



 Cite this: *RSC Adv.*, 2021, 11, 25364

Direct synthesis of shaped MgAPO-11 molecular sieves and the catalytic performance in *n*-dodecane hydroisomerization†

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In industrial application, molecular sieves are usually used in a certain shape. This requires the addition of binder and causes the reduction of both the molecular sieve content and catalytic performance. Herein, pseudo-boemite was mixed with deionized water at room temperature, followed by the drop-wise addition of phosphoric acid, magnesium acetate solution, hydrofluoric acid, di-*n*-propylamine and 1-ethyl-3-methyl imidazolium bromide with vigorous stirring. The molar ratio of Al₂O₃ : P₂O₅ : MgO : HF : DPA : [EMIm]Br : H₂O in the gel was 1 : 1 : 0.03 : 0.18 : 0.4 : 1 : 45. Then the gel was dried, extruded and directly crystallized to form a shaped MgAPO-11 molecular sieve. X-ray diffraction, scanning electron microscopy, N₂ adsorption, ammonia temperature programmed desorption, pyridine adsorption infrared spectroscopy and nuclear magnetic resonance spectroscopy were used to investigate the physicochemical properties of the samples. X-ray diffraction, scanning electron microscopy and N₂ adsorption tests show that the shaped MgAPO-11 molecular sieve is fully crystallized and possesses hierarchical porosity. Mg is incorporated into the molecular sieve framework and the Pt catalyst supported by the obtained shaped MgAPO-11 exhibits excellent catalytic performance with *n*-dodecane conversion of 94% and isomer selectivity of 95% at 280 °C. Such a method for the direct synthesis of shaped molecular sieves shows potential for the green synthesis of molecular sieves in industry.

Received 14th May 2021

Accepted 14th July 2021

DOI: 10.1039/d1ra03758g

rsc.li/rsc-advances

Introduction

Hydroisomerization of *n*-alkanes is an important process to improve the octane number of gasoline and the low temperature fluidity of petrochemical diesel and lubricating oil.^{1–3} At present, catalysts used in hydroisomerization are mainly the bifunctional ones with functions of hydrogenation/dehydrogenation and isomerization/cracking.² The isomerization/cracking function is usually provided by molecular sieves with a one-dimensional medium pore structure and moderate acidity, such as MgAPO-11,^{4,5} SAPO-11 (ref. 6 and 7) and ZSM-22.^{8,9}

In industrial applications, most of the catalysts based on molecular sieves must be manufactured into specific shapes and sizes to alter the pressure drop and activity. The expectation is that performance optimization will eventually be achieved

while maintaining sufficient strength to withstand typical operational events. Generally, a process to prepare a shaped catalyst is complex, including the synthesis of molecular sieve powder, forming of a shaped support and impregnation of noble metal.¹⁰ Molecular sieve powder is usually synthesized by a hydrothermal method followed by washing; the whole process could lead to a lot of waste liquid. In the shaping process, silica, alumina, attapulgite, kaolin *etc.* are usually required as binders. The use of binders usually can bring a significant impact upon catalyst behaviour. This impact may be beneficial or deleterious to the performance. Recently, Lakiss *et al.*¹¹ reported that, in the dealkylation of 1,3,5 tri-isopropyl benzene, the bound FAU-type zeolite systematically displayed catalytic performance superior to the parent powder. They investigated the mechanism and found that, with the addition of binder, new catalytic sites are created and located on the external or mesoporous surface of the zeolites, *i.e.* at the zeolite–binder interface. In some cases, these binders are catalytically inert.¹² The addition of binders may partially block the micropores, dilute the active components and affect the catalytic performance of the catalysts.¹⁰ Dorado *et al.* studied the effect of binder on shaped Pd/HZSM-5 and Pd/HBeta catalysts and demonstrated that the presence of binder decreased strong acid site density and reduced *n*-butane conversion.¹³ Extrusion with silica or boehmite adversely

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† Electronic supplementary information (ESI) available. See DOI: 10.1039/d1ra03758g



affected both the catalyst lifetime and selectivity to light olefins in methanol-to-hydrocarbons.¹⁴ Therefore, a green route for the direct synthesis of shaped molecular sieves, which can not only simplify the preparation process, but also eliminate the negative impact of binders on the catalytic performance, is imperative for the industrial application of molecular sieves.

To directly synthesize shaped molecular sieves, two strategies have been reported. One strategy is the synthesis of molecular sieve-coating on a shaped substrate. For this method, shaped silicon dioxide, ceramic or aluminum foam is used as the substrate, and then the substrate is crystallized under a certain condition, so that the molecular sieve is attached to the surface of the substrate, and the shaped molecular sieve is directly obtained.^{15,16} For example, Bauer *et al.*¹⁶ used aluminum foam as the active support, made the support crystallized and partially converted the support into molecular sieves to prepare shaped SAPO-34, SAPO-18 and AlPO₄-5 molecular sieves. Although the obtained shaped molecular sieves contain partial substrates, the post-processing step after synthesis can be avoided by using this method.

The other strategy is to convert the non-molecular sieve component in the shaped body into molecular sieves, so as to solve the binder problem. For this strategy, shaped precursors are extruded or molded into a desired shape. Through crystallization, the shape of the precursor was kept and non-molecular sieve components in the shaped precursor are transformed into crystalline molecular sieves. The shaped precursors can be extruded from the mixture of molecular sieve and binder^{17–21} or from active amorphous gel.^{22,23} For example, Schumann *et al.*^{17,18} mixed 13X or LTA molecular sieve with the binder and extruded the mixture, then made the extrudates crystallize under hydrothermal condition to produce shaped molecular sieves. The binder in the shaped body was converted into molecular sieves, so the content of molecular sieve could be improved. Wang *et al.*²² mixed zeolite β, sodium aluminate, fumed silica, silica sol, NaOH, tetrapropylammonium bromide to form an active amorphous gel and extruded the gel. Under hydrothermal conditions, shaped ZSM-5 molecular sieve was directly produced without the shape change.

Recently, we reported a method for direct synthesis of shaped AlPO₄-11 molecular sieve,²⁴ which can be regarded as an improved SAC method. The phosphorus source, aluminum source, organic amine, hydrofluoric acid, and ionic liquid were mixed, extruded and crystallized to form columned support without the addition of any other reagent. Ionic liquid acts as a lubricant to facilitate the extrusion process and can retain water in the gel during the heating. Water promoted columned support crystallization. It is worth noting that the columned support was composed of 100% molecular sieves and have a high mechanical strength of 69 N cm⁻¹. This method not only simplifies the preparation process, but also overcomes the binder-covering problem and mechanical problem simultaneously. From an economic and environmental standpoint, such method shows potential for the green synthesis of molecular sieves in industry.

In AlPO₄-11 molecular sieve, the strict alternation of AlO₄ and PO₄ tetrahedra makes the framework electrically neutral.

Therefore, if AEL type molecular sieve is used as a support in the isomerization reaction, the introduction of Lewis or Brønsted acid site into the inactive framework by isomorphous substitution is a solution.²⁵ Several studies have focused on the incorporation of heteroatoms like Al,^{26,27} Ce,²⁸ Ga²⁹ and Ti³⁰ into the electronically neutral framework *via* direct synthesis. The coordination of heteroatoms in the electronically neutral framework can affect the catalytic properties because the incorporation of heteroatoms allows creation of Lewis and Brønsted acid sites.³¹ In this way, a potential catalyst for many catalytic reactions could be found. Recently, Azhagapillai *et al.* synthesized the aluminium substituted mesoporous KIT-6 material and examined its catalytic performance in isobutylbenzene acylation reaction. The density of Brønsted acid sites increases with the decrease in Si/Al ratio and aids isobutylbenzene conversion.³²

In this work, we further promoted the method for direct synthesis of shaped molecular sieve to synthesize the shaped MgAPO-11 through the addition of magnesium acetate in the initial material, extrusion of the mixture and crystallization. The Pt catalyst supported by the obtained shaped MgAPO-11 showed good catalytic performance in *n*-dodecane hydroisomerization. The incorporation of magnesium and the crystallization process of molecular sieve were also studied in detail.

