

Article

Effect of Surface-Modified TiO² Nanoparticles on the Anti-Ultraviolet Aging Performance of Foamed Wheat Straw Fiber/Polypropylene Composites

Lihui Xuan ¹ , Guangping Han ¹ , Dong Wang ¹ , Wanli Cheng 1,*, Xun Gao ¹ , Feng Chen 1,2 and Qingde Li 1,3

- ¹ Key Laboratory of Bio-based Material Science and Technology (Ministry of Education), Northeast Forestry University, Harbin 150040, China; leeh91@hotmail.com (L.X.); guangping.han@nefu.edu.cn (G.H.); zcwangd@hotmail.com (D.W.); gaoxunbeihuauniversity@hotmail.com (X.G.); c.f84@hotmail.com (F.C.); liqingde2017@hotmail.com (Q.L.)
- ² Wood Engineering Department, Liaoning Forestry Vocational-Technical College, Shenyang 110101, China
-
- ³ College of Arts and Design, Qiqihar University, Qiqihar 161000, China ***** Correspondence: nefucwl@nefu.edu.cn; Tel./Fax: +86-451-8219-1871

Academic Editor: Homayoun Hadavinia Received: 20 March 2017; Accepted: 19 April 2017; Published: 26 April 2017

Abstract: Surface modification and characterization of titanium dioxide (TiO₂) nanoparticles and their roles in thermal, mechanical, and accelerated aging behavior of foamed wheat straw fiber/polypropylene (PP) composites are investigated. To improve the dispersion of nanoparticles and increase the possible interactions between wheat straw fiber and the PP matrix, the surface of the $TiO₂$ nanoparticles was modified with ethenyltrimethoxy silane (A171), a silane coupling agent. The grafting of A171 on the TiO² nanoparticles' surface was characterized by Fourier transform infrared spectroscopy (FTIR). The wheat straw fibers treated with A171 and modified $TiO₂$ nanoparticles were characterized by FTIR and thermogravimetric analysis (TGA). FTIR spectra confirmed that the organic functional groups of A171 were successfully grafted onto the $TiO₂$ nanoparticles and wheat straw fibers, and the modified $TiO₂$ nanoparticles were adsorbed onto the wheat straw fibers. Thermogravimetric analysis showed that a higher thermal stability of the wheat straw fiber was obtained with the modified TiO² nanoparticles. The flexural, tensile, and impact properties were improved. A higher ultraviolet (UV) stability of the samples treated with modified $TiO₂$ nanoparticles was exhibited by the study of the color change and loss in mechanical properties.

Keywords: surface-modified TiO₂ nanoparticles; foamed wheat straw fiber/polypropylene composites; mechanical properties; thermostability; UV stability

1. Introduction

Most agricultural residues are currently burned off in the field. The burning off of wheat straw and other agricultural residues is responsible not only for a series of environmental problems, but also for the sustainable development of precious resources. Therefore, using a waste biomass resource has been the main objective, scientifically and sufficiently, of many studies. Recently, many studies have been carried out on composite panels produced using various agricultural residues and plastic [\[1–](#page-11-0)[3\]](#page-11-1). The most commonly investigated mechanical properties of agricultural fiber plastic composites (AFPC), such as tensile, flexural, and impact properties, are comparable to wood-plastic composites (WPC). Similar to WPC, AFPC can be extensively used for furniture, packaging, transportation, decorative materials, and so forth. However, several deficiencies, such as low strength-to-weight ratio, poor toughness, weak brittleness, and inconvenience of installation, have limited the application of AFPC [\[4\]](#page-11-2).

Incorporating a blowing or foaming agent in the production of the AFPC could effectively avoid some of these disadvantages [\[5](#page-11-3)[,6\]](#page-11-4). ZnO-modified azodicarbonamide (AC) has recently been widely used in research as a blowing agent owing to its compatible decomposition temperature range with the processing temperature of the plastic matrix in the AFPC [\[7\]](#page-11-5). The application of the AFPC as a building material necessitates that they are used outdoors and are often exposed to sunlight, which has resulted in concerns about their short service life. Of particular concern is the durability of AFPC after UV exposure. To overcome the inclination of AFPC to aging caused by UV, various approaches have been used to impart an anti-aging agent to the composite materials [\[8](#page-11-6)[,9\]](#page-11-7).

 $TiO₂$ is of interest for various applications owing to its numerous merits, such as low cost, nontoxicity, high photocatalytic activity, optical and electronic properties, antibacterial activity, UV protection, and environmental friendliness. When the $TiO₂$ particle size is reduced to the nanoscale, a significant effect on the anti-ultraviolet aging performance, thermal stability, and mechanical properties in the polymer materials has been demonstrated [\[10\]](#page-11-8). Novel properties of nanocomposites can be obtained by successfully incorporating TiO₂ nanoparticles into matrix materials. However, owing to their nanoscale size, extremely high surface area, and the ability to form highly-reactive hydroxyl and superoxide radicals, $TiO₂$ nanoparticles agglomerate very easily in various media and show a poor dispersion capacity in materials, which may reduce the resultant correlation properties of the nanocomposite materials. Therefore, it is necessary to control the nanoparticle morphology by dispersion modification to make them uniformly combine with other components. Many efforts have been made to overcome this problem and to enhance the filler–matrix interaction [\[11\]](#page-11-9). Surface modification of $TiO₂$ nanoparticles with silane coupling is an effective way to reduce its surface energy and to improve its dispersion properties. Moreover, grafting active functional groups onto the nanoparticle surface can increase the possibility of forming chemical bonds between modified nanoparticles and matrix materials. Surface-modified $TiO₂$ nanoparticles have been widely applied to polymer nanocomposites. However, research to study the effects of $TiO₂$ nanoparticles on foamed fiber/plastic composites is relatively rare, even though $TiO₂$ nanoparticles exhibit obvious UV absorption that may cause an increase in the UV resistance of composites.

