

OPEN

Inherent Surface Properties of Adsorbent-Free Ultrathin Bi₂Se₃ Topological Insulator Platelets

Blaž Belec^{1*}, Katja Ferfolja¹, Tanja Goršak^{2,3}, Nina Kostevšek⁴, Sandra Gardonio¹, Mattia Fanetti¹ & Matjaz Valant^{1,5}

We report on a hydrothermal synthesis of hexagonal ultra-thin Bi₂Se₃ platelets, which was performed without any organic reactants. The synthesis resulted in the particles with a surface, clean of any organic adsorbents, which was confirmed with a high-resolution transmission electron microscopy, zeta-potential measurements and thermogravimetric measurements coupled with a mass spectroscopy. Due to the absence of the adsorbed organic layer on the Bi₂Se₃ platelet surface, we were able to measure their inherent surface and optical properties. So far this has not been possible as it has been believed that such hexagonal Bi₂Se₃ platelets can only be prepared by a solvothermal synthesis, for which it was unable to avoid the organic surface layer. Here we explain the mechanism behind the successful hydrothermal synthesis and show a striking difference in zeta potential behaviour and UV-vis absorption characteristics caused by the adsorbed layer. The surface of the hydrothermally synthesized Bi₂Se₃ platelets was so clean to enable the occurrence of the localized surface plasmon resonance due to the bulk and topological surface electronic states.

Topological insulators (TI) are a novel class of materials that recently have been intensively studied due to their attractive electron properties that result from metallic, linearly dispersing, spin-polarized surface states, widely known as topological surface states (TSS). The TI have a bulk band gap but, opposite to ordinary insulators, possess TSS on their surfaces that are robust to non-magnetic impurities and disorder because of the combined effect of time-reversal symmetry and spin-momentum locking^{1–3}. These peculiar properties make the TI as very promising materials for novel applications like quantum computing^{4–6}, THz detectors^{7,8}, plasmonics^{9–11}, spintronics^{3,12,13}, and for medical diagnostics and treatment^{14,15}.

Bi₂Se₃ is one of the most promising TI materials for the above mentioned applications, due to a narrow bulk band gap of 0.3 eV and the single Dirac cone at the Γ point of the Brillouin zone. Apart from its special electronic properties, Bi₂Se₃ displays a distinctive crystal structure. It crystallizes in a rhombohedral crystal structure, in which five covalently bonded atomic sheets (Se-Bi-Se-Bi-Se) form a quintuple layer arranged along the z-axis. The quintuple layers are linked together by weak van der Waals forces^{16,17}. This distinctive feature makes the Bi₂Se₃ crystals to grow in a variety of different low-dimensional nanostructures^{18–30}. From the viewpoint of applications (e.g. spintronic, quantum computing etc.), the most interesting are plate-like Bi₂Se₃ nanoparticles. The most suitable form for these applications would be oriented Bi₂Se₃ thin films prepared, for example, with a directed assembly of the plate-like Bi₂Se₃ crystallites by an electrophoretic deposition^{31–34}. Due to its geometry, the Bi₂Se₃ platelets self-assemble with its large surfaces parallel to the substrate, forming thin film-like structures that can be used as building blocks in the heterostructures³⁵.

Up till now, the most exploited method for the synthesis of the Bi₂Se₃ platelets is the solvothermal method^{21,22,24–27,29,36}. However, a disadvantage of this method is in use of toxic, expensive organic solvents (e.g. ethylene-glycol, diethylene-glycol, dimethylfuran, ethyleamine, etc.) and capping agents (e.g. acetic acid, PVP, ascorbic acid, etc.), which raises environmental issues. But the most serious disadvantage of the solvothermal synthesis is in a fact that it yields particles coated with an adsorbed thin amorphous layer of the organic solvents or surfactants^{37,38}. The formed organic layer is detrimental in the application of TI in various devices. It impedes

¹Materials Research Laboratory, University of Nova Gorica, Vipavska 11c, 5270, Ajdovščina, Slovenia. ²Department for Materials Synthesis, Jožef Stefan Institute, Jamova 39, 1000, Ljubljana, Slovenia. ³Jožef Stefan International Postgraduate School, Jamova 39, 1000, Ljubljana, Slovenia. ⁴Department for Nanostructured Materials, Jožef Stefan Institute, Jamova 39, 1000, Ljubljana, Slovenia. ⁵Institute of Fundamental and Frontier Sciences, University of Electronic Science and Technology of China, 610054, Chengdu, China. *email: blaz.belec@ung.si

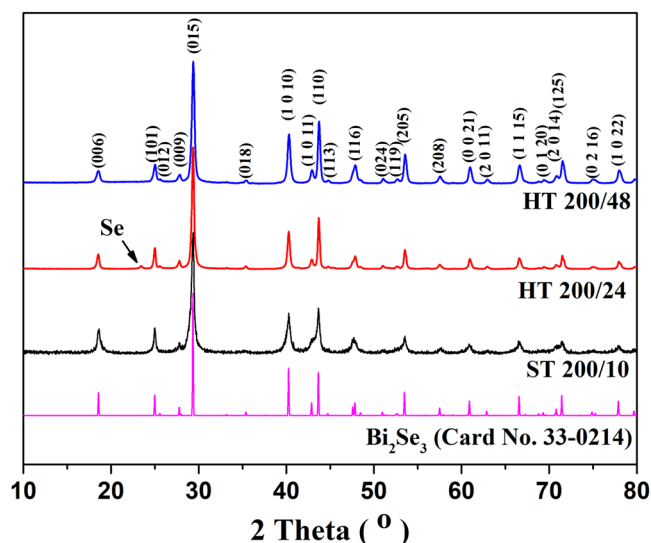


Figure 1. XRD patterns of Bi_2Se_3 platelets synthesized hydrothermally (HT) and solvothermally (ST) at 200 °C and different reaction times compared with the theoretical XRD pattern (Card No. 33-0214).

a direct contact, required between TI and other materials in heterostructures that exploit the inherent surface electronic properties of TI^{1–3,39–41}. Consequently, the solvothermally synthesized Bi_2Se_3 platelets cannot be used in such applications. It has already been pointed out that it would be essential to prepare the Bi_2Se_3 platelets without the presence of the organic solvents or surfactants²⁵. One of the possible approaches is the hydrothermal synthesis. Despite a successful preparation of a variety of Bi_2Se_3 nanostructures with well-defined morphologies, there are no reports on a successful synthesis of the shape-specific product with only the hexagonal Bi_2Se_3 platelets, using the hydrothermal method, that would allow avoiding the presence of organic adsorbents. The samples prepared hydrothermally usually contain, besides the hexagonally shaped platelets, also rod-like or sheet-like particles^{19,42}. Moreover, Ota *et al.*⁴² claimed that the monodispersed hexagonally shaped platelets could only be prepared by the solvothermal method.

