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A General Method for Preparing Anatase TiO₂ Treelike-Nanoarrays on Various Metal Wires for Fiber Dye-Sensitized Solar Cells

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Anatase TiO_2 tree-like nanoarrays were prepared on various metal wires (Ti, W, Ni, etc.) through one-step facile hydrothermal reaction. The anatase TiO_2 tree-like nanoarrays consist of long TiO_2 nanowire trunks with direct charge transport channels, and a large number of short TiO_2 nanorod branches with large surface areas. Fiber dye-sensitized solar cells (FDSSCs) based on the anatase TiO_2 tree-like nanoarrays deposited on Ti wires can achieve outstanding power conversion efficiency (PCE) of 6.32%, while FDSSCs on W wires have lower PCE of 3.24% due to the formation of WO_3 layer, which might enhance recombination of charges. When the substrate is changed to a Nicole oxide wire, a novel p-n heterojunction can be obtained. This universal method is simple, facile, and low cost for preparing anatase TiO_2 treelike-nanoarrays on various metal wires, which may find potential applications in fabrication of optoelectronic devices.

T itanium dioxide (TiO₂), one of the most important wide band-gap semiconductors and metal oxides, has attracted great attention due to its excellent optical and electrical properties, high chemical and optical stability, low toxicity and cost, and appropriate electronic band structure¹⁻⁴. TiO₂ plays critical roles in many fields, such as dye-sensitized solar cells (DSSCs)⁵, lithium ion batteries⁶, gas sensors⁷, and photocatalysts⁸, *etc.* The performance of TiO₂-based devices is found to be strongly dependent on TiO₂ morphology, crystalline phase and structure⁹⁻¹⁵. The anatase TiO₂ treelike-nanoarrays are of particular interest for large surface areas, suitable porosity, and their potential applications as continuous pathway for charge transport. To date, considerable efforts have been devoted to fabricating anatase TiO₂ treelike-nanoarrays¹⁶⁻¹⁸. Normally, the first step of growing anatase TiO₂ treelike-nanoarrays is titanium hydrolysis reaction in alkaline solution, which results in one-dimensional (1D) titanate on Ti substrates; the second step is ion-exchange process in acidic solution to obtain 1D titanic acid; the final step is growing anatase TiO₂ treelike-nanoarrays. However, the procedures are too complicated, and the substrates must be Ti related materials to offer titanium source. Therefore, the controlled growth of anatase TiO₂ treelike-nanoarrays on various substrates through a relatively simple method is highly desirable for possible optoelectronic applications with various requirements^{19,20}.

As a cheap, environment-friendly and simple fabrication process, DSSCs have huge potentials for replacing silicon-based solar cells²¹. Nowadays, the PCE of the traditional plane DSSCs is up to 12.3%²². However, the substrate is rigid fluorine-doped tin oxide (FTO) conducting glass, which limits transportation and applications of flat DSSCs. Therefore, great attention has been paid to exploring substitutional substrates such as metal, plastic conducting and carbonous substrates. The newly developed FDSSCs can be synthesized based on any flexible conducting wire substrates, such as metal wire²³⁻²⁶, carbon fiber²⁷⁻²⁹ and optical fiber³⁰, which significantly extend the source of possible substrates. Meanwhile, such structure may find applications in solving electrolyte leakage problems, uncomplicated encapsulation and modularization, omnidirectional light harvesting, weaving materials for clothing, and external bond layer of buildings.

 TiO_2 1D nanoarrays, with continuous pathway for charge transport, have been exploited as excellent photoanodes for FDSSCs. Up to now, TiO_2 nanotubes anodised on Ti wires as photoanodes have obtained the highest PCE of nearly 7% after a high concentration of $TiCl_4$ solution processing to achieve sufficient surface areas³¹. However, only one Ti wire is involved in the reaction for TiO_2 nanotube, which is a waste of time and money.

