

Structure and Role of a Ga-Promoter in Ni-Based Catalysts for the Selective Hydrogenation of CO₂ to Methanol

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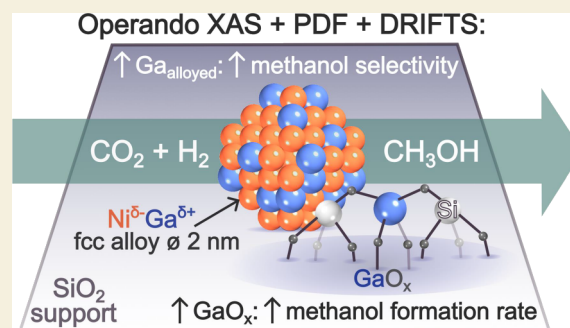
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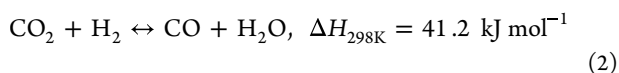
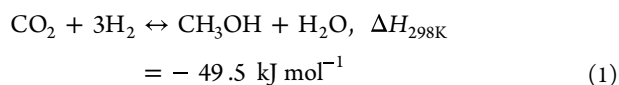
ABSTRACT: Supported, bimetallic catalysts have shown great promise for the selective hydrogenation of CO₂ to methanol. In this study, we decipher the catalytically active structure of Ni–Ga-based catalysts. To this end, model Ni–Ga-based catalysts, with varying Ni:Ga ratios, were prepared by a surface organometallic chemistry approach. In situ differential pair distribution function (d-PDF) analysis revealed that catalyst activation in H₂ leads to the formation of nanoparticles based on a Ni–Ga face-centered cubic (fcc) alloy along with a small quantity of GaO_x. Structure refinements of the d-PDF data enabled us to determine the amount of both alloyed Ga and GaO_x species. In situ X-ray absorption spectroscopy experiments confirmed the presence of alloyed Ga and GaO_x and indicated that alloying with Ga affects the electronic structure of metallic Ni (viz., Ni^{δ−}). Both the Ni:Ga ratio in the alloy and the quantity of GaO_x are found to minimize methanation and to determine the methanol formation rate and the resulting methanol selectivity. The highest formation rate and methanol selectivity are found for a Ni–Ga alloy having a Ni:Ga ratio of ~75:25 along with a small quantity of oxidized Ga species (0.14 mol_{GaO_x} mol_{Ni}^{−1}). Furthermore, operando infrared spectroscopy experiments indicate that GaO_x species play a role in the stabilization of formate surface intermediates, which are subsequently further hydrogenated to methoxy species and ultimately to methanol. Notably, operando XAS shows that alloying between Ni and Ga is maintained under reaction conditions and is key to attaining a high methanol selectivity (by minimizing CO and CH₄ formation), while oxidized Ga species enhance the methanol formation rate.

KEYWORDS: bimetallic catalysts, CO₂ hydrogenation, nickel, gallium, in situ, operando, X-ray absorption spectroscopy, X-ray total scattering, pair distribution function



1. INTRODUCTION

The selective hydrogenation of CO₂ to methanol (eq 1) enables us to convert a greenhouse gas, CO₂, into a value-added product of high global demand, making this process a sustainable alternative to the commercial methanol synthesis based on syngas and reforming technology.¹ Notably, the industrial catalyst used in the commercial process, viz., Cu/ZnO/Al₂O₃, suffers from deactivation and low methanol selectivity when used under CO₂ hydrogenation conditions (eq 2).^{1,2} The main causes for catalyst deactivation include sintering of the active Cu⁰ nanoparticles³ and ZnO,⁴ oxidation of Cu,⁴ and poisoning of the active sites by hydroxyl groups at high concentrations of H₂O or CO₂.⁵



In that context, research efforts have been undertaken to improve Cu-based catalysts,⁶ as well as explore alternative metal-based catalysts such as Ni, Pd, and Au,^{2,7,8} focusing on understanding the nature of the active sites and the role of additional metals/metal oxides, which is critical to allow for a knowledge-driven catalyst design. For example, Ga-promoted transition metal catalysts, such as Cu–Ga,^{9–12} Ni–Ga,^{13–17} and Pd–Ga,^{18–22} have shown promising activity for the CO₂ hydrogenation to methanol. In these catalysts, Ga was present in the form of an alloy that in many cases^{9,12,14,18} coexisted with oxidized Ga species under reaction conditions. The promotional effect of Ga in Ni is particularly noteworthy as Ni-based catalysts are typically very effective in the methanation reaction (eq 3).²³

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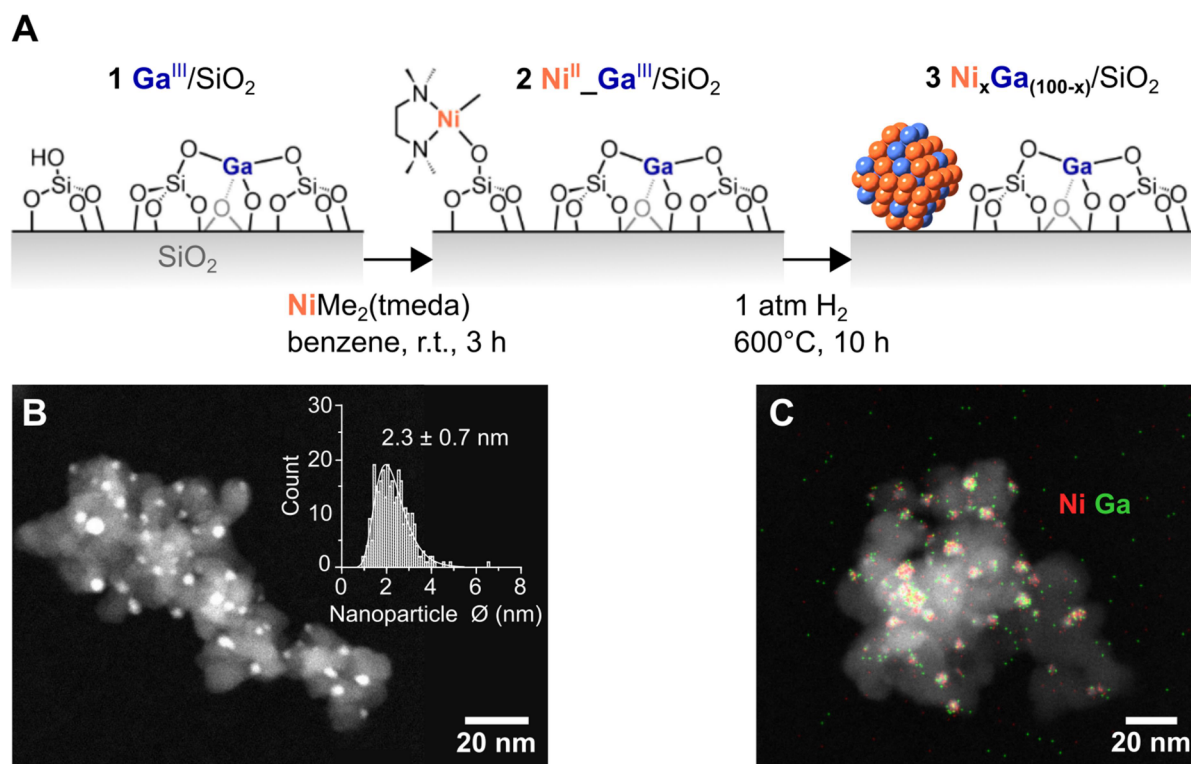
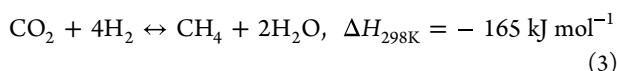


Figure 1. (A) Synthesis of bimetallic $\text{Ni}_x\text{Ga}_{(100-x)}/\text{SiO}_2$ catalysts. (B) STEM image with the corresponding particle size distribution and (C) STEM-EDX image with overlaid Ni and Ga EDX signals of as-reduced $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$, selected here as a representative example for the TEM images of all $\text{Ni}_x\text{Ga}_{(100-x)}/\text{SiO}_2$ presented in the Supporting Information.

The introduction of Ga, forming Ni–Ga intermetallics (ordered alloys), has been shown however to improve the selectivity toward methanol, reaching up to ca. 60%.¹⁴



This finding has triggered research into understanding the role of Ga in the CO_2 hydrogenation pathway, the relationship between Ni–Ga phases and catalyst performance, and ultimately the structure of the active sites in this promising catalyst family.^{13–15,24,25} Previous works based on density functional theory (DFT) calculations have linked the $\delta\text{-Ni}_5\text{Ga}_3$ phase to high methanol yields,¹³ while more recent studies on Ga-doped Ni(211) surfaces have argued that Ni is the active site and that alloying with Ga modifies the electronic structure of Ni through an electron transfer from Ga, promoting in turn the formation of oxygenates (methanol and CO) over methane.^{15,26,27}

Despite these previous research efforts, the structure of the active sites in the Ni–Ga system has not been elucidated unequivocally. Two major challenges have limited progress in answering this central research question, viz., the lack of model catalyst systems with precise control over size, phase, and composition (i.e., Ni:Ga ratio) and the lack of detailed information concerning the catalyst structure (electronic and geometric) under operando conditions. With regard to the synthesis of Ni–Ga catalysts, previous works reported challenges in obtaining well-defined nanoparticles of a single phase as conventional approaches such as impregnation lead typically to catalysts containing multiple phases (e.g., both $\delta\text{-Ni}_5\text{Ga}_3$ and $\alpha'\text{-Ni}_3\text{Ga}$),^{14,16,25,28} or mixtures of alloys and gallium oxide, which makes it impossible to identify the

catalytically active motif.^{14,25,29,30} Furthermore, it has been reported that the oxidation of Ga to Ga_2O_3 species can promote methanol formation.¹⁴ Indeed, dealloying under CO_2 hydrogenation conditions has been reported for related bimetallic M–M' systems (Cu–Ga,^{9,12} Cu–Zn,³¹ and Pd–Ga¹⁸), generating a mixture of both metallic M (Cu/Pd) and (partially) oxidic M' (Ga/Zn) species. Similarly, the ratio of M/M' in bimetallic catalysts has been shown to have a pronounced influence on the catalyst activity, transitioning from promoting to poisoning effects.¹⁰ In the specific case of Ni–Ga, the $\delta\text{-Ni}_5\text{Ga}_3$ phase has been proposed to be particularly active for CO_2 hydrogenation to methanol, putting forward questions related to the optimal Ni:Ga ratio as well as the stability and the role of the alloy and/or the oxide interface on methanol selectivity and formation rate.

