

Article

Multifunctional Hypercrosslinked Porous Organic Polymers Based on Tetraphenylethene and Triphenylamine Derivatives for High-Performance Dye Adsorption and Supercapacitor

Mohamed Gamal Mohamed ^{1,2} , Ahmed. F. M. EL-Mahdy ¹ , Tso-Shiuan Meng ¹,
Maha Mohamed Samy ¹ and Shiao-Wei Kuo ^{1,3,*} 

¹ Department of Materials and Optoelectronic Science, Center of Crystal Research, National Sun Yat-Sen University, Kaohsiung 80424, Taiwan; mgamal.eldin34@gmail.com (M.G.M.); ahmed1932005@gmail.com (A.F.M.E.-M.); s199483226@gmail.com (T.-S.M.); maha35430@gmail.com (M.M.S.)

² Chemistry Department, Faculty of Science, Assiut University, Assiut 71516, Egypt

³ Department of Medicinal and Applied Chemistry, Kaohsiung Medical University, Kaohsiung 807, Taiwan

* Correspondence: kuosw@faculty.nsysu.edu.tw; Tel.: +886-7-525-4099

Received: 22 September 2020; Accepted: 19 October 2020; Published: 21 October 2020



Abstract: We successfully prepared two different classes of hypercrosslinked porous organic polymers (HPPs)—the tetraphenylethene (TPE) and (4-(5,6-Diphenyl-1H-Benzimidazol-2-yl)-triphenylamine (DPT) HPPs—through the Friedel–Crafts polymerization of tetraphenylethene and 4-(5,6-diphenyl-1H-benzimidazol-2-yl)-triphenylamine, respectively, with 1,4-bis(chloromethyl)benzene (Ph-2Cl) in the presence of anhydrous FeCl₃ as a catalyst. Our porous materials exhibited high BET surface areas (up to 1000 m² g⁻¹) and good thermal stabilities. According to electrochemical and dyes adsorption applications, the as-prepared DPT-HPP exhibited a high specific capacitance of 110 F g⁻¹ at a current density of 0.5 A g⁻¹, with an excellent cycling stability of over 2000 times at 10 A g⁻¹. In addition, DPT-HPP showed a high adsorption capacity up to 256.40 mg g⁻¹ for the removal of RhB dye from water.

Keywords: tetraphenylethene; triphenylamine; porous polymers; dyes adsorption; supercapacitor

1. Introduction

Microporous organic polymers (MOPs) have attracted much attention as next-generation materials in the industry and academic areas because of their good thermal stability, low density, low regeneration energy, high pore volume and large BET (Brunauer–Emmett–Teller) surface area, synthetic diversity, and easier preparation [1–7]. MOPs have been used in many potential applications, such as water treatment, drug delivery, chemical sensing, heterogeneous catalysis, energy storage, hydrogen evolution, nanofiltration, oil scavenging, carbon dioxide reduction, gas separation and gas storage [8–17]. There are different kinds of MOPs, such as conjugated microporous polymers (CMPs) [15–18], covalent organic frameworks (COFs) [16–20], polymers of intrinsic microporosity (PIMs) [21–23], covalent triazine-based frameworks (CTFs) [24–28], and hypercrosslinked polymers (HCPs) [29–32].

Among MOPs, HCPs can be easily synthesized through the simple Friedel–Crafts alkylation reaction in the presence of building aromatic monomers and an external crosslinker [33–35]. These materials exhibited many unique features, such as large surface areas, excellent chemical and thermal properties, controlled reaction conditions, and easy functionalizations. Moreover, HCPs can be synthesized to afford a framework with a well-defined porous topology by controlling the length

of the crosslinker [36–41]. The preparation HCPs can be easily synthesized through three methods: (i) knitting rigid aromatic building blocks with an external crosslinker, such as 1,3,5-trichlorotriazine, 1,4-bis(chloromethyl)benzene or formaldehyde dimethyl acetal, (ii) the one-step polycondensation of functional monomers, and (iii) using post-crosslinking polymer precursors [39,42]. At present, HCP materials are still interesting porous polymers for energy storage applications because of their hierarchical porous structures, the extended π -conjugated systems in their frameworks, and their large surface areas [43–45]. The enhancement of the electronic conductivity, electroactive surface area and supercapacitor performance of the HCP materials can be achieved through the incorporation of heteroatoms such as sulfur, nitrogen, boron, and phosphorus into their polymeric frameworks, and also by using versatile building blocks for increasing their pore volumes and BET surface areas [46,47]. The preparation of heterocyclic imidazole derivatives has drawn much attention because of their unusual fluorescent, two-photon absorption, and phosphorescent properties. Such materials have been also used as optical materials, electroluminescent materials, three-dimensional optical data storages, and photographic materials [48–55].

Herein, we prepared two types of hypercrosslinked porous organic polymers, TPE-HPP and DPT-HPP, using the Friedel–Crafts polymerization of tetraphenylethene (TPE) and 4-(5,6-diphenyl-1H-benzimidazol-2-yl)-triphenylamine (DPT), respectively, with 1,4-bis(chloromethyl)benzene (Ph-2Cl) as an external crosslinker in the presence of anhydrous FeCl_3 . The chemical structures of these materials were confirmed by solid-state ^{13}C cross-polarization/magic-angle spinning (CP/MAS) nuclear magnetic resonance (NMR) and Fourier-transform infrared (FTIR) spectroscopy. Furthermore, the porous structure, morphology and crystallinity of these two porous materials were also investigated using different instruments, such as SEM, TEM, BET and PXRD measurements. We found that the resulting materials displayed high thermal stabilities and surface areas. Thus, we expect that our new hypercrosslinked polymers TPE-HPP and DPT-HPP could be used in energy storage and dye-uptake applications.

2. Experimental Section

2.1. Materials

Benzophenone (99%), potassium carbonate (K_2CO_3 , 99.995%), rhodamine B (RhB, 95%), and 1,4-bis(chloromethyl)benzene (98%) were purchased from Alfa Aesar. Titanium tetrachloride (TiCl_4 , 99.9%), zinc (Zn, 98%), phosphorus trichloride (POCl_3 , 99%), benzil (98%), ammonium acetate ($\text{CH}_3\text{COONH}_4$, 98%), acetic acid (CH_3COOH , 99%), and 1,2-dichloroethane (DCE, 99.8%) were purchased from Alfa Aesar. Triphenylamine (TPA, 98%) and ethyl acetate (EA) were ordered from Acros Organics. Anhydrous magnesium sulfate (MgSO_4 , 99.5%) and iron(III) chloride (FeCl_3 , 99.99%), tetrahydrofuran (THF), acetone, methanol (CH_3OH), dimethyl sulfoxide (DMSO), and dichloromethane (DCM), were purchased from Showa Corporation (Japan). 4-(Diphenylamino)benzaldehyde (TPA-CHO) was prepared as reported in the previous study [50].

2.2. Synthesis of Tetraphenylethylene (TPE)

A mixture of benzophenone (5.00 g, 0.027 mol) and zinc (6.52 g, 0.099 mol) in dry THF (200 mL) was stirred at 0 °C for 30 min. Then, TiCl_4 (5.43 mL, 0.049 mol) was added slowly over 10 min and then the reaction mixture was refluxed at 80 °C for 24 h. The reaction mixture was cooled to room temperature and quenched with a 10% aqueous solution of K_2CO_3 , followed by the removing of THF under vacuum. The crude product was extracted with ethylacetate (EA), and the organic EA layer was dried over anhydrous MgSO_4 , filtered and removed using a rotary evaporator. The crude product was then washed with ethanol and filtered to afford a white crystalline solid (yield: 97%). $^1\text{H-NMR}$ (500 MHz, δ , ppm, CDCl_3 , Figure S1): 7.05–7.15 (m, 20H). $^{13}\text{C-NMR}$ (125 MHz, δ , ppm, CDCl_3 , Figure S2): 140.7; 141.0; 131.3; 127.7; 126.4.

2.3. Synthesis of 4-(5,6-Diphenyl-1H-Benzimidazol-2-yl)-Triphenylamine (DPT)

A mixture of TPA-CHO (1.21 g, 5.74 mmol), benzil (1.00 g, 4.76 mmol), and ammonium acetate (1.84 g, 23.78 mmol) in glacial acetic acid (10 mL) was refluxed at 90 °C for 12 h under a N₂ atmosphere. The reaction mixture was then poured into an aqueous solution of sodium hydrogen sulfite (200 mL, 3 wt.%) to afford a yellow powder (yield: 80%). 500 MHz, δ , ppm, DMSO-*d*₆, (Figure S3): 12.54 (s, 1H), 7.99 (d, *J* = 8.4 Hz, 2H), 7.32–7.52 (m, 14H), 7.03–7.10 (m, 8H). ¹³C-NMR (125 MHz, δ , ppm, DMSO-*d*₆, Figure S4): 148.2, 147.8, 146.4, 130.5, 129.3, 127.3, 125.3, 125.2, 124.3, 123.6. *m/z* (high-resolution mass spectrometry, HRMS) found 463.2050 (M⁺), Anal. Calc. for C₃₃H₂₅N₃: 463.2048 (Figure S5).

2.4. Synthesis of TPE-HPP

A mixture of tetraphenylethylene (0.523 g, 1.573 mmol) and 1,4-bis(chloromethyl)benzene (1.652 g, 9.44 mmol) in 1,2-dichloroethane (20 mL) was added to anhydrous FeCl₃ (1.531 g, 9.44 mmol) under an N₂ atmosphere. The mixture was refluxed at 70 °C for 48 h. After cooling to room temperature, the insoluble residue was filtered off and successively washed several times with methanol, water, DCM, acetone and THF. The product was gained as a brown powder (yield: 75%).