As it is presented in this paper, we have successfully solved the problem that limits the application of the Bi_2Se_3 platelets in technology for years. Herein, we report a facile hydrothermal synthesis of the Bi_2Se_3 hexagonal platelets without using any organic reagents. In contrary to the previous report⁴², we demonstrated that the hexagonal platelets can be obtained hydrothermally by precise control of supersaturation. The obtained particles have a clean, non-modified surface, free of any adsorbed amorphous layer. Such particles expose the real inherent properties of the Bi_2Se_3 surface, which we have measured. As we show here, the difference in the surface and optical properties compared to the surfaces with the adsorbed amorphous layer is striking.

Results and Discussion

XRD patterns of the products, synthesized hydrothermally and solvothermally, are compared in Fig. 1. The diffraction patterns are very similar. The peaks can be indexed according to the Bi_2Se_3 rhombohedral structure (space group $R3m$, JCPDS 33-0214). The strong and sharp diffraction peaks indicate that Bi_2Se_3 is well crystallized. However, in the case of *HT 200/24*, an additional peak of Se is observed. The Se peak is not observed anymore for the *HT 200/48* sample. The selenium could appear as a result of an incomplete reaction between bismuth and selenium, due to the short reaction time.

Figure 2a shows an SEM image of the hydrothermally synthesized product *HT 200/48*. The product contains the hexagonally shaped platelets with a broad size distribution ranging from several tens of nanometres to a micrometre in diameter. The EDXS mapping (Fig. 2b) shows an increased signal of both elements, Bi and Se, over the region where the platelets are positioned, while the EDXS quantitative analysis on >100 platelets gave an atomic ratio Bi/Se = 0.66 ± 0.02 , which is in a good agreement with the Bi_2Se_3 stoichiometry.

The literature reports claim that the reaction product with only hexagonally shaped Bi_2Se_3 platelets can only be obtained by the solvothermal method at temperatures above 150 °C^{19,42}. It was explained that in the case of the hydrothermal synthesis, the reaction rate is much lower, which affects the product morphology. The slower reaction rate is a result of reduced ionic activity due to the more protic (polar) nature of water compared to the aprotic solvents (e.g. ethylene glycol). The reduced ionic activity promotes the growth of the particles in the preferential direction, forming particles with the rod-like morphology. According to the proposed growth mechanism⁴², the initial nanoparticles with a non-uniform sheet-like morphology transform into the rod-like particles at 150 °C⁴².

Here we show that contrary to the reported results, the hexagonally-shaped Bi_2Se_3 can be prepared with the hydrothermal synthesis at a somewhat higher temperature, 200 °C (Fig. 2a). Since the reaction rate is lower, due to the protic nature of water, the hydrothermal synthesis takes higher temperature and longer time compared to the solvothermal method. The SEM analysis revealed, that the Bi_2Se_3 platelets were formed already after 24 h, but some initial Se is still detected by XRD (Fig. 1); the reaction time is too short for the reaction between Bi and Se to be completed. Although the *HT 200/24* product is not single phase Bi_2Se_3 , a majority of it are hexagonally shaped platelets (Fig. 2c).

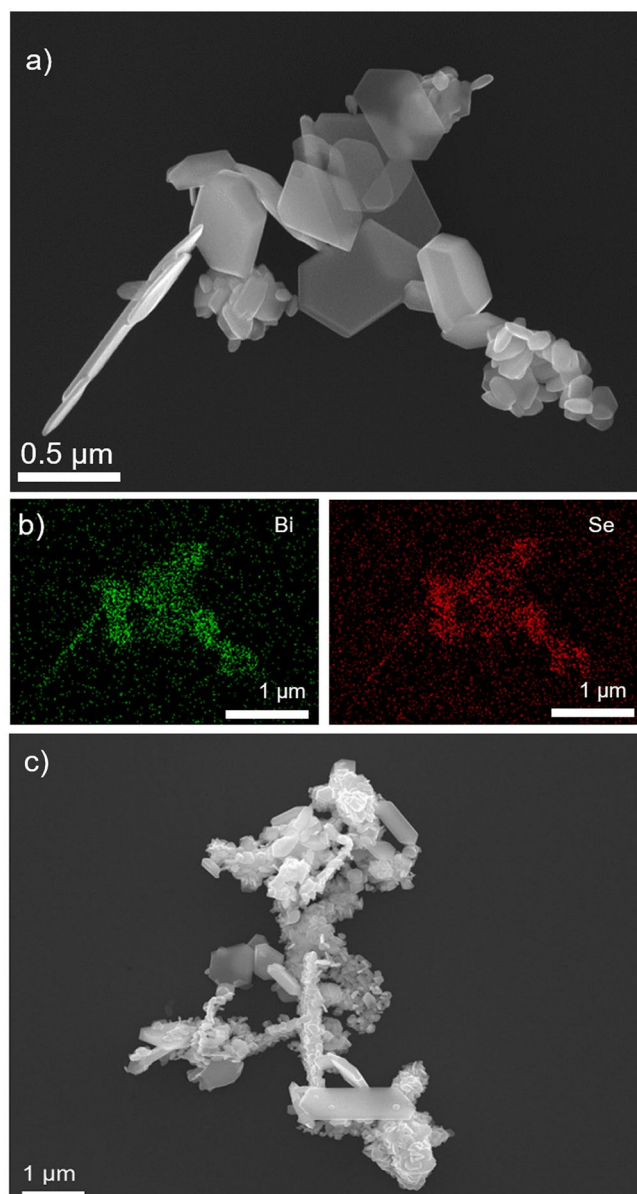


Figure 2. (a) SEM image of hydrothermally synthesized Bi_2Se_3 particles at 200 °C after 48 h with (b) corresponding EDXS elemental mapping and (c) SEM image of hydrothermally synthesized Bi_2Se_3 particles prepared at 200 °C after 24 h.

