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Hierarchical non-woven fabric NiO/TiO₂ film as an efficient anode material for lithium-ion batteries†

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Hierarchical non-woven fabric NiO/TiO₂ film is prepared using a facile two-step synthesis by electrospinning and subsequent hydrothermal reaction to yield TiO₂ film decorated with NiO nanosheets. As an anode material for Li-ion batteries, the NiO/TiO₂ composite exhibits discharge and charge capacities of 1251.3 and 1284.3 mA h g⁻¹, with good cycling performance and rate capability. Characterization shows that the NiO nanosheets are distributed on the surface of the TiO₂ nanofibers. The total specific capacity of NiO/TiO₂ is higher than the sum of pure TiO₂ and NiO, indicating a positive synergistic effect of TiO₂ and NiO on the improvement of electrochemical performance. The results suggest that non-woven fabric NiO/TiO₂ film is a promising anode material for lithium ion batteries.

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1. Introduction

With the rapid development of human society, the demand for green energy is increasing dramatically.^{1,2} Owing to the high voltage, high capacity, and long cycle life, rechargeable lithium-ion batteries (LIBs) have become power sources for portable electronic devices and electric vehicles.³ Graphite, a commercial anode material for LIBs, exhibits a low theoretical capacity (372 mA h g⁻¹) and poor rate capability, and thus cannot meet the increasing requirements of high-performance electrode materials.^{4,5} Comparatively, transition metal oxides (TMOs), due to their high theoretical capacities and superior electrochemical properties, have attracted much attention as alternative anode materials. In particular, nickel oxide (NiO) is regarded as one of the most appealing anode materials because of its high theoretical capacity (718 mA h g⁻¹), high density (6.67 g cm⁻³), low cost, environmental benignity, and natural abundance.^{6,7}

Unfortunately, the poor electrical and ionic conductivities of NiO hinder its electrochemical performances.^{8–10} Besides, the sluggish kinetics of the conversion reaction and the large volume expansion (95.69%) of the large-sized electrode materials severely restrict the development of NiO anodes.¹¹ To address these shortcomings, Poizot *et al.* first proposed the nano-sized TMOs that undergo conversion reactions with lithium and provide two- or three-fold higher reversible capacity than graphitic anodes.¹² After that, great efforts have been made

to improving the performance of NiO as the anode material through fabricating various nanostructures, including nanoparticles,¹³ nanosheets,¹⁴ nanotubes,^{15,16} nanofibers,¹⁷ hollow microspheres,^{18,19} and sandwich-like films.²⁰ The nanoscaled materials not only shorten the electron and Li⁺ diffusion pathways, but also increase the electrode–electrolyte interface, enhancing the cycling stability and high-rate capacity.^{21,22} Nevertheless, the issues for nanoscaled materials are that they can easily become aggregated and get more structural defects, resulting in poor cycling performance.²³ Thus, sophisticated nanomaterials, particularly with designed unique structures, are highly needed.

Herein, a novel hierarchical non-woven fabric film composed of NiO nanosheets (NSs) grown on TiO₂ nanofibers is prepared *via* an electrospinning method combined with a hydrothermal post-treatment and used as an efficient anode material for LIBs. The special three-dimensional (3D) nano-architecture possesses outstanding advantages. First, the direct growth of NiO nanosheets on TiO₂ nanofibers improves the conductivity and separates active materials from each other, completely avoiding the aggregation and agglomeration. In addition, this 3D morphology helps to shorten the pathways for Li⁺ and electron transport and enlarges exposed surface for more lithium-exchange channels, leading to high rate capability and cycling stability.

2. Experimental

2.1 Material fabrication

All chemicals or materials were used directly without further purification. First, the non-woven fabric film composed of TiO₂ nanofibers is prepared according to our previous report.^{24,25} Then, the as-prepared TiO₂ non-woven fabric film was employed

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as the scaffold for the growth of NiO nanosheets *via* a hydrothermal reaction. Briefly, the non-woven fabric film was tilted against the wall of the autoclave at a certain angle. 5 mmol of $\text{NiSO}_4 \cdot 6\text{H}_2\text{O}$ and 10 mmol of hexamethylenetetramine (HMT) were added to 60 mL of water to form a homogeneous green solution. Subsequently, the solution was transferred into the autoclave, which was then tightly sealed and kept in an electric oven at 100 °C for 12 h. After cooling down naturally to room temperature, the non-woven fabric film was then taken out, washed with water thoroughly, and dried at 60 °C overnight. The obtained non-woven fabric film was placed in a tube furnace and annealed in air at the temperature of 450 °C with a heating rate of 2 °C min^{-1} . After cooling to room temperature, the resultant non-woven fabric film was donated as NiO/TiO₂ film.