2.5. Synthesis of DPT-HPP

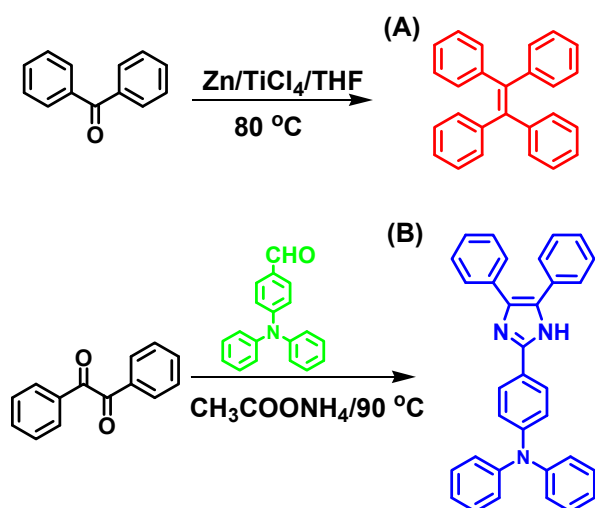
A mixture of 4-(5,6-Diphenyl-1H-benzimidazol-2-yl)-triphenylamine (0.521 g, 1.124 mmol) and 1,4-bis(chloromethyl)benzene (1.181 g, 6.744 mmol) in 1,2-dichloroethane (20 mL) was added to anhydrous FeCl₃ (1.093 g, 6.744 mmol) under a N₂ atmosphere. The mixture was refluxed at 70 °C for 48 h. After being cooled to room temperature, the insoluble residue was filtered off and successively washed several times with methanol, water, DCM, and tetrahydrofuran (THF). The product was gained as a dark green powder (yield: 80%).

3. Results and Discussions

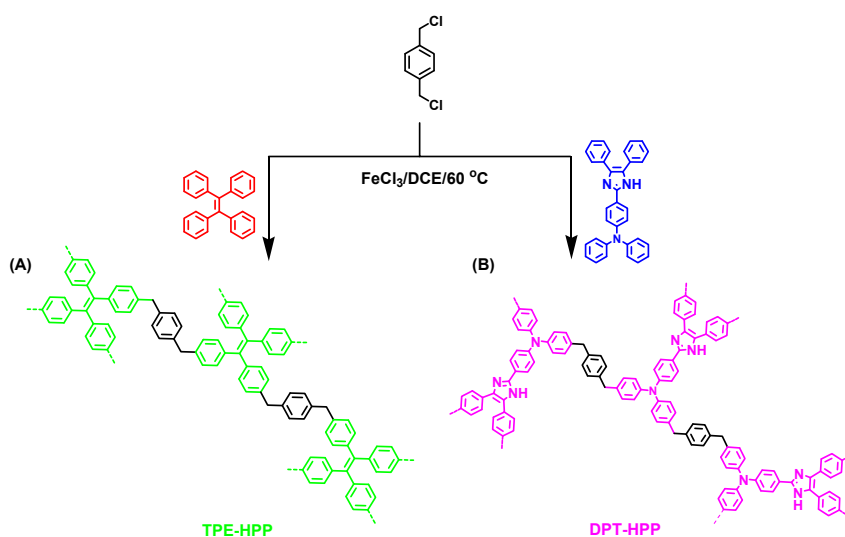
3.1. Synthesis of TPE-HPP and DPT-HPP

Scheme 1A,B show the synthetic methods for the preparation of TPE and DPT monomers. The TPE monomer was firstly prepared through the reaction of benzophenone with zinc in the presence of TiCl₄ [56,57], while the DPT monomer was synthesized by the reaction of benzil with 4-(diphenylamino)benzaldehyde (TPA-CHO) and ammonium acetate in acetic acid at 90 °C [50]. Scheme 2 represented the schematic method for preparing the two different HPPs—TPE and DPT HPPs—via a Friedel–Crafts reaction of TPE and DPT with 1,4-bis(chloromethyl)benzene in the presence of anhydrous FeCl₃ as a catalysis at 60 °C for 24 h. After the reactions were completed, the crude materials were washed several times with MeOH, THF, acetone, DCM, and DMF to remove any excess from the unreacted FeCl₃ and monomers. In addition, the TPE-HPP and DPT-HPP were insoluble in all organic solvents, such as DMF, acetone, THF, CH₂Cl₂ and MeOH.

The chemical molecular structures of TPE and DPT monomers were confirmed by using ¹H and ¹³C-NMR spectroscopy in CDCl₃ and DMSO-*d*₆, respectively (Figure 1). The ¹H-NMR spectrum (Figure 1A) of TPE monomer displayed proton signals centered at 7.14 and 7.04 ppm for the aromatic rings, while the ¹H-NMR spectrum (Figure 1B) of DPT displayed a characteristic proton signal at 9.74 ppm for the NH group, in addition to the aromatic proton signals at 8.04–7.04 ppm. Figure 1C,D presented the ¹³C-NMR analyses of TPE and DPT. The signals of the carbon nuclei of the aromatic rings in the TPE and DPT moieties appeared in the range 144.06–126.56 ppm for TPE and 130.56–122.25 ppm for DPT. Furthermore, the ¹³C-NMR spectrum of DPT (Figure 1D) displayed a signal at 147.31 ppm for the C=N carbon nuclei resonance. The molecular weight of the DPT was *m/z* 364, which is consistent with its calculated value (*m/z* 364.30) (Figure S5). The NMR and high-resolution mass results confirmed the successful synthesis of the TPE and DPT monomers.



Scheme 1. Preparation of (A) TPE and (B) DPT.



Scheme 2. Preparation of (A) TPE-HPP and (B) DPT-HPP.

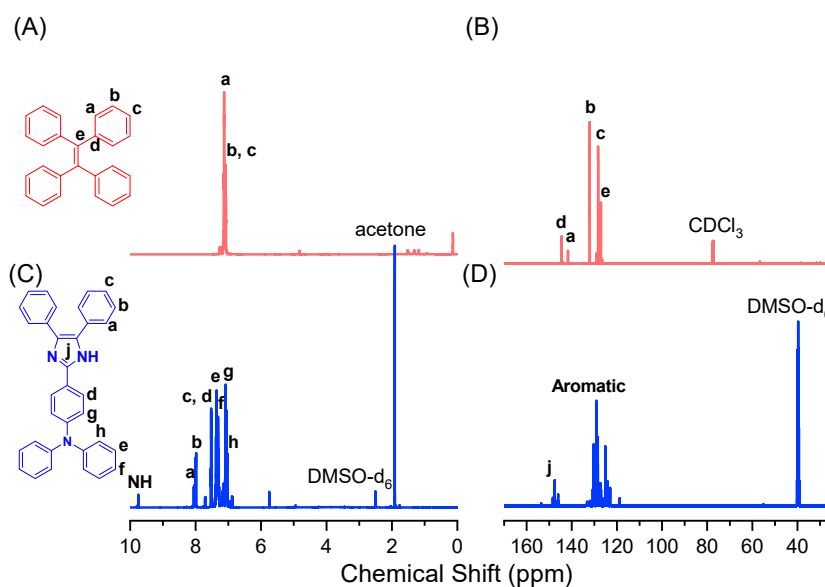


Figure 1. (A,B) $^1\text{H-NMR}$ spectra of TPE and DPT. (C,D) $^{13}\text{C-NMR}$ spectra of TPE and DPT.

Figure 2 showed the FTIR profiles of Ph-2Cl, TPE, DPT, TPE-HPP and DPT-HPP, recorded at room temperature. The FTIR spectra of Ph-2Cl, TPE, DPT, TPE-HPP and DPT-HPP showed a characteristic absorption band in the range 3050–3084 cm^{-1} for the stretching C–H aromatic groups, in addition to an absorption band at 1600 cm^{-1} for the C=C bonds. The FTIR spectra of TPE-HPP and DPT-HPP (Figure 2D,E) featured absorptions signals in the range 2970–2850 cm^{-1} representing their aliphatic C–H units. These results confirmed the successful incorporation of 4-bis(chloromethyl)benzene as an external crosslinker into the polymer matrices. In addition, the absorption bands for the stretching NH groups of the DPT and DPT-HPP appeared at 3435 cm^{-1} (Figure 2C,E).

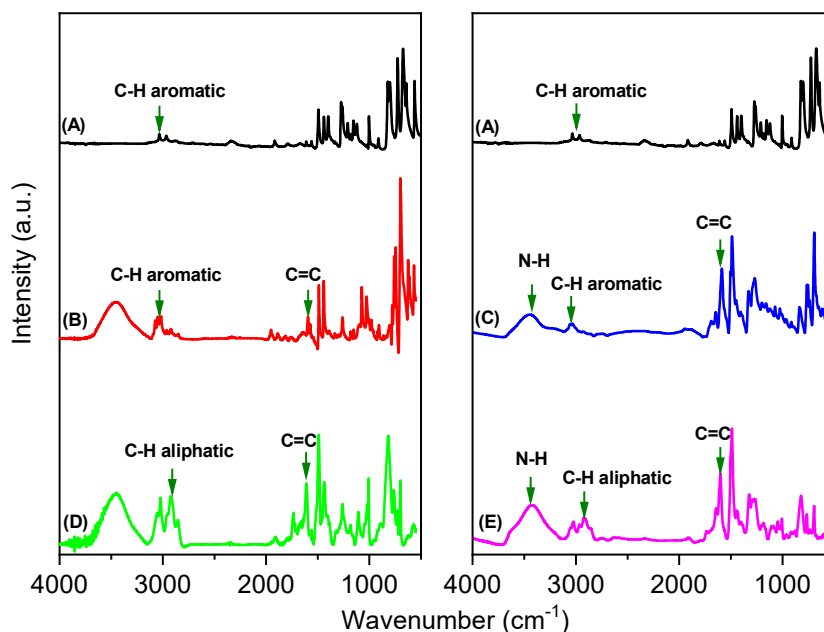


Figure 2. FTIR analyses of (A) Ph-2Cl, (B) TPE, (C) DPT, (D) TPE-HPP and (E) DPT-HPP.

Furthermore, the chemical structures of TPE-HPP and DPT-HPP were checked by using the solid-state ^{13}C CP/MAS NMR measurement, as shown in Figure 3. The characteristic signals for the TPE-HPP appeared at 139.67–128.77 and 39.58 ppm, respectively, for its carbon atoms in the aromatic rings and methylene units (Figure 3A), while such carbon atoms appeared at 139.49–128.98 ppm and 39.79 ppm, respectively, for the DPT-HPP (Figure 3B).

