

Article

Dielectric Barrier Discharge Plasma Jet (DBDjet) Processed Reduced Graphene Oxide/Polypyrrole/Chitosan Nanocomposite Supercapacitors

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Abstract: Reduced graphene oxide (rGO) and/or polypyrrole (PPy) are mixed with chitosan (CS) binder materials for screen-printing supercapacitors (SCs) on arc atmospheric-pressure plasma jet (APPJ)-treated carbon cloth. The performance of gel-electrolyte rGO/CS, PPy/CS, and rGO/PPy/CS SCs processed by a dielectric barrier discharge plasma jet (DBDjet) was assessed and compared. DBDjet processing improved the hydrophilicity of these three nanocomposite electrode materials. Electrochemical measurements including electrical impedance spectroscopy (EIS), cyclic voltammetry (CV), and galvanostatic charging-discharging (GCD) were used to evaluate the performance of the three types of SCs. The Trasatti method was used to evaluate the electric-double layer capacitance (EDLC) and pseudocapacitance (PC) of the capacitance. The energy and power density of the three types of SCs were illustrated and compared using Ragone plots. Our experiments verify that, with the same weight of active materials, the combined use of rGO and PPy in SCs can significantly increase the capacitance and improve the operation stability.

Keywords: atmospheric-pressure plasma; dielectric barrier discharge; reduced graphene oxide; polypyrrole; supercapacitor; plasma

1. Introduction

Supercapacitors (SCs) have a higher capacitance and lower working voltage than regular capacitors; they are usually used in high-power-density applications requiring fast charging and discharging [1–4]. Their energy storage mechanisms are mainly the electrical double-layer capacitance (EDLC) and pseudocapacitance (PC). EDLC is manifested by rapid ion adsorption/desorption at the electrode/electrolyte interface [5]. This generally occurs in carbon-based electrode materials such as carbon black, carbon nanofibers, carbon nanotubes, and graphenes [6–9]. By contrast, PC is manifested by the Faraday oxidation/reduction reaction of active materials on the electrode surface [10–12]. Because graphene has high conductivity, carrier mobility, and specific surface area, it has been widely used in energy devices such as fuel cells [13], solar cells [14], oxygen evolution reaction electrodes [15], and SCs [16]. However, in practical applications, graphene may

agglomerate and stack, and this may limit the charge accumulation in the electric double layer [17,18] and result in insufficient energy density. To further increase the capacitance, graphene is often compounded with conductive polymers to couple EDLC and PC [19]. Conducting polymers, such as polypyrrole (PPy), polyaniline, polythiophene, and their derivatives, are candidate active materials for realizing PCs owing to their good conductivity, excellent mechanical flexibility, high theoretical capacitance, and fast redox reaction properties [2,20–22]. Various studies have aimed to combine the advantages of SCs and traditional batteries to achieve high energy density and high power density. In this study, reduced graphene oxides (rGOs) and PPy were selected for the hybrid nanocomposite, and chitosan (CS) was used as a binder and dispersant to screen-print rGO/PPy/CS pastes onto carbon cloth.

Plasma has been extensively used for surface activation and modification [23–25]. Reactive plasma species may create active sites or defects, introduce dopants, and graft surface functional groups on materials surfaces [12,26,27]. All of these tailor the surface properties and may alter the surface wettability [1]. Plasma treatment is frequently performed to improve the quality of joining, bonding, or adhesion [28]. Low-pressure plasma operated with a vacuum system has been extensively used in industrial applications. Atmospheric-pressure plasma (APP) can be operated in a regular pressure environment without using a vacuum system; this is particularly advantageous for biomedicine and agriculture applications [29,30]. By avoiding the need for a vacuum system, the operation cost of APPs can be reduced, making APP an economical viable tool to perform plasma processing when the samples can tolerate environmental cleanliness conditions [4,27,31–33]. Owing to the vigorous interactions of APP reactive species on carbon-based materials, APPs have been extensively used for the rapid processing of carbonaceous materials [4,7,12,27,31,34,35].

In this study, we use a moderate temperature (~500 °C) nitrogen arc atmospheric pressure plasma jet (APPJ) to make a carbon cloth substrate hydrophilic. Because PPy cannot tolerate high temperature, near room temperature (<40 °C) He-2%O₂ dielectric barrier discharge plasma jet (DBDjet) is used to treat the screen-printed rGO/PPy/CS for further improving its hydrophilicity, thereby facilitating its contact with the gel electrolyte. With the same material weight loading (i.e., the same weight of active materials), we compare the performance of SCs made with rGO/CS, PPy/CS, and rGO/PPy/CS. Our experimental results show improved capacitance and long-term operation stability when using rGO and PPy together as the active materials for SCs.

2. Materials and Methods

2.1. Preparation of rGO/PPy/CS, rGO/CS, and PPy/CS Pastes

CS (obtained from shrimp shells, deacetylation degree: >75%, Sigma Aldrich, Burlington, MA, USA) acetic acid solution was prepared by stirring a mixture of 0.3 g of CS powder and 20 mL of 0.1 M acetic acid (purity: >99.5%, AUECC, Kaohsiung, Taiwan) at 50 °C for 2 h. After natural cooling, the mixture was then stirred at room temperature for 1 h before use. Next, three types of active materials were introduced: 0.1 g of rGO (thickness: <5 nm, chip diameter: 0.1–5 μm, oxygen content: 5–10%, purity: >99%, Golden Innovation Business, New Taipei City, Taiwan), 0.1 g of PPy (conductivity: 10–50 S/cm, water: <1.0%, Sigma Aldrich, Burlington, MA, USA), and mixture of 0.05 g rGO and 0.05 g PPy. Then the powdered active material was mixed with 1.5 g ethanol (purity: 95%, Echo Chemical, Miaoli, Taiwan) and 3.6 g of CS acetic acid solution and stirred with a magnetic stirrer. Finally, the resulting mixture was concentrated by using a rotary evaporator at 50 °C for 150 s.

2.2. Fabrication of rGO/PPy/CS, rGO/CS, and PPy/CS Electrodes

A rectangular carbon cloth (3 cm × 4 cm) was used as the current collector. First, the carbon cloth was pre-treated using a nitrogen DC pulse APPJ (as shown in Figure 1) for 30 s. The treatment process is described in a previous study [1]. After the APPJ treatment, three types of active materials were screen printed on the carbon cloth with an area of

$1.5 \times 2 \text{ cm}^2$. Next, the sample was calcined at $80 \text{ }^\circ\text{C}$ for 10 min in an oven. Finally, a DBDjet (as shown in Figure 1) was applied to post-treat the sample three times with a scanning speed of 2 mm/s. The carrier gas of the DBDjet was mixture with 98% helium and 2% oxygen.