2.2 Structure characterization

The compositions of the as-prepared materials were examined by X-ray diffraction (XRD, D8-ADVANCE) with Cu K α radiation. The morphology and elemental distributions of the samples were investigated by a field emission scanning electron microscope (FE-SEM, JSM-7001F) equipped with energy dispersive X-ray spectroscopy (EDS) and a transmission electron microscope (TEM, JEM-2100).

2.3 Electrochemical measurements

Electrochemical measurements were carried out in CR2016 coin-type cells. The testing cells were assembled in an argon-filled glove box using NiO/TiO₂ non-woven fabric film and Li foil as the working electrode and counter electrode, respectively. Celgard 2400 as the separator, and 1 mol L⁻¹ LiPF₆ in a 1 : 1 (volume) mixture of ethylene carbonate and ethyl methyl carbonate as the electrolyte. The charge/discharge test was performed between 0.01 V and 3 V at various current densities using a NEWER battery tester. Cyclic voltammetry (CV, 0.01 V to 3 V, 0.1 mV s⁻¹) and electrochemical impedance spectroscopy (EIS, 0.01 to 10⁵ Hz) were conducted using electrochemical workstation (CHI 660E) with 5 mV amplitude. All tests were carried out at room temperature.

3. Results and discussion

The 1D nanostructures with fewer surface defects are well recognized as efficient for fast electron transport.²⁶ To preserve the long fibrous structure of the electrospinning nanofiber, TiO₂ non-woven fabric film was used as the current collector, and NiO NSs are grown on the surface of nanofibers after the hydrothermal treatment. The detailed fabrication procedure is illustrated in Scheme 1. Briefly, an electrospinning process was used to synthesize TiO₂ non-woven fabric film, which was then employed as the scaffold for the growth of NiO nanosheets *via* a hydrothermal reaction. The composite material film was obtained by calcining in air at 450 °C for 2 h to decompose Ni(OH)₂ to NiO.

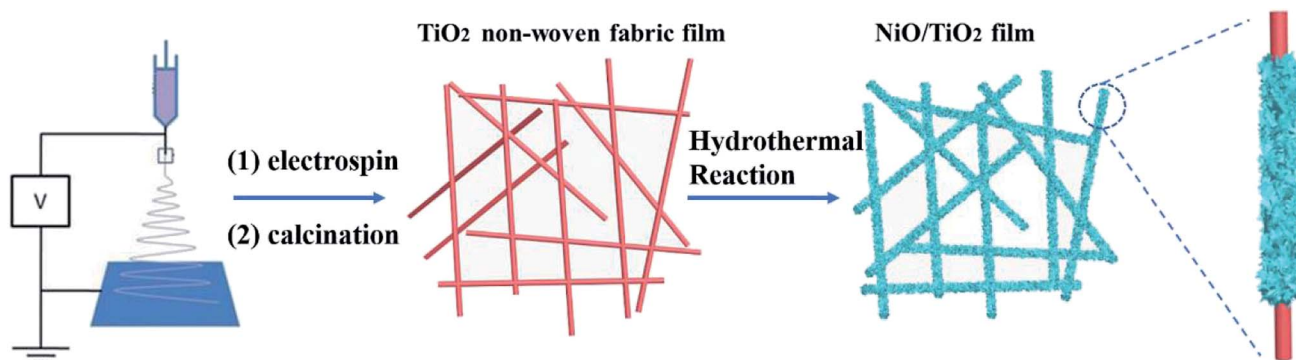
The XRD patterns of the samples are shown in Fig. 1. The TiO₂ film exhibits diffraction peaks that agree well with the

