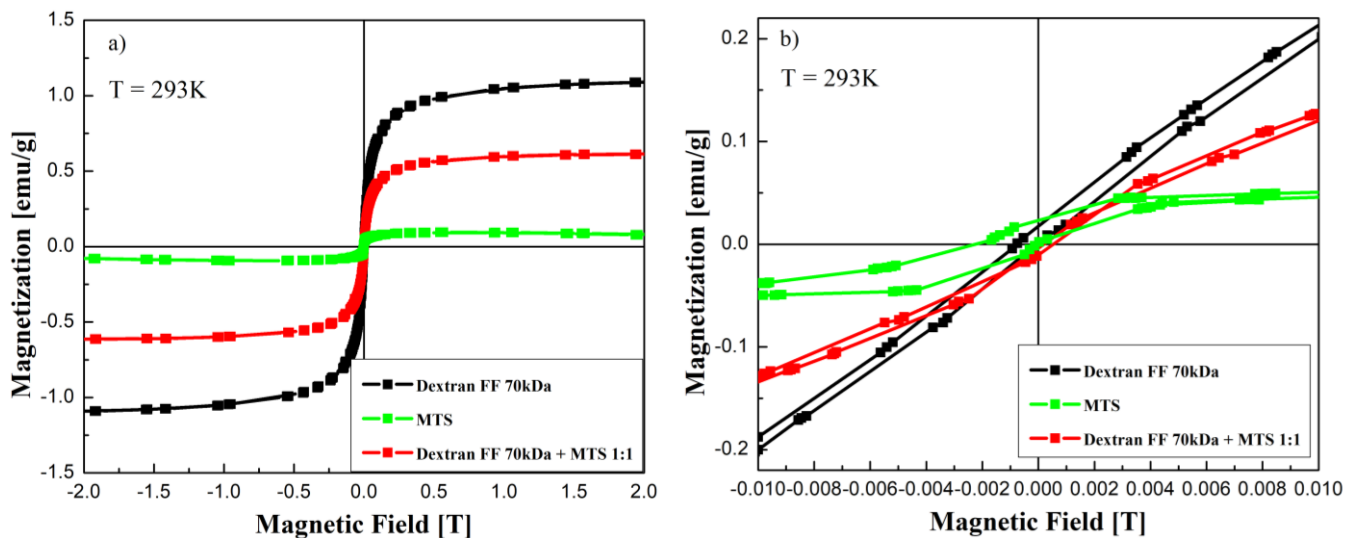


Figure 4. Magnetization versus magnetic field for magnetic fluid, magnetosome chains, and the mixture of magnetic fluid and magnetosomes (ratio 1:1), (a) and a zoom resolution where an indication of non-zero coercivity can be observed in the case of magnetosomes (b).

Additionally, the magnetic core diameter of tested samples was calculated by fitting the initial magnetization curve to the model of magnetization in polydisperse magnetic fluid. The particle size distribution, in this case, is usually described by the log-normal distribution function [30]:

$$f(d) = \frac{1}{\sqrt{2\pi d\beta}} \exp \left[-\frac{\left(\ln \frac{d}{d_0}\right)^2}{2\beta^2} \right] \quad (1)$$

where d is the diameter, d_0 and β are the parameters obtained by fitting this function. The magnetite nanoparticle distribution functions are shown in Figure 5. In order to estimate the mean diameter $\langle d \rangle$ and mean standard deviation $\langle \sigma \rangle$, the following formulas were used [31]: $d = d_0 \exp\left(\frac{\beta^2}{2}\right)$ and $\sigma = d \sqrt{\exp(\beta^2) - 1}$.

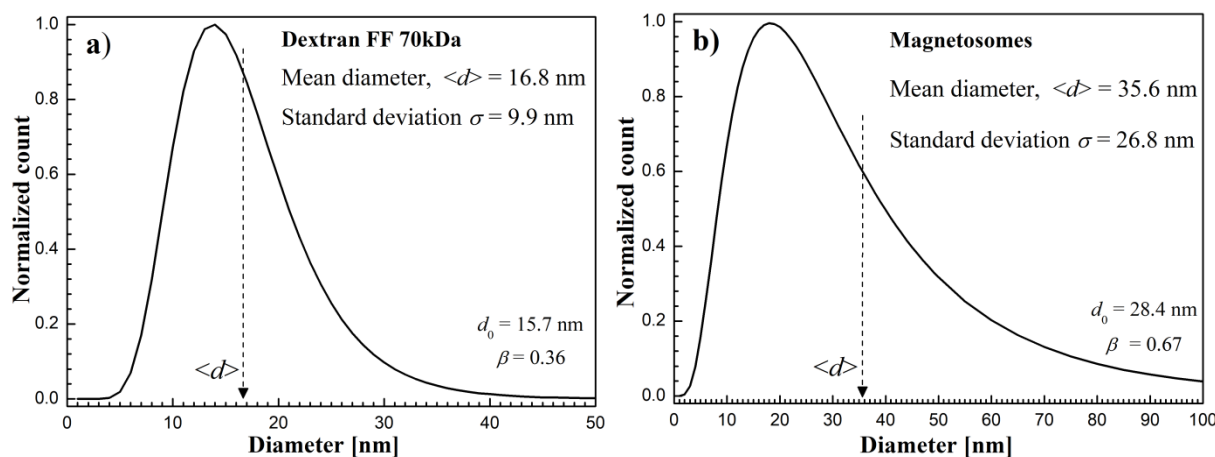


Figure 5. Log-normal particle core size distribution function obtained from VSM data for Dextran FF (a) and magnetosomes sample (b).

In the case of Dextran FF magnetite nanoparticles, the mean size was estimated to be 16.8 nm. The magnetosomes magnetite cores were estimated to be 35.6 nm. The magnetosomes core size is comparable to the study of Gojzewski et al. [32]. The obtained mean size and its standard deviation were 43 ± 12 nm. However, it should be noted that, in this case, it is not individual nanoparticles but particles arranged in chain-like structures. Electron microscopy images of MTS are not presented now, but they can be seen, for example, in the following papers [24,25,32,33].