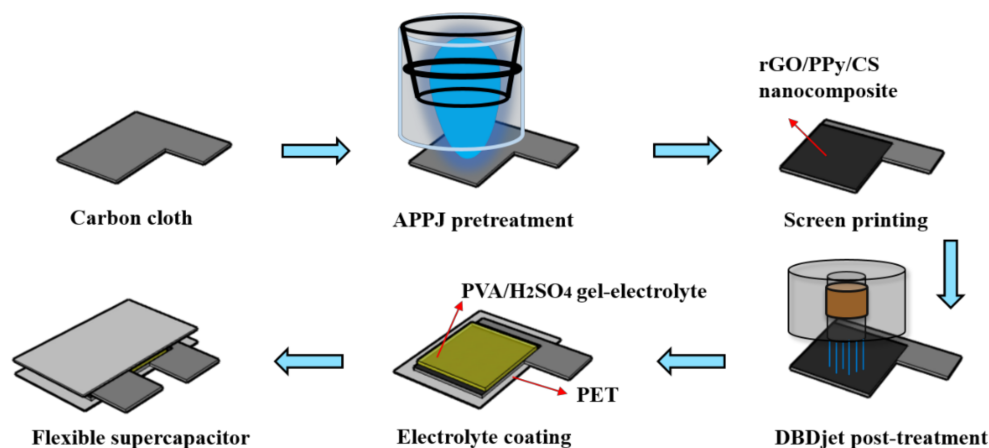


Figure 1. Schematic diagram of fabrication process of flexible SCs.

2.3. Preparation of Gel-Electrolyte Solution

First, 1.5 g of polyvinyl alcohol (PVA; MW: 850,000–124,000, 99+% hydrolysis, Sigma Aldrich) powder was slowly added to 15 mL of 1 M sulfuric acid (purity: 95–97%, AUECC, Burlington, MA, USA). Then, the mixture was stirred at 200 rpm at $70 \text{ }^\circ\text{C}$ for 6 h. Then, the gel-electrolyte solution was naturally cooled and stirred at 300 rpm at room temperature.

2.4. Fabrication of Symmetric Sandwich-Type SCs

First, 1 mL of gel electrolyte solution was dropped on the DBDjet post-treated rGO/CS, PPy/CS, or rGO/PPy/CS nanocomposite-coated carbon cloth. The sample was then naturally dried for 24 h. The gel electrolyte coating step was repeated three times. Finally, the symmetrical sandwich-type SC was combined with two gel electrolyte-coated electrodes with the same active materials. Light pressing was applied against the sides of the gel electrolyte to assure the flatness of the device. Figure 1 shows a schematic of the process flowchart.

2.5. Characterization of rGO/PPy/CS, rGO/CS, and PPy/CS Nanocomposites and SCs

The water contact angles of the rGO/CS, PPy/CS, or rGO/PPy/CS nanocomposite-coated carbon cloths were measured by using a goniometer (Sindatek, Model 100SB, Taipei City, Taiwan). Scanning electron microscopy (SEM, JSM-IT100, JEOL, Tokyo, Japan) and X-ray photoelectron spectroscopy (XPS, Sigma Probe, Thermo VG-Scientific) were used to check the surface morphology and surface chemical bonding status of the rGO/PPy/CS, rGO/CS, and PPy/CS-coated carbon cloths. An electrochemical workstation (Autolab PGSTAT204, Metrohm, Utrecht, The Netherlands) was used to perform cyclic voltammetry (CV) (0–0.8 V, potential scan speed: 2–200 mV/s), galvanostatic charging discharging (GCD, 0–0.8 V, constant current: 0.25, 0.5, 1, 3, and 5 mA), and electrochemical impedance spectroscopy (EIS, 0.1–100,000 Hz) measurements in a two-electrode configuration to characterize the SC [1].

3. Results and Discussion

3.1. Surface Morphology of rGO/CS, PPy/CS, and rGO/PPy/CS Electrodes

Figure 2 shows the SEM images of the electrodes made with rGO/CS, PPy/CS, and rGO/PPy/CS pastes on carbon cloth after DBDjet post-treatment, respectively. The higher-magnification SEM images with rGO/CS paste reveal the sheet-like structure of rGO nano-

flakes. rGOs with a large specific surface area typically provide EDLC for SCs [36]. The SEM images of the PPy/CS electrode show spherical PPy particles with coarse surfaces [37]. The higher-magnification SEM images with rGO/PPy/CS electrode show a combined structure, with a sheet-like structure of rGOs and spherical particles of PPy.

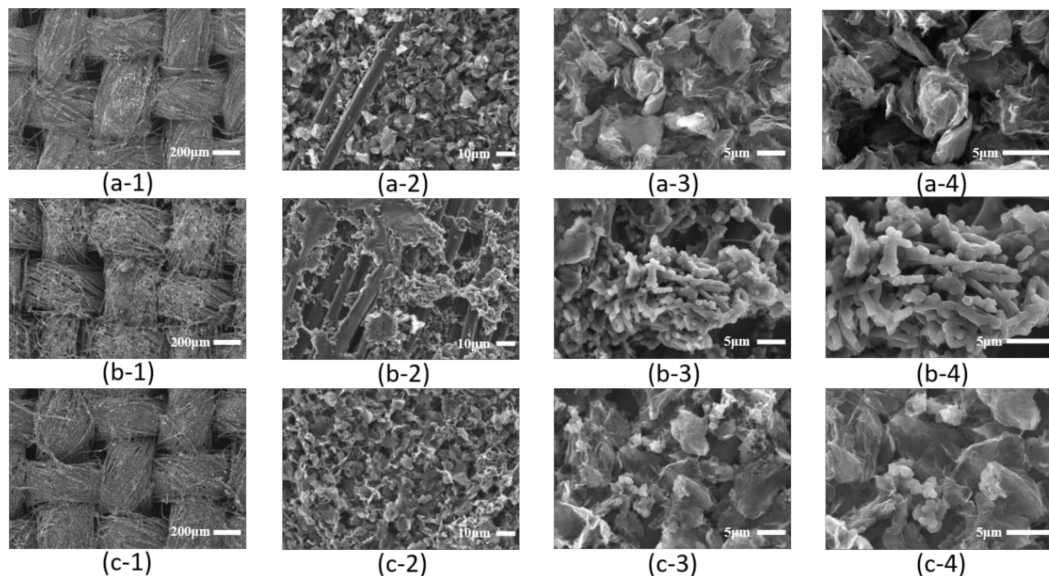


Figure 2. SEM images. rGO/CS electrodes with (a-1) 70 \times , (a-2) 1000 \times , (a-3) 3000 \times , and (a-4) 5000 \times magnification; PPy/CS electrodes with (b-1) 70 \times , (b-2) 1000 \times , (b-3) 3000 \times , and (b-4) 5000 \times magnification; and rGO/PPy/CS electrodes with (c-1) 70 \times , (c-2) 1000 \times , (c-3) 3000 \times , and (c-4) 5000 \times magnification.

