



## Intermetallic Semiconductors

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# Interplay of Atomic Interactions in the Intermetallic Semiconductor Be<sub>5</sub>Pt

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Abstract: Semiconducting substances form one of the most important families of functional materials. However, semiconductors containing only metals are very rare. The chemical mechanisms behind their ground-state properties are only partially understood. Our investigations have rather unexpectedly revealed the semiconducting behaviour (band gap of 190 meV) for the intermetallic compound Be<sub>5</sub>Pt formed at a very low valence-electron count. Quantum-chemical analysis shows strong charge transfer from Be to Pt and reveals a threedimensional entity of vertex-condensed empty Be<sub>4</sub> tetrahedrons with multi-atomic cluster bonds interpenetrated by the framework of Pt-filled vertex-condensed Be<sub>4</sub> tetrahedrons with twoatomic polar Be-Pt bonds. The combination of strong Coulomb interactions with relativistic effects results in a band gap.

## Introduction

From the first PbS rectifier to the Si-based field-effect transistor and light-emitting diodes, semiconducting materials were crucial for the development of electronic devices, paving the way to the modern information age.<sup>[1,2]</sup> The continuous interest in new semiconducting materials is fuelled by the discovery of unexpected properties, which have led to emerging applications from thermoelectric materials with large Seebeck coefficients<sup>[3,4]</sup> to flexible electronics.<sup>[5]</sup> On the contrary, temperature-driven semiconductor-to-metal transitions attract the interest of basic research.<sup>[6]</sup> Semiconducting

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behaviour in crystalline solids, caused by the existence of an energy gap at the Fermi level, can be rooted in several mechanisms. From the chemical point of view, the occurrence of semiconducting behaviour among intermetallic compounds relies heavily on i) the combination of elements, ii) the adopted crystal structure, and iii) the valence-electron count. Since its discovery in the 1940s,<sup>[7]</sup> Mg<sub>2</sub>Sn represents this rather rare case. From the very beginning, its study has led to significant contributions to the scientific understanding of metals, alloys, and semiconductors.<sup>[8-11]</sup> The mechanism of gap formation is closely coupled to the type and spatial organisation of chemical bonding.

Most common semiconductors can be assigned to the group of so-called tetrahedral structures, where all constituting atoms are fourfold (tetrahedrally) coordinated. Notable examples are the elemental semiconductors Si and Ge, with a diamond-type crystal structure, and the thereof derived binary III-V and II-VI semiconductors GaAs, CdTe, ZnSe with sphalerite- or wurtzite-type structures (Supporting Information, Table S1). In these "electron-precise" compounds, an average count of four valence electrons per atom allows the formation of a three-dimensional network of fourfold bonded atoms connected by two-centre-two-electron covalent (polar) bonds.

Reduction of the average number of electrons below four may have different consequences, depending on the electronegativity difference between the components.<sup>[12,13]</sup> For large differences, intermetallic semiconductors follow the bonding scenario by Zintl.<sup>[14,15]</sup> Here, the cationic components donate their valence electrons to form two-centre-two-electron covalent bonds only in the anionic sublattices of fourfold bonded (including lone pairs) atoms extended in three (intermetallic clathrates at appropriate compositions<sup>[16-19]</sup>), pseudo-two (SrSi<sub>2</sub>), or zero (BaSi<sub>2</sub>) dimensions (Table S1). Ternary semiconductors with the CaAl<sub>2</sub>Si<sub>2</sub>-type structure follow the Zintl scheme, for example, CaMn<sub>2</sub>Bi<sub>2</sub> and CaCd<sub>2</sub>Sb<sub>2</sub>, although for the transition metals, only the s electrons are counted (Table S1). In all these substances, which typically contain Group 13-15 (p-block) elements in the anionic part, the ionic interaction between the electron donors and the anionic sublattice completes the chemical bonding picture and supports the gap formation.

