Hydroxyapatite-Bioceramic/Expanded Perlite Hybrid Composites Coating on Ti₆Al₄V by Hydrothermal Method and in vitro Behavior

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ABSTRACT: This study was aimed to coat a hybrid bioceramic composite onto Ti₆Al₄V by using hydrothermal method. The Hybrid bioceramic composite for coating was prepared by reinforcing different rations of expanded perlite (EP) and 5 wt.% chitosan into synthesized Hydroxyapatite (HA). Coating was performed at 1800°C for 12 hours. The coated specimens were gradually subjected to a sintering at 6000°C for 1 hour. For in vitro analysis, the specimens were kept in Ringer's solution for 1, 10, and 25 days. All specimens were examined by SEM, EDX, FTIR, and surface roughness analyses for characterizing. It was concluded that as the reinforcement ratio increased, there was an increase in coating thickness and surface roughness. The optimum reinforcement ratio for expanded perlite can be 10 wt.% (A3-B3). With increasing ratio of calcium (Ca) and phosphate (P) (Ca/P), the surface becomes more active in body fluid and then observed the formation of the hydroxycarbonate apatite (HCA) layer. As the waiting time increased, there was an increase in the formation of an apatite structure.

KEYWORDS: Expanded perlite, hydroxyapatite, bioceramics, hydrothermal method

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Introduction

Increase in the human population has directly caused developments in science and technology. Materials used in repairing tissue damages occurring as a result of accidents and injuries experienced by humans and produced and developed to help the functioning of the body are known as biomaterials. The most important property expected from a biomaterial is that it is biocompatible. That is, a biomaterial is expected to be compatible with the tissues surrounding it and not induce unwanted side effects on the tissue. This is why it is desired that every material used in the body is biocompatible.¹ Implants that are used as biomaterials are categorized as metallic, ceramic, polymer, and composite materials.²

Materials such as titanium and its alloys, cobalt and its alloys, stainless-steel, and gold are used as metallic biomaterials as they have high strength and wear resistance.^{3,4} Titanium and its alloys are preferred in several biomedical applications as they do not react with the oxygen in the air and do not form oxides, and they are durable. Additionally, due to the compatible structure of biomaterials made out of titanium and its alloys with bone, they are prevalently preferred for use.⁵⁻⁸ Hydroxyapatite (HA) is a biocompatible biomaterial that is usually used in coating bones, teeth, and implants. Hydroxyapatite (HA) $[Ca_{10} (PO_4)_6 (OH)_2]$ constitutes 70% of the bone structure. It is also preferred as it has high bioactivity and biodegradability and does not have any toxic effects.9,10 In recent years, in the fields of orthopedics and dental implants,11 several studies have been carried out Hydroxyapatite (HA) coating to the metal implants (Ti, Ti₆Al₄V, 316 L stainless-steel).¹²

Various techniques have been developed to coat Hydroxyapatite (HA) onto the surfaces of titanium and its alloys. Examples of these may be listed as CVD (chemical vapor deposition), PVD (physical vapor deposition), plasma spray, vacuum plasma, HVOF (high-velocity oxygen fuel thermal spray process), sol-gel submersion, and EPD (electrophoretic coating). All of these methods have some disadvantages about the poor connection between coating and the surface.13 Plasma spray coating is the most frequently used method in uncemented hip prosthetic surgeries. However, plasma spray Hydroxyapatite (HA) coatings have some unwanted properties such as composition, structural control, and weak adhesive strength.¹⁴ The hydrothermal method that has recently been used as a coating method, was preferred in our study since its reaction temperature is lower than 200°C, process time is short, the coating is more homogenous, and the method is low-cost.¹⁵⁻¹⁸

Studies with bioactive glasses have a significant place in the repair of damaged tissues. Bioactive glasses have high biocompatibility properties with the help of their composition consisting of SiO₂, Na₂O, CaO, and P₂O₅. This is because bioactive glass contains almost 60% SiO₂ and Na₂O, CaO, and P₂O₅ by weight in its composition, and these components become highly active in aqueous environments such as body fluids. Moreover, MgO and Al₂O₃, which may be found in the composition of bioactive glasses, can form a bond with body tissues.¹⁹⁻²¹ Also, chemical composition of the expanded perlite (EP) contains 60% to 75% SiO_2 , 10% to 15% Al_2O_3 and with different ratio of K₂O, MgO, CaO, and Fe₂O₃. Considering this perspective, expanded perlite shows many similarities to



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COMPOSITION	SIO ₂	AL ₂ O ₃	K ₂ O	MGO	CAO	FE_2O_3	NA ₂ O ₃	TIO ₂	MNO ₂	SO ₃
wt.%	74	14.33	4.95	0.28	0.5	0.97	2.9	0.12	0.07	0.03

Table 1. Chemical composition of Expanded perlite (EP) (in wt.%).