Herein, we report anatase TiO2 treelike-nanoarrays consisting of long TiO₂ nanowire trunks and short nanorod branches on various metal wires (Ti, W, Ni. etc.) for possible applications in FDSSCs. The FDSSCs based on Ti wires show outstanding performance of 6.32% PCE. The vertically aligned long TiO₂ nanowire trunks enhance the charge transport, and the short nanowire branches provide large surface areas for dye loading. Moreover, anatase TiO₂ treelikenanoarrays can be simultaneously synthesized on several metal wires (Ti, W, Ni, etc.) through one-step hydrothermal process in one container. Thus, the hydrothermal method in this study provides large possibilities of growing TiO₂ on various metal wires as templates according to the requirements of the TiO₂ related devices. This hydrothermal method is the same as growing ZnO nanoarrays on different substrates³²⁻³⁴. This method is simple, facile, low cost, and suitable for large-scale production, which shows great potentials for future applications.

Results

Structure of anatase TiO₂ treelike-nanoarrays. Figure 1(a) shows a bare Ti wire of $\sim 500~\mu m$ in diameter. The seeded-Ti wires were immersed into a solution containing 1.24 g potassium titanium oxide oxalate dihydrate, 50 mL diethylene glycol and 20 mL water at 180°C for 12 h (see Method Section). The dimension of the wires increases slightly and the surface becomes rough, as shown in Figure 1(b). The morphologies of Ti wire before and after depositing TiO₂ are shown in Figure 1S (Supplementary Information), and the later is covered with a white layer of TiO_2 film. The phase purity and structure of the hydrothermal sample reacted for 12 h was characterized using X-ray diffraction (XRD). Besides the sharp Ti peaks from the substrates of Ti wires, which correspond to (101) and (200) planes of the anatase TiO₂ structure, there are obviously several broadened diffraction peaks, as shown in Figure 1(c). The results suggest that the as-products are anatase TiO₂ (PCPDFWIN file No. 84-1286). According to the "iii" sample in Figure 1(c), the peaks of anatase TiO_2 are much sharper after annealing treatment, which indicates better crystallinity.

Figure 2 shows typical scanning electron microscopy (SEM) images of the TiO_2 nanoarrays with different reaction time (3, 6, 9, 12, 15 h). The reaction temperature is $180^{\circ}C$ for each sample deposition. Figures 2 (a) and (b) show SEM images of TiO_2 synthesized for

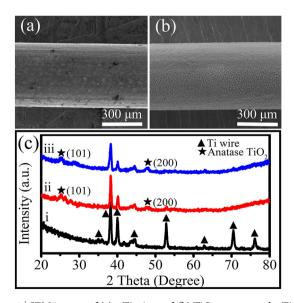


Figure 1 | SEM images of (a) a Ti wire and (b) TiO_2 grown on the Ti wire for 12 h, (c) XRD patterns of Ti plate substrate (i) by using the same growth procedure, TiO_2 depositing on Ti wire for 12 h before (ii) and after annealing (iii).

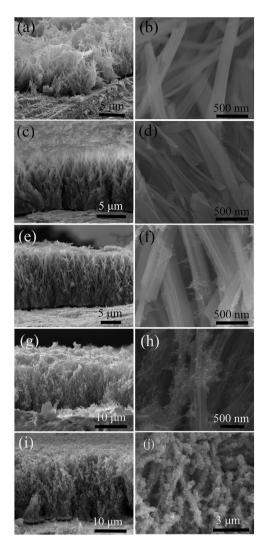


Figure 2 | Cross-sectional SEM images of morphological evolution of the TiO_2 nanoarrays synthetized at different time: (a) and (b) 3 h; (c) and (d) 6 h; (e) and (f) 9 h; (g) and (h) 12 h; (i) and (j) 15 h.

