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OPEN Tuning the ferromagnetic phase in the CDW compound SmNiC₂ via chemical alloying

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We report a study on tuning the charge density wave (CDW) ferromagnet SmNiC₂ to a weakly coupled superconductor by substituting La for Sm. X-ray diffraction measurements show that the doped compounds obey Vegard's law, where La (Lu) alloying expands (shrinks) the lattice due to its larger (smaller) atomic size than Sm. In the series $Sm_{1-x}La_xNiC_{2r}$ CDW transition ($T_{CDW} = 148$ K) for SmNiC₂ is gradually suppressed, while the ferromagnetic (FM) ordering temperature (T_c) at 17 K slightly increases up to x = 0.3. For x > 0.3, T_c starts to decrease and there is no signature that could be related with the CDW phase. Electrical resistivity, magnetic susceptibility and specific heat measurements point toward the possible presence of a FM quantum critical point (QCP) near x = 0.92, where the T_c is extrapolated to zero temperature. Superconductivity in LaNiC₂ (T_{sc} = 2.9 K) is completely suppressed with small amount of Sm inclusion near the proposed FM critical point, indicating a competition between the two ordered phases. The tunable lattice parameters via chemical substitution (La,Lu) and the ensuing change among the ordered phases of ferromagnetism, CDW and superconductivity underscores that SmNiC₂ provides a rich avenue to study the rare example of a FM QCP, where the broken symmetries are intricately correlated.

Recently, study on a quantum critical point (QCP) has attracted great interest because of the possibility of previously unknown phases of matter from the singular quantum fluctuations associated with the QCP¹. There exist a plethora of reports on antiferromagnetic (AFM) QCP systems, but examples with a ferromagnetic (FM) QCP are rare²⁻⁴. The CDW ferromagnet SmNiC₂ shows a quasi-one-dimensional electronic structure, which leads to a CDW state from the Fermi surface nesting, e.g., Peierls instability⁵. Study on the effects of hydrostatic pressure in SmNiC, has suggested that singular quantum fluctuations may create novel quantum phases in a vicinity of the projected FM quantum critical point (QCP) near 3.8 GPa (ref. 6). These novel phases, however, hide the presence of the QCP, requiring an alternative approach to probe the nature of the candidate FM QCP. Recently, Kim et al. predicted that both hydrostatic and chemical pressure have similar effects in tuning the electronic states of SmNiC₂ (ref. 7).

Ternary rare-earth nickel carbides (RNiC₂) were first reported by Bodak⁸, which are formed in an orthorhombic Amm2 (space group #38) crystal structure with Ni and the rare-earth (R) metal chains being aligned along the crystallographic a-axis. This system is of particular interest because both CDW and long range magnetic ordering phases have been observed together. Among the RNiC₂ compounds only two - SmNiC₂ and LaNiC₂ - exhibit properties other than an AFM ground state. Magnetism in $RNiC_2$ originates from the lanthanide sublattice and displays an AFM ordering in most cases (R = Ce, Pr, Nd, Gd, Tb, Dy, Ho, Er, Tm)^{9,10}. The exceptions, SmNiC₂ and LaNiC₂, are a ferromagnet and a superconductor, respectively. We note that the crystal structure of LuNiC₂ has been reported¹¹, but to our knowledge physical properties of this compound remain unknown.

In SmNiC₂, the low-temperature synchrotron x-ray diffraction reveals that a CDW state with the wave vector $\mathbf{q} = (0.5, 0.52, 0)$ forms below 148 K and disappears below the Curie temperature 17.7 K, indicating destruction of the CDW due to appearance of the ferromagnetic ordering¹². This is different from $NdNiC_2$ where a CDW is still observed below the Nèel temperature (in the AFM state)¹³. The coexistence of a CDW with superconductivity is also observed in Er₅Ir₄Si₁₀ and Lu₅Ir₄Si₁₀ (refs 14 and 15).