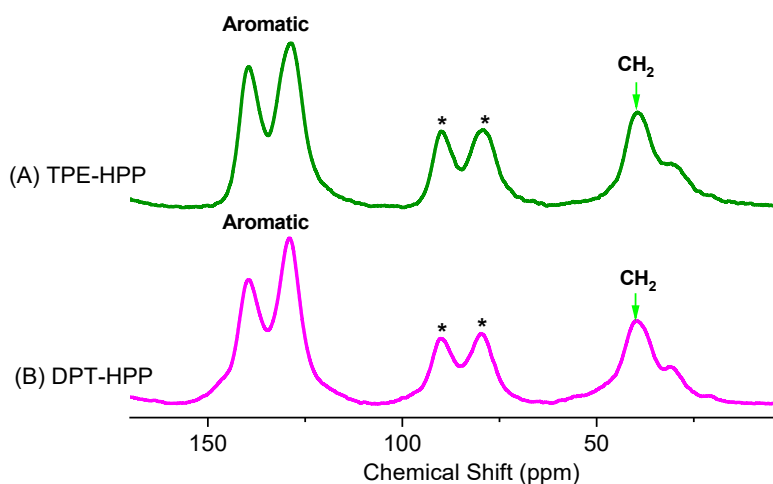


Figure 3. Solid-state ^{13}C CP/MAS NMR spectra of (A) TPE-HPP and (B) DPT-HPP where * is the side band of solid state NMR.

3.2. TGA, XRD, BET, SEM, and TEM Analyses for TPE-HPP and DPT-HPP

The thermal stabilities of TPE-HPP and DPT-HPP and their corresponding monomers were examined by using TGA analyses under an N₂ atmosphere, as presented in Figure 4A,B. The TGA measurements showed that the T_{d5} , T_{d10} and char yield values were 158, 273 °C and 62% for the Ph-2Cl monomer, respectively, while they were 246 °C, 263 and 0%, for the TPE monomer, respectively. The DPT monomer exhibited decomposition temperatures (T_{d5} and T_{d10}) of 186 and 272 °C, respectively, and a char yield value of 11%. After the crosslinking of TPE and DPT with Ph-2Cl through Friedel–Crafts reactions, the values of T_{d5} for the resulting TPE-HPP and DPT-HPP were 387 and 263 °C, respectively, and the values of T_{d10} were 551 and 481 °C, respectively. In addition, the char yields of TPE-HPP and DPT-HPP were 75% and 74%, respectively. The values of T_{d5} and T_{d10} , and the char yield for TPE-HPP, DPT-HPP and their corresponding monomers, are summarized in Table 1.

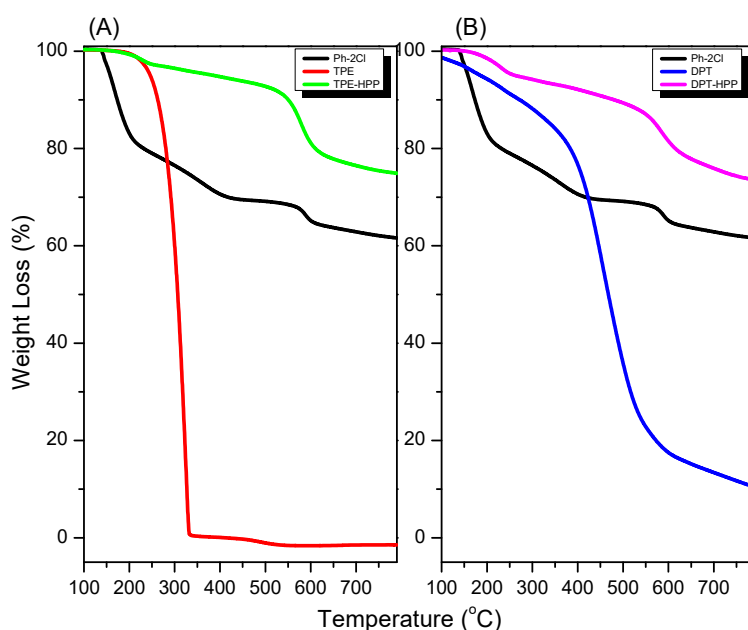


Figure 4. (A,B) TGA analyses of Ph-2Cl, TPE, DPT, TPE-HPP, and DPT-HPP.

Table 1. Thermal stability and porosity properties of TPE-HPP and DPT-HPP, and comparison with their corresponding monomers.

Sample	T_{d5} (°C)	T_{d10} (°C)	Char Yield (wt.%)	Surface Area (m ² g ⁻¹)	Pore Size (nm)
Ph-2Cl	158	273	62	-	-
TPE	246	263	0	-	-
DPT	186	272	11	-	-
TPE-HPP	387	551	75	922	1.01–1.98
DPT-HPP	263	481	74	1230	1.21–2.15

Based on XRD analyses, both TPE-HPP and DPT-HPP were shown to be amorphous materials without crystalline characteristics (Figure S6). Figure 5A,B show the N₂ adsorption/desorption and pore size distribution curves of the TPE-HPP and DPT-HPP, respectively, recorded at 1 bar. The results revealed that both TPE-HPP and DPT-HPP featured type II with type IV adsorption isotherms, according to the IUPAC classification. In addition, the BET profiles of TPE-HPP and DPT-HPP displayed that the N₂ uptake capacities were increased at low-pressure and high-pressure regions, indicating the presence of micro and meso pores in the TPE-HPP and DPT-HPP frameworks.

The BET surface area, total pore volume, and pore size diameter of the TPE-HPP were $922 \text{ m}^2 \text{ g}^{-1}$, $0.86 \text{ cm}^3 \text{ g}^{-1}$ and 1.10–1.98 nm, respectively (Figure 5C), while the DPT-HPP revealed a BET surface area, total pore volume and pore size diameter of $1230 \text{ m}^2 \text{ g}^{-1}$, $1.03 \text{ cm}^3 \text{ g}^{-1}$ and 1.21–2.15 nm, respectively (Figure 5D). Comparing with the reported HCPs, the specific surface area of our DPT-HPP ($1230 \text{ m}^2 \text{ g}^{-1}$) was among the highest values of the other hypercrosslinked microporous polymers; for example, this includes the tetraphenylanthraquinone-based HCP ($1980 \text{ m}^2 \text{ g}^{-1}$), binaphthol-based HCP ($1015 \text{ m}^2 \text{ g}^{-1}$), and a porous organic polymer containing triazine and carbazole moieties ($913 \text{ m}^2 \text{ g}^{-1}$) [20,58–60]. Furthermore, the SEM and TEM measurements were performed to confirm the porous properties of the TPE-HPP and DPT-HPP, as displayed in Figure 6. The SEM imaging of TPE-HPP and DPT-HPP showed the existence of interparticulate porosity between the agglomerated primary microgel particles (Figure 6A,B), while the high-resolution transmission electron imaging (Figure 6C,D) of the TPE-HPP and DPT-HPP showed the presence of an alternate bright and dark microstructure with a hole shape.

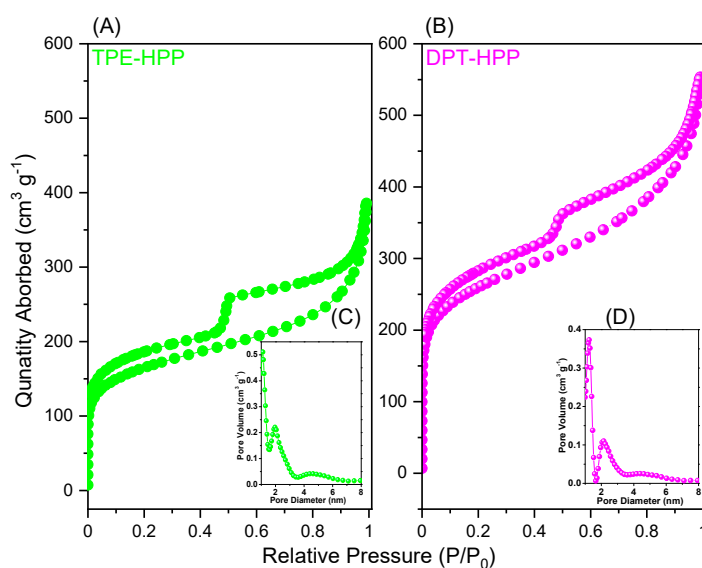


Figure 5. N_2 adsorption/desorption isotherms of (A) TPE-HPP and (B) DPT-HPP. Pore size distribution curves of (C) TPE-HPP and (D) DPT-HPP.

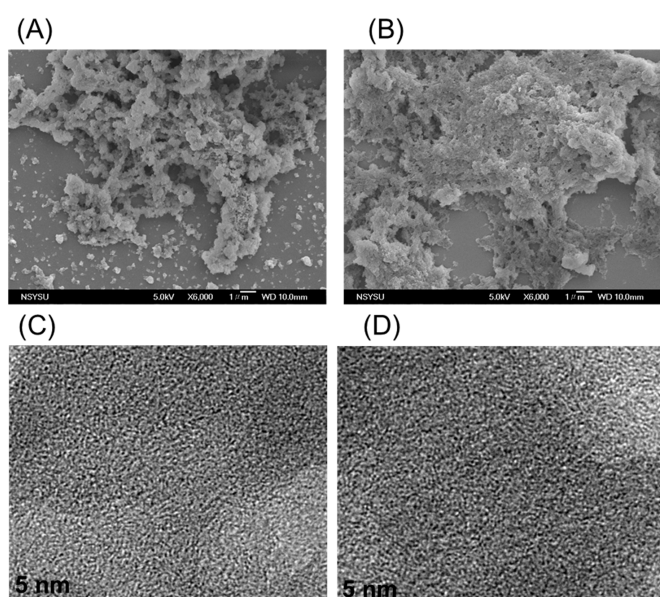


Figure 6. SEM images of (A) TPE-HPP and (B) DPT-HPP. TEM images of (C) TPE-HPP and (D) DPT-HPP.