3.2. Hydrophilicity Assessment and XPS Results of rGO/CS, PPy/CS and rGO/PPy/CS Electrodes

Figure 3a shows that the water contact angle of pristine carbon cloth is 115.7 $^\circ$, indicating its hydrophobic nature. After nitrogen arc APPJ pre-treatment, owing to the synergistic effect of high temperature and reactive plasma species, the carbon cloth became hydrophilic, and the droplet was immediately absorbed into the carbon cloth, as shown in Figure 3b [38]. Figure 3c-1,d-1,e-1 respectively shows the water contact angles of screen-printed rGO/CS, PPy/CS, and rGO/PPy/CS pastes on carbon clothes before DBDjet treatment. The PPy/CS-coated carbon cloth shows a water contact angle of 90.8 $^\circ$ owing to the strong hydrophobicity of PPy. After DBDjet treatment on rGO/CS, PPy/CS, and rGO/PPy/CS-coated carbon clothes, the material surfaces became hydrophilic and absorbed water droplets, as shown in Figure 3c-2,d-2,e-2, respectively. The low temperature DBDjet processing can avoid the thermal damage on the coated materials (especially PPy) and simultaneously improve the wettability [1], thereby enhancing the interfacial contact with the follow-up-coated gel electrolytes.

XPS analyses were conducted to identify the chemical elements. The electrode containing rGOs showed higher C-C/C-O peak area ratios in XPS C1s spectra compared to that without rGOs. Further, the N1s spectra indicated the -NH $^+$ - content with the PPy coating, suggesting the successful deposition of PPy. Figures S1–S4 show the XPS results, and Tables S1–S4 show the corresponding element proportions (Supplementary Materials).

3.3. EIS

Figure 4 shows the Nyquist plots of SCs with rGO/CS, PPy/CS, and rGO/PPy/CS nanocomposites. The inset of Figure 4 shows magnified view of the high frequency region. Table 1 lists the capacitive contribution of SCs with rGO/CS, PPy/CS, and rGO/PPy/CS nanocomposites. In the low frequency region, the curves are straight lines, corresponding to the Warburg diffusion impedance (W_0). A straight line with a steeper slope indicates an electric double layer (EDL) with a compact structure. Conversely, one with a milder slope

tends to indicate an ion diffusion in the capacitor. The SC with rGO/CS nanocomposites is more EDL-like; by contrast, that with PPy/CS nanocomposites is more ion-diffusion-like. Ideally, the SC with rGO/PPy/CS nanocomposites should exhibit a combination of these two effects. In the high-frequency region, a small recessed semicircle can be observed, indicating the parallel combination of the charge transfer resistance (R_{CT}) and the constant phase element (CPE). The lower charge transfer resistance (indicated by a smaller semicircle) corresponds to a higher electron propagation speed. This is an important factor in fast redox systems, especially for SCs. However, among all SCs, there is no obvious difference in this regard. The difference of the Warburg diffusion impedance dominates the influence of each material on the SC. CPE corresponds to the interfacial capacitance between the electrode and the electrolyte. However, there is no obvious difference between each CPE value. Here, it should be noted that the intercept at the x -axis represents the series resistance (R_S).

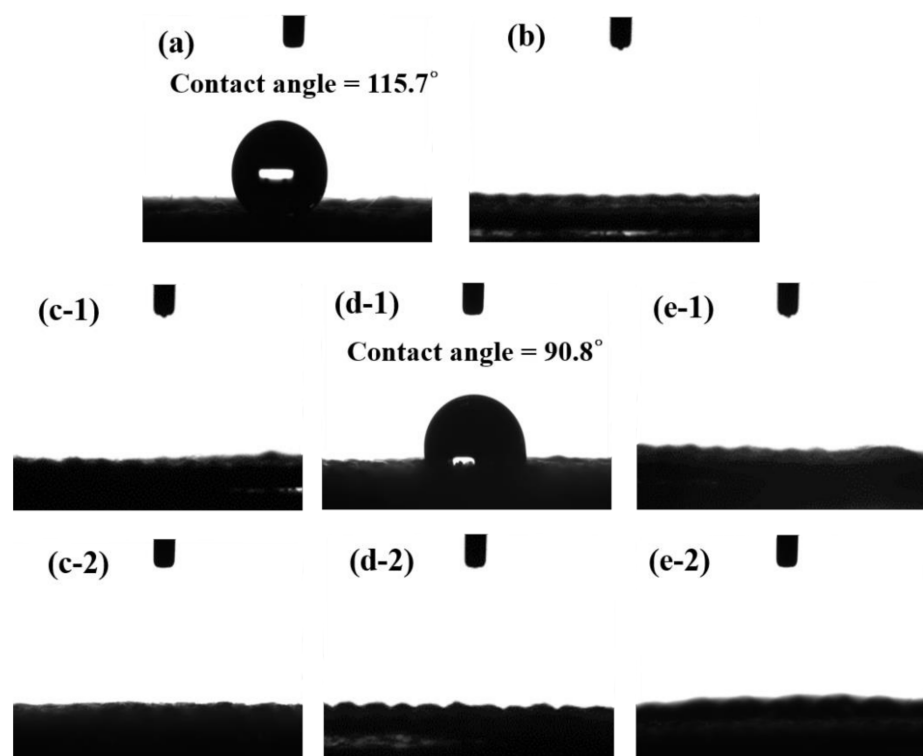


Figure 3. Water contact angle of pure carbon cloth (a) without APPJ treatment and (b) with APPJ treatment; rGO/CS-coated carbon cloth (c-1) without DBDjet treatment and (c-2) with DBDjet treatment; PPy/CS-coated carbon cloth (d-1) without DBDjet treatment and (d-2) with DBDjet treatment; and rGO/PPy/CS-coated carbon cloth (e-1) without DBDjet treatment and (e-2) with DBDjet treatment.

Table 1. Capacitive contribution of SCs with rGO/CS, PPy/CS, and rGO/PPy/CS nanocomposites.