Thus, the aim of this study was to investigate the practical feasibility of using $TiO₂$ nanoparticles as an additive in the manufacturing of foamed wheat straw fiber/PP composites. To achieve an appropriate dispersion of TiO₂ nanoparticles and to yield a better compatibility between the wheat straw fiber and PP, a silane coupling agent, A171, was used for surface modification of the TiO₂ nanoparticles. The effects of the modification with $TiO₂$ nanoparticles on the mechanical properties and resistance to UV aging of the composites were evaluated.

2. Materials and Methods

2.1. Raw Materials and Preparation

Wheat stalks were collected from a farmland in Weifang, China. The wheat stalks were granulated to the length of 40–80 um (Figure [1\)](#page-2-0), and the materials were dried to a moisture content of approximately 3% prior to use. PP (density = 920 kg/m³, melt flow index (MFI) = 2.5 g/10 min) was used as a matrix. The coupling agent, ethenyltrimethoxy silane (A171, $CH₂=CHSi(OCH₃)₃$, boiling point: 285 ◦C) was received from Quanxi Chemical Co., Ltd. (Nanjing, China). Azodicarbonamide (AC, decomposition temperature: 180–210 °C, gas evolution: 215–235 mL/g), as a blowing agent, was purchased from Hengrui Chemical Co., Ltd. (Jinan, China). TiO₂ nanoparticles (purity: 99.8%, 25 nm, anatase titania), zinc oxide (ZnO), calcium carbonate (CaCO₃), industrial paraffin, distilled water, ethanol, and acetic acid were purchased from Aladdin Reagent (Shanghai, China).

2.2. Modification of TiO² Nanoparticles with the A171 Coupling Agent

Although the commercial and functionalized $TiO₂$ nanoparticles can be obtained from many suppliers nowadays, there are some problems affecting the control quality and the stability of the

coupling agent. Thus, the TiO₂ nanoparticles used in this study were functionalized in the laboratory. The silane coupling agent, A171, was used for the modification of nano-sized TiO₂ particles. The TiO₂ nanoparticles were dispersed in a certain amount of ethanol, and acetic acid was added into the nanoparticles were dispersed in a certain amount of ethanol, and acetic acid was added into the mixture to achieve a pH of 4. The mixture was mechanically stirred at 500 rpm at room temperature for 30 min and was then ultrasonicated for 20 min. Then, the dispersion was transferred into a 250 mL three-necked round-bottom flask fitted with a condenser and a thermometer and A171 (15% of the three-necked round-bottom flask fitted with a condenser and a thermometer and A171 (15% of the mass of the TiO₂ nanoparticles) was added. The reaction for the functional groups of A171 to the TiO₂ nanoparticle surfaces was performed at 60 °C and allowed to proceed for 1 h. The dispersion was purified of the free A171 by five cycles of centrifugation for 20 min (10,000 rpm), redispersion of the purified of the free A171 by five cycles of centrifugation for 20 min (10,000 rpm), redispersion of the sediment in absolute ethanol, and ultrasonication for 10 min. A part of the dispersion was oven-dried sediment in absolute ethanol, and ultrasonication for 10 min. A part of the dispersion was oven-dried at 100 °C for 24 h and was cooled in a vacuum for 1 h to obtain a modified TiO₂ nanoparticle powder for back-up detection. for back-up detection.

2.3. Modification of Wheat Straw Fiber with the A171 Coupling Agent and Modified TiO₂ Nanoparticle Powder

Ninety-five percent ethanol was prepared by mixing pure ethanol and distilled water. A171 was Ninety-five percent ethanol was prepared by mixing pure ethanol and distilled water. A171 was added into the prepared 95% ethanol with stirring to reach a 4% concentration. Acetic acid was added into the mixture to reach pH 4 and mixed with mechanical stirring at room temperature for some into the mixture to reach pH 4 and mixed with mechanical stirring at room temperature for some hours for hydrolysis. The wheat straw fiber was placed into a high-speed mixer. Under the condition hours for hydrolysis. The wheat straw fiber was placed into a high-speed mixer. Under the condition of 60 °C, 1% prepared silane solution and 0%–3% modified TiO₂ nanoparticles were sprayed on the surface of the fiber and then the fibers were stirred for 30 min. The fiber samples were taken out and surface of the fiber and then the fibers were stirred for 30 min. The fiber samples were taken out and then dried at room temperature to fully evaporate the ethanol (Figure 1). then dried at room temperature to fully evaporate the ethanol (Figur[e 1](#page-2-0)).

Figure 1. Optical microscope images of (a) untreated wheat straw fiber and (b) wheat straw fiber treated by A171 and modified $TiO₂$ nanoparticles.

2.4. Blend Design and Sample Fabrication 2.4. Blend Design and Sample Fabrication

Through a comprehensive analysis of the relevant formulations of foamed composites and Through a comprehensive analysis of the relevant formulations of foamed composites and nanocomposites, the design for various formulations is shown in Table 1. According to Table 1, nanocomposites, the design for various formulations is shown in Table [1.](#page-3-0) According to Table [1,](#page-3-0) modified wheat straw fiber, PP, AC, ZnO, CaCO₃, and industrial paraffin were uniformly mixed in a high-speed mixer. The extrudates were prepared in the Engineering Composite Laboratory of Northeast Forestry University using a SJSH30/SJ45 twin-screw extruder (Nanjing Rubber Machinery Northeast Forestry University using a SJSH30/SJ45 twin-screw extruder (Nanjing Rubber Machinery Plant, Nanjing, China) and then granulated using a granulator. The blending temperature profile was controlled at 100, 130, 160, 165, 165, 180, and 175 °C from the feeding zone to the die. After the granulation was masticated by a two-roll mill, the composites were fabricated by prepressing for min, hot-pressing under 5 MPa for 10 min, and cold-pressing for 5 min, in that order. 15 min, hot-pressing under 5 MPa for 10 min, and cold-pressing for 5 min, in that order.