Beside the ionic activity, which influences the reaction rate, the important parameter affecting the growth and morphology of particles is supersaturation (σ) of the reaction system^{27,43}. According to the crystal growth theory, σ represents the driving force for crystal growth. Furthermore, the growth rate (R) of the crystal is linearly dependent on σ . Therefore, R linearly increases with σ ⁴³. In the case of the Bi_2Se_3 hydrothermal synthesis, σ can be controlled by the reduction rate of Se to Se^{2-} ^{19,27,42}. The reduction rate can be controlled by a concentration of a reductive agent (hydrazine) and/or by NaOH concentration. NaOH, present in the reaction, provides OH^- ions that convert HSe^- into Se^{2-} in the final step of Se reduction process⁴². In the reported literature^{19,42}, the authors had used a high concentration of reductive agent and/or NaOH, thus providing high σ . In our experiments, the concentration of the reductive agent was low, providing low σ . Moreover, to further decrease the hydrazine reducing capacity, we added 25 μL of 11 M HCl. According to Liu *et al.*²⁷, the reductive capacity of the reductive agents can be decreased with a strong acid. Comparing the reaction conditions used in our experiments and those reported in the literature, we assume that the reason for obtaining the hexagonally shaped Bi_2Se_3 platelets with the hydrothermal method is in much lower supersaturation level.

Figure 3a,b show representative TEM images of the Bi_2Se_3 platelets prepared solvothermally (*ST 200/10*) and hydrothermally (*HT 200/48*), respectively. A comparison of platelet width distributions, measured from the TEM images and expressed as an equivalent diameter (Fig. 4a), revealed that the solvothermally synthesized platelets display a broad, monomodal size distribution with a diameter ranging from 200–400 nm. The hydrothermally

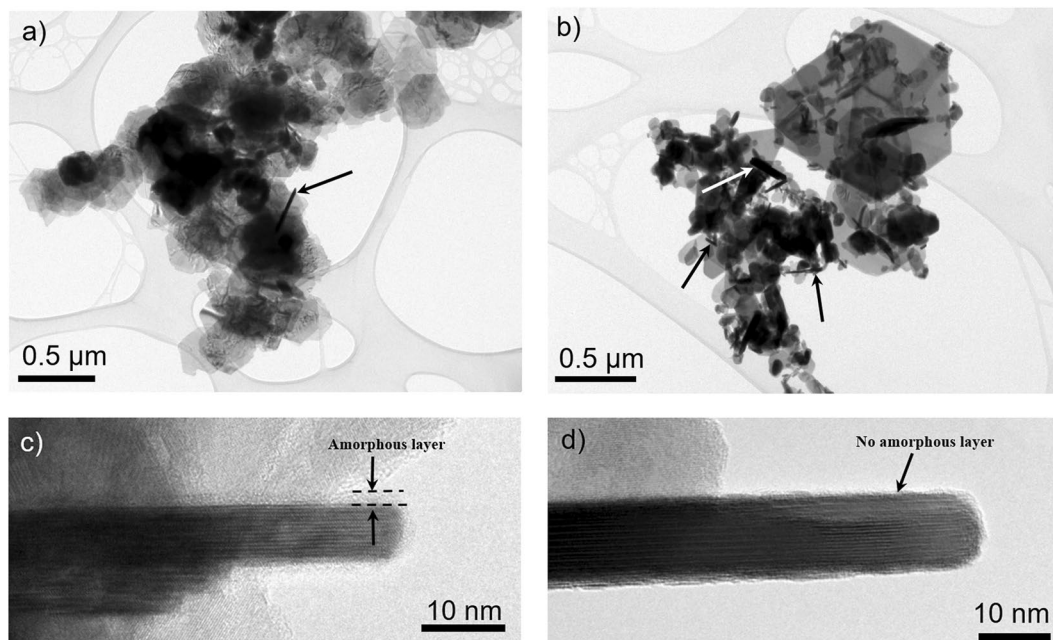


Figure 3. Representative TEM image of Bi_2Se_3 platelets synthesized (a) solvothermally at 200°C for 10 h and (b) hydrothermally at 200°C for 48 h. HR-TEM image of (c) solvothermally and (d) hydrothermally synthesized Bi_2Se_3 platelets orientated edge-on demonstrating the differences resulting from the different synthesis methods.