Experimental

Chemicals and reagents

Pseudo-boemite (78.6 wt% Al₂O₃) was supplied from Sasol Co., Ltd. Hydrofluoric acid (40 wt%) and orthophosphoric acid (85 wt%) were supplied from Tianjin Kermel Chemical Reagent Co., Ltd. 1-Ethyl-3-methyl imidazolium bromide ([EMIm]Br) was produced by ourselves. Magnesium acetate (99%) and di-*n*-propylamine (DPA, 98 wt%) were supplied from China Pharmaceutical Group Chemical Reagent Co., Ltd. *n*-Dodecane (99%) and chloroplatinic acid (99.9%) were supplied from Fluka Co., Ltd. All chemicals and reagents were used directly without secondary purification.

Synthesis of shaped MgAPO-11 molecular sieve

The synthesis process of shaped MgAPO-11 molecular sieve includes two steps: the preparation of gel extrudates and the crystallization. The gel extrudates were typically prepared as follows: pseudo-boemite, phosphoric acid and deionized water were mixed and subjected to vigorous agitation at room temperature for 3 h. A separate solution of magnesium acetate solution, hydrofluoric acid, DPA and [EMIm]Br was added to above mixture. Stirring of the resulting slurry at room temperature for 3 h produced a uniform and viscous gel. The molar ratio of Al₂O₃ : P₂O₅ : MgO : HF : DPA : [EMIm]Br : H₂O in the mixture was 1 : 1 : 0.03 : 0.18 : 0.4 : 1 : 45. The obtained mixture was dried at 80 °C with a weight loss of ~60% to obtain a malleable solid. The malleable solid was directly extruded by an extruding machine. The malleable solid was extruded through the template with holes of 2 mm diameter, and the obtained cylindrical extrudates possess a diameter of 2 mm.

Then, the cylindrical extrudates were measured and cut manually with a length of 5–7 mm. The cylindrical extrudate of initial gel was denoted as IG-Mg (initial gel containing Mg).

The crystallization process was as follows: the IG-Mg was put into the autoclave without adding any other reagent. After crystallization at 200 °C for 4 h, IG-Mg was transformed into molecular sieve and the shaped MgAPO-11 molecular sieve was obtained, denoted as SM-Mg (shaped molecular sieve MgAPO-11). The template was removed by calcination for 11 h in air at 550 °C.

Shaped AlPO₄-11 molecular sieve was prepared by this method according to the procedure reported elsewhere for comparison, denoted as SM (shaped molecular sieve AlPO₄-11).²⁴

The industrial catalyst support SAPO-11-ind was purchased from AOS Catalyst (Dalian) Co., Ltd. for comparison of mechanical strength.

Preparation of Pt/SM-Mg and Pt/SM catalysts

The calcined SM-Mg and SM were impregnated with H₂PtCl₆ aqueous solution *via* incipient wetness impregnation method and the Pt loading amount is 0.5 wt%. After impregnation, the samples were dried at 120 °C for 2 h and reduced under H₂ (100 ml min⁻¹) at 400 °C for 4 h to obtain the catalysts. According to the support, the catalysts were labeled as Pt/SM-Mg and Pt/SM, respectively.

Characterization

X-ray diffraction (XRD) patterns of the shaped gel and shaped molecular sieves were measured on PANalytical X'Pert PRO X-ray diffractometer. Taking the sample crystallized for 4 h as reference, the sum intensity of the peaks at 2θ of 8.2°, 9.6°, 13.3°, 15.8°, 20.6° and 21.1° for each sample was compared with that of the sample crystallized for 4 h to obtain the relative crystallinity. Micromeritics ASAP 2420 was used to measure the surface area and pore volume of the samples. The morphology of the samples was observed by JSM 7800F field emission scanning electron microscope (SEM). Ammonia temperature programmed desorption (NH₃-TPD) was tested on a Micromeritics AutoChem II 2920 instrument. Firstly, 0.1 g sample (20–40 mesh) was treated at 350 °C with He for 1 h and cooled to 100 °C. Secondly, 5 vol% NH₃/He was injected for adsorption until saturation was reached. The temperature was then raised to 600 °C at a rate of 10 °C min⁻¹, and the desorption process was monitored with a thermal conductivity detector. Pyridine adsorption infrared (Py-IR) spectrum was measured on a Bruker Tensor27 spectrophotometer. The sample was vacuumed at 400 °C for 60 min and cooled to room temperature, and then the spectra were recorded as blank. After adsorption of pyridine and vacuum treatment at 150 °C and 300 °C, respectively, the corresponding spectrograms were recorded at room temperature. The nuclear magnetic resonance (NMR) spectrum of solid samples was performed on Agilent DD2-500 MHz NMR spectrometer. The radial mechanical compressive strength was evaluated on DL II intelligent particle strength tester. The sample was placed in the center of the sample table. After the

instrument was started, the force was applied automatically. When the sample is touched, the display would have real-time data output. With the force continued to increase until the sample was broken, the maximum value of the force was locked, that is, the compressive strength value of the sample. 20 specimens were tested and the average value was taken. A Bruker Multi Ram spectrometer was used to record the Raman spectra of the samples. The Fourier-transform infrared (FT-IR) spectra of the samples were recorded by the Bruker TENSOR27 FT-IR spectrometer. The sample and KBr were mixed and ground with a weight ratio of 1 : 50. Prior to FT-IR characterization, the samples were grinded, washed with deionized water three times and then dried at 80 °C overnight.

Catalyst testing

n-Dodecane hydroisomerization was used as a model reaction to evaluate the isomerization performance of catalysts Pt/SM-Mg and Pt/SM. The reaction was carried out in a fixed-bed reactor with an inner diameter of 8 mm at atmospheric pressure. The mass of catalyst was 0.76 g in each reaction. The reaction temperature ranged from 210 to 310 °C, the molar ratio of hydrogen-to-hydrocarbon was 15, and the liquid hourly space velocity (LHSV) was 1.0 h⁻¹. The products were analyzed online by Agilent 7890A gas chromatography with HP-PONA capillary column and FID detector.

Results and discussion

Structure and morphology analysis

The photograph and XRD patterns of IG-Mg and SM-Mg are presented in Fig. 1. As shown in Fig. 1A, IG-Mg is a cylindrical extrudate with a length of 5–7 mm and a diameter of 2 mm. The crystallized SM-Mg keeps well the shape of IG-Mg. In Fig. 1B, IG-Mg shows a broad peak at around 15–30° and four sharp diffraction peaks at 10.2°, 22.5°, 23.5° and 25.6° over the broad peak. The broad peak at around 15–30° can be attributed to amorphous phase, indicating the amorphous nature of IG-Mg. The peaks are attributable to dipropylammonium bromide (JCPDS: 00-050-2274), which was produced by the reaction of protonated di-*n*-propylamine with bromide ions during the drying process. In the XRD pattern of SM-Mg, both the diffraction of the amorphous phase and dipropylammonium bromide disappear and the characteristic diffraction of the AEL-type

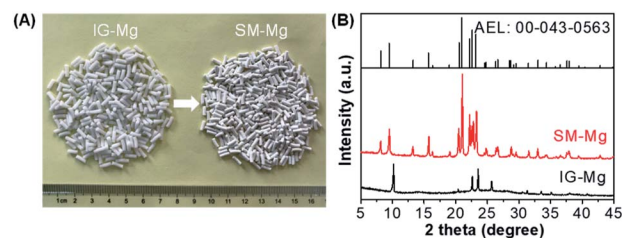


Fig. 1 (A) Photographs and XRD patterns of IG-Mg and SM-Mg. The JCPDS of AEL-type molecular sieve was also given in (B).

molecular sieve appear clearly (JCPDS: 00-043-0563). This indicates the sample is well crystallized.