The appearance of a band gap in intermetallic compounds of the main-group elements with XYZ composition and MgAgAs-type crystal structure (half-Heusler phases) can be described from the bonding point of view in three ways. Depending on the electronegativities of the components, it

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may either be ionic interaction between all three components as in LiMgN, or a Zintl-like bonding picture as in LiAlSi, or a combination of non-polar and polar characters of interactions still allowing the application of the Pearson 8–N rule.<sup>[20]</sup> In all cases, the cubic coordination of one of the components points to a more complex bonding picture. It can be described by a combination of polar three- and five-centre interactions becoming evident for the half-Heusler phases of transition metals,<sup>[21]</sup> for example, TiNiSn and TiCoSb (Table S1). As a general rule in half-Heusler compounds, the Fermi level falls into the band gap at counts of 18 electrons per formula unit, assuming a closed-shell configuration for the component in the cubic coordination.<sup>[22]</sup>

Another important group of semiconducting intermetallic compounds is formed by transition metals and Group 13-15 (p-block) elements: MoSi<sub>2</sub>, CrSi<sub>2</sub>, RuGa<sub>2</sub>, OsAl<sub>2</sub>, MnCoAl<sub>4</sub>, as well as the Nowotny-Chimney-Ladder (NCL) phases Ru<sub>2</sub>Si<sub>3</sub>, Fe<sub>2</sub>Ge<sub>3</sub>, and ReGaSi (see Table S1). Characteristic for these substances is a relatively small electronegativity difference between the components, suggesting the formation of polar bonds. Indeed, despite the p-block elements making up the majority of the composition, the direct interactions are mainly found between the transition-metal and the p-block atoms, for example, in ZrBi<sub>1.62</sub>.<sup>[23]</sup> For a valence-electron count of 14 per formula unit with one transition metal, these compounds show distinct stabilization and are often found to be semiconducting.<sup>[24]</sup> Some intermetallic compounds with low average electron count and small electronegativity differences follow neither the 18- nor the 14-electron count but are nevertheless (narrow-gap) semiconductors: FeGa<sub>3</sub>, RuGa<sub>3</sub>, OsGa<sub>3</sub>, ZnSb, CdSb, or FeSi (Table S1). Here, the appearance of the gap is interpreted as the formation of several threecentre interactions among the majority of two-centre ones.<sup>[25]</sup>

In the case of large electronegativity difference and reduced average electron count, the formation of a band gap was observed when electron transfer formally leads to closedshell configurations for both partners, as in the semiconductors Mg<sub>2</sub>Si, Mg<sub>2</sub>Ge, and Mg<sub>2</sub>Sn which obey the Zintl count (Table S1). This exceptional behaviour also appears for combinations of the most electronegative with the most electropositive metals, with examples given by CsAu, RbAu, and Cs<sub>2</sub>Pt (Table S1). The charge transfer and the resultant presence of auride ions Au<sup>-</sup> ([Xe]6s<sup>2</sup>4f<sup>14</sup>5d<sup>10</sup>) was confirmed by photoemission and Mössbauer experiments.<sup>[26,27]</sup> Up to now, such behaviour was only reported for intermetallic semiconductors containing alkaline metals. Recently, strong charge transfer was observed in the Be-rich complex intermetallic compound Be<sub>21</sub>Pt<sub>5</sub>. Its band structure already shows a small pseudo-gap slightly above the Fermi level.<sup>[28]</sup> This was the motivation to search for semiconducting materials in the beryllium-rich part of the Be-Pt system. The chemical challenge here is that the formed compounds challenge conventional valence-counting rules due to the very low number of valence electrons per atom. The present work describes the interplay of two-centre polar and multicentre covalent bonding which gives rise to the experimentally observed semiconducting ground state for the intermetallic compound Be<sub>5</sub>Pt (first observed in ref. [29]) only when relativistic effects (already at the level of scalar-relativistic treatment) are taken into account.

## **Results and Discussion**

The compound Be<sub>5</sub>Pt adopts a crystal structure of the Be<sub>5</sub>Au type (space group  $F\bar{4}3m$ , a = 5.9710(3) Å) as observed from the X-ray single-crystal diffraction data (Figure 1, left, and Tables S1 and S2). The X-ray powder-diffraction pattern of the prepared product (Figure S1) confirms its single-phase



**Figure 1.** Crystal structures of  $Be_5Pt$  (left) and  $MgCu_2$  (right). The Be2/ Cu atoms (small grey spheres) form a network of vertex-condensed tetrahedra (yellow). The Be1 atoms in  $Be_5Pt$  replace half of the Mg positions in  $MgCu_2$ , the second half (at the origin) is occupied by Pt atoms. Coordination polyhedra of Pt (large dark-blue spheres) and Be1 (small grey spheres) are shown in blue and red, respectively (left).

character. The lattice parameters of the Be<sub>5</sub>Pt phase in the samples with the off-stoichiometric compositions are equal within three estimated standard deviations, which indicates a negligible homogeneity range.