Е	Modified TiO ₂ Nanoparticles (wt.%)	Silane Coupling Agent $(wt, %$	Wheat Straw Fiber $(wt, %$	PP (wt, %	AС (wt, %)	ZnO (wt, %	CaCO ₃ (wt, %	Paraffin (wt.%)
\overline{A}			29	62.5		0.5		
В			29	61.5		0.5		
	l.5		29	61		0.5		
			29	60.5		0.5		
	2.5		29	60		0.5		
			29	59.5		0.5		

Table 1. Formulations of various samples for the preparation of the foamed wheat straw fiber/PP composites.

2.5. Material Characterization $\frac{1}{2}$

2.5.1. FTIR Study $D_{\rm 2}$ 1 \sim 29 \pm \sum 2.5 1 20 \sum 2.5 \sum 2.5

The treated and untreated wheat straw fiber and $TiO₂$ nanoparticles were ground and FTIR absorption data were obtained using a Nicolet 6700 FTIR spectrometer (Thermo Fisher Science, *2.5. Material Characterization* Agawam, MA, USA) with an ATR unit (Universal ATR Diamond Zn/Se) at a resolution of $4\ \mathrm{cm^{-1}}$ for 32 scans in the spectral range of 600–4000 cm^{-1} .

2.5.2. Thermogravimetric Analysis and using a Nicolet 6700 FTIR spectrometer (Thermo Fisher Science, Thermo Fis

Thermal properties of the untreated and treated wheat straw fiber samples were measured by using a TGA 309F3 thermal analyzer (TA Instruments, New Castle, DE, USA) at a heating rate of 10 °C/min from room temperature up to 600 °C under a nitrogen atmosphere. ermal properties of the untreated and the

2.5.3. Accelerated Aging Test Thermal properties of the untreated and treated wheat straw fiber samples were measured by

Fifteen replicates of each of the six formulations were placed in a xenon arc-type light exposure apparatus operated according to the American Society for Testing Material (ASTM) G 154-04 standard practice, with treatment for up to 600 h (Figure [2\)](#page-3-1). To understand the effect of UV exposure, the samples Figures, were removed for colorimetric analysis at an interval of 100 h. A NF333 photometer (Nippon Denshoku Co., Tokyo, Japan) was used to measure the color according to the International Commission on Illumination (CIE). The standard colorimetric parameters (brightness L*, red-green chromaticity index a^* , and yellow-blue chromaticity index b^*) were measured at five locations on each sample, and the color change (ΔE) was determined with the procedure outlined in ASTM D 2244: approved to the Chording the analysis at all direction of Test in Λ Testiscopy proveducted μ chromaticity index *a**, and yellow-blue chromaticity index *b**) were measured at five locations on each

$$
\Delta E = \left(\Delta L^{*2} + \Delta a^{*2} + \Delta b^{*2}\right)^{1/2} \tag{1}
$$

where ΔL^* , Δa^* , and Δb^* represent the differences between the initial and final values of L^* , a^* , and *b**, respectively. *b**, respectively.

Figure 2. The samples of (a) before the UV aging test; (b) being UV aging tested; and (c) after the UV aging test. aging test.

2.5.4. Mechanical Properties 2.5.4. Mechanical Properties \mathbf{I}

Samples were oven-dried at 103 °C for 24 h before testing to ensure the same conditioning for samples before and after UV exposure. Tensile tests were conducted in accordance with ASTM D 638. The size of each sample was 165 mm length and 19 mm width. Flexural (three-point bending) strength The size of each sample was 165 mm length and 19 mm width. Flexural (three-point bending) strength was measured according to ASTM D 790 using a support span of 56 mm and a cross head speed of was measured according to ASTM D 790 using a support span of 56 mm and a cross head speed of 2 mm/min. Five replicates were run for each formulation. All tests were performed on a universal 2 mm/min. Five replicates were run for each formulation. All tests were performed on a universal testing machine (Shenzhen REGER Instrument Co., Ltd., Shenzhen, China). The impacting strengths testing machine (Shenzhen REGER Instrument Co., Ltd., Shenzhen, China). The impacting strengths were measured according to ASTM D 6110 on a XJ-T0G composited impacting tester (Hebei Chengde were measured according to ASTM D 6110 on a XJ-T0G composited impacting tester (Hebei Chengde Mechanical Instrument Co., Ltd., Chengde, China). Ten replicates for each composition were tested for Mechanical Instrument Co., Ltd., Chengde, China). Ten replicates for each composition were tested impact strength. The mean value of parallel specimens was used as the test result. Specimens of the mechanical properties test are shown in Figure 3. the mechanical properties test are shown in Fi[gu](#page-4-0)re 3.

Figure 3. The samples' (a) tensile property being tested; (b) the flexural property being tested; and (**c**) the impact property being tested. (**c**) the impact property being tested.