characteristic peaks of anatase TiO₂ crystal. Namely, the peaks at 25.4, 48.1 and 54.6° correspond to the (101), (200) and (105) crystal planes of TiO₂ (JCPDS card no. 21-1272), respectively. All the reflections of NiO sample can be perfectly indexed as NiO (JCPDS card no. 65-2901) in terms of the relative intensity and position. The well-resolved diffraction peaks appearing at 37.08, 43.23, and 62.81° are correspondingly assigned to the (111), (200), and (220) planes, respectively. For the NiO/TiO₂ film, all the diffraction peaks are in agreement with the standard patterns of TiO₂ (JCPDS card no. 21-1272) and NiO (JCPDS card no. 65-2901), and no impurity diffraction peaks are observed.

The morphologies of the samples were examined by field-emission scanning electron microscope (SEM) and transmission electron microscope (TEM). Fig. 2 shows the SEM images of the TiO₂ nanofibers (Fig. 2a and b) and the NiO/TiO₂ composite (Fig. 2c and d). It can be seen that the electrospinning TiO₂ nanofibers have a relatively smooth surface with diameters ranging from 350 to 700 nm, which are intact during the film fabrication process. The long fiber structure should be effective in facilitating electron transport. The nanofibers stack loosely with each other, leaving lots of voids, which facilitate the electrolyte penetration as well as Li⁺ transport between the electrode and the electrolyte. After the hydrothermal treatment, as shown in Fig. 2c and d, each nanofiber is coated with NiO nanosheets with the average thickness of ~7 nm and height of ~350 nm. The TEM analyses (Fig. 2e and f) also demonstrate the highly branched structure, in which NSs are disorderly grown on the surface. The high-resolution TEM (HR-TEM) image (Fig. 2g) of a single NS shows a typical lattice distance of 0.21 nm, which is indexed to the (200) plane of NiO. Compared with the pristine TiO₂ nanofiber film, the large interspaces between fibers are filled with NSs, resulting in more active sites of the composite film. The corresponding element mapping of Ti, O, and Ni in Fig. 3 also confirms the uniform dispersion of NiO nanosheets on the TiO₂ fiber surface. However, in the absence of TiO₂ nanofibers, NiO NSs tend to self-assemble into nanoflowers with sizes ranging from 6 to 10 μm (Fig. S1†).

Cyclic voltammetry (CV) tests and galvanostatic discharge/charge measurements were investigated in half-cell configurations to evaluate the lithium storage properties of the as-prepared NiO/TiO₂ composite film as an anode electrode in LIBs. Fig. 4a shows the CV curves of the NiO/TiO₂ film for the 1st and 2nd cycles at a scan rate of 0.01 mV s⁻¹ over the voltage range of 0.01–3 V. In the first cathodic scan, a strong peak appears at ~0.47 V, which corresponds to the formation of a solid electrolyte interphase (SEI) film and the reduction of NiO to Ni.²⁷ In the following cycle, this cathodic peak significantly shifts to about 0.9 V, which agrees well with previous reports.^{28,29} The anodic scans show similar profiles, in which two oxidation peaks are found at 1.49 V and 2.1 V, indicating the decomposition of the SEI film and the oxidation of Ni to NiO, respectively.³⁰

Fig. 4b and c show the discharge/charge curves of NiO and NiO/TiO₂ electrodes at 0.1C and 0.2C (1C = 718 mA h g⁻¹), respectively. The NiO electrode exhibits discharge and charge capacities of 615.8 and 662.6 mA h g⁻¹ at 0.1C, and 626.4 and 579.6 mA h g⁻¹ at 0.2C. Impressively, the NiO/TiO₂ electrode



Scheme 1 Schematic diagram for the fabrication of the hierarchical NiO/TiO₂ film.