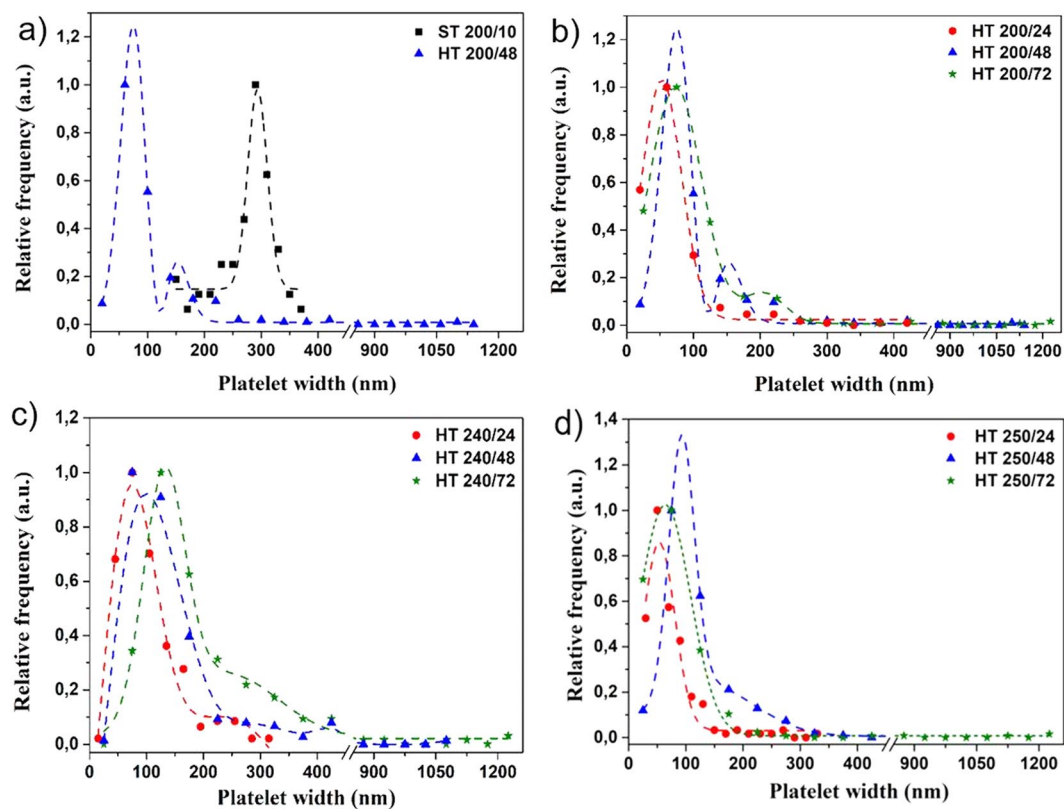


Figure 4. Bi_2Se_3 width measurements obtained from TEM images: a comparison between (a) the platelets synthesized by the solvothermal (ST) and hydrothermal (HT) method and a comparison between the hydrothermally synthesized platelets prepared at (b) 200°C , (c) 240°C and (d) 250°C with different reaction time.

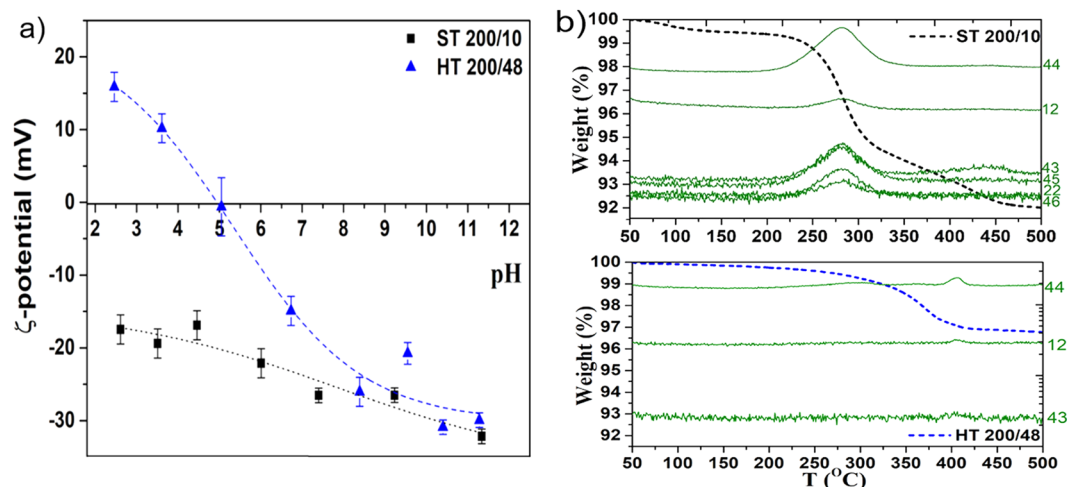


Figure 5. (a) Zeta-potential behavior of the hydrothermally (HT) and solvothermally (ST) synthesized Bi_2Se_3 platelets. (b) Mass loss during TG-MS experiment in nitrogen atmosphere for HT and ST samples heated from 50 to 500 °C. The green curves are intensities of chosen representative m/z fragments of evolved gases.

synthesized platelets display a bimodal size distribution in a nanometre size regime, with diameters ranging from ≈ 20 –100 nm and ≈ 150 –250 nm. Moreover, the hydrothermal product also contains some larger platelets with a diameter up to ≈ 1.2 μm . The platelets' thickness in both cases, the hydrothermal and solvothermal synthesis, was in a range from 10–20 nm, depending on the platelet diameter. However, due to the narrower size distribution in the case of the solvothermal synthesis, only the larger platelets, with a diameter close to 0.5 μm are 20 nm thick or more. A majority of the solvothermally prepared particles are 10 nm thick or even less. The difference in the size distribution can be ascribed to the use of different solvents. The hydrothermal performed synthesis results in the bimodal size distribution in the nanometre size regime and appearance of larger, micrometre sized platelets because the initially formed platelets grow with the secondary recrystallization (Ostwald ripening)^{44–46}. In the case of the solvothermal method, the surfactants and shaping agents suppress the secondary recrystallization, therefore, block the excessive platelet growth. This leads to the narrower and monomodal size distribution⁴⁶. We have also performed a synthesis with longer reaction time (up to 72 h) and/or at a higher temperature (240–250 °C). The size measurements of the hydrothermally synthesized Bi_2Se_3 platelets (>200 platelets per sample) indicate that there is not much difference between the sample prepared at prolonged time of 72 h at 200 °C compared to sample prepared at 48 h (Fig. 4b). The same is true also for the products obtained at 240 °C (Fig. 4c). All the products display a shoulder-like feature above 200 nm, indicating the bimodal size distribution of the platelets in the nanometre regime. For the samples prepared at 250 °C, only the sample treated for 48 h showed the broad shoulder-like feature, while samples treated for 24 or 72 h display the monomodal size distribution (Fig. 4d). However, with the increase in the reaction time and/or temperature, the average platelet size does not increase. This suggests that the system reached its thermodynamic equilibrium⁴⁴.

The most important difference between the hydrothermally and solvothermally synthesized Bi_2Se_3 particles is shown on the HR-TEM images in Fig. 3c,d. The images display the platelets' largest lattice periodicity of 0.95 nm, which corresponds to the (300) lattice planes of the rhombohedral Bi_2Se_3 structure^{42,47}. The observation of the platelets orientated edge-on revealed that the platelets synthesized by the solvothermal method, are covered with an approximately 2 nm thick amorphous layer (marked on Fig. 3c). Contrary, the surface of the hydrothermally synthesized Bi_2Se_3 platelets is clean (Fig. 3d), without the presence of an amorphous layer. It is well known, that in the case of the solvothermal synthesis, the molecules of organic solvent or surfactant adsorb on the particles' surface in the form of an amorphous layer^{31,48–51}. In our case, the amorphous layer could consist of the adsorbed molecules of ethylene glycol or PVP. To provide additional proof for the presence of the organic layer adsorbed on the surface of the solvothermal Bi_2Se_3 platelets and its absence on the hydrothermally synthesized surfaces, zeta potential (ζ) measurements and thermogravimetry coupled with a mass spectroscopy (TG-MS) were performed.