3. Results and Discussion 3. Results and Discussion

3.1. FTIR Analysis 3.1. FTIR Analysis

FTIR spectra of unmodified TiO₂ nanoparticles and A171-modified TiO₂ nanoparticles are presented in Figure [4.](#page-5-0) In the spectrum for the unmodified TiO₂ (curve-4a), the stretching vibration of the absorbed water, as well as surface -OH groups, which are present in the TiO₂ nanoparticles, were confirmed by the broad absorption band between 3400 cm^{−1} and 3200 cm^{−1} and the peak at around 1630 cm^{−1} [12]. This [pea](#page-11-10)ks were mainly observed because the TiO₂ nanoparticles were easily polarized in the presence of water and produced a large quantity of hydroxyl groups on the surface. Furthermore, the electrons in the valence band of nano TiO₂ were excited to generate electron-hole pairs, and part of the electron-hole pairs migrated to the surface, forming the hydroxyl groups. For pairs, and part of the electron-hole pairs migrated to the surface, forming the hydroxyl groups. For the A171-treated TiO₂ (curve-4b), the broad absorption peak between 3400 cm^{−1} and 3200 cm^{−1} was weakened. This indicated that a condensation reaction between the silanol groups of hydrolyzed weakened. This indicated that a condensation reaction between the silanol groups of hydrolyzed A171 and the surface hydroxyl group of TiO₂ nanoparticles occurred [\[13\]](#page-11-11). In addition, a broad band at around 1276.16 cm $^{-1}$ corresponding to a Si–O–Si bond was observed, which confirmed a condensation reaction between silanol groups hydrol[yze](#page-11-9)d by A171 [11]. Compared with pure TiO₂ nanoparticles, the characteristic bands of 2958.43 cm^{−1} and 1620 cm^{−1} were attributed to the –CH₃ stretching and C=C stretching, respectively, demonstrating that the functional groups in A171 were grafted onto the TiO₂ nanoparticle surface. [Zha](#page-11-9)o et al. [11] showed that according to the IR spectra of modified nano TiO₂, the peak at 940 cm^{−1}–910 cm^{−1} was assigned to the stretching vibration band of Ti–O–Si. Chen and [Yak](#page-11-12)ovlev [14] also reported that the Si-O-Ti bonds confirmed the bonding of organosilane functional groups on the particle surface. Since the residual (non-reacted) and physically-adsorbed A171 molecules were removed by extraction in ethanol and centrifugal separation, the peaks mentioned showed that the chemical bonding of A171 onto the nanoparticle surfaces was realized through Ti–O–Si bonds, which could improve the dispersion of TiO₂ nanoparticles, mechanical properties, and UV aging resistance of composites containing $TiO₂$ nanoparticles [\[15\]](#page-11-13).

Figure 4. FTIR spectra of (a) untreated TiO₂ and (b) A171-grafted TiO₂ nanoparticles.

Figure [5](#page-6-0) shows the FTIR spectra of pure wheat straw fiber and modified wheat straw fiber. pure wheat straw fiber was characterized by the absorption bands (shown in curve-5a) appearing at The pure wheat straw fiber was characterized by the absorption bands (shown in curve-5a) appearing at between 3598.7 cm^{-1} and 3257.18 cm^{-1}, which were associated with the stretching vibration band intramolecular –OH of cellulose, hemicellulose, polysaccharide, and monosaccharide in the wheat of intramolecular –OH of cellulose, hemicellulose, polysaccharide, and monosaccharide in the wheat straw fiber. The C–H stretching vibrations of –CH3, as well as –CH2, which were present in the straw fiber. The C–H stretching vibrations of –CH3, as well as –CH2, which were present in the cellulose, were confirmed by the peaks at 2853.17 cm−1 and 2918.25 cm−1, respectively. The cellulose cellulose, were confirmed by the peaks at 2853.17 cm−¹ and 2918.25 cm−¹ , respectively. The cellulose was characterized by a high-intensity peak at 1034.14 cm−1 for –OH bending and shoulder peaks was characterized by a high-intensity peak at 1034.14 cm−¹ for –OH bending and shoulder peaks appeared at around 1158.53 cm^{-1} and 897.70 cm^{-1} for C–O–C stretching. The absorption peak at 896.75 cm^{-1} and 897.70 cm^{-1} for C–O–C stretching. $\mu_{\rm F}$ peared at abound 1100,00 cm and 697 as can be the interest of concerning. The about pear at 896.25 cm⁻¹ was ascribed to the intramolecular hydrogen bonding [\[16\]](#page-11-14). After surface modification organosilane, as presented in spectra curve-5b, the characteristic peaks for wheat straw fiber and by organosilane, as presented in spectra curve-5b, the characteristic peaks for wheat straw fiber and σ y expansionally as presented in specific call the set in distribution peaks at 2853.17 cm⁻¹ and 2918.25 cm⁻¹ were significantly weakened, indicating the quantity of C–H presented in cellulose was changed by the significantly weakened, indicating the quantity of C–H presented in cellulose was changed by the silane coupling agent. The peak that appeared at 1134.42 cm−1 for the Si–O–Si bond was due to the silane coupling agent. The peak that appeared at 1134.42 cm−¹ for the Si–O–Si bond was due to the condensation reaction of silanol groups hydrolyzed by A171. The absorption peak at 1200.47 cm−¹ condensation reaction of silanol groups hydrolyzed by A171. The absorption peak at 1200.47 cm−¹ was ascribed to the Si–O–cellulose bond, indicating that a strong polycondensation grafting reaction was ascribed to the Si–O–cellulose bond, indicating that a strong polycondensation grafting reaction was assisted to the strate or conducted some, martelling that a strong perfect that the mangements calculated between silanol groups and wheat straw fiber occurred [\[12\]](#page-11-10). No characteristic peaks for Ti–O–cellulose or hydrogen bonds formed between the TiO₂ surface –OH and the wheat straw fiber surface –OH or hydrogen bonds formed between the TiO₂ surface –OH and the wheat straw fiber surface –OH of hydrogen echas formed convent the rice conduct of that the middle straw fiber strate of the wheeler the wheat straw fiber and the modified TiO₂ nanoparticles did not occur. This proved that the main bonding mechanism of the modified TiO₂ nanoparticles and wheat straw fiber was physical sorption. However, Sun et al. showed that $TiO₂$ was chemically bonded to the wood surface through the hydrogen groups during the hydrothermal was chemically bonded to the wood surface through the hydrogen groups during the hydrothermal process [17]. This may arise from the different reaction conditions. process [\[17\]](#page-11-15). This may arise from the different reaction conditions.