The structural arrangement of Be<sub>5</sub>Au can be derived from the cubic one of the Laves phase MgCu<sub>2</sub> (space group  $Fd\bar{3}m$ , Figure 1, right) by ordered occupation of the Mg positions (Wyckoff site 8a in  $Fd\bar{3}m$ ) by Pt (site 4a in  $F\bar{4}3m$ ) and Be1 (site 4c in  $F\bar{4}3m$ ), respectively. From the structural point of view, Be<sub>5</sub>Pt does not reveal similarity to any specific family of semiconducting intermetallic compounds mentioned in the introduction. On the one hand, it does not show tetrahedral heteroatomic coordination for all atoms like in the classical III-V and II-VI semiconductors (while Pt and Be1 are fourfold coordinated by heteroatoms, Be2 has only 3 Pt atoms at the closest distance). On the other hand, it also does not contain isolated, zero-, one-, or two-dimensional polyanions, characteristic for the Zintl phases or compounds like Mg<sub>2</sub>Si, CsAu, or Cs<sub>2</sub>Pt. Analysis of the interatomic distances suggests the formation of one substructure built of Pt tetrahedrally coordinated by Be1 at a distance of 2.59 Å (atomic radii of Pt and Be are 1.38 Å and 1.133 Å, respectively). Another substructure is formed by homo-coordinated Be2 atoms, forming vertex-condensed tetrahedra with relatively short distances of 2.09 Å (3  $\times$ ) and 2.13 Å (3  $\times$ ) with next-nearest Pt atoms at 2.48 Å (in analogy to the Laves phases). Usually, the presence of such short distances in both substructures should be a fingerprint of covalent two-centre interactions. The separation between the substructures-the Be1-Be2 distance of 2.48 Å-is larger than the Be2-Be2 distance and the distance of 2.08 Å in elemental beryllium, signalling a diversification of chemical interactions in Be<sub>5</sub>Pt. How justified the negligence or consideration of Be1–Be2 contacts is, will be discussed further in the context of chemical-bonding analysis. The combination of high CN values (9 and 16) together with the very low number of valence electrons per atom signals diversification of chemical interactions in Be<sub>5</sub>Pt on the one hand and makes it an unlikely candidate for a semiconductor on the other hand.

The calculated electronic density of states (DOS) of Be<sub>5</sub>Pt reveals, quite unexpectedly, a narrow band gap of  $E_g = 85 \text{ meV}$  (Figure 2) pointing towards semiconducting behaviour. The Pt 5d states dominate the valence band of Be<sub>5</sub>Pt



**Figure 2.** Calculated electronic density of states (DOS, fully relativistic treatment) for Be<sub>5</sub>Pt: total DOS, partial atomic contributions, and sum of the partial DOS weighted by photoionization cross-sections and broadened for accounting for life-time effects (DOS<sub>w</sub>) are shown as coloured lines; the inset shows details of the DOS around the Fermi level.

between -7 and -4 eV. A special feature is the strong structuring of the Pt 5d contributions with several local maxima. The DOS within 3 eV below the Fermi level  $E_F$  is dominated by Be 2p states.