To shed light on these questions, we synthesized tailored Ni–Ga-based catalysts with varying Ni:Ga ratios and constant particle sizes ($\sim 2 \text{ nm}$) using surface organometallic chemistry (SOMC)^{32,33} and thermolytic molecular precursor (TMP)³⁴ approaches and interrogated (quantitatively) their structure under operando conditions using a combination of X-ray-based characterization techniques. In situ and operando differential pair distribution function (d-PDF) analysis of X-ray total scattering data and X-ray absorption spectroscopy (XAS) revealed that after activation in H_2 , the most selective catalyst contained nanoparticles with an fcc Ni–Ga alloy structure with a ratio Ni:Ga = 75:25 and a small quantity of oxidized Ga species, GaO_x ($0.14 \text{ mol}_{\text{GaO}_x} \text{ mol}_{\text{Ni}}^{-1}$). The presence of GaO_x appreciably increased the methanol formation rate while maintaining a high methanol selectivity when compared with a catalyst that contains an identical Ni–Ga alloy composition (Ni:Ga = 75:25) but a smaller GaO_x content ($0.06 \text{ mol}_{\text{GaO}_x} \text{ mol}_{\text{Ni}}^{-1}$). The structure of the catalysts formed after activation

Table 1. Catalyst Composition, as Determined by ICP, Molar Ratio of Ni:Ga (–) in the fcc-Ni_xGa_(100–x) Alloy, and Molar Amounts of GaO_x and Ga_{alloyed} Normalized by the Catalysts' Total Ni Content, as Determined by PDF and ICP Analyses

Catalyst	x_{ICP} in Ni _x Ga _(100–x) (–)	Molar ratio of Ni:Ga in the fcc-Ni _x Ga _(100–y) alloy (–)	Ga _{alloyed} (mol _{Ga_{alloyed}} mol _{Ni} ^{–1})	GaO _x (mol _{GaO_x} mol _{Ni} ^{–1})
Ni ₆₅ Ga ₃₅ /SiO ₂	67.5	74:26	0.34	0.14
Ni ₇₀ Ga ₃₀ /SiO ₂	71.9	75:25	0.33	0.06
Ni ₇₅ Ga ₂₅ /SiO ₂	76.4	82:18	0.23	0.08

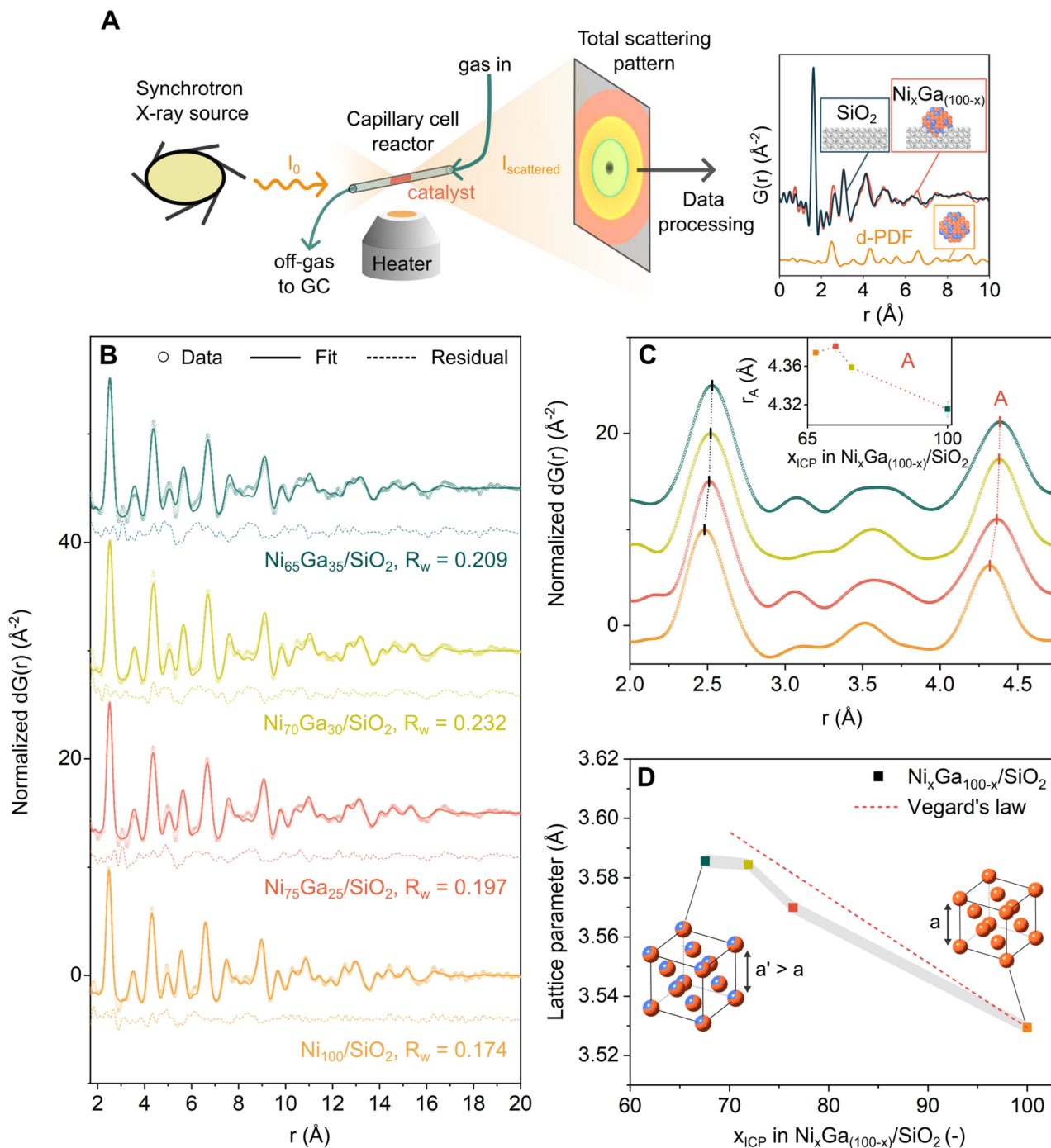


Figure 2. (A) Schematic representation of a X-ray total scattering experiment and the resulting d-PDF data. (B) d-PDF data fitted to a fcc-Ni_xGa_(100–y) alloy. (C) Zoom into the region $r = 2$ –4.75 Å of the d-PDF data. The inset plots the fitted position of the metal–metal pair correlation labeled A as a function of x_{ICP} in the catalysts. (D) Cubic lattice parameter as a function of the fraction of Ni, as determined by elemental analysis. The error bars are represented by the area shaded in gray. Conditions: 1 bar H₂, 230 °C, after in situ activation.

was maintained under reaction conditions (20 bar CO₂:H₂:N₂ = 1:3:1, 230 °C), i.e., the quantities of Ga alloyed and GaO_x

species remained constant. Furthermore, operando diffuse reflectance infrared Fourier transform spectroscopy (DRIFTS)

provided additional insight into the surface species under reaction conditions whereby only the most active catalyst showed bands due to formate species. Combining these findings, we were able to conclude that the alloying of Ni with Ga is key to attaining a high methanol selectivity, while the presence of oxidized Ga species enhances appreciably the rate of methanol formation.

2. RESULTS AND DISCUSSION

2.1. Catalyst Synthesis

We prepared a series of silica-supported Ni–Ga-based catalysts with varying Ni:Ga ratios via a SOMC-TMP approach (Figure 1A).³⁵ Briefly, in all of the Ni-containing materials, Ni was introduced by grafting $[\text{Ni}(\text{CH}_3)_2(\text{tmeda})]$ onto the surface OH groups of $\text{Ga}^{\text{III}}/\text{SiO}_2$ (or SiO_2 , in case of the monometallic Ni material), whereby $\text{Ga}^{\text{III}}/\text{SiO}_2$ was produced by grafting $[\text{Ga}(\text{OSi}(\text{OtBu})_3)_3(\text{THF})]$ onto SiO_2 as reported previously.³⁶ All materials were subsequently treated under H_2 at 600 °C for 12 h, which yielded the as-prepared catalysts, denoted as $\text{Ni}_x\text{Ga}_{(100-x)}/\text{SiO}_2$, where x is the nominal catalyst composition. The nominal Ni loading was kept constant at ca. 2 wt %, while the Ga loading was varied to obtain the desired nominal ratios of Ni:Ga (100:0, 75:25, 70:30, 65:35). The reference material $\text{Ga}_{100}/\text{SiO}_2$ (ca. 0.9 wt % Ga) was obtained by treating $\text{Ga}^{\text{III}}/\text{SiO}_2$ under H_2 at 600 °C for 12 h. The final composition of the as-prepared catalysts was determined by elemental analysis (inductively coupled plasma optical emission spectroscopy, ICP-OES) and will be denoted as x_{ICP} (Table 1 and Supporting Information Table S1).

Ex situ high-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM) images of the as-prepared materials recorded under air-free conditions showed well-dispersed nanoparticles on the SiO_2 support with an average diameter of around 2 nm (Figure 1B and Supporting Information Figures S1, S5, S8, and S12). These STEM images further confirmed that the SOMC-based synthesis approach yielded catalysts of very similar particle size, independent of the Ni:Ga ratio. Hence, any change in product formation rates could not be attributed to changes in the surface area of the nanoparticles. Furthermore, STEM-EDX images of $\text{Ni}_{100}/\text{SiO}_2$, $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$, $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$, and $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ showed a spatial overlap between the Ni and Ga EDX signals in all of the as-prepared materials (Figure 1C and Supporting Information Figures S7, S10, and S14). Turning to the reference $\text{Ga}_{100}/\text{SiO}_2$, Ga was found to be highly dispersed with no visible nanoparticle formation (Supporting Information Figures S3 and S4), in line with previous studies.³⁵

2.2. Structure of the Catalysts after In Situ Activation

Prior to CO_2 hydrogenation experiments, the catalysts were activated in 1 bar H_2 at 600 °C for 1 h, and this process was monitored by in situ d-PDF and XAS. Here, we focus on the analysis of the catalyst structure obtained after their in situ activation. The evolution of the air-exposed catalysts during activation is described in the Supporting Information. d-PDF was obtained by subtracting the PDF signal of the support (i.e., SiO_2) from the PDF data of the entire catalyst (Figures 2A and S22).^{37,38} For the d-PDF analysis of in situ activated $\text{Ni}_{100}/\text{SiO}_2$, $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$, $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$, and $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$, synchrotron X-ray total scattering data were collected at the reaction temperature of 230 °C in 1 bar H_2 . The SiO_2 -subtracted X-ray total scattering and d-PDF data of the in situ activated catalysts are shown in Figures S21 and 2B (data collected at 230 °C). The