3.3. CO₂ Uptake, Dye Adsorption and Electrochemical Performance

Because our porous materials had high surface areas (up to 1000 m² g⁻¹), large pore-volumes, and meso and microporous architectures, we expected that our new hypercrosslinked polymers TPE-HPP and DPT-HPP could be used as potential candidates for gas capture, energy storage, and the removal of dyes from the waste-water. CO₂ isotherm measurements at 298 and 273 K (Figure 7A,B) were performed to investigate the CO₂ adsorption capacity of TPE-HPP and DPT-HPP. At 298 K and 1 bar pressure (Figure 7A), the CO₂ adsorption capacity reached 1.65 and 1.85 mmol g⁻¹ for the TPE-HPP and DPT-HPP, respectively, while the CO₂ capture was 1.91 and 2.06 mmol g⁻¹ for the TPE-HPP and DPT-HPP, respectively, at 273 K (Figure 7B). Both TPE-HPP and DPT-HPP exhibited higher values of CO₂ uptake than that of the tetraphenylanthraquinone-based HCPs such as An-CPOP-1 (1.30 mmol at 298 K and 1.40 mmol g⁻¹ at 273 K) and An-CPOP-2 (1.40 mmol g⁻¹ at 298 K and 1.52 mmol g⁻¹ at 273 K) [20]. In addition, our materials showed higher CO₂ capture than silole-based HCPs (PDMPAS, 1.02 mmol g⁻¹ at 298 K), COF-102 (1.56 mmol g⁻¹ at 273 K) and C2M1-Al (1.44 mmol g⁻¹ at 273 K) [61–63]. As shown by the above results, DPT-HPP had an excellent performance for CO₂ capture compared with the previously reported HCPs materials, which can be attributed to the strong dipole–quadrupole interactions between the DPT-HPP materials and the CO₂ molecules.

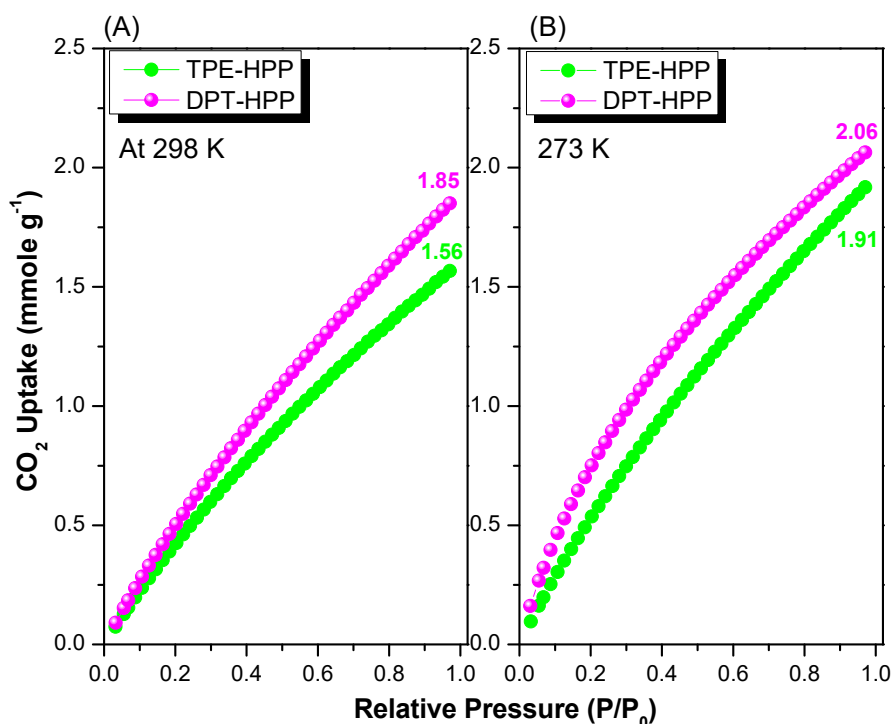


Figure 7. CO₂ uptake of TPE-HPP and DPT-HPP at (A) 298 K and (B) 273 K.

It has been previously reported that the nitrogen atoms in the organic polymers can strongly interact with dye molecules through non-covalent interactions, such as dipole–dipole, and electrostatic interactions. This interaction ability of nitrogen atoms is due to their basic properties, which resulted from their high charge densities. Therefore, researchers documented that the incorporation of N-functionalized groups, such as azo (N=N) and imine (C=N), into organic polymers can dramatically enhance their adsorption efficiencies towards small organic pollutants in water treatment applications [64]. In addition, the surface area of the adsorbent materials also plays an important role in determining the adsorption efficiency; increasing the surface area massively enhances the adsorption efficiency of the adsorbents [65–68]. Therefore, we have evaluated the suitability of our hypercrosslinked polymers TPE-HPP and DPT-HPP to be used as adsorbents for water treatment through testing their adsorption capacities to Rhodamine B (RhB) dye. For quantitative evaluation

of the adsorptions of TPE-HPP and DPT-HPP polymers, we tracked the RhB adsorption processes by recording the ultraviolet-visible spectra of aqueous dye solutions at 0, 15, 30, 60 and 90 min after the addition of our hypercrosslinked polymers. As shown in Figure 8A, after the addition of 5 mg of TPE-HPP hypercrosslinked polymer to aqueous RhB solution, the intensity of the maximum adsorption peak of RhB ($\lambda_{\max} = 554 \text{ nm}$) gradually decreased as the time increased from 0 to 30 min, and vanished entirely within 60 min. The insert in Figure 8C shows images of RhB/TPE-HPP supernatants at different times—here, the color of the aqueous RhB solution has switched from red-violet to colorless within 60 min. As a result, the TPE-HPP hypercrosslinked polymer revealed high performance for the elimination of RhB dye from water; the removal efficiency of RhB using TPE-HPP reached up to 97% within 60 min (Figure 8C). As shown in Figure 8B, the intensity of the maximum adsorption peak of RhB gradually decreased from 0 to 10 min after the addition of 5 mg of DPT-HPP hypercrosslinked polymer to aqueous RhB solution, and then completely disappeared after 15 min. This quick adsorption of the DPT-HPP polymer was also confirmed by switching the color of the aqueous RhB solution from red-violet to colorless within 15 min, as shown in Figure 8D. Thus, our DPT-HPP hypercrosslinked polymer exhibited a higher efficiency for the elimination of organic RhB from water, and the removal efficiency of RhB using DPT-HPP polymer reached up to 97% within 15 min (Figure 8D). These results indicated that the presence of nitrogen atoms in the backbone structure of the DPT-HPP polymer and its higher surface area ($1230 \text{ m}^2 \text{ g}^{-1}$) can enhance the adsorption process and provide a higher dye removal efficiency than that of the TPE-HPP polymer, which does not contain nitrogen atoms and has a lower surface area ($922 \text{ m}^2 \text{ g}^{-1}$).

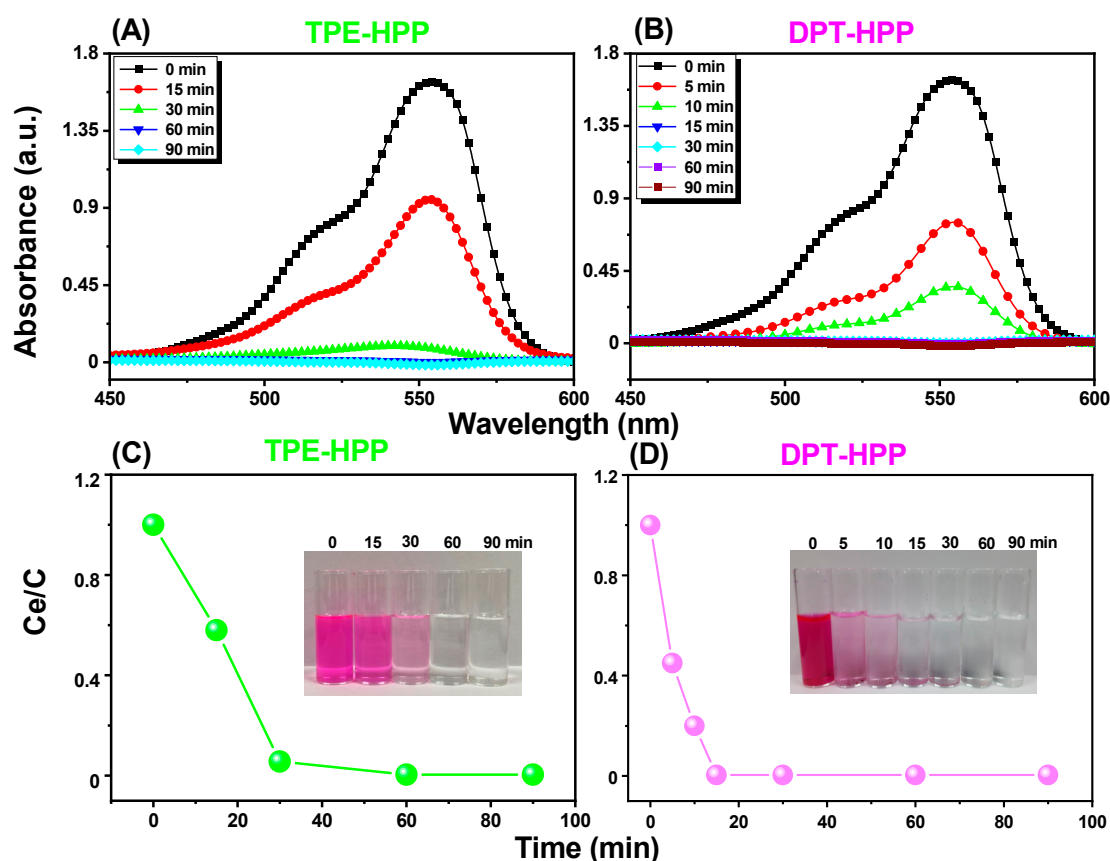


Figure 8. (A,B) UV-Vis spectra of aqueous RhB recorded at various periods after the addition of TPE-HPP and DPT-HPP (initial concentration of both RhB: 25 mg L^{-1}). (C,D) Adsorption rates of RhB (initial concentration: 25 mg L^{-1}) after the addition of TPE-HPP and DPT-HPP at various periods. Insets: Corresponding photographs. C_e (mg L^{-1}) is the equilibrium concentration of RhB at various measured times; C (mg L^{-1}) is the initial concentration of RhB (0 min).

We implemented the Langmuir isothermal model [69] to investigate the kinetic adsorption behavior of the dye RhB on the surface of hypercrosslinked polymers TPE-HPP and DPT-HPP. By fitting the equilibrium adsorption data of our synthesized polymers, the resultant Langmuir isothermal models revealed a linear relationship (C_e/Q_e vs. C_e) for both TPE-HPP and DPT-HPP polymers (Figure 9A). As summarized in Table 2, the correlation coefficients (R_L^2) of these Langmuir models were 0.99266 for TPE-HPP and 0.96436 for DPT-HPP polymer. According to the resultant Langmuir equations, we calculated the maximum adsorption capacities (Q_m) of our hypercrosslinked polymers, which were found to be 107.41 mg g⁻¹ for TPE-HPP and 256.40 mg g⁻¹ for DPT-HPP (Figure 9B). The Q_m values confirmed the higher adsorption capacity of DPT-HPP over TPE-HPP. Furthermore, the cyclic adsorption–regeneration test of aqueous RhB dye solution has been utilized to study the reusability of TPE-HPP and DPT-HPP (Figure S7). The adsorption efficiencies of TPE-HPP and DPT-HPP exhibited negligible change after four regeneration cycles, confirming the suitability of our TPE-HPP and DPT-HPP as efficient adsorbents for the removal of RhB from water.