3 h on the Ti wire, with length of about 8 μ m. When the reaction time is prolonged to 6 h, there are fractional branches on the trunks of TiO₂ nanowire arrays, as shown in Figures 2 (c) and (d). After 9 h of reaction time (Figures 2 (e) and (f)), obvious branches can be observed, and the length of the TiO₂ nanoarrays increases to about 14 μ m. Figures 2 (g) and (h) show the sample morphologies reacted for 12 h. There are a large number of TiO₂ nanorod branches on the TiO_2 nanowire stems, and the length is about 18.5 μ m. Figures 2 (i) and (j) show SEM images of TiO₂ treelike-nanoarrays deposited on Ti substrate for 15 h. The branches become denser and longer, and the length of TiO_2 nanowire increases to 22 µm. Therefore, TiO_2 nanowire stems grow longer with increasing reaction time, and short nanorod branches start to germinate on the surface of TiO₂ stems, and become denser and longer. In this way, the TiO₂ tree-like nanoarrays consisting of long $\rm TiO_2$ nanowire stems and short $\rm TiO_2$ nanorod branches on various wires are successfully obtained with one-step facile hydrothermal reaction.

Transmission electron microscopy (TEM) technique was further used to characterize the morphologies and structures of the TiO_2 samples, as shown in Figure 3. Figures 3 (a) and (b) are TEM and high-resolution TEM (HRTEM) images of a TiO_2 nanowire sample synthesized for 3 h. The interplanar distances of 0.189 and 0.351 nm in the HRTEM image correspond to (200) and (101) planes of the anatase crystal phase of TiO_2 , respectively (PCPDFWIN file

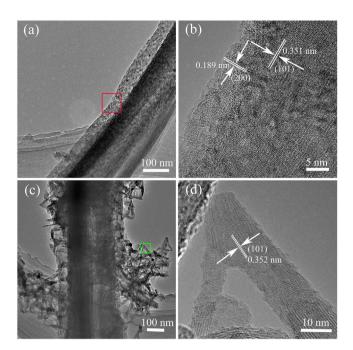


Figure 3 | Transmission Electron Microscopic (TEM) and high resolution transmission electron microscopic images (HRTEM) of the TiO_2 samples deposited for 3 h (a) and (b), 12 h (c) and (d). The HRTEM images (b) and (d) correspond to the color-marked regions in the TEM images (a) and (c), respectively.

No. 84-1286). The edge of the nanowire is rough, which is beneficial for the regrowth of branches. Figure 3 (c) shows TEM image of a TiO_2 nanotree synthesitized for 12 h, in which a large number of branches come out on the stem. The corresponding HRTEM image (Figure 3 (d)) shows interplanar spacing of 0.352 nm, which matches well with the (101) planes of anatase TiO_2 .

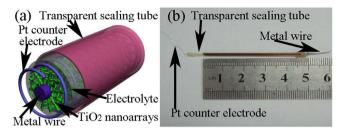


Figure 4 | The FDSSC Schematic (a) and its Optical image (b).

Fabrication of FDSSCs. The TiO₂ tree-like nanoarrays on metal wires are considered as preferential photoanodes for FDSSCs owing to their superior electron transport capabilities and comparatively large surface areas, and the performance of DSSC based on anatase TiO₂ is better than that of the rutile one¹¹. The process of fabricating FDSSCs is seen in Method Section. Figures 4 (a) and (b) show the schematic setup and optical image of FDSSC, respectively. The counter electrode is twisted around the dyesensitized photoanode, and then immersed in a transparent tube filled with electrolyte. The FDSSCs is encapsulated in the tube, the end of which is connected to the electrode^{23–29}. The melted plastic was used to seal the both ends of the tube to finish the fabrication of FDSSCs.

Photovoltaic performance characterization. Figure 5 (a) indicates the current density-voltage (I–V) curves of the FDSSCs fabricated with photoanodes of TiO₂ nanoarrays prepared with different reaction time (3, 6, 9, 12, 15 h) on Ti wires. Table 1 summarized the corresponding detailed photovoltaic properties, including the open-circuit voltage (V_{oc}), short-circuit current density (J_{sc}), fill factor (FF), and PEC. The PCE increases gradually from 2.58% to 6.32% when the hydrothermal reaction time increases from 3 to 12 h, and then decreases slightly to 5.50% as the reaction time prolonged to 15 h. The J_{sc} variation has the same trend as PCE, which increases significantly from 5.42 to 13.95 mA/cm², and then decreases slightly

