



Article A Theoretical Study of Fe Adsorbed on Pure and Nonmetal (N, F, P, S, Cl)-Doped Ti₃C₂O₂ for Electrocatalytic Nitrogen Reduction

Heng Luo^{1,2}, Xiaoxu Wang³, Chubin Wan^{1,*}, Lu Xie², Minhui Song² and Ping Qian^{1,2}

- ¹ Department of Physics, University of Science and Technology Beijing, Beijing 100083, China; s20190798@xs.ustb.edu.cn (H.L.); qianping@ustb.edu.cn (P.Q.)
- ² Beijing Advanced Innovation Center for Materials Genome Engineering, University of Science and Technology Beijing, Beijing 100083, China; b20170555@xs.ustb.edu.cn (L.X.); s20191385@xs.ustb.edu.cn (M.S.)
- ³ DP Technology, Beijing 100083, China; wangxx@dp.tech
- * Correspondence: cbwan@ustb.edu.cn

Abstract: The possibility of using transition metal (TM)/MXene as a catalyst for the nitrogen reduction reaction (NRR) was studied by density functional theory, in which TM is an Fe atom, and MXene is pure $Ti_3C_2O_2$ or $Ti_3C_2O_{2-x}$ doped with N/F/P/S/Cl. The adsorption energy and Gibbs free energy were calculated to describe the limiting potentials of N₂ activation and reduction, respectively. N₂ activation was spontaneous, and the reduction potential-limiting step may be the hydrogenation of N₂ to *NNH and the desorption of *NH₃ to NH₃. The charge transfer of the adsorbed Fe atoms to N₂ molecules weakened the interaction of N \equiv N, which indicates that Fe/MXene is a potential catalytic material for the NRR. In particular, doping with nonmetals F and S reduced the limiting potential of the two potential-limiting steps in the reduction reaction, compared with the undoped pure structure. Thus, Fe/MXenes doped with these nonmetals are the best candidates among these structures.

Keywords: DFT; MXene; nitrogen reduction; electrocatalysis; Gibbs free energy

1. Introduction

Ammonia is a raw material for the production of various fertilizers and is a potential energy source that is easy to store and transport, environmentally friendly, and relatively safe. Ammonia synthesis is important in agricultural production and energy development. However, most ammonia synthesis still relies on the Hubble–Bosch method proposed in the 20th century, which requires harsh reaction conditions (400–600 °C and 20–40 MPa) [1–3]. This method consumes a large amount of energy and causes significant greenhouse gas emissions [4]. In addition, other negative effects, such as adverse effects on the equipment under high-temperature and high-pressure conditions, need to be considered. Therefore, the development of environmentally friendly and less energy-demanding methodologies for NH₃ synthesis is urgently needed. Electrocatalytic ammonia synthesis has attracted increasing attention owing to its high efficiency and environmental friendliness. The introduction of electrical energy has a remarkable influence on N₂ activation and changes the reaction pathways [5], which is beneficial for the development of new stable and efficient catalysts.

New catalysts can be developed from unique structures, such as core–shell Ni–Au nanoparticles for CO₂ hydrogenation [6], or from new materials. The excellent physical, electronic, and chemical properties of two-dimensional (2D) materials have attracted extensive scientific research [7–13]. In addition, 2D materials, such as molybdenum disulfide, graphene, and metal–organic frameworks (MOFs) [14–16], have emerged as potential candidates for electrochemical nitrogen reduction reactions (NRRs). Notably, MXene, a new member of the 2D material family that joined in 2011 [17], has developed rapidly in the past nine years [17–19]. The general formula of MXene is $M_{n+1}X_nT_x$, where M represents early transition metals (TMs), X represents carbon or nitrogen, T_x represents the surface



Citation: Luo, H.; Wang, X.; Wan, C.; Xie, L.; Song, M.; Qian, P. A Theoretical Study of Fe Adsorbed on Pure and Nonmetal (N, F, P, S, Cl)-Doped Ti₃C₂O₂ for Electrocatalytic Nitrogen Reduction. *Nanomaterials* **2022**, *12*, 1081. https://doi.org/ 10.3390/nano12071081

Academic Editor: Frederik Tielens

Received: 3 March 2022 Accepted: 23 March 2022 Published: 25 March 2022

Publisher's Note: MDPI stays neutral with regard to jurisdictional claims in published maps and institutional affiliations.



Copyright: © 2022 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (https:// creativecommons.org/licenses/by/ 4.0/). functional groups O, OH, or F, and n = 1, 2, 3. MXenes are synthesized by the chemical etching of A layers in the MAX ($M_{n+1}AX_n$) phase. Although a variety of 2D MXenes have been theoretically predicted [20], only a few have been synthesized. MXenes are applied in a wide range of fields, including electrocatalysis [21], hydrogen storage [22,23], lithium-ion batteries [24,25], and supercapacitors [26]. MXene is a potential candidate for electrochemical NRRs (e-NRRs) because of its large specific surface area, adjustable structure, and excellent stability [27–29].

MXene-based electrocatalysts for the e-NRR can be divided into two categories: pure MXene and MXene-based hybrid electrocatalysts [30]. Pure MXene is a potential candidate for the e-NRR. For example, Azofra et al. [31] found that M_3C_2 exhibited good N_2 capture and activation behavior. However, bare-metal atoms on the surface of M_3C_2 are considered active sites [31,32], which tend to bind to functional groups such as oxygen groups; thus, the electrical conductivity is decreased, and the active sites are inactivated. Pure MXene still faces challenges as a catalyst for the e-NRR; therefore, MXene hybrids have been designed. Li et al. [33] loaded nanosized Au particles onto Ti_3C_2 nanosheets (Au/Ti_3C_2) for the e-NRR. Their research indicated that the hybrid is conducive to N₂ chemisorption and decreases the activation energy barrier. Au/Ti₃C₂ shows excellent catalytic performance. MnO₂-decorated Ti₃C₂T_x (MnO₂-Ti₃C₂T_x) has also been studied as an efficient electrocatalyst for ammonia synthesis under environmental conditions [34]. MnO₂ and $Ti_3C_2T_x$ synergistically promote electrocatalytic activity to achieve superior catalytic activity. In addition, single-atom catalysts (SACs) have been widely studied because of their low cost, superior performance, and full use of metal atoms. Gao et al. [5] studied the reaction pathways and overpotentials of $Ti_3C_2O_2$ -supported TM (Fe, Co, Ru, Rh) SACs. These MXene hybrids, including noble metal-MXene, TM oxide-MXene, and MXene-based SACs, have effectively changed the catalytic performance, providing more possibilities for the screening of new efficient and stable catalysts.