Figure 5a shows the zeta-potentials (ζ) of the platelets as a function of suspension pH. The Bi_2Se_3 layered structure terminates with Se^{2-} ions²⁵. The main difference between both surfaces is that in the case of the solvothermally synthesized platelets, the surface charge results from the adsorbed organic molecules and not from the platelet surface directly. Considering that for unmodified particles the atoms on the particle surface dictate the surface charge³⁷, we can explain the positive ζ of the hydrothermally synthesized platelets in acidic pH by adsorption of hydronium ion (H_3O^+). At the neutral and alkaline pH, the ζ values are negative due to the adsorption of hydroxyl ions (OH^-) from water. For the solvothermally synthesized platelets, ζ is negative along the whole pH region between pH 2.5 and 11. This is due to the hydroxyl ($-\text{OH}$) groups of the ethylene glycol molecules (used as a solvent for the solvothermal synthesis) adsorbed onto the surface of the platelets during the reaction.

The presence of ethylene glycol on the surface of the solvothermally synthesized platelets was also confirmed by TG-MS. The TG curve of the solvothermally prepared Bi_2Se_3 platelets, shown in Fig. 5b, is very similar to those reported in literature^{18,52}. During the sample heating to 500 °C in a N_2 atmosphere, all together around 8% of the mass was lost. A significant mass loss can be observed at ≈ 250 °C. The MS curves corresponding to this

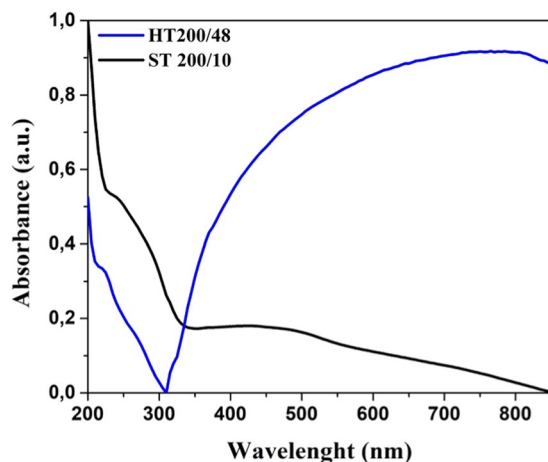


Figure 6. UV-vis absorption spectra of Bi_2Se_3 platelets synthesized hydrothermally (HT) and solvothermally (ST).

weight loss ($m/z = 42, 43, 45$ and 46) belong to ethylene glycol fragments⁵³. At the same temperature also the C ($m/z = 12$) and CO_2 ($m/z = 44$) fragments are observed, which were also ascribed to the ethylene glycol. The third decrease in the mass can be seen at $\approx 440^\circ\text{C}$ and is accompanied by an increase of the most intensive fragment 43. This indicates on two fractions of ethylene glycol molecules, one directly bonded on the platelet surface with strong van der Waals forces and one bonded on longer distance with weaker electrostatic forces. On the contrary, the mass loss of the hydrothermally synthesized sample is smaller, only 3%. (Fig. 5b) The significant decrease in mass was observed only above 300°C and was accompanied by an increase in the CO_2 ($m/z = 44$) and C ($m/z = 12$) fragments. Since the hydrothermal synthesis was performed in the absence of any organic molecules, the observed CO_2 and C fragments are due to CO_2 surface adsorption during post-processing (e.g. powder washing, drying in air etc). No intensity increase of the glycol-specific fragments (e.g. $m/z = 43$) was observed. The results obtained with the TG-MS analysis additionally support the ζ -potential and TEM results and confirm that in the case of the solvothermal synthesis, the Bi_2Se_3 platelets are coated with the thin amorphous layer, which is ethylene glycol, while hydrothermal synthesized surfaces are free of the organics.

Figure 6 shows the UV-vis spectra of the Bi_2Se_3 platelets, synthesized solvothermally and hydrothermally, dispersed in water at room temperature. The solvothermally prepared platelets display absorption peaks in the range of $\approx 210\text{--}300$ and $\approx 350\text{--}550$ nm, but towards the near infrared region the absorbance decreases without any absorption peaks. The behaviour is similar to what is reported in the literature^{14,15,36,52,54}. Contrary, the absorption spectrum of the hydrothermally synthesized Bi_2Se_3 platelets is strikingly different. In the spectral region of $\approx 210\text{--}300$ nm is very similar to what is observed for the solvothermally prepared platelets. However, at the higher wavelengths it displays an intense and broad adsorption peak that dominates the absorption over the entire measured range. A calculated plasmonic figure of merit predicts that the bulk intraband transition in Bi_2Se_3 lead to the plasmonic absorption in the range of $200\text{--}400$ nm, while interband and intraband transitions involving TSS yield the plasmonic response in the range of $400\text{--}700$ nm and above $1\ \mu\text{m}$, respectively⁵⁵. By comparing our measurements and the calculation from ref. ⁵⁵, the observed absorption features can be interpreted as due to a localized surface plasmon resonance (LSPR). Interestingly, only in the case of the platelets prepared with the hydrothermal method, the contribution due to TSS seems to be manifested.

Based on the microscopic characterization reported above, we propose that the quenching of the TSS plasmonic contribution in the case of the solvothermally prepared Bi_2Se_3 platelets, in the optical range, is due to the presence of the adsorbed organic layer. There are different reasons. One possibility is that the plasmon resonances due to TSS are shifted outside the visible range because of the organic layer like it is observed in metal nanoparticles⁵⁶. The other reason can be that the adsorbed organic layer changes the structure of the top atomic layers of Bi_2Se_3 platelets. Therefore, the thickness of the intact quintuple layers of Bi_2Se_3 reduces, reaching the threshold thickness where there is the opening of the gap with the consequent disappearance of the TSS contribution to LSPR⁵⁷.