In contrast to the calculation results, the electrical resistivity measured on a bulk sample of Be<sub>5</sub>Pt shows a metal-like temperature dependence (Figure 3a). However, the microstructure of the sample (Figure 3b) reveals small inclusions of elemental beryllium on the grain boundaries and within the grains, likely due to the peritectic formation of Be<sub>5</sub>Pt and existence of the Be-Be<sub>5</sub>Pt eutectic. Obviously, these metallic inclusions distort the results of the bulk measurement. In order to obtain information about the intrinsic conductivity, a microscale specimen was cut off from the single-phase grain of the bulk sample with the focusedion-beam technique<sup>[30]</sup> and provided with contacts to make a microdevice (Figure 3c). The so-measured electrical resistivity of Be<sub>5</sub>Pt decreases monotonically with increasing temperature, corresponding to a semiconducting behaviour inferred from the band-structure calculations (Figure 3d). The size of the band gap was estimated applying the Arrhenius relation  $\ln \rho(T) = \rho_0 \exp(-E_g/2k_BT)$ . A linear fit of  $\ln \rho$  vs. 1/T in the temperature range from 335 K to 400 K (inset in Figure 3D) yields a value of  $E_{\sigma} = 190 \text{ meV}$  for the energy gap. This is in a good agreement with results of the band-structure calculations, particularly considering the typ-



**Figure 3.** Electrical resistivity of Be<sub>5</sub>Pt: a) temperature dependence of electrical resistivity for the bulk sample; b) microstructure of the bulk sample showing inclusions of elemental beryllium; c) shape of the microscale device used for measurement (V and I are contacts for voltage and current, respectively); d) temperature dependence of electrical resistivity for the microscale samples; the inset shows an Arrhenius plot of  $\ln \rho$  vs. 1/T with a linear fit (black line).

ical underestimation of the band gap within the local-density approximation.

To substantiate the reasons behind the appearance of the band gap from a chemical point of view, analysis of chemical bonding was performed by applying quantum-chemical techniques in position space. The calculated electron density was evaluated according to the quantum theory of atoms in molecules (QTAIM<sup>[31]</sup>). In contrast to many other intermetallic compounds, the electron density in Be<sub>5</sub>Pt reveals two types of maxima. The "usual" maxima are located at the nuclei of the Pt and Be atoms, giving rise to the atomic basins (Figure 4a). The so-called non-nuclear maxima (NNM) are located within the Be2 tetrahedra (Figure S2). Appearance of NNM was also observed for elemental lithium,[32] beryllium,<sup>[33]</sup> and some of their compounds. They were considered not as an exception, but as a normal evolution step of the chemical-bonding picture in homonuclear groups with decreasing interatomic distances.<sup>[34]</sup> Fortunately for the further



**Figure 4.** Chemical bonding in Be<sub>5</sub>Pt: a) shape of the atomic QTAIM basins and effective charges; b) bonding basins for the 2at Be1–Pt (red), multi-atomic (10at) Be2<sub>4</sub>Pt<sub>6</sub> (orange), and 8at Be2<sub>4</sub>Pt<sub>4</sub> (light pink) interactions with their populations; lpc is the polar character of the 2at bond; c) general bonding picture in Be<sub>5</sub>Pt with the interpenetrating networks of multi-atomic (Be2–Pt, green-orange-pink) and polar two-atomic (Be1–Pt, red) interactions.

analysis, the basins of the NNM in Be<sub>5</sub>Pt have common surfaces mainly with those of Be2 atoms (Figure S2), that is, the Be2 atoms make the largest contribution to the population of the NNM basins. Thus, for the charge evaluation, the populations of the NNM basins were considered together with basins of Be2. Integration of the electron density within the QTAIM basins yields their electronic populations and after subtraction of the atomic numbers, the effective charges are obtained. These amount to -4.00 for Pt and +1.36 and +0.66 for Be1 and Be2 (Figure 4a), respectively. This confirms the charge transfer from Be to Pt and the difference in the effective charge between Be species as identified by HAXPES (see below).