In this study, a 2D MXene, Ti₃C₂O₂, was modified with nonmetals (N, F, P, S, and Cl) and adsorbed TM (Fe atom, Fe/Ti₃C₂O_{2-x}) to study the catalytic performance of the e-NRR. Gibbs free energy (ΔG) was used to analyze the reaction pathway and limit the potential of each catalyst, and the main potential-limiting steps of the reaction were determined as $*N_2 + H \rightarrow *NHH$ and $*NH_3 \rightarrow NH_3$.

2. Computational Methods

Density functional theory (DFT) calculations were performed using the Vienna ab initio simulation package v. 5.4.4. (University of Vienna, Vienna, Austria) [35,36]. The generalized gradient approximation with Perdew–Burke–Ernzerhof was used as an exchange-correlation function [37]. The projector-augmented wave method was adopted to describe the effect of the core electrons on the valence electron density [38]. The cut-off energy was set to 600 eV. The convergence criteria for the energy and force were 10^{-5} eV and 10^{-2} eV/Å, respectively. The thickness of the vacuum layer was more than 20 Å to avoid interactions in the *z*-direction, and the *x*-and *y*-directions were set as periodic boundary conditions. A 3 × 3 × 1 supercell was used for all the structures. For geometric optimization, the Brillouin zones were sampled with 4 × 4 × 1 Monkhorst–Pack meshes [39], and DFT-D3 was used to accurately describe Van der Waals interactions [40]. Charge transfer was computed by Bader charge population analysis [41,42] and the electron localization function (ELF) was analyzed using the VESTA code [43].

The substitution energies (ΔE_{sub}) of doping different nonmetallic elements (N/F/P/S/Cl) on the surface of Ti₃C₂O₂ can be expressed as

$$\Delta E_{\rm sub} = E_{\rm NM-Ti_3C_2O_{2-x}} - E_{\rm Ti_3C_2O_2} + E_{\rm O} - E_{\rm NM} \tag{1}$$

where E_O and E_{NM} represent the energies of a single O atom and nonmetallic elements (N, F, P, S, Cl), respectively, and were calculated using H₂ [44], H₂O [45], NH₃ [46], HF [47], H₃PO₄ [48], H₂S [49], and HCl [50] from the Open Quantum Materials Database (OQMD) [51,52].

The adsorption energy (ΔE_{ads}) of Fe anchored on NM-Ti₃C₂O_{2-x} (NM represents the surface nonmetals, O, N, F, P, S, and Cl) was calculated using the following formula:

$$\Delta E_{ads} = E_{Fe/NM-Ti_3C_2O_{2-x}} - E_{NM-Ti_3C_2O_{2-x}} - E_{Fe}$$
(2)

 ΔG was calculated as described by Nørskov et al. [53]. Under standard reaction conditions, the chemical potential of a proton and electron pair (μ [H⁺ + e⁻]) is equal to half that of gaseous hydrogen (μ [H₂]). ΔG was calculated using the following formula:

$$\Delta G = \Delta E_{\rm DFT} + \Delta ZPE - T\Delta S - neU + \Delta G_{\rm pH}$$
(3)

where ΔE is the potential energy change calculated by DFT, ΔZPE is the zero-point energy correction, and it is calculated by calculating the frequency of the adsorbed species. $T\Delta S$ is the entropy correction, which is usually available from some database, where T = 298 K; $\Delta G_{\rm pH}$ and *neU* are the contributions from the pH and electrode potential (*U*), respectively; *n* is the number of electrons transferred; *U* is the applied bias. $\Delta G_{\rm pH}$ is defined as

$$\Delta G_{\rm pH} = -k_{\rm B}T\ln\left[\mathrm{H}^+\right] = \mathrm{pH} \times K_{\rm B}T\ln10 \tag{4}$$

where $k_{\rm B}$ is Boltzmann's constant. For all the calculations, the pH was set to zero. The $\Delta E_{\rm ads}$ values of different adsorbates were calculated as follows:

$$\Delta E_{\rm ads} = E_{\rm cat-mol} - E_{\rm cat} - E_{\rm mol} \tag{5}$$

where $\Delta E_{\text{cat-mol}}$ is the energy of the entire adsorption structure, E_{cat} is the energy of the catalyst, and E_{mol} is the energy of the adsorbate molecules such as N₂ and N_xH_y.

3. Results and Discussion

3.1. Geometric Structure

Bare Ti₃C₂ is a hexagonal lattice with $P\bar{3}m1$ group symmetry, five atomic layers of Ti–C–Ti, two exposed Ti layers, and an experimental lattice constant of 3.057 Å [54]. After structural optimization, a = b = 3.020 Å, which was in good agreement with the experimental values. Bare MXenes are unstable under relevant NRR operating conditions [55], and they are always functionalized by electronegative functional groups [56], as they are chemically exfoliated from the bulk MAX phase by HF [17,57]. O-terminated Ti₃C₂ was used for further experiments. There are different possibilities for the adsorption of O on Ti₃C₂. According to previous studies [5], the most stable structure is O adsorbed at the hollow sites of the contralateral surface Ti atoms, as shown in Figure 1a,b. Nonmetallic elements (N/F/P/S/Cl) were used to modify the Ti₃C₂O₂ surface. ΔE_{sub} indicates the stability of a surface before and after doping with nonmetallic elements. The ΔE_{sub} values for N, F, P, S, and Cl were 1.79, -1.04, 0.81, -0.27, and -1.01 eV, respectively. The structure became more stable after doping with F, S, and Cl when $\Delta E_{sub} < 0$ and became more unstable after doping with doping with other nonmetallic elements.

Pure Ti₃C₂O₂ and Ti₃C₂O₂ modified with nonmetallic elements (Figure S1) were used to support single Fe atoms. Two different hollow sites (H1 and H2) and an O-top site on the surface were considered, as shown in Figure 1a. The O-top was unstable, and the E_{ads} values of Fe adsorbed on H1 and H2 are listed in Table 1. Except for the F-doped structures, the Fe atoms preferred to adsorb on the H1 site, as the E_{ads} was smaller. Notably, in the F-doped structure, the Fe atom was adsorbed on the next-nearest H1 site (Figure 1e). As shown in Table 1, the doping of N, F, P, and S facilitates the adsorption of Fe, while it is more difficult for Fe to adsorb on the Cl-doped structure. Figure 1c–h show the most stable adsorption positions for the different catalysts.