3.2. Heating Effect Results in RMF

The results of the rotating magnetic field's influence on the tested samples are shown in Figures 6–8. Temperature changes before and after switching on the RMF for subsequent samples are shown in Figures 6a, 7a and 8a. The obtained values of temperature changes proved to be small due to the low concentration of magnetic material in the samples (from 0.019% in the sample of MTS to 0.29% in the sample Dextran FF) and did not exceed 100 mK in the experiment. In order to determine the temperature slope in time (dT/dt), the linear function was fitted to the recorded temperature measurement points in time after switching on the RMF.

In turn, Figures 6b, 7b and 8b show the dependence of (dT/dt) on the value of the RMF intensity amplitude H . In the case of the Dextran FF sample, the measurement points (dT/dt)– H are arranged along the power function $(dT/dt) = (H/2068)^{1.97}$. For the remaining tested samples, the measuring points are arranged in a manner similar to the linear relationship.

As shown in Figure 6b, in the magnetic fluid sample of dextran-coated nanoparticles, there is mainly a relaxation mechanism of heat energy release according to the Néel mechanism. This fact is proved by the value of the exponent $n = 1.97$, close to 2, which is characteristic of superparamagnetic nanoparticles lacking magnetic hysteresis [30].

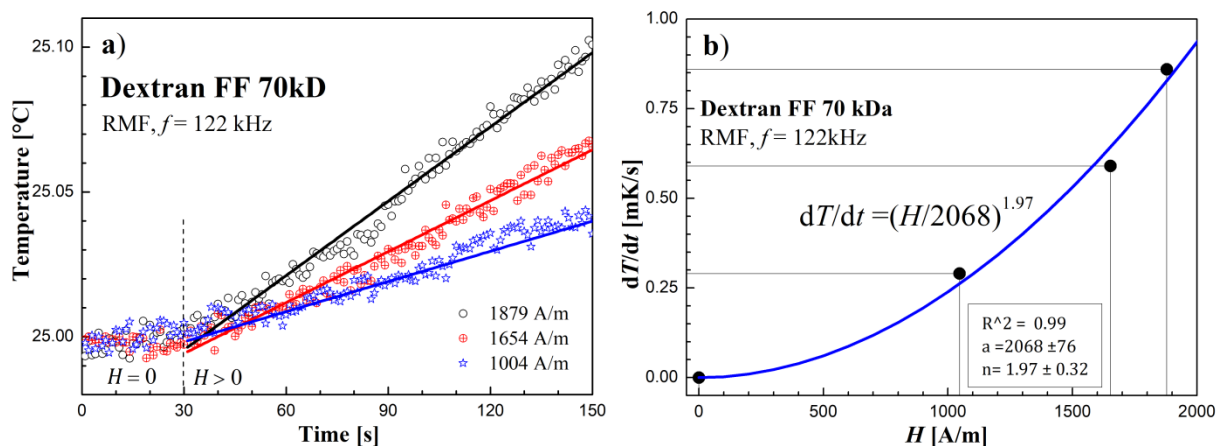


Figure 6. (a) Time courses of temperature changes in the Dextran FF sample. (b) dT/dt values depending on the amplitude of the magnetic field strength H in the RMF at the frequency $f = 122$ kHz.

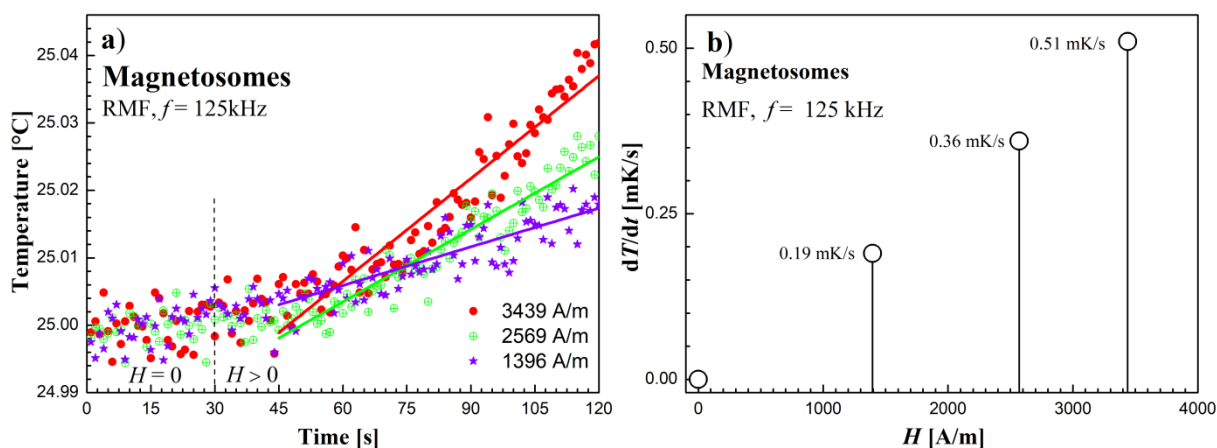


Figure 7. (a) Time courses of temperature changes in the magnetosomes sample. (b) dT/dt values depending on the amplitude of the magnetic field strength H in the RMF at the frequency $f = 125$ kHz.