Morphology of IG-Mg and SM-Mg was observed by SEM. As seen from the surface and cross section images shown in Fig. 2(A–D), IG-Mg is smooth and compact, and contains macropores. The macropores were formed by the removal of water in IG-Mg. The SEM images also show that IG-Mg extrudates are non-crystallized gel monolith. The crystallized SM-Mg is completely composed of crystallites with uniform morphology (Fig. 2E–H). Furthermore, SM-Mg possesses the diffraction pattern of AEL-type molecular sieve with high intensity. These results suggest that SM-Mg is entirely composed of AEL-type molecular sieve.

Texture properties of SM-Mg were studied by N_2 physisorption. As Fig. 3 shows, the physisorption isotherm of SM-Mg is a combination of the types I and IV. In the range of $p/p_0 < 0.01$, the isotherm rises rapidly with a large uptake, which was caused by the filling of micropores. In the range of $0.4 < p/p_0 < 0.99$, the isotherm exhibits a H3 type hysteresis loop, which is attributed to the capillary condensation of N_2 in the mesopores.^{33,34} The mesopores might be formed by the accumulation of molecular sieve crystallites. As shown in Table 1, the BET surface area and micropore volume of SM-Mg are $177.1 \text{ m}^2 \text{ g}^{-1}$ and $0.07 \text{ cm}^3 \text{ g}^{-1}$, respectively. Additionally, enriched macropores can be observed among the intergrown crystallite clusters in the SEM

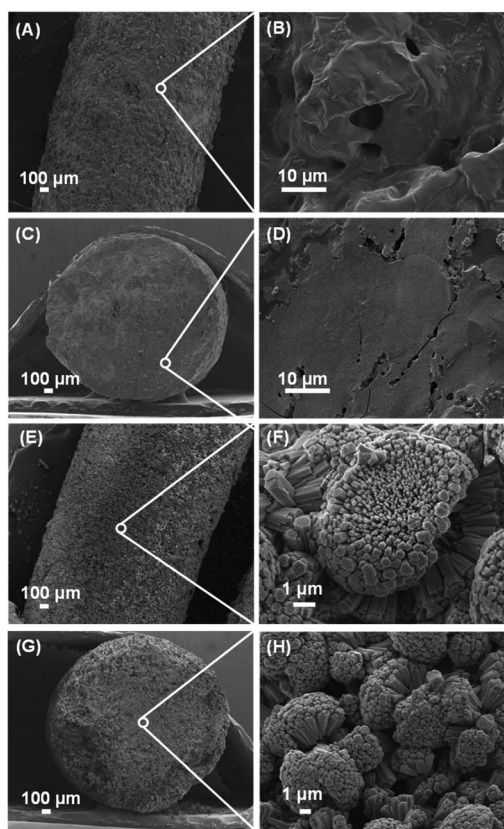


Fig. 2 SEM images of the gel extrudate and the product. (A and B) Surface of IG-Mg; (C and D) cross section of IG-Mg (E and F) surface of SM-Mg; (G and H) cross section of SM-Mg.

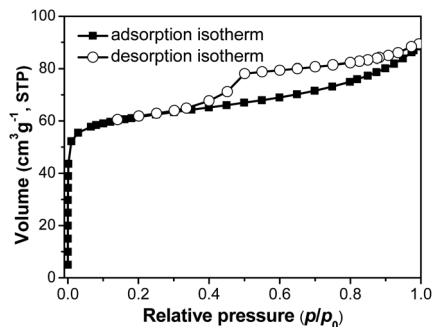


Fig. 3 N_2 physisorption isotherm of SM-Mg.

images (Fig. 2E–H). This phenomenon suggests that SM-Mg has a hierarchical porosity composed of micropores, mesopores and macropores. The radial compressive strength of SM-Mg was further tested and the result is 66 N cm^{-1} (Table 1). As a comparison, the mechanical strength of industrial catalyst support SAPO-11-ind was measured and the value was 78 N cm^{-1} . Although the mechanical strength of SM-Mg is slightly lower than that of SAPO-11-ind, such mechanical strength can still ensure that SM-Mg meet the requirement of being used as an isomerization catalyst support.

To study the thermal stability of SM-Mg and SM, TG measurement were performed. As shown in Fig. S1,† the weight losses in the range of $40\text{--}800 \text{ }^\circ\text{C}$ is mainly the decomposition of [EMIm]Br and organic amines, except for the desorption of a small amount of physisorbed water. From the TG analysis, it can be known that both SM-Mg and SM have good thermal stability.

Based on the results of XRD, SEM, N_2 physisorption, TG and mechanical strength test, it can be deduced that SM-Mg is composed of molecular sieve crystallites, possesses hierarchical porosity and good thermal stability, exhibits excellent mechanical strength. This makes SM-Mg a promising support for industrial catalysts.

Incorporation of Mg

The incorporation of Mg can bring about the acidity change in the molecular sieve framework and the acidity is characterized by NH_3 -TPD and Py-IR.

Fig. 4 shows the NH_3 -TPD profiles of SM and SM-Mg. SM shows only one desorption peak, while SM-Mg shows two obvious desorption peaks. The peak at round $180 \text{ }^\circ\text{C}$ is observed on the profiles for both SM and SM-Mg. This peak belongs to the desorption of weakly adsorbed NH_3 , such as these adsorbed at the defect sites and on the weak acid P-OH.^{4,35} The peak at round $350 \text{ }^\circ\text{C}$ appears only on the NH_3 -TPD profile of SM-Mg. This peak may be caused by the desorption of NH_3 adsorbed at strong acid sites formed by replacing framework Al^{3+} with Mg^{2+} .⁴ Compared with SM, the acidity of SM-Mg was increased significantly. This was caused by the incorporation of Mg into the molecular sieve framework.

To distinguish the type of acid sites, Py-IR spectra were recorded for SM-Mg at $150 \text{ }^\circ\text{C}$ and $300 \text{ }^\circ\text{C}$. As shown in Fig. 5, the

Table 1 Textural properties and mechanical strength of SM-Mg

Sample	S_{BET}^a ($\text{m}^2 \text{g}^{-1}$)	S_{ext}^b ($\text{m}^2 \text{g}^{-1}$)	V_{total}^c ($\text{cm}^3 \text{g}^{-1}$)	V_{micro}^c ($\text{cm}^3 \text{g}^{-1}$)	V_{meso}^d ($\text{cm}^3 \text{g}^{-1}$)	Str^e (N cm^{-1})
SM-Mg	177.1	38.3	0.16	0.07	0.09	66

^a BET surface area. ^b *t*-plot external surface area. ^c *t*-plot micropore volume. ^d BJH desorption cumulative volume of pores. ^e Mechanical strength.

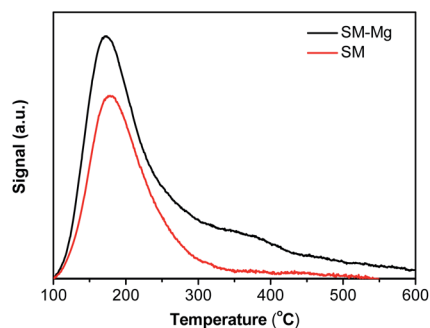


Fig. 4 NH_3 -TPD curves of SM-Mg and SM. Samples underwent temperature-programmed desorption under NH_3 with detailed parameters shown in sample characterization.

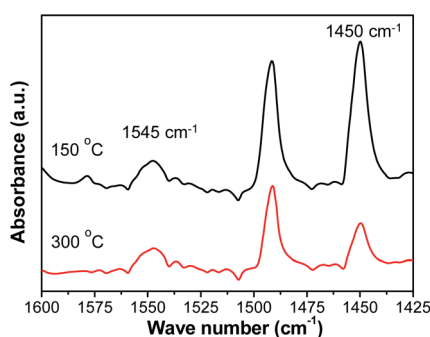


Fig. 5 Py-IR spectra of SM-Mg with pyridine desorption temperatures of 150 °C and 300 °C.

peak at 1545 cm^{-1} should be assigned to the Brønsted acid sites and the peak at 1450 cm^{-1} is associated with Lewis acid sites.³⁶ The peak areas decrease markedly with the increasing of temperature, which is due to desorption of weakly adsorbed pyridine at elevated temperatures. The quantitative results of Brønsted and Lewis acid sites are given in Table 2. These results indicate that the SM-Mg contains both Brønsted and Lewis acid sites. AlPO_4 -11 possesses a neutral framework that is built up

Table 2 Acidity of SM-Mg measured by Py-IR at different temperatures

Sample	Brønsted ($\mu\text{mol Py g}^{-1}$)		Lewis ($\mu\text{mol Py g}^{-1}$)	
	150 °C	300 °C	150 °C	300 °C
SM-Mg	15.3	14.1	45.2	16.7

from strict alternation of AlO_4 and PO_4 tetrahedra.³⁷ Therefore, it is not suitable to be used in isomerization reaction. With the isomorphous substitution of Mg^{2+} for Al^{3+} , electrically negative framework is produced. The negative framework charges are balanced by H^+ and then Brønsted acid sites are formed.²⁵ In this way, SM-Mg has the catalytic acidity and can be used as a support in isomerization reaction.