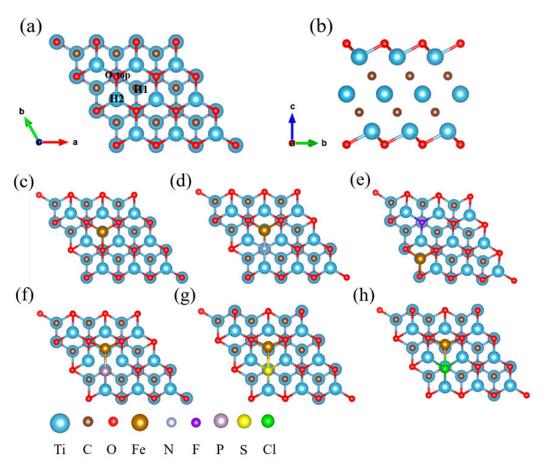


Figure 1. (a) Top view and different adsorption sites on $Ti_3C_2O_2$ and (b) side view of $Ti_3C_2O_2$. The most stable structure of Fe adsorbed on (c) $Ti_3C_2O_2$, (d) N-doped $Ti_3C_2O_2$, (e) F-doped $Ti_3C_2O_2$, (f) P-doped $Ti_3C_2O_2$, (g) S-doped $Ti_3C_2O_2$, and (h) Cl-doped $Ti_3C_2O_2$.

Table 1. Adsorption energies of Fe adsorbed on different sites and N2 adsorbed on different catalysts,the charge on N2, and the charge transferred after N2 adsorption.

Species	$E_{\rm ads}$ of Fe (eV)		E_{ads} of N ₂ (eV)	Charge Transferred on N ₂ (e)	
	H1	H2	L_{ads} of N_2 (eV)	charge manorefield off N ₂ (e)	
Fe/Ti ₃ C ₂ O ₂	-3.57	-3.30	-0.92	0.19	
$Fe/N-Ti_3C_2O_{2-x}$	-4.32	-3.90	-0.77	0.15	
$Fe/F-Ti_3C_2O_{2-x}$	-3.61	-3.60	-0.78	0.18	
$Fe/P-Ti_3C_2O_{2-x}$	-5.12	-4.68	-0.55	0.13	
$Fe/S-Ti_3C_2O_{2-x}$	-4.33	-4.02	-0.59	0.16	
$Fe/Cl-Ti_3C_2O_{2-x}$	-3.39	-3.11	-0.85	0.21	

3.2. N₂ Adsorption

Based on the Fe/NM-Ti₃C₂O_{2-x} structure, N₂ adsorption was calculated using E_{ads} . There are two different positions for N₂ adsorption, and advanced research has shown that N₂ adsorption is closer end to end than side to side [5]. Figure 2a–f show the most stable structure of N₂ adsorbed on different catalysts from end to end, and Figure 2g–l show the ELF of these structures. E_{ads} ranged from -0.55 eV to -0.92 eV, which indicates that the N₂ adsorption has strong spontaneity, and the absolute value of E_{ads} from small to large was in the order: Fe/P-Ti₃C₂O_{2-x} < Fe/S-Ti₃C₂O_{2-x} < Fe/N-Ti₃C₂O_{2-x} < Fe/F-Ti₃C₂O_{2-x} < Fe/F-Ti₃C₂O_{2-x} < Fe/Ti₃C₂O_{2-x}, Fe/F-Ti₃C₂O_{2-x}, Fe/F-Ti₃C₂O_{2-x}, Fe/F-Ti₃C₂O_{2-x}, and Fe/Cl-Ti₃C₂O_{2-x} are 1.128, 1.125, 1.129, 1.123, 1.126, and 1.130 Å, respectively. Compared with the N \equiv N bond length in the gas phase (1.11 Å), all of them became longer. The calculation of charge transfer is shown in Table 1. The results show that N₂ gains electrons in all these catalysts and the translated charges increase with an increase in the number of valence electrons from N to O or from P to S and Cl in the same period. However, doping with F did not obey this rule, which may be due to the special adsorption site of Fe. Fe was adsorbed on the first nearest H1 site and followed the trend from N to O and F. These findings were consistent with those of Wang et al. [58]. A strong positive correlation exists between the electron gains of N₂ and the change in bond length: N₂ on Fe/Cl-Ti₃C₂O_{2-x} gained the most electrons and had the largest increase in bond length relative to the gas phase, whereas N₂ on Fe/P-Ti₃C₂O_{2-x} gained the least electrons and had the smallest increment in bond length relative to the gas phase.

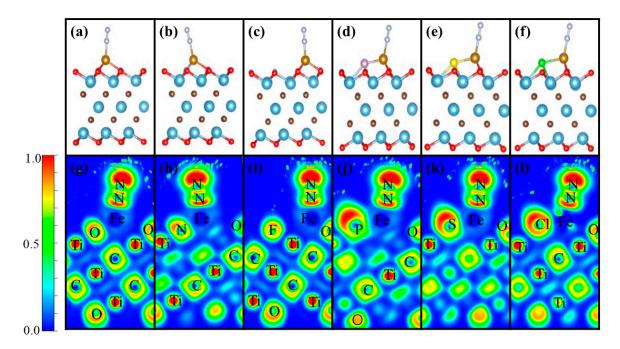


Figure 2. Most stable structures of N₂ adsorbed on (a) Fe/Ti₃C₂O₂, (b) Fe/N-Ti₃C₂O_{2-*x*}, (c) Fe/F-Ti₃C₂O_{2-*x*}, (d) Fe/P-Ti₃C₂O_{2-*x*}, (e) Fe/S-Ti₃C₂O_{2-*x*}, and (f) Fe/Cl-Ti₃C₂O_{2-*x*} and ELFs of N₂ adsorbed on (g) Fe/Ti₃C₂O₂, (h) Fe/N-Ti₃C₂O_{2-*x*}, (i) Fe/F-Ti₃C₂O_{2-*x*}, (j) Fe/P-Ti₃C₂O_{2-*x*}, (k) Fe/S-Ti₃C₂O_{2-*x*}, and (l) Fe/Cl-Ti₃C₂O_{2-*x*}.

The partial density of states of N₂ adsorbed on Fe/Ti₃C₂O₂ or Fe/NM-Ti₃C₂O_{2-*x*} (Figure 3) shows spin-up and spin-down of the d orbital of the Fe atom and the p orbital of the N atom. At the Fermi level, almost no spin-up was observed, whereas the spin-down was more obvious, and the d orbital of Fe effectively overlapped with the P orbital of N near the Fermi level. The electrons in the occupied d orbital of Fe/NM-Ti₃C₂O_{2-*x*} transferred to the antibonding orbitals of N₂, as shown in Table 1, and the adsorbed N₂ on different catalysts gained electrons from 0.13 e to 0.21 e, thus lowering the bond energy of N₂.