Conclusion

Many methods have been proposed to synthesize Bi_2Se_3 platelets. Among them, the solvothermal synthesis is the most exploited one^{21,22,24–27,29}. Unfortunately, the disadvantage of this method is the use of organic reactants (solvents or capping agents), which tend to adsorb on the surface of the particles, as it was demonstrated in this work. The adsorbed amorphous layer limits the applicability of such Bi_2Se_3 platelets, since it appears to extinguish the plasmonic resonance involving TSS. The phenomena might be considered as an indirect signature of the TSS disruption of the TI, the property on which many applications rely^{1–3,39–41}. As we demonstrated in this article, the hydrothermal synthesis without using any organic reactants can be applied for the synthesis of Bi_2Se_3 platelets despite the opposite claims reported in the literature. This can be achieved by careful control of supersaturation of reactant species. The synthesized product contained only hexagonally-shaped particles with a clean surface, free of any organic adsorbents. Due to this, we were the first to measure the true inherent surface and optical properties of the Bi_2Se_3 platelets. We demonstrated, that the properties are significantly affected by the adsorbed layer.

We showed this on the case of the particle zeta potential and the UV-vis absorption characteristics. In contrary to the solvothermally synthesized particles, the TSS derived plasmonic features of the hydrothermally synthesized, are not disturbed due to the absence of the amorphous layer. Consequently, the inherent surface properties are manifested. Based on the comparison between experimental and calculated optical properties, we demonstrated that such surfaces display LSPR due to the electronic bulk states and TSS. The LSPR plays an important role in the photothermic effect of the Bi₂Se₃ platelets. Therefore, such Bi₂Se₃ platelets have great potential to be used as a photothermic conversion agent in cancer therapy^{7,10,11,14}. Since the surface of the hydrothermally synthesized platelets is clean, they have great potential to be used as building blocks for devices where the intimate contact between the materials in the heterostructure must be assured.

Methods

Chemicals. Bismuth (III) nitrate pentahydrate (Bi(NO₃)₃·5H₂O), bismuth (III) oxide (Bi₂O₃, 99.98%), selenium powder (Se, ≥99.5%), hydrochloric acid (HCl), hydrazine hydrate (N₂H₄·H₂O, 35%), sodium hydroxide (NaOH), ethylene glycol (C₂H₆O₂, 99%), polyvinylpyrrolidone (PVP, MW = 8000) were used. All the chemicals were purchased from Alfa Aesar and were used without further purification.

For the synthesis of Bi₂Se₃ platelets by the hydrothermal method, the stoichiometric amounts of bismuth nitrate (1 mmol) and selenium (1.5 mmol) were dissolved in 20 mL of deionized water under vigorous stirring, followed by an addition of 11 M HCl (25 μL) and hydrazine (1.6 mL). The obtained grey slurry was sealed in a Teflon-lined autoclave and heated at various temperatures ranging from 200–250 °C for 24–72 hours. Afterwards, the autoclave was allowed to cool naturally to room temperature. Then the product was washed several times with deionized water. The platelets, synthesized with the hydrothermal method, are denoted as “HT T/t”, where “T” denotes the synthesis temperature in °C and “t” time of the reaction in hours.

To compare and evaluate the hydrothermal synthesis, the Bi₂Se₃ platelets were also prepared by the solvothermal method, according to the procedure described in ref.⁵⁸. In short, PVP and the stoichiometric amounts of Bi and Se were dissolved in 20 mL of ethylene glycol. The grey slurry was sealed in a Teflon-lined autoclave and heated to 200 °C for 10 h. When the reaction was completed, the product was washed several times with deionized water. The sample obtained with the solvothermal method is denoted as “ST 200/10”.

The synthesized product was characterized by an X-ray powder diffractometer Rigaku MiniFlex with Cu Kα radiation (λ = 1541 Å, 30 kV, 10 mA). The morphology and chemical composition of the platelets were analyzed by a scanning (SEM) and a transmission (TEM) electron microscopy. For the SEM analysis, the suspension containing the platelets was deposited on a Si-wafer and dried. The SEM analysis was performed using a field-emission scanning electron microscope (FESEM, JEOL JSM 7100 TTLS) equipped with an energy dispersive X-ray spectrometer (EDXS, Oxford X-Max80). For the TEM analysis, the platelets were suspended in ethanol and deposited on a copper-grid-supported lacey carbon film. The TEM analysis was performed using a field-emission electron microscope (JEOL JEM 2100UHR) operating at 200 kV and equipped with an Oxford X-Max80T energy dispersive X-ray spectroscopy detector (EDXS). The width of the platelets expressed as an equivalent diameter was determined from the TEM images, on which 200–300 platelets per sample were accounted for the statistic using Gatan Digital Micrograph Software. The obtained data, representing the frequency count of the size distribution, were fitted using the single or multiple peak Gaussian fit mode. Electro-chemical properties (ζ-potential) of the platelets dispersed in water were measured as a function of the suspension pH using a ZetaPALS instrument (Brookhaven Instruments Corporation). The pH of the aqueous suspension was adjusted with diluted hydrochloric acid and sodium hydroxide. The thermogravimetric analysis was performed by a TGA/DSC 2 (Mettler Toledo) thermal analyzer coupled with a mass spectrometer Thermstar 300 (Vacuum Pfeiffer). The samples were heated from 50 °C to 500 °C at 10 °C/min under a nitrogen atmosphere with a gas flow of 50 mL/min. The light absorption properties were analyzed with a classical UV-vis spectroscopy. The spectroscopy was performed by a PerkinElmer Lambda 950 spectrometer using a quartz cuvette with a size of 1 × 1 × 3 cm. A measurement range, λ, from 300 to 900 nm was used with a scanning rate of 1 nm/s. Prior to the measurements, the water suspension of the platelets was stabilized according to the ζ-potential and sonicated to break any possible agglomerates.