	R_S (Ω)	R_{CT} (Ω)	W_0 (Ω)	C_{PE} ($\mu s^\alpha/\Omega$)
rGO/CS	4.05	0.16	1.0	0.0002
PPy/CS	5.72	0.72	704.7	0.0015
rGO/PPy/CS	4.53	0.45	1.3	0.002

3.4. CV Measurement

Figure 5 shows the cyclic voltammetry (CV) measurement results of three different nanocomposite SCs at different scan rates. The enclosed area monotonically increased with

the potential scan rate. The potential window and potential scan rate were set as 0–0.8 V and 2–200 mV/s, respectively. The areal capacitance can be calculated as [39]

$$C_A = \frac{A}{s \times v \times \Delta V} \quad (1)$$

Here, C_A is the areal capacitance (mF/cm^2); A , the convolution area of the CV curves (mW); ΔV , the potential window (V); v , the potential scan rate (V/s); and s , the geometric area of the active part (cm^2). An SC with rGO/CS nanocomposites mainly exhibits the EDLC, and therefore, the CV curve is more squarish [40]. By comparison, an SC with PPy/CS nanocomposites shows a much more irregular CV curve. This indicates the apparent oxidation–reduction reaction of PPy [41]. The SC with rGO/PPy/CS nanocomposites shows an obvious redox peak, and its areal capacitance ($45.32 \text{ mF}/\text{cm}^2$) is much higher than that of rGO/CS ($23.37 \text{ mF}/\text{cm}^2$) and PPy/CS ($24.34 \text{ mF}/\text{cm}^2$) at the scan rate of 2 mV/s. In this case, the areal capacitance of the SC with rGO/PPy/CS nanocomposite is boosted by both EDLC and PC. The combination of rGO and PPy dramatically improves the SC performance.

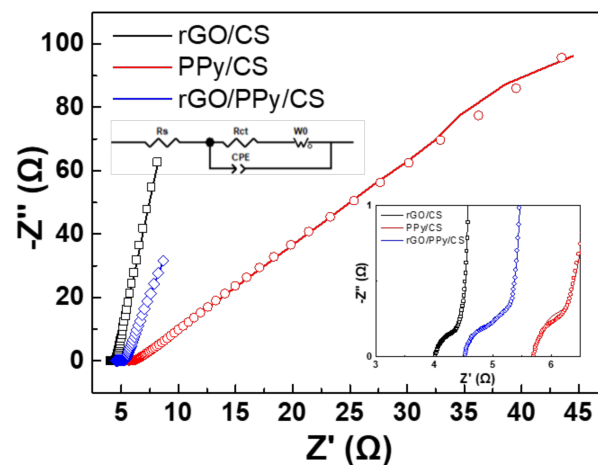


Figure 4. Nyquist plots of SCs with rGO/CS, PPy/CS, and rGO/PPy/CS nanocomposites.

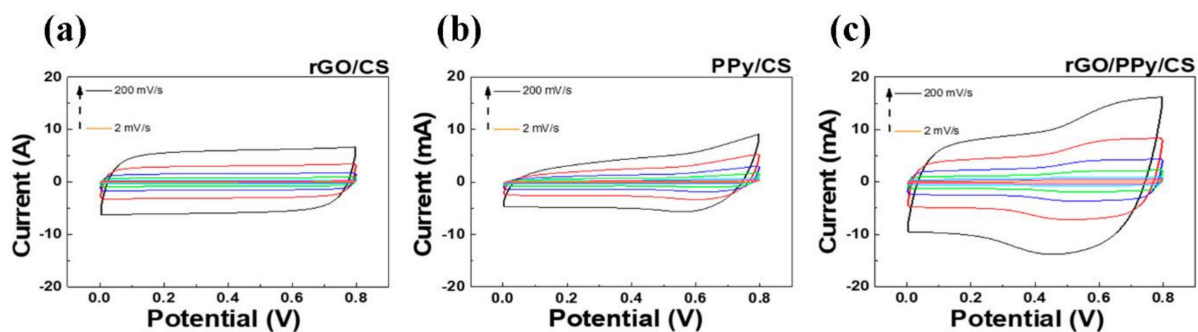


Figure 5. Cyclic voltammetry curves of (a) rGO/CS, (b) PPy/CS, (c) rGO/PPy/CS nanocomposite SCs at different scan rates.

3.5. Trasatti's Plots

Trasatti's plots can be used to determine the contributions of the EDLC and the PC [42]. Trasatti's theory is used to divide the storage mechanism into two different types: surface charge (C_{out}) and diffusion control charge (C_{in}). Different mechanism dominates in different potential scan rate ranges [43]. C_{out} is mainly formed by the accumulated charge on the active material surface, which is the origin of EDLC. Conversely, the charges stored in the PC-based material contribute to C_{in} . By extrapolation, the two types of capacitance

values can be evaluated. When the voltage scan rate approaches infinity, the C_{out} dominates the capacitance value. When the voltage scan rate approaches zero, there is enough reaction time for the charge to diffuse into the active material, and charges can be stored internally and externally ($C_{total} = C_{in} + C_{out}$) [44,45].

Figure 6 shows Trasatti's plots of SCs with rGO/CS, PPy/CS, and rGO/PPy/CS nanocomposites. Figure 6a shows the relationship between $1/C_A$ and $V^{0.5}$; this straight-fitted line can be used to extrapolate C_{total} . Figure 6b shows the relationship between C_A and $v^{-0.5}$; it can be used to extrapolate C_{out} . Table 2 shows the calculated capacitance contribution. The SCs with rGO/CS nanocomposites exhibit higher EDLC (79.4%), whereas those with PPy/CS nanocomposites exhibit roughly equal contributions of EDLC and PC (50%:50%). The rGOs added into PPy/CS to form rGO/PPy/CS boost the EDLC effect to achieve the best performance with the corresponding capacitive contribution percentage of EDLC and PC being ~75%:25%.

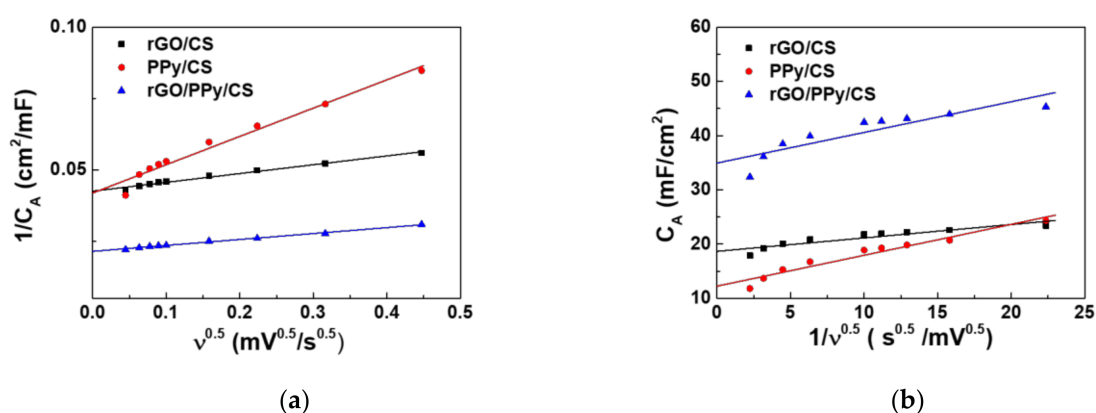


Figure 6. Trasatti's plots of SCs with rGO/PPy/CS, rGO/CS, and PPy/CS nanocomposites: (a) $1/C_A$ vs. $v^{1/2}$. (b) C_A vs. $1/v^{1/2}$.

