

Two-Dimensional Half-Metallic and Semiconducting Lanthanide-Based MXenes

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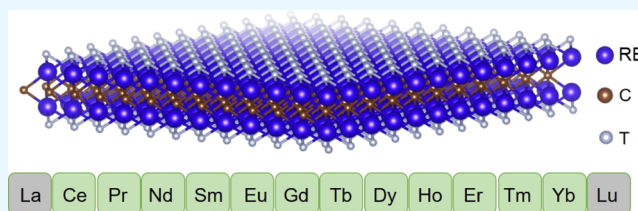


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ABSTRACT: As a large family of two-dimensional materials, MXenes have attracted intensive attention in recent years. For more functional applications, it is of great significance to determine new MXene members. Here, we theoretically expand the M elements of MXenes to the lanthanide series. Based on density functional theory calculations, the bare lanthanide-based carbides M_2C ($M = Ce, Pr, Nd, Sm, Eu, Gd, Tb, Dy, Ho, Er, Tm, \text{ and } Yb$) and the corresponding fluorine- and hydroxyl-terminated configurations are investigated. Most of the fluorine- and hydroxyl-terminated MXenes investigated are half-metals. Specifically, in the half-metallic Eu_2CF_2 , the spin-down states show a band gap larger than 2 eV, implying this configuration's potential applications in spin generation and injection. Both Gd_2CT_2 ($T = F \text{ and } OH$) are magnetic semiconductors. The former shows an indirect band gap of 1.38 eV, while the latter presents a direct one of 0.882 eV. These two configurations also show large magnetic moments higher than $13.7 \mu_B$ per unit cell. All the hydroxyl-terminated MXene members show relatively low work functions, with the lowest value of 1.46 eV determined in $Tm_2C(OH)_2$. These predicted electronic properties imply that the lanthanide-based MXenes could have potential applications in spintronics, information storage, near-infrared detectors, field effect transistors, and field emitter cathodes.



INTRODUCTION

With the enhanced degree of integration, conventional silicon-based electronics encounter increasing challenges. For instance, the behavior of the circuit with the chip size at the nanoscale becomes unstable due to the quantum tunneling effect. Moreover, the performance of silicon transistors degrades at high temperatures because of the small band gap and unsatisfactory thermal conductivity. Regarding the development of next-generation electronics, to find promising semiconducting materials to replace or combine with silicon has been demonstrated to be an effective approach.¹ On the other hand, spintronics have been demonstrated to show many advantages over conventional semiconductor devices, such as nonvolatility, increased data processing speed, and decreased electric power consumption.² In order to utilize the spin degree of freedom, the corresponding spintronic materials should possess high spin polarization.³ As a consequence, many efforts have been devoted to developing more semiconducting or spintronic materials.^{2,4,5} Combined with the increasing demands on paper displays and wearable electronics,^{6,7} two-dimensional (2D) semiconducting or spintronic films have attracted increasing attention. Regarding the 2D semiconductors, a number of configurations such as hexagonal boron nitride (h-BN),⁸ phosphorene,⁴ and molybdenum disulfide (MoS_2),^{5,9} have been synthesized, but improvements are also required for these configurations. For example, the band gap of h-BN is extremely large (5.5 eV), and phosphorene is prone to chemical degradation upon exposure to ambient conditions.¹⁰ For 2D spintronic materials, an

intrinsic 2D chromium triiodide CrI_3 was synthesized through mechanical exfoliation,^{11,12} and a 2D ferromagnetic van der Waals crystal $Cr_2Ge_2Te_6$ was also developed.¹³ However, these two synthesized 2D materials are magnetic metals with low Curie temperatures. Consequently, it is crucial to predict and synthesize more 2D semiconducting and spintronic materials.^{14–20}

In 2011, a new family of 2D materials named MXenes was proposed,²¹ which is generally synthesized from the selective etching of layered parent $M_{n+1}AX_n$ phases (or denoted as MAX phases, where M denotes an early transition metal; A is an A-group element, such as Al and Si; X is C and/or N; and n is generally 1–3). More than 150 MAX phases have been synthesized,^{22,23} and the corresponding MXene family generally possesses versatile chemical elements and various configurations. Since hydrofluoric acid is usually adopted as etchant, the surface of MXene is normally functionalized by fluorine and hydroxyl groups.²⁴ To date, more than 40 MXene members have been synthesized,^{21,25–30} which show potential applications ranging from energy storage,^{31,32} electromagnetic interference shielding,³³ reinforcement materials,³⁴ cata-

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lysts,^{35–37} to sensors.³⁷ Among these 2D carbides, most configurations are metallic.³⁶ Only Mo_2CT_x ³⁸ and $\text{Mo}_2\text{TiC}_2\text{T}_x$ ³⁹ present semiconducting-like behaviors in experiments, with T_x denoting surface functional group. Theoretically, several MXene members were determined to be semiconducting, such as M_2CO_2 ($\text{M} = \text{Sc}, \text{Ti}, \text{Zr}, \text{Hf}, \text{Mn}$ and W), M_2CF_2 (Sc, Cr and Mo) and $\text{Sc}_2\text{C}(\text{OH})_2$.^{40–48} Mn_2CF_2 was found to be a half-metal with a high Curie temperature of 520 K.⁴⁴ M_2C and M_2N ($\text{M} = \text{Ti}$ and V) were also predicted to show half-metallic or spin gapless semiconducting feature under strains.⁴⁹ Obviously, it is feasible to tune the electronic properties of MXenes by varying their transition metals and surface groups.

Specially, MXenes with its transition metal in group IIIB generally show semiconducting characteristics regardless of functional groups, such as M_2CT_2 ($\text{M} = \text{Sc}, \text{Y}, \text{La}$ and Lu ; $\text{T} = \text{O}, \text{F}$ and OH).^{50,51} $\text{M}_2\text{C}(\text{OH})_2$ ($\text{M} = \text{Sc}, \text{Y}$ and Lu) are the only three direct-band-gap semiconducting MXene members up to date.^{42,50} Sc_2CO_2 presents large out-of-plane polarization, which could have potential applications in piezoelectric and ferroelectric materials.⁵² Both Sc_2CT_2 ($\text{T} = \text{F}$ and OH) and Lu_2CT_2 ($\text{T} = \text{F}$ and OH) could be used in high-frequency semiconductor devices on the basis of their promising electron mobilities, where the electron mobility of $\text{Lu}_2\text{C}(\text{OH})_2$ is nearly 10 times that of silicon.^{43,51} Moreover, the intermediate states $\text{Sc}_2\text{C}(\text{OH})_x\text{O}_{2-x}$ among the transition from $\text{Sc}_2\text{C}(\text{OH})_2$ to Sc_2CO_2 show magnetic semiconducting properties.⁵⁴

Based on the excellent semiconducting and spintronic properties reported for the MXenes with group IIIB transition metals, here we expand the M element of MXene to the lanthanide series. Based on the first-principles calculations, 12 bare M_2C MXenes ($\text{M} = \text{Ce}, \text{Pr}, \text{Nd}, \text{Sm}, \text{Eu}, \text{Gd}, \text{Tb}, \text{Dy}, \text{Ho}, \text{Er}, \text{Tm}$ and Yb) are first studied. The left lanthanide-based M_2C ($\text{M} = \text{La}$ and Lu) have been reported previously.^{50,51} All of the structures investigated are found to be stabilized in the T-type configuration. Further, the structures, stabilities and electronic properties of corresponding fluorine- and hydroxyl-functionalized members are discussed. Most functionalized configurations are also predicted to be stable. The fluorine- and hydroxyl-functionalized structures are generally half-metallic except for M_2CT_2 ($\text{M} = \text{Ce}, \text{Tm}, \text{Gd}$ and Yb). M_2CT_2 ($\text{M} = \text{Ce}, \text{Tm}$ and Yb ; $\text{T} = \text{F}$ and OH) are magnetic metals. Gd_2CT_2 ($\text{T} = \text{F}$ and OH) are magnetic semiconductors with their band gaps are 1.38 and 0.882 eV, respectively. Moreover, the hydroxyl-terminated MXenes generally show relatively low work functions.