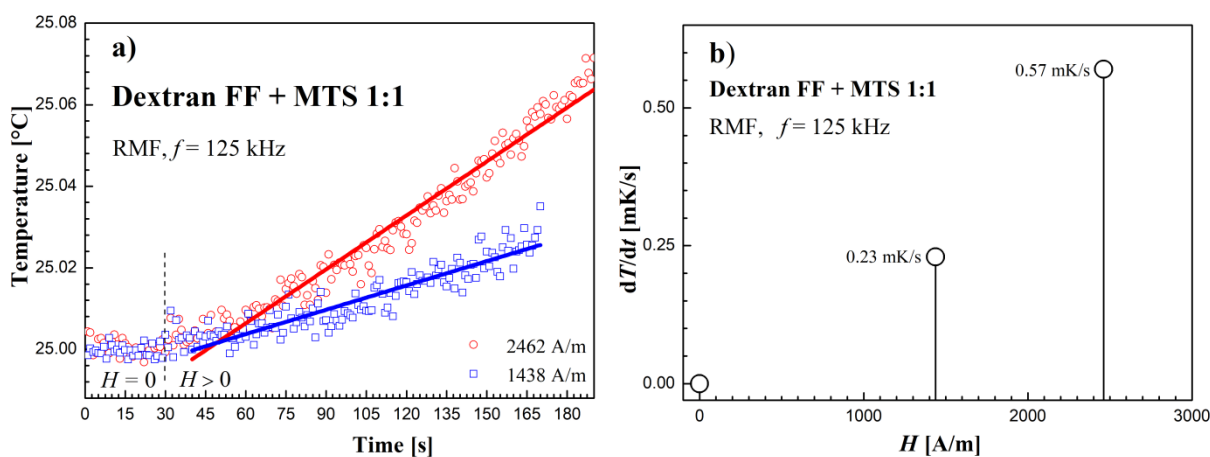


Figure 8. (a) Time courses of temperature changes in the dextran with magnetosomes sample. (b) dT/dt values depending on the amplitude of the magnetic field strength H in the RMF at the frequency $f = 125$ kHz.

3.3. Heating Effect Results in AMF

The AFM calorimetric effect results at similar frequencies are shown in Figures 9–11. Figures 9a, 10a and 11a show the temperature changes of the samples before and after switching on the magnetic field. Due to the achieved higher AMF intensity values (over 10 kA/m), higher temperature increases were obtained. In this case, the thermal fluctuations did not matter much.

The dependence of the measurement points of the dT/dt parameter on the AMF intensity H , shown in the Figures 9b, 10b and 11b, was supplemented with a power function of the $dT/dt = (H/a)^n$ type, where a and n are numerical parameters, obtained from the fitting procedure.

The dependence of the power function dT/dt on the AMF intensity shown in Figure 9b reveals a comparable value of the exponent of this function ($n \simeq 2$) that was obtained in RMF, which proves that the mechanisms of heat released are similar in RMF and AMF.

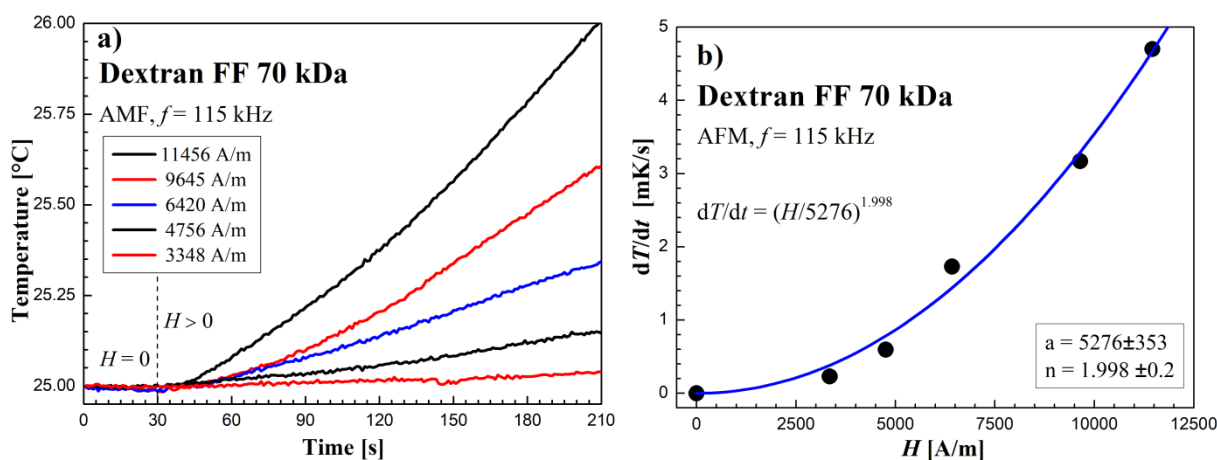


Figure 9. Time courses of temperature changes in the Dextran FF sample (a,b) dT/dt values depending on the amplitude of the magnetic field strength H in the AMF at the frequency $f = 115$ kHz.

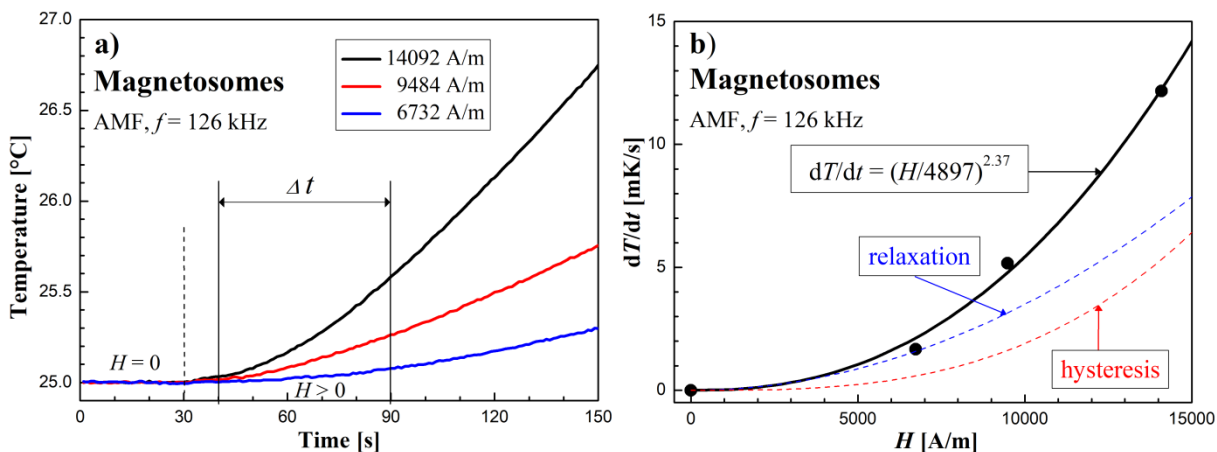


Figure 10. Time courses of temperature changes in the magnetosomes sample (a,b) dT/dt values depending on the amplitude of the magnetic field strength H in the AMF at the frequency $f = 126$ kHz.