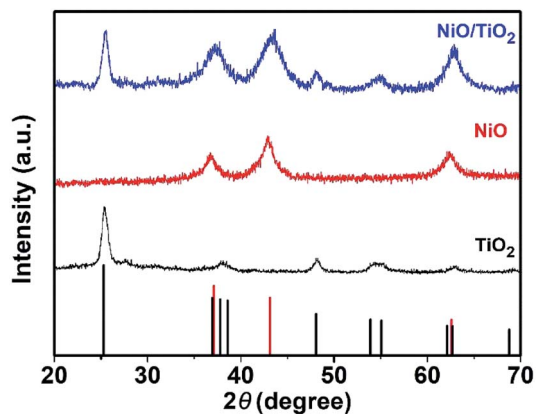


Fig. 1 XRD patterns of the TiO₂, NiO and NiO/TiO₂, respectively.

exhibits discharge and charge capacities of 1251.3 and 1284.3 mA h g⁻¹ at 0.1C, and 1075 and 1154.4 mA h g⁻¹ at 0.2C, with the coulombic efficiency of 97% and 93%, respectively. Thus, the discharge and charge capacities of the NiO/TiO₂ electrode are about twofold of those delivered by the NiO electrode.

The electrochemical performance is crucial for the potential anode materials in LIBs. The rate performance of the NiO/TiO₂ electrode is first examined at different charge/discharge rates from 0.05C to 0.5C for 5 cycles, and then returns to 0.05C. As illustrated in Fig. 4d, the NiO/TiO₂ electrode exhibits high discharge capacities at lower current densities, which is 1388.9 mA h g⁻¹ at the rate of 0.05C and decreases to 509.5 mA h g⁻¹ at 0.5C. When the current returns to 0.05C, the discharge capacity recovers to 1015 mA h g⁻¹, showing good

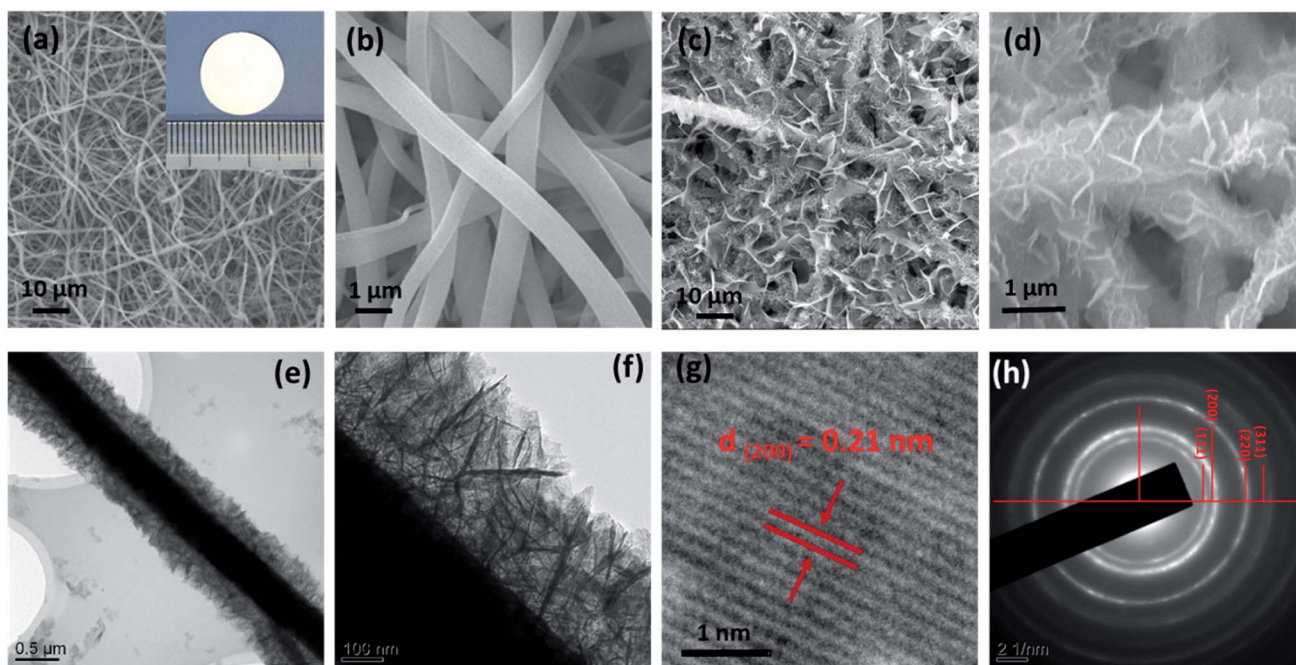


Fig. 2 (a) Low- and (b) high-magnification SEM images of the TiO₂ fibers. The inset figure shows the TiO₂ electrode. (c) Low- and (d) high-magnification SEM images of the NiO/TiO₂ composite film. (e and f) TEM images of the NiO/TiO₂ composite. (g) HRTEM and (h) SAED images of NiO NS.