reciprocal space data of the activated catalysts showed broad, yet clear Bragg peaks for all of the catalysts that can be indexed, independent of the Ni:Ga ratio, as a face-centered cubic (fcc)-type structure (Supporting Information Figure S21). Modeling of the d-PDF confirmed that all the nanoparticles have an fcc type structure and revealed an average diameter of the nanoparticles of ca. 2 nm that was invariant to the catalysts' elemental composition, in line with STEM-EDX measurements (Supporting Information Figure S27). In addition, fitting of the d-PDF data revealed an increase in the cubic lattice parameter of the fcc- $\text{Ni}_y\text{Ga}_{(100-y)}$ alloy [where $y:(100-y)$ is the Ni:Ga ratio in the alloy] with increasing Ga content, which was attributed to the incorporation of Ga into the (nano)alloy structure, causing a tensile strain in the fcc lattice (Figure 2D, Supporting Information Table S3). In line with the formation of Ni–Ga alloys, d-PDF modeling also revealed an increase in the atomic displacement factors (ADFs) with respect to $\text{Ni}_{100}/\text{SiO}_2$ due to a broader distribution of interatomic distances in the alloys (Supporting Information Figure S28).³⁹ The agreement factors (R_w) in all of the fittings were in the range of 0.17–0.23 which are typical values for the PDF of small nanoparticles.^{40,41} It is possible that defects inside and/or on the surface of the nanoparticles led to some misfit between the experimental data and the calculated PDF (e.g., slight misfit in the intensity of the peaks between 5 and 7 Å, Figure 2B), which are often present in nanoparticles;^{42,43} however, the peak positions of the PDF are well explained by the models. The lattice parameters extracted from the fitting of the d-PDF data were independent of the r -range fitted (1.7–25 Å) as shown in Figures 2D and S31. This indicates that the local and midrange structures of the alloy are comparable. Concerning the detection of oxidized species by d-PDF, it has to be stated that the detection of Ga–O pairs by d-PDF is challenging (expected at ca. 1.7–2.0 Å)⁴⁴ due to the presence of termination (noise) ripples arising from the finite Q -range,⁴⁵ and the low scattering cross-section of oxygen atoms relative to that of the higher Z elements Ni and Ga.⁴⁶ Notably, the fcc Ni–Ga alloys (bulk) follow Vegard's law, i.e., there is a linear relationship between the Ni:Ga ratio and the cell parameter (Supporting Information Table S4).⁴⁷ However, when plotting the cell parameters extracted from the supported nanoparticles against the catalysts' composition determined by ICP (Figure 2D) together with the predicted cell parameters using Vegard's law (dashed curve), we observe that the determined cell parameters were lower than the values expected from Vegard's law, indicating that not all of the Ga in the catalyst was incorporated into the fcc structure of the (alloy) nanoparticles and hence remained as oxidized Ga species (GaO_x), as confirmed by XAS (vide infra). Thus, we next quantified the amount of alloyed Ga species ($\text{Ga}_{\text{alloyed}}$) via Vegard's law, and the amount of GaO_x as the difference between the catalyst's total Ga content measured by ICP and $\text{Ga}_{\text{alloyed}}$ (see Table 1 and the Supporting Information Section 3.1 for details). A key observation from this analysis was that the lattice parameters of the alloy phases in $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$ and $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ were equal within the error of the fitting, corresponding to an fcc alloy of composition Ni:Ga = 75:25 in both materials, despite the fact that $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ had a higher total Ga content than $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$ (Tables 1 and S1 in the Supporting Information). Thus, $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ contained a larger quantity of GaO_x (0.14 mol _{GaO_x} mol_{Ni}^{−1}) than $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$ (0.06 mol _{GaO_x} mol_{Ni}^{−1}). Furthermore, it is worth noting that the composition of the alloy nanoparticles in $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$ and

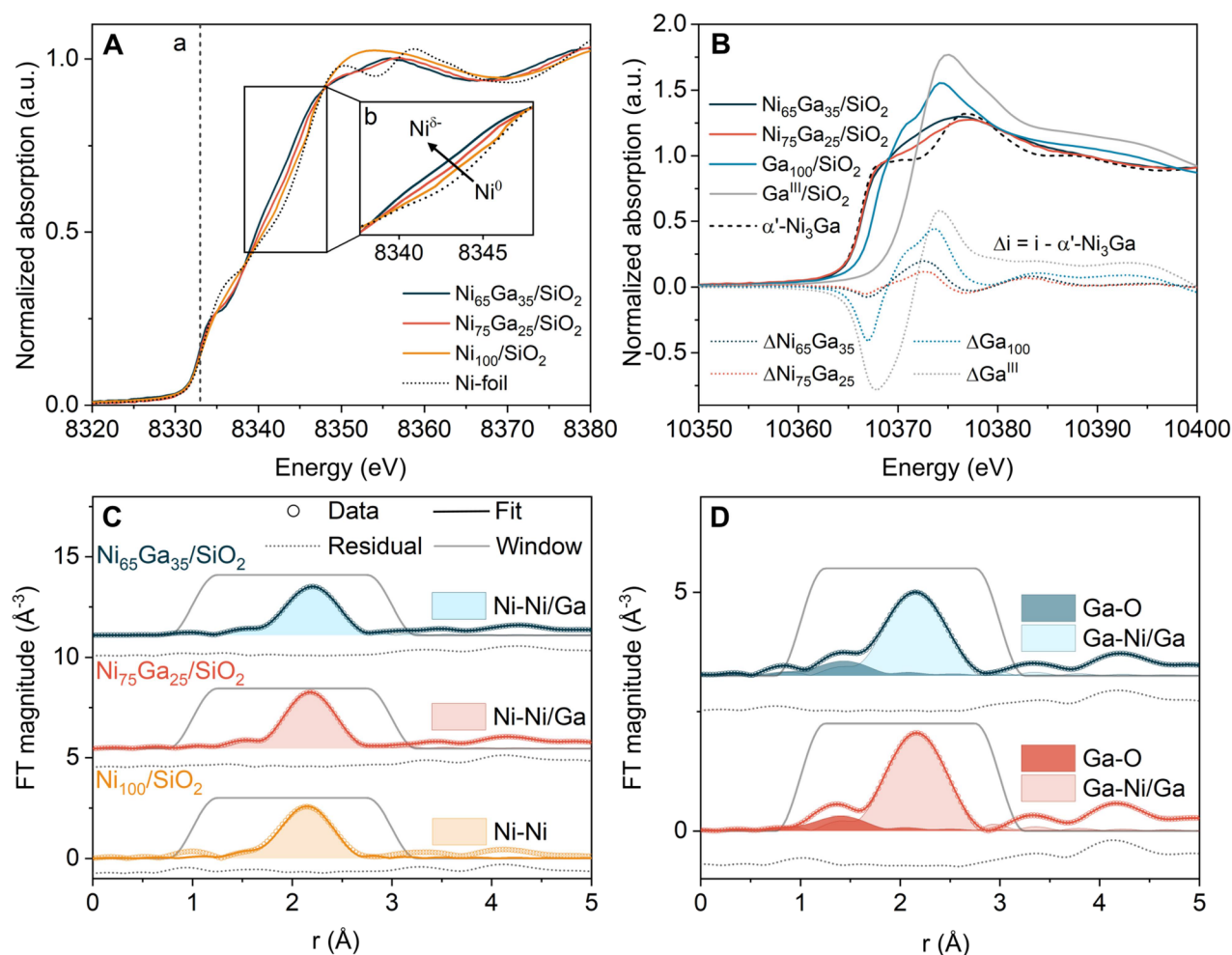


Figure 3. XAS data of the activated catalysts and reference materials: (A) normalized Ni K-edge XANES of $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$, $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$, and $\text{Ni}_{100}/\text{SiO}_2$ and a Ni-foil reference. Line “a” denotes the position of the maximum of the first derivative of the Ni K-edge XANES of Ni foil (8333 eV). Inset “b” shows a shift of the Ni absorption edge to lower energies with increasing Ga content, indicating the formation of $\text{Ni}^{\delta-}$ species upon alloying of Ni with Ga. (B) Normalized Ga K-edge XANES of $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$, $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$, and $\text{Ga}_{100}/\text{SiO}_2$, as well as the α' - Ni_3Ga and $\text{Ga}^{\text{III}}/\text{SiO}_2$ references. The dotted lines represent difference XANES ($\Delta i = i - \alpha' - \text{Ni}_3\text{Ga}$). Experimental and fitted Fourier transforms of the (C) Ni K-edge and (D) Ga K-edge EXAFS data (plots not corrected for the phase shift). The contributions of the individual Ni–M, Ga–M (M = Ni, Ga), and Ga–O coordination spheres to the fits are shown as shaded areas. Conditions: 1 bar H_2 , 50 °C, after in situ activation.

$\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ was the same as the composition of the (intermetallic) α' - Ni_3Ga phase. From our data, we could not exclude an ordering of the metals in the catalysts as in the α' - Ni_3Ga intermetallic phase due to the very similar structure of fcc alloys and α' - Ni_3Ga . The random fcc alloy (with a composition Ni:Ga = 75:25) differs from the ordered α' - Ni_3Ga only in the occupancy of Ni and Ga in the fcc sites: in the random alloy, Ga and Ni atoms randomly occupy the same sites, whereas in α' - Ni_3Ga , Ga sits only at the corners and Ni in the center of the faces of the cubic unit cell (see Supporting Information Figure S30). Due to the reduced symmetry of the α' - Ni_3Ga structure (being a superstructure of the fcc structure) compared to that of the fcc random alloy, additional low-intensity reflections would be expected in its diffraction pattern.⁴⁸ However, since Ni and Ga have very similar scattering properties, it is challenging to detect such weak reflections. Similarly, only tiny differences would be expected in the magnitude of the PDF peaks. However, previous reports have shown that intermetallic phases are only formed above a certain critical nanoparticle size,^{49–51} and thus, we hypothesize that it is unlikely that ordering took place in ~ 2 nm-sized (nano)alloys.

To probe the electronic and local structure of Ga and Ni, we performed XAS experiments at the Ni and Ga K-edges. XANES analysis provided insight into the metal oxidation states, while EXAFS yielded quantitative information concerning the local structure of Ni and Ga in the materials. In this context, it is worth noting that Ga K-edge XAS is more sensitive in probing for the presence of GaO_x species than d-PDF. In the in situ XAS experiments, the same capillary reactor cell was used as in the d-PDF experiments, allowing to directly confront the respective results. Due to limited synchrotron beamtime, the in situ XAS experiments were performed on the catalyst with the highest Ga content ($\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$), one with lower Ga content ($\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$) and the references $\text{Ni}_{100}/\text{SiO}_2$ and $\text{Ga}_{100}/\text{SiO}_2$ (collected ex situ in airtight capillaries).

The in situ acquired Ni K-edge XANES spectra of the catalysts (after activation and collected at 50 °C) are presented in Figure 3A. The edge position [determined as the maximum of the first derivative of the normalized absorption $\mu(E)$] was at ca. 8333 eV in all of the catalysts, indicating that Ni is in its metallic state, and no oxidized Ni species were present in the activated catalysts. Taking a closer look into the second edge feature labeled as “b”

revealed a shift to lower energies for $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ (by ca. -1.2 eV at $\mu_{\text{norm}} = 0.6$ au) and $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ (by ca. -0.5 eV at $\mu_{\text{norm}} = 0.6$ au) with respect to $\text{Ni}_{100}/\text{SiO}_2$. Moreover, the shape of the XANES features of $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ and $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ in the white line region was different from that of $\text{Ni}_{100}/\text{SiO}_2$ and the Ni foil. According to previous studies, the observed XANES features in $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ and $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ were interpreted as an electron transfer from Ga to Ni in the Ni–Ga alloy ($\text{Ga}^{\delta+}/\text{Ni}^{\delta-}$), whereby the electron transfer was most pronounced in $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ due to the higher content of Ga in the alloy, in line with d-PDF analysis (vide supra).^{52,53}