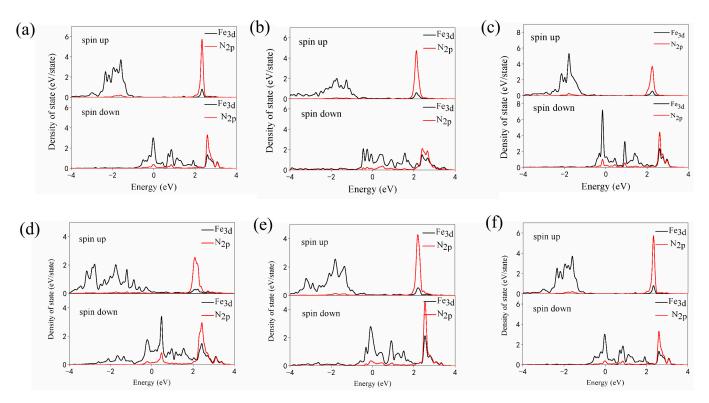


Figure 3. Partial density of states of N₂ adsorbed on (**a**) Fe/Ti₃C₂O₂, (**b**) Fe/N-Ti₃C₂O_{2-x}, (**c**) Fe/F-Ti₃C₂O_{2-x}, (**d**) Fe/P-Ti₃C₂O_{2-x}, (**e**) Fe/S-Ti₃C₂O_{2-x}, and (**f**) Fe/Cl-Ti₃C₂O_{2-x}.

3.3. N₂ Reduction Mechanism

The overall e-NRR reaction on the cathode is

$$N_2(g) + 6(H^+ + e^-) \rightarrow 2NH_3(g)$$
 (6)

and the anode reactions provide protons and electrons. Liu et al. [59] summarized the mechanism of the e-NRR. The e-NRR is divided into dissociation and association mechanisms by different hydrogenation (protonation and reduction) sequences and the breaking of the N \equiv N triple bond. In the dissociation mechanism, the N \equiv N bond is broken during the adsorption process (* denotes the adsorption site).

$$2* + N_2 \rightarrow 2*N \tag{7}$$

Then, two separated N atoms on the surface of the catalysts receive protons and electrons, and ammonia is formed in the last hydrogenation step:

$$*N + H^+ + e^- \to *NH \tag{8}$$

$$*NH + H^+ + e^- \rightarrow *NH_2 \tag{9}$$

$$*NH_2 + H^+ + e^- \rightarrow *NH_3 \tag{10}$$

$$NH_3 \rightarrow NH_3$$
 (11)

In the association mechanism, the $N \equiv N$ bond breaks at a certain hydrogenation step. According to the hydrogenation sequence, it can be further classified into distal, alternating, and enzymatic pathways. The hydrogenation step in the enzymatic pathway is similar to that in the alternating pathway; the difference is that N_2 adsorbs side to side in the enzymatic pathway, but ends in the distal and alternating pathways. For the distal and alternating pathways, the first two steps are

$$* + N_2 \rightarrow *N_2 \tag{12}$$

$$*N_2 + H^+ + e^- \rightarrow *N_2H \tag{13}$$

In the distal pathway, the N atom moves away from the catalytically gained protons and electrons, releasing the first NH₃ molecule, as follows:

$$*N_2H + H^+ + e^- \rightarrow *NNH_2 \tag{14}$$

$$*NNH_2 + H^+ + e^- \rightarrow *N + NH_3 \tag{15}$$

Hydrogenation then occurs on the remaining N atom and releases the second NH_3 molecule according to Reactions (8)–(11). In the alternating pathway, hydrogenation occurs on two newton atoms alternatively, and NH_3 is formed until the $N\equiv N$ bond is completely broken.

$$*N_2H + H^+ + e^- \to *NHNH$$
(16)

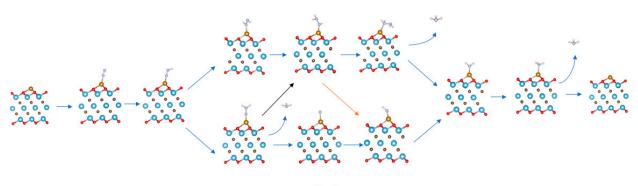
$$*NHNH + H^{+} + e^{-} \rightarrow *NHNH_{2}$$
(17)

$$*NHNH_2 + H^+ + e^- \rightarrow *NH_2NH_2 \tag{18}$$

$$*NH_2NH_2 + H^+ + e^- \rightarrow *NH_2 + NH_3$$
⁽¹⁹⁾

After the first NH_3 is released, the remaining $*NH_2$ obtains protons and electrons and releases the second ammonia according to Reactions (10) and (11). Figure 4 shows the other mixed pathways that follow neither the distal nor alternating pathways but a combination of two paths. Optimized structures of all the possible elementary steps in NRR is showed in Figure S2.

Alternating



Distal

Figure 4. Possible pathway and reaction intermediates for NRR with the associated mechanism. Dark brown, blue, red, brown, light blue, and light pink represent C, Ti, O, Fe, N, and H, respectively.

The ΔG values calculated by DFT calculations considered all correction terms, including the zero-point energy, temperature, and entropy corrections. Table 2 illustrates the E_{ZPE} and entropy corrections (*TS*) of different reaction intermediates on Fe/Ti₃C₂O₂ using the *TS* values obtained from the National Institute of Standards and Technology [60] at T = 298 K. The catalyst as a substrate is immobilized, although the surface is different, we compared the zero-point energy with the study of Ling [61]; the difference is marginal, as N₂ reduction also occurred on the transition metal atoms in Ling's research, and only the E_{ZPE} of NH₃ was significantly different. NH₃ is a gas phase, not an adsorbent, so other research was also compared [5]. The calculated E_{ZPE} and *TS* of H₂ are 0.27 and 0.4 eV [60], respectively.

Adsorption Species	$E_{\rm ZPE}$ (eV)	$E'_{\rm ZPE}$ (eV)	E _{ZPE} Difference (eV)	<i>TS</i> [60] (eV)
N ₂	0.15	0.15 [61]	0	0.59
*N≡N	0.19	0.20 [61]	0.01	0.23
*N=NH	0.47	0.49 [61]	0.02	0.20
*N-NH ₂	0.78	0.82 [61]	0.04	0.25
*N	0.09	0.08 [61]	0.01	0.06
*NH	0.31	0.35 [61]	0.04	0.14
*NH ₂	0.63	0.65 [61]	0.02	0.18
*NH ₃	1.00	1.02 [61]	0.02	0.23
*NH=NH	0.81	0.80 [61]	0.01	0.25
*NH-NH ₂	1.11	1.13 [61]	0.02	0.31
*NH2-NH2	1.50	1.49 [61]	0.01	0.27
NH ₃	0.92	0.96 [5]	0.04	0.60

Table 2. E_{ZPE} and *TS* of different reaction intermediates on Fe/Ti₃C₂O₂, *T* = 298 K.