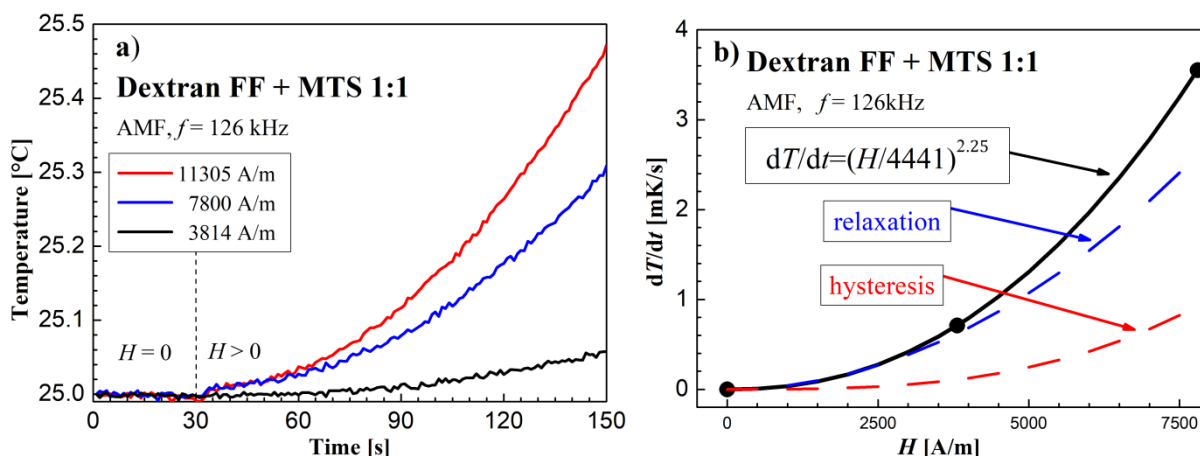


Figure 11. Time courses of temperature changes in the dextran with magnetosomes sample (a,b) dT/dt values depending on the amplitude of the magnetic field strength H in the AMF at the frequency $f = 126$ kHz.

In the case of the sample with magnetosomes, where the magnetic cores are much larger than that of Dextran FF, it can be seen from Figure 10b that the exponent $n = 2.37$. This means that apart from the magnetic relaxation mechanism, there is also an additional source of heat release, which is magnetic hysteresis. Remembering that in the case of losses due to magnetic hysteresis, the thermal power $P \propto (dT/dt)$ separated according to this mechanism is proportional to the cube of the AMF amplitude, $P_{\text{his}} \propto H^3$, we can write the resultant power function $(H/a)^n$ obtained from the matching as:

$$\frac{dT}{dt} = \left(\frac{H}{4897}\right)^{2.37} = \left(\frac{H}{5348}\right)^2 + \left(\frac{H}{8072}\right)^3 \quad (2)$$

Formula (2) thus includes two heat sources in a sample of magnetosomes exposed to AMF. At the same time, it can be seen from Figure 10b that the dominant mechanism of the heat released, however, is magnetic relaxation. Then we can formally present the parameter (dT/dt) as consisting of two effects from relaxation $(H/r)^2$ and from hysteresis $(H/h)^3$:

$$\frac{dT}{dt} = \left(\frac{H}{a}\right)^n = \left(\frac{H}{r}\right)^2 + \left(\frac{H}{h}\right)^3 = \left(\frac{H}{5348}\right)^2 + \left(\frac{H}{8072}\right)^3 \quad (3)$$

3.4. Comparison of the Heating Effects of Samples for Both Types of Magnetic Fields

From the comparative dependences (Figures 12–14) of the heating rates (dT/dt) in RMF and AMF, we can observe much higher heating dynamics (i.e., efficiency) in RMF for all three types of particles (individual, chain-like, and mixed). The most significant difference in the efficacy of RMF to AMF, at comparable f values, can be observed in the case of individual nanoparticles of the Dextran FF sample. In the case of systems containing chain-like nanoparticles (MTS and Dextran FF + MTS), the dT/dt values are always higher when the RMF is applied, but the differences are not as significant as in the case of individual nanoparticles of the Dextran FF sample. These facts correspond to the findings of Bekovic's works [15,16], where he presented higher heat losses in RMF compared to AMF. Experimentally, this finding is also presented in [17], where, in oil-based magnetic fluid, RMF caused a heating effect of more than twice as large as AMF.

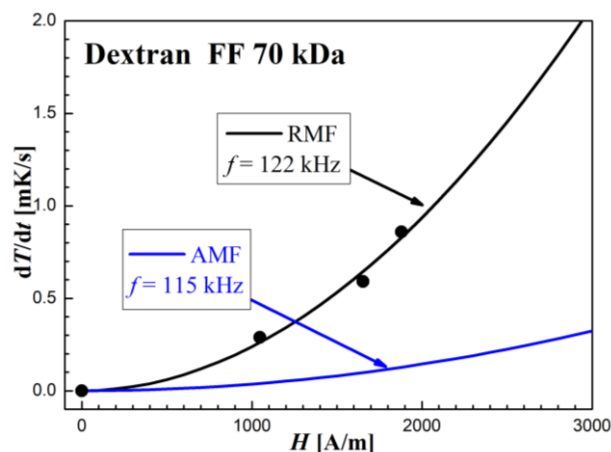


Figure 12. Comparison of the dT/dt dependence on the amplitude of the intensity H measured in rotating and oscillating magnetic fields in the Dextran FF sample.