COMPUTATIONAL DETAILS

All the first-principles calculations are carried out using the Vienna Ab initio Simulation Package (VASP) code.⁵⁵ The Perdew–Burke–Ernzerhof (PBE) scheme of generalized gradient approximation (GGA)⁵⁶ is adopted for the exchange–correlation functional. A plane-wave cutoff energy of 500 eV based on the projector augmented-wave pseudopotential⁵⁷ is utilized for describing the ion–electron interactions. All the structures are relaxed until the forces on each atom are less than 1.0×10^{-4} eV/Å, and the energy tolerance is set as 1.0×10^{-6} eV/unit cell. A Γ -centered $12 \times 12 \times 1$ k -point mesh sampled in the hexagonal Brillouin zone (BZ) is adopted for structural optimization, and a 60 k -point grid along the high-symmetry routes of BZ is employed for plotting the electronic energy band. To eliminate the interactions of neighboring layers, a vacuum layer with its

thickness larger than 15 Å is adopted. The layer thickness of MXene is chosen as the monomer thickness in its corresponding multilayer configuration, as reported in our previous work.⁵⁸ To test the dynamic stability of MXenes, $4 \times 4 \times 1$ supercells are considered for the phonon calculations from the density functional perturbation theory (DFPT)⁵⁹ combined with the Phonopy⁶⁰ and VASP⁵⁵ software. The work function is calculated as the difference between the vacuum level and Fermi level of each configuration, and the vacuum level is obtained by plotting the average potential along the z -axis perpendicular to the MXene surface. All the calculations are based on the spin-polarization scheme. All the structures are visualized in the VESTA code.⁶¹

In order to test the thermodynamic stability of the functionalized configurations, the formation energy is calculated according to the following equation⁴⁰

$$E_f = E_{\text{tot}}(\text{M}_2\text{CT}_2) - E_{\text{tot}}(\text{M}_2\text{C}) - E_{\text{tot}}(\text{T}_2) \quad (1)$$

where $E_{\text{tot}}(\text{M}_2\text{CT}_2)$ denotes the total energy of the functionalized MXene, $E_{\text{tot}}(\text{M}_2\text{C})$ is the total energy of bare MXene, and $E_{\text{tot}}(\text{T}_2)$ is the total energy of F_2 or $\text{O}_2 + \text{H}_2$, as the fluorine and hydroxyl groups are studied in our work.

RESULTS AND DISCUSSION

In order to investigate the lanthanide-based MXenes, their structures and stabilities are first studied. The 2D bare transition-metal carbides could be categorized as T- and H-types,⁶² and both types of structures are shown in Figure 1. For

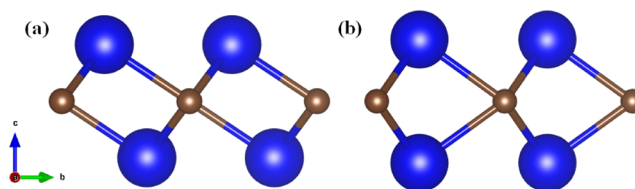


Figure 1. (a,b) Side views of the T- and H-type M_2C configurations. Blue and brown balls show the lanthanide and carbon atoms, respectively.

the T-type configuration shown in Figure 1a, both the top and bottom transition metals are located in the triangle centers of neighboring carbon atoms. Many reported MXenes appeared in this type;²⁶ for the H-type structure, the transition-metal atoms in the top layer are located on the top sites of the bottom transition metals. The well-known MoS_2 is stabilized in the H-type configuration.⁵

Based on the unit cells, the relative total energies between the T- and H-type M_2C ($\text{M} = \text{Ce}, \text{Pr}, \text{Nd}, \text{Sm}, \text{Eu}, \text{Gd}, \text{Tb}, \text{Dy}, \text{Ho}, \text{Er}, \text{Tm},$ and Yb) are provided in Table S1 in Supporting Information. From the table, it is observed that the T-type configurations generally show lower total energies, which implies that all the bare M_2C MXenes are stabilized in the T-type. The lattice parameters of all the T- and H-type M_2C are listed in Table S2. The lattice parameters in M_2C generally decrease with the increasing atomic number of M , which could be ascribed to M with a larger atomic number that has a smaller atomic radius.⁶³ Further, the decomposition energies of the T-type M_2C into its potential competing phases are studied and provided in Table S3. Most of the structures are metastable, which show negative decomposition energies, but their absolute values are smaller than 200 meV/atom.⁶⁴ M_2C ($\text{M} = \text{Sm}, \text{Dy}, \text{Er},$ and Tm) show positive decomposition