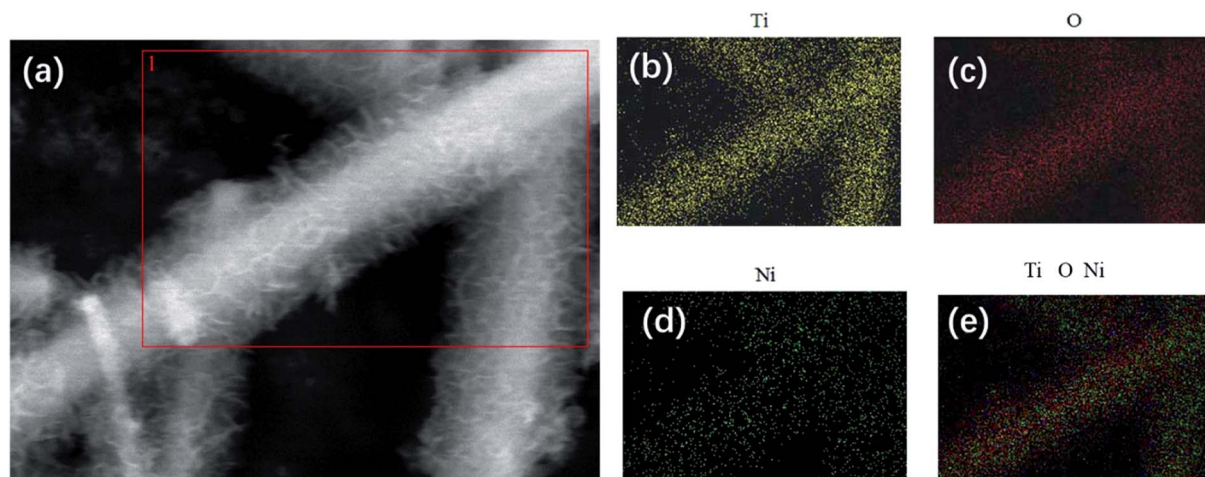


Fig. 3 (a) SEM image of NiO/TiO₂ and (b–e) the corresponding EDX element mapping of (b) Ti, (c) O, (d) Ni, and (e) the reconstructed chemical map.

stability. Fig. 4e presents the cycling performance of the two electrodes. The initial discharge capacity of NiO/TiO₂ electrode is 1285.6 mA h g⁻¹, and the capacity maintains a high value of 487.1 mA h g⁻¹ after 50 cycles at 0.1C. While for the NiO electrode, the initial discharge capacity is 655.2 mA h g⁻¹, decreasing to only 198.9 mA h g⁻¹ after 50 cycles.

To confirm the electrochemical kinetics related to Li⁺ diffusion, electrochemical impedance spectroscopy (EIS) measurements were carried out. Fig. 4f presents the Nyquist plots of the electrodes in the frequency range of 10⁻² to 10⁵ Hz at the open-circuit potential. The semicircle in the high-frequency region is

attributed to the charge transfer resistance (R_{ct}) at the electrode–electrolyte interface, and the inclined line in the low-frequency range (the Warburg impedance) corresponds to Li-ion diffusion in the electrode.³¹ The NiO/TiO₂ electrode shows the lowest R_{ct} and Warburg impedance among the testing electrodes. The low R_{ct} is ascribed to the high number of electroactive sites due to the good dispersion of NiO nanosheets, while the tunneled TiO₂ nanofibers shorten the Li⁺ diffusion pathways, reducing the Warburg impedance.