Figure 1. SEM images of the Expanded perlite (EP).

Table 2. The chemical composition of the Ti₆Al₄V alloy substrate.

С	FE	Ν	0	AL	V	н
<0.08	<0.25	<0.05	<0.02	5.5-6.76	3.5-4	<0.015

commercial inert glass, bioceramic glass and especially bioactive glass.^{22,23} chitosan is frequently used in tissue engineering due to its properties such as biocompatibility, bioactivity, and biodegradability.^{24,25} Furthermore, it supports cell adhesion and multiplication. Chitosan allows the formation of a porous scaffolding structure that supports the tissue scaffold. Due to these properties, it is prevalently used in studies conducted on Hydroxyapatite.²⁶

In this study, we aimed to coat a titanium alloy (Ti_6Al_4V) with Hydroxyapatite (HA)- Expanded Perlite (EP) and Hydroxyapatite (HA)- Expanded Perlite (EP)-Chitosan using by hydrothermal coating method. This method which has several superior aspects to the other researched coating methods was preferred in our study since its reaction temperature is lower than 200°C, process time is short, the coating is more homogenous, and the method is low-cost. Also, Expanded perlite was investigated with respect to its properties as component of composite systems considered for biomedical applications. chemical composition of the expanded perlite (EP) shows nearly same amount with biomaterials especially bioceramic glass. It has high reserve with low cost. Although the effect of perlite with different factors has been studied and observed in many field, there is a lack of knowledge of perlite at biocomposite materials, and still need for more research to be able to determine the optimum settings for the biocomposite coating in different HA-reinforcement combinations. In this respect, many advantages in terms of properties, biocompatibility, as well as low cost could be ascribed to the realization of Expanded Perlite. In none of the previous studies, expanded perlite as a additives for the biomaterials have been subjected to this hydrothermal method in coating. So, it was targeted in this study to compare the coating thicknesses forming on the surfaces of the specimens, and the conditions in which the best coating obtained were examined according to the SEM-EDX, XRD, and FTIR results.

Materials and Methods

The expanded perlite used in the experiments was obtained from the firm of the Nanography (Nano technology informatics manufacturing and consulting C LLC. (Ankara, Turkey)). The size of expanded perlite is 50 microns, and the chemical composition of the expanded perlite is shown in Table 1. Also, Figure 1 shows the SEM images of the expanded perlite. The P_2O_5 and Ca(NO₃) used as the sources of Hydroxyapatite (HA) and the Ti₆Al₄V alloy used as a base material in the experiments were obtained from the firm of fytronix scientific instruments (Elazığ, Turkey).

Preparation of Ti₆Al₄V plates

The chemical composition of the Ti_6Al_4V alloy used as the base material in the experiments is shown in Table 2. The surfaces of the Ti_6Al_4V alloy material were grinded (400-1200 mesh). The surfaces of the specimens were then cleaned



Figure 2. The schematic representation of the synthesis of hybrid coating on Ti₆Al₄V.

Table 3. The codes and contents of used specimens.

WITH TRE	IOUT HEAT ATMENTED (A)	HEAT TREATMENTED (B)			
A1	HA + %5EP	B1	HA+%5EP		
A2	HA + %5EP + %5 Chitosan	B2	HA + %5EP + %5 Chitosan		
A3	HA + %10EP	B3	HA+%10EP		
A4	HA + %15EP	B4	HA + %15EP		
A5	HA + %15EP + %5 Chitosan	B5	HA + %15EP + %5 Chitosan		

with ethanol, acetone and distilled water. The cleaned specimens were kept in a 66.3% H_2SO_4 and 10.6% HCl solution at a weight ratio of 1:1 at 60°C for 1 hour. After the specimens were dried at room temperature, and a group of specimens was subjected to a preliminary heat treatment process at 600°C.