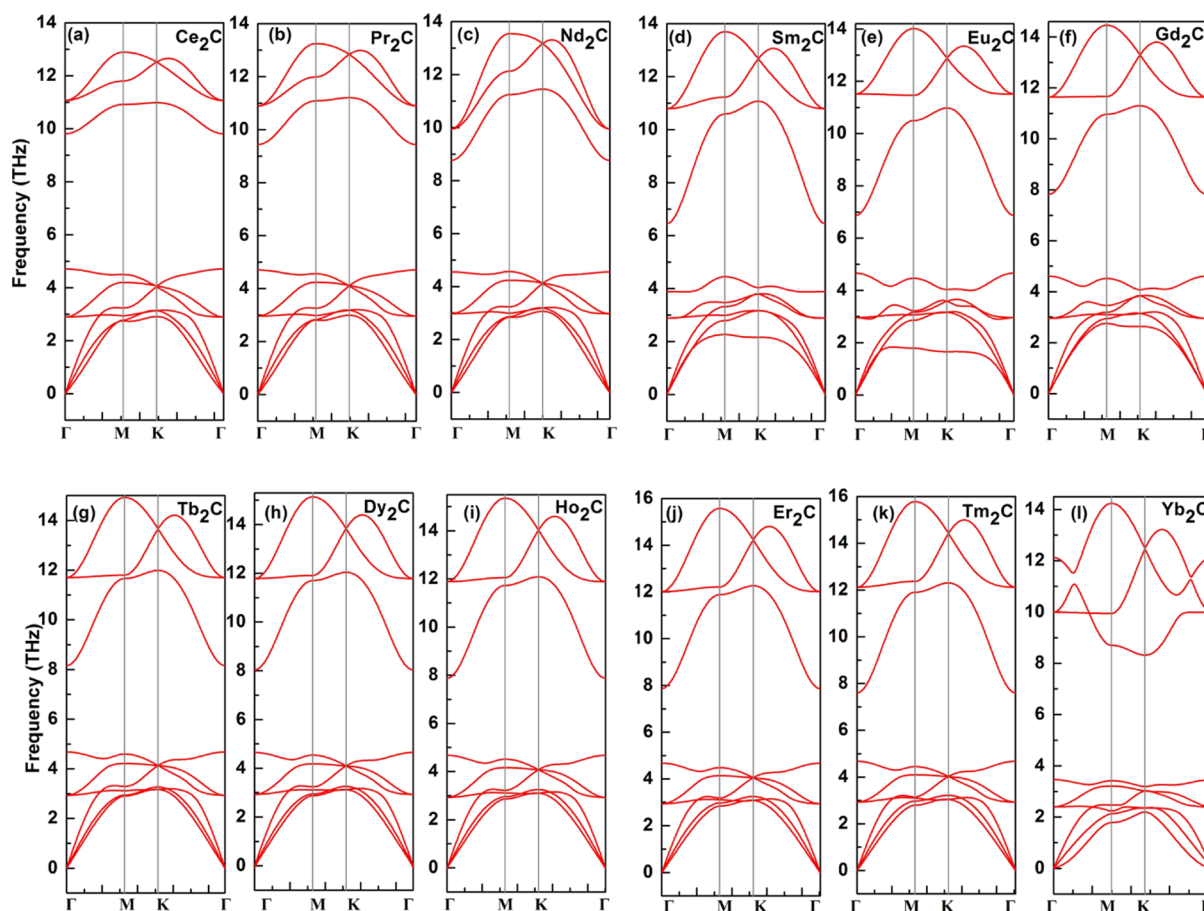


Figure 2. (a–l) Phonon dispersions of the T-type M_2C ($M = \text{Ce, Pr, Nd, Sm, Eu, Gd, Tb, Dy, Ho, Er, Tm, and Yb}$) MXenes.

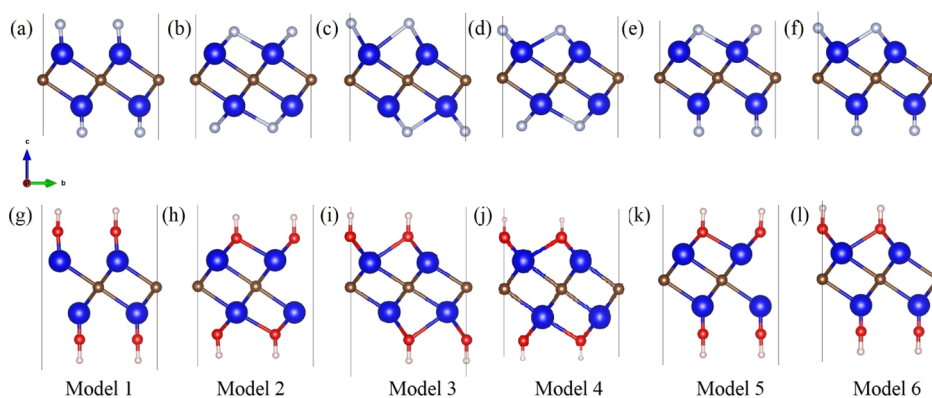


Figure 3. (a–f) Side views of models 1–6 of the fluorine-functionalized MXenes. (g–l) Side views of models 1–6 of the hydroxyl-terminated MXenes. Blue, brown, silver, red, and white balls show the lanthanide, carbon, fluorine, oxygen, and hydrogen atoms, respectively.

energies, which imply that these configurations are thermodynamically stable. The dynamic stabilities of these T-type configurations are also evaluated. All the phonon dispersions of these configurations are calculated and provided in Figure 2. From this figure, the phonon frequencies in all the dispersions are observed to be positive, which implies that all the T-type M_2C configurations are dynamically stable.^{43,59} In addition, there are evident phonon gaps in these phonon dispersions, which could be ascribed to the remarkable difference in the mass of carbon and lanthanide atoms. The evident phonon gaps might enable the applications of these M_2C configurations in phononic crystals.⁶⁵

Based on the stable T-type configurations, spin-polarized band structures for all the M_2C configurations are calculated,^{66,67} and the corresponding results are provided in Figure S1 in Supporting Information. From the figure, all the bare M_2C MXenes are observed to be magnetic metals. The corresponding magnetic moments based on the unit cell of M_2C and the magnetic contribution from the f orbital in each M are provided in Table S4. Apparently, the magnetism of M_2C is mainly contributed by the f orbital. Further, the functionalized M_2C MXenes are studied. The functional group on MXene is dependent on the type of etchant with the selective etching approach.^{21,26} Using hydrofluoric acid as the

etchant, the surface of MXene is generally functionalized by the F and OH groups.²¹ Therefore, we focus on the fluorine and hydroxyl functional groups in this work. According to previous reports,^{40,42} the locations of functional groups could vary in different MXenes. To determine the stable functionalized structure, six possible configurations, as shown in Figure 3 are investigated. In this figure, the first and second rows present six different structural models of the fluorine- and hydroxyl-functionalized MXenes, respectively. To facilitate elaboration, the configurations from the left to the right side are denoted as model 1, 2, 3, 4, 5, and 6, respectively. In model 1, the functional groups are located on the top sites of the neighboring lanthanide atoms on both sides, while the groups are on the top sites of the bottom metals on both sides in model 2. In model 3, the functional groups are on the top sites of the central carbon atoms on both sides. In model 4, the functional groups are on the top sites of central carbon atoms on one side, which are on the top sites of the bottom metals on the opposite side. Regarding model 5, the groups are on the top sites of neighboring metals on one side, which are on the top sites of the hollow centers of neighboring metals on the other side. In the last model, the groups are on the top sites of neighboring metals on one side, which are on the top sites of the central carbon atoms on the opposite side. After structural relaxation, the relative total energies of these models for fluorine-functionalized MXenes are provided in Table 1, and

Table 1. Relative Total Energies (in eV) of Fluorine-Functionalized MXenes M_2CF_2 ($M = Ce, Pr, Nd, Sm, Eu, Gd, Tb, Dy, Ho, Er, Tm,$ and Yb)

MXenes	model 1	model 2	model 3	model 4	model 5	model 6
Ce_2CF_2	3.16	0.00	0.714	0.353	1.42	1.89
Pr_2CF_2	3.14	0.00	0.746	0.368	1.44	1.93
Nd_2CF_2	3.13	0.00	0.779	0.382	1.46	1.96
Sm_2CF_2	3.12	0.00	0.845	0.406	1.48	2.01
Eu_2CF_2	3.15	0.00	0.890	0.425	1.49	2.05
Gd_2CF_2	3.08	0.00	0.906	0.422	1.48	2.04
Tb_2CF_2	3.13	0.00	0.928	0.426	1.46	2.04
Dy_2CF_2	3.02	0.00	0.946	0.427	1.45	2.09
Ho_2CF_2	2.99	0.00	0.972	0.426	1.43	2.04
Er_2CF_2	3.05	0.00	1.08	0.517	1.51	2.13
Tm_2CF_2	2.93	0.00	0.997	0.422	1.40	2.04
Yb_2CF_2	4.50	0.00	0.547	0.256	2.15	2.61

Table 2. Relative Total Energies (in eV) of Hydroxyl-Terminated MXenes $M_2C(OH)_2$ ($M = Ce, Pr, Nd, Sm, Eu, Gd, Tb, Dy, Ho, Er, Tm,$ and Yb)