Superconductivity in the first member of RNiC₂ family, LaNiC₂ with $T_{sc} = 2.9$ K, was first reported by W. H. Lee, et al. almost two decades ago⁸. Superconducting critical temperature can be increased to 8K in the solid

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Figure 1. Powder x-ray diffraction (XRD) patterns. (a) XRD for the series $\text{Sm}_{1-x}\text{La}_x\text{NiC}_2$ for x = 0, 0.2, 0.4, 0.6, 0.8, 1 and $\text{Sm}_{1-y}\text{Lu}_y\text{NiC}_2$ for y = 0.2, 0.4, 0.6. (b) Enlarged image of the main peak (111) showing a shift from right to left for SmNiC₂ and LaNiC₂ respectively. Open circles are experimental points, whereas calculated diffraction patterns is represented by a solid red line. Difference between experiment and a model is shown by a blue line. The vertical ticks correspond to the Bragg peaks for Sm_{1-x}La_xNiC₂.

solution $La_{0.5}Th_{0.5}NiC_2$ (ref. 16). In contrast, Y doping of $La_{1-x}Y_xNiC_2$ decreases T_{sc} (ref. 17). Nuclear quadrupole relaxation (NQR)¹⁸ and specific heat¹⁹ measurements suggest that $LaNiC_2$ is a conventional BCS superconductor. A pure singlet pairing state has been reported for the non-centrosymmetric superconductors: Li_2Pd_3B (ref. 20), BaPtSi₃ (ref. 21), Re₃W (ref. 22), Mg₁₀Ir₁₉B₁₆ (ref. 23). However, strong evidences for an unconventional character of superconductivity in LaNiC₂ have been recently suggested by muon spin relaxation (μ SR)^{24,25} and penetration depth experiments²⁶. In addition, a phenomenological two-gap BCS model was recently proposed by Kim *et al.*⁷. The uncertainty on the nature of superconductivity for LaNiC₂ makes it more important to study the relationship between superconductivity and ferromagnetism.

Here we report crystallographic and physical properties of $\text{Sm}_{1-x}\text{La}_x\text{NiC}_2$ and $\text{Sm}_{1-y}\text{Lu}_y\text{NiC}_2$, where La and Lu were substituted for Sm to introduce negative and positive chemical pressures, respectively. X-ray diffraction study showed that the volume change in the La (or Lu) alloyed compounds obeys Vegards' law: La (Lu) alloying expands (shrinks) the volume linearly. Since La is an element with empty 4*f* orbital and Lu has a fully filled 4*f* orbital with J = 0, the chemical alloying does not introduce magnetism to the system, but is expected to induce chemical pressure effects through a change in the distance between constituent elements. Magnetic susceptibility, electrical resistivity and heat capacity data point toward the possible presence of a FM quantum critical point near x = 0.92, where the Curie temperature T_C is extrapolated to zero temperature and the specific heat divided by temperature is strongly enhanced with decreasing temperature down to the lowest measuring temperature. Comprehensive magnetic and superconducting phase diagram for both $\text{Sm}_{1-x}\text{La}_y\text{NiC}_2$ and $\text{Sm}_{1-y}\text{Lu}_y\text{NiC}_2$ is constructed for the first time.

Results

X-ray Diffraction Measurements. Powder XRD patterns are presented in Fig. 1(a) for $\text{Sm}_{1-x}\text{La}_x\text{NiC}_2$ and $\text{Sm}_{1-y}\text{Lu}_y\text{NiC}_2$. Main reflection peaks clearly shift towards lower and higher angles for $\text{Sm}_{1-x}\text{La}_x\text{NiC}_2$ and $\text{Sm}_{1-y}\text{Lu}_y\text{NiC}_2$, respectively. This is consistent with the larger (smaller) ionic radius of La^{3+} (Lu^{3+}) than Sm^{3+} and confirms successful chemical alloying. The panels (b–e) of Fig. 1 present the Le Bail refinements performed using the FULLPROF diffraction suite²⁷ for representative $\text{Sm}_{1-x}\text{La}_x\text{NiC}_2$ (x = 0, 0.2, 0.8 and 1.0) samples. The refinements confirm an orthorhombic *Amm2* crystal structure (s.g. #38) and give the lattice parameters for $\text{Sm}_{1-x}\text{R}_x\text{NiC}_2$ (R = La and Lu). More detailed inspection of the $\text{Sm}_{1-x}\text{La}_x\text{NiC}_2$ series of the XRD patterns reveals a