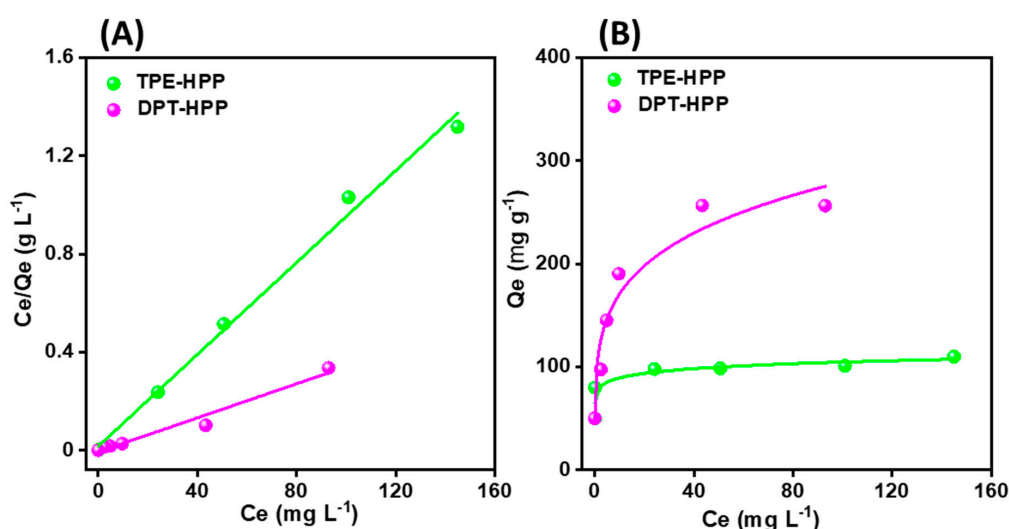


Figure 9. (A) Langmuir isothermal models and (B) adsorption isothermal curves for the adsorption of the RhB on TPE-HPP and DPT-HPP.

Table 2. Fitted parameters for the adsorption of RhB on the hypercrosslinked polymers TPE-HPP and DPT-HPP.

	Q_m (mg g ⁻¹)	K_L	R_L^2
TPE-HPP	107.41	0.5649	0.99266
DPT-HPP	256.40	0.6250	0.96436

To investigate and understand the electrochemical ability of the two porous materials (TPE-HPP and DPT-HPP) in a three-electrode system with 1 M KOH as an aqueous electrolyte, both cyclic voltammetry (CV) and galvanostatic charge-discharge (GCD) were carried out, as shown in Figure 10. The CV curves (Figure 10A,B) of TPE-HPP and DPT-HPP were recorded in the potential window from 0.00 to −1.00 V (vs. Hg/HgO) with different sweep rates from 5 to 200 mV s⁻¹. As shown in their CV profile, both samples displayed humps with rectangle-like shapes, indicating that their capacitive performance is derived from electric double-layer capacitance (EDLC). In addition, the rectangle-like shapes (Figure 10A,B) of these materials were maintained even at high sweep rates and current densities, confirming the facile kinetics and excellent rate capabilities of TPE-HPP and DPT-HPP. The GCD profiles (Figure 10C,D) of the TPE-HPP and DPT-HPP revealed that these materials possessed triangular shapes with a slight bend, suggesting a combination of pseudocapacity and EDLC features for these two materials, which arises from the presence of the electroactive methylene groups [46]. In addition,

the DPT-HPP sample possessed the redox-active triphenylamine units that can undergo a reversible redox process [30]. As previously reported, in two successive one-electron steps, the tetraphenylethene (TPE) can undergo a redox process, forming its reduced form [70], while the triphenylamine (TPA) possesses a nitrogen atom as its electroactive center connected, to three electron-rich phenyl rings, in which a reversible radical redox process occurs during charge and discharge processes [39,71,72].

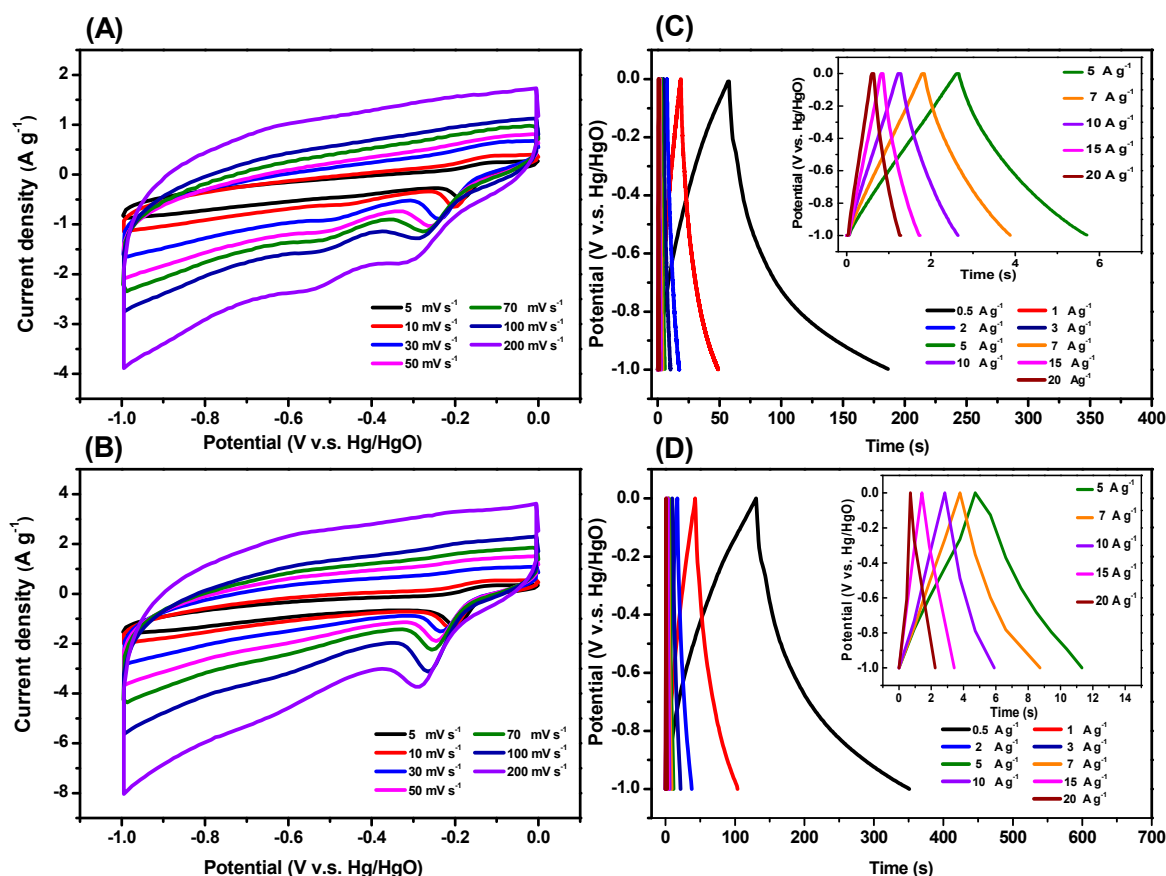


Figure 10. (A,B) CV and (C,D) GCD profiles, recorded in 1 M KOH, of the (A,C) TPE-HPP and (B,D) DPT-HPP.

As presented in the GCD curves, DPT-HPP had a longer discharging time compared to that of TPE-HPP. Thus, the value of the specific capacitance of the DPT-HPP (110.50 F g^{-1}) was higher than that of the TPE-HPP (67.00 F g^{-1}), recorded at a current density of 0.5 A g^{-1} (Figure 11). This higher specific capacitance of the DPT-HPP can be attributed to its higher surface area ($1230 \text{ m}^2 \text{ g}^{-1}$), large pore size (2.15 nm), pore volume ($1.03 \text{ cm}^3 \text{ g}^{-1}$) and the fact of its containing heteroatoms such as N atoms in its framework. We found that Car-TPA, Car-TPP, Car-TPT COFs, TPA-COF-1, TPA-COF-2, TPA-COF-3, TPT-COF-4, TPT-COF-5 and TPTCOF-6 exhibited the specific capacitances of 13.6, 14.5, 17.4 F g^{-1} , 51.3, 14.4, 5.1, 2.4, 0.34 and 0.24 F g^{-1} , respectively, at 0.2 A g^{-1} [30,73]. Khattak and his group revealed that the specific capacitance value of DABTFP COF was 98 F g^{-1} at 0.5 A g^{-1} [74]. Furthermore, the TpPa-COF/PANI composite exhibited a capacitance of 95 F g^{-1} at 0.2 A g^{-1} [75]. In addition, DAAQ-TFP COF showed capacitance of 48 F g^{-1} at 0.1 A g^{-1} [76]. We also reported that An-CPOP-1 and An-CPOP-2 exhibited high specific capacitances of 72.75 and 98.4 F g^{-1} , respectively, at 0.5 A g^{-1} , and excellent cycling stability [39].