MXenes	model 1	model 2	model 3	model 4	model 5	model 6
$Ce_2C(OH)_2$	2.60	0.00	0.368	0.180	1.26	1.48
$Pr_2C(OH)_2$	2.63	0.00	0.399	0.192	1.28	1.52
$Nd_2C(OH)_2$	2.65	0.00	0.433	0.204	1.54	1.29
$Sm_2C(OH)_2$	2.66	0.00	0.469	0.232	1.32	1.60
$Eu_2C(OH)_2$	2.69	0.00	0.553	0.245	1.33	1.62
$Gd_2C(OH)_2$	2.68	0.00	0.610	0.274	1.35	1.66
$Tb_2C(OH)_2$	2.63	0.00	0.639	0.255	1.32	1.66
$Dy_2C(OH)_2$	2.64	0.00	0.564	0.261	1.34	1.67
$Ho_2C(OH)_2$	2.67	0.00	0.759	0.300	1.34	1.67
$Er_2C(OH)_2$	2.61	0.00	0.632	0.305	1.33	1.70
$Tm_2C(OH)_2$	2.63	0.00	0.810	0.262	1.60	1.28
$Yb_2C(OH)_2$	3.64	0.00	0.302	0.136	1.83	2.05

the corresponding values for the hydroxyl-terminated structures are given in Table 2. From these tables, it is observed that model 2 shows the lowest energy for all the fluorine- and hydroxyl-functionalized configurations investigated. This stable configuration is similar to the structures of many other MXenes reported.⁴² Model 1 generally presents the highest energy, and model 4 shows a little higher value than that of model 2. Taking $Ce_2C(OH)_2$ as an example, the total energy of model 4 is only 0.180 eV higher than that of model 2.

Based on model 2, the thermodynamic stabilities of these lanthanide-based MXenes are further examined. According to eq 1, the formation energies for all the functionalized configurations are calculated and provided in Table 3. From

Table 3. Formation Energies E_f (in eV) of the Lanthanide-based MXenes Terminated by the Fluorine and Hydroxyl Groups Based on Their Unit Cells

M_2CT_2	T = F	T = OH
M = Ce	-14.8	-18.4
M = Pr	-16.1	-19.7
M = Nd	-18.1	-21.7
M = Sm	-24.6	-28.1
M = Eu	-22.2	-25.8
M = Gd	-31.7	-35.4
M = Tb	-27.6	-31.4
M = Dy	-24.1	-27.8
M = Ho	-20.5	-24.9
M = Er	-19.1	-22.8
M = Tm	-13.4	-20.4
M = Yb	-12.3	-16.1

the table, the formation energies of all the fluorine- and hydroxyl-terminated MXenes are observed to be negative. The negative values with the largest and smallest magnitudes are determined in $Gd_2C(OH)_2$ and Yb_2CF_2 , which are -35.4 and -12.3 eV, respectively. These values with large magnitudes imply that the lanthanide-based MXenes are prone to be terminated by the fluorine and hydroxyl groups in hydrofluoric acid.^{21,26} The formation energy of the hydroxyl-terminated MXene generally shows a larger magnitude than that of the fluorine-functionalized structure, which implicates that the fluorine group could transform into the hydroxyl one upon washing and/or storing in water.²⁴ Moreover, the magnitude of the formation energy of M_2CT_2 (T = F and OH) generally

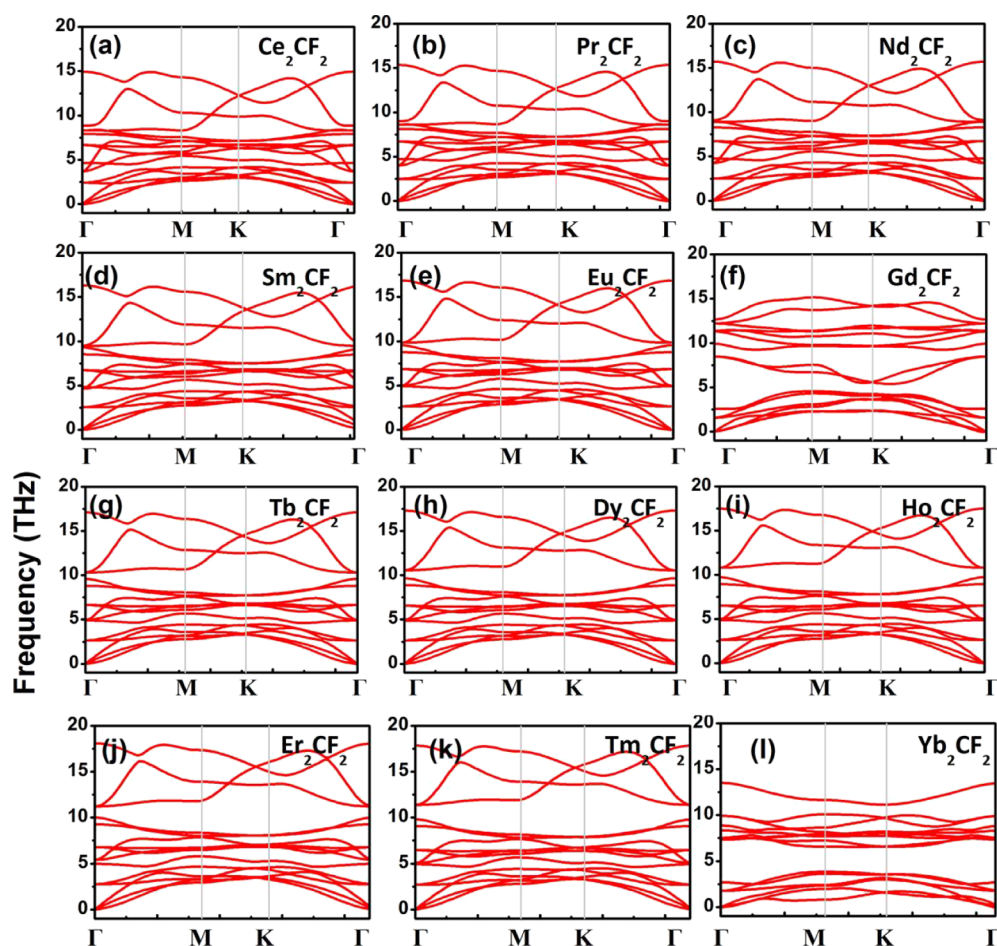


Figure 4. (a–l) Phonon dispersions of M_2CF_2 MXenes ($M = \text{Ce, Pr, Nd, Sm, Eu, Gd, Tb, Dy, Ho, Er, Tm, and Yb}$).

increases, with its M from Ce to Gd, and then decreases from Gd to Yb. This behavior could be related to the occupation fraction of the f orbital of the lanthanide atom. Gd with half-full f electrons shows a stronger chemical bond with its surface group.

Further, the dynamic stabilities of these functionalized configurations are studied. The phonon dispersions of M_2CF_2 and $M_2C(OH)_2$ ($M = \text{Ce, Pr, Nd, Sm, Eu, Gd, Tb, Dy, Ho, Er, Tm, and Yb}$) are provided in Figures 4 and 5, respectively. All of the phonon dispersions except for that of $\text{Ce}_2\text{C(OH)}_2$ are positive, implying that these configurations are dynamically stable. The hydroxyl-terminated structures generally show much higher frequencies than the corresponding fluorine-terminated members, which could be ascribed to the relatively light mass of the hydrogen element. Upon the same functional group, the highest phonon frequency also varies in different lanthanide-based MXenes. This is a reflection of different bond strengths in lanthanide-based MXenes as a stronger bond strength generally introduces a higher phonon frequency.⁵⁸ Based on the discussions above, most of the lanthanide-based MXenes investigated are thermodynamically and dynamically stable, which could be realized in further experiments.