Figure 2. Relative change in lattice parameters: $\Delta L/L_0 = (L_x - L_0)/L_0$ and $\Delta L/L_0 = (L_y - L_0)/L_0$ for La and Lu doped samples, respectively. The lattice parameter along the *a*-axis (solid circles) shows the maximal change. Inset shows the crystal structure of SmNiC₂. Samarium and nickel atoms are represented by purple and grey balls, respectively. Small carbon atoms are marked by brown balls. Note that the C-C dimers are located along the *b* – axis.

slight broadening and splitting in the main reflection peaks for $0.2 \le x \le 0.6$. An excellent Le Bail fit ($R_{wp} = 11.7$, $R_{exp} = 8.65$, $\chi^2 = 1.84$) was obtained by assuming the presence of two phases with the same crystal structure and different lattice parameters as can be seen for $Sm_{0.8}La_{0.2}NiC_2$ (see Fig. 1c). This suggests that two distinct compounds $Sm_{1-x}La_xNiC_2$ with slightly different La concentrations (*x*) are present for the intermediate composition range. The majority phase is the one with smaller *x* – the Sm-richer variant. Such behaviour is not observed for Lu-doped series. The refined lattice parameters for $Sm_{1-x}La_xNiC_2$ are shown in Fig. S1 of the Supplementary Information. The *a* and *c* lattice parameters obey Vegard's law for the whole La concentration range, while the *b* parameter is almost constant up to x = 0.3 and linearly increases with further increasing La level. The rigid C-C dimers along the *b* direction are likely responsible for the negligible change in the *b* lattice constant for the lower La level¹⁰. The intermediate region, in which two phases with different La/Sm ratio are present, is shadowed on the diagram and the lattice parameters for the second, minority phase are shown by triangles.

The relative change $(\Delta L/L_0)$ of each lattice parameter vs. nominal concentration of La and Lu for Sm_{1-x}La_x. NiC₂ and Sm_{1-y}Lu_yNiC₂ is presented in Fig. 2. ΔL is the change of each lattice parameter compared to the lattice parameter (L₀) of the parent (SmNiC₂) compound. For example $\Delta a/a_0$ was calculated from $(a_x - a_0)/a_0$, where a_x and a_0 are the *a*-axis lattice parameters for Sm_{1-x}La_xNiC₂ and SmNiC₂, respectively. With increase in *x*, the lattice parameter along the *a*-axis expands more rapidly than that along the *b*- and *c*-axis: as large as a 7% increase in the *a*-axis is observed, while less than 1% change occurs along the *b*-axis. In contrast, Lutetium doping (*y*) causes a decrease of the lattice parameters, which is most pronounced along the *a*-axis.