To further investigate whether Mg is incorporated into the molecular sieve framework, ^{27}Al and ^{31}P MAS NMR experiments were carried out and the changes of local chemical environment of P and Al atoms were studied.

As Fig. 6 shows, the amorphous IG-Mg has one weak resonance at 42.5 ppm and two strong resonances at 8.0 and -13 ppm in ^{27}Al NMR spectrum. The weak broad resonance at 42.5 ppm can be ascribed to four-coordinated Al in the amorphous aluminium phosphate.^{38–40} The resonance at 8 ppm should arise from six-coordinated Al in the unreacted pseudo-boemite.⁴¹ The resonance at -13 ppm can be assigned to six-coordinated Al in the amorphous aluminium phosphate, which arises from AlO_4 sites that are additionally coordinated by $\text{OH}^-/\text{H}_2\text{O}$ species.^{38–40} Since the six-coordinated Al atoms are dominant, IG-Mg is not in a crystalline state. Meanwhile, the ^{31}P NMR spectrum of IG-Mg exhibits a broad peak at around -11 ppm, which can be attributed to P connected with less than four AlO_4 in amorphous aluminum phosphate.^{38,39,42,43} These P species include $\text{P}(\text{OAl})(\text{OH})_3$, $\text{P}(\text{OAl})_2(\text{OH})_2$ and $\text{P}(\text{OAl})_3(\text{OH})$.^{38,44} The ^{27}Al and ^{31}P NMR results indicate the amorphous nature of IG-Mg.

SM-Mg shows one dominant resonance at 37.5 ppm and two weak resonances at 13.0 and -6 ppm in the ^{27}Al MAS NMR spectrum. For MgAPO-11 , the resonance at 37.5 ppm is caused by four-coordinated Al. Akolekar *et al.*⁴⁵ found a sharp ^{27}Al signal at 30–40 ppm in the NMR study of MgAPO-11 . They

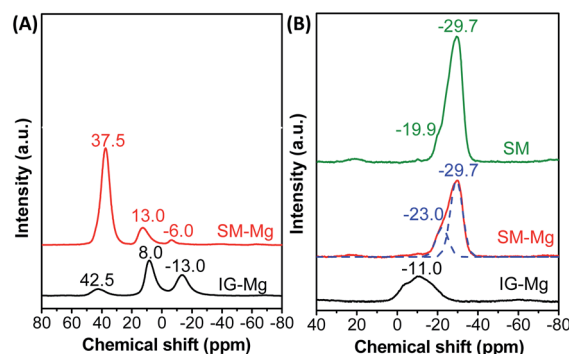


Fig. 6 ^{27}Al (A) and ^{31}P (B) MAS NMR spectra of IG-Mg, SM-Mg and SM.

attributed this signal to four-coordinated Al in crystallized aluminophosphate. The weak resonance at 13.0 ppm should be assigned to five-coordinated Al, which is caused by the interaction of framework Al with F^- or OH^-/H_2O .⁴⁶ The other weak resonance at -6 ppm arises from six-coordinated Al in $Al(OP)_4F_2$.⁴⁷ After crystallization, the six-coordinated Al signal of unreacted pseudo-boemite and amorphous aluminium phosphate disappears. The four-coordinated Al signal becomes dominant and only a little five- or six-coordinated Al signals appear. Such five- and six-coordinated Al signals associate with the species interacted with fluoride ions or water and are often observed in the NMR spectra of molecular sieves synthesized in F-containing system.^{39,48} The coordination environment of Al changed during crystallization, and the framework is mainly composed of four-coordination Al. This result evidently proves that SM-Mg is completely crystallized.

The ^{31}P MAS NMR spectrum of SM shows a strong peak at -29.7 ppm and a weak shoulder peak at -19.9 ppm. The weak shoulder peak at -19.9 ppm is assigned to tetrahedral P in $-CLO$ type aluminophosphate molecular sieves.⁴⁹ According to the previous work,²⁴ when the crystallization time was between 1 h to 24 h, the product always contains intermediate molecular sieve phase ($-CLO$). After 48 h, AEL-type molecular sieve was obtained. Fig. S2† displays the ^{31}P NMR spectra of SM crystallized for different time. The weak peak at -11.0 ppm is attributed to the structural P-OH groups in the framework, while the intense peak at -19.9 ppm is assigned to tetrahedral P. For SM-Mg, the ^{31}P MAS NMR spectrum shows an asymmetric line-shape for the signal at around 20–35 ppm, which consists of complex multiplets of overlapping peaks. This signal could be divided into two peaks, namely a strong peak at -29.7 ppm and a shoulder peak at -23 ppm. In general, the ^{31}P signal is sensitive to the change of chemical environment. For $AlPO_4$ molecular sieve, the ^{31}P NMR spectrum exhibits only one single symmetric resonance at around -30 ppm.^{24,50} The incorporation of Mg into the framework of aluminophosphate molecular sieves has obvious effect on ^{31}P chemical shift. For example, Deng *et al.*⁵¹ found that the ^{31}P spectrum of MgAPO-5 has a broad and asymmetric signal. Though deconvolution, four peaks at -29.2 , -22.2 , -17.2 and -11.3 ppm were obtained, which were assigned to P(4Al, 0Mg), P(3Al, 1Mg), P(2Al, 2Mg) and P(1Al, 3Mg), respectively. Prasad and Haw⁵² found that the ^{31}P signal of MgAPO-20 could be deconvolved into four peaks at -33.9 , -28.0 , -21.3 and -14.8 ppm, which were assigned to P atoms connecting with 0, 1, 2 and 3 Mg atoms, respectively. Wang *et al.*⁵³ characterized MgAPO-11 by ^{31}P NMR and assigned the peaks at -29 and -23 ppm to P(4Al) and P(3Al, 1Mg) units in the framework, respectively. The above NMR results indicate that Mg can substitute for Al in the framework, leading to the formation of four P coordination modes: P(4Al), P(3Al, 1Mg), P(2Al, 2Mg), and P(1Al, 3Mg). Except the P(4Al) peak at -29.7 ppm, only one peak at -23 ppm was detected in the ^{31}P MAS NMR spectrum of SM-Mg. Because of the low Mg content ($Al_2O_3 : MgO = 1 : 0.03$) and the well dispersion of the magnesium source in the reaction gel, P(1Al, 3Mg) and P(2Al, 2Mg) unit would not appear. Accordingly, the peak at -23 ppm

should be attributed to P(3Al, 1Mg) unit formed by Mg substituting for Al.

As the above results show, Mg was successfully incorporated into the molecular sieve framework and Brønsted acid sites were formed. Thus, the corresponding SM-Mg is endowed with catalytic acidity.

Formation process of SM-Mg

XRD is employed to follow the crystallization process of SM-Mg. Fig. 7 shows the XRD patterns of samples crystallized for different time. The relative crystallinity curve is exhibited in Fig. 8. As Fig. 7 shows, the XRD pattern of IG-Mg presents the diffraction peaks of dipropylammonium bromide and a broad weak reflection related to amorphous phase. After crystallized for 0.5 h, the dipropylammonium bromide peaks disappear and only the broad weak reflection can be seen. This indicates that the structure was rearranged and the sample was completely composed of amorphous phase. As shown in Fig. 8, the relative crystallinity is 0 at this time. As the crystallization time increasing to 4 h, the diffraction peaks of AEL-type molecular sieve appear, and the relative crystallinity increase rapidly. Further increase of the crystallization time causes no significant increase of the relative crystallinity. These results indicate that the sample was completely crystallized after 4 h.