To evaluate the magnetic stability of all the fluorine- and hydroxyl-terminated MXenes, nonmagnetic (NM), ferromagnetic (FM), and antiferromagnetic (AFM) states for the fluorine- and hydroxyl-terminated MXenes are examined.^{68–72} The structural diagrams of FM and AFM configurations are shown in Figure 6. The corresponding relative total energies

for these NM, FM, and AFM configurations are presented in Table S5. Apparently, most FM configurations show the lowest total energies for these M_2CF_2 and $M_2C(OH)_2$ structures. Only Ho_2CF_2 and $\text{Dy}_2\text{C(OH)}_2$ show the lowest total energies in their AFM states. The energy difference between NM and FM is generally larger than that between FM and AFM. Eu_2CF_2 shows the largest energy difference between its FM and AFM states, which implies that this configuration might possess a high Curie temperature.

After checking the structural stabilities, the structural parameters are further investigated. The lattice parameters, layer thicknesses, and bond lengths for the fluorine-terminated MXenes are provided in Table 4, and the corresponding values for the hydroxyl-functionalized members are presented in Table 5. Apparently, the lattice parameters of both fluorine- and hydroxyl-terminated MXenes investigated here are generally larger than those of MXenes reported previously,⁴² which is due to the larger atomic radii of lanthanides than those of early transition metals. Moreover, the lattice parameters and bond lengths generally decrease with the increasing atomic number of lanthanides, which could be ascribed to the decreasing atomic radius of lanthanides.⁷³ In particular, the Eu-containing MXenes are against this trend. Taking Eu_2CF_2 as an example, its lattice parameter of 3.74 Å is larger than that of 3.71 Å of its former Sm_2CF_2 . The Eu–C bond is much larger than the other M–C bonds. Similar behaviors are also determined in the $\text{Eu}_2\text{C(OH)}_2$ MXene. These special structural parameters for Eu-containing MXenes

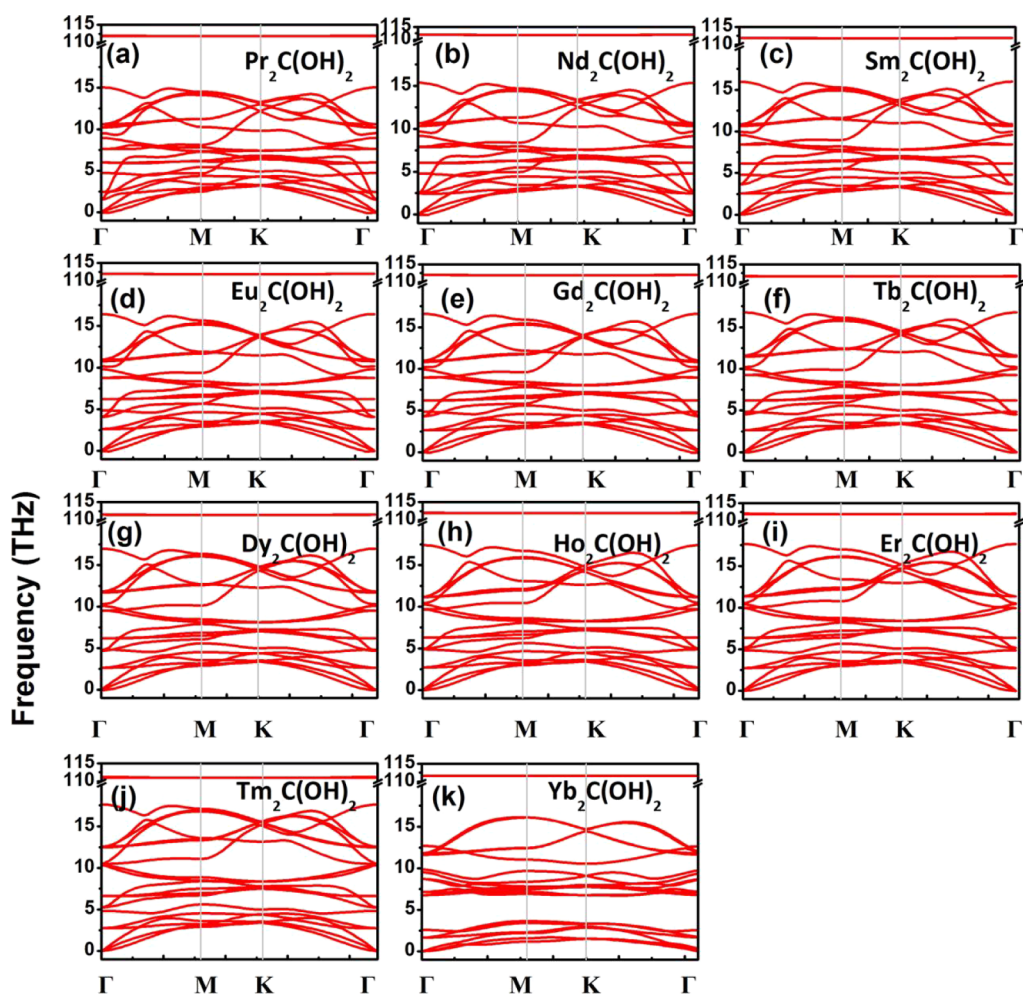


Figure 5. (a–k) Phonon dispersions of $M_2C(OH)_2$ MXenes ($M = \text{Pr, Nd, Sm, Eu, Gd, Tb, Dy, Ho, Er, Tm, and Yb}$).

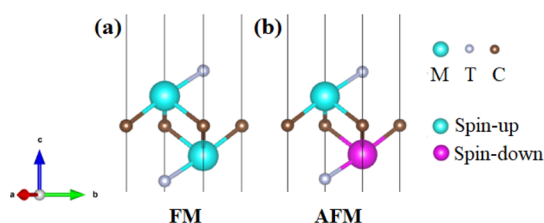


Figure 6. (a,b) Side views of the FM and AFM configurations for M_2CT_2 ($T = \text{F and OH}$) MXenes, respectively. Pink and green balls represent the M atoms with spin-up and spin-down electron configurations, respectively. Brown and small gray balls represent the carbon atom and the surface group T, respectively.

could be caused by the special valence electron configuration of $4f^7 5d^0 6s^2$ in Eu. For the other lanthanide elements, except for Yb, the valence electrons are composed by $5d^1 6s^2$ and varying numbers of 4f electrons. Generally, these predicted structural parameters will give some guidance for future synthesis and characterization of these lanthanide-based MXenes.