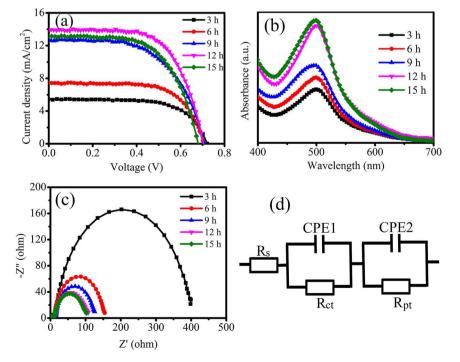


Figure 5 | Characteristics of the FDSSCs with TiO₂ photoanodes based on Ti wires prepared for different times. (a) I–V curves. (b) UV-Vis spectra of solutions containing N719 desorbed from sensitized TiO₂ nanoarrays. (c) EIS measurements were conducted in the dark under a bias of 0.7 V. (d) an equivalent circuit for fiting Nyquist plots.

to 13.15 mA/cm². Thus, it can be concluded that the differences of the resulting PEC could be mainly attributed to the diverse J_{sc} . In general, J_{sc} can be approximated by the following expression^{35,36}:

$$J_{sc} = q\eta_{1h}\eta_{inj}\eta_{cc}I_0$$

where *q* is the elementary charge, η_{lh} is the light-harvesting efficiency of a cell, η_{inj} is the charge-injection efficiency, η_{cc} is the chargecollection efficiency, and I_0 is the light flux. Among all of these parameters, η_{1h} is mainly related to the amount of adsorbed dye, η_{cc} is largely dependent on the competition between recombination and charge collection, and η_{inj} is suggested to be of the same value for all the photoelectrodes based on the TiO₂ nanoarrays and the N719 dye.

The length of nanowire stem and the number of the nanorod branches both increase with prolonged reaction time, which could offer larger surfaces to absorb more dye molecules. Figure 5 (b) shows UV-Vis spectra of solutions containing N719 dye molecules desorbed from sensitized TiO₂ nanoarrays synthesized with different reaction time. The corresponding absorbed dye amounts in Table 1 were calculated using Lambert-Beer's law. As the reaction time of TiO₂ photoanodes increasing from 3 to 15 h, the amounts of absorbed dye increases from 27.28 to 62.45 nmol/cm², which results in gradually increasing η_{1h} . Therefore, the J_{sc} and PCE of the FDSSCs improve with increasing deposition time of TiO₂, which could be resulted from the dye adsorption of TiO₂ working eletrodes.

However, excessively long TiO₂ nanoarrays may lead to decreased J_{sc} and PCE, which is the case of FDSSCs with the TiO₂ photoelectrodes synthesized for 15 h. The corresponding charge-transfer resistance reduces, resulting in a larger recombination rate of photo-induced electrons and holes at the TiO2/dye/electrolyte interfaces. To further elucidate the charge recombination process, electrochemical impedance spectroscopic (EIS) measurements were conducted in the dark under a bias of 0.70 V37. Figures 5 (c) and (d) are the Nyquist plots and the corresponding simplified equivalent circuit for the FDSSCs, respectively. R_s is the series resistance related to the transport resistance of the metal wire substrates. R_{ct} is the charge-transfer resistance accounting for the recombination of photo-induced electrons and holes at the TiO2/dye/electrolyte interfaces, which corresponds to the larger semicircle in the low-frequency region. R_{pt} is the charge-transfer resistance at the counter electrode/electrolyte interface corresponding to the smaller semicircle in the high-frequency region³⁸. The fitting values of R_{ct} of the TiO₂ photoelectrode synthetized for 3, 6, 9, 12 and 15 h are 356.1, 134.9, 103.0, 85.99 and 77.39 Ω , respectively, as listed in Table 1. Nevertheless, the values of R_s and R_{Pt} are nearly the same since the Ti wire substrate and the counter electrode/electrolyte interface are the same, as shown in Figure 5 (c). Longer TiO_2 nanarrays have smaller R_{ct} values, implying rapid recombination rates of photoinduced electrons and holes at the $\mathrm{TiO}_2/\mathrm{dye}/\mathrm{electrolyte}$ interfaces, which may lead to decreased J_{sc} and PCE of the FDSSC. Both of the R_{ct} and absorbed dye amounts affect the J_{sc} and PCE of the FDSSC, while both parameters correlate to the length of the TiO₂ nanowire arrays and the number of nanorod branches. The TiO2 tree-like