Figure 5. FTIR spectra of (a) untreated wheat straw fiber and (b) wheat straw fiber treated by A171 and modified TiO₂ nanoparticles.

3.2. Morphology and Dispersibility of Surface-Modified Ti O_2 Nanoparticles

and modified TiO2 nanoparticles.

 S and S analysis was conducted to examine the morphology and dispersibility of surface-modified to surface-modified to m SEM analysis was conducted to examine the morphology and dispersibility of surface-modified TiO₂ nanoparticles. Typical surface SEM images of pure TiO₂ nanoparticles and surface-modified TiO₂ nanoparticles are illustrated in Figure [6.](#page-6-1) As can be seen from Figure [6a](#page-6-1), the spherical shape, as well as agglomerates, comprised the unmodified TiO₂ nanoparticles, indicating a poor dispersion in the as aggiomerates, comprised the unmodified $11O_2$ nanoparticles, mulcating a poor dispersion in the
control group. From F[igu](#page-6-1)re 6b, an improved dispersion was observed in the surface-modified TiO₂ nanoparticles, showing that the surface modification prevented the aggregation of TiO₂ nanoparticles to some extent. Certainly, there were still a small amount of agglomerates in the surface-modified TiO₂ nanoparticles, which might have an impact on some properties.

3.3. Thermogravimetric Analysis **Figure 6.** SEM images of (a) untreated TiO₂ and (b) A171-grafted TiO₂ nanoparticles.

To estimate the optimum amount of modified TiO2 nanoparticles required to enhance the *3.3. Thermogravimetric Analysis 3.3. Thermogravimetric Analysis*

properties of the wheat straw fiber, the wheat straw fiber treated by various percentages of modified To estimate the optimum amount of modified $TiO₂$ nanoparticles required to enhance the To estimate the optimum amount of modified $TiO₂$ nanoparticles required to enhance the properties of the wheat straw fiber, the wheat straw fiber treated by various percentages of modified TiO₂ nanoparticles were analyzed by the TGA technique. The TGA curves of the wheat straw fiber with different percentages of modified TiO₂ nanoparticles is shown in Figure [7.](#page-7-0) Three main mass loss regions were observed. In the thermograms obtained from the $TiO₂$ nanoparticle-treated wheat *Straw fibers, all of the samples showed an initial mass loss in the region of 50–100 °C, which was* attributed to the evaporation of water and residual solvents. It was evident that the mass loss was greater in the untreated wheat straw fibers compared with the treated fibers in the first stage, which arose from the untreated wheat straw fibers not being completely dry. The weight loss at this stage depends mainly on the initial moisture content of the fiber, therefore, the thermal properties should be analyzed from the second stage of thermal degradation. The second main degradation region was located between 200 °C and 350 °C, and the third stage mass loss occurred above 350 °C. The second degradation region was attributed to the cellulose and also to the degradation of lignin and other low molecular weight substances, which agreed with the results from the study of Huang et al. [\[18\]](#page-12-0). As can be seen, the treated wheat straw fiber had a higher decomposition temperature than the untreated wheat straw fiber, indicating the thermal stability of the fiber was enhanced by adding the modified TiO₂ nanoparticles. The degradation temperature of the treated wheat straw fiber was slightly increased with the TiO₂ nanoparticle content. Compared with the untreated wheat straw fiber, a reduced rate of the degradation of the wheat straw fiber became evident with the gradual increase in the additional amount of modified TiO₂ nanoparticles. This further demonstrated that the modified $TiO₂$ nanoparticles played a role on the thermal stability of the wheat straw fiber. Another slight no grandparticles played a role on the thermal stability of the wheat straw fiber. Thermal straw fiber slight weight loss after 350 °C may be attributed to the water formed from the condensation of the hydroxyl group on the particle surface [\[15\]](#page-11-13). In the thermal degradation process, 61% and 77% weight loss were observed for the untreated and treated wheat straw fiber, respectively. The percentages of modified observed for the untreated and treated wheat straw fiber, respectively. The percentages of modified TiO₂ particles had no obvious effect on the thermal properties of wheat straw fiber within the range selected for this experiment. For higher 1 wt.% content, the effect of TiO₂ nanoparticles on the thermal properties of wheat straw fiber was small. Hence, it was concluded that the thermal stability of the properties of wheat straw fiber was small. Hence, it was concluded that the thermal stability of the obtained wheat straw fiber was slightly improved with the dispersion of modified TiO₂ nanoparticles. loss after 350 °C may be attributed to the water formed from the condensation of the hydroxyl group

Figure 7. TGA curves of wheat straw fiber with different surface-modified TiO₂ nanoparticle contents.

3.4. Color Change during UV Aging 3.4. Color Change during UV Aging

The values of ΔL^* , Δa^* , Δb^* , and ΔE at different exposure times for UV-aged composites made from six different formulations are shown in Figure 8. From the figure from six different formulations are shown in Figure [8.](#page-8-0) From the figure, it was obvious that the brightness factor ∆*L** and the calculated total ∆*E* increased for all the samples with an increasing exposure, which was indicative of color fading of the UV-aged composite samples. Although all of the samples experienced an increase in the value of ΔL^* and ΔE , the increment of the value was