Received: 15 July 2019; Accepted: 28 November 2019;

Published online: 13 December 2019

References

- Hasan, M. Z. & Kane, C. L. Colloquium: Topological insulators. *Rev. Mod. Phys.* **82**, 3045–3067 (2010).
- Qi, X.-L. & Zhang, S.-C. Topological insulators and superconductors. *Rev. Mod. Phys.* **83**, 1057–1110 (2011).
- Kong, D. & Cui, Y. Opportunities in chemistry and materials science for topological insulators and their nanostructures. *Nat. Chem.* **3**, 845–849 (2011).
- Collins, G. P. Computing with quantum knots. *Sci. Am.* **294**, 57–63 (2006).
- Kitaev, A. & Preskill, J. Topological entanglement entropy. *Phys. Rev. Lett.* **96**, 110404 (2006).
- Ornes, S. Core Concept: Topological insulators promise computing advances, insights into matter itself. *Proc. Natl. Acad. Sci. USA* **113**, 10223–10224 (2016).
- Autore, M. *et al.* Terahertz plasmonic excitations in Bi₂Se₃ topological insulator. *J. Phys. Condens. Matter* **29**, 183002 (2017).
- Zhang, X., Wang, J. & Zhang, S.-C. Topological insulators for high-performance terahertz to infrared applications. *Phys. Rev. B* **82**, (2010).
- Autore, M. *et al.* Plasmon-Phonon Interactions in Topological Insulator Microrings. *Adv. Optical Mater.* **3**, 1257–1263 (2015).
- Di Pietro, P. *et al.* Observation of Dirac plasmons in a topological insulator. *Nat. Nanotechnol.* **8**, 556–560 (2013).
- Autore, M. *et al.* Observation of Magnetoplasmons in Bi₂Se₃ Topological Insulator. *ACS Photonics* **2**, 1231–1235 (2015).
- Zhang, H. *et al.* Topological insulators in Bi₂Se₃, Bi₂Te₃ and Sb₂Te₃ with a single Dirac cone on the surface. *Nat. Phys.* **5**, 438–442 (2009).
- Chen, Y. L. *et al.* Experimental realization of three-dimensional topological insulator, Bi₂Se₃. *Science* **325**, 1–3 (2009).