Table 1 | Photovoltaic parameters of the FDDSCs with working electrodes of TiO_2 nanoarrays depositing on Ti wires for different times

Samples	J _{sc} (mA/cm²)	V _{oc} (V)	FF	PCE(%)	Adsorbed dye (nmol/cm²)	e R _{ct} (Ω)
3 h	5.42	0.722	0.659	2.58	27.28	356.1
6 h	7.42	0.705	0.671	3.51	33.35	134.9
9 h	12.75	0.706	0.578	5.20	39.71	103.0
12 h	13.95	0.705	0.641	6.32	59.72	85.99
15 h	13.15	0.681	0.614	5.50	62.45	77.39

nanoarrays synthetized for 12 h possess suitable absorbed dye amounts and recombination rate of generated electrons and holes, and achieve the highest J_{sc} , which ensures excellent photovoltaic performance.

The TiO₂ treelike-nanoarrays deposited on W and Ni oxide wires were shown in Figure 2S (Supplementary Information). Figure 6 (a) shows the I-V curve of FDSSCs reacted for 12 h based on a W wire. The corresponding PCE (3.26%) and J_{sc} (9.91 mA/cm²) are both lower than that based on Ti wire. Annealing of Ti wire leads to the formation of TiO₂ anatase phase which has the same band structure as the TiO₂ working electrode. Annealing of the W wire, on the other hand, introduces a WO₃ layer between the W substrate and the TiO₂ photoelectrode. The conduction band level of WO₃ is lower than that of TiO2³⁹. Thus, electron transfer from the TiO2 layer to the WO3 layer is associated with the dissipation of less amount of energy^{40,41}. To analyse the photovoltaic parameters, EIS measurements were conducted in the dark under a bias of 0.70 V. The Ti and W based FDSSCs showed no noticeable resistance in the first semicircle of EIS data (see Fig. 6 (b)). However, the W based FDSSCs showed a drastic decrease in the internal resistance ($R_{ct} = 45.71 \Omega$), which means a larger recombination rate in photoelectrode, and which might be due to the high-level conduction band mismatch between the TiO₂ working electrode and the WO3 layer, as shown in Figure 6 (c). Although the W and Ti based FDSSCs have the same interface structures of TiO₂/dye/electrolyte, the FDSSCs based on W wires have another interface of TiO₂/WO₃ labeled as "1" in Figure 6 (c), which enhances the recombination of charges. Thus, the semicircle in the low-frequency region is dependent on both of the interfaces of TiO₂/ dye/electrolyte and TiO₂/WO₃ for the FDSSCs based on the W wire substrates. While WO₃ is n-type semiconductor, a p-type NiO₂ layer was introduced between substrate and TiO₂ for comparison. A Ni wire as substrate was heated at 550°C for two hours to introduce a nickel oxide layer on the Ni wire. It was found that the resulting device tested under bias had no efficiency (Figure 3S in Supplementary Information) since the photo-generated electron transfer from the TiO₂ to the Ni substrate is hindered by the p-type NiO layer, as schemed in Figure 6 (c) and labeled as "2". Figure 6 (d) shows that the p-type NiO and n-type TiO₂ form a p-n junction (Figure 4S in Supplementary Information). Although the FDDSCs based on the above p-n junction show no performance, they might have potential applications in other photoelectric devices, such as UV-detector, light-emitting diode, etc.