Turning to Ga K-edge XANES, the respective data of the activated catalysts are plotted in Figure 3B together with α' - Ni_3Ga as a reference for a Ni–Ga alloy and $\text{Ga}^{\text{III}}/\text{SiO}_2$ as a reference for a material containing only Ga^{3+} (where Ga^{3+} is in tetrahedral coordination with oxygen as reported previously).³⁶ In addition, we collected Ga K-edge XANES data of activated $\text{Ga}_{100}/\text{SiO}_2$, which also contains Ga^{3+} indicated by the white line feature at ca. 10375.0 eV. However, activated $\text{Ga}_{100}/\text{SiO}_2$ also exhibits a feature at ca. 10371.5 eV, which can be assigned to Ga^+ and/or Ga hydride species ($[\text{GaH}_2]^+$ or $[\text{GaH}]^{2+}$), that were formed during activation in H_2 .^{54,55} Here, we refer to this mixture of Ga species as GaO_x (representing a mixture of $\text{Ga}^{3+}/\text{Ga}^+/\text{Ga}$ hydride species).^{54,55} The energy positions of the Ga K-edge in $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ and $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ were within 0.1 eV of that of the bulk α' - Ni_3Ga reference, indicating that most Ga species were in an alloyed state. Moreover, the Ga K-edge XANES spectra of both catalysts exhibited two main features at 10368.0 and 10377.0 eV, which were due to Ga alloyed with Ni. In fact, these features were present in α' - Ni_3Ga and have been assigned to transitions to unoccupied Ga 4p states which are hybridized with Ni 3d/non-d bands of α' - Ni_3Ga above the Fermi level.⁵³ However, the white line feature of $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ and $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ (in the range 10370–10375 eV) had a higher intensity compared to that of α' - Ni_3Ga , suggesting the presence of a minor quantity of oxidized Ga species in the activated catalysts. Indeed, by evaluating the difference spectra (differences with respect to α' - Ni_3Ga , Figure 3B), we concluded that $\text{Ni}_x\text{Ga}_{(100-x)}/\text{SiO}_2$ contained predominantly $\text{Ga}_{\text{alloyed}}$ with some minor quantity of GaO_x species.^{54,56,57} Next, we performed linear combination fittings (LCFs) of the Ga K-edge XANES data of $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ and $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ using α' - Ni_3Ga and $\text{Ga}_{100}/\text{SiO}_2$ as references for the $\text{Ga}_{\text{alloyed}}$ and GaO_x species, respectively. LCF analysis yielded a ratio $\text{GaO}_x:\text{Ga}_{\text{alloyed}} = 31:69$ for $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ and $\text{GaO}_x:\text{Ga}_{\text{alloyed}} = 19:81$ for $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ (corresponding to 0.15 and 0.05 $\text{mol}_{\text{GaO}_x} \text{mol}_{\text{Ni}}^{-1}$ in $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ and $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$, respectively; Supporting Information Table S6 and Figures S33 and S34) in line with the values obtained in Table 1.

We note that linear combination using the α' - Ni_3Ga and $\text{Ga}_{100}/\text{SiO}_2$ references resulted in the best agreement between the experimental data and the fit when compared to using the combination α' - Ni_3Ga and $\text{Ga}^{\text{III}}/\text{SiO}_2$ or α' - Ni_3Ga and β - Ga_2O_3 , indicating the presence of GaO_x in $\text{Ni}_x\text{Ga}_{(100-x)}/\text{SiO}_2$. Supplementary LCF results including an extended selection of Ga K-edge XANES references (Ga-foil, β - Ga_2O_3) can be found in the Supporting Information (Table S6, Figures S33 and S34). However, as the quantitative analysis of Ga K-edge XANES data containing multiple (alloyed and oxidized) species can be subject to errors, as the Ga features are affected in a convoluted fashion by the oxidation state, electronic interactions (e.g., from alloying), as well as the coordination environment in the oxide and the size and phases of the alloyed nanoparticles, additional

EXAFS analysis was performed to probe in more detail the presence of Ga–O atomic pairs.

The k^2 -weighted EXAFS oscillations ($k^2X(k)$) and the corresponding magnitude of the Fourier transform (FT) at the Ni K-edge of $\text{Ni}_{100}/\text{SiO}_2$, $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$, and $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ are presented in Figures S38 and S40 in the Supporting Information and Figure 3C. We observed a first neighboring Ni–M shell ($M = \text{Ni, Ga}$) at ca. 2.2 Å and no evidence of a Ni–O shell, in line with XANES analysis. The EXAFS data at the Ga K-edge of $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ and $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ are presented in Figures 3D, S38 and S40 in the Supporting Information and evidenced a prominent Ga–M ($M = \text{Ni or Ga}$) shell and a weak Ga–O shell, indicative of a minor quantity of oxidized Ga species in line with XANES. Fitting the Ga and Ni K-edge EXAFS data allowed us to determine the average interatomic distances, coordination numbers (CN), and σ^2 (the mean square variation in path length also referred to as Debye Waller factors) of the first Ni–M, Ga–M, and Ga–O shells (Supporting Information Table S8). This analysis revealed an increase in the average Ni–M distance with increasing Ga content, i.e., $2.48 < 2.50 < 2.53$ Å for $\text{Ni}_{100}/\text{SiO}_2$, $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$, and $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$, respectively (Supporting Information Figure S41), in line with d-PDF analysis that showed an increase in the lattice parameter as Ga is incorporated into the Ni–Ga alloy fcc structure. The Ga–M distances were determined to be 2.52 Å in both $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ and $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$, which were close to the Ni–M distances determined by EXAFS fitting of the Ni K-edge data. The σ^2 values obtained for Ga–M were slightly higher than those for Ni–M, suggesting a somehow higher degree of disorder around Ga. The average CN for $\text{Ni}_{100}/\text{SiO}_2$ was 9(1) (number in parentheses represents the standard deviations obtained from the fittings), whereas in $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$, the $\text{CN}(\text{Ga–M})$ was close to $\text{CN}(\text{Ni–M})$ [i.e., $\text{CN}(\text{Ni–M}) = 9.0(6)$, $\text{CN}(\text{Ga–M}) = 9(2)$] in line with an homogeneous distribution of Ga within the nanoalloy (i.e., no Ga or Ni surface segregation). In $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$, the $\text{CN}(\text{Ni–M}) = 8.8(6)$ was slightly larger than $\text{CN}(\text{Ga–M}) = 8.1(8)$, which we attribute to a larger quantity of oxidized Ga species (GaO_x) in this catalyst, leading to a decrease in $\text{CN}(\text{Ga–M})$. Indeed, to fit the Ga K-edge EXAFS data of $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ and $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$, a Ga–O path was required. The Ga–O distances were at ca. 1.84–1.87 Å with a CN of 0.6(3) and 0.7(1) for $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ and $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$, respectively, providing further evidence for the presence of GaO_x species in both catalysts. However, it was not possible to quantify the amount of GaO_x species via EXAFS.

To summarize, the combined d-PDF and XAS analyses show that $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$, $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$, and $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ contained alloyed nanoparticles of ~ 2 nm in size with an fcc structure along with small quantities of GaO_x species. Using the extracted cell parameters and the ICP-determined Ni and Ga contents in the catalysts, the alloy composition and the quantity of GaO_x were determined. $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ and $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$ contained alloy nanoparticles with a Ni:Ga ratio of ca. 75:25, but $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ contained more GaO_x species (0.14 $\text{mol}_{\text{GaO}_x} \text{mol}_{\text{Ni}}^{-1}$) compared to $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$ (0.06 $\text{mol}_{\text{GaO}_x} \text{mol}_{\text{Ni}}^{-1}$). On the other hand, $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ contained an alloy with a higher Ni:Ga ratio of 82:18 compared to $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ and $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$ along with a GaO_x content of 0.08 $\text{mol}_{\text{GaO}_x} \text{mol}_{\text{Ni}}^{-1}$, i.e., in between that of $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ and $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$. XAS supported the findings of the d-PDF analysis in that Ni and Ga formed Ni–Ga alloys and that there were minor quantities of oxidized Ga species (ca. 0.05 and 0.15 $\text{mol}_{\text{GaO}_x}$

$\text{mol}_{\text{Ni}}^{-1}$ in $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ and $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$, respectively). Ni K-edge XANES data pointed to a different electronic structure (electron transfer from Ga to Ni) in the Ni–Ga catalysts compared to the monometallic Ni reference ($\text{Ni}_{100}/\text{SiO}_2$) due to the alloying of Ni with Ga.

2.3. Catalytic Performance and Role of Alloyed Ga and GaO_x

We determined the catalytic performance of the different materials under the following CO_2 hydrogenation conditions: fixed bed reactor, 25 bar, $\text{CO}_2:\text{N}_2:\text{H}_2 = 1:1:3$, 230 °C, and a gas hourly space velocity of 60 $\text{L gcat}^{-1} \text{h}^{-1}$. All product (methanol, CO, and CH_4) formation rates are normalized by the Ni content (Figure 4A) as Ni has been proposed as the active site whereby its electronic structure is modified by neighboring Ga atoms.^{15,26}

Due to the small and invariant size of the Ni–Ga nanoparticles in the catalysts (ca. 2 nm according to d-PDF analysis), corresponding to a surface:volume ratio of 3:1 nm^{-1} , we can infer that normalizing the product formation rates by the total Ni content (ca. 2 wt % in all catalysts) offers a meaningful basis for examining the impact of Ga addition on the catalytic activity of Ni-based nanoparticles. The catalytic performance per g of catalyst is reported in Table S1 in the Supporting Information. Interestingly, the rate of CO_2 conversion decreases first by ca. 75% when transitioning from $\text{Ni}_{100}/\text{SiO}_2$ (2.8 $\text{mmol}_{\text{CO}_2} \text{mol}_{\text{Ni}}^{-1} \text{s}^{-1}$) to $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ (0.7 $\text{mmol}_{\text{CO}_2} \text{mol}_{\text{Ni}}^{-1} \text{s}^{-1}$) and slightly increases again in catalysts with higher nominal Ga contents in the following order: $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2 < \text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2 < \text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ (Figure 4A). Notably, this change of CO_2 conversion rates was accompanied by a change in product selectivity, viz., from mostly methane when using pure Ni to methanol for Ni–Ga. The methanol formation rates of $\text{Ni}_{100}/\text{SiO}_2$ and $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ were very similar, i.e., ca. 0.2 $\text{mmol}_{\text{MeOH}} \text{mol}_{\text{Ni}}^{-1} \text{s}^{-1}$ but were significantly higher for $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$ (0.6 $\text{mmol}_{\text{MeOH}} \text{mol}_{\text{Ni}}^{-1} \text{s}^{-1}$) and $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ (1.1 $\text{mmol}_{\text{MeOH}} \text{mol}_{\text{Ni}}^{-1} \text{s}^{-1}$), while $\text{Ga}_{100}/\text{SiO}_2$ did not convert any CO_2 to any significant extent (Table S1). We would like to note here that methane and CO, but not methanol, have been typically reported products from CO_2 hydrogenation over monometallic Ni/ SiO_2 catalysts. We hypothesize that methanol formation over $\text{Ni}_{100}/\text{SiO}_2$ could be due to the presence of very small Ni clusters which interact strongly with the SiO_2 support and show a different selectivity profile compared to larger Ni nanoparticles.⁵⁸ Generally, a remarkable shift in methanol selectivity was observed upon the addition of Ga for the $\text{Ni}_x\text{Ga}_{(100-x)}/\text{SiO}_2$ series (Figure 4, Supporting Information Figure S17). Specifically, the methanol selectivity increased in the following order: $\text{Ni}_{100}/\text{SiO}_2$ (7%) < $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ (33%) < $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$ (48%) < $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ (54%). In parallel, while $\text{Ni}_{100}/\text{SiO}_2$ displayed a high methane selectivity (88%), the methane selectivity decreased to 11% in $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$, and no methane was observed for $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$ and $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$.