Table 2. Capacitive contributions of SCs with rGO/CS, PPy/CS, and rGO/PPy/CS nanocomposites.

	Ctotal (mF/cm ²)	Cin (mF/cm ²)	Cout (mF/cm ²)	Capacitive Contribution (EDLC:PC) (%)
rGO/CS	23.5	18.7	4.9	79.4:20.6
PPy/CS	23.9	12.2	11.6	51.3:48.7
rGO/PPy/Cs	46.6	34.9	11.7	75.0:25.0

3.6. GCD

Figure 7 shows the GCD results of the SCs with rGO/PPy/CS, rGO/CS, and PPy/CS nanocomposites. GCD measurements can be used to calculate the areal capacitance values as [46]

$$C_A = \frac{2I \times T}{s \times \Delta V} \quad (2)$$

Here, C_A is the areal capacitance (mF/cm²); I , the discharging current (mA); T , the discharging time (s); ΔV , the potential window (V); and s , the geometric area of the active material (cm²). Table 3 lists the areal capacitance values calculated from the GCD curves in Figure 7. The SC with rGO/PPy/CS nanocomposites shows the best performance with an areal capacitance of 72.79 mF/cm² under a discharging current of 1 mA; this is 2–3 times higher than that achieved under the other two conditions. This result agrees with the CV results. With the PPy/CS composite, the slope of the GCD curve changes for a low discharging current. This phenomenon indicates that PPy performs a more obvious redox reaction [1]. The parameter used to compare the performance of SCs is gravimetric-specific capacitance. In our case, however, the measurement of weight change due to screen-printed materials in our SCs could lead to large error in determining gravimetric-specific

capacitance because of the materials loading, and therefore, we used areal capacitance to compare our results. In addition, the areal capacitance and/or volumetric capacitance is more useful to describe the specification in commercial SCs. Table S5 (Supplementary Materials) lists the C_A performance of various composite SCs. The C_A value of SC with rGO/PPy/CS nanocomposite is on par with most of the results in Table S5.

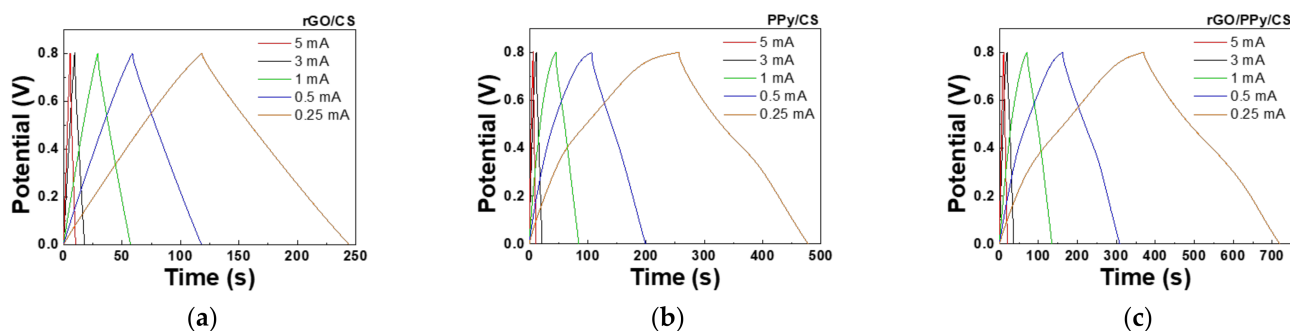


Figure 7. GCD curves of the SCs with (a) rGO/CS, (b) PPy/CS, and (c) rGO/PPy/CS nanocomposites under different discharge currents.

Table 3. Areal capacitance of SCs analyzed from GCD measurement.

	Areal Capacitance (mF/cm ²)				
	Discharging Current (mA)				
	5	3	1	0.5	0.25
rGO/CS	19.30	21.25	23.42	24.64	26.23
PPy/CS	20.97	25.01	32.77	38.57	46.34
rGO/PPy/CS	38.44	43.32	52.91	61.33	72.79

3.7. Ragone Plots

Ragone plots are used to evaluate the energy and power density of the SCs. The areal energy density and areal power density can respectively be calculated as [47]

$$E_A = \frac{C_A \times \Delta V^2}{7.2} \quad (3)$$

$$P_A = \frac{3.6 \times E_A}{T} \quad (4)$$

where E_A is the energy density ($\mu\text{Wh}/\text{cm}^2$); C_A , the areal capacitance from the GCD measurement (mF/cm^2); ΔV , the potential window (V); P_A , the power density (mW/cm^2); and T , the discharging time (s).

Figure 8 shows Ragone plots of the SCs with rGO/CS, PPy/CS, and rGO/PPy/CS nanocomposites. The SC with rGO/PPy/CS nanocomposites exhibited the best achieved energy density of $6.5 \mu\text{Wh}/\text{cm}^2$ under a discharging current of 0.25 mA and the best achieved power density of $1.333 \text{ mW}/\text{cm}^2$ under a discharging current of 5 mA. The nanocomposite materials increased the energy density. By contrast, the SCs with rGO/CS and PPy/CS nanocomposites showed poor performance, indicating the advantages of mixing rGOs, PPy, and CS to form nanocomposites.

3.8. CV Stability and Bending Tests

Figure 9a shows the CV stability test results of the SCs with a potential scanning rate of 200 mV/s. After 10,000 cycles, rGO/CS SC retains 90% of the initial areal capacitance, whereas PPy/CS retains only 43% owing to the poor PPy stability in long-term operation. rGO/PPy/CS SC retains 88% of the initial capacitance value. Adding rGOs to PPy/CS clearly improves the CV stability.

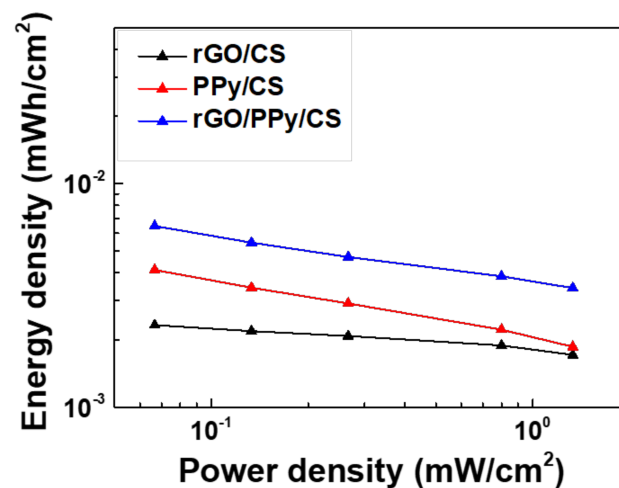


Figure 8. Ragone's plots of SCs with rGO/CS, PPy/CS and rGO/PPy/CS nanocomposite.