Discussion

Anatase TiO₂ tree-like nanoarrays consisting of long TiO₂ nanowire trunks and a large number of short TiO₂ nanorod branches have been successfully synthesized on various metal wires (Ti, W, Ni, etc.) through one-step facile hydrothermal reaction. The FDSSCs based on anatase TiO2 tree-like nanoarrays on Ti wires show outstanding performance with PCE of 6.32%. The performance of FDSSCs based on a W wire is lower than that based on a Ti wire since the annealing of the WO₃ layer enhances charge recombination. If the substrate is an oxidized Ni wire, the cell has no efficiency since the p-type NiO layer prevent electronic transmission, while a novel p-n heterojunction can be obtained. This method is expected to be simple, facile, and low cost to prepare anatase TiO₂ treelikenanoarrays on different substrates, which could meet various requirements of optoelectronic applications to the substrates, and open up a promising avenue for new TiO₂-based applications of nanodevices.

Methods

Materials. Ti (or W, Ni, *etc.*) wires (99.9%, with the diameter of 0.50 mm) were purchased from China New Metal Materials Technology Co., Ltd. Boric acid (H_2BO_3 , 99.5%), ethanol (99.7%), acetone (99.5%), potassium titanium oxide oxalate dihydrate ($C_4K_2O_9Ti$ · $2H_2O$, 98.5%), and diethylene glycol (DEG) were purchased from Sinopharm Chemical Reagent Co. Ltd. Ruthenium 535-bisTBA (N719) was



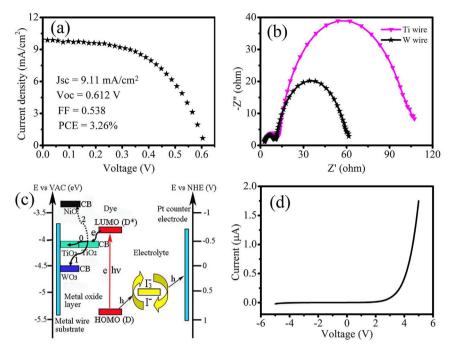


Figure 6 | (a) I–V curves of the FDSSCs fabricated with TiO_2 photoanodes prepared on W wire for 12 h. (b) EIS measurements were conducted in the dark under a bias of 0.7 V for the FDSSCs based on W and Ti wires with TiO_2 photoanodes synsentized for 12 h. (c) Energy states diagram for different metal wire-based FDDSC featuring the operation principle, and the dashed line shows the photogenerated electrons can not transfer from TiO_2 nanoarray to NiO film ("0", "1", "2" note the different oxide layers of TiO_2 , WO₃, and NiO corresponding to the metal wires of Ti, W, and Ni, respectively). (d) I–V curve of TiO_2 nanoarrays depositing on an oxidized Ni wire for 12 h under bias from -5 V to 5 V, and the rectification characteristic indicates a novel p-n junction of NiO/TiO₂.

purchased from Solaronix. Guanidinium thiocyanate (GuSCN, 99.0%) was from Amresco. Ammonium hexafluorotitanate (IV) ((NH₄)₂TiF₆, 98%), lithium iodide (LiI, 99.999%), iodine (I₂, 99.99%), 1-methyl-3- propylimidazolium iodide (PMII, 98%), 4-tert-butylpyridine (TBP, 96%) and tert-butyl alcohol (99.5%) were obtained from Aladdin. Acetonitrile (99.8%) and valeronitrile (99%) were from Alfa Aesar. All solvents and chemicals were reagent grade and were used as received without further purification.

Deposition TiO₂ seeds. The metal wires (Ti, W, Ni, etc.) were polished using an abrasive paper, washed using clean water with detergents, and subsequently ultrasonicated in deionized water, acetone and ethanol for 20 min, respectively. After dried in air, the Ni wires were placed in a muffle furnace and heated at 550°C for 2 hours. The color of the Ni wires was changed from silver to grey indicating that the surface of the Ni wires was oxidized. The treated-metal wires were immersed into the solution containing 0.1 M (NH₄)₂TiF₆ and 0.2 M H₃BO₃ at room temperature for 30 min to form a seed layer of TiO₂ nanoparticls⁴².