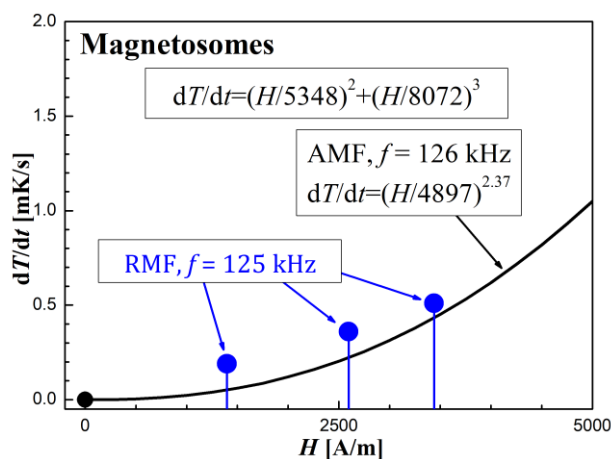


Figure 13. Comparison of the dT/dt dependence on the amplitude of the intensity H measured in rotating and oscillating magnetic field in the Magnetosomes sample.

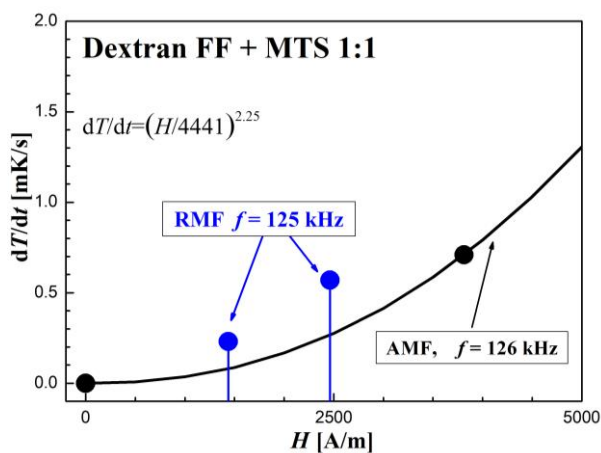


Figure 14. Comparison of the dT/dt dependence on the amplitude of the intensity H measured in rotating and oscillating magnetic field in the sample of Dextran FF + MTS (in a ratio 1:1).

Table 1 provides numerical values of a , n , r , and h parameters obtained by fitting experimental data of magnetite nanoparticle systems' heat response during the application of RMF and AMF. Having determined a , n , r , and h coefficients from Equation (3), it is

possible to calculate the share of thermal energy from the magnetic hysteresis in relation to the total losses in heating from the Equation (3) [34]:

$$\frac{P_{hys}}{P_{total}} = \frac{H \cdot r^2}{h^3 + H \cdot r^2} \quad (4)$$

Table 1. Numerical values of parameters a and n , a power function of the $dT/dt = (H/a)^n$ type for samples determined in RMF and in AMF.

Samples	RMF		AMF			
	a	n	a	n	r	h
Dextran FF	2068	1.97	5276	1.998	$\cong 2068$	∞
MTS			4897	2.37	5348	8072
Dextran FF + MTS			4441	2.25	4832	7996

Figure 15 shows this proportion for comparison for the samples with magnetosomes and for the mixed two samples (magnetosomes and Dextran FF). As can be seen in both cases, the source of the loss in magnetic hysteresis is the sample with magnetosomes, but the sample “mix” gives a smaller share of the loss on hysteresis than the sample with the magnetosomes alone, which is obvious. The Dextran FF sample itself showed no hysteresis loss due to its superparamagnetic properties.

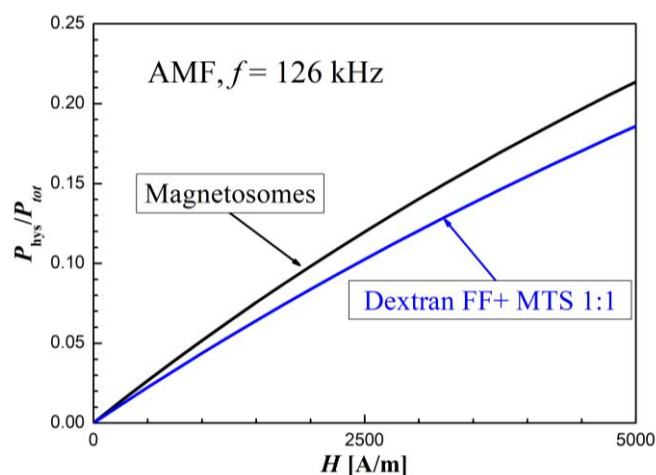


Figure 15. The share of heat energy losses in the magnetosomes sample and in the Dextran FF + MTS sample resulting from magnetic hysteresis in relation to the total loss in a heating effect.

4. Specific Absorption Rate (SAR) Analysis

The primary goal of the work was not to maximize hyperthermic parameters (dT/dt and SAR) but to point out the influence of the method of generating the magnetic field (AMF and RMF). On the other hand, the ability of magnetic particle systems to dissipate the energy of an external alternating magnetic field is given by the SAR ($W \cdot kg^{-1}$; the ratio of the absorbed power to the mass of magnetic particles in the sample) normalized to the unit mass of magnetic material [35]. Moreover, Tables 2–4 below summarize the calculated SAR values at the given fields and frequencies, which were used in the measurement of the heating dependencies. In order to calculate SAR, the following formulas and parameters were used:

$$SAR = C_S \frac{m_S}{m_{NP}} \cdot \left(\frac{dT}{dt} \right), \quad \frac{m_S}{m_{NP}} = \frac{1}{\Phi_V} \cdot \frac{\rho_S}{\rho_{NP}} \quad (5)$$

$$C_S \sim 4184 [J/kg \cdot K], \quad \rho_{NP} = 5180 [kg \cdot m^{-3}], \quad \rho_S = 1000 [kg \cdot m^{-3}], \quad m_{NP} = \Phi_V \cdot \rho_{NP}$$

Table 2. AMF parameters (H and f) and heating rate (dT/dt) of tested samples.