Preparation of coating solution

Calcium nitrate Ca(NO₃) and di-phosphorus pentoxide (P_2O_5) were dissolved in 15 ml distilled water for Hydroxyapatite (HA) synthesis, so that the calcium (Ca) and phosphate (P) stochiometric ratio would be 1.67 (Ca/P). The expanded perlite used at ratios of 5%, 10%, and 15% as a reinforcement material was dissolved in 15 ml distilled water. The obtained solutions were dissolved into each other on a magnetic stirrer. It was ensured that the solution on the magnetic stirrer had 10 pH by using NH_3 . The resulting solution and the Ti_6Al_4V alloy were subjected to the coating process in a hydrothermal method (12 hours, 180°C). Figure 2 shows the steps of the hydrothermal process. The codes of the obtained specimens are presented in Table 3.

Characterization of synthesized hydroxyapatite coating

The characterizations of the hybrid coatings were carried out by scanning electron microscope attached to energy dispersive spectroscopy (SEM-EDS, Jeol, JSM-7001F), atomic force microscope (AFM, Park System 100-E), X-ray diffraction (XRD, Bruker D8), and Fourier-transform infrared spectroscopy (FTIR, Thermo Scientific[™] Nicolet[™] iS^{™5}) spectroscopy. Also, the surface roughness of the specimens was measured using an SRT-6210 digital portable surface roughness Measurement device.

In vitro tests

In vitro tests were conducted to determine the biocompatibility of the biocomposite specimens. Ringer's lactate solution was used for the in vitro tests. Ringer's solution is frequently used in in vitro tests on organs or tissues and assessment of titanium coatings, and it resembles the chemical composition of human blood plasma to a large extent.²⁷⁻³⁰ The chemical compositions of Ringer's solution and human blood are presented in Table 4. For each coated specimens were kept in 75 ml Ringer's lactate solution at 37°C for 1, 10, and 25 days for the in vitro processes.

Table 4. Chemical composition of SBF and Ringer's solution.³¹

CONCENTRATION (MMOLL ⁻¹)							
ION	HUMAN BLOOD PLASMA	RINGER'S SOLUTION					
Na ⁺	142	147.2					
K+	5	4					
Mg ²⁺	1.5	-					
Ca ²⁺	2.5	2,2					
CI-	103	155.7					
HCO ₃ ⁻	27	-					
HPO₄ [−]	1	-					
SO42-	0.5	-					

Experimental Results and Discussion

Micrography results

Figures 3 to 6 show the SEM images -EDX analysis of the coated A specimens.

It was observed that a connected and porous structure was present on the surface coatings of all A specimens, but there was no homogenous distribution. Lack of a homogenous distribution on the coating surface makes osteointegration easier.32 It is thought that the porous structure observed was caused by the structure of the expanded perlite and HA.33 Morphology of the HA is rough and porous. These pores are called macro pores because of the pores diameter which it is bigger than 100 µm. Porosity increases the surface area and provides a broader area for bone cell growth.³⁴ A bioactive glass has a composition of 44.97% to 58.47% SiO₂, 17.79% to 24.49% CaO, 17.76% to 24.55% Na2O, and 5.98% to 5.99% P₂O₅.³⁵ This glass was kept in simulated body fluid and formed a calcium phosphate layer. By conducting biological and toxicological tests on the obtained specimens, they supported their use in bone applications as they had high phosphate contents.³⁶ In addition to containing high amounts of Al₂O₃ and SiO₂ in its structure, expanded perlite also contains MgO, CaO, and Fe_2O_3 . It also contains trace amounts of Na_2O_5 , CaF_2 , and P_2O_5 . The component density observed in the EDX analysis (Figure 3) supported the contents of expanded perlite.

When the A1 and A2 specimens were compared in terms of chitosan reinforcement, it was observed that the coating thickness increased from approximately 4.863 to 9.902 μ m. This was because chitosan supports cell adhesion and multiplication due to its chemical structure and allows the formation of internally connected pores that provide in vitro and in vivo bone tissue formation.³⁷

As shown in Figure 5, as the ratio of expanded perlite was increased, there were variable change in the coating thickness. While the coating thickness at the expanded perlite ratio of 5 wt.% was approximately $4.86 \,\mu\text{m}$, it was observed to increase

approximately 2-fold (9.19 μ m) when this ratio was increased to 10 wt.%. However, when the expanded perlite reinforcement ratio became 15 wt.%, the thickness was observed unexpectedly as 2.87 μ m. This was the lowest coating thickness value that we obtained. While the reason for this is not completely known, we observed that reinforcement components have some limiting ratio that we encountered in some examples from the literature. A study about the limiting nature of reinforcement components was conducted by Demirkol et al produced composites by adding commercial inert glass to synthetic HA at ratios of 5 and 10 wt.%. They selected the temperature range of 1000°C to 1300°C to observe the effects of sintering. They observed better results in composites with the sintering temperature of 1300°C where 5 wt.% commercial inert glass was used.³⁸