Electrical Resistivity. Resistivity measurements were performed for $Sm_{1-x}La_xNiC_2$ ($0 \le x \le 1$) and $\text{Sm}_{1-v}\text{Lu}_v\text{NiC}_2$ ($0 \le y \le 0.4$) series. Temperature dependence of the normalized resistivity is shown in Fig. 3 for $Sm_{1-x}La_xNiC_2$, where resistivity values are normalized to those at 300 K for comparison. Depending on the La concentration, three features are observed. For the parent SmNiC_2 and slightly La doped (x < 0.3) Sm_{1-} $_x$ La_xNiC₂, a sharp inflection at high temperature is seen due to a charge density wave (CDW) formation. T_{CDW} was assigned as the minimum of the temperature derivative of resistivity $(d\rho/dT)$. For the parent compound SmNiC₂ the obtained T_{CDW} (=148 K) is comparable with the previous reports^{10,28}. With increase in La content, the CDW transition temperature starts to decrease rapidly, reaches to 34 K for Sm_{0.7}La_{0.3}NiC₂, and is not observed for x > 0.3. The increase in resistivity just below T_{CDW} is due to a CDW gap opening on the Fermi surface. For SmNiC₂ resistivity reaches a maximum at about 120 K and decreases with further decrease of temperature. Another feature that is clearly visible for SmNiC₂ is a sharp drop in resistivity at 17.2 K (see Fig. 3(b)), which is caused by the ferromagnetic (FM) phase transition, as will be supported by magnetic measurements. Although such behaviour is typically seen for ferromagnetic compounds below the Curie temperature, one order of magnitude decrease of resistivity is rare and may originate from the destruction of the CDW state at the same temperature. When a CDW is present, ferromagnetism is robust. In fact, there is a slight increase in the Curie temperature from 17.2 K (SmNiC₂) to 17.8 K (Sm_{0.25}La_{0.75}NiC₂) with increasing *x*. For x > 0.3, the CDW transition is no longer observed and the Curie temperature starts to decrease with increasing La concentration, which suggests a strong correlation between the CDW and ferromagnetic phases.

Figure 3(c) presents resistivity data for Sm_{1-x}La_xNiC₂ system with high La concentration ($x \ge 0.9$). A sharp superconductivity transition is observed for LaNiC₂ and Sm_{0.02}La_{0.98}NiC₂ with $T_{sc} = 3.4$ K and 2 K, respectively. The samples with slightly lower x (0.97 and 0.95) do not show a transition to the zero-resistance superconducting state above 1.8 K, suggesting that the resistivity drop is due to filamentary nature of the SC phase. The rapid suppression of superconductivity is often observed in the presence of magnetic impurities that act as strong scattering centers to destroy superconducting Cooper pairs. One such example is La_{1-x}Gd_x alloy, where only 1% of Gd alloying suppresses T_{sc} from almost 6 K for La to 1 K for La_{0.99}Gd_{0.01} (ref. 29).

In contrary to La alloying, the low-*T* normalized resistivity of SmNiC₂ is enhanced by Lu alloying (see Fig. S2 in the Supplementary Information). The slope of the resistivity $(d\rho/dT)$ at high temperatures decreases with



Figure 3. Electrical resistivity for La-doped SmNiC₂. (a) Temperature dependence of the normalized electrical resistivity for Sm_{1-x}La_xNiC₂, $0 \le x \le 1$. Low-temperature resistivity plots of the ferromagnetic $(0 \le x \le 0.8)$ and superconducting transitions $(0.9 \le x \le 1)$ are shown in (**b**,**c**), respectively.

increase of Lu concentration, too. The 20% Lu-doped compound ($Sm_{0.8}Lu_{0.2}NiC_2$) reveals both CDW and FM behaviour. The Curie temperature is suppressed to 10 K, whereas the CDW transition temperature increases to 152 K, a 4 K increase from the pure compound. It is in contrast to the La-doping ($Sm_{1-x}La_xNiC_2$): at the same concentration level (x = 0.2), T_{CDW} is suppressed to 126 K, a 24 K decrease.