Samples (RMF)	H [A·m ⁻¹]	f [kHz]	dT/dt mK·s ⁻¹	Φ_V [%]	Φ_V [-]	m_{NP} [kg]	m_S/m_{NP}	SAR [W·kg ⁻¹]
Dextran FF	1047	122	0.29	0.29	0.0029	15.0	66.7	80.9
	1653	122	0.59					165
	1879	122	0.86					240
MTS	1396	125	0.19	0.019	0.00019	0.984	1016	808
	2569	125	0.36					1530
	3439	125	0.51					2168
Dextran FF + MTS	1438	125	0.23	0.154	0.00154	7.98	125	120.3
	2462	125	0.57					298.1

Table 3. RMF parameters (H and f) and heating rate (dT/dt) of tested magnetic samples.

Samples (AMF)	H [A·m ⁻¹]	f [kHz]	dT/dt [mK·s ⁻¹]	Φ_V [%]	Φ_V [-]	SAR [W·kg ⁻¹]	ILP = SAR/ $f \cdot H^2$ nH·m ² /kg
Dextran FF	3348	115	0.23	0.29	0.0029	66	0.05
	4756	115	0.595			166	0.06
	6420	115	2.0			558	0.12
	9645	115	3.17			884	0.08
	11,456	115	4.7			1311	0.09
MTS	6732	126	1.67	0.019	0.00019	7103	1.24
	9484	126	5.17			21,988	1.94
	14,092	126	12.17			51,759	2.07
Dextran FF + MTS	3814	126	0.71	0.154	0.00154	373	0.20
	7800	126	3.55			1864	0.24
	11,305	126	5.33			2798	0.17

Table 4. Comparison of the heating effect in the tested samples of different physical parameters in an RMF.

Physical Parameter	Symbol	Unit	Dextran FF	MTS	Dextran FF + MTS
Frequency	f	Hz	1.22×10^5	1.25×10^5	1.25×10^5
Amplitude	H	A·m ⁻¹	1879	3439	2462
Volume concentration	Φ_V	-	0.0029	0.00019	$(0.00019 + 0.0029)/2 = 0.00154$
Heating rate	dT/dt	K·s ⁻¹	0.00086	0.00051	0.00057
Specific absorption rate	SAR	W·kg ⁻¹	240	2168	299
	$(dT/dt)/(\Phi_V \cdot f \cdot H^2)$	K·m ² ·A ⁻²	6.88×10^{-13}	18.2×10^{-13}	4.88×10^{-13}
	ILP = SAR/ $(f \cdot H^2)$	nH·m ² /kg	0.557	1.47	0.69

Table 4 compares the normalized parameter $(dT/dt)/(\Phi_V \cdot f \cdot H^2)$ [30,36] for the three samples tested in RMF. It was established that the highest value (18.2×10^{-13}) is achieved by this parameter for the sample from magnetosomes, in which, due to the large size of nanoparticles, there are probably additional power losses caused by magnetic hysteresis.

Furthermore, the sample from magnetosomes has the lowest volume concentration value of 0.019%. Another formula in the literature, $ILP = SAR/(f \cdot H^2)$ [3–7], indicates that our samples give a comparable value of this parameter to the other literature data. However, the magnetosome material provides the greatest thermal effect of any of our samples. It appears that both parameters can be successfully used when comparing different calorimetric effects as long as dT/dt is proportional to H^2 . This is the case when the nanoparticles show a calorimetric effect from magnetic relaxation and, in addition, when the magnetic field is not too high. At higher amplitudes of the magnetic field intensity (when $\xi > 1$), the nonlinearity effect occurs, and then the power function index $n < 2$.

5. Conclusions

The paper shows the influence of the methodology of the application of alternating magnetic fields on the heating of magnetic colloidal systems of various characters. The assumption that a higher heat generation efficiency is achieved in the case of RMF application than in AMF conditions was confirmed. The particle size, as well as the factor of their arrangement (in the form of individual synthetically stabilized nanoparticles, chain-like particles with a natural envelope, or their mix in a ratio of 1:1), also played an important role in the heating mechanism. The heat energy released principle of individual nanoparticles (Dextran FF) originated in the relaxation mechanism according to Néel and Brown relaxation in both RMF and AMF. This is indicated by the value of the exponent n exponent close to 2, which is characteristic of superparamagnetic nanoparticles. When the systems contained the magnetosomes (MTS and Dextran FF + MTS 1:1 sample), the power law exponent $n > 2$; in that case, except for magnetic relaxation, a magnetic hysteresis contribution to heating is present. For the Dextran FF sample, the dT/dt values were significantly different when compared to RMF and AMF. This difference was not so significant for samples containing magnetosomes, although dT/dt was still higher in the RMF. This suggests that individual nanoparticles have a more pronounced response to RMF than those in chain-like structures. This is probably related to the more dominant contribution originating from the relaxation processes in the hyperthermic experiment. In any case, as already indicated, in magnetic hyperthermia, it is necessary to take into account many factors that affect heat generation. RMF increases the effectiveness of hyperthermia over the often studied AMF. However, higher efficiency is markedly visible in the individual nanoparticle systems of magnetic fluid.

Author Contributions: All authors conceptualized the outline and agreed on the content of the manuscript. Investigation, A.S., M.M., K.Z. and I.S.; writing—original draft preparation, M.M. and A.S.; writing—review and editing, M.M., A.S., P.K., I.S., M.T. All authors have read and agreed to the published version of the manuscript.

Funding: This research was funded by M-ERA.NET 2-FMF and SAS-MOST Joint Research Project SK-TW AZCAL, as well as MODEX (ITMS2014+: 313011T548), NATO Science for Peace and Security Programme (G5683), VEGA 2/0043/2, by the Slovak Research and Development Agency under the Contract No. APVV-19-0324, and by the project of the School of Science (SNS) from Adam Mickiewicz University of Poznań.

Institutional Review Board Statement: Not applicable.

Informed Consent Statement: Not applicable.

Data Availability Statement: Not applicable.

Conflicts of Interest: The authors declare no conflict of interest.