Furthermore, the evolution process of micropores is also shown in Fig. 8. When the crystallization time is 0.5 h, the micropore volume reaches $0.01 \text{ m}^3 \text{ g}^{-1}$. This indicates that molecular sieve structure has appeared in the product. The amount of micropore volume can reflect the crystallization degree of molecular sieves. When the crystallization is completed, the micropore volume reaches the maximum value of $0.07 \text{ m}^3 \text{ g}^{-1}$.

In our previous work, shaped $AlPO_4$ -11 was directly synthesized through this method.²⁴ It took a long crystallization time (2 d) and an intermediate ($-CLO$ type $AlPO_4$) appeared during crystallization. In this study, it takes only 4 h to get fully crystallized shaped MgAPO-11. No intermediate was detected during the crystallization process. Compared with the synthesis of shaped $AlPO_4$ -11, Mg significantly accelerated the crystallization of shaped MgAPO-11. In the ionothermal synthesis of MgAPO-11, Wang *et al.*⁵³ also found that the addition of Mg accelerated the molecular sieve crystallization and inhibited the

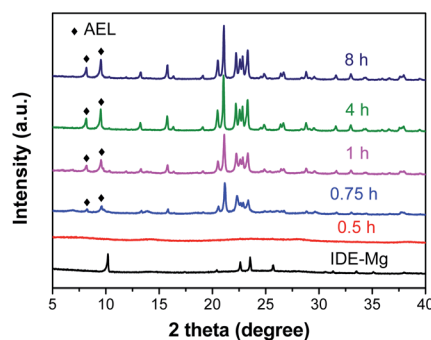


Fig. 7 XRD patterns of the samples crystallized for different time.

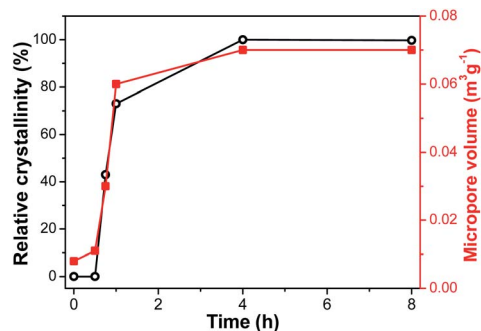


Fig. 8 Time-dependent curves of relative crystallinity and micropore volume.

transition to dense phase. Perez-Pariente *et al.*⁵⁴ pointed out that the synthesis of metal-incorporated aluminophosphates is strongly correlated to the presence of the hetero-metal atoms. A typical example is the synthesis of AFI- and CHA-type molecular sieves.⁵⁵ Only the AFI-type molecular sieve was obtained from hetero-metal-free gel, while CHA-type molecular sieve was obtained from Zn-containing gel. Similar phenomena were also observed in the synthesis of AEL- and CHA-type molecular sieves.⁵⁶ When V or Ti was added into the starting gel, the CHA-type molecular sieve was more likely to occur in the product than the AEL-type one. Furthermore, Lewis *et al.*⁵⁷ found that strong interaction can be formed between metal species and template. The interaction between the metal species and the template lead to the formation of a specific molecular sieve. The metal species play a cooperative framework-template role in the synthesis of molecular sieve. In our work, the addition of Mg favored the rapid formation of MgAPO-11 without any intermediate formation. This indicates that the Mg species may contribute a cooperative structure-directing effect along with DPA. The mechanism of such effect would be further studied.

FT-IR and Raman spectra were used to investigate the crystallization process of SM-Mg. The results are presented in Fig. 9 and 10, respectively. The FT-IR and Raman peaks of the samples can be divided into two groups: the peaks of adsorbed/embedded [EMIm] and DPA in the samples and the peaks of Al-O-P species. It should be emphasized that, before FT-IR or Raman characterization, the as-synthesized sample was ground into powder, washed with deionized water for three times, and then dried at 80 °C. Therefore, the signals of organics detected

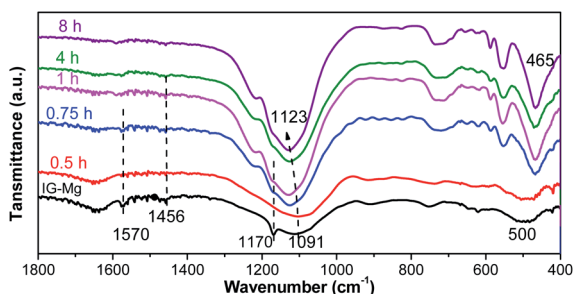


Fig. 9 FT-IR spectra of samples crystallized for different time.

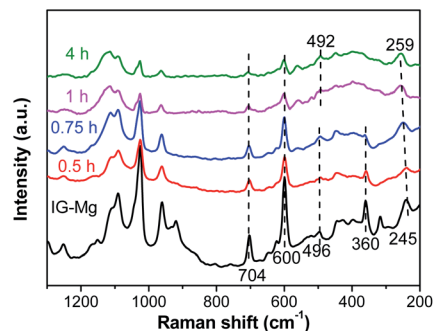


Fig. 10 Raman spectra of samples crystallized for different time.

in the FT-IR and Raman spectra should be assigned to [EMIm]/DPA that did not be washed out.

As shown in Fig. 9, the FT-IR peaks at 1570, 1465 and 1170 cm^{-1} are belonging to [EMIm]Br and DPA⁵⁸ and the peaks gradually weaken and disappear with the progress of crystallization. This indicates that the interaction between the [EMIm] Br/DPA and phosphorus species is strong at the beginning. As the crystallization proceeds, molecular sieve is formed, the interaction between the [EMIm]Br/DPA and phosphorus becomes weak, and the [EMIm]Br/DPA is easily washed out. It is because at the beginning, acid-base reaction occurs between [EMIm]Br/DPA and phosphoric acid to form a salt, there is a strong interaction between them, so [EMIm]Br/DPA cannot be washed out. As the crystallization proceeds, the molecular sieve framework is formed though the connection of phosphorus, aluminum and magnesium by sharing a common oxygen vertex. Therefore, phosphorus, aluminum, magnesium and oxygen bond together, while [EMIm]Br and DPA are free and exist in the cavity acting as either a space filler or structure directing agent. In this way, the interaction between [EMIm]Br/DPA and molecular sieve framework becomes weak and can be washed out of the sample. In the Raman spectra (Fig. 10), the peaks assigned to [EMIm]Br at 600 and 704 cm^{-1} (ref. 58) decrease significantly during the crystallization. This result further confirms that the strong acid-base interaction between [EMIm] Br/DPA and phosphorus species in amorphous phase is transformed into the weak host-guest interaction of [EMIm]Br/DPA with the molecular sieve framework. Because of the weakening of the interaction, [EMIm]Br can be gradually washed out of the sample.

During the crystallization, the peak position and type of the Al-O-P bonds have also changed. As shown in Fig. 9, the FT-IR spectrum of the initial gel (IG-Mg) exhibits the peaks at 500 and 1091 cm^{-1} . The peak at 500 cm^{-1} is attributed to Al-O-P bending vibration.^{59,60} The intensity of this peak increases with crystallization time. Redshift occurs and leads to the peak shifting to 465 cm^{-1} when the crystallization is completed (4 h). This indicates that the structure of the sample is gradually rearranged. At the same time, the peak assigned to the asymmetric stretching vibrations of the tetrahedra (TO_4)^{59,60} shifts from 1091 cm^{-1} to 1123 cm^{-1} . This result indicates that the amorphous gel is gradually transformed to molecular sieve.