Magnetization Measurement. Magnetic characterization of the $Sm_{1-x}La_xNiC_2$ series is presented in Fig. 4. There are no anomalies at around 13 K and 25 K that originates from the presence of SmNi₂ binary phase⁹. The absence of anomalous features suggests high quality of our samples. The molar magnetic susceptibility ($\chi_{\rm M}$) at $\mu_{o}H = 0.1 \text{ T}$ (Fig. 4a) shows a rapid increase with decreasing temperature below 17 K and saturates between 0.5 to 1.2 emu/Sm mol for different La concentrations at low temperatures. At this moment, it is difficult to point out a correlation between the saturated moments and alloying concentrations because the samples are in a polycrystalline form. The FM transition temperature determined by the minimum of the temperature derivative of susceptibility ($d\chi/dT$), as shown in Fig. 4b, is in good agreement with the Curie temperature $T_{\rm C} = 17.2$ K from the resistivity measurement. Ferromagnetism in $Sm_{1-x}La_xNiC_2$ is robust. T_C initially increases with La concentration, reaches a maximum of 18 K at x = 0.25, and then decreases with further increasing La level, showing a small dome shape in the T-x phase diagram. It is interesting to note that the ferromagnetic phase persists up to the nominal La concentration x = 0.86: a ferromagnetic transition at as low as 2.5 K was observed for the composition x = 0.86in the series Sm_{1-x}La_xNiC₂. In order to confirm the ferromagnetic state and precisely estimate the Curie temperature, the Arrott plot analysis was performed. A series of isothermal magnetisation curves in the immediate vicinity of the Curie temperature were measured. In a plot of H/M vs M^2 , the isotherm which passes through the origin gives the best estimate of Curie temperature because $H/M = \beta M^2$ at $T = T_c$ (ref. 30). The Arrott plot presented in Fig. 5 shows that the ferromagnetic transition temperature for $Sm_{0.2}La_{0.8}NiC_2$ is 5.5 K, very close to $T_C = 5.3$ K estimated for the same sample from the minimum of $d\chi/dT$. Using the Arrott plot, $T_{\rm C} = 3.5$ K was estimated for Sm_{0.14}La_{0.86}NiC₂ (not shown here).

A molar magnetic susceptibility at 300 K for the Sm_{1-x}La_xNiC₂ series is shown in Fig. S3 of the Supplementary Information, which linearly decreases with increasing La concentration. As expected, the extrapolated line reaches zero at LaNiC₂. The same experimental data normalized per Sm-mol is almost independent of *x* and is about $\chi(300\text{K}) = 1.8 \times 10^{-3} \text{ emu/Sm-mol}$, indicating that La is successfully substituted for Sm.

According to H. Onodera *et al.*, spontaneous magnetization (*M*) along the *a*-axis of the single crystal (~ $0.32 \mu_B$) is smaller than $0.72 \mu_B$ for Sm³⁺, while it is negligible along the *b*- and *c*-axis⁹. The fact that we have obtained $0.19 \mu_B$ at 2 K could be ascribed to the polycrystalline form of the measured sample. Such a small value has been explained by the mixed valence state of Sm²⁺ and Sm³⁺ ions or crystalline electric field (CEF) effects. A mixed valent state is common in Sm containing compounds. Above the ferromagnetic transition temperature the entire series did not follow the Curie-Weiss behaviour, which might be pertinent to the mixed valent state of Sm ions. It is imperative to study the exact nature of magnetic interactions present in SmNiC₂ by measurements like neutron diffraction to gain a deep insight on the valence of Sm ions.



Figure 4. Magnetization for La-doped SmNiC₂. (a) Molar magnetic susceptibility $\chi_{\rm M}$ plotted against temperature for Sm_{1-x}La_xNiC₂, $0 \le x \le 0.80$. (b) First temperature derivative of the susceptibility is representatively shown to determine the FM transition temperature $T_{\rm C}$ for x = 0, 0.2, 0.4, 0.6, 0.8. Arrow marks the $T_{\rm C}$ for x = 0, where $d\chi_{\rm M}/dT$ is a minimum. (c) Evolution of $T_{\rm C}$ as a function of La concentration x.