14. Guozhi, J., Peng, W., Yanbang, Z. & Kai, C. Localized surface plasmon enhanced photothermal conversion in Bi₂Se₃ topological insulator nanoflowers. *Sci. Rep.* **6**, 25884 (2016).
15. Li, Z. *et al.* Multifunctional Bismuth Selenide Nanocomposites for Antitumor Thermo-Chemotherapy and Imaging. *ACS Nano* **10**, 984–997 (2016).
16. Sobolev, V. V., Shutov, S. D., Popov, Y. V. & Sestatskii, S. N. Reflectivity spectra of the rhombohedral crystals Bi₂Te₃, Bi₂Se₃, and Sb₂Te₃ over the range from 0.7 to 12.5 eV. *Phys. Stat. Sol.* **30**, 349–355 (1968).
17. Zhang, W., Yu, R., Zhang, H.-J., Dai, X. & Fang, Z. First-principles studies of the three-dimensional strong topological insulators Bi₂Te₃, Bi₂Se₃ and Sb₂Te₃. *New J. Phys.* **12**, 065013 (2010).
18. Salavati-Niasari, M., Bazarganipour, M. & Davar, F. Hydrothermal synthesis and characterization of bismuth selenide nanorods via a co-reduction route. *Inorganica Chim. Acta* **365**, 61–64 (2011).
19. Liu, K., Wang, J., Liu, H. & Xiang, D. Preparation and characterization of nanostructured Bi₂Se₃ and Sn_{0.5}-Bi₂Se₃. *Rare Metals* **28**, 112–116 (2009).
20. Liu, F. *et al.* The effect of temperature on Bi₂Se₃ nanostructures synthesized via chemical vapor deposition. *J. Mater. Sci.: Mater. Electron* **26**, 3881–3886 (2015).
21. Batabyal, S. K., Basu, C., Das, A. R. & Sanyal, G. S. Solvothermal synthesis of bismuth selenide nanotubes. *Mat. Lett.* **60**, 2582–2585 (2006).
22. Wang, D., Yu, D., Mo, M., Liu, X. & Qian, Y. Preparation and characterization of wire-like Sb₂Se₃ and flake-like Bi₂Se₃ nanocrystals. *J. Cryst. Growth* **253**, 445–451 (2003).
23. Cui, H. *et al.* Sonochemical synthesis of bismuth selenide nanobelts at room temperature. *J. Cryst. Growth* **271**, 456–461 (2004).
24. Wu, X. *et al.* Solution growth of two-dimensional Bi₂Se₃ nanosheets for two-color all-optical switching. *Materials* **10** (2017).
25. Min, Y. *et al.* Quick, controlled synthesis of ultrathin Bi₂Se₃ nanodiscs and nanosheets. *J. Am. Chem. Soc.* **134**, 2872–2875 (2012).
26. Liu, H. T., Dai, J., Zhang, J. J. & Xiang, W. D. Solvothermal Synthesis of Bi₂Se₃ Hexagonal Nanosheet Crystals. *Adv. Mater. Res.* **236–238**, 1712–1716 (2011).
27. Liu, X. *et al.* One-pot synthesis of Bi₂Se₃ nanostructures with rationally tunable morphologies. *Nano. Research* **8**, 3612–3620 (2015).
28. Gao, L. *et al.* The high-yield growth of Bi₂Se₃ nanostructures via facile physical vapor deposition. *Vacuum* **140**, 58–62 (2017).
29. Kadel, K., Kumari, L., Li, W., Huang, J. Y. & Provencio, P. P. Kadel. Synthesis and thermoelectric properties of Bi₂Se₃ nanostructures. *Nanoscale Res. Lett.* **6**, 1–5 (2011).
30. Jiang, Y., Hao, M., Jiang, L., Liu, F. & Liu, Y. Shape and stoichiometry control of bismuth selenide nanocrystals in colloidal synthesis. *RSC Adv.* **6**, 47840–47843 (2016).
31. Lisjak, D. *et al.* A surface-chemistry study of barium ferrite nanoplates with DBSA-modified surfaces. *Appl. Surf. Sci.* **305**, 366–374 (2014).
32. Lin, T.-H., Huang, W.-H., Jun, I.-K. & Jiang, P. Bioinspired assembly of colloidal nanoplatelets by electric field. *Chem. Mater.* **21**, 2039–2044 (2009).
33. Verde, M., Peiteado, M., Caballero, A. C., Villegas, M. & Ferrari, B. Electrophoretic deposition of transparent ZnO thin films from highly stabilized colloidal suspensions. *J. Colloid. Interface Sci.* **373**, 27–33 (2012).
34. Kort, K. R. & Banerjee, S. Oriented electrophoretic deposition of GdOCl nanoplatelets. *J. Phys. Chem. B* **117**, 1585–1591 (2013).
35. Kou, L., Ma, Y., Sun, Z., Heine, T. & Chen, C. Two-Dimensional Topological Insulators: Progress and Prospects. *J. Phys. Chem. Lett.* **8**, 1905–1919 (2017).
36. Xie, H. *et al.* Metabolizable ultrathin Bi₂Se₃ nanosheets in imaging-guided photothermal therapy. *Small* **12**, 4136–4145 (2016).
37. Berg, J. M., Romoser, A., Banerjee, N., Zebda, R. & Sayes, C. M. The relationship between pH and zeta potential of ~ 30 nm metal oxide nanoparticle suspensions relevant to in vitro toxicological evaluations. *Nanotoxicology* **3**, 276–283 (2009).
38. Shaw, D. J. *Introduction to colloid and surface chemistry*. 4th. Ed. edn. (Butterworth-Heinemann Publications, 2000).
39. Schubert, G., Fehske, H., Fritz, L. & Vojta, M. Fate of topological-insulator surface states under strong disorder. *Phys. Rev. B* **85**, <https://doi.org/10.1103/PhysRevB.85.201105> (2012).
40. Siroki, G., Haynes, P. D., Lee, D. K. K. & Giannini, V. Protection of surface states in topological nanoparticles. *Phys. Rev. Mater.* **1**, (2017).
41. Jakobs, S. *et al.* Controlling the spin texture of topological insulators by rational design of organic molecules. *Nano Lett.* **15**, 6022–6029 (2015).
42. Ota, J. R., Roy, P., Srivastava, S. K., Popovitz-Biro, R. & Tenne, R. A simple hydrothermal method for the growth of Bi₂Se₃ nanorods. *Nanotechnology* **17**, 6 (2006). 1700–1705.
43. Jolivet, J. P. *Metal oxide chemistry and synthesis: From solutions to solid state*. (John Wiley & Sons, 1994).
44. Kashchiev, D. *Nucleation: Basic theory with applications*. (Butterworth-Heinemann, Elsevier Science, 2000).
45. Drogenik, M., Kristl, M., Žnidaršič, A., Hanzel, D. & Lisjak, D. Hydrothermal synthesis of barium hexaferrite nanoparticles. *J. Am. Ceram. Soc.* **90**, 2057–2061 (2007).
46. Primc, D., Makovec, D., Lisjak, D. & Drogenik, M. Hydrothermal synthesis of ultrafine barium hexaferrite nanoparticles and the preparation of their stable suspensions. *Nanotechnology* **20**, 1–9 (2009).
47. Dang, W., Peng, H., Li, H., Wang, P. & Liu, Z. Epitaxial heterostructures of ultrathin topological insulator nanoplate and graphene. *Nano Lett.* **10**, 2870–2876 (2010).
48. Nagarajan, R. & Hatton, T. A. *Nanoparticles, synthesis, stabilization, passivation and functionalization*. (Oxford university Press, 2008).
49. Kralj, S., Makovec, D., Čampelj, S. & Drogenik, M. Producing ultra-thin silica coatings on iron-oxide nanoparticles to improve their surface reactivity. *J. Magn. Magn. Mater.* **322**, 1847–1853 (2010).
50. Lisjak, D., Ovtar, S. & Drogenik, M. The stability of BaFe₁₂O₁₉ nanoparticles in polar solvents. *J. Mater. Sci.* **46**, 2851–2859 (2011).
51. Han, K. N., Healy, T. W. & Fuerstenau, D. W. The mechanism of adsorption of fatty acids and other surfactants at the oxide-water interface. *J. Coll. and Interf. Sci.* **44**, 407–414 (1973).
52. Peng, J., Xiong, Y., Lin, Z., Sun, L. & Weng, J. Few-layer bismuth selenides exfoliated by hemin inhibit amyloid-beta1-42 fibril formation. *Sci. Rep.* **5**, 10171 (2015).
53. MS: <https://webbook.nist.gov/cgi/cbook.cgi?ID=C107211&Mask=200#Mass-Spec>. (accessed January 2019)
54. Sun, L. *et al.* Preparation of few-layer bismuth selenide by liquid-phase-exfoliation and its optical absorption properties. *Sci. Rep.* **4**, 4794 (2014).
55. Yin, J. *et al.* Plasmonics of topological insulators at optical frequencies. *NPG Asia Materials* **9**, e425 (2017).
56. Xu, Z., Hou, Y. & Sun, S. Magnetic core/shell Fe₃O₄/Au and Fe₃O₄/Au/Ag nanoparticles with tunable plasmonic properties. *J. Am. Ceram. Soc.* **129**, 8698–8699 (2007).
57. Eddrief, M., Vidal, F. & Gallas, B. Optical properties of Bi₂Se₃: from bulk to ultrathin films. *J. Phys. D: Appl. Phys.* **49**, 505304 (2016).
58. Ferfolja, K., Valant, M., Mikulska, I., Gardonio, S. & Fanetti, M. Chemical Instability of an Interface between Silver and Bi₂Se₃ Topological Insulator at Room Temperature. *J. Phys. Chem. C* **122**, 9980–9984 (2018).

Acknowledgements

The work was financially supported from the Slovenian Research Agency, research core funding No. P2-0412. All authors have given approval to the final version of the manuscript.

Author contributions

B.B. and M.V. conceived the study. B.B. performed the synthesis, XRD, microscopy and carried our particle size measurements. K.F. performed thermal analysis. T.G. performed ζ -potential measurements. N.K. performed UV-VIS measurements. B.B., M.V., T.G., S.G., M.F. wrote the paper. All the authors contributed in interpretation of data.

Competing interests

The authors declare no competing interests.

Additional information

Correspondence and requests for materials should be addressed to B.B.

Reprints and permissions information is available at www.nature.com/reprints.

Publisher's note Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.



Open Access This article is licensed under a Creative Commons Attribution 4.0 International License, which permits use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons license, and indicate if changes were made. The images or other third party material in this article are included in the article's Creative Commons license, unless indicated otherwise in a credit line to the material. If material is not included in the article's Creative Commons license and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder. To view a copy of this license, visit <http://creativecommons.org/licenses/by/4.0/>.

© The Author(s) 2019