As mentioned previously, MXenes with their M elements in group IIIB generally show promising semiconducting properties;^{42,43,50–54} thus, the electronic properties of these lanthanide-based MXenes are further investigated. Based on the GGA-PBE functional, all the electronic structures for the fluorine- and hydroxyl-functionalized MXenes are provided in

Table 4. Lattice Parameter a , Layer Thickness h , and Bond Lengths M–C and M–F (in Å) for M_2CF_2 ($M = \text{Ce, Pr, Nd, Sm, Eu, Gd, Tb, Dy, Ho, Er, Tm, and Yb}$) MXenes

MXenes	a	h	M–C	M–F
Ce_2CF_2	3.69	6.82	2.57	2.46
Pr_2CF_2	3.68	6.77	2.56	2.43
Nd_2CF_2	3.69	6.76	2.58	2.43
Sm_2CF_2	3.71	6.71	2.60	2.41
Eu_2CF_2	3.74	6.73	2.63	2.41
Gd_2CF_2	3.65	6.65	2.53	2.41
Tb_2CF_2	3.59	6.60	2.49	2.39
Dy_2CF_2	3.56	6.58	2.48	2.37
Ho_2CF_2	3.56	6.56	2.48	2.36
Er_2CF_2	3.54	6.53	2.48	2.35
Tm_2CF_2	3.52	6.51	2.47	2.33
Yb_2CF_2	3.54	6.58	2.51	2.33

Figures 7 and 8, respectively. For the fluorine-terminated configurations, Ho_2CF_2 is stabilized in its AFM states, as discussed previously, and its spin-up and spin-down states are degenerate in electronic structures. M_2CF_2 ($M = \text{Ce, Tm, and Yb}$) are FM metals, and both the spin-up and spin-down states intersect with their Fermi levels. For the hydroxyl-terminated MXenes, $\text{Dy}_2C(OH)_2$ is an AFM metal, while $M_2C(OH)_2$ ($M = \text{Tb, Er, Tm, and Yb}$) are FM metals. Gd_2CT_2 ($T = \text{F and OH}$) are the only two FM semiconducting members

Table 5. Lattice Parameter a , Layer Thickness h , and Bond Lengths M–C and M–O (in Å) for $M_2C(OH)_2$ ($M = Pr, Nd, Sm, Eu, Gd, Tb, Dy, Ho, Er, Tm,$ and Yb) MXenes

MXenes	a	h	M–C	M–O
$Pr_2C(OH)_2$	3.69	8.60	2.57	2.51
$Nd_2C(OH)_2$	3.71	8.59	2.58	2.50
$Sm_2C(OH)_2$	3.72	8.58	2.60	2.49
$Eu_2C(OH)_2$	3.76	8.60	2.63	2.50
$Gd_2C(OH)_2$	3.65	8.50	2.53	2.48
$Tb_2C(OH)_2$	3.59	8.42	2.49	2.45
$Dy_2C(OH)_2$	3.59	8.39	2.49	2.44
$Ho_2C(OH)_2$	3.56	8.37	2.48	2.43
$Er_2C(OH)_2$	3.55	8.35	2.47	2.41
$Tm_2C(OH)_2$	3.53	8.33	2.47	2.39
$Yb_2C(OH)_2$	3.54	8.42	2.51	2.39

determined here. All the rest of the configurations are half-metals. For these half-metals, it is worth pointing out that the metal state of the half-metal is spin-up when the atomic number of M in M_2CT_2 is smaller than that of Gd . On the contrary, the metal state of half-metal is spin-down when the atomic number of M is larger than that of Gd . This behavior might be related to the occupation fractions of f orbital in these lanthanide atoms. For example, the band gap of the spin-down states in Eu_2CF_2 is as large as 2.39 eV. This high spin-polarization in these half-metals enables these configurations in spintronic devices, especially for pure spin generation and injection.⁷⁴

Based on the GGA-PBE functional, the band gaps in Gd_2CF_2 and $Gd_2C(OH)_2$ are 0.732 and 0.400 eV, respectively. Apparently, the band gap of $Gd_2C(OH)_2$ is smaller than that of the fluorine-terminated configuration. The band gap in Gd_2CF_2 is indirect, with the valence band maximum (VBM) in its electronic energy band located at the BZ center, and the corresponding conduction band minimums (CBMs) lie on the high-symmetry M point. $Gd_2C(OH)_2$ is a direct band gap semiconductor, with both its VBM and CBM located at the BZ center. As the GGA-PBE functional generally underestimates the band gaps for semiconductors, the more precise HSE06 functional is also adopted. The electronic energy bands of Gd_2CT_2 ($T = F$ and OH) from HSE06 are provided in Figure 9. Both the band gaps of Gd_2CF_2 and $Gd_2C(OH)_2$ increase significantly after the HSE06 correction. For Gd_2CF_2 , its band gap increases from the GGA of 0.732 to 1.381 eV. This value is a little higher than the band gap of bulk silicon, implying that Gd_2CF_2 is suitable for semiconductor devices. For $Gd_2C(OH)_2$, its band gap increases to 0.882 eV after correction. On the basis of the direct and appropriate band gap, $Gd_2C(OH)_2$ could have promising applications in infrared adsorption devices.⁷⁵ Moreover, these two structures are intrinsic 2D magnetic semiconductors, which are rarely reported in previous works. The properties of magnetic semiconductors indicate that Gd_2CT_2 ($T = F$ and OH) have potential applications in the channel materials of spin devices⁷⁴ and quantum computing.⁷⁶

In order to understand the origin of magnetism in these systems, the magnetic moments of the fluorine- and hydroxyl-terminated MXenes are calculated and listed in Tables 6 and 7,