In the comparison of the A4 and A5 specimens, it was observed that the thickness of coating increased more with the addition of chitosan when the expanded perlite ratio was 5 wt.%. This showed us that chitosan reinforcement increased thickness at low expanded perlite reinforcement ratios. When the A4 (2.87 μ m) and A5 (6.76 μ m) specimens were compared, it was seen that the thickness increased when chitosan was reinforced into the specimen with the lowest thickness obtained. These suggested that chitosan supports cell adhesion and multiplication.³⁹

Figures 7 to 10 show the SEM-EDX analysis of the B specimens. In the B group of specimens, a porous surface morphology with a low crack density and different zone were observed on the SEM images. This porous structure makes osteointegration easier in bioceramic-coated implants.³² The porous and rough nature of the obtained coating is an important factor in increasing bioactivity in addition to providing the use of bone-producing cells in these areas.³² While the macropore diameter may be greater than 100 µm, the micropore diameter may be smaller than 10 µm. Porous structures smaller than 1 µm play a role in bioactivity as they facilitate the interaction of proteins.⁴⁰ Additionally, for good adhesion between the coating and the metal, crack-free structures need to be obtained. If sufficient adhesion is achieved between the coating and the implant, coating-tissue interaction is achieved, and the implant is fixed onto the tissue. This means a faster and better integration of the implant to the tissue.⁴¹⁻⁴³ It is also thought that the dense structures observed in the EDX analysis originated from the content of HA and rich content of expanded perlite.

In the comparison of the B1 and B2 specimens, chitosan reinforcement into the coating under the same conditions and reinforcements by 5 wt.%, increased the coating thickness from 4.453 to $9.12 \,\mu$ m.

As seen in Figure 9, in the B specimen group, as the expanded perlite reinforcement increased, the coating thickness increased linearly. When the expanded perlite reinforcement ratio was 5%, the coating thickness was $4.453 \,\mu\text{m}$. When the expanded perlite reinforcement ratios were 10% and 15%, the coating thicknesses became respectively 6.85 and 9.12 μ m.