As shown in Figure 5a–f, for all structures, the first protonation was likely to generate *NNH species; the ΔG values for Fe/Ti₃C₂O₂, Fe/N-Ti₃C₂O_{2-x}, Fe/F-Ti₃C₂O_{2-x}, Fe/P-Ti₃C₂O_{2-x}, Fe/S-Ti₃C₂O_{2-x}, and Fe/Cl-Ti₃C₂O_{2-x} increased to 0.90, 1.04, 0.85, 0.99, 0.88, and 1.01 eV, respectively. The second step is more likely to form *NNH₂ instead of the *NHNH species in the alternate path, as the energy requirements are higher, and the increments in ΔG for Fe/Ti₃C₂O₂, Fe/N-Ti₃C₂O_{2-x}, Fe/F-Ti₃C₂O_{2-x}, Fe/P-Ti₃C₂O_{2-x}, Fe/S-Ti₃C₂O_{2-x}, and Fe/Cl-Ti₃C₂O_{2-x} were 0.1, 0.06, 0.12, -0.05, 0.12, and 0.07 eV to form *NNH₂, respectively. In the subsequent hydrogenation steps, the intermediate configuration in the alternating pathway was easier to form than the first NH₃ molecule desorption in the distal pathway. The first NH₃ is not desorbed until the fifth proton is added, and adsorptive *NH₃ is formed when the sixth proton is added. The reaction *NNH₂ \rightarrow *NHNH₂ \rightarrow *NH₂NH₂ \rightarrow *NH₂ \rightarrow *NH₃ is exothermic, and larger energy input is required until the adsorptive *NH₃ is desorbed to form the second NH₃ molecule. The ΔG values of Fe/Ti₃C₂O₂, Fe/N-Ti₃C₂O_{2-x}, Fe/F-Ti₃C₂O_{2-x}, Fe/P-Ti₃C₂O_{2-x}, Fe/S-Ti₃C₂O_{2-x}, and Fe/Cl-Ti₃C₂O_{2-x} were 1.95, 1.11, 0.97, 1.07, 1.09, 0.99 eV, respectively. However, it was reported that the use of an acidic electrolyte can promote NH₃ desorption, as the protonation of adsorbed NH_3 to form $NH4^+$ can easily proceed [62,63], so the actual energy barrier is even smaller. For all these structures, the two potential limiting steps were the first hydrogenation of N₂ to form the *NNH species and the last process of NH₃ desorption to form the second NH₃ molecule. Compared with the original structure, nonmetallic doping was beneficial for the desorption of the last NH₃ molecule, but only the doping of F and S was beneficial for the formation of *NNH and NH₃.

Figure 6 shows the most possible reaction pathway for different catalysts. All these structures are likely to follow the mixed pathway: $N_2 \rightarrow *N_2 \rightarrow *NNH \rightarrow *NNH_2 \rightarrow *NHNH_2 \rightarrow *NH_2NH_2 \rightarrow *NH_2 \rightarrow *NH_3 \rightarrow NH_3$. In addition, the doping of nonmetals has a remarkable effect on NRR. For N_2 adsorption, E_{ads} is reduced, compared with the nondoped structure, which may be the reason why NH₃ desorption is easier in the last step. In the hydrogenation process, the doping of different nonmetals also makes each step of the hydrogenation easier or harder. The doping of N, P, and Cl makes it difficult for *N₂ to form *NNH, whereas F and S facilitate the formation of *NNH from *N₂. From *NNH to *NNH₂, only the doping of P shows an obvious impact and makes the transformation occur spontaneously. In comparison, the other doped nonmetals do not show a great effect. The doping of nonmetal also does not have much influence on *NNH₂ \rightarrow *NH₂ \rightarrow *NH₂

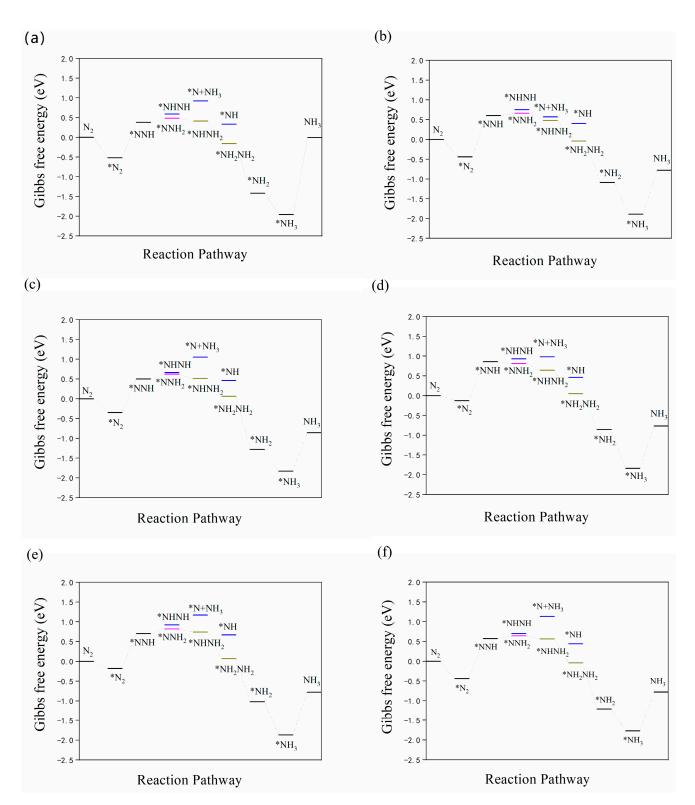


Figure 5. Gibbs free energy diagrams of (**a**) Fe/Ti₃C₂O₂, (**b**) Fe/N-Ti₃C₂O_{2-x}, (**c**) Fe/F-Ti₃C₂O_{2-x}, (**d**) Fe/P-Ti₃C₂O_{2-x}, (**e**) Fe/S-Ti₃C₂O_{2-x}, and (**f**) Fe/Cl-Ti₃C₂O_{2-x}.

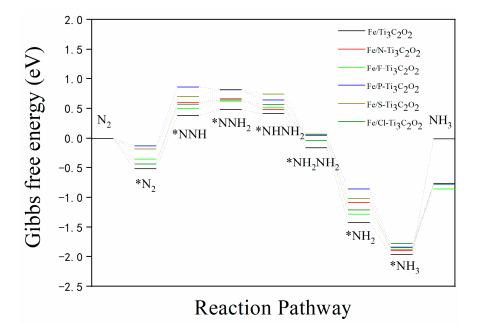


Figure 6. Nitrogen reduction reaction pathways for all structures.