To further analyze the role of the composition of the alloy and the quantity of GaO_x in the catalytic performance of the series of catalysts studied, the rate of methanol formation, the methanol selectivity, S_{MeOH} , and the quantities of $\text{Ga}_{\text{alloyed}}$ and GaO_x normalized by the catalysts' Ni contents are plotted as a function of the Ni:Ga ratio determined by ICP (x_{ICP} in $\text{Ni}_x\text{Ga}_{(100-x)}/\text{SiO}_2$, Figure 4B,C). Contrasting the individual trends, we observe that S_{MeOH} increased quasi-linearly (in the range considered here) with the amount of $\text{Ga}_{\text{alloyed}}$ (Figure 4B, inset b). On the other hand, the methanol formation rate related nonlinearly with $\text{Ga}_{\text{alloyed}}$ and increased significantly once the composition of the alloy nanoparticle had reached the

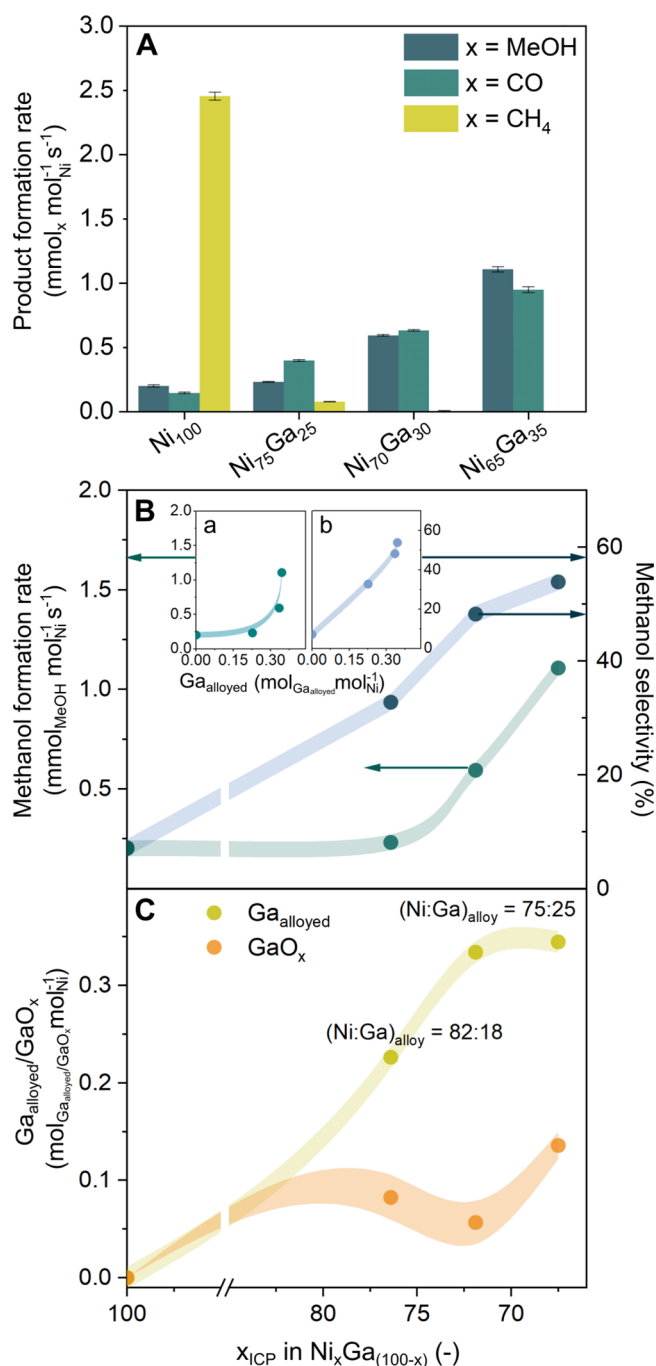


Figure 4. (A) Product formation rates over $\text{Ni}_x\text{Ga}_{(100-x)}/\text{SiO}_2$. (B) Methanol formation rate and methanol selectivity (averaged over the first 180 min of TOS) as a function of x_{ICP} in $\text{Ni}_x\text{Ga}_{(100-x)}/\text{SiO}_2$, as determined by ICP. The insets plot the methanol formation rates and methanol selectivity as a function of the $\text{Ga}_{\text{alloyed}}$ content normalized by the catalysts' Ni content. (C) $\text{Ga}_{\text{alloyed}}$ and GaO_x contents as a function of x_{ICP} . The labels denote the molar ratio Ni:Ga in the alloy, as determined from PDF analysis. Shaded lines are guides to the eye.

composition Ni:Ga = 75:25 (Figure 4B, inset a). Moreover, it was observed that for catalysts with an optimal alloy composition of Ni:Ga = 75:25 (ratio determined after activation according to d-PDF), an increase in the quantity of GaO_x led to a significant increase in the rate of methanol formation (at a stable methanol selectivity).

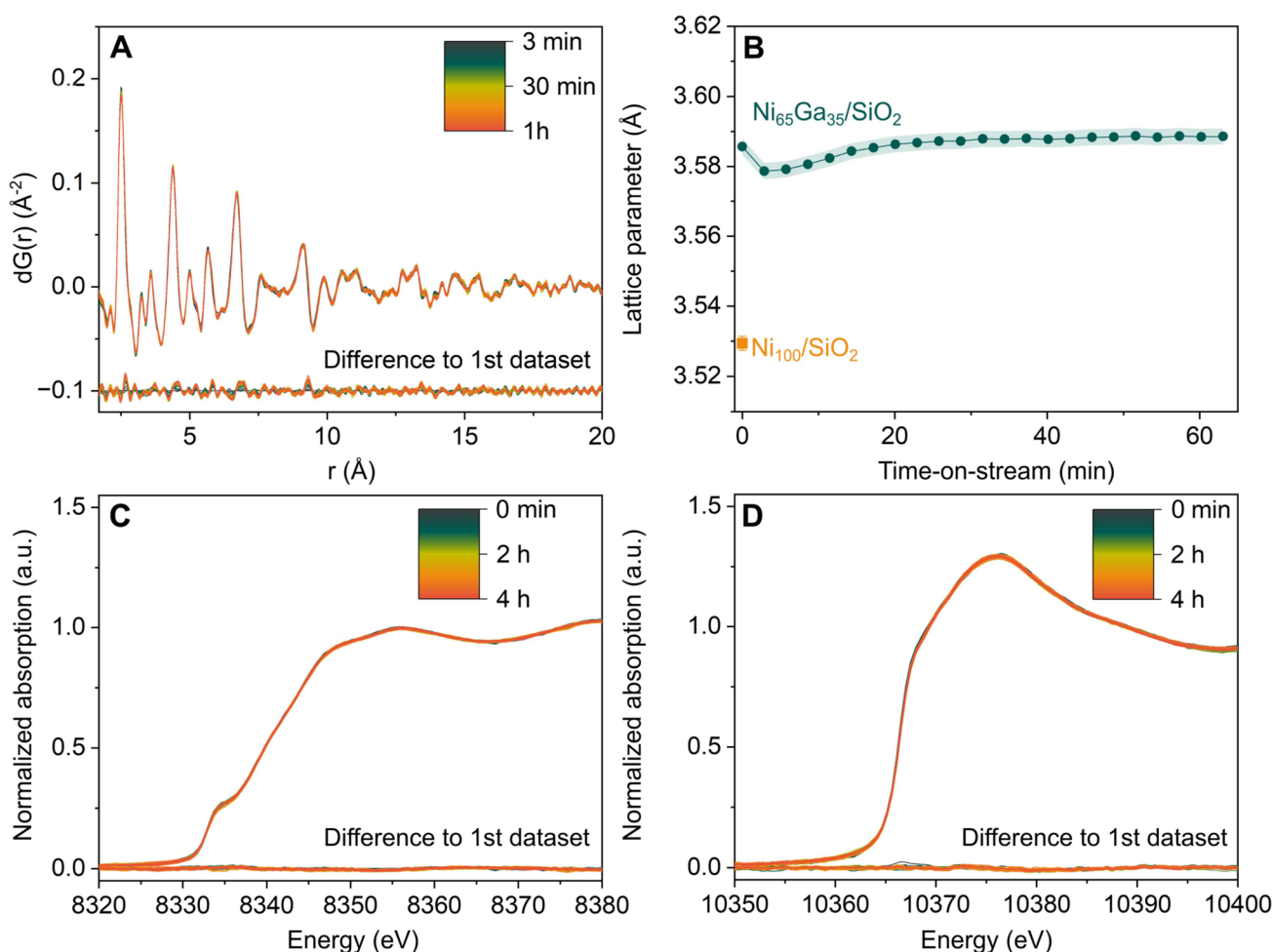


Figure 5. Operando characterization of $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$. (A) d-PDF data collected over 60 min of TOS. The difference between each scan collected at TOS with respect to the first data collected after 3 min under reaction conditions. (B) Lattice parameter obtained via the fitting of the d-PDF shown in (A) as a function of TOS. The first value of the cell parameter (TOS = 0) corresponds to the value determined after catalyst activation (data collected in H_2 , Figure 2D). The lattice parameter obtained for activated $\text{Ni}_{100}/\text{SiO}_2$ is plotted for comparison. Normalized XANES collected over 4 h TOS at the (C) Ni K-edge and (D) Ga K-edge. Conditions: 20 bar, $\text{CO}_2:\text{H}_2:\text{N}_2 = 1:3:1$, 230 °C.

2.4. Catalysts' Structure during CO_2 Hydrogenation to Methanol Conditions

To track the catalyst structure and possible changes under reaction conditions, we further collected operando d-PDF and XAS data on the most active catalyst, i.e., $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$, under CO_2 hydrogenation conditions (data collection started at the time of the gas switch from 20 bar N_2 to 20 bar $\text{CO}_2:\text{H}_2:\text{N}_2 = 1:3:1$ at 230 °C after the activation in hydrogen). We monitored the catalyst's structure over ca. 3.5 h (total scattering) and 4 h (XAS) time-on-stream (TOS) (Figure 5). During the operando experiments, the outlet gas stream was analyzed online by gas-chromatography. In addition, similar operando d-PDF experiments were performed for $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$ and $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ for 1 h TOS.

d-PDF analysis revealed that the nanocrystalline fcc phase in $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ remained stable under reaction conditions (Figure 5A, Supporting Information Figure S25); also, no growth in particle size was observed. We detected a minor increase by ca. 0.3% in the lattice parameter (from 3.579 to 3.589 Å, Figure 5B) within the first 60 min of TOS (to a lesser extent in $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$ and $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$, Supporting Information Figure S24). The slight increase in the lattice parameters could be due to a temperature increase caused by the onset of the

exothermic CO_2 to methanol reaction (after the gases are switched from 20 bar N_2 at TOS = 0 min to 20 bar $\text{CO}_2:\text{H}_2:\text{N}_2 = 1:3:1$). It should be noted that a dealloying of the nanoparticles (i.e., oxidation of some Ga) would lead to a decrease in the lattice parameter, i.e., the opposite than what is observed experimentally.