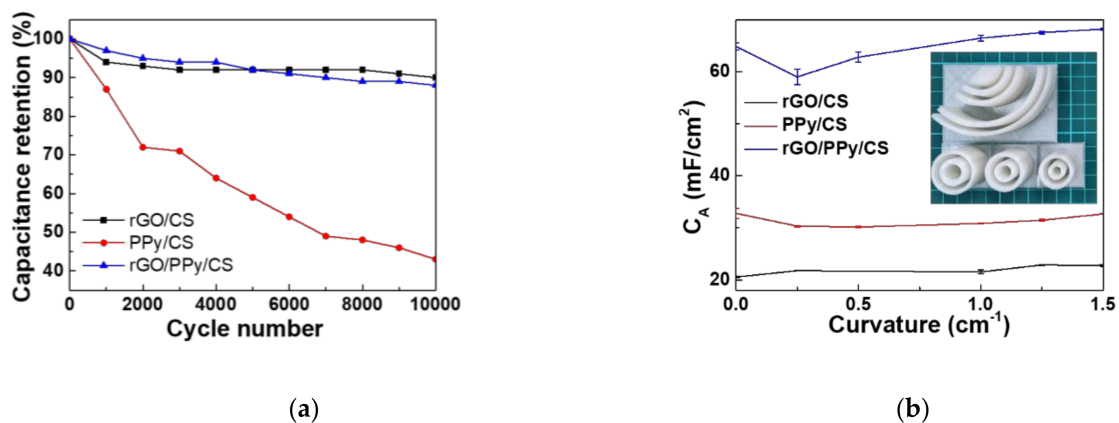


Figure 9. (a) Cycling stability test with a potential scan rate of 200 mV/s. (b) Bending test under different curvatures with the discharging current of 1 mA.

Figure 9b shows the bending test results of the SCs under different curvatures. The capacitance values are measured by GCD under the discharging current of 1 mA. The capacitance values fluctuate randomly as the curvature increases. Interestingly, the capacitance values increase at a certain curvature. All SCs function well under bending.

4. Conclusions

This study evaluates the screen-printed rGO/CS, PPy/CS, and rGO/PPy/CS SCs with active materials of the same weight. The GCD results show that the areal capacitance of rGO/CS, PPy/CS, and rGO/PPy/CS SCs is 26.23, 46.34, and 72.79 mF/cm², respectively. Combining rGO and PPy clearly results in improved areal capacitance and improved 10,000-cycle CV stability. The EIS results indicate that the SC with rGO/CS nanocomposites is more EDL-like, whereas the one with PPy/CS nanocomposites is more ion-diffusion-like. Trasatti analysis indicates that the addition of rGOs increases the contribution of EDLC. All three types of SCs perform well under bending.

Supplementary Materials: The following are available online at <https://www.mdpi.com/article/10.3390/polym13203585/s1>. Figure S1. XPS analysis of electrodes for (a) rGO/CS, (b) PPy/CS, and (c) rGO/PPy/CS pastes on carbon cloth. Table S1. Atomic ratios of C, O, N, Na, S, and Si as calculated from XPS spectra. Figure S2. XPS spectra of C1s for (a) rGO/CS, (b) PPy/CS, and (c) rGO/PPy/CS paste on carbon cloth. Table S2. XPS C1s spectra of electrode with rGO/CS, PPy/CS, and rGO/PPy/CS pastes on carbon cloth. Figure S3. XPS spectra of N1s for (a) rGO/CS, (b) PPy/CS,

and (c) rGO/PPy/CS paste on carbon cloth. Table S3. XPS N1s spectra of electrode with rGO/CS, PPy/CS, and rGO/PPy/CS pastes on carbon cloth. Figure S4. XPS spectra of O1s for (a) rGO/CS, (b) PPy/CS, and (c) rGO/PPy/CS pastes on carbon cloth. Table S4. XPS O1s spectra of electrode with rGO/CS, PPy/CS, and rGO/PPy/CS pastes on carbon cloth. Table S5. Comparison of areal capacitance of SCs with different materials listed in literatures.

Author Contributions: Conceptualization, J.-Z.C., I.-C.C. (I-Chun Cheng) and C.-C.H.; methodology, J.-Z.C., I.-C.C. (I-Chun Cheng), I.-C.C. (I-Chun Cheng) and C.-C.H.; software, C.L.; validation, J.-Z.C., C.L. and C.-W.H.; investigation, C.L.; data curation, C.L.; writing—original draft preparation, C.L.; writing—review and editing, J.-Z.C.; supervision, J.-Z.C. All authors have read and agreed to the published version of the manuscript.