Cycling performance under high charging/discharging rate was also examined to explore the potential of the NiO/TiO₂

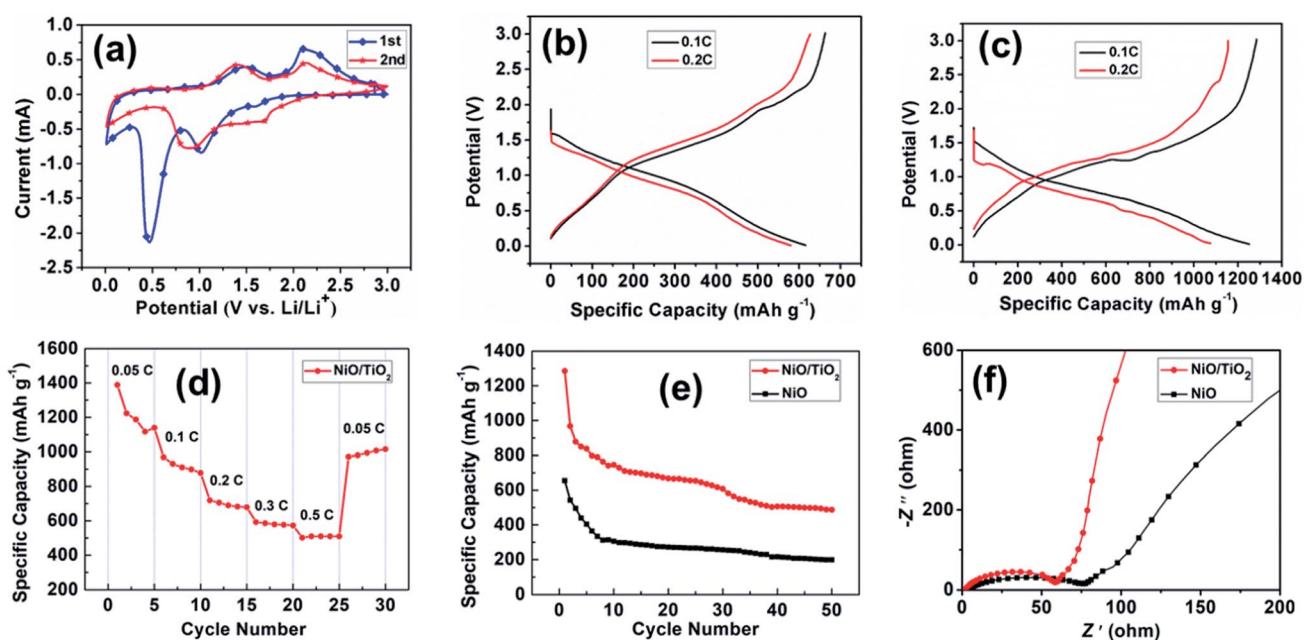


Fig. 4 (a) Cyclic voltammograms of the NiO/TiO₂ film at a scan rate of 0.1 mV s⁻¹ for the first two cycles. (b and c) Charge–discharge voltage profiles of (b) NiO and (c) NiO/TiO₂ electrodes at current densities of 0.1C and 0.2C. (d) Rate performance of the NiO/TiO₂ electrode at different current densities. (e) Cycling performance of NiO and NiO/TiO₂ electrodes at a current density of 0.1C. (f) Nyquist plots of NiO and NiO/TiO₂ electrodes.

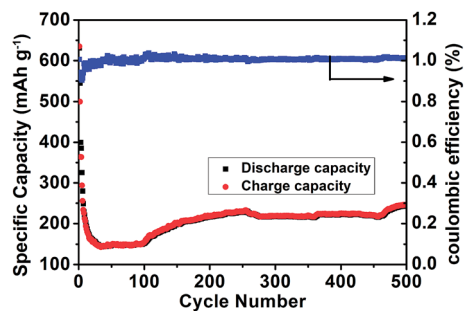


Fig. 5 Long cycling stability of the NiO/TiO₂ electrode at 6C.

electrode. Fig. 5 records its capacity variation under high current density of 6C (for NiO, 1C = 718 mA g⁻¹) for 500 cycles. To start with, the discharge capacity of NiO/TiO₂ decreases to 152.4 mA h g⁻¹ in the beginning 100 cycles. Then excitingly, it increases steadily in the following cycles. Afterwards, it gradually rises to 243.9 mA h g⁻¹ at the 500th cycle. Even though the capacity experiences some up and down movements, which may be related with the domination of the interfacial Li⁺ storage mechanism, the long life under such high power is still remarkable.³² In addition, high coulombic efficiencies (about 100% in average) are also found after the first cycle. The Fig. S2† reveal that after suffering intense cycling test, the NiO/TiO₂ remains original structure with uniform solid electrolyte interface (SEI) layer coating on the surface, which prevents the aggregation of the active materials. In addition, the integrity of the electrode is well preserved.