Further details of atomic interactions in Be<sub>5</sub>Pt were obtained from the topological analysis of the electron localisability indicator (ELI) in its ELI-D representation  $(Y_D)$ .<sup>[35,36]</sup> ELI-D has recently developed into a powerful quantumchemical tool for the analysis of chemical bonding, particularly for intermetallic compounds.<sup>[20,21,36,37]</sup> The ELI-D distribution reveals three types of bonding interactions in Be<sub>5</sub>Pt, visualised by three different types of attractors (ELI-D maxima; Figures 4b and S3). Two sets of ELI-D attractors are located at the centres of the tetrahedra formed by Be2. The basins of attractors, centred at  $\binom{3}{4}, \frac{3}{4}, \frac{3}{4}$  and  $\binom{1}{2}, \frac{1}{2}, \frac{1}{2}$ , have common surfaces with 8 and 14 atomic-core basins. Having populations of 5.45 and 5.00 electrons, respectively, they represent multi-atomic cluster bonds resulting from the low valence-electron concentration, similar to the ones discovered recently in Be<sub>21</sub>Pt<sub>5</sub>. From the intersection analysis, the contribution of Be1 to the cluster bond pinned at  $\binom{1}{2}, \frac{1}{2}$ ,  $\frac{1}{2}$  is quite small (0.08 electrons, Table S4), so this cluster bond is considered as ten-atomic (10a) instead of the 14atomic one, as it was found for  $\mathrm{Be}_{21}\mathrm{Pt}_5.^{[28]}$  The basin of the cluster bond at  $\binom{3}{4}, \frac{3}{4}, \frac{3}{4}$  does not reveal any contributions from the six neighbouring Be1 atoms, thus the interaction is an eight-atomic (8a) one. The cluster bonds in  $Be_5Pt$  interlink the  $(Be2)_4$  tetrahedrons with anionic Pt species.

A second type of interaction is visualised by the ELI-D attractors located on the bond line between Pt and Be1 (basin population of 0.65 e<sup>-</sup>, Figure 4b and Table S4). Their basins have only contributions from Pt (0.51 e<sup>-</sup>) and Be1 (0.12 e<sup>-</sup>) with negligible amount of  $0.02 e^-$  from six Be2, so it is a twoatomic (2a) interaction. The polar character lpc of 0.61 for this bond qualifies it as a strongly polar one (lpc=0 represents a non-polar bond, lpc=1 reflects a fully polar (closed-shell) interaction. Icads to filling and energy lowering of the Pt 5d states<sup>[26,27]</sup> (see the findings of the HAXPES experiments below). These polar 2a-Be1–Pt bonds stretch the substructure with tetrahedrally bonded platinum and beryl-lium atoms.

Interestingly, there are no special topological features in the ELI-D distribution between Be1 and Be2 in  $Be_5Pt$ (Figure S4). Such behaviour is characteristic of a non-bonding interaction caused by the repulsion of the positively charged beryllium species due to the strong bond polarity of Be–Pt interactions.

Aside from the charge transfer, relativistic effects were suggested to support the opening of the gap in the DOS for polar intermetallic compounds.<sup>[38]</sup> In order to investigate this issue, fully, scalar-, and non-relativistic calculations were performed for Be<sub>5</sub>Pt, Be<sub>5</sub>Pd, and hypothetical Be<sub>5</sub>Ni. However, only the relativistic calculation for Be<sub>5</sub>Pt yields the narrow band gap (Figures 2 and S7). Accounting for relativistic effects increases the charge transfer (Figure S5), increases the population of the 2a-Be1-Pt interaction, and reduces the populations of both multi-centre cluster bonds (Figure S6, Table S4), that is, the bonding appears to be more covalent in the relativistic approximation which leads to band narrowing and an opening of the gap. This effect is much smaller for Be<sub>5</sub>Pd and practically disappears for the hypothetical Be<sub>5</sub>Ni. The most striking effect of relativity is on the interactions between Be1 and Be2. In the semiconducting Be<sub>5</sub>Pt, the Be1 contributions to the cluster bonds and the Be2 contributions to the 2a-Be1-Pt bonds are negligible. When going from Be<sub>5</sub>Pt to Be<sub>5</sub>Ni and from relativistic to non-relativistic treatments, the Be1 participation in the cluster bonds increases (Table S4), implying more attractive interactions between Be1 and Be2 species. Further confirmation was provided by the presence of semiconducting behaviour and increase of the band gap up to 0.44 eV in Be<sub>5</sub>MgPta hypothetical composition with a larger component-electronegativity difference than in Be<sub>5</sub>Pt (Figure S8). In contrast, the electronic structure of the prototype Be<sub>5</sub>Au is typical for a metal (Figure S10).