Synthesis of TiO₂ tree-like nanoarrays on metal wires. First, 1.240 g C₄O₉Ti·2H₂O was dissolved in 50 mL DEG by magnetic stirring for 45 min, and then 20 mL deionized water added to the solution and stirred for 15 min again. The final solution was transferred to a Teflon-lined stainless steel autoclave within a number of seeded-metal wires. Afterward, the autoclave was loaded into an oven at 180°C for different times (3, 6, 9, 12, 15 h) and then cooled down to room temperature naturally. After the hydrothermal reaction, the as-products were collected from the solution, rinsed with deionized water and dried at 80°C over night.

Fabrication of FDSSCs. The as-prepared anatase TiO₂ nanoarrays on metal wires were used as photoelectrodes for FDSSCs. Before dye sensitizing, the as-prepared TiO₂ nanoarrays were immersed into a 0.12 M TiCl₄ solution at 70°C for 40 min. After washing with water and drying in air, the samples were sintered at 500°C for 30 min. After cooling down to about 80°C, the TiO₂ electrodes were putted into 0.5 mM N719 dye in acetonitrile/tert-butanol (volume ratio 1 : 1), and kept for 20 h at room temperature. The sensitized electrodes were then washed in acetonitrile to remove physically-adsorpted dye molecules before the cell assembly. Afterwards, platinum wire with diameter 0.08 mm as the counter electrode was twisted around the dye-sensitized photoanode carefully. Subsequently, the both electrodes were immersed in a transparent plastic tube or a glass tube (Φ_{inner} :0.9 mm, Φ_{outer} : 1.2 mm) filled with electrolyte. The liquid electrolyte composition was 0.6 M PMII, 0.05 M LiI, 0.03 M I₂, 0.1 M GuSCN, and 0.5 M TBP in acetonitrile and valeronitrile (V : V = 85 : 15.)⁴³. The active length of cell was 4.5 cm. The effective active area of the cell is calculated by multiplying the diameter of photoanode and effective length of the cell.

Measurement and characterization. The X-ray diffraction (XRD, PANalytical B.V., The Netherlands) with Cu-Ka radiation measurements were carried out to analyze the phase purity of the samples. The surface morphology of the samples was recorded by field emission scanning electron microscopy (FESEM, FEI NOVA NanoSEM 450). Transmission electron microscopy (TEM) and high-resolution transmission electron microscopy (HRTEM) images were observed by FEI Titan G² 60-300. The TEM sample was prepared by drop casting ethanolic dispersion of TiO2 samples onto a carbon coated Cu grid. The current density-voltage (I-V) measurements were performed under AM 1.5G (100 mW/cm², calibrated with a Si photodiode) conditions using a solar simulator illumination (Newport, USA) and an Autolab electrochemic workstation (modelAUT84315, The Netherlands). For testing the adsorbed dye amount of the TiO2 working eletrodes, the sensitized-TiO2 samples desorbed the dye into 0.1 M NaOH solution. UV-Vis absorption spectrometry (UV-2550, Shimadzu) was measured to calculate the amount of the adsorbed dye amount, expressed in terms of moles of dye anchored per projected unit area of the photoelectrode. The electrochemical impedance spectroscopy (EIS) measurements were scanned in dark condition at a bias of 0.70 V with an amplitude of 10 mV in a frequency range from 100 kHz to 0.1 Hz. The aperture area is equal to the diameter of photoanode multiplied by the active length of the FDSSCs.

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Author contributions

L.C. carried out all experiments and wrote the draft of the manuscript; Y.H.G. & L.C. devised the original concept, designed the experiments, discussed the interpretation of results and revised the manuscript; L.Y.L. & J.S. contributed the TEM microstructure experiment; F.F.T. participated in part of experiments; L.Y.L. & L.S.N. revised the manuscript. All authors discussed the results and participated in manuscript revision.

Additional information

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