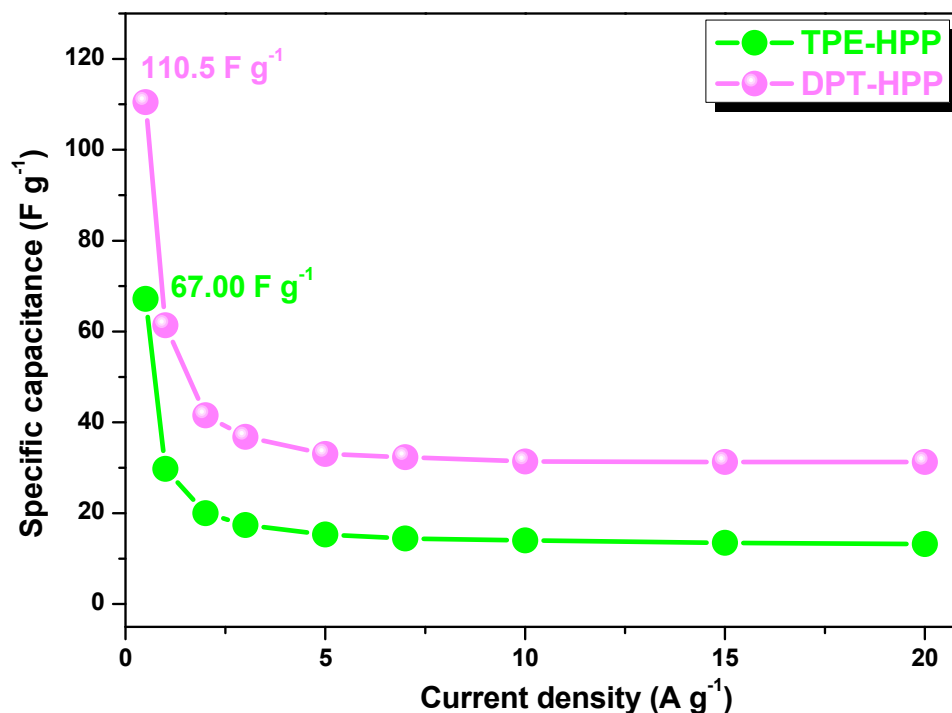


Figure 11. Specific capacitances profile of TPE-HPP and DPT-HPP recorded at various current densities.

Table S1 summarizes the specific surface areas and specific capacitances of TPE-HPP, DPT-HPP, other reported POPs, and related materials for supercapacitor applications. Furthermore, the cycling stability of TPE-HPP and DPT-HPP was examined over 2000 repetitions at 10 A g⁻¹ (Figure 12), which revealed that TPE-HPP and DPT-HPP had excellent cycling stabilities and capacitance retentions of 86% for TPE-HPP and 92% for DPT-HPP. According to the Ragone plot (Figure S8), these two TPE-HPP and DPT-HPP electrodes showed good energy and power densities.

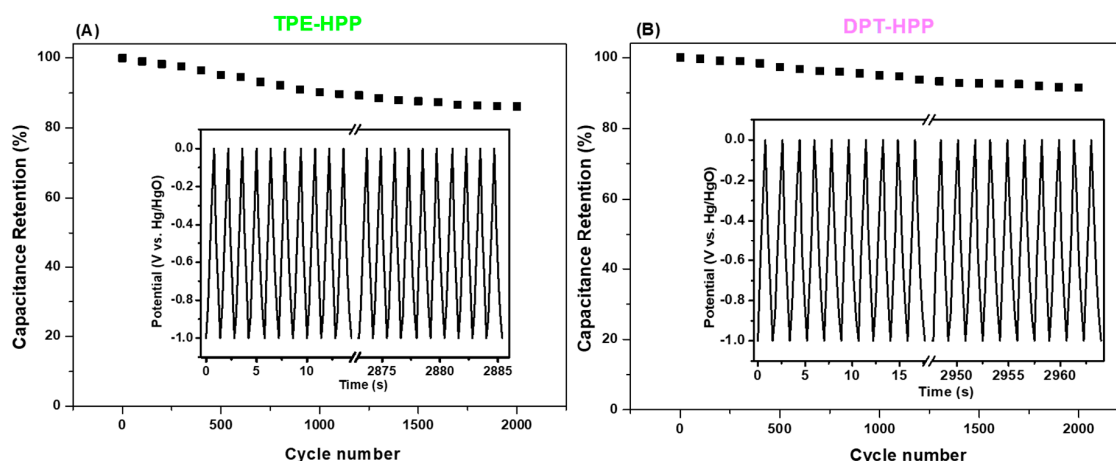


Figure 12. Cycling stability performance of (A) TPE-HPP and (B) DPT-HPP measured at a current density of 10 A g⁻¹ for 2000 cycles.

4. Conclusions

According to the FTIR and NMR analyses, two types of HCPs were successfully synthesized through the simple and effective Friedel–Crafts polymerization of tetraphenylethene and 4-(5,6-diphenyl-1H-benzimidazol-2-yl)-triphenylamine with 1,4-bis(chloromethyl)benzene in the presence of FeCl₃ as a catalyst. Based on dyes and electrochemical measurements, we found

that the as-prepared DPT-HPP displayed a significant adsorption capacity (up to 256.40 mg g⁻¹) for RhB dye from water, and a supercapacitor performance (up to 110.50 F g⁻¹ at a current density of 0.5 A g⁻¹) with an excellent cycling stability (up to 95.3% capacitance retention over 2000 cycles). Consequently, our new DPT-HPP can be considered as an excellent candidate for energy storage and dye adsorption applications.

Supplementary Materials: The following are available online at <http://www.mdpi.com/2073-4360/12/10/2426/s1>, **Figure S1:** ¹H NMR spectrum of TPE. **Figure S2:** ¹³C NMR spectrum of TPE. **Figure S3:** ¹H NMR spectrum of DPT. **Figure S4:** ¹³C NMR spectrum of DPT. **Figure S5:** FT-MS pattern of DPT. **Figure S6:** XRD pattern of (A) TPE-HPP and (B) DPT-HPP. **Figure S7:** Reusability of (A) TPE-HPP and (B) DPT-HPP for the removal of RhB MB within 60 min. **Figure S8:** Ragone plots of the energy density and power density of the (A) TPE-HPP and (B) DPT-HPP electrodes in 1 M KOH. **Table S1:** Comparison between the specific surface area/specific capacitance of TPE-HPP and DPT-HPP with those of previously reported POPs and related materials for supercapacitor applications.

Author Contributions: M.G.M., A.F.M.E.-M., T.-S.M., M.M.S. did the experiment, and S.-W.K. contributed to the literature review and the writing of this paper. All authors have read and agreed to the published version of the manuscript.

Funding: This study was supported financially by the Ministry of Science and Technology, Taiwan, under contracts MOST 106-2221-E-110-067-MY3, 108-2218-E-110-013-MY3, 108-2638-E-002-003-MY2, and 108-2221-E-110-014-MY3.

Conflicts of Interest: The authors declare no conflict of interest.

References

1. Wang, T.X.; Liang, H.P.; Anito, D.A.; Ding, X.; Han, B.H. Emerging applications of porous organic polymers in visible-light photocatalysis. *J. Mater. Chem. A* **2020**, *8*, 7003–7034. [[CrossRef](#)]
2. Furukawa, H.; Yaghi, O.M. Storage of Hydrogen, Methane, and Carbon Dioxide in Highly Porous Covalent Organic Frameworks for Clean Energy Applications. *J. Am. Chem. Soc.* **2009**, *131*, 8875–8883. [[CrossRef](#)] [[PubMed](#)]
3. Sang, Y.; Chen, G.; Huang, J. Oxygen-rich porous carbons from carbonyl modified hyper-cross-linked polymers for efficient CO₂ capture. *J. Polym. Res.* **2020**, *27*, 36. [[CrossRef](#)]
4. Lv, H.; Wang, W.; Li, F. Porous organic polymers with built-in N-heterocyclic carbenes: Selective and efficient heterogeneous catalyst for the reductive N-formylation of amines with CO₂. *Chem. Eur. J.* **2018**, *24*, 16588–16594. [[CrossRef](#)] [[PubMed](#)]
5. Aly, K.I.; Sayed, M.M.; Mohamed, M.G.; Kuo, S.W.; Younis, O. A facile synthetic route and dual function of network luminescent porous polyester and copolyester containing porphyrin moiety for metal ions sensor and dyes adsorption. *Micropor. Mesopor. Mater.* **2020**, *298*, 110063. [[CrossRef](#)]
6. Sychalska, K.; Zając, D.; Baluta, S.; Halicka, K.; Cabaj, J. Functional Polymers Structures for (Bio)Sensing Application—A Review. *Polymers* **2020**, *12*, 1154. [[CrossRef](#)] [[PubMed](#)]
7. Byun, Y.; Je, S.H.; Talapanani, S.N.; Coskun, A. Advances in Porous Organic Polymers for Efficient Water Capture. *Chem. Eur. J.* **2019**, *25*, 10262–10283. [[CrossRef](#)]
8. Abuzeid, H.R.; EL-Mahdy, A.F.M.; Ahmed, M.M.M.; Kuo, S.W. Triazine-functionalized covalent benzoxazine framework for direct synthesis of N-doped microporous carbon. *Polym. Chem.* **2019**, *10*, 6010–6020. [[CrossRef](#)]
9. Ignacz, G.; Fei, F.; Szekely, G. Ion-Stabilized Membranes for Demanding Environments Fabricated from Polybenzimidazole and Its Blends with Polymers of Intrinsic Microporosity. *ACS Appl. Nano Mater.* **2018**, *11*, 6349–6356. [[CrossRef](#)]
10. Mohamed, M.G.; Atayde, E.C., Jr.; Matsagar, B.M.; Na, J.; Yamauchi, Y.; Wu, K.C.W.; Kuo, S.W. Construction Hierarchically Mesoporous/Microporous Materials Based on Block Copolymer and Covalent Organic Framework. *J. Taiwan Inst. Chem. Eng.* **2020**, *122*, 180–192. [[CrossRef](#)]
11. Topuz, F.; Abdulhamid, M.A.; Nunes, S.P.; Szekely, G. Hierarchically porous electrospun nanofibrous mats produced from intrinsically microporous fluorinated polyimide for the removal of oils and non-polar solvents. *Environ. Sci. Nano* **2020**, *7*, 1365–1372. [[CrossRef](#)]
12. Cousins, K.; Zhang, R. Highly Porous Organic Polymers for Hydrogen Fuel Storage. *Polymers* **2019**, *11*, 690. [[CrossRef](#)] [[PubMed](#)]