Next, we evaluated whether changes in the oxidation state of Ga or Ni occurred under CO_2 hydrogenation conditions. To probe possible changes in the XANES data with TOS, we calculated the difference (Δ) between the n th and the first XANES scan (with n ranging between the first and the final scan after ca. 4 h). The maximum difference in the magnitude of ΔXANES at the Ni- and Ga-K-edges was <0.05 in normalized absorption units (Figure 5C,D), indicating that no reduction nor oxidation of Ga took place under reaction conditions. This also implied that the small quantities of GaO_x that were present after activation of the catalysts remained constant with TOS. We however do not exclude the possibility of alloying/dealloying of Ga taking place over extended catalyst operation times. In line with our XANES data and the observation of a stable fcc alloy phase from operando total scattering/PDF analysis, also the fitted EXAFS parameters of the reacted catalyst $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ remained constant with respect to the activated catalyst (i.e.,

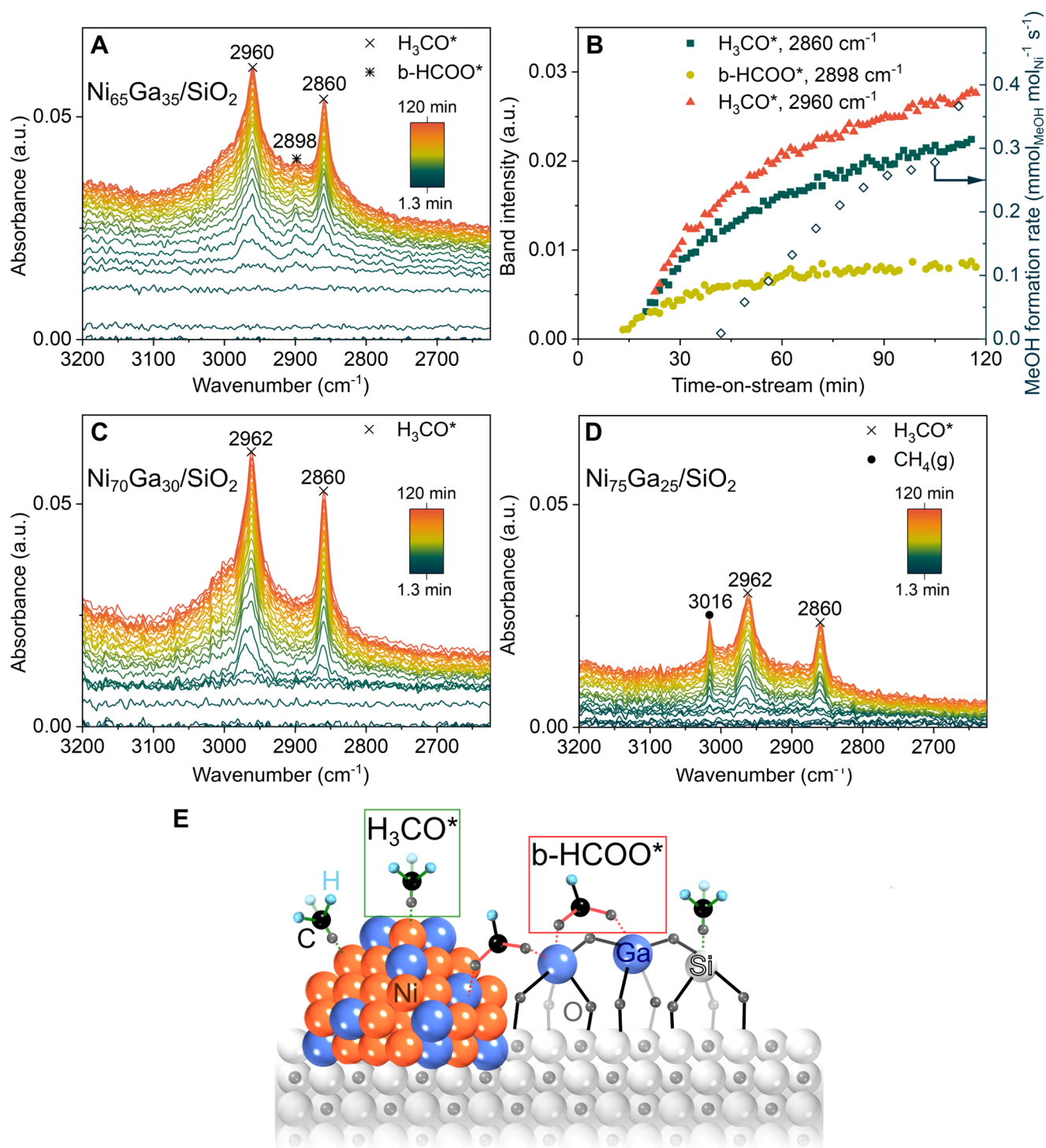


Figure 6. (A) Operando DRIFTS for $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ and (B) the corresponding baseline subtracted, intensity of the methoxy (H_3CO^*) and bidentate formate (b-HCOO^*) bands and the measured methanol formation rate as a function of TOS. Operando DRIFTS for (C) $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$ and (D) $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$. TOS denotes the time after switching from 20 bar N_2 to reaction conditions. Conditions: 20 bar, $\text{CO}_2:\text{H}_2:\text{N}_2 = 1:3:1$, 230 $^\circ\text{C}$, GHSV = 60 $\text{L g}_{\text{cat}}^{-1} \text{h}^{-1}$. (E) Schematic showing how methoxy and formate adsorbate species are potentially adsorbed on the catalyst surface.

within the error of the fitting, see Supporting Information Table S8). These results indicated that the Ni–Ga alloy nanoparticles retained their structure and composition under CO_2 hydrogenation conditions, with no evidence for dealloying/additional alloying, in agreement with the study of Hejral et al. using unsupported Ni–Ga nanoparticles.²⁵ The behavior observed for Ni–Ga/ SiO_2 contrasts what have been observed for other bimetallic catalysts (i.e., Cu–Zn,³¹ Cu–Ga,^{10,12} Pd–Ga¹⁸)

synthesized via the same SOMC approach used in this work, where dealloying, even if partial, was always observed under CO_2 hydrogenation conditions.^{12,18,31,59,60} Note that the absence of bulk dealloying was also observed for the Au–Zn/ SiO_2 system (Au:Zn ca. 33:67).⁶¹ Also here, the Au–Zn alloy formed after catalyst activation (H_2 , 300 $^\circ\text{C}$) and remained stable during CO_2 hydrogenation conditions (10 bar $\text{CO}_2:\text{H}_2:\text{Ar} = 1:3:1$, 230 $^\circ\text{C}$),

while a minor oxidation of surface Zn was observed using operando XAS with a modulation of the gas-phase composition.

2.5. Surface Reaction Intermediates and Bound Adsorbates

To assess whether differences in the catalysts' structure and composition (e.g., Ni:Ga ratio in the fcc alloy and quantity of GaO_x) affect the type and amounts of surface adsorbate species and reaction intermediates, we performed operando DRIFTS experiments on $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$, $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$, and $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$, i.e., catalysts with distinctive catalytic performance and structure. It is worth remembering that $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ and $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$ contained fcc alloy nanoparticles with an identical Ni:Ga ratio (75:25) but different quantities of GaO_x (0.14 and 0.06 $\text{mol}_{\text{GaO}_x} \text{mol}_{\text{Ni}}^{-1}$ in $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ and $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$, respectively) allowing us to assess the role of GaO_x species on the observed surfaces species. Further, $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ contained an fcc alloy with a higher Ni:Ga ratio (82:12) compared to $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ and $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$. Importantly, online analysis of the off-gas during the operando DRIFTS experiments reproduced the trend of the methanol formation rate observed in the catalytic packed bed measurements (Supporting Information Figures S46–48), viz., the methanol formation rate decreases in the order $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2 > \text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2 > \text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$.

In the operando DRIFTS data acquired, there are two spectral regions of interest (i) 1900–2200 cm^{-1} and (ii) 2600–3200 cm^{-1} , in which $M_n\text{-CO}_m$ ($M = \text{Ni, Ga}$) vibrations⁶² and C–H stretching vibrations⁶³ are expected, respectively. Bands at ca. 2200–2450, 3500, and 3770 cm^{-1} are assigned to gaseous $\text{CO}_2(\text{g})$ (see Table S10 in the Supporting Information for all the referenced band assignments).

In the region 1900 cm^{-1} – 2200 cm^{-1} (Supporting Information Figures S46–48), bands at 2077, 2094, and 2129 cm^{-1} were observed for all of the catalysts and were due to pressurized gaseous $\text{CO}_2(\text{g})$.⁶⁴ The first and most prominent band to appear after less than 10 min once the reaction gas mixtures has been introduced into the IR cell (range of 2049–2064 cm^{-1}) could be assigned to $\text{Ni}(\text{CO})_n$ ($n = 1\text{--}4$) surface adsorbate species and possibly also to gaseous $\text{Ni}(\text{CO})_4(\text{g})$.^{24,65–67}

In the region 2600–3200 cm^{-1} , two bands due to C–H stretching vibrations at 2860 and 2960–2962 cm^{-1} appeared on all catalysts independent of their composition. These bands started to appear at ca. 20–30 min after the introduction of a reaction gas mixture (Figure 6A–D). Previous studies on supported metal nanoparticle catalysts for methanol synthesis have assigned these bands to the asymmetric and symmetric stretching modes of methoxy groups ($-\text{OCH}_3$) bonded to either the metal nanoparticles or the support.⁶³ The band positions observed in our study (2860 and 2960–2962 cm^{-1}) matched well with values reported for methoxy species adsorbed onto SiO_2 .^{68–70} Interestingly, on $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ (i.e., the most active and selective catalyst), an additional band at 2898–2900 cm^{-1} was observed that can be assigned to $\nu(\text{CH})$ of bidentate formate (b-HCOO);¹⁸ this band is absent in the other two catalysts, i.e., $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$ and $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$. However, we do not exclude the presence of formate species in these catalysts, which could be present in a smaller extent compared to that of $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ and are thus not detected and overshadowed with overlapping methoxy species in the IR spectrum. Note that two additional bands due to b-HCOO would be expected in the region 2800–3000 cm^{-1} , which, however, overlap with the two bands assigned to methoxy species.^{18,71,72} Based on previous in

situ IR studies on gallium oxides, the band at 2898–2900 cm^{-1} was likely due to b-HCOO on gallium oxide species,^{63,73} which was in agreement with our d-PDF and XAS analyses that indicated more abundant GaO_x species (= binding sites of b-HCOO) on $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ when compared to that on $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$ and $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$. Hence, our experiments suggested that GaO_x species (likely in proximity of the alloy nanoparticles) further promote the formation of formate species. According to the work of Collins et al.,⁷³ bidentate formate species on gallium oxide also shows bands in the region 1300–1600 cm^{-1} (i.e., $\nu_{\text{as}}(\text{CO}_2)$ at 1580 cm^{-1} , $\delta(\text{CH})$ at 1386 cm^{-1} , $\nu_{\text{s}}(\text{CO}_2)$ at 1372 cm^{-1}) which, however, could not be observed unequivocally in our system due to the strong absorption of the incident IR radiation by SiO_2 in this spectral region, leading to a poor signal-to-noise ratio.⁷⁴ The presence of formate (in $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$) and methoxy bands in $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$, $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$, and $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ was in line with the previously proposed mechanism for methanol formation, i.e., the formate–methoxy pathway (Figure 6E)²⁴ and in line with what has been proposed for Cu.⁷⁵ In addition, and in agreement with the formation of some methane in $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ (Figure 6D), operando DRIFTS on $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ showed the ν_3 stretching vibration of $\text{CH}_4(\text{g})$ at 3016 cm^{-1} , along with weak rotational bands of CH_4 in the range 2600–3200 cm^{-1} (Supporting Information Figure S48). No bands due to CH_4 were observed on $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ and $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$, i.e., catalysts that also did not show any methane formation or only negligible quantities in the packed bed experiments (Figure 6 A,C, Supporting Information Figures S46 and S47).