To shed more light on the reasons for the appearance of semiconducting behaviour in Be<sub>3</sub>Pt, the electronic structure of the valence-band and core levels was investigated by hard X-ray photoelectron spectroscopy (HAXPES). The valence-band spectrum of Be<sub>3</sub>Pt (Figure 5 a) is in good agreement with the calculated density of states weighted by photoionization cross-sections DOS<sub>w</sub> (Figure 2). The localised Pt 5d levels are shifted well below  $E_F$  compared to Pt metal. Around the





**Figure 5.** Hard-X-ray photoelectron spectra of Be<sub>5</sub>Pt: a) valence band (orange) in comparison with Pt metal (blue), the inset shows a detailed view around  $E_F$ ; b) Pt 4f core levels (orange) in comparison with Pt metal (blue); c) polarization-dependent Be 1s core-level spectra of the Be1 and Be2 positions recorded in parallel (orange) and perpendicular (pink) detector geometry.

Fermi level, the HAXPES spectrum shows a strongly reduced intensity and a clear shift of the onset compared to Pt metal, confirming the absence of occupied states (insets in Figures 5a and 2). The Pt 4f core levels in Be<sub>5</sub>Pt are shifted to higher binding energies  $E_{\text{bind}}$  by +2.12 eV compared to Pt metal (Figure 5b). Concomitantly, the peak shape changes from an asymmetric one in Pt metal to a symmetric one in Be<sub>5</sub>Pt, in line with a transition from metallic to semiconducting behaviour. From electronegativity (EN) arguments (on the Pauling scale: EN(Pt) = 2.2,  $EN(Be) = 1.57^{[39]}$ ) and the quantum-chemical calculations (see above), a charge transfer from Be to Pt is expected. The more complete filling of the Pt 5d valence shell enhances the shielding of the core hole on Pt and reduces relaxation from the valence band. In metals, this effect is of a similar magnitude to the chemical shift, hindering straightforward deduction of charge transfer from core-level shifts.<sup>[40]</sup> Additionally, the filling of Pt 5d states shifts the core levels to higher binding energies. Thus, this does not equate to a chemical shift due to a positive charge on Pt but supports the notion of filling of the Pt 5d states, analogous to the effect observed in Ga-Pd intermetallic  $\mathsf{compounds}^{[41]}$  In the HAXPES spectra of the Be1s core level, unexpectedly, two lines were observed which are separated by  $\Delta E_{\text{bind}} = 1.68 \text{ eV}$  (Figure 5c). Both lines are shifted to higher binding energies, compared to Be metal  $(E_{\text{bind}} = 111.3 \text{ eV}^{[42]})$ . The s-character of both lines was confirmed by the line-intensity variation between spectra recorded with the electron analyser in parallel and perpendicular orientation relative to the polarization plane of the incident beam (for details, see the Materials and Methods section in the Supporting Information). The peak-area ratio of the two lines at 113.86 eV and 112.18 eV is approximately 1:4.3. This corresponds closely to the atomic ratio of the Be1 (fourfold site) and Be2 (sixteenfold site) in the crystal structure of Be<sub>5</sub>Pt. Electronic structure calculations reproduced the separation of the Be1 and Be2 core levels, yielding a  $\Delta E_{\text{bind}}$  of 1.55 eV (well in agreement with the experimental value above). This is strong evidence for a vastly different atomic environment and chemical bonding of Be1 and Be2 (see the crystal structure and chemical-bonding analysis above). Assuming that final-state effects are comparable for both sites, the higher binding energy on Be1 may be interpreted as a more positive effective charge on Be1 than on Be2 (see the QTAIM results above). All in all, the good agreement of the experimental valence-band spectrum of Be<sub>5</sub>Pt (Figure 5a) with the calculated density of states weighted by photoionization cross-sections DOS<sub>w</sub> (Figure 2) as well as the correct explanation of the Pt and Be core levels justifies the reliability of the calculation results and, indirectly, confirms the presence of the negatively charged platinum species in the structure as obtained from the analysis of chemical bonding above.

## Conclusion

The binary compound  $Be_3Pt$  is a rare representative of semiconductors that are formed by metallic elements only. The appearance of the gap in the band structure is a consequence of the relativistic effects and the unique chemical-bonding situation in the compound characterised by polar two-atomic Pt–Be1 and multi-atomic Pt–Be2 bonds. In fact, the combination of charge transfer caused by the electronegativity difference between the components and relativistic effects controls the semiconducting behaviour.

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## Conflict of interest

The authors declare no conflict of interest.

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