13. Jansen, J.C.; Esposito, E.; Fuoco, A.; Carta, M. Microporous Organic Polymers: Synthesis, Characterization, and Applications. *Polymers* **2019**, *11*, 844. [[CrossRef](#)]
14. Mohamed, M.G.; EL-Mahdy, A.F.M.; Takashi, Y.; Kuo, S.W. Ultrastable conductive microporous covalent triazine frameworks based on pyrene moieties provide high-performance CO₂ uptake and supercapacitance. *New J. Chem.* **2020**, *44*, 8241–8253. [[CrossRef](#)]
15. Mohamed, M.G.; Ebrahim, S.M.; Hammam, A.S.; Kuo, S.W.; Aly, K.I. Enhanced CO₂ capture in nitrogen-enriched microporous carbons derived from Polybenzoxazines containing azobenzene and carboxylic acid units. *J. Polym. Res.* **2020**, *27*, 197. [[CrossRef](#)]
16. Cui, Y.; Du, J.; Liu, Y.; Yu, Y.; Wang, S.; Pang, H.; Liang, Z.; Yu, J. Design and synthesis of multifunctional porous N-rich polymer containing s-triazine and Troöger's base for CO₂ adsorption, catalysis and sensing. *Polym. Chem.* **2018**, *9*, 2643–2649. [[CrossRef](#)]
17. Jiang, W.; Yue, H.; Shuttleworth, P.S.; Xie, P.; Li, S.; Guo, J. Adamantane-Based Micro- and Ultra-Microporous Frameworks for Efficient Small Gas and Toxic Organic Vapor Adsorption. *Polymers* **2019**, *11*, 486. [[CrossRef](#)]
18. McKeown, N.B.; Budd, P.M.; Book, D. Microporous polymers as potential hydrogen storage materials. *Macromol. Rapid Commun.* **2007**, *28*, 995–1002. [[CrossRef](#)]
19. Li, J.G.; Lee, P.Y.; Ahmed, M.M.M.; Mohamed, M.G.; Kuo, S.W. Varying the Hydrogen Bonding Strength in Phenolic/PEO-b-PLA Blends Provides Mesoporous Carbons Having Large Accessible Pores Suitable for Energy Storage. *Macromol. Chem. Phys.* **2020**, *221*, 2000040. [[CrossRef](#)]
20. Mohamed, M.G.; Hung, W.S.; EL-Mahdy, A.F.M.; Ahmed, M.M.M.; Dai, L.; Chen, T.; Kuo, S.W. High-Molecular-Weight PLA-b-PEO-b-PLA Triblock Copolymer Templated Large Mesoporous Carbons for Supercapacitors and CO₂ Capture. *Polymers* **2020**, *12*, 1193. [[CrossRef](#)]
21. Wang, S.; Liu, Y.; Ye, Y.; Meng, X.; Du, J.; Song, X.; Liang, Z. Ultrahigh volatile iodine capture by conjugated microporous polymer based on N,N,N',N'-tetraphenyl-1,4-phenylenediamine. *Polym. Chem.* **2019**, *10*, 2608–2615.
22. Li, W.T.; Zhuang, Y.T.; Wang, J.Y.; Yang, T.; Yu, Y.L.; Chen, M.L.; Wang, J.H. A three-dimensional porous organic framework for highly selective capture of mercury and copper ions. *ACS Appl. Polym. Mater.* **2019**, *1*, 2797–2806.
23. Kumar, R.; Shunmugam, R. Unique design of porous organic framework Showing efficiency toward removal of toxicants. *ACS Omega* **2017**, *2*, 4100–4107. [[PubMed](#)]
24. Tan, Z.; Su, H.; Guo, Y.; Liu, H.; Liao, B.; Amin, A.M.; Liu, Q. Ferrocene-Based Conjugated Microporous Polymers Derived from Yamamoto Coupling for Gas Storage and Dye Removal. *Polymers* **2020**, *12*, 719.
25. Xie, Y.; Wang, T.T.; Liu, X.H.; Zou, K.; Deng, W.Q. Capture and conversion of CO₂ at ambient conditions by a conjugated microporous polymer. *Nat. Commun.* **2013**, *4*, 1960. [[PubMed](#)]
26. Jiang, J.X.; Su, F.; Trewin, A.; Wood, C.D.; Campbell, N.L.; Niu, H.; Dickinson, C.; Ganin, A.Y.; Rosseinsky, M.J.; Khimyak, Y.Z.; et al. Conjugated Microporous Poly(aryleneethynylene) Networks. *Angew. Chem. Int. Ed.* **2007**, *46*, 8574–8578.
27. Côté, A.P.; Benin, A.I.; Ockwig, N.W.; O'Keeffe, M.; Matzger, A.J.; Yaghi, O.M. Porous, Crystalline, Covalent Organic Frameworks. *Science* **2005**, *310*, 1166–1170. [[PubMed](#)]
28. Mohamed, M.G.; Lee, C.C.; EL-Mahdy, A.F.M.; Luder, J.; Yu, M.H.; Li, Z.; Zhu, Z.; Chueh, C.C.; Kuo, S.W. Exploitation of Two-Dimensional Conjugated Covalent Organic Frameworks Based on Tetraphenylethylene with Bicarbazole and Pyrene Units and Applications in Perovskite Solar Cells. *J. Mater. Chem. A* **2020**, *8*, 11448–11459.
29. EL-Mahdy, A.F.M.; Mohamed, M.G.; Mansoure, T.H.; Yu, H.H.; Chen, T.; Kuo, S.W. Ultrastable tetraphenyl-p-phenylenediamine-based covalent organic frameworks as platforms for high-performance electrochemical supercapacitors. *Chem. Commun.* **2019**, *55*, 14890–14893.
30. EL-Mahdy, A.F.M.; Kuo, C.H.; Alshehri, A.; Young, C.; Yamauchi, Y.; Kim, J.; Kuo, S.W. Strategic design of triphenylamine- and triphenyltriazine-based two-dimensional covalent organic frameworks for CO₂ uptake and energy storage. *J. Mater. Chem. A* **2018**, *6*, 19532–19541.
31. Vopička, O.; Friess, K.; Hynek, V.; Sysel, P.; Zgazar, M.; Sipek, M.; Pilnacek, M.K.; Lanc, M.; Jansen, J.C.; Mason, C.R.; et al. Equilibrium and transient sorption of vapours and gases in the polymer of intrinsic microporosity PIM-1. *J. Membr. Sci.* **2013**, *434*, 148–160. [[CrossRef](#)]

32. Mohamed, M.G.; EL-Mahdy, A.F.M.; Ahmed, M.M.M.; Kuo, S.W. Direct Synthesis of Microporous Bicarbazole-Based Covalent Triazine Frameworks for High-Performance Energy Storage and Carbon Dioxide Uptake. *ChemPlusChem* **2019**, *84*, 1767–1774. [[CrossRef](#)] [[PubMed](#)]
33. Liu, M.; Guo, L.; Jin, S.; Tan, B. Covalent triazine frameworks: Synthesis and applications. *J. Mater. Chem. A* **2019**, *7*, 5153–5172. [[CrossRef](#)]
34. EL-Mahdy, A.F.M.; Hung, Y.H.; Mansoure, T.H.; Yu, H.H.; Hsu, Y.S.; Wu, K.C.W.; Kuo, S.W. Synthesis of [3+3] β -ketoenamine-tethered covalent organic frameworks (COFs) for high-performance supercapacitance and CO₂ storage. *J. Taiwan Inst. Chem. Eng.* **2019**, *103*, 199–208. [[CrossRef](#)]
35. Li, Z.; Li, H.; Wang, D.; Suwansoontorn, A.; Du, G.; Liu, Z.; Hasan, M.D.; Nagao, Y. A simple and cost-effective synthesis of ionic porous organic polymers with excellent porosity for high iodine capture. *Polymer* **2020**, *204*, 122796.
36. PENCHAH, H.R.; GHAEMI, A.; GILANI, H.G. Benzene-Based Hyper-Cross-Linked Polymer with Enhanced Adsorption Capacity for CO₂ Capture. *Energy Fuels* **2019**, *33*, 12578–12586.
37. Li, B.; Gong, R.; Wang, W.; Huang, X.; Zhang, W.; Li, H.; Hu, C.; Tan, B. A New Strategy to Microporous Polymers: Knitting Rigid Aromatic Building Blocks by External Cross-Linker. *Macromolecules* **2011**, *4*, 2410–2414. [[CrossRef](#)]
38. Pan, L.; Chen, Q.; Zhu, J.H.; Yu, J.G.; He, Y.J.; Han, B.H. Hypercrosslinked porous polycarbazoles via one-step oxidative coupling reaction and Friedel–Crafts alkylation. *Polym. Chem.* **2015**, *6*, 2478–2487. [[CrossRef](#)]
39. Mohamed, M.G.; Zhang, X.; Mansoure, T.H.; EL-Mahdy, A.F.M.; Huang, C.F.; Danko, M.; Xin, Z.; Kuo, S.W. Hypercrosslinked porous organic polymers based on tetraphenylanthraquinone for CO₂ uptake and high-performance supercapacitor. *Polymer* **2020**, *205*, 122857. [[CrossRef](#)]
40. Ma, T.; Zhao, X.; Matsuo, Y.; Song, J.; Zhao, R.; Faheem, M.; Chen, M.; Zhang, Y.; Tian, Y.; Zhu, G. Fluorescein-based fluorescent porous aromatic framework for Fe³⁺ detection with high sensitivity. *J. Mater. Chem. C* **2019**, *7*, 2327–2332. [[CrossRef](#)]
41. Wang, S.; Tu, M.; Peng, T.; Zhang, C.; Li, T.; Hussain, I.; Wang, J.; Tan, B. Porous hypercrosslinked polymer-TiO₂-graphene composite photocatalysts for visible-light-driven CO₂ conversion. *Nat. Commun.* **2019**, *10*, 676–686. [[CrossRef](#)] [[PubMed](#)]
42. Fontanals, N.; Marce, R.M.; Borrull, F.; Cormack, P.A.G. Hypercrosslinked materials: Preparation, characterisation, and applications. *Polym. Chem.* **2015**, *6*, 7231–7244. [[CrossRef](#)]
43. Tsyurupa, M.P.; Davankov, V.A. Porous structure of hypercrosslinked polystyrene: State-of-the-art mini review. *React. Funct. Polym.* **2006**, *66*, 768–779. [[CrossRef](#)]
44. Chen, D.; Fu, Y.; Yu, W.; Yu, G.; Pan, C. Versatile Adamantane-based porous polymers with enhanced microporosity for efficient CO₂ capture and iodine removal. *Chem. Eng. J.* **2018**, *334*, 900–906. [[CrossRef](#)]
45. Wang, X.; Mu, P.; Zhang, C.; Chen, Y.; Zeng, J.; Wang, F.; Jiang, J.X. Control Synthesis of Tubular Hyper-Cross-Linked Polymers for Highly Porous Carbon Nanotubes. *ACS Appl. Mater. Interfaces* **2017**, *9*, 20779–20786. [[CrossRef](#)]
46. Lu, Y.; Liang, J.; Deng, S.; He, Q.; Deng, S.; Hu, Y.; Wang, D. Hypercrosslinked polymers enabled micropore-dominant N, S Co-Doped porous carbon for ultrafast electron/ion transport supercapacitors. *Nano Energy* **2019**, *65*, 103993. [[CrossRef](#)]
47. Jiang, X.; Liu, Y.; Liu, J.; Fu, X.; Luo, Y.; Lyu, Y. Hypercrosslinked conjugated microporous polymers for carbon capture and energy storage. *New J. Chem.* **2017**, *41*, 3915–3919. [[CrossRef](#)]
48. Pola, S.; Kuo, C.H.; Peng, W.T.; Islam, M.M.; Chao, I.; Tao, Y.T. Contorted Tetrabenzocoronene Derivatives for Single Crystal Field Effect Transistors: Correlation between Packing and Mobility. *Chem. Mater.* **2012**, *24*, 2566–2571. [[CrossRef](#)]
49. Tian, M.; Wang, C.; Wang, L.; Luo, K.; Zhao, A.; Guo, C. Study on the synthesis and structure–effect relationship of multi-aryl imidazoles with their fluorescence properties. *Luminescence* **2014**, *29*, 540–548. [[CrossRef](#)]
50. Li, Y.H.; Chen, Y.C. Triphenylamine-hexaarylbiimidazole derivatives as hydrogen acceptor photoinitiators for free radical photopolymerization under UV and LED light. *Polym. Chem.* **2020**, *11*, 1504–1513. [[CrossRef](#)]
51. Lin, W.; Yuan, L.; Tan, W.; Feng, J.; Long, L. Construction of fluorescent probes via protection/deprotection of functional groups: A ratiometric fluorescent probe for Cu²⁺. *Chem. Eur. J.* **2008**, *15*, 1030–1035. [[CrossRef](#)]