3. CONCLUSIONS

In this study, we report the synthesis of model catalysts containing Ni–Ga nanoparticles supported on silica ($\text{Ni}_x\text{Ga}_{(100-x)}/\text{SiO}_2$) using a SOMC/TMP approach enabling a precise control over loading and particle size and yielding highly active Ni–Ga-based catalysts for the selective hydrogenation of CO_2 into methanol, in contrast to pure Ni that favors methanation. Combined electron microscopy, operando d-PDF, and XAS studies showed that these silica-supported catalysts contained, after their in situ activation in H_2 , ~ 2 nm sized, Ni–Ga fcc alloy particles with a narrow size distribution along with a small fraction of GaO_x species. Furthermore, Ni K-edge XAS showed that the electronic structure of Ni was affected by alloying and can be described as $\text{Ni}^{\delta-}\text{Ga}^{\delta+}$. Using Vegard's law, the composition of the alloyed nanoparticles was determined as Ni:Ga = 75:25 in activated $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ and $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$, whereas the alloy in $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$ was more Ni-rich (Ni:Ga = 82:12). Furthermore, considering the total Ga content in the materials, as determined by elemental analysis, each catalyst also contained residual GaO_x . Notably, the GaO_x content in $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$ (0.14 $\text{mol}_{\text{GaO}_x} \text{mol}_{\text{Ni}}^{-1}$) was more than double than that in $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$ (0.06 $\text{mol}_{\text{GaO}_x} \text{mol}_{\text{Ni}}^{-1}$), despite having an alloy with the same composition. $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$, having a more Ni-rich alloy, also contained GaO_x (0.08 $\text{mol}_{\text{GaO}_x} \text{mol}_{\text{Ni}}^{-1}$), the quantity of which lies in between that of the other two catalysts.

Regarding catalysis, alloying Ni with Ga decreased its methanation activity, while promoting methanol formation rates up to a certain Ga content (Ni:Ga = 75:25). $\text{Ni}_{65}\text{Ga}_{35}/\text{SiO}_2$, containing an alloy with a Ni:Ga ratio of 75:25 and the largest amount of GaO_x , showed the highest methanol activity (1.1 $\text{mmol}_{\text{MeOH}} \text{mol}_{\text{Ni}}^{-1} \text{s}^{-1}$) and selectivity ($S_{\text{MeOH}} = 54\%$) and

improved performance compared to $\text{Ni}_{70}\text{Ga}_{30}/\text{SiO}_2$ ($0.6 \text{ mmol}_{\text{MeOH}} \text{ mol}_{\text{Ni}}^{-1} \text{ s}^{-1}$ and $S_{\text{MeOH}} = 48\%$) which contained an alloy with the same composition (Ni:Ga = 75:25) but a lower GaO_x content. $\text{Ni}_{75}\text{Ga}_{25}/\text{SiO}_2$, containing a Ni-rich alloy and an intermediate amount of GaO_x , showed the poorest performance ($0.2 \text{ mmol}_{\text{MeOH}} \text{ mol}_{\text{Ni}}^{-1} \text{ s}^{-1}$ and $S_{\text{MeOH}} = 33\%$) along with the monometallic Ni catalyst ($0.2 \text{ mmol}_{\text{MeOH}} \text{ mol}_{\text{Ni}}^{-1} \text{ s}^{-1}$ and $S_{\text{MeOH}} = 7\%$) that favored methanation ($S_{\text{CH}_4} = 88\%$), as expected for pure Ni. Hence, both the amount of Ga in the alloy (Ni:Ga = 75:25) and Ga in the form of GaO_x are important for the methanol production rate and selectivity. Importantly, operando d-PDF and XAS analyses of the catalysts under CO_2 hydrogenation conditions revealed that the fcc structure of the alloy and the oxidation states of Ni and Ga remained invariant with respect to the activated state (treated under H_2), indicating that the bulk alloy was not affected by the reaction conditions, in sharp contrast to what has been observed for Cu–Ga and Pd–Ga. Furthermore, monitoring the catalysts by operando DRIFTS indicates that the presence of GaO_x helps in stabilizing formate species, an important reaction intermediate in the methanol formation pathway, further supporting the importance of GaO_x .

Overall, our results showed that alloy nanoparticles with a Ni:Ga ratio of 75:25 result in high methanol activity and selectivity—considerably higher than for Ni-rich alloys—while the presence of GaO_x further increases the rate of methanol formation. These results pointed to the importance of site-isolation of Ni by Ga that leads to modifications of the electronic structure of Ni, viz., $\text{Ni}^{\delta+}\text{Ga}^{\delta-}$. Ni site isolation likely prevents methanation activity, while the presence of GaO_x further promotes the selective formation of methanol. Therefore, the regulation of the quantity of both alloyed Ga and GaO_x species is crucial in achieving a high methanol selectivity and a high rate of methanol formation.

4. EXPERIMENTAL SECTION

4.1. Materials

All preparation and operations were performed in a M. Braun glovebox under an argon atmosphere or using high vacuum and standard Schlenk techniques. Pentane was purged with argon for 30 min and dried using a MB SPS 800 solvent purification system in which columns used for pentane purification were packed with activated copper and alumina. Benzene was either distilled from purple Na^0 /benzophenone or obtained from the MB SPS system and used without further purification. All solvents were stored over 4 Å molecular sieves after being transferred into a glovebox. Four Å molecular sieves were activated under high vacuum overnight at 300 °C. Fine quartz wool (Elemental Microanalysis) was calcined at 800 °C for 12 h and transferred into a glovebox for storage. SiO_{2-700} was prepared by heating Aerosil (200 m^2/g) to 500 °C (ramp of 300 °C/h) in air and subsequent calcination in air for 12 h. Afterward, the material was evacuated at high vacuum (10^{-5} mbar) at 500 °C for 8 h, followed by heating to 700 °C (ramp of 60 °C/h) and keeping the material at 700 °C for approximately 24 h. Titration of SiO_{2-700} using $[\text{Mg}(\text{CH}_2\text{Ph})_2(\text{THF})_2]$ purified via sublimation prior to use yielded an Si–OH density of 0.3 mmol/g, corresponding to 0.9 accessible Si–OH groups per nm^2 . The molecular complexes $[\text{Ga}(\text{OSi}(\text{OtBu})_3)_3(\text{THF})]$ and $[\text{Ni}(\text{CH}_3)_2(\text{tmeda})]$ were prepared according to adapted literature procedures.^{36,76} Other reagents were purchased from Sigma-Aldrich or Acros Organics and used as received. The supported species $\text{Ga}^{\text{III}}/\text{SiO}_2$ (precursor for $\text{Ni}_x\text{Ga}_{(100-x)}/\text{SiO}_2$ syntheses) were prepared according to an adapted literature procedure (see below).³⁶

4.2. Synthesis of $\text{Ga}^{\text{III}}/\text{SiO}_2$

In a typical synthesis, SiO_{2-700} (0.729 g, 0.219 mmol –OH) was added to a 20 mL vial. Next, benzene (about 5 mL) was added slowly while stirring to give a white suspension. $[\text{Ga}(\text{OSi}(\text{OtBu})_3)_3(\text{THF})]$ (0.147 mmol/g SiO_{2-700} nominal loading; 0.100 g, 0.107 mmol) was added slowly to the suspension as a white solution in benzene (about 5 mL) while stirring (1200 rpm). The resulting suspension was stirred at room temperature (rt) for 12 h (100 rpm). The benzene on top of the silica material was decanted, and the material was washed with benzene (5 mL) two times to wash off any unreacted complex. The material was then washed with pentane before it was dried in vacuo for 2 h to remove any residual solvent yielding $\text{Ga}^{\text{III}}/\text{SiO}_2$ as a white solid. The white material was then transferred into a tubular quartz reactor. The reactor was set under high vacuum (10^{-5} mbar) and successively heated to 300 °C (ramp of 5 °C/min) for 1 h, 400 °C (ramp of 5 °C/min) for 1 h, 500 °C (ramp of 5 °C/min) for 1 h, and finally 600 °C (ramp of 5 °C/min) for 10 h, yielding $\text{Ga}^{\text{III}}/\text{SiO}_2$ as a white/gray material.

4.3. Synthesis of $\text{Ga}_{100}/\text{SiO}_2$

Here, ca. 0.580 g of $\text{Ga}^{\text{III}}/\text{SiO}_2$ (synthesized as described above) was added to a tubular quartz flow-reactor supported with a porous quartz frit. The reactor was heated to 600 °C (ramp of 5 °C/min) under a steady flow of H_2 and then treated under H_2 at 600 °C for 12 h. The reactor was subsequently evacuated under high vacuum (10^{-5} mbar) while cooling to room temperature, yielding $\text{Ga}_{100}/\text{SiO}_2$ as a white/gray material. Elemental analysis yielded: Ga, 0.88 wt %.

4.4. Synthesis of $\text{Ni}_{100}/\text{SiO}_2$

SiO_{2-700} (0.726 g, 0.218 mmol –OH) was added to a 20 mL vial. Benzene (about 5 mL) was added slowly while stirring to give a white suspension. $[\text{Ni}(\text{CH}_3)_2(\text{tmeda})]$ (0.342 mmol/g SiO_{2-700} nominal loading; 0.051 g, 0.248 mmol) was added slowly to the suspension as a deep yellow solution in benzene (about 5 mL) while stirring (1200 rpm), resulting in an immediate pink/deep-red coloration of SiO_2 . The resulting suspension was stirred at room temperature for 75 min (125 rpm) after which no yellow color of the supernatant was observable anymore. The benzene on top of the silica material was decanted, and the material was washed with benzene (5 mL) two times to wash off any unreacted complex. The material was then washed with pentane before it was dried in vacuo for 30 min to remove any residual solvent, yielding $\text{Ni}^{\text{II}}/\text{SiO}_2$ as a pink/deep-red solid. The material was transferred to a tubular quartz flow-reactor supported with a porous quartz frit. The reactor was heated to 600 °C (ramp of 5 °C/min) under a steady flow of H_2 and held under H_2 at 600 °C for 12 h. The reactor was subsequently evacuated under high vacuum (10^{-5} mbar) while cooling to room temperature, yielding $\text{Ni}_{100}/\text{SiO}_2$ as a dark brown/black material. Elemental analysis yielded: Ni, 2.12 wt %.