4. Conclusions

The reaction pathway of the TM atom, Fe, adsorbed on pure $Ti_3C_2O_2$ and surface nonmetal (N/F/P/S/Cl)-doped $Ti_3C_2O_2$ as the N₂ reduction reaction catalyst was calculated using DFT. The main limiting steps of the reaction are $*N_2 + H \rightarrow *NNH$ and $*NH_3 \rightarrow NH_3$, and the limiting potentials of the two steps can reach 0.85–1.01 and 0.97–1.95 eV, respectively. Compared with pure $Ti_3C_2O_2$, nonmetal doping has an impact on catalytic performance. The doped nonmetal (N/F/P/S/Cl) reduces the energy barrier to form NH₃ in the last step, and only the doping of F and S is beneficial to the formation of *NNH in the first step and the desorption of $*NH_3$ in the last step. Therefore, the materials doped with F and S are considered better candidate materials for NRR among the tested catalysts. Our research demonstrates a feasible way to search for new NRR catalysts by modifying the surface of MXenes and loading TM atoms as new catalysts.

Supplementary Materials: The following are available online at https://www.mdpi.com/article/10 .3390/nano12071081/s1, Figure S1: Top and side views of $Ti_3C_2O_2$ and its nonmetal doped structure, Figure S2: Optimized structures of all the possible elementary steps in NRR, taking $Ti_3C_2O_2$ as an example. Other nonmetal-doped $Ti_3C_2O_2$ show similar geometric structure.

Author Contributions: The study was planned and designed by H.L., X.W., C.W. and P.Q. DFT calculations were performed by H.L. and X.W. The manuscript was prepared by H.L., L.X. and M.S. H.L., X.W., C.W., L.X., M.S. and P.Q. discussed the results and commented on the manuscript together. All authors have read and agreed to the published version of the manuscript.

Funding: This research was funded by the National Key Research and Development Program of China (Grant Nos. 2021YFB0700500 and 2018YFB0704300) and the National Natural Science Foundation of China (Grant No. 11975043).

Institutional Review Board Statement: Not applicable.

Informed Consent Statement: Not applicable.

Data Availability Statement: The datasets generated during and/or analyzed during the current study are available from the corresponding author.

Conflicts of Interest: The authors declare no conflict of interest.

References

- 1. Van der Ham, C.J.M.; Koper, M.T.M.; Hetterscheid, D.G.H. Challenges in reduction of dinitrogen by proton and electron transfer. *Chem. Soc. Rev.* **2014**, *43*, 5183–5191. [CrossRef] [PubMed]
- Guo, C.; Ran, J.; Vasileff, A.; Qiao, S.Z. Rational design of electrocatalysts and photo(electro)catalysts for nitrogen reduction to ammonia (NH₃) under ambient conditions. *Energy Environ. Sci.* 2018, 11, 45–56. [CrossRef]
- 3. Zhao, Y.; Zhao, Y.; Shi, R.; Wang, B.; Waterhouse, G.I.N.; Wu, L.Z.; Tung, C.H.; Zhang, T. Tuning Oxygen Vacancies in Ultrathin TiO₂ Nanosheets to Boost Photocatalytic Nitrogen Fixation up to 700 nm. *Adv. Mater.* **2019**, *31*, 1806482. [CrossRef] [PubMed]
- 4. Smill, V.; Streatfeild, R.A. Enriching the earth: Fritz Haber, Carl Bosch, and the transformation of world food production. *Electron. Green J.* **2002**, *43*, 622–623. [CrossRef]
- Gao, Y.; Zhuo, H.; Cao, Y.; Sun, X.; Zhuang, G.; Deng, S.; Zhong, X.; Wei, Z.; Wang, J. A theoretical study of electrocatalytic ammonia synthesis on single metal atom/MXene. *Cuihua Xuebao/Chin. J. Catal.* 2019, *40*, 152–159. [CrossRef]
 Wang, R. The dynamics of the peel. *Nat. Catal.* 2020, *3*, 333–334. [CrossRef]
- 6, 1600087. [CrossRef]
 8. Cong, L.; Xie, H.; Li, J. Hierarchical Structures Based on Two-Dimensional Nanomaterials for Rechargeable Lithium Batteries. *Adv. Energy Mater.* 2017, 7, 1601906. [CrossRef]
- 9. Fu, Q.; Bao, X. Surface chemistry and catalysis confined under two-dimensional materials. *Chem. Soc. Rev.* 2017, 46, 1842–1874. [CrossRef]
- 10. Novoselov, K.S.; Mishchenko, A.; Carvalho, A.; Neto, A.H.C. 2D materials and van der Waals heterostructures. *Science* **2016**, *353*, aac9439. [CrossRef]
- 11. Shan, A.; Teng, X.; Zhang, Y.; Zhang, P.; Xu, Y.; Liu, C.; Li, H.; Ye, H.; Wang, R. Interfacial electronic structure modulation of Pt-MoS2 heterostructure for enhancing electrocatalytic hydrogen evolution reaction. *Nano Energy* **2022**, *94*, 106913. [CrossRef]
- 12. Su, Y.; Cao, S.; Shi, L.; Qian, P. Investigation of biaxial strain behavior and phonon-limited mobility for γ graphyne: First-principles calculation. *J. Appl. Phys.* **2021**, *130*, 195703. [CrossRef]
- 13. Chan, H.; Wang, H.; Song, K.; Zhong, M.; Shi, L.; Qian, P. Origin of phonon-limited mobility in two-dimensional metal dichalcogenides. *J. Phys. Condens. Mat.* 2022, 34, 013003. [CrossRef] [PubMed]
- Suryanto, B.H.R.; Kang, C.S.M.; Wang, D.; Xiao, C.; Zhou, F.; Azofra, L.M.; Cavallo, L.; Zhang, X.; Macfarlane, D.R. Rational Electrode-Electrolyte Design for Efficient Ammonia Electrosynthesis under Ambient Conditions. ACS Energy Lett. 2018, 3, 1219–1224. [CrossRef]
- 15. Deng, J.; Liu, C. Boron-Doped Graphene Catalyzes Dinitrogen Fixation with Electricity. Chem 2018, 4, 1773–1774. [CrossRef]
- Luo, S.; Li, X.; Zhang, B.; Luo, Z.; Luo, M. MOF-Derived Co₃O₄@NC with Core-Shell Structures for N₂ Electrochemical Reduction under Ambient Conditions. ACS Appl. Mater. Interfaces 2019, 11, 26891–26897. [CrossRef]
- 17. Naguib, M.; Kurtoglu, M.; Presser, V.; Lu, J.; Niu, J.; Heon, M.; Hultman, L.; Gogotsi, Y.; Barsoum, M.W. Two-dimensional nanocrystals produced by exfoliation of Ti₃AlC₂. *Adv. Mater.* **2011**, *23*, 4248–4253. [CrossRef]
- Zhang, P.; Wang, D.; Zhu, Q.; Sun, N.; Fu, F.; Xu, B. Plate-to-Layer Bi₂MoO₆/MXene-Heterostructured Anode for Lithium-Ion Batteries. *Nano-Micro Lett.* 2019, 11, 81. [CrossRef]
- 19. Liu, Y.T.; Zhang, P.; Sun, N.; Anasori, B.; Zhu, Q.Z.; Liu, H.; Gogotsi, Y.; Xu, B. Self-Assembly of Transition Metal Oxide Nanostructures on MXene Nanosheets for Fast and Stable Lithium Storage. *Adv. Mater.* **2018**, *30*, 1707334. [CrossRef]
- 20. Tang, Q.; Zhou, Z.; Chen, Z. Innovation and discovery of graphene-like materials via density-functional theory computations. *Wiley Interdiscip. Rev. Comput. Mol. Sci.* **2015**, *5*, 360–379. [CrossRef]
- 21. Liu, A.; Liang, X.; Ren, X.; Guan, W.; Gao, M.; Yang, Y.; Yang, Q.; Gao, L.; Li, Y.; Ma, T. Recent Progress in MXene-Based Materials: Potential High-Performance Electrocatalysts. *Adv. Funct. Mater.* **2020**, *30*, 2003437. [CrossRef]
- 22. Hu, Q.; Sun, D.; Wu, Q.; Wang, H.; Wang, L.; Liu, B.; Zhou, A.; He, J. MXene: A new family of promising hydrogen storage medium. *J. Phys. Chem. A* 2013, *117*, 14253–14260. [CrossRef] [PubMed]
- 23. Wu, X.; Wang, Z.; Yu, M.; Xiu, L.; Qiu, J. Stabilizing the MXenes by Carbon Nanoplating for Developing Hierarchical Nanohybrids with Efficient Lithium Storage and Hydrogen Evolution Capability. *Adv. Mater.* **2017**, *29*, 1607017. [CrossRef] [PubMed]
- 24. Naguib, M.; Come, J.; Dyatkin, B.; Presser, V.; Taberna, P.L.; Simon, P.; Barsoum, M.W.; Gogotsi, Y. MXene: A promising transition metal carbide anode for lithium-ion batteries. *Electrochem. Commun.* **2012**, *16*, 61–64. [CrossRef]
- Luo, J.; Tao, X.; Zhang, J.; Xia, Y.; Huang, H.; Zhang, L.; Gan, Y.; Liang, C.; Zhang, W. Sn⁴⁺ Ion Decorated Highly Conductive Ti₃C₂ MXene: Promising Lithium-Ion Anodes with Enhanced Volumetric Capacity and Cyclic Performance. ACS Nano 2016, 10, 2491–2499. [CrossRef]
- 26. Yan, J.; Ren, C.E.; Maleski, K.; Hatter, C.B.; Anasori, B.; Urbankowski, P.; Sarycheva, A.; Gogotsi, Y. Flexible MXene/Graphene Films for Ultrafast Supercapacitors with Outstanding Volumetric Capacitance. *Adv. Funct. Mater.* **2017**, *27*, 1701264. [CrossRef]
- 27. Zhong, Y.; Xia, X.H.; Shi, F.; Zhan, J.Y.; Tu, J.P.; Fan, H.J. Transition metal carbides and nitrides in energy storage and conversion. *Adv. Sci.* **2015**, *3*, 1500286. [CrossRef]
- 28. Li, Z.; Wu, Y. 2D Early Transition Metal Carbides (MXenes) for Catalysis. Small 2019, 15, 1804736. [CrossRef]
- 29. Zhao, Q.; Zhu, Q.; Miao, J.; Zhang, P.; Wan, P.; He, L.; Xu, B. Flexible 3D Porous MXene Foam for High-Performance Lithium-Ion Batteries. *Small* **2019**, *15*, 1904293. [CrossRef]