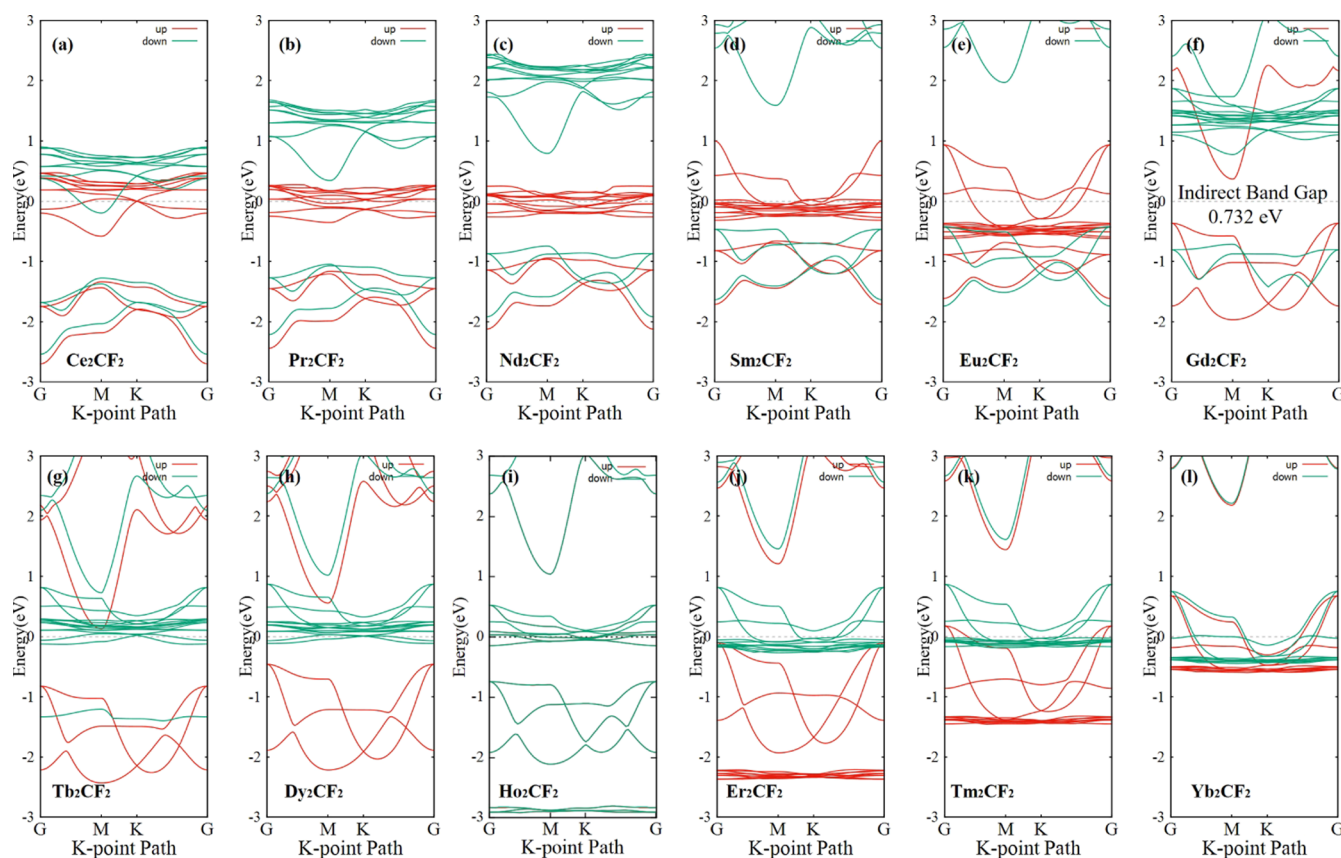


Figure 7. (a–l) Electronic energy bands of M_2CF_2 ($M = Ce, Pr, Nd, Sm, Eu, Gd, Tb, Dy, Ho, Er, Tm,$ and Yb) MXenes based on the GGA-PBE functional. The Fermi level is set as 0 eV.

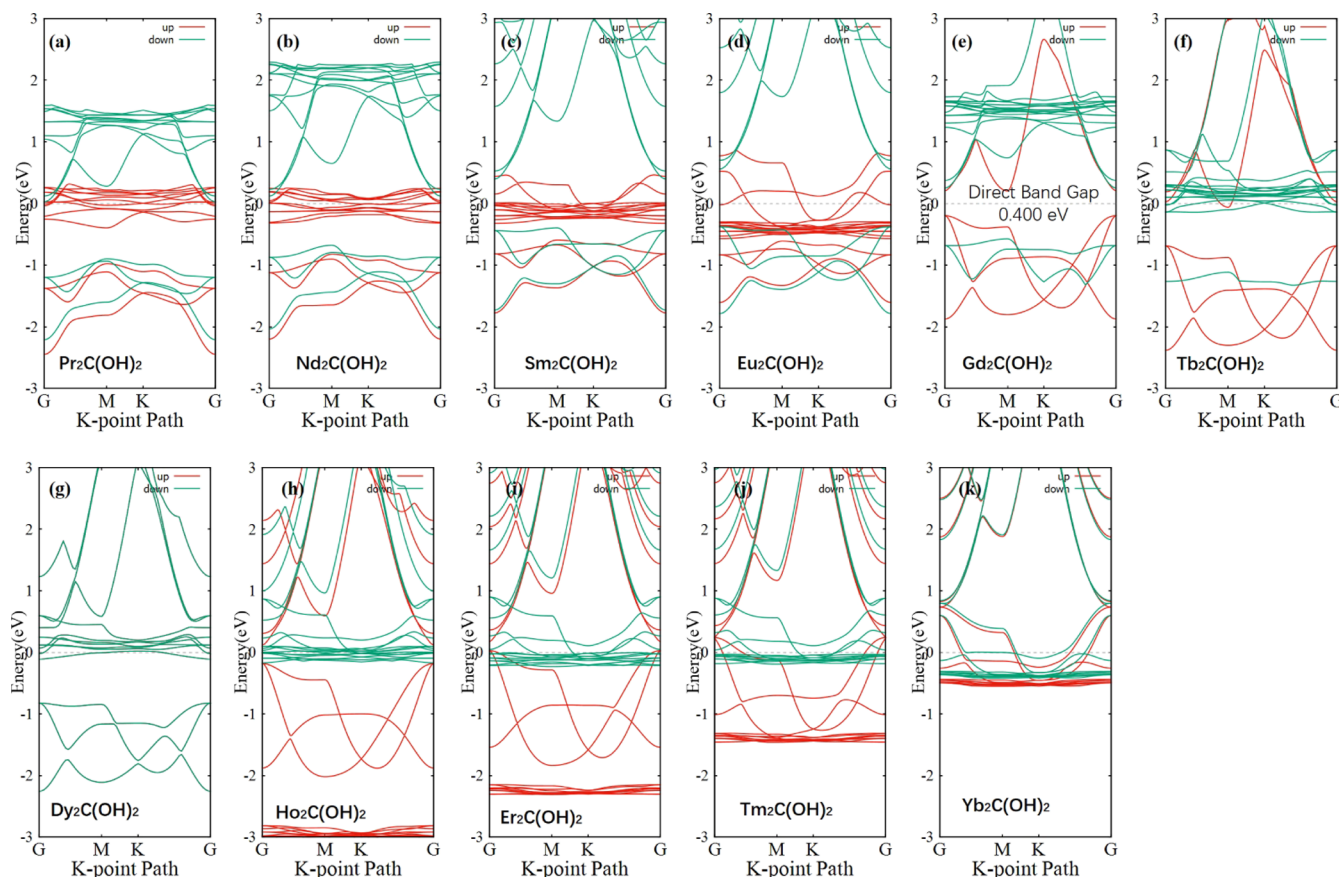


Figure 8. (a–k) Electronic energy bands of $M_2C(OH)_2$ ($M = \text{Pr, Nd, Sm, Eu, Gd, Tb, Dy, Ho, Er, Tm, and Yb}$) MXenes based on the GGA-PBE functional. The Fermi level is set as 0 eV.

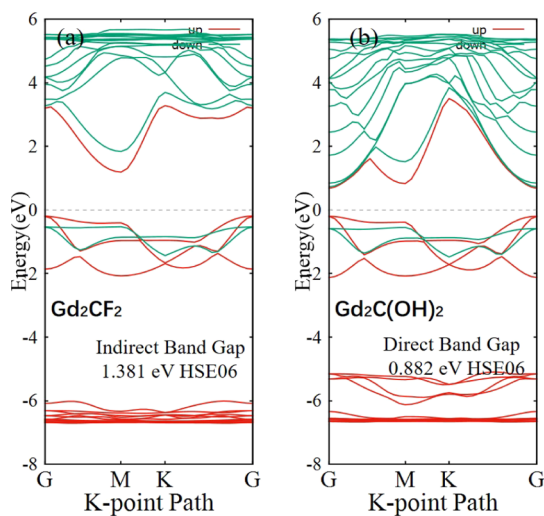


Figure 9. (a,b) Electronic energy bands of Gd_2CF_2 and $Gd_2C(OH)_2$ after the HSE06 correction. The Fermi energy is set as 0 eV.

respectively. The projected density of states (PDOS) of M in these lanthanide-based M_2CF_2 and $M_2C(OH)_2$ MXenes are calculated and show in Figures S2 and S3, respectively. Apparently, the magnetism is correlated with the occupation fraction of f orbitals in these M atoms, and their spin-up and spin-down states are separated in those FM configurations. Based on the unit cells, the magnetic moments vary significantly among these lanthanide-based MXenes. $Yb_2C(OH)_2$ shows the lowest magnetic moment of $0.390 \mu_B$. The