52. Bu, L.; Sawada, T.; Kuwahara, Y.; Shosenji, H.; Yoshida, K. Crystallographic structure and solid-state fluorescence enhancement behavior of a 2-(9-anthryl)phenanthroimidazole-type clathrate host upon inclusion of amine molecules. *Dyes Pigm.* **2003**, *59*, 43–52. [[CrossRef](#)]
53. Boydston, A.J.; Vu, P.D.; Dykhno, O.L.; Chang, V.; Wyatt, A.R.; Stockett, A.S.; Ritschdorff, E.T.; Shear, J.B.; Bielawski, C.W. Modular fluorescent benzobis(imidazolium) salts: Syntheses, photophysical analyses, and applications. *J. Am. Chem. Soc.* **2008**, *130*, 3143–3156. [[CrossRef](#)] [[PubMed](#)]
54. Jayabharathi, J.; Thanikachalam, V.; Devi, K.B.; Srinivasan, N. Physicochemical studies of some novel y-shaped imidazole derivatives a sensitive chemisensor. *Spectrochim. Acta A* **2011**, *82*, 513–520. [[CrossRef](#)]
55. Xie, N.; Chen, Y. Synthesis and photophysical properties of 1,4-bis(4,5-diarylimidazol) benzene dyes. *J. Photochem. Photobiol. A* **2007**, *189*, 253–257. [[CrossRef](#)]
56. Zhou, J.X.; Luo, X.S.; Liu, X.; Qiao, Y.; Wang, P.; Mecerreyes, D.; Bogliotti, N.; Chen, S.L.; Huang, M.H. Azo-linked porous organic polymers: Robust and time-efficient synthesis via NaBH₄-mediated reductive homocoupling on polynitro monomers and adsorption capacity towards aniline in water. *J. Mater. Chem. A* **2018**, *6*, 5608–5612. [[CrossRef](#)]
57. Xie, S.; Wong, A.Y.H.; Kwok, R.T.K.; Li, Y.; Su, H.; Lam, J.W.Y.; Chen, S.; Tang, B.Z. Fluorogenic Ag⁺-Tetrazolate Aggregation Enables Efficient Fluorescent Biological Silver Staining. *Angew. Chem. Int. Ed.* **2018**, *57*, 5750–5753. [[CrossRef](#)]
58. Jing, X.; Zou, D.; Cui, P.; Ren, H.; Zhu, G. Facile synthesis of cost-effective porous aromatic materials with enhanced carbon dioxide uptake. *J. Mater. Chem. A* **2013**, *1*, 13926–13931. [[CrossRef](#)]
59. Zhang, X.; Lu, J.; Zhang, J. Porosity Enhancement of Carbazolic Porous Organic Frameworks Using Dendritic Building Blocks for Gas Storage and Separation. *Chem. Mater.* **2014**, *26*, 4023–4029. [[CrossRef](#)]
60. Yao, S.; Yang, X.; Yu, M.; Zhang, Y.; Jiang, J.X. High surface area hypercrosslinked microporous organic polymer networks based on tetraphenylethylene for CO₂ capture. *J. Mater. Chem. A* **2014**, *2*, 8054–8059. [[CrossRef](#)]
61. Liu, G.; Wang, Y.; Shen, C.; Ju, Z.; Yuan, D. A facile synthesis of microporous organic polymers for efficient gas storage and separation. *J. Mater. Chem. A* **2015**, *3*, 3051–3058. [[CrossRef](#)]
62. Li, T.; Yan, X.; Liu, Y.; Zhang, W.D.; Fu, Q.T.; Zhu, H.; Li, Z.; Gu, Z.G. A 2D covalent organic framework involving strong intramolecular hydrogen bonds for advanced supercapacitors. *Polym. Chem.* **2020**, *11*, 47–52. [[CrossRef](#)]
63. Shu, G.; Zhang, C.; Li, Y.; Jiang, J.X.; Wang, X.; Li, H.; Wang, F. Hypercrosslinked silole-containing microporous organic polymers with N-functionalized pore surfaces for gas storage and separation. *J. Appl. Polym. Sci.* **2018**, *135*, 45907–45916. [[CrossRef](#)]
64. Yuan, Y.; Huang, H.; Chen, L.; Chen, Y. N,N'-Bicarbazole: A Versatile Building Block toward the Construction of Conjugated Porous Polymers for CO₂ Capture and Dyes Adsorption. *Macromolecules* **2017**, *50*, 4993–5003. [[CrossRef](#)]
65. Ben, T.; Ren, H.; Ma, S.; Cao, D.; Lan, J.; Jing, X.; Wang, W.; Xu, J.; Deng, F.; Simmons, J.M.; et al. Targeted synthesis of a porous aromatic framework with high stability and exceptionally high surface area. *Angew. Chem. Int. Ed.* **2009**, *48*, 9457–9460. [[CrossRef](#)]
66. El-Mahdy, A.F.M.; Liu, T.E.; Kuo, S.W. Direct synthesis of nitrogen-doped mesoporous carbons from triazine-functionalized resol for CO₂ uptake and highly efficient removal of dyes. *J. Hazard. Mater.* **2020**, *391*, 122163. [[CrossRef](#)]
67. Wang, S.; Shao, L.; Sang, Y.; Huang, J. Hollow Hyper-Cross-Linked Polymer Microspheres for Efficient Rhodamine B Adsorption and CO₂ Capture. *J. Chem. Eng. Data* **2019**, *64*, 1662–1670. [[CrossRef](#)]
68. Huang, J.H.; Huang, K.L.; Liu, S.Q.; Wang, A.W.; Yan, C. Adsorption of Rhodamine B and methyl orange on a hypercrosslinked polymeric adsorbent in aqueous solution. *Colloids Surf. A Physicochem. Eng. Asp.* **2008**, *330*, 55–61. [[CrossRef](#)]
69. Langmuir, I. The adsorption of gases on plane surfaces of glass mica and platinum. *J. Am. Chem. Soc.* **1918**, *40*, 1361–1403. [[CrossRef](#)]
70. Schreivoge, A.; Maurer, J.; Winter, R.; Baro, A.; Laschat, S. Synthesis and Electrochemical Properties of Tetrasubstituted Tetraphenylethenes. *Eur. J. Org. Chem.* **2006**, *2006*, 3395–3404. [[CrossRef](#)]
71. Su, C.; Yang, F.; Ji, L.; Xu, L.; Zhang, C. Polytriphenylamine derivative with high free radical density as the novel organic cathode for lithium ion batteries. *J. Mater. Chem. A* **2014**, *2*, 20083–20088. [[CrossRef](#)]

72. Su, C.; He, H.; Xu, L.; Zhao, K.; Zheng, C.; Zhang, C. A mesoporous conjugated polymer based on a high free radical density polytriphenylamine derivative: Its preparation and electrochemical performance as a cathode material for Li-ion batteries. *J. Mater. Chem. A* **2017**, *5*, 2701–2709. [[CrossRef](#)]
73. EL-Mahdy, A.F.M.; Young, C.; Kim, J.; You, J.; Yamauchi, Y.; Kuo, S.W. Hollow Microspherical and Microtubular [3 + 3] Carbazole-Based Covalent Organic Frameworks and Their Gas and Energy Storage Applications. *ACS Appl. Mater. Interfaces* **2019**, *11*, 9343–9354. [[CrossRef](#)] [[PubMed](#)]
74. Khattak, A.M.; Ghazi, Z.A.; Liang, B.; Khan, N.A.; Iqbal, A.; Li, L.; Tang, Z. A redox-active 2D covalent organic framework with pyridine moieties capable of faradaic energy storage. *J. Mater. Chem. A* **2016**, *4*, 16312–16317. [[CrossRef](#)]
75. Liu, S.; Yao, L.; Lu, Y.; Hua, X.; Liu, J.; Yang, Z.; Wei, H.; Mai, Y. All-organic covalent organic framework/polyaniline composites as stable electrode for high-performance supercapacitors. *Mater. Lett.* **2019**, *236*, 354–357. [[CrossRef](#)]
76. DeBlase, C.R.; Silberstein, K.E.; Truong, T.T.; Abruña, H.D.; Dichtel, W.R. β -Ketoenamine-Linked Covalent Organic Frameworks Capable of Pseudocapacitive Energy Storage. *J. Am. Chem. Soc.* **2013**, *135*, 16821–16824. [[CrossRef](#)]

Publisher's Note: MDPI stays neutral with regard to jurisdictional claims in published maps and institutional affiliations.



© 2020 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (<http://creativecommons.org/licenses/by/4.0/>).