not proportional to the increase of exposure. Within the 300–400 h block, there was an apparent growth rate, and then the growth rate gradually decreased. These findings were consistent with the study of Fabiyi [\[19\]](#page-12-1). The samples without TiO² nanoparticles exhibited larger ∆*L** and ∆*E* values than samples with TiO₂ nanoparticles, indicating the enhanced UV aging-resistant performances for TiO₂ nanoparticle-added materials, which arose from the UV shielding ability of TiO₂ nanoparticles. Du et al. [\[9\]](#page-11-7) and Hou et al. [\[20\]](#page-12-2) reported that the color fading of wood-plastic composites could be attributed to the photodegradation nature of the fiber. Therefore, nano TiO₂ could inhibit color fading by absorbing some of the UV radiation, which resulted in less UV radiation being available to degrade the color of the samples. Furthermore, surface modification of TiO₂ nanoparticles by silane has been found to be effective in improving its dispersion. The TiO₂ particles with better dispersion enhanced the UV protection and improved the weathering performance [\[21\]](#page-12-3). As for the effect of the TiO₂ nanoparticle content on the color fading of composites, no significant differences were recorded among the TiO₂ nanoparticle-incorporated formulations, showing that the TiO₂ nanoparticle content level had little effect on the anti-ultraviolet aging performance within the selected range of this experiment. This may be related to the presence of a small amount of agglomerates after the surface modification of the TiO₂ nanoparticles. When comparing the effect of TiO₂ nanoparticles on the values of ∆*a*^{*} and ∆*b*^{*}, it can be seen that these values decreased for all samples. Compared with ΔL^* , the absolute values of Δa^* and Δb^* were relatively small, showing the Δa^* and Δb^* were not the main influencing factors of ΔE . Since the trend of ΔE followed that of ΔL^* , ΔL^* was the most important parameter influencing the total *ΔE*. A similar observation was reported by Hou [\[20\]](#page-12-2). This color fading pattern, therefore, revealed that composite formulations with moderated weathering performance could be achieved by incorporating TiO₂ nanoparticles. Du et al. [\[9\]](#page-11-7) also reported that the UV stability of wood–polymer composites was increased by incorporating ultraviolet absorbers. m_2 nanoparticles. When comparing the effect of m_2 nanoparticles on the values of m_2 and m_3

Figure 8. The change in color of foamed wheat straw fiber/PP composites with different modified **Figure 8.** The change in color of foamed wheat straw fiber/PP composites with different modified TiO2 nanoparticle content after UV weathering. TiO² nanoparticle content after UV weathering.

3.5. Mechanical Properties Study

The results of flexural, tensile, and impact properties of foamed wheat straw fiber/PP composites *3.5. Mechanical Properties Study* with different formulations before and after UV weathering are summarized in Figure [9.](#page-9-0) It was observed that the composites containing modified TiO₂ nanoparticles exhibited higher mechanical properties than those of composites without fillers. This might be a result of the nanoscale particles having a high aspect ratio, which offers an active surface area, and could play a role on the adsorption of hydroxyl groups on the surface of the fiber and chemical bonds in PP; hence, the mechanical properties were reinforced. An improvement in the mechanical properties arising from the incorporation of nanoparticles into the composites ha[s a](#page-12-5)lready been reported [12,22]. Xiao [23] showed that the mechanical properties for wood fiber/PP composite improved owing to the incorporation of Al_2O_3 nanoparticles. The trends of the mechanical properties were similar to each other, which showed a rising tendency along with the increasing concentration of modified $TiO₂$ nanoparticles, except for a small increase at a low level of modified TiO₂ nanoparticle loading (1%–2.5%). As clearly seen from Figure [9,](#page-9-0) the control samples showed lower mean mechanical values of 11.02 MPa, 5.02 MPa, and 0.53 KJ/m² for flexural, tensile, and impact properties, respectively. When the modified TiO₂ manoparticle content was increased to 3% , the mechanical values correspondingly increased to anoparticle content was increased to 3% , the mechanical values correspondingly increased to 18.13 MPa, 10.29 MPa, and 3.28 KJ/m². In general, the addition of modified nano TiO₂ affected the mechanical properties. However, in the range of the modified $TiO₂$ nanoparticle additions selected the mechanical properties. However, in the range of the modified $TiO₂$ nanoparticle additions selected in this experiment, no optimal values were obtained. The effect of higher modified $TiO₂$ nanoparticle contents (above 3 wt.%) on the mechanical properties of epoxy nanocomposites were examined in contents (above 3 wt.%) on the mechanical properties of epoxy nanocomposites were examined in previous studies [\[24\]](#page-12-6). The influence of a high modified TiO₂ nanoparticle content on the mechanical properties of the foamed wheat straw fiber/PP composites remains to be studied. properties of the foamed wheat straw fiber/PP composites remains to be studied. mechanical properties. However, in the range of the modified TiO2 nanoparticle additions selected in

Figure 9. The change in mechanical properties of wheat straw fiber/PP composites with different **Figure 9.** The change in mechanical properties of wheat straw fiber/PP composites with different modified $TiO₂$ nanoparticle content after UV weathering.

From Figure [9](#page-9-0) and Table [2,](#page-10-0) the flexural, tensile, and impact properties were reduced after UV treatment. The reduction in the mechanical properties was greater in the control samples. The modified $TiO₂$ nanoparticle-added formulations exhibited less of a reduction in all flexural, tensile, and impact properties. This might be a result of the modified $TiO₂$ nanoparticles inhibiting the rate of photodegradation. Related research has shown that ultraviolet light absorbed by TiO₂ nanoparticles is converted into heat, and the remaining UV energy is not sufficient to destroy the composites. Therefore, the resistance of foamed wheat straw fiber/PP composites to UV instability could be significantly improved with the addition of modified $TiO₂$ nanoparticles. However, the loss of mechanical properties for the composites incorporating any level of modified $TiO₂$ nanoparticles was unremarkable for the studied period of UV irradiation. These findings were consistent with the above analysis, where the range of modified TiO₂ nanoparticle additions selected in this experiment was limited. Moreover, the effect of a small amount of agglomerates formed in the surface modified $TiO₂$ nanoparticles could not be ignored. A study for a lower reduction in the mechanical properties and better UV resistance of these composites by adding more $TiO₂$ nanoparticles is still needed. As with the composites without UV aging, the mechanical properties were consistently high for all composites containing modified TiO² nanoparticles compared with the control specimens. This may arise from the interfacial interaction between the PP matrix, modified $TiO₂$ nanoparticles, and fibers, which is powerful enough to delay aging resulting from exposure to UV irradiation. Furthermore, the UV aging resulted in a relatively higher reduction of flexural properties in all composites with different formulations when compared with the tensile and impact properties, indicating the ultraviolet radiation had a greater influence on the flexural properties. This was mainly because the PP was oxidized under ultraviolet light and the aerobic environment and, as a result, the PP chains were fractured. The chain scission of the PP increased the microcracks on the composite surfaces, which led to a decrease in the flexural properties and hindered the stress transfer from the filler to the matrix, causing a decrease in the composite strength. In addition, since flexural strength depends on the tensile strength of the composite, the decreased flexural strength can be attributed to the decreased tensile modulus.