As shown in Fig. 10, the Raman spectrum of IG-Mg exhibits peaks at 245, 360 and 496 cm^{-1} . The intensity of the peak at 360 cm^{-1} , which is assigned to isolated octahedral AlO_6 (ref. 61 and 62), gradually decreases and disappears. During the crystallization, most of the octahedral AlO_6 species in the initial gel would be consumed, and then tetrahedral AlO_4 species would appear. In this process, phosphoric ions connect Al atoms by replacing hydroxyl groups and the Al-O-P bond could be formed.⁶³ As the peak corresponding to the AlO_6 species disappears, the peak of 4-membered rings (4-MR) gradually shifts from 496 cm^{-1} to 492 cm^{-1} , possibly due to the increase in the number of 4-MR.⁶⁴ Moreover, the vibration peak of 10-member ring shifts from 245 cm^{-1} to 259 cm^{-1} and the peak intensity is enhanced. This indicates that the 10-membered ring is becoming more and more regular and finally the AEL-type framework is formed.^{65,66}

Water is essential for the synthesis of molecular sieves. Matsukata *et al.*^{67,68} found that the formation of a pseudo-*BEA structure required an exposure of the dry gel to steam for a few hours, and nanoparticle intermediates containing a pseudo-*BEA structure can be transformed into highly ordered *BEA molecular sieve. They presumed that the dehydration of silanols in the precursor during the thermal treatment in the presence of a minimum amount of water possibly promoted the crystallization. In the ionothermal synthesis of $\text{AlPO}_4\text{-5}$ and $\text{AlPO}_4\text{-11}$, Ma *et al.*⁶⁹ found that both the induction and growth time of crystallization curves were dramatically reduced after the addition of reagent quantities of water. As a nucleophilic reagent, the added water would improve hydrolysis reactions and thereby facilitate the formation of solution active species. Furthermore, the added water could also promote the synthesis kinetics by promoting both the production and the transport of H^+ and OH^- hydrates. In this work, [EMIm]Br in the gel extrudates plays a role of retaining water. Water promoted the depolymerization and repolymerization, participated in the bond breaking and rebonding in the six- and five-coordination species to form the four-coordination species. Under the direction of the structure-directing agent (DPA), the gel structure was rearranged to form the molecular sieve framework. During the crystallization, there is no liquid in the system and no dissolution occurs. The crystallization process may follow the solid phase transformation mechanism.

Superior performance in hydroisomerization of *n*-dodecane

The isomerization properties of catalysts Pt/SM-Mg and Pt/SM were evaluated using *n*-dodecane hydroisomerization as a model reaction. The results are shown in the Fig. 11. Generally, *n*-alkanes hydroisomerization is carried out on a bifunctional catalyst. The bifunctional catalyst should contain metallic sites for hydrogenation–dehydrogenation and acid sites for isomerization.² The hydroisomerization of *n*-alkanes follows the following steps:² (1) *n*-alkanes are adsorbed on metallic sites and converted to *n*-alkenes by dehydrogenation. (2) *n*-Alkenes are transferred from the metallic sites to acid sites. (3) *n*-Alkenes are adsorbed at acid sites and transform into carbenium ions through the protonation reaction, followed

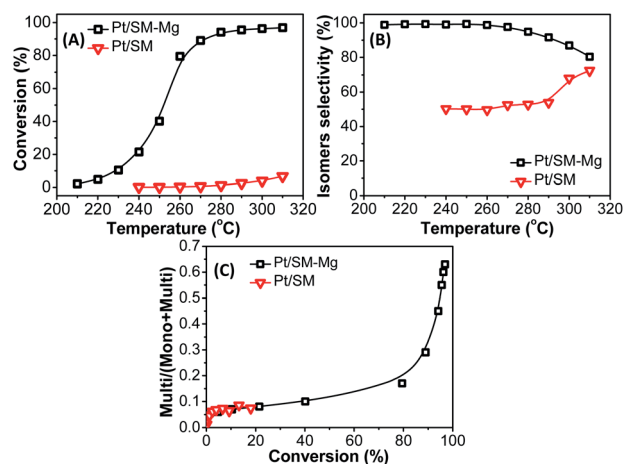


Fig. 11 Catalytic performance of catalysts Pt/SM-Mg and Pt/SM in *n*-dodecane hydroisomerization. (A) *n*-Dodecane conversion, (B) isomers selectivity, (C) multi/(multi + mono) ratio against conversion. Multi and mono represent the multi-branched and mono-branched *i*-dodecanes, respectively. Reaction conditions: LHSV = 1 h^{-1} , atmosphere H_2 , H_2/n -dodecane (mol) = 15.

by the isomerization/cracking reaction. (4) The isomerized carbenium ions deprotonate at acid sites to form *i*-alkenes. (5) *i*-Alkenes are transferred from acid sites to metallic sites. (6) On metallic sites, *i*-alkenes are converted to *i*-alkanes by hydrogenation and then desorbed. Therefore, to improve the selectivity of isomerization catalysts, it is necessary to effectively control the balance between metal and acid functions. The isomerization performance of metal catalysts without acidity is always poor.

With the increase of reaction temperature in the range of 210–310 °C, the *n*-dodecane conversion of Pt/SM-Mg and Pt/SM gradually increase. For catalysts based on $\text{AlPO}_4\text{-11}$ molecular sieves, namely Pt/SM, the *n*-dodecane conversion increases slowly. When the temperature reaches 310 °C, the conversion is only 7%. In addition, the isomerization selectivity of Pt/SM is less than 80%. These results should be mainly attributed to the lack of acidity in SM. Reaction occurred on Pt/SM is mainly the thermal cracking of C–C bonds. As a result, both activity and selectivity of the hydroisomerization reaction are low. In contrast, with the incorporation of Mg, the acidity of SM-Mg increases obviously. The acid and metal functions on Pt/SM-Mg are well balanced, so SM-Mg exhibits higher improved activity and selectivity. With the increase of reaction temperature, the conversion increases rapidly. When the temperature is 280 °C, the conversion reaches 94%. Moreover, in the range of 210 °C to 280 °C, the selectivity is above 95%. Even if the temperature is increased to 310 °C, the selectivity can still be over 81%. Such high activity and selectivity is due to the suitable Mg incorporation, which increases the number of acid sites and guarantees the moderate acidity of the molecular sieve.

i-Alkanes, especially the multi-branched *i*-alkanes are desirable for both increasing the octane number of gasoline and improving the low-temperature flow property of diesel and lubricant. Therefore, maintaining a high content of multi-

branched isomers in the isomer fractions (multi/(multi + mono)) is important for achieving a high isomerization performance. The multi/(multi + mono) ratio against the conversion is shown in Fig. 11. With the increase of conversion, the multi/(multi + mono) ratio of both Pt/SM-Mg and Pt/SM increase gradually. However, the multi/(multi + mono) ratio of Pt/SM is always at a very low level, with a maximum of only 0.073, which is caused by the low catalytic activity of Pt/SM. In contrast, the multi/(multi + mono) ratio of Pt/SM-Mg is relatively higher. Before the conversion increases to 80%, the multi/(multi + mono) value increases slowly. As the conversion exceeds 80%, the multi/(multi + mono) value increases rapidly, reaching a maximum of 0.63. This is because at a low conversion, due to the good diffusion performance, the mono-branched olefin intermediate is easy to desorb from the acid site and diffuse to the metal center, where hydrogenation reaction occurs and the mono-branched isomers are obtained. At a high conversion, the mono-branched olefin intermediate undergoes a further skeleton isomerization to form multi-branched olefin intermediate, and then hydrogenation occurs on the metal center to generate multi-branched isomers.