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References

1. Tseng, C.-H.; Hsin, J.-C.; Tsai, J.-H.; Chen, J.-Z. Dielectric-barrier-discharge jet treated flexible supercapacitors with carbon cloth current collectors of long-lasting hydrophilicity. *J. Electrochem. Soc.* **2020**, *167*, 116511. [[CrossRef](#)]
2. Liu, Y.; Murtaza, I.; Shuja, A.; Meng, H. Interfacial modification for heightening the interaction between PEDOT and substrate towards enhanced flexible solid supercapacitor performance. *Chem. Eng. J.* **2020**, *379*, 122326. [[CrossRef](#)]
3. Guan, B.; Li, Y.; Yin, B.; Liu, K.; Wang, D.; Zhang, H.; Cheng, C. Synthesis of hierarchical NiS microflowers for high performance asymmetric supercapacitor. *Chem. Eng. J.* **2017**, *308*, 1165–1173. [[CrossRef](#)]
4. Liu, L.; Ye, D.; Yu, Y.; Liu, L.; Wu, Y. Carbon-based flexible micro-supercapacitor fabrication via mask-free ambient micro-plasma-jet etching. *Carbon* **2017**, *111*, 121–127. [[CrossRef](#)]
5. Meng, J.; Nie, W.; Zhang, K.; Xu, F.; Ding, X.; Wang, S.; Qiu, Y. Enhancing electrochemical performance of graphene fiber-based supercapacitors by plasma treatment. *ACS Appl. Mater. Interfaces* **2018**, *10*, 13652–13659. [[CrossRef](#)]
6. Zhang, X.; Wang, W.; Lu, J.; Hua, L.; Heng, J. Reversible heat of electric double-layer capacitors during galvanostatic charging and discharging cycles. *Thermochim. Acta* **2016**, *636*, 1–10. [[CrossRef](#)]
7. Chien, H.-H.; Cheng, Y.-C.; Hao, Y.-C.; Hsu, C.-C.; Cheng, I.-C.; Yu, S.; Chen, J.-Z. Nitrogen DC-pulse atmospheric-pressure-plasma jet (APPJ)-processed reduced graphene oxide (rGO)-carbon black (CB) nanocomposite electrodes for supercapacitor applications. *Diam. Relat. Mater.* **2018**, *88*, 23–31. [[CrossRef](#)]
8. Yang, C.-H.; Kuok, F.-H.; Liao, C.-Y.; Wan, T.-H.; Chen, C.-W.; Hsu, C.-C.; Cheng, I.-C.; Chen, J.-Z. Flexible reduced graphene oxide supercapacitor fabricated using a nitrogen dc-pulse atmospheric-pressure plasma jet. *Mater. Res. Express* **2017**, *4*, 025504. [[CrossRef](#)]
9. Chen, J.-Z.; Wang, C.; Hsu, C.-C.; Cheng, I.-C. Ultrafast synthesis of carbon-nanotube counter electrodes for dye-sensitized solar cells using an atmospheric-pressure plasma jet. *Carbon* **2016**, *98*, 34–40. [[CrossRef](#)]
10. Wang, J.-G.; Yang, Y.; Huang, Z.-H.; Kang, F. Synthesis and electrochemical performance of MnO₂/CNTs-embedded carbon nanofibers nanocomposites for supercapacitors. *Electrochim. Acta* **2012**, *75*, 213–219. [[CrossRef](#)]
11. Hosseini, M.G.; Shahryari, E. A novel high-performance supercapacitor based on chitosan/graphene oxide-MWCNT/polyaniline. *J. Colloid Interface Sci.* **2017**, *496*, 371–381. [[CrossRef](#)]
12. Chien, H.-H.; Liao, C.-Y.; Hao, Y.-C.; Hsu, C.-C.; Cheng, I.-C.; Yu, S.; Chen, J.-Z. Improved performance of polyaniline/reduced-graphene-oxide supercapacitor using atmospheric-pressure-plasma-jet surface treatment of carbon cloth. *Electrochim. Acta* **2018**, *260*, 391–399. [[CrossRef](#)]
13. Antolini, E. Graphene as a new carbon support for low-temperature fuel cell catalysts. *Appl. Catal. B—Environ.* **2012**, *123*, 52–68. [[CrossRef](#)]