										El Al	I Series	Net	unn. C [wt.%]	norm. C [wt.%]	Atom. C E [at.%]	rror (1 Sigma) [wt.%]
		- DPC A	1		Stall.	-	and a start	and the second		Ti 2	K-series	14012	42.29	55.30	33.92	1.26
	1.1.1.150						And Call and	And State Providence	Col. Chi and	0 8	K-series	802	22.12	11 23	9 23	5.51
			1. 1. 1. 1.						1 A.M.	P 1	K-series	1299	2.08	2.72	2.58	0.13
	· · · ·		1	Service R.	1.20				199.8	V 2	K-series	192	0.60	0.78	0.45	0.07
Contraction of the	9	1	- C		1				÷ .	A1 1	K-series	65	0.14	0.18	0.20	0.05
	Charles .		Sec. 1						e B	Si 1	K-series	48	0.07	0.09	0.10	0.04
Al	and set of the	and the	A ST AND	ALC: NO					118	N 7	K-series	0	0.00	0.00	0.00	0.00
at the			S 1	19 10 10			2411		3.696	Na 1	K-series	0	0.00	0.00	0.00	0.00
10 µm	EHT = 20.00 kV WD = 7.5 mm	Signal A = SE1 Meg = 2.00 K X	Date :24 Jul 2019 Time :11:05:14	ZEISS	2 µm	EHT = 20.00 kV WD = 9.0 mm	Signal A = SE1 Mag = 5.00 K X	Date :24 Jul 2019 Time :13:11:59	ZEINN			Total	; 76.47	100.00	100.00	
		and the second		31						El A	N Series	Net	unn. C [wt.%]	norm. C [wt.%]	Atom. C En [at.%]	rror (1 Sigma) [wt.%]
	a contraction	and the second	2. 2. de	1 and						Ca	0 K-serie	6723	30.64	48.67	31.09	1.02
1	1 11 16	The second	The and		-				7	0	K-serie:	3 481	21.49	34.14	54.63	6.09
1. 3	Stame &				10 S 24	100	a with some	- 1 PE - 13 P	1. 19	P	5 K-serie	2834	8.94	14.20	11.73	0.44
1000			100000000		and 2	14 10 1	1111 3	11 88 18 6	11 + 202	Si	4 K-serie:	401	1.10	1.74	1.59	0.11
1.4	and a state of the		Calcar F. C.	2	30 2 34			129111	FALL	Al 1	3 K-serie:	3 103	0.39	0.62	0.59	0.08
			10 0 54		1 1 1	5181	1.12 1.2	11 1 1 1 1 1 1 1 1	6712523	Ti 2	2 K-serie	3 28	0.21	0.33	0.18	0.07
A3			and the second		1 3	1/2/11/	11 B 1	1 / 1 / / 12	118 11 11	Na	1 K-serie	5 19	0.14	0.22	0.08	0.05
1					11.25	1 2 191	19 - 19 C E	1111111	1 1 39	N	K-serie:	3 0	0.00	0.00	0.00	0.00
2 µm	EHT = 20.00 kV WD = 7.5 mm	Signal A = SE1 Meg = 5.00 K X	Date :24 Jul 2019 Time :11:20:29	ZEISS	2 µm	EHT = 15.00 KV WD = 9.0 mm	Signal A = SE1 Meg = 5.00 K X	Date :29 Nov 2019 Time :12:28:01	ZEISS			Total	1: 62.9	5 100.0	0 100.00	
				9 19 19			-			El A	N Series	Net	unn. C [wt.%]	norm. C [wt.%]	Atom. C E [at.%]	rror (1 Sigma) [wt.%]
		and the second	5. • • • • • • • •		1.0	And the second second	and the second sec	1000	and the second	Ti 2	K-series	6336	44.62	56.36	34.70	1.42
100	See.	Service States			-	-			ALC: NOT THE OWNER	0 8	K-series	352	23.37	29.52	54.38	7.34
		5.5	Service and the							Ca 2	K-series	1541	6.90	8.71	6.41	0.31
		15 15	1 H H H H							P 1	K-series	764	2.87	3.63	3.45	0.20
2010		2 2 .								V 2	K-series	187	1.35	1.71	0.99	0.14
100	1.1.1									A1 1	K-series	11	0.05	0.07	0.07	0.04
A4	Constant -		and the second							N /	K-series	0	0.00	0.00	0.00	0.00
	TT PUC	Sand Carles	1							Na 1	K-series	0	0.00	0.00	0.00	0.00
10 µm	EHT = 20.00 kV	Signal A = SE1	Date :24 Jul 2019	11115	2 µm	EHT = 20.00 KV	Signal A = SE1	Date :24 Jul 2019								
H	WD = 7.5 mm	Meg = 2.00 K X	Time :11:33:01		H	WD = 8.5 mm	Meg = 5.00 K X	Time :13:25:22	26155			Total	: 79.1	6 100.0	00 100.00	

Figure 3. SEM images and EDX analysis of A1, A3, and A4 specimens with different magnifications.



Figure 4. SEM images of cross-section of the hybrid coatings for A1 and A2 specimens.

The study by Say et al⁴² supported the results of our study. They observed that the adhesive strength of HA was lower than coating with HA-SiO₂. Other studies have also observed that adding silicate into the HA improved thermal stability, dissolvability and mechanical properties.⁴⁴⁻⁴⁶

As seen in Figure 10, when 5 wt.% chitosan was added to the 15 wt.% expanded perlite-reinforced specimens, the coating

thickness decreased from 13.89 to $9.43 \,\mu$ m. That is, when B4 and B5 were compared, addition of $5 \,w$ t.% chitosan into $15 \,w$ t.% expanded perlite reinforcement reduced the coating thickness. Since the coatings will be in contact with the liquids in the human body, they will show wear behavior over time. Wear tests are also required to measure the wear behavior of the coatings and to calculate the friction coefficients. Work on this will continue.



Figure 5. SEM images of cross-section of the synthesized hybrid coatings for A1, A3, and A4 specimens.



Figure 6. SEM images of cross-section of the synthesized hybrid coatings for A4 and A5 specimens.