Modified Nano TiO ₂ Level (wt.%)				1	$1.5\,$	2	2.5	3
		Flexural strength (MPa)	11.02	13.66	13.85	15.56	15.74	18.13
	Before UV exposure	Flexural modulus (MPa)	413.59	653.62	680.33	767.42	795.99	984.05
		Flexural strength (MPa)	6.89	11.2	11.63	13.38	13.85	16.5
Flexural properties	After UV exposure	Flexural strength loss (%)	37.48	18.01	16.03	14.01	12.01	8.99
		Flexural modulus (MPa)	256.06	535.97	578.28	659.98	708.43	895.49
		Flexural modulus loss (%)	38.09	18.00	15.00	14.00	11.00	9.00
	Before UV exposure	Tensile strength (MPa)	5.02	7.66	7.89	8.04	8.18	10.29
Tensile properties	After UV exposure	Tensile strength (MPa)	3.72	6.51	6.79	7.08	7.2	9.36
		Tensile strength loss (%)	25.90	15.01	13.94	11.94	11.98	9.04
	Before UV exposure	Impact $(KJ/m2)$	0.53	1.86	1.92	2.25	3.03	3.28
Impact properties		Impact $(KJ/m2)$	0.46	1.69	1.73	2	2.8	3.12
		Impact loss $(\%)$	13.21	9.14	9.90	11.11	7.59	4.88
	After UV exposure							

Table 2. The change and reduction ratio in mechanical properties of wheat straw fiber/PP composites with different modified $TiO₂$ nanoparticle content after UV weathering.

4. Conclusions

TiO² nanoparticles modified by A171 were used as the active fillers for foamed wheat straw fiber/PP composites imparting an anti-UV performance to the materials. Surface modification of the $TiO₂$ nanoparticles with a silane coupling agent and an interaction between the wheat straw fiber and the silane coupling agent were successfully achieved. Thermogravimetric analysis showed that a higher thermal stability was achieved for the wheat straw fiber treated with modified $TiO₂$ nanoparticles. Improvements in the flexural, tensile, and impact properties were obtained, which could be attributed to the active surface area of the modified $TiO₂$ nanoparticles. The samples treated with modified TiO₂ nanoparticles were shown to exhibit a higher UV stability than that of untreated nanoparticles by studying the color change and loss in mechanical properties. Basic research on the application of $TiO₂$ nanoparticles incorporated into foamed wheat straw fiber/PP composites was conducted and will aid in future research in this field.

Acknowledgments: This work was financially supported by the National Natural Science Foundation of China (Grant No. 31570550).

Author Contributions: Lihui Xuan and Wanli Cheng conceived of and designed the experiments. Lihui Xuan, Dong Wang, Xun Gao, Feng Chen, and Qingde Li performed the experiments. Lihui Xuan, Wanli Cheng and Guangping Han analyzed and discussed the data. Lihui Xuan wrote the manuscript, with revisions by Wanli Cheng and Guangping Han.

Conflicts of Interest: The authors declare no conflict of interest.