14. Yin, Z.; Zhu, J.; He, Q.; Cao, X.; Tan, C.; Chen, H.; Yan, Q.; Zhang, H. Graphene-based materials for solar cell applications. *Adv. Energy Mater.* **2014**, *4*, 1300574. [[CrossRef](#)]
15. Arivu, M.; Masud, J.; Umaphathi, S.; Nath, M. Facile synthesis of Ni₃B/rGO nanocomposite as an efficient electrocatalyst for the oxygen evolution reaction in alkaline media. *Electrochem. Commun.* **2018**, *86*, 121–125. [[CrossRef](#)]
16. Rames, S.; Yadav, H.M.; Lee, Y.J.; Hong, G.W.; Kathalingam, A.; Sivasamy, A.; Kim, H.S.; Kim, H.S.; Kim, J.H. Porous materials of nitrogen doped graphene oxide@SnO₂ electrode for capable supercapacitor application. *Sci. Rep.* **2019**, *9*. [[CrossRef](#)]
17. Zheng, W.; Li, S.; Yu, X.; Chen, C.; Huang, H.; Huang, Y.; Li, L. Synthesis of hierarchical reduced graphene oxide/SnO₂/polypyrrole ternary composites with high electrochemical performance. *Mater. Res. Bull.* **2016**, *80*, 303–308. [[CrossRef](#)]
18. Lim, S.P.; Huang, N.M.; Lim, H.N. Solvothermal synthesis of SnO₂/graphene nanocomposites for supercapacitor application. *Ceram. Int.* **2013**, *39*, 6647–6655. [[CrossRef](#)]
19. Xiang, C.; Li, M.; Zhi, M.; Manivannan, A.; Wu, N. A reduced graphene oxide/Co₃O₄ composite for supercapacitor electrode. *J. Power Sources* **2013**, *226*, 65–70. [[CrossRef](#)]
20. An, H.; Wang, Y.; Wang, X.; Zheng, L.; Wang, X.; Yi, L.; Bai, L.; Zhang, X. Polypyrrole/carbon aerogel composite materials for supercapacitor. *J. Power Sources* **2010**, *195*, 6964–6969. [[CrossRef](#)]
21. Ramya, R.; Sivasubramanian, R.; Sangaranarayanan, M. Conducting polymers-based electrochemical supercapacitors—Progress and prospects. *Electrochim. Acta* **2013**, *101*, 109–129. [[CrossRef](#)]
22. Snook, G.A.; Kao, P.; Best, A.S. Conducting-polymer-based supercapacitor devices and electrodes. *J. Power Sources* **2011**, *196*, 1–12. [[CrossRef](#)]
23. Liu, Y.; Qu, G.; Sun, Q.; Jia, H.; Wang, T.; Zhu, L. Endogenously activated persulfate by non-thermal plasma for Cu(II)-EDTA decomplexation: Synergistic effect and mechanisms. *Chem. Eng. J.* **2021**, *406*, 126774. [[CrossRef](#)]
24. Bai, B.C.; Lee, H.-U.; Lee, C.W.; Lee, Y.-S.; Im, J.S. N₂ plasma treatment on activated carbon fibers for toxic gas removal: Mechanism study by electrochemical investigation. *Chem. Eng. J.* **2016**, *306*, 260–268. [[CrossRef](#)]
25. Zhang, T.; Wu, J.; Chen, J.; Pan, Q.; Wang, X.; Zhong, H.; Tao, R.; Yan, J.; Hu, Y.; Ye, X. Activating Titanium Metal with H₂ Plasma for the Hydrogen Evolution Reaction. *ACS Appl. Mater. Interfaces* **2021**, *13*, 24682–24691. [[CrossRef](#)]
26. Tao, L.; Duan, X.; Wang, C.; Duan, X.; Wang, S. Plasma-engineered MoS₂ thin-film as an efficient electrocatalyst for hydrogen evolution reaction. *Chem. Commun.* **2015**, *51*, 7470–7473. [[CrossRef](#)] [[PubMed](#)]
27. Abuzairi, T.; Okada, M.; Mochizuki, Y.; Poespawati, N.R.; Purnamaningsih, R.W.; Nagatsu, M. Maskless functionalization of a carbon nanotube dot array biosensor using an ultrafine atmospheric pressure plasma jet. *Carbon* **2015**, *89*, 208–216. [[CrossRef](#)]
28. Hsu, C.-T.; Tsai, J.-H.; Huang, T.-M.; Hsin, J.-C.; Chen, J.-Z.; Lee, B.-S.; Yang, T.-C. Atmospheric pressure plasma jet treatment enhances the effect of Alloy Primer on the bond strength between polymethyl methacrylate and stainless steels: Application for retention of magnetic attachment to resin denture base. *Colloids Surf. B Biointerfaces* **2021**, *197*, 111440. [[CrossRef](#)]
29. Puač, N.; Gherardi, M.; Shiratani, M. Plasma agriculture: A rapidly emerging field. *Plasma Process. Polym.* **2018**, *15*, 1700174. [[CrossRef](#)]
30. Graves, D.B. Low temperature plasma biomedicine: A tutorial review. *Phys. Plasmas* **2014**, *21*, 080901. [[CrossRef](#)]
31. Chen, J.-Z.; Hsu, C.-C.; Wang, C.; Liao, W.-Y.; Wu, C.-H.; Wu, T.-J.; Liu, H.-W.; Chang, H.; Lien, S.-T.; Li, H.-C. Rapid atmospheric-pressure-plasma-jet processed porous materials for energy harvesting and storage devices. *Coatings* **2015**, *5*, 26–38. [[CrossRef](#)]
32. Gerullis, S.; Kretzschmar, B.S.M.; Pfuch, A.; Beier, O.; Beyer, M.; Grünler, B. Influence of atmospheric pressure plasma jet and diffuse coplanar surface barrier discharge treatments on wood surface properties: A comparative study. *Plasma Process. Polym.* **2018**, *15*, 1800058. [[CrossRef](#)]
33. Alotaibi, F.; Tung, T.T.; Nine, M.J.; Kabiri, S.; Moussa, M.; Tran, D.N.; Losic, D. Scanning atmospheric plasma for ultrafast reduction of graphene oxide and fabrication of highly conductive graphene films and patterns. *Carbon* **2018**, *127*, 113–121. [[CrossRef](#)]
34. Kuok, F.-H.; Liao, C.-Y.; Wan, T.-H.; Yeh, P.-W.; Cheng, I.-C.; Chen, J.-Z. Atmospheric pressure plasma jet processed reduced graphene oxides for supercapacitor application. *J. Alloys Compd.* **2017**, *692*, 558–562. [[CrossRef](#)]
35. Chen, J.-Z.; Liao, W.-Y.; Hsieh, W.-Y.; Hsu, C.-C.; Chen, Y.-S. All-vanadium redox flow batteries with graphite felt electrodes treated by atmospheric pressure plasma jets. *J. Power Sources* **2015**, *274*, 894–898. [[CrossRef](#)]
36. Chen, Y.; Zhang, X.; Yu, P.; Ma, Y. Stable dispersions of graphene and highly conducting graphene films: A new approach to creating colloids of graphene monolayers. *Chem. Commun.* **2009**, 4527–4529. [[CrossRef](#)] [[PubMed](#)]
37. Xu, J.; Wang, D.; Yuan, Y.; Wei, W.; Duan, L.; Wang, L.; Bao, H.; Xu, W. Polypyrrole/reduced graphene oxide coated fabric electrodes for supercapacitor application. *Org. Electron.* **2015**, *24*, 153–159. [[CrossRef](#)]
38. Chang, J.-H.; Lin, M.-F.; Kuo, Y.-L.; Yang, C.-R.; Chen, J.-Z. Flexible rGO-SnO₂ supercapacitors converted from pastes containing SnCl₂ liquid precursor using atmospheric-pressure plasma jet. *Ceram. Int.* **2021**, *47*, 1651–1659. [[CrossRef](#)]
39. Keawploy, N.; Venkatkarthick, R.; Wangyao, P.; Zhang, X.; Liu, R.; Qin, J. Eco-friendly conductive cotton-based textile electrodes using silver-and carbon-coated fabrics for advanced flexible supercapacitors. *Energy Fuels* **2020**, *34*, 8977–8986. [[CrossRef](#)]
40. Li, Y.; Zhang, Y.; Zhang, H.; Xing, T.-L.; Chen, G.-Q. A facile approach to prepare a flexible sandwich-structured supercapacitor with rGO-coated cotton fabric as electrodes. *RSC Adv.* **2019**, *9*, 4180–4189. [[CrossRef](#)]
41. Dubal, D.P.; Lee, S.H.; Kim, J.G.; Kim, W.B.; Lokhande, C.D. Porous polypyrrole clusters prepared by electropolymerization for a high performance supercapacitor. *J. Mater. Chem.* **2012**, *22*, 3044–3052. [[CrossRef](#)]

42. Ardizzone, S.; Fregonara, G.; Trasatti, S. "Inner" and "outer" active surface of RuO₂ electrodes. *Electrochim. Acta* **1990**, *35*, 263–267. [[CrossRef](#)]
43. Tiwari, P.; Jaiswal, J.; Chandra, R. Hierarchical growth of MoS₂@ CNT heterostructure for all solid state symmetric supercapacitor: Insights into the surface science and storage mechanism. *Electrochim. Acta* **2019**, *324*, 134767. [[CrossRef](#)]
44. Huang, C.; Zhang, J.; Young, N.P.; Snaith, H.J.; Grant, P.S. Solid-state supercapacitors with rationally designed heterogeneous electrodes fabricated by large area spray processing for wearable energy storage applications. *Sci. Rep.* **2016**, *6*, 1–15. [[CrossRef](#)]
45. Shao, J.; Zhou, X.; Liu, Q.; Zou, R.; Li, W.; Yang, J.; Hu, J. Mechanism analysis of the capacitance contributions and ultralong cycling-stability of the isomorphous MnO₂@ MnO₂ core/shell nanostructures for supercapacitors. *J. Mater. Chem. A* **2015**, *3*, 6168–6176. [[CrossRef](#)]
46. Gund, G.S.; Dubal, D.P.; Chodankar, N.R.; Cho, J.Y.; Gomez-Romero, P.; Park, C.; Lokhande, C.D. Low-cost flexible supercapacitors with high-energy density based on nanostructured MnO₂ and Fe₂O₃ thin films directly fabricated onto stainless steel. *Sci. Rep.* **2015**, *5*, 1–13. [[CrossRef](#)]
47. Zhang, Y.; Liu, Y.; Bai, Y.; Liu, Y.; Xie, E. Boosting the electrochemical properties of carbon materials as bipolar electrodes by introducing oxygen functional groups. *RSC Adv.* **2020**, *10*, 35295–35301. [[CrossRef](#)]