B1		El AN Series Net unn. C norm. C Atom. C Error (1 Sigma) [wt.4] [wt.4]
10 µm EHT = 20.00 KV Signal A = SE1 Date :24 Jul 2019 ZEXX	2 µm EHT = 20.00 KV Signal A = SE1 Date :2	Total: 81.26 100.00 100.00
	1 1 XIV XIV 10	El AN Series Net unn. C norm. C Atom. C Error (1 Sigma) [wt.%] [wt.%] [at.%] [wt.%]
The IS X 1 Th		0 8 K-series 152 22.80 30.87 53.54 9.71
		Ti 22 K-series 1130 21.90 29.67 17.19 0.98
Strand Carton & Internet		Ca 20 K-series 1/41 21.52 29.15 20.18 0.89 P 15 K-series 736 6.77 9.17 8.22 0.44
and a second of the second		V 23 K-series 22 0.43 0.58 0.31 0.12
and the start of the fill	1 1 A 1 A 1 A 1 A 1 A 1 A 1 A 1 A 1 A 1	Si 14 K-series 33 0.27 0.37 0.36 0.08
	1.	Al 13 K-series 12 0.14 0.19 0.20 0.07
-B3		N 7 K-series 0 0.00 0.00 0.00 0.00
man the train the states of th	and the second of the second	Na 11 K-series 0 0.00 0.00 0.00 0.00
10 µm EHT = 20.00 KV Signal A = SE1 Date :24 Jul 2019 22355 WD = 9.0 mm Mag = 2.00 K X Time :14:34:47	2 µm EHT = 15.00 kV Signal A = SE1 Date :2 WD = 9.0 mm Mag = 5.00 KX Time :1	8Nov 2019 21355 Total: 73.83 100.00 100.00
		El AN Series Net una. C norm. C Atom. C Error (1 Signa) [wt.4] [wt.4] [at.4] [ut.4] 0 8 K-series 550 29.05 39.29 61.27 7.88 Ca 20 K-series 4043 21.91 29.48 318.44 71 22 K-series 1468 12.44 16.43 8.77 0.53 F 15 K-series 2633 10.07 13.64 10.97 0.49 Al 13 K-series 29 0.14 0.19 0.17 0.65 St 14 K-series 30.01 0.15 0.14 0.05
B4		Na 11 K-series 11 0.11 0.15 0.16 0.06
		N 7 K-series 0 0.00 0.00 0.00 0.00
10 µm EHT = 20.00 KV Signal A = SE1 Date :24 Jul 2019 ZEXX WD = 9.5 mm Mag = 2.00 KX Time:14:41.21 ZEXX	2 µm EHT = 20.00 kV Signal A = SE1 Date :2 WD = 9.0 mm Mag = 5.00 K X Time :1	1.042019 ZIXX Total: 73.95 100.00 100.00

Figure 7. SEM images and EDX analysis with different magnifications for B1, B3, and B4 specimens.



Figure 8. SEM images of cross-section of the hybrid coatings for B1 and B2 specimens.



Figure 9. SEM images of cross-section of the hybrid coatings for B1, B3, and B4 specimens.



Figure 10. SEM images of cross-section of the hybrid coatings for B4 and B5 specimens.

X-ray analysis

The X-ray analysis of the B1 and B2 specimens are shown in Figure 11.

 Ti_6Al_4V and HA peaks were intensively observed In the XRD analyses of the B1 and B2 specimens. The P_2O_5 as the phosphate source and the Ca(NO₃) used as the calcium source led to the formation of HA. The B2 specimen additionally contained 5 wt.% chitosan reinforcement in difference to the B1 specimen provided the same peak values. The reason for this

was that the structure of chitosan resembles naturally obtained HA. Chitosan is prevalently preferred in the world of medicine as it is biodegradable, non-toxic and has qualities like cell adhesion and minimal reaction with foreign bodies. It also supports bone formation in in vitro and in vivo environments due to its osteoconductive property. The polysaccharide scaffold structure of chitosan is similar to glycosaminoglycan that are the main extracellular matrix component of bone.^{47,48} This is why it is frequently preferred in tissue engineering. In addition to these



Figure 11. The X-ray diffraction (XRD) pattern of B1 and B2 specimens.

properties, as it cannot show a bioactive property by itself, it is used together with HA.49,50 Other peaks observed in the XRD analysis were TiO₂, SiO₂, and Al₂O₃. The TiO₂ film that is formed creates an interface connection between the Ti₆Al₄V base and the HA coating.⁵¹ It is thought that the formation of the TiO₂ structure originates from the oxidation of titanium atoms during the sintering process. It is considered that the SiO_2 and $A_{l2}O_3$ structures come from the contents of expanded perlite. This is because the content of expanded perlite includes high ratios of SiO₂ and Al₂O₃.⁵² Al₂O₃, which is bioinert, is frequently used in studies, usually by reinforcing into HA. It was determined that, when used alongside HA, Al₂O₃ increased hardness and strength.53 Furthermore, it is frequently preferred in load-bearing hip prosthetics due to its good biocompatibility, corrosion resistance within body fluids and mechanical properties.⁵⁴ Moreover, the study by Li et al⁵⁵ found that SiO₂ was a component of bioactive glass and had the capacity to form apatite in SBF. It was supported by the information in the literature that the peaks obtained within the scope of our study were also used in other studies on biocompatibility. This supported the usability of expanded perlite (EP) in the place of bioglass in studies where bioglass is used as a reinforcement, biocompatibility is desired.