It has also been reported that MgAPO-11 and SAPO-11 was used as the support for *n*-dodecane hydroisomerization. Yang *et al.*⁴ synthesized MgAPO-11 and SAPO-11 powder by hydrothermal method and Wang *et al.*⁵³ synthesized MgAPO-11 powder by ionothermal method. Then the powder was shaped into particles for Pt/MgAPO-11 catalyst support. Table 3 shows the results of *n*-dodecane hydroisomerization on Pt/MgAPO-11 and Pt/SAPO-11 catalysts prepared by different methods. For Pt/SM-Mg catalyst, the superior catalytic performance with *n*-dodecane conversion of 94% and isomers selectivity of 95% is obtained at 280 °C. For Pt/MgAPO-11-hydr catalyst, the conversion and isomers selectivity were up to 81% and 78%, respectively. The Pt/MgAPO-11-iono catalyst had a catalytic performance with a conversion of 93% and a selectivity of 94%. The Pt/SAPO-11 catalyst exhibited a conversion of 94% and a selectivity of 81% at 300 °C. Comparing with that of SAPO-11, the higher acid strength of the MgAPO-11 was probably the reason for its higher activity and isomer yield.⁴ The above as-synthesized molecular sieves are all in powder. The molecular sieve powders usually need to be extruded with binders to make them more suitable for practical applications. However, with the addition of binder, the acid sites are easily covered and the micropores may be blocked, leading to dilution of the active species. In this work, the shaped MgAPO-11 molecular sieve support is completely composed of molecular sieve without

binder, which can be directly used in industrial reactions. The preparation of shaped support composed of molecular sieve can make the external surface and pore mouth of molecular sieve uncovered, so the catalytic performance shaped support can be well played. The result suggests that the direct synthesis of shaped molecular sieves is not only simple, but also can produce molecular sieve with high catalytic performance, which is worthy of further promotion.

Conclusions

In summary, shaped MgAPO-11 molecular sieve with high relative crystallinity, excellent mechanical strength and hierarchical porosity has been synthesized directly. Mg is successfully incorporated into the molecular sieve. Crystallization of the shaped MgAPO-11 molecular sieve is investigated in detail, and it is considered that the crystallization process follows solid phase transformation mechanism. In the hydroisomerization of *n*-dodecane, the Pt/SM-Mg catalyst exhibits the excellent catalytic performance with *n*-dodecane conversion of 94% and isomers selectivity of 95% at 280 °C. The directly synthesis of shaped molecular sieve has proven to be a high-yield, high-safety and low-pollution methodology. As these features show, this method is an eco-friendly route and has potential significance in industrial applications.

Author contributions

Ping Wang: conceptualization, methodology, validation, formal analysis, investigation, resources, data curation, writing – original draft, writing – review & editing, visualization. Hao Liu: formal analysis; visualization. Congxin Wang: supervision, funding acquisition. Guang Lv: data curation; investigation. Donge Wang: supervision. Huaijun Ma: supervision. Zhijian Tian: conceptualization, writing – review & editing, visualization, supervision, project administration, funding acquisition.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

The acknowledgements come at the end of an article after the conclusions and before the notes and references. The authors thank the National Key R&D Program of China (No.

Table 3 The results of *n*-dodecane hydroisomerization on Pt/MgAPO-11 and Pt/SAPO-11 catalysts prepared by different methods^a

Catalysts	Al ₂ O ₃ /Mg molar ratio	Pt loading amount	T (°C)	con	sel	Ref.
Pt/SM-Mg	1 : 0.03	0.5 wt%	280	94%	95%	This work
Pt/MgAPO-11-hydr	1 : 0.03	0.5 wt%	280	81%	78%	4
Pt/MgAPO-11-iono	1 : 0.03	0.5 wt%	280	93%	94%	54
Pt/SAPO-11	—	0.5 wt%	300	94%	81%	4

^a con: conversion, sel: selectivity, ref: reference.

2017YFB0306701), National Natural Science Foundation of China (No. 21406224) and Youth Innovation Promotion Association of CAS (No. 2017228). Besides, Ping Wang is extremely grateful for Ms Huimin Gong from Dalian Institute of Chemical Physics for taking the SEM photographs and her wise advice.

References

- 1 A. Corma and H. Garcia, *Chem. Rev.*, 2003, **103**, 4307–4365.
- 2 H. Deldari, *Appl. Catal., A*, 2005, **293**, 1–10.
- 3 S. J. Miller, H. S. Lacheen and C. Y. Chen, *Ind. Eng. Chem. Res.*, 2016, **55**, 6760–6767.
- 4 X. M. Yang, Z. S. Xu, Z. J. Tian, H. J. Ma, Y. P. Xu, W. Qu and L. W. Lin, *Catal. Lett.*, 2006, **109**, 139–145.
- 5 M. Hartmann and S. P. Elangovan, *Chem. Eng. Technol.*, 2003, **26**, 1232–1235.
- 6 C. X. Wang, Z. J. Tian, L. Wang, R. S. Xu, Q. H. Liu, W. Qu, H. J. Ma and B. C. Wang, *ChemSusChem*, 2012, **5**, 1974–1983.
- 7 T. Blasco, A. Chica, A. Corma, W. J. Murphy, J. Agundez-Rodriguez and J. Perez-Pariente, *J. Catal.*, 2006, **242**, 153–161.
- 8 G. Lv, C. Wang, P. Wang, L. Sun, H. Liu, W. Qu, D. Wang, H. Ma and Z. Tian, *ChemCatChem*, 2019, **11**, 1431–1436.
- 9 J. A. Martens, G. Vanbutsele, P. A. Jacobs, J. Denayer, R. Ocakoglu, G. Baron, J. A. M. Arroyo, J. Thybaut and G. B. Marin, *Catal. Today*, 2001, **65**, 111–116.
- 10 S. Mitchell, N. L. Michels and J. Perez-Ramirez, *Chem. Soc. Rev.*, 2013, **42**, 6094–6112.
- 11 L. Lakiss, J.-P. Gilson, V. Valtchev, S. Mintova, A. Vicente, A. Vimont, R. Bedard, S. Abdo and J. Bricker, *Microporous Mesoporous Mater.*, 2020, **299**, DOI: 10.1016/j.micromeso.2020.110114.
- 12 A. Martin, H. Berndt, U. Lohse and U. Wolf, *J. Chem. Soc., Faraday Trans.*, 1993, **89**, 1277–1282.
- 13 F. Dorado, R. Romero and P. Canizares, *Appl. Catal., A*, 2002, **236**, 235–243.
- 14 N. L. Michels, S. Mitchell and J. Perez Ramirez, *ACS Catal.*, 2014, **4**, 2409–2417.
- 15 A. Sachse, A. Galarneau, F. Fajula, F. Di Renzo, P. Creux and B. Coq, *Microporous Mesoporous Mater.*, 2011, **140**, 58–68.
- 16 J. Bauer, R. Herrmann, W. Mittelbach and W. Schwieger, *Int. J. Energy Res.*, 2009, **33**, 1233–1249.
- 17 K. Schumann, B. Unger, A. Brandt and F. Scheffler, *Microporous Mesoporous Mater.*, 2012, **154**, 119–123.
- 18 K. Schumann, B. Unger, A. Brandt, G. Fischer, H. Richter and J. Jänchen, *Chem. Ing. Tech.*, 2014, **86**, 106–111.
- 19 K. Shams and S. J. Mirmohammadi, *Microporous Mesoporous Mater.*, 2007, **106**, 268–277.
- 20 J. Zhou, J. W. Teng, L. P. Ren, Y. D. Wang, Z. C. Liu, W. Liu, W. M. Yang and Z. K. Xie, *J. Catal.*, 2016, **340**, 166–176.
- 21 W. Zhang, S. Gao, S. Xie, H. Liu, X. Zhu, Y. Shang, S. Liu, L. Xu and Y. Zhang, *Chin. J. Catal.*, 2017, **38**, 168–175.
- 22 D. J. Wang, Z. N. Liu, H. Wang, Z. K. Xie and Y. Tang, *Microporous Mesoporous Mater.*, 2010, **132**, 428–434.
- 23 K. T. Jung and Y. G. Shul, *Microporous Mesoporous Mater.*, 1998, **21**, 281–288.
- 24 P. Wang, C. X. Wang, G. Lv, P. Li, G. J. Hou, W. Qu, H. J. Ma, D. E. Wang and Z. J. Tian, *Microporous Mesoporous Mater.*, 2020, **295**, DOI: 10.1016/j.micromeso.2019.109962.
- 25 M. Hartmann and L. Kevan, *Res. Chem. Intermed.*, 2002, **28**, 625–695.
- 26 A. Prabhu and M. Palanichamy, *Microporous Mesoporous Mater.*, 2013, **168**, 126–131.
- 27 A. Prabhu, L. Kumaresan, M. Palanichamy and V. Murugesan, *Appl. Catal., A*, 2009, **360**, 59–65.
- 28 A. Prabhu, A. Al Shoaibi, C. Srinivasakannan, M. Palanichamy and V. Murugesan, *J. Rare Earths*, 2013, **31**, 477–484.
- 29 A. Prabhu, A. Al Shoaibi and C. Srinivasakannan, *Appl. Catal., A*, 2013, **466**, 137–141.
- 30 L. Kumaresan, A. Prabhu, M. Palanichamy and V. Murugesan, *J. Taiwan Inst. Chem. Eng.*, 2010, **41**, 670–675.
- 31 A. Prabhu, L. Kumaresan, M. Palanichamy and V. Murugesan, *Appl. Catal., A*, 2010, **374**, 11–17.
- 32 A. Prabhu, S. Balachandran and P. Muthusamy, *Mater. Sci. Energy Technol.*, 2021, **4**, 128–135.
- 33 M. Pramanik, M. Nandi, H. Uyama and A. Bhaumik, *Green Chem.*, 2012, **14**, 2273–2281.
- 34 A. K. Patra, A. Dutta, M. Pramanik, M. Nandi, H. Uyama and A. Bhaumik, *ChemCatChem*, 2014, **6**, 220–229.
- 35 M. Hochtl, A. Jentys and H. Vinek, *Microporous Mesoporous Mater.*, 1999, **31**, 271–285.
- 36 M. Guisnet, P. Ayrault, C. Coutanceau, M. F. Alvarez and J. Datka, *J. Chem. Soc., Faraday Trans.*, 1997, **93**, 1661–1665.
- 37 S. T. Wilson, B. M. Lok, C. A. Messina, T. R. Cannan and E. M. Flanigen, *J. Am. Chem. Soc.*, 1982, **104**, 1146–1147.
- 38 Y. N. Huang, D. Machado and C. W. Kirby, *J. Phys. Chem. B*, 2004, **108**, 1855–1865.
- 39 R. S. Xu, W. P. Zhang, J. Xu, Z. J. Tian, F. Deng, X. W. Han and X. H. Bao, *J. Phys. Chem. C*, 2013, **117**, 5848–5854.
- 40 C. S. Blackwell and R. L. Patton, *J. Phys. Chem.*, 1988, **92**, 3965–3970.
- 41 J. G. Longstaffe, B. H. Chen and Y. N. Huang, *Microporous Mesoporous Mater.*, 2007, **98**, 21–28.
- 42 Z. C. Zhao, W. P. Zhang, R. S. Xu, X. W. Han, Z. J. Tian and X. H. Bao, *Dalton Trans.*, 2012, **41**, 990–994.
- 43 Y. N. Huang, B. A. Demko and C. W. Kirby, *Chem. Mater.*, 2003, **15**, 2437–2444.
- 44 Y. N. Huang, R. Richer and C. W. Kirby, *J. Phys. Chem. B*, 2003, **107**, 1326–1337.
- 45 D. B. Akolekar and R. F. Howe, *J. Chem. Soc., Faraday Trans.*, 1997, **93**, 3263–3268.
- 46 R. D. Gougeon, E. B. Brouwer, P. R. Bodart, L. Delmotte, C. Marichal, J. M. Chezeau and R. K. Harris, *J. Phys. Chem. B*, 2001, **105**, 12249–12256.
- 47 Z. M. Yan, B. H. Chen and Y. Huang, *Solid State Nucl. Magn. Reson.*, 2009, **35**, 49–60.
- 48 Y. Wei, Z. J. Tian, H. Gies, R. S. Xu, H. J. Ma, R. Y. Pei, W. P. Zhang, Y. P. Xu, L. Wang, K. D. Li, B. C. Wang, G. D. Wen and L. W. Lin, *Angew. Chem., Int. Ed.*, 2010, **49**, 5367–5370.
- 49 Y. Wei, Z. J. Tian, H. Gies, R. S. Xu, H. J. Ma, R. Y. Pei, W. P. Zhang, Y. P. Xu, L. Wang, K. D. Li, B. C. Wang,