- 30. Sun, J.; Kong, W.; Jin, Z.; Han, Y.; Ma, L.; Ding, X.; Niu, Y.; Xu, Y. Recent advances of MXene as promising catalysts for electrochemical nitrogen reduction reaction. *Chin. Chem. Lett.* **2020**, *31*, 953–960. [CrossRef]
- Azofra, L.M.; Li, N.; Macfarlane, D.R.; Sun, C. Promising prospects for 2D d²-d⁴ M₃C₂ transition metal carbides (MXenes) in N₂ capture and conversion into ammonia. *Energy Environ. Sci.* 2016, *9*, 2545–2549. [CrossRef]
- Li, N.; Chen, X.; Ong, W.J.; Macfarlane, D.R.; Zhao, X.; Cheetham, A.K.; Sun, C. Understanding of Electrochemical Mechanisms for CO₂ Capture and Conversion into Hydrocarbon Fuels in Transition-Metal Carbides (MXenes). ACS Nano 2017, 11, 10825–10833. [CrossRef] [PubMed]
- 33. Liu, D.; Zhang, G.; Ji, Q.; Zhang, Y.; Li, J. Synergistic Electrocatalytic Nitrogen Reduction Enabled by Confinement of Nanosized Au Particles onto a Two-Dimensional Ti₃C₂ Substrate. *ACS Appl. Mater. Interfaces* **2019**, *11*, 25758–25765. [CrossRef]
- 34. Kong, W.; Gong, F.; Zhou, Q.; Yu, G.; Ji, L.; Sun, X.; Asiri, A.M.; Wang, T.; Luo, Y.; Xu, Y. An MnO₂-Ti₃C₂T: X MXene nanohybrid: An efficient and durable electrocatalyst toward artificial N₂ fixation to NH₃ under ambient conditions. *J. Mater. Chem. A* 2019, 7, 18823–18827. [CrossRef]
- 35. Kresse, G.; Furthmüller, J. Efficiency of ab-initio total energy calculations for metals and semiconductors using a plane-wave basis set. *Comput. Mater. Sci.* **1996**, *6*, 15–50. [CrossRef]
- 36. Kresse, G.; Furthmüller, J. Efficient iterative schemes for ab initio total-energy calculations using a plane-wave basis set. *Phys. Rev. B Condens. Matter Mater. Phys.* **1996**, *54*, 11169. [CrossRef]
- Perdew, J.P.; Burke, K.; Ernzerhof, M. Generalized gradient approximation made simple. *Phys. Rev. Lett.* 1996, 77, 3865–3868. [CrossRef]
- 38. Blöchl, P.E. Projector augmented-wave method. Phys. Rev. B 1994, 50, 17953–17979. [CrossRef]
- 39. Monkhorst, H.J.; Pack, J.D. Special points for Brillouin-zone integrations. Phys. Rev. B 1976, 13, 5188–5192. [CrossRef]
- 40. Grimme, S.; Antony, J.; Ehrlich, S.; Krieg, H. A consistent and accurate ab initio parametrization of density functional dispersion correction (DFT-D) for the 94 elements H-Pu. *J. Chem. Phys.* **2010**, *132*, 154104. [CrossRef]
- Henkelman, G.; Arnaldsson, A.; Jónsson, H. A fast and robust algorithm for Bader decomposition of charge density. *Comput. Mater. Sci.* 2006, *36*, 354–360. [CrossRef]
- Sanville, E.; Kenny, S.D.; Smith, R.; Henkelman, G. Improved grid-based algorithm for Bader charge allocation. *J. Comput. Chem.* 2007, 28, 899–908. [CrossRef] [PubMed]
- 43. Momma, K.; Izumi, F. VESTA 3 for three-dimensional visualization of crystal, volumetric and morphology data. *J. Appl. Crystallogr.* **2011**, 44, 1272–1276. [CrossRef]
- 44. Ozerov, R.; Kogan, V.; Zhdanov, G.; Kukhto, O. The crystalline structure of solid isotopes of hydroxy. *Sov. Phys. Cryst.* **1962**, *6*, 507–508.
- 45. Bernal, J.D.; Fowler, R.H. A theory of water and ionic solution, with particular reference to hydrogen and hydroxyl ions. *J. Chem. Phys.* **1933**, *1*, 515–548. [CrossRef]
- 46. Olovsson, I.; Templeton, D.H. X-ray study of ammonia and ammonia monohydrate. *Am. Crystallogr. Assoc. Progr. Abstr.* **1959**, *12*, 832–836.
- 47. Johnson, M.W.; Sándor, E.; Arzi, E. The crystal structure of deuterium fluoride. *Acta Crystallogr. Sect. B Struct. Crystallogr. Cryst. Chem.* **1975**, *31*, 1998–2003. [CrossRef]
- Furberg, S.; Landmark, P.; Gardell, S.; Magnéli, A.; Magnéli, A.; Pestmalis, H.; Åsbrink, S. The Crystal Structure of Phosphorous Acid. Acta Chem. Scand. 1957, 11, 1505–1511. [CrossRef]
- 49. Cockcroft, J.K.; Fitch, A.N. The solid phases of deuterium sulphide by powder neutron diffraction. *Z. Krist. New Cryst. Struct.* **1990**, 193, 1–19. [CrossRef]
- 50. Sándor, E.; Farrow, R.F.C. Crystal structure of solid hydrogen chloride and deuterium chloride. *Nature* **1967**, *213*, 171–172. [CrossRef]
- 51. Saal, J.E.; Kirklin, S.; Aykol, M.; Meredig, B.; Wolverton, C. Materials design and discovery with high-throughput density functional theory: The open quantum materials database (OQMD). *JOM* **2013**, *65*, 1501–1509. [CrossRef]
- 52. Kirklin, S.; Saal, J.E.; Meredig, B.; Thompson, A.; Doak, J.W.; Aykol, M.; Rühl, S.; Wolverton, C. The Open Quantum Materials Database (OQMD): Assessing the accuracy of DFT formation energies. *NPJ Comput. Mater.* **2015**, *1*, 15010. [CrossRef]
- 53. Rossmeisl, J.; Logadottir, A.; Nørskov, J.K. Electrolysis of water on (oxidized) metal surfaces. *Chem. Phys.* 2005, 319, 178–184. [CrossRef]
- 54. Mashtalir, O.; Naguib, M.; Mochalin, V.N.; Dall'Agnese, Y.; Heon, M.; Barsoum, M.W.; Gogotsi, Y. Intercalation and delamination of layered carbides and carbonitrides. *Nat. Commun.* **2013**, *4*, 1716. [CrossRef] [PubMed]
- Johnson, L.R.; Sridhar, S.; Zhang, L.; Fredrickson, K.D.; Raman, A.S.; Jang, J.; Leach, C.; Padmanabhan, A.; Price, C.C.; Frey, N.C.; et al. MXene Materials for the Electrochemical Nitrogen Reduction-Functionalized or Not? ACS Catal. 2020, 10, 253–264. [CrossRef]
- Gao, G.; O'Mullane, A.P.; Du, A. 2D MXenes: A New Family of Promising Catalysts for the Hydrogen Evolution Reaction. ACS Catal. 2017, 7, 494–500. [CrossRef]
- 57. Tong, Y.; He, M.; Zhou, Y.; Zhong, X.; Fan, L.; Huang, T.; Liao, Q.; Wang, Y. Electromagnetic wave absorption properties in the centimetre-band of Ti₃C₂T_x MXenes with diverse etching time. *J. Mater. Sci. Mater. Electron.* **2018**, *29*, 8078–8088. [CrossRef]
- Wang, X.; Su, Y.; Song, M.; Song, K.; Chen, M.; Qian, P. Design single nonmetal atom doped 2D Ti₂CO₂ electrocatalyst for hydrogen evolution reaction by coupling electronic descriptor. *Appl. Surf. Sci.* 2021, 556, 149778. [CrossRef]