References

1. Moise, S.; Byrne, J.M.; Haj, A.J.E.; Telling, N.D. The Potential of Magnetic Hyperthermia for Triggering the Differentiation of Cancer Cells. *Nanoscale* **2018**, *10*, 20519–20525. [[CrossRef](#)]
2. Bruvera, I.J.; Actis, D.G.; Calatayud, M.P.; Mendoza Zélis, P. Typical Experiment vs. in-Cell like Conditions in Magnetic Hyperthermia: Effects of Media Viscosity and Agglomeration. *J. Magn. Magn. Mater.* **2019**, *491*, 165563. [[CrossRef](#)]

3. Ebrahimisadr, S.; Aslibeiki, B.; Asadi, R. Magnetic Hyperthermia Properties of Iron Oxide Nanoparticles: The Effect of Concentration. *Phys. C Supercond. Appl.* **2018**, *549*, 119–121. [[CrossRef](#)]
4. Rezanezhad, A.; Hajalilou, A.; Eslami, F.; Parvini, E.; Abouzari-Lotf, E.; Aslibeiki, B. Superparamagnetic Magnetite Nanoparticles for Cancer Cells Treatment via Magnetic Hyperthermia: Effect of Natural Capping Agent, Particle Size and Concentration. *J. Mater. Sci. Mater. Electron.* **2021**, *32*, 24026–24040. [[CrossRef](#)]
5. Aslibeiki, B.; Ehsani, M.H.; Nasirzadeh, F.; Mohammadi, M.A. The Effect of Interparticle Interactions on Spin Glass and Hyperthermia Properties of Fe₃O₄ Nanoparticles. *Mater. Res. Express* **2017**, *4*, 075051. [[CrossRef](#)]
6. Aslibeiki, B.; Eskandarzadeh, N.; Jalili, H.; Ghotbi Varzaneh, A.; Kameli, P.; Orue, I.; Chernenko, V.; Hajalilou, A.; Ferreira, L.P.; Cruz, M.M. Magnetic Hyperthermia Properties of CoFe₂O₄ Nanoparticles: Effect of Polymer Coating and Interparticle Interactions. *Ceram. Int.* **2022**, *48*, 27995–28005. [[CrossRef](#)]
7. Jalili, H.; Aslibeiki, B.; Hajalilou, A.; Musalu, O.; Ferreira, L.P.; Cruz, M.M. Bimagnetic Hard/Soft and Soft/Hard Ferrite Nanocomposites: Structural, Magnetic and Hyperthermia Properties. *Ceram. Int.* **2022**, *48*, 4886–4896. [[CrossRef](#)]
8. Yu, X.; Ding, S.; Yang, R.; Wu, C.; Zhang, W. Research Progress on Magnetic Nanoparticles for Magnetic Induction Hyperthermia of Malignant Tumor. *Ceram. Int.* **2021**, *47*, 5909–5917. [[CrossRef](#)]
9. Roca, A.G.; Wiese, B.; Timmis, J.; Vallejo-Fernandez, G.; O’Grady, K. Effect of Frequency and Field Amplitude in Magnetic Hyperthermia. *IEEE Trans. Magn.* **2012**, *48*, 4054–4057. [[CrossRef](#)]
10. Ortega, D.; Pankhurst, Q.A. Magnetic Hyperthermia. *Nanoscience* **2013**, *1*, e88. [[CrossRef](#)]
11. Brezovich, I.A. Low Frequency Hyperthermia: Capacitive and Ferromagnetic Thermoseed Methods. *Med. Phys. Monogr.* **1988**, *16*, 82–110.
12. Hergt, R.; Dutz, S. Magnetic Particle Hyperthermia—Biophysical Limitations of a Visionary Tumour Therapy. *J. Magn. Magn. Mater.* **2007**, *311*, 187–192. [[CrossRef](#)]
13. Skumiel, A.; Leszczyński, B.; Molcan, M.; Timko, M. The Comparison of Magnetic Circuits Used in Magnetic Hyperthermia. *J. Magn. Magn. Mater.* **2016**, *420*, 177–184. [[CrossRef](#)]
14. Dan, M.; Bae, Y.; Pittman, T.A.; Yokel, R.A. Alternating Magnetic Field-Induced Hyperthermia Increases Iron Oxide Nanoparticle Cell Association/Uptake and Flux in Blood—Brain Barrier Models. *Pharm. Res.* **2015**, *32*, 1615. [[CrossRef](#)]
15. Beković, M.; Trbušić, M.; Trlep, M.; Jesenik, M.; Hamler, A. Magnetic Fluids’ Heating Power Exposed to a High-Frequency Rotating Magnetic Field. *Adv. Mater. Sci. Eng.* **2018**, *2018*, 6143607. [[CrossRef](#)]
16. Beković, M.; Trlep, M.; Jesenik, M.; Hamler, A. A Comparison of the Heating Effect of Magnetic Fluid between the Alternating and Rotating Magnetic Field. *J. Magn. Magn. Mater.* **2014**, *355*, 12–17. [[CrossRef](#)]
17. Skumiel, A.; Kopcansky, P.; Timko, M.; Molcan, M.; Paulovicova, K.; Wojciechowski, R. The Influence of a Rotating Magnetic Field on the Thermal Effect in Magnetic Fluid. *Int. J. Therm. Sci.* **2022**, *171*, 107258. [[CrossRef](#)]
18. Skumiel, A. A New Way to Generate a Rotating Magnetic Field in the High Frequency Range. *J. Magn. Magn. Mater.* **2022**, *541*, 168529. [[CrossRef](#)]
19. Konopacki, M.; Jędrzejczak-Silicka, M.; Szymańska, K.; Mijowska, E.; Rakoczy, R. Effect of Rotating Magnetic Field on Ferromagnetic Structures Used in Hyperthermia. *J. Magn. Magn. Mater.* **2021**, *518*, 167418. [[CrossRef](#)]
20. Le Nagard, L.; Zhu, X.; Yuan, H.; Benzerara, K.; Bazylinski, D.A.; Fradin, C.; Besson, A.; Swaraj, S.; Stanescu, S.; Belkhou, R.; et al. Magnetite Magnetosome Biomineralization in Magnetospirillum Magneticum Strain AMB-1: A Time Course Study. *Chem. Geol.* **2019**, *530*, 119348. [[CrossRef](#)]
21. Amor, M.; Busigny, V.; Louvat, P.; Tharaud, M.; Gélabert, A.; Cartigny, P.; Carlut, J.; Isambert, A.; Durand-Dubief, M.; Ona-Nguema, G.; et al. Iron Uptake and Magnetite Biomineralization in the Magnetotactic Bacterium Magnetospirillum Magneticum Strain AMB-1: An Iron Isotope Study. *Geochim. Cosmochim. Acta* **2018**, *232*, 225–243. [[CrossRef](#)]
22. Molday, R.S.; Mackenzie, D. Immunospecific Ferromagnetic Iron-Dextran Reagents for the Labeling and Magnetic Separation of Cells. *J. Immunol. Methods* **1982**, *52*, 353–367. [[CrossRef](#)]
23. Sipošova, K.; Pospiskova, K.; Bednarikova, Z.; Safarik, I.; Safarikova, M.; Kubovcikova, M.; Kopcansky, P.; Gazova, Z. The Molecular Mass of Dextran Used to Modify Magnetite Nanoparticles Affects Insulin Amyloid Aggregation. *J. Magn. Magn. Mater.* **2017**, *427*, 48–53. [[CrossRef](#)]
24. Molcan, M.; Gojzewski, H.; Skumiel, A.; Dutz, S.; Kovac, J.; Kubovcikova, M.; Kopcansky, P.; Vekas, L.; Timko, M. Energy Losses in Mechanically Modified Bacterial Magnetosomes. *J. Phys. D Appl. Phys.* **2016**, *49*, 365002. [[CrossRef](#)]
25. Molcan, M.; Kopcansky, P.; Timko, M.; Rajnak, M.; Gojzewski, H.; Čebers, A. Dispersion of Magnetic Susceptibility in a Suspension of Flexible Ferromagnetic Rods. *J. Mol. Liq.* **2020**, *305*, 112823. [[CrossRef](#)]
26. Dzarova, A.; Royer, F.; Timko, M.; Jamon, D.; Kopcansky, P.; Kovac, J.; Choueikani, F.; Gojzewski, H.; Rousseau, J.J. Magneto-Optical Study of Magnetite Nanoparticles Prepared by Chemical and Biomineralization Process. *J. Magn. Magn. Mater.* **2011**, *323*, 1453–1459. [[CrossRef](#)]
27. Timko, M.; Molcan, M.; Hashim, A.; Skumiel, A.; Muller, M.; Gojzewski, H.; Jozefczak, A.; Kovac, J.; Rajnak, M.; Makowski, M.; et al. Hyperthermic Effect in Suspension of Magnetosomes Prepared by Various Methods. *IEEE Trans. Magn.* **2013**, *49*, 250–254. [[CrossRef](#)]
28. Alphandéry, E.; Ding, Y.; Ngo, A.T.; Wang, Z.L.; Wu, L.F.; Pileni, M.P. Assemblies of Aligned Magnetotactic Bacteria and Extracted Magnetosomes: What Is the Main Factor Responsible for the Magnetic Anisotropy? *ACS Nano* **2009**, *3*, 1539–1547. [[CrossRef](#)]