- G. D. Wen and L. W. Lin, *Angew. Chem., Int. Ed.*, 2010, **49**, 5367–5370.
- 50 C. Blackwell and R. Patton, *J. Phys. Chem.*, 1984, **88**, 6135–6139.
- 51 F. Deng, Y. Yue, T. C. Xiao, Y. U. Du, C. H. Ye, L. D. An and H. L. Wang, *J. Phys. Chem.*, 1995, **99**, 6029–6035.
- 52 S. Prasad and J. F. Haw, *Chem. Mater.*, 1996, **8**, 861–864.
- 53 L. Wang, Y. P. Xu, B. C. Wang, S. J. Wang, J. Y. Yu, Z. J. Tian and L. W. Lin, *Chem. - Eur. J.*, 2008, **14**, 10551–10555.
- 54 J. Perez-Pariente and M. Sanchez-Sanchez, in *Structure and Reactivity of Metals in Zeolite Materials*, ed. J. P. Pariente and M. Sanchez-Sanchez, 2018, vol. 178, pp. V–VII.
- 55 M. G. O'Brien, A. M. Beale, C. R. A. Catlow and B. M. Weckhuysen, *J. Am. Chem. Soc.*, 2006, **128**, 11744–11745.
- 56 M. Sanchez-Sanchez, A. A. Romero, I. Pinilla-Herrero and E. Sastre, *Catal. Today*, 2017, **296**, 239–246.
- 57 D. W. Lewis, C. R. A. Catlow and J. M. Thomas, *Chem. Mater.*, 1996, **8**, 1112–1118.
- 58 S. A. Katsyuba, E. E. Zvereva, A. Vidis and P. J. Dyson, *J. Phys. Chem. A*, 2007, **111**, 352–370.
- 59 W. Kong, W. Dai, N. Li, N. Guan and S. Xiang, *J. Mol. Catal. A: Chem.*, 2009, **308**, 127–133.
- 60 R. J. Kalbasi and E. Izadi, *C. R. Chim.*, 2011, **14**, 1002–1013.
- 61 B. Zhang, J. Xu, F. T. Fan, Q. Guo, X. Q. Tong, W. F. Yan, J. H. Yu, F. Deng, C. Li and R. R. Xu, *Microporous Mesoporous Mater.*, 2012, **147**, 212–221.
- 62 F. T. Fan, Z. C. Feng, K. J. Sun, M. L. Guo, Q. Guo, Y. Song, W. X. Li and C. Li, *Angew. Chem., Int. Ed.*, 2009, **48**, 8743–8747.
- 63 S. Oliver, A. Kuperman, A. Lough and G. A. Ozin, *Chem. Mater.*, 1996, **8**, 2391–2398.
- 64 Y. C. Xiao, N. Sheng, Y. Y. Chu, Y. Q. Wang, Q. M. Wu, X. L. Liu, F. Deng, X. J. Meng and Z. C. Feng, *Microporous Mesoporous Mater.*, 2017, **237**, 201–209.
- 65 A. J. Holmes, S. J. Kirkby, G. A. Ozin and D. Young, *J. Phys. Chem.*, 1994, **98**, 4677–4682.
- 66 B. H. Chen and Y. Huang, *J. Phys. Chem. C*, 2007, **111**, 15236–15243.
- 67 S. Inagaki, K. Nakatsuyama, E. Kikuchi and M. Matsukata, *Stud. Surf. Sci. Catal.*, 2005, **158**, 343–350.
- 68 S. Inagaki, K. Nakatsuyama, E. Kikuchi and M. Matsukata, *Bull. Chem. Soc. Jpn.*, 2010, **83**, 69–74.
- 69 H. J. Ma, Z. J. Tian, R. S. Xu, B. C. Wang, Y. Wei, L. Wang, Y. P. Xu, W. P. Zhang and L. W. Lin, *J. Am. Chem. Soc.*, 2008, **130**, 8120–8121.