- 59. Liu, D.; Chen, M.; Du, X.; Ai, H.; Lo, K.H.; Wang, S.; Chen, S.; Xing, G.; Wang, X.; Pan, H. Development of Electrocatalysts for Efficient Nitrogen Reduction Reaction under Ambient Condition. *Adv. Funct. Mater.* **2021**, *31*, 2008983. [CrossRef]
- 60. National Institute of Standards and Technology. Available online: https://janaf.nist.gov/ (accessed on 30 December 2021).
- 61. Ling, C.; Ouyang, Y.; Li, Q.; Bai, X.; Mao, X.; Du, A.; Wang, J. A General Two-Step Strategy–Based High-Throughput Screening of Single Atom Catalysts for Nitrogen Fixation. *Small Methods* **2019**, *3*, 1800376. [CrossRef]
- Chun, H.J.; Apaja, V.; Clayborne, A.; Honkala, K.; Greeley, J. Atomistic Insights into Nitrogen-Cycle Electrochemistry: A Combined DFT and Kinetic Monte Carlo Analysis of NO Electrochemical Reduction on Pt(100). ACS Catal. 2017, 7, 3869–3882. [CrossRef]
- 63. Clayborne, A.; Chun, H.-J.; Rankin, R.B.; Greeley, J. Elucidation of Pathways for NO Electroreduction on Pt(111) from First Principles. *Angew. Chem.* 2015, 127, 8373–8376. [CrossRef]