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Heat Capacity Measurements. Heat capacity $C_p(T)$ of the polycrystalline SmNiC₂ (black dots) and LaNiC₂ (blue solid line) is plotted as a function of temperature in Fig. 6(a). At the highest temperature, C_p (at 300 K) reaches approximately 80% of the value expected by Dulong-Petit law 3nR value ~100 J mol⁻¹K⁻¹, suggesting that the Debye temperature for SmNiC₂ exceeds 300 K. A small anomaly at around 153K (inset b) is likely caused by the CDW ordering, although this temperature is 5 K higher than a CDW temperature estimated from the resistivity and magnetization measurements. A huge anomaly is visible at low temperature and details are presented in Fig. 6(c). In the zero-field data, a λ -shape transition occurs at $T_C = 17$ K, which is in agreement with the Curie temperature estimated by resistivity and magnetization techniques. With increasing magnetic field, the transition is broadened and split into two peaks, indicating an additional field-induced phase transition. A simple subtraction of the LaNiC₂ specific heat from the SmNiC₂. The magnetic entropy recovered at T_C (= 17.1 K) accounts for 80% of *R*ln2, the entropy expected for the doublet ground state of the J=5/2 multiplet for Sm³⁺ (4f⁶). Incomplete recovery of the entropy at T_C could be ascribed to either the fluctuating valence between Sm³⁺ and Sm²⁺ or the Kondo screening effects of Sm 4f spins by the itinerant electrons.



Figure 6. Specific heat of La-doped SmNiC₂. (a) Temperature dependence of the specific heat C_p for SmNiC₂ (black dots) and LaNiC₂ (blue line). The inset (**b**,**c**) show C_p in the vicinity of the CDW and PM-FM transition, respectively. The inset (**d**) displays the magnetic entropy as a function of temperature for SmNiC₂.



Figure 7. Low temperature specific heat data of Sm_{1-x}La_xNiC₂ (0.86 < x < 1.0). The vertical line mark the Curie temperature estimated from the Arrott plot of the magnetization. Panel (b) shows C_p/T versus T^2 for LaNiC₂ under zero magnetic field (open circles) and 3T (open squares). The solid red line represents fit to $C_p/T = \gamma + \beta T^2$. Panel (c) shows C_p/T versus T_{sc} in the vicinity of superconducting transition. Solid line shows the equal entropy construction.

Heat capacity data (C_p/T) for La-rich samples is selectively presented on a semi-logarithmic scale in Fig. 7. A sharp SC transition for LaNiC₂ (open circles) is visible at 2.9 K in Fig. 7(a). When Sm is alloyed in LaNiC₂, the SC jump in the specific heat is quickly suppressed and there is no signature for the SC transition down to 1.8 K, the lowest measured temperature, for 3% Sm concentration (x = 0.97). With further increasing Sm concentration, the



Figure 8. Temperature vs concentration phase diagram. Top panel shows the change in volume of the orthorhombic series $\text{Sm}_{1-x}\text{La}_x\text{NiC}_2$ ($0 \le x \le 1$) and $\text{Sm}_{1-y}\text{Lu}_y\text{NiC}_2$ ($0 \le y \le 0.4$) with respect to composition. The bottom panel shows the phase diagram of temperature vs. composition. For $\text{Sm}_{1-x}\text{La}_x\text{NiC}_2$, the CDW ordering disappears above x = 0.3, while the ferromagnetic phase survives up to x = 0.86. The superconductivity is rapidly suppressed for x < 1 in $\text{Sm}_{1-x}\text{La}_x\text{NiC}_2$. For the Lu-doped $\text{Sm}_{1-y}\text{Lu}_y\text{NiC}_2$, T_{CDW} disappears for y > 0.2 and the magnetic phase decreases with increasing Lu concentration y. The open triangles are data points obtained from the magnetization measurements, while the solid circles are from resistivity measurements.

low-temperature specific heat divided by temperature (C/T) increases with decreasing temperature and shows a peak at 14% Sm concentration due to the FM ordering, where the brown vertical line marks the Curie temperature estimated from the Arrott plot of magnetization. Even though the lowest measured temperature is limited to 1.8 K, the singular enhancement in C/T is clearly visible as Sm concentration approaches 10%, indicating the possible presence of a FM quantum critical point at that concentration.