References

- 1. El-Saied, H.; Basta, A.H.; Hassanen, M.E.; Korte, H.; Helal, A. Behaviour of rice byproducts and optimizing the conditions for production of high performance natural fiber polymer composites. *J. Polym. Environ.* **2012**, *20*, 838–847. [\[CrossRef\]](http://dx.doi.org/10.1007/s10924-012-0439-0)
- 2. El-Shekeil, Y.; Sapuan, S.; Abdan, K.; Zainudin, E. Influence of fiber content on the mechanical and thermal properties of Kenaf fiber reinforced thermoplastic polyurethane composites. *Mater. Des.* **2012**, *40*, 299–303. [\[CrossRef\]](http://dx.doi.org/10.1016/j.matdes.2012.04.003)
- 3. Liu, D.; Song, J.; Anderson, D.P.; Chang, P.R.; Hua, Y. Bamboo fiber and its reinforced composites: Structure and properties. *Cellulose* **2012**, *19*, 1449–1480. [\[CrossRef\]](http://dx.doi.org/10.1007/s10570-012-9741-1)
- 4. Petchwattana, N.; Covavisaruch, S.; Chanakul, S. Mechanical properties, thermal degradation and natural weathering of high desity polyethylene/rice hull composites compatibilized with maleic anhydride grafted polyethylene. *J. Polym. Res.* **2012**, *19*, 9921. [\[CrossRef\]](http://dx.doi.org/10.1007/s10965-012-9921-6)
- 5. Kord, B. Effect of chemical blowing agent on the hygroscopic behavior of HDPE/rice husk flour composites. *J. Thermoplast. Compos. Mater.* **2013**, *26*, 1114–1126. [\[CrossRef\]](http://dx.doi.org/10.1177/0892705711434192)
- 6. Petchwattana, N.; Covavisaruch, S. Influeces of modified chemical blowing agents on foaming of wood plastic composites prepared from poly (viyl chloride) and rice hull. *Adv. Mater. Res.* **2011**, *306*, 869–873. [\[CrossRef\]](http://dx.doi.org/10.4028/www.scientific.net/AMR.306-307.869)
- 7. Zhang, J.; Shi, L.; Zhang, J.Y. The mechanism of thermal decomposition of azodicarbonamide and the influence of zinc oxide. *J. Beijing Univ. Chem. Technol.* **2011**, *3*, 39–43. (In Chinese)
- 8. Du, H.; Wang, W.; Wang, Q.; Zhang, Z.; Sui, S.; Zhang, Y. Effects of pigments on the UV degradation of wood-flour/HDPE composites. *J. Appl. Polym. Sci.* **2010**, *118*, 1068–1076. [\[CrossRef\]](http://dx.doi.org/10.1002/app.32430)
- 9. Du, H.; Wang, W.; Wang, Q.; Sui, S.; Song, Y. Effects of ultraviolet absorbers on the UV-degradation of rice-hull/HDPE composites. *J. Appl. Polym. Sci.* **2012**, *126*, 906–915. [\[CrossRef\]](http://dx.doi.org/10.1002/app.36558)
- 10. Li, X.W.; Song, R.G.; Jiang, Y.; Wang, C.; Jiang, D. Surface modification of TiO₂ nanoparticles and its effect on the properties of fluoropolymer/TiO² nanocomposite coatings. *Appl. Surf. Sci.* **2013**, *276*, 761–768. [\[CrossRef\]](http://dx.doi.org/10.1016/j.apsusc.2013.03.167)
- 11. Zhao, J.; Milanova, M.; Warmoeskerken, M.M.; Dutschk, V. Surface modification of TiO₂ nanoparticles with silane coupling agents. *Colloid Surf. A* **2012**, *413*, 273–279. [\[CrossRef\]](http://dx.doi.org/10.1016/j.colsurfa.2011.11.033)
- 12. Devi, R.R.; Maji, T.K. Interfacial effect of surface modified $TiO₂$ and $SiO₂$ nanoparticles reinforcement in the properties of wood polymer clay nanocomposites. *J. Taiwan Inst. Chem. Eng.* **2013**, *44*, 505–514. [\[CrossRef\]](http://dx.doi.org/10.1016/j.jtice.2012.11.018)
- 13. Li, X.G. Structure and properties of wet modified titanium dioxide with silane coupling agent KH-570. *Adhesion* **2012**, *9*, 39–41. (In Chinese)
- 14. Chen, Q.; Yakovlev, N.L. Assorption and interaction of organosilanes on TiO₂ nanoparticles. *Appl. Surf. Sci.* **2010**, *257*, 1395–1400. [\[CrossRef\]](http://dx.doi.org/10.1016/j.apsusc.2010.08.036)
- 15. Sabzi, M.; Mirabedini, S.M.; Zohuriaan-Mehr, J.; Atai, M. Surface modification of TiO₂ nano-particles with silane coupling agent and investigation of its effect on the properties of polyurethane composite coating. *Prog. Org. Coat.* **2009**, *65*, 222–228. [\[CrossRef\]](http://dx.doi.org/10.1016/j.porgcoat.2008.11.006)
- 16. He, C.X.; Fu, L.M.; Xiong, J. Study on the microstructure and properties of wheat straw with different treatments. *J. Nanjing Agric. Univ.* **2016**, *39*, 325–331. (In Chinese)
- 17. Sun, Q.F.; Lu, Y.; Liu, Y. Growth of hydrophobic TiO₂ on wood surface using a hydrothermal method. *J. Mater. Sci.* **2011**, *46*, 7706–7712. [\[CrossRef\]](http://dx.doi.org/10.1007/s10853-011-5750-y)
- 18. Huang, K.; Zhang, G.Z.; Hou, X.L. Preparation and Characterization of Acetylated Rice Straw. *J. Cellulose Sci. Technol.* **2011**, *19*, 28–34.
- 19. Fabiyi, J.S.; McDonald, A.G. Effect of wood species on property and weathering performance of wood plastic composites. *Compos. Part A Appl. Sci. Manuf.* **2010**, *41*, 1434–1440. [\[CrossRef\]](http://dx.doi.org/10.1016/j.compositesa.2010.06.004)
- 20. Hou, R.Y.; He, C.X.; Xue, J.; Min, Y.; Chuanchuan, D. UV accelerated ageing properties of wheat straw/PP wood plastic composite. *Acta Mater. Compos. Sin.* **2013**, *5*, 86–93. (In Chinese)
- 21. Rong, Y.; Chen, H.Z.; Wu, G.; Wang, M. Preparation and characterization of titanium dioxide nanoparticles/ polystyrene composites via radical polymerization. *Mater. Chem. Phys.* **2005**, *91*, 370–374. [\[CrossRef\]](http://dx.doi.org/10.1016/j.matchemphys.2004.11.042)
- 22. Jaya, V.S.; Mohanty, S.; Rahail Parvaiz, M.; Nayak, S.K. Influence of nanoclays and nano-tio₂ on the mechanical and thermal properties of polycarbonate nanocomposites. *Macromol. Res.* **2012**, *20*, 1006. [\[CrossRef\]](http://dx.doi.org/10.1007/s13233-012-0197-9)
- 23. Xiao, L.X.; Yuan, G.M.; He, X. Study on the mechanical properties of nano-Al₂O₃/PP/wood fiber composites. *J. Cent. South Univ. For. Technol.* **2016**, *2*, 130–134. (In Chinese)
- 24. Pinto, D.; Bernardo, L.; Amaro, A.; Lopes, S. Mechanical Properties of Epoxy Nanocomposites using Titanium Dioxide as Reinforcement—A Review. *Constr. Build. Mater.* **2015**, *95*, 506–524. [\[CrossRef\]](http://dx.doi.org/10.1016/j.conbuildmat.2015.07.124)

© 2017 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license [\(http://creativecommons.org/licenses/by/4.0/\)](http://creativecommons.org/licenses/by/4.0/.).