Fig. 6 outlines the proposed electron transfer and Li⁺ diffusion mechanism during the lithiation–delithiation process. The reasons for the high electrochemical performance can be explained as follows. First, high conductive TiO₂ non-woven fabric film as the current collector can enhance the electron

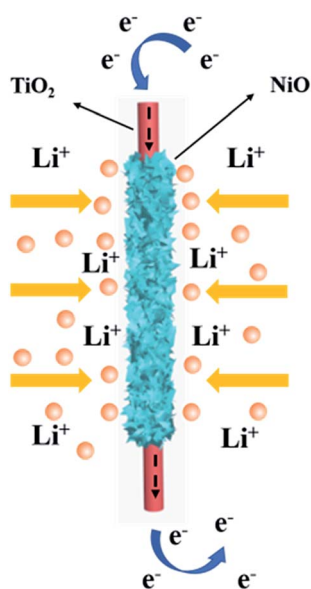


Fig. 6 Schematic illustration of the lithiation–delithiation process of the NiO/TiO₂ composite film.

transfer and effectively suppress the aggregation of NiO NSs, enabling the electrode material to achieve good rate capability. Second, the large interfacial areas can provide more sites for Li⁺ storage as well as facilitate the Li⁺ transport between the electrode and the electrolyte.^{33,34}

4. Conclusions

In conclusion, a non-woven fabric film composed of ultrathin NiO nanosheet arrays vertically grown on TiO₂ nanofibers has been fabricated through an electrospinning method followed by a solvothermal process. The as-fabricated NiO/TiO₂ electrode exhibits excellent electrochemical performance, which is attributed to the ultrathin thickness of NiO nanosheets and the direct contact with the 3D non-woven fabric TiO₂ film. This facile synthesis method of ultrathin NiO nanosheet arrays opens up new strategies for developing advanced electrodes for high-performance LIBs.

Conflicts of interest

There are no conflicts to declare.

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References

- 1 M. Armand and J. M. Tarascon, *Nature*, 2008, **451**, 652–657.
- 2 P. G. Bruce, B. Scrosati and J. M. Tarascon, *Angew. Chem., Int. Ed.*, 2008, **47**, 2930–2946.
- 3 M. A. Rahman, G. Song, A. I. Bhatt, Y. C. Wong and C. Wen, *Adv. Funct. Mater.*, 2016, **26**, 647–678.
- 4 S. Wu, N. Du, H. Wu, C. Xiao, W. Zhao and D. Yang, *RSC Adv.*, 2016, **6**, 109649–109656.
- 5 Z. Wang, M. Zhang and J. Zhou, *ACS Appl. Mater. Interfaces*, 2016, **8**, 11507–11515.
- 6 M. V. Reddy, G. V. S. Rao and B. V. R. Chowdari, *Chem. Rev.*, 2013, **113**, 5364–5457.
- 7 J. Cabana, L. Monconduit, D. Larcher and M. R. Palacín, *Adv. Mater.*, 2010, **22**, E170–E192.
- 8 L. Zhang, J. Mu, Z. Wang, G. Li, Y. Zhang and Y. He, *J. Alloys Compd.*, 2016, **671**, 60–65.
- 9 J. Wu, W. Yin, W.-W. Liu, P. Guo, G. Liu, X. Liu, D. Geng, W.-M. Lau, H. Liu and L. M. Liu, *J. Mater. Chem. A*, 2016, **4**, 10940–10947.
- 10 Y. Feng, H. Zhang, Y. Zhang, Y. Bai and Y. Wang, *J. Mater. Chem. A*, 2016, **4**, 3267–3277.
- 11 P. Simon and Y. Gogotsi, *Nanosci. Tech.*, 2009, vol. 7, pp. 320–329.
- 12 P. Poizot, S. Laruelle, S. Grugeon, L. Dupont and J.-M. Tarascon, *Nature*, 2000, **407**, 496–499.