Table 6. Magnetic Moments (in μ_B) of M_2CF_2 ($M = \text{Ce, Pr, Nd, Sm, Eu, Gd, Tb, Dy, Ho, Er, Tm, and Yb}$) MXenes

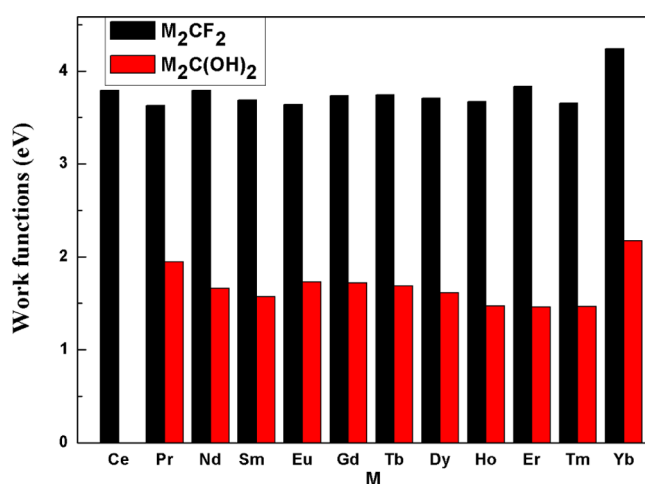
MXenes	total magnetization	$M(f\text{-orbital})$
Ce_2CF_2	1.33	0.628
Pr_2CF_2	3.75	1.88
Nd_2CF_2	5.87	3.05
Sm_2CF_2	10.1	5.31
Eu_2CF_2	12.3	6.43
Gd_2CF_2	13.8	6.84
Tb_2CF_2	11.8	5.83
Dy_2CF_2	9.77	4.77
Ho_2CF_2	0.00	± 3.69
Er_2CF_2	5.72	2.67
Tm_2CF_2	3.53	1.60
Yb_2CF_2	0.458	0.190

largest magnetic moment is determined in Gd_2CF_2 , the value of which is as large as $13.8 \mu_B$. The total magnetisms in M_2CF_2 ($M = \text{Dy and Ho}$) and $M_2C(OH)_2$ is zero because their stable magnetic configurations are all in the AFM states. Apparently, the magnetism is correlated with the occupation fraction of f orbitals in these lanthanide atoms. Gd_2CT_2 ($T = \text{F and OH}$) generally show large magnetic moments because the f orbital in Gd is half-full. This behavior could be directly seen from the magnetic contributions of f orbitals listed in Tables 6 and 7. The f orbital of each Gd atom contributes $6.84 \mu_B$ in both Gd_2CT_2 ($T = \text{F and OH}$). These high magnetic moments could enable these lanthanide-based MXenes' applications in information storage, especially in 2D electronic devices.

Table 7. Magnetic Moments (in μ_B) of $M_2C(OH)_2$ ($M = Pr, Nd, Sm, Eu, Gd, Tb, Dy, Ho, Er, Tm,$ and Yb) MXenes

MXenes	total magnetization	M(f-orbital)
$Pr_2C(OH)_2$	3.68	1.85
$Nd_2C(OH)_2$	5.82	3.02
$Sm_2C(OH)_2$	10.1	5.29
$Eu_2C(OH)_2$	12.3	6.42
$Gd_2C(OH)_2$	13.7	6.84
$Tb_2C(OH)_2$	11.8	5.86
$Dy_2C(OH)_2$	0.00	± 4.79
$Ho_2C(OH)_2$	7.74	3.75
$Er_2C(OH)_2$	5.69	2.69
$Tm_2C(OH)_2$	3.48	1.60
$Yb_2C(OH)_2$	0.390	0.161

As the hydroxyl-terminated MXenes have been demonstrated to show low work functions previously,⁵³ the work functions of these lanthanide-based MXenes M_2CT_2 ($M = Ce, Pr, Nd, Sm, Eu, Gd, Tb, Dy, Ho, Er, Tm,$ and Yb ; $T = F$ and OH) are also investigated here. As shown in Figure 10, the

**Figure 10.** Work functions of M_2CT_2 MXenes ($M = Ce, Pr, Nd, Sm, Eu, Gd, Tb, Dy, Ho, Er, Tm,$ and Yb ; $T = F$ and OH).

hydroxyl-terminated MXenes generally show much lower work functions than the fluorine-terminated ones. The work functions of the fluorine-functionalized structures are in the range from 3.62 to 4.24 eV, with the minimum and maximum values determined in Pr_2CF_2 and Yb_2CF_2 , respectively. With respect to the hydroxyl-terminated configurations, the corresponding values are in the range from 1.46 to 2.17 eV. These values are generally lower than 2.1 eV of the typical low work function material cesium.⁷⁷ The lowest work function is determined in $Tm_2C(OH)_2$. Noteworthily, the work functions in several hydroxyl-terminated configurations are even lower than that in $Sc_2C(OH)_2$, which was found to be the lowest work function material reported previously.⁵³ The low work function in these hydroxyl-terminated MXenes could be ascribed to the high potential of surface hydrogen atoms.⁷⁸ Based on these low work functions, these lanthanide-based MXenes could have potential applications in field emitter cathodes.

CONCLUSIONS

In summary, the stabilities, structural parameters, and electronic properties of lanthanide-based MXenes have been investigated in this work. According to the formation energies and phonon dispersions calculated, most fluorine- and hydroxyl-terminated MXenes are stable. The fluorine- and hydroxyl-terminated MXenes are generally magnetic, and most of them are half-metals. High spin polarization has been determined in these half-metals. The spin-up states in the half-metal Eu_2CF_2 show a large band gap of 2.39 eV. Gd_2CT_2 ($T = F$ and OH) are the only two magnetic semiconductors. From HSE06, it is observed that Gd_2CF_2 is an indirect semiconductor with a band gap of 1.38 eV, while $Gd_2C(OH)_2$ shows a direct band gap of 0.882 eV. The magnetisms in these lanthanide-based MXenes are mainly related to the occupation fractions of 4f orbitals in the lanthanide atoms. Based on the half-full 4f orbital of Gd, both the unit cells of Gd_2CT_2 ($T = F$ and OH) show magnetic moments higher than $13.7 \mu_B$. Moreover, the hydroxyl-terminated structures generally show relatively low work functions. The work function in $Tm_2C(OH)_2$ is as low as 1.46 eV, which could be the lowest work function to the best of our knowledge. Based on these predicted magnetic half-metallic, and semiconducting properties, with low work functions, these lanthanide-based MXenes could have widespread potential applications such as in spintronics, quantum computation, near-infrared detectors, field effect transistors, information storage, and field emitter cathodes. We look forward to the synthesis of these lanthanide-based MXenes in future experiments.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acsomega.2c03964>.

Relative total energies between the T- and H-type M_2C MXenes; lattice parameters for the T- and H-type M_2C MXenes; potential competing phases and the corresponding decomposition energies for the T-type M_2C MXenes; electronic energy bands for the T-type M_2C MXenes based on the GGA-PBE functional and the corresponding magnetic moments and relative total energies between the NM, FM, and AFM states of the M_2CT_2 MXenes; and PDOS of M in M_2CF_2 and $M_2C(OH)_2$ MXenes (PDF)

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Notes

The authors declare no competing financial interest.

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