Figure 7(b,c) magnifies the specific heat of LaNiC₂ near the SC transition temperature. The normal-state specific heat measured at $\mu_0 H = 3$ T, shown by open squares in Fig. 7b, was fitted to $C_p = \gamma T + \beta T^3$, where the first and second terms represent electronic and lattice contributions, respectively. The fit to these data allows us to estimate $\gamma = 7.3(1)$ mJ mol⁻¹K⁻², and $\beta = 0.088(5)$ mJ mol⁻¹K⁻⁴. The simple Debye model connects the β coefficient and the Debye temperature Θ_D through $\Theta_D = (12\pi^4 n R/5\beta)^{1/3} = 445$ K, where R = 8.314 J mol⁻¹K⁻¹ and *n* is the number of atoms per formula unit (*n* = 4 for LaNiC₂). This value is slightly lower than reported $\Theta_D = 456$ K for YNiC₂ (ref. 31) and can be explained by a simple mass relationship: larger La mass than Y should result in lower Θ_D .

From an equal entropy construction shown by the solid lines in Fig. 7c, superconducting critical temperature was obtained to be $T_{sc} = 2.9$ K, which is lower than that from the resistivity measurement ($T_{sc} = 3.4$ K). The ratio between the specific heat jump (ΔC) at T_c and the Sommerfeld coefficient (γ), $\Delta C/\gamma T_{sc}$, is 1.33, which is close to the BCS predicted value of 1.42 for a weakly coupled superconductor.

Discussion

It is interesting to note that the CDW phase is driven by the one-dimensional (1D) anisotropy along the *a*-axis in the electronic structure⁷. The 1D anisotropy increases with chemical or physical pressure mainly because the Ni (or Sm) chain along the *a*-axis is affected the most (either compressed or expanded). The phase diagram for the series Sm_{1-x}La_xNiC₂ is shown in Fig. 8. The results from both transport and magnetic measurements are used to plot the phase diagram. In the top panel of Fig. 8, the unit cell volume is plotted against *x* and *y*, which evidently shows that both Sm_{1-x}La_xNiC₂ and Sm_{1-y}Lu_yNiC₂ follow Vegard's law. In the bottom plot of Fig. 8, we can see a slight increase in T_{CDW} with the inclusion of Lu in the lattice due to a positive chemical pressure, while there occurs a decrease in T_{CDW} with increase in La due to a negative chemical pressure in the lattice. The CDW transition is getting suppressed from 148 K (*x*=0) to 34 K (*x*=0.3) with La doping concentration *x* and completely destroyed for x > 0.3.

When a CDW is present, there is a slight increase in the FM Curie temperature from 17.2 K to 17.8 K (which is confirmed from both transport and magnetic measurements). Recently B-doped SmNiC₂ reported a similar increase in $T_{\rm C}$ from 17.5 K to 23.1 K until the CDW is present³². Once a CDW is suppressed in Sm_{1-x}La_xNiC₂, however, the Curie temperature starts to decrease and is suppressed down to 2.5 K at x = 0.86. Inset of the bottom panel magnifies the temperature-La concentration phase diagram near the pure LaNiC₂, where solid lines are guides to eyes. Both $T_{\rm C}$ and $T_{\rm sc}$ could be extrapolated to zero Kelvin at x = 0.92, underscoring the possibility of a FM QCP that was proposed by the singular enhancement in the low-*T* specific heat. Electrical resistivity measurements show that the first-order FM transition in SmNiC₂ changes to the second order or a weakly first order for higher La concentration, suggesting that the disorder introduced by La substitution may be important to the realization of the FM QCP in SmNiC₂ (see Fig. S4 in the Supplementary Information). In the case of Lu doped SmNiC₂, $T_{\rm C}$ is strongly suppressed from 17.5 K to 8.8 K for y = 0.4. As shown in Fig. S2 in the Supplementary Information, however, the disparate resistivity behaviour near $T_{\rm C}$ for y = 0.4 demands further investigation on the precise nature of magnetic ordering for different Lu doped level.