29. Molcan, M.; Petrenko, V.I.; Avdeev, M.V.; Ivankov, O.I.; Garamus, V.M.; Skumiel, A.; Jozefczak, A.; Kubovcikova, M.; Kopcansky, P.; Timko, M. Structure Characterization of the Magnetosome Solutions for Hyperthermia Study. *J. Mol. Liq.* **2017**, *235*, 11–16. [[CrossRef](#)]
30. Rosensweig, R.E. Heating Magnetic Fluid with Alternating Magnetic Field. *J. Magn. Magn. Mater.* **2002**, *252*, 370–374. [[CrossRef](#)]
31. Skumiel, A.; Kaczmarek, K.; Flak, D.; Rajnak, M.; Antal, I.; Brząkała, H. The Influence of Magnetic Nanoparticle Concentration with Dextran Polymers in Agar Gel on Heating Efficiency in Magnetic Hyperthermia. *J. Mol. Liq.* **2020**, *304*, 112734. [[CrossRef](#)]
32. Gojzewski, H.; Makowski, M.; Hashim, A.; Kopcansky, P.; Tomori, Z.; Timko, M. Magnetosomes on Surface: An Imaging Study Approach. *Scanning* **2012**, *34*, 159–169. [[CrossRef](#)]
33. Molcan, M.; Kaczmarek, K.; Kubovcikova, M.; Gojzewski, H.; Kovac, J.; Timko, M.; Józefczak, A. Magnetic Hyperthermia Study of Magnetosome Chain Systems in Tissue-Mimicking Phantom. *J. Mol. Liq.* **2020**, *320*, 114470. [[CrossRef](#)]
34. Izydorzak, M.; Skumiel, A.; Leonowicz, M.; Kaczmarek-Klinowska, M.; Pomogailo, A.D.; Dzhardimalieva, G.I. Thermophysical and Magnetic Properties of Carbon Beads Containing Cobalt Nanocrystallites. *Int. J. Thermophys.* **2011**, *33*, 627–639. [[CrossRef](#)]
35. Davydov, A.S.; Belousov, A.V.; Krusanov, G.A.; Kolyvanova, M.A.; Kovalev, B.B.; Komlev, A.S.; Krivoschapkin, P.V.; Morozov, V.N.; Zverev, V.I. Promising Magnetic Nanoradiosensitizers for Combination of Tumor Hyperthermia and X-ray Therapy: Theoretical Calculation. *J. Appl. Phys.* **2021**, *129*, 033902. [[CrossRef](#)]
36. Halbreich, A.; Roger, J.; Pons, J.N.; Geldwerth, D.; Da Silva, M.F.; Roudier, M.; Bacri, J.C. Biomedical Applications of Maghemite Ferrofluid. *Biochimie* **1998**, *80*, 379–390. [[CrossRef](#)]