In vitro analysis

The SEM images of the B1, B2, and B3 specimens kept in Ringer's solution for 1, 10, and 25 days were given in Figure 12. EDX analysis was conducted for the specimens kept in the solution for 10 days (Figure 12).

The SEM analyses shown in Figure 12 reveal highly porous and rough structures. The property of bioactive glasses and ceramics is formation of a hydroxycarbonate apatite (HCA) layer with collagen tissue fibers. For the formation of the HCA layer, calcium-based phosphates and silicates are highly important.^{56,57} The content of expanded perlite and the structures of bioactive glasses and ceramics are highly similar.⁵⁸ When Figure 11 is examined, it is seen that the HCA layer (white layers) was formed even after 1 day in the B2 and B4 specimens. The EDS analyses revealed Ca, P, Mg, Na, O, and Si. Na ions were encountered in the EDX analysis of the specimens kept in the SBF (in vitro) solution. It is believed that this came from Ringer's solution. It is known that these elements, which show a very low toxic effect in the human body, have effects on bioactivity and biocompatibility.^{29,59}

The XRD analysis of the B2 specimen with and without in vitro solution treatment is presented in Figure 13.

Researchers have examined HA composites with glass reinforcement containing components such as SiO₂, CaO, MgO, Na₂O, and P₂O₅. The reason for glass reinforcement is that its chemical content is biocompatible and does not show toxic properties.⁶⁰⁻⁶³ It is an important fact that the content of bioactive glasses contains approximately 60% SiO₂ and a high ratio of CaO/P₂O₅. This is because, as this ratio increases, the surface becomes more active in fluid environments like body fluid and blood, and the interactions on the surface increase, then increasing the formation of the HCA layer.

That is, the surface becomes more biocompatible.62,63 Phases like calcium phosphate and calcium oxide formed during the formation of the HCA layer are important. For this reason, it is thought that these phases formed as shown in the XRD analysis increased the biocompatibility. TiO₂ peaks were observed in many intervals in the structure before and after the in-vitro treatment. This TiO₂ film formed creates an interface connection between the Ti₆Al₄V base and the HA coating layer.⁵² It was observed that the Al₂O₃ and SiO₂ peaks obtained before the in vitro treatment disappeared after the in-vitro treatment. HA $(Ca_{10}(PO_4)_6.5(OH)_2)$ peak was encountered in the XRD analysis conducted after the in vitro treatment. Four different types of calcium phosphate are used to increase dis- $Ca_8H_2(PO_4)_6.5(H_2O),$ solvability: $Ca_{2}(PO_{4})_{3}.n(H_{2}O),$ Ca₁₀(PO₄)₆.5(OH)₂, and CaHPO₄.2H₂O.⁶⁴ The most intense peak observed after HA was NaOH. The Na+ ion is the most prevalently found ion in Ringer's solution and human blood plasma.³¹ It is believed that, as Na+ ions remained on the surface of the specimen kept in Ringer's solution, these provided a peak in the XRD analysis.

FTIR analysis

FTIR analysis was conducted by keeping the B1, B2, and B4 specimens in Ringer's solution for 10 days. The results of the FTIR analysis are shown in Figure 14. In these spectra, it is easy to observe the characteristic absorption bands of HA. O-H bands in the structure of HA are seen at 3750, 630, and 335 cm⁻¹. The characteristic bands of phosphate compounds are encountered within 900 to 1200 and 543 to 601 cm⁻¹. CO₂ stretching bands at 2362 and 2366 cm⁻¹, H₂O stretching bands at 1636 cm⁻¹, and CO₃⁻² stretching bands at 1458 and 1413 cm⁻¹ in the crystal framework structure is observed.⁶⁵⁻⁶⁷ The Ca-PO₄ band is observed at 275 and 295 cm⁻¹, and the Ca-OH band is observed at 335 cm⁻¹. The characteristic bands of HA, PO₄³⁻ groups, are observed at 1090, 14 to 962, 18 to 601, 41 to 473, and 82 to 569 cm⁻¹.⁶⁸⁻⁷¹ IT was observed that carbonate groups



Figure 12. SEM images of affected surfaces of coated samples after in vitro tests in SBF and EDX analysis of B1, B2, and B3 specimens after 10 days.