4.5. Synthesis of $\text{Ni}_x\text{Ga}_{(100-x)}/\text{SiO}_2$ ($x = 65, 70, 75$)

$\text{Ga}^{\text{III}}/\text{SiO}_2$ materials (0.700, 0.707, and 0.704 g) were synthesized as described above with different nominal Ga loadings (0.114, 0.147, and 0.184 mmol/g SiO_{2-700} nominal loading; 0.074, 0.097, and 0.121 g of $[\text{Ga}(\text{OSi}(\text{OtBu})_3)_3(\text{THF})]$). All materials were obtained as off white/gray materials. In a next step, the $\text{Ga}^{\text{III}}/\text{SiO}_2$ materials (0.640, 0.605, and 0.545 g) were added to 20 mL vials. Next, benzene (about 5 mL) was added slowly while stirring to give a grayish suspension. $[\text{Ni}(\text{CH}_3)_2(\text{tmeda})]$ (0.342 mmol/g $\text{Ga}^{\text{III}}/\text{SiO}_2$ nominal loading; 0.045, 0.042, and 0.038 g; 0.219, 0.207, and 0.187 mmol) was added slowly to the suspension as a deep yellow solution in benzene (about 5 mL) while stirring (1200 rpm), resulting in an immediate pink/deep-red coloration of the SiO_2 . The resulting suspensions were stirred at room temperature for 65 min (125 rpm) after which no yellow color of the supernatant was observable anymore. The benzene on top of the silica material was decanted, and the materials were washed with benzene (5 mL) two times to wash off any unreacted complex. The materials were then washed with pentane before they were dried in vacuo for ca. 45 min to remove any residual solvent yielding pink/deep-red solids. The materials were then transferred to small, tubular quartz vessels supported with a porous quartz frit. These small tubes were plugged with a small amount of quartz wool to avoid spilling. The small vessels were subsequently transferred to a large tubular quartz flow-

reactor supported with a porous quartz frit. The reactor was heated to 600 °C (ramp of 5 °C/min) under a steady flow of H₂ and then treated under H₂ at 600 °C for 12 h. The reactor was subsequently evacuated under high vacuum (10^{−5} mbar) while cooling to room temperature, yielding Ni₇₅Ga₂₅/SiO₂, Ni₇₀Ga₃₀/SiO₂, and Ni₆₅Ga₃₅/SiO₂ as dark brown/black materials.

The synthesis yielded ca. 500 mg per sample. The materials were characterized by elemental analysis, catalytic CO₂ hydrogenation tests, in situ and operando X-ray total scattering, and operando DRIFTS. A second batch of Ni₆₅Ga₃₅/SiO₂ and Ni₇₅Ga₂₅/SiO₂ catalysts was prepared to allow for further characterization. The second catalyst batch was characterized by XAS (Figures 3, S32–44), ICP, catalytic tests (labeled with the suffix XAS). To ensure reproducibility, PDF was also collected for Ni₆₅Ga₃₅/SiO₂-XAS.

4.6. Characterization

4.6.1. Elemental Analysis (Ga, Ni). The elemental composition of the catalysts was determined by ICP-OES using an Agilent 5100 VDV instrument. Typically, 2–3 mg of the sample was dissolved in 5 mL of aqua regia, followed by microwave digestion at 175 °C for 30 min (Anton Paar, Multiwave GO). The resulting solution was cooled down to room temperature and diluted to 25 mL with deionized water. For the calibration of the instrument, a multielement standard (multielement standard solution S, Sigma-Aldrich) was used. Each measurement was repeated three times, and the average values are reported in Table S1.

4.6.2. Transmission Electron Microscopy. HAADF-STEM images were recorded on a FEI Talos F200X and—where indicated—a JEOL JEM-ARM300F Grand Arm “Vortex” instrument operated at 200 and 300 keV, respectively. Powdered samples were mixed in solid form with a Lacey-C 400 mesh Cu grid inside a glovebox under an Ar atmosphere before it was mounted onto a vacuum transfer tomography holder from Fischione Instruments (model #2560) (Talos) or a Double Tilt Atoms Defend Holder System (Mel-Build Corporation, serial number: DT-TR-006-J001) inside the glovebox, which was subsequently transferred to the chamber of the TEM in the absence of air. All imaging was done in air-free conditions if not indicated otherwise. For all materials, 300 nanoparticles were counted to yield a particle size distribution (PSD) analysis. The determination of the nanoparticle diameters for the PSD was done by manual measurements using the software ImageJ (version 1.52a). All values extracted from the specific particle size distributions assumed a normal distribution (most of the shown distribution curves are log-normal). The “±” symbol in the particle size distributions plots indicates the standard deviation of the mean particle size. The following catalysts were characterized by HAADF-STEM/EDX: Ni₁₀₀/SiO₂, Ni₇₅Ga₂₅/SiO₂, Ni₇₀Ga₃₀/SiO₂, and Ni₆₅Ga₃₅/SiO₂ (as-prepared and after a CO₂ hydrogenation test, the latter are referred to as “spent”).

4.6.3. Total X-ray Scattering and X-ray Absorption Spectroscopy. In situ/operando X-ray total scattering and XAS experiments were performed at the beamlines ID15A and BM31 of the European Synchrotron Radiation Facility, respectively. The same setup was used for both types of experiments, consisting of a capillary cell reactor (quartz capillary with an outer diameter of 1 and 0.02 mm wall thickness) containing ca. 2–3 mg of catalyst placed between two quartz wool plugs. Heating of the capillary was achieved from below via a hot air blower, and gases could be flown through it at a defined pressure via mass flow controllers and a backpressure regulator placed after the outlet of the capillary. A typical in situ/operando experiment consists of an in situ activation treatment (temperature ramp to 600 °C at 10 °C/min in 1 bar 5 mL/min H₂ and then waiting for 1 h), cooling down to the reaction temperature of 230 °C in 1 bar 5 mL/min H₂ (or 50 °C, to collect EXAFS, see Figure S18), and then pressurizing the reactor to 20 bar in 15 mL/min N₂ (at 230 °C). Once the pressure had equilibrated, the gases were changed to the reaction gas mix of CO₂:H₂:N₂ = 1:3:1 (at 20 bar, 5 mL/min). The off-gas during the reaction was analyzed using a compact gas chromatograph (Global Analyzer Solutions compact GC^{4.0} equipped with FID and TCD detectors, 1 injection/7 min).

X-ray total scattering data was collected continuously at a rate of 1 measurement/2.62 min and up to $Q_{\text{max, instr}} = 30 \text{ \AA}^{-1}$ (incident X-ray energy of 90.0 keV). Total scattering data of the pristine silica support was measured under in situ activation and reaction conditions and used as background to calculate the d-PDF data of the Ni_xGa_(100−x)/SiO₂ (Supporting Information Figure S22). In addition, total scattering data of the CeO₂ NIST reference materials were obtained to determine the experimental resolution parameters Q_{damp} and Q_{broad} . For the conversion from total scattering data (reciprocal space) to the PDF (real space), the software PDFgetX3⁷⁷ v. 2.2.1 was used, whereas the modeling of the d-PDF was done in PDFGui⁷⁸ v 1.0. The total scattering data was processed within the range $Q_{\text{min}} = 1.5 \text{ \AA}^{-1}$ and $Q_{\text{max}} = 23 \text{ \AA}^{-1}$ with $r_{\text{poly}} = 1.0$, which is approximately the r -limit for the maximum frequency in the $F(Q)$ correction polynomial.

Ni and Ga K-edge XAS scans were collected consecutively. Ni and Ga XANES were collected between 8250 and 8550 eV and 10300–10600 eV, respectively, with a step of 0.3 eV (ca. 50 s/spectrum). Ni and Ga EXAFS were collected between 8200 and 8970 eV and 10200–11100 eV with a step of 0.5 eV (ca. 3 min/spectrum). More details on the X-ray total scattering and XAS data collection and processing can be found in section 3 of the Supporting Information.

4.6.4. Diffuse Reflectance Infrared Fourier Transform Spectroscopy. Operando DRIFTS experiments were performed using a Nicolet 6700 FT-IR equipped with a Harrick Praying Mantis DRIFTS accessory and high-temperature reaction chamber. Data were collected from 650 to 4000 cm^{−1} at a spectral resolution of 4 cm^{−1} using a mercury cadmium telluride detector. In a typical experiment, ca. 20 mg of powder was placed onto a piece of quartz wool in the sample cup of the reaction cell. First, the catalysts were in situ activated in 15% H₂/N₂ (20 mL/min) at 590 °C (10°/min) for 1 h and subsequently cooled down to 230 °C. Next, the cell was pressurized to 20 bar in N₂ (20 mL/min). Once the pressure had stabilized, a measurement was collected, which served as the background for all the subsequent measurements under reaction conditions. In a next step, the atmosphere was switched to CO₂:H₂:N₂ = 1:3:1 (at 20 bar, 20 mL/min), and measurements were continuously collected every 1.3 min for ca. 2 h. The identical compact GC as the one used for the operando XAS/PDF experiments was used for the analysis of the off-gas. A blank test (empty reaction cell) was performed to determine the time needed for the equilibration of the feed gases from the time when the gases are switched from 20 bar N₂ to the reaction gas mixture. The exchange time was approximately 126 min (see Figure S45).

4.7. Catalytic Tests

For the catalytic tests, 100 mg of the as-prepared powder was transferred into a Hastelloy C276 reactor tube (internal diameter 9.1 mm) inside a M. Braun glovebox under a N₂ atmosphere. The powder was placed in between two pieces of quartz wool (Acros Organics, 9–30 μm) and onto a frit in the middle of the reactor tube (Hastelloy C276, pore size 2 μm). The reactor was then transferred from the glovebox and mounted into a Microactivity-Efficient flow reactor system (PID Eng & Tech), without exposing the catalyst to air (the reactor is provided with valves for such purpose). The spent catalysts were transferred back into the glovebox and stored for further TEM analysis.

The CO₂ hydrogenation tests were performed as follows. First, the catalyst was heated to 600 °C (10 °C/min) in 1 bar H₂ (50 mL/min) and held at 600 °C for 1 h. This step is referred to as in situ activation. Subsequently, the catalyst was cooled down to the reaction temperature of 230 °C (still in 1 bar H₂, 50 mL/min), followed by a switch of the gas flow to 80 mL/min N₂ and a pressure increase to 25 bar. After the stabilization of the pressure and temperature, the gas feed was switched to 25 bar of CO₂:H₂:N₂ = 1:3:1 (100 mL/min, resulting in a GHSV of 60 L gcat^{−1} h^{−1}), and the off-gas was continuously analyzed by a gas chromatograph (PerkinElmer Clarus 580 equipped with FID and TCD detectors, 1 injection/30 min). For further information on the catalytic tests, we refer the reader to Section 2 of the Supporting Information.

■ ASSOCIATED CONTENT

SI Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/jacsau.3c00677>.

Experimental details, additional microscopy, X-ray absorption and IR spectroscopy, as well as X-ray total scattering/PDF and catalytic test data (PDF)

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Author Contributions

CRedit (Contributor Roles Taxonomy)⁷⁹ was used for standardized contribution descriptions: N.K.Z.: investigation, methodology, formal analysis, validation, visualization, writing (original draft), writing (review and editing). L.R.: investigation, methodology, formal analysis, validation, visualization, writing (original draft), writing (review and editing). S.C.: investigation and methodology. P.M.A.: conceptualization, project administration, supervision, writing (review and editing). C.M.: conceptualization, project administration, supervision, writing (review and editing), resources, funding acquisition. C.C.: conceptualization, project administration, supervision, writing (review and editing), resources, funding acquisition. N.K.Z. and L.R. contributed equally to this work.

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Notes

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■ ABBREVIATIONS

d-PDF, differential