m FM ordering persists up to x = 0.92 in LaNiC₂-SmNiC₂ system, where the dilution of Sm local moments by La substitution exceeds the percolation limit³³. Once a CDW is destroyed, the relationship in LaNiC₂-SmNiC₂ solid solution is almost similar to La-Gd alloy system, where the SC transition temperature of La elemental metal decreases rapidly with increasing Gd content and is completely suppressed at 0.9 at% Gd. La-Gd alloy containing just 3% Gd becomes ferromagnetic at 1.3 K (ref. 29). However, Gd in Y did not show any FM until 10 at% Gd³⁴. With the introduction of other rare earths in binary alloys the Neel and Curie points are generally lowered. We have also synthesized other rare earth substitution such as Y alloying to SmNiC₂. For 20% Y concentration, the CDW is completely destroyed and the Curie temperature is also dropped to 10 K (not shown here).

Conclusions

We have successfully synthesized SmNiC₂ solid solution with La (or Lu) and investigated the tuneable behaviour from the CDW ferromagnet to the weakly coupled superconductor. La alloying in SmNiC₂ expands the lattice parameters (negative pressure), while Lu alloying shrinks the lattice parameters (positive pressure). The CDW transition temperature ($T_{CDW} = 148$ K) in Sm_{1-x}La_xNiC₂ decreases with increasing La inclusion *x* because of the poor Fermi surface nesting conditions from the La alloying. La (or Lu) alloying also dilutes the density of Sm local moments because there is no f electron in the substituent, therefore suppressing the FM phase. The Curie temperature, however, does not decrease monotonically with La concentration, but shows a maximum near x = 0.3, underscoring that the CDW and ferromagnetic phases compete against each other. Superconductivity is observed only for La rich compounds (x > 0.92), where the SC transition temperature ($T_{sc} = 3.4$ K) for LaNiC₂ is quickly suppressed with increasing Sm contents and to zero Kelvin near 8% Sm concentration. When combined with the fact that both Curie temperature and SC transition temperature is suppressed to zero Kelvin near $x_c = 0.92$, the singular enhancement of the low-*T* specific heat at the critical concentration x_c points toward the presence of a ferromagnetic quantum critical point (QCP). We note that disorder introduced by the La substitution may be conducive to the realization of the FM QCP in SmNiC₂ (ref. 35). More study is in progress to elucidate the nature of the candidate QCP.

Methods

The series of compounds $Sm_{1-x}La_xNiC_2$ ($0 \le x \le 1$) and $Sm_{1-y}Lu_yNiC_2$ ($0 \le y \le 0.4$) were synthesized by the arc-melting technique, using constituent elements of purity 99.9% or higher. The weight loss after arc melting was less than 1%, indicating that the nominal concentration is close to the actual alloying level. Since WDS analysis corroborates this conclusion, the nominal concentration was used throughout this manuscript. The arc-melted samples were annealed at 1173K for ten days in a sealed evacuated quartz tube. The annealed samples were quenched in NaCl-ice water mixture. Structural characterization was performed by the powder x-ray diffraction (PXRD) method using a Rigaku diffractometer with Cu K α radiation. The lattice parameters of the samples were determined by LeBail profile refinements of PXRD carried out using the FULLPROF software²⁷. Resistivity measurements were performed using a standard four probe technique employing a Quantum Design Physical Property Measurement System (PPMS). The contacts were made by spot welding of platinum wires on the sample surface. Heat capacity was measured in temperature range 1.9 K < T < 300 K at fields up to 9 T by using the thermal relaxation technique (PPMS system). Magnetic measurements were carried out using a Quantum Design Magnetic Property Measurement System (MPMS).

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Author Contributions

G.P., I.K. and T.K. synthesized the La- and Lu-doped SmNiC₂ compounds. T.K. performed powder x-ray diffraction analysis. J.S. measured heat capacity and G.P., I.K. and S.S. performed resistivity and magnetization measurements. The results were discussed with all authors. The manuscript was written by G.P., T.K. and T.P. with inputs from all authors.

Additional Information

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