- 13 A. K. Rai, L. T. Anh, C.-J. Park and J. Kim, *Ceram. Int.*, 2013, **39**, 6611–6618.
- 14 X. Shang, X. Li, H. Yue, S. Xue, Z. Liu, X. Hou and D. He, *Mater. Lett.*, 2015, **157**, 7–10.
- 15 H. Pang, Q. Lu, Y. Li and F. Gao, *Chem. Commun.*, 2009, 7542–7544.
- 16 S. A. Needham, G. X. Wang and H. K. Liu, *J. Power Sources*, 2006, **159**, 254–257.
- 17 V. Aravindana, P. S. Kumara, J. Sundaramurthy, W. C. Ling, S. Ramakrishn and S. Madhavia, *J. Power Sources*, 2013, **227**, 284–290.
- 18 D. Xie, W. Yuan, Z. Dong, Q. Su, J. Zhang and G. Du, *Electrochim. Acta*, 2013, **92**, 87–92.
- 19 A. K. Mondal, D. Sua, Y. Wang, S. Chen, Q. Liu and G. Wang, *J. Alloys Compd.*, 2014, **582**, 522–527.
- 20 J. Zhong, X. L. Wang, X. H. Xia, C. D. Gu, J. Y. Xiang, J. Zhang and J. P. Tu, *J. Alloys Compd.*, 2011, **509**, 3889–3893.
- 21 H. Zhang, Y. Feng, Y. Zhang, L. Fang, W. Li, Q. Liu, K. Wu and Y. Wang, *ChemSusChem*, 2014, **7**, 2000–2006.
- 22 J. Chen and F. Cheng, *Acc. Chem. Res.*, 2009, **42**, 713–723.
- 23 Y. Feng, H. Zhang, W. Li, L. Fang and Y. Wang, *J. Power Sources*, 2016, **301**, 78–86.
- 24 Y. Cao, Y. Dong, H. Feng, H. Chen and D. Kuang, *Electrochim. Acta*, 2016, **189**, 259–264.
- 25 Y. Cao, Y. Dong, H. Chen, D. Kuang and C. Su, *RSC Adv.*, 2016, **6**, 78202–78209.
- 26 C. A. Bonino, L. Ji, Z. Lin, O. Toprakci, X. Zhang and S. A. Khan, *ACS Appl. Mater. Interfaces*, 2011, **3**, 2534–2542.
- 27 B. Varghese, M. V. Reddy, Y. Zhu, S. L. Chang, T. C. Hoong, G. V. S. Rao, B. V. R. Chowdari, A. T. S. Wee, C. T. Lim and C. H. Sow, *Chem. Mater.*, 2008, **20**, 3360–3367.
- 28 J. Liang, H. Hu, H. Park, C. Xiao, S. Ding, U. Paik and X. Lou, *Energy Environ. Sci.*, 2015, **8**, 1707–1711.
- 29 X. Xu, H. Tan, K. Xi, S. Ding, D. Yu, S. Cheng, G. Yang, X. Peng, A. Fakeeh and R. V. Kumarc, *Carbon*, 2015, **84**, 491–499.
- 30 H. Li, H. Ma, M. Yang, B. Wang, H. Shao, L. Wang, R. Yu and D. Wang, *Mater. Res. Bull.*, 2017, **87**, 224–229.
- 31 S. Nilmoung, T. Sinprachim, I. Kotutha, P. Kidkhunthod, R. Yimnirun, S. Rujirawat and S. Maensiribce, *J. Alloys Compd.*, 2016, **688**, 1131–1140.
- 32 H. Fang, W. Zou, J. Yan, Y. L. Xing and S. C. Zhang, *ChemElectroChem*, 2018, **5**, 2458–2463.
- 33 W. Li, L. Zeng, Y. Wu and Y. Yu, *Sci. China Mater.*, 2016, **59**, 287–321.
- 34 Y. Zhou, Y. Liu, W. Zhao, F. Xie, R. Xu, B. Li, X. Zhou and H. Shen, *J. Mater. Chem. A*, 2016, **4**, 5932–5941.