Figure 13. The X-ray diffraction (XRD) pattern of B2 specimen (a) without in vitro (b) with 10 days in vitro treatment.

except for the chitosan-containing specimen. If a biomaterial is rougher, its surface area greats, and the implant adheres better to the surface. Increased surface area also positively affects bone formation.36,41,43

Conclusions

In the scope of this study, it was aimed to coat a hybrid composite (hydroxyapatite/perlite) onto the Ti₆Al₄V implant alloy by using the hydrothermal method. As a result, the following results were observed.

- In the microstructure analysis of the specimens with and • without preliminary heat treatment, it was revealed that porous and rough surfaces were formed. It was observed that the surface coatings had an interconnected and porous structure, but no homogenous distribution. Studies have observed that lack of a homogenous distribution on the coating surface makes osteointegration easier. Chitosan provides cell adhesion and multiplication.
- In the specimens without preliminary heat treatment, when the expanded perlite reinforcement ratio is higher than 10 wt.%, it was observed that the coating thickness decreased. In the specimens with preliminary head treatment, as the expanded perlite reinforcement increased, the coating thickness was observed to increase. The best coating thickness was obtained when the expanded perlite reinforcement ratio are 15 wt.% for specimen with preliminary heat treatment. So; the optimum expanded perlite reinforcement ratio for specimens without preliminary heat treatment is 10wt.% and it may exceed 15 wt.% for specimens with preliminary heat treatment.
- The result of the XRD analysis showed that the coating formed on the Ti₆Al₄V alloy had a structure of hydroxyapatite and expanded perlite. According to the results of the FTIR analysis, the band intervals formed belonged to the hydroxyapatite and phosphate characteristic bands.
- A hydroxyapatite peak was encountered in the XRD analysis conducted after the in vitro treatment, and it is thought that the layer formation process was going on.
- The increase in the surface roughness values after the in vitro treatment showed that our structure was apatited. With the surface roughness measurements made after the in vitro treatment, it was observed that the surface roughness of the chitosan-containing specimen decreased. Additionally, for the same measurements, it was observed that the surface roughness values increased by an increase in the expanded perlite reinforcement ratio by weight.

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Table 5. Surface roughness parameters before in vitro and after in vitro for B1, B2, and B3 specimens.

	Rq: 5,12	Rq: 0,87	Rq: 36,86							
	Rt: 13,93	Rt: 1,81	Rt: 16,46							
	1 500	1 1 1	C 11 11							
were found at 10	41 and 598 cm	⁻¹ , and they	formed bone-like							
apatite structures	apatite structures by solidifying in SBF.72 These data were									
compatible with the study by Lutišanová et al ⁷³ conducted with										
SBF. It is thought that the increase in the peak amplitudes										
observed in Figure 2 was caused by the increase in the rein-										
forcement compo	nents by weigh	t.								

Surface roughness results. The pre-in vitro and post-in vitro surface roughness measurements of the coated specimens are given in Table 5. Between the pre- and post-in vitro measurements, it was observed that the surface roughness increased

Figure 14. The FTIR spectrum of B1, B2, and B4 specimens.

	B1	B2	B4
Before in vitro	Ra: 1237	Ra: 0864	Ra: 0315
	Rz: 5327	Rz: 2445	Rz: 0892
	Rq: 1420	Rq: 0,79	Rq: 0363
	Rt: 5390	Rt: 2609	Rt: 0918
1 day	Ra: 3650	Ra: 0,82	Ra: 5,38
	Rz: 10,33	Rz: 2,33	Rz: 15,23
	Rq: 3,95	Rq: 0,88	Rq: 6,46
	Rt: 10,41	Rt: 2,36	Rt: 15,38
10 days	Ra: 2,94	Ra: 1,25	Ra: 3,99
	Rz: 8,33	Rz: 3,53	Rz: 11,29
	Rq: 3,06	Rq: 1,27	Rq: 4,64
	Rt: 8,41	Rt: 3,57	Rt: 11,4
25 days	Ra: 4,88	Ra: 0,63	Ra: 5,76
	Rz: 13,8	Rz: 1,79	Rz: 16,30
	Rq: 5,12	Rq: 0,87	Rq: 36,86
	Rt: 13,93	Rt: 1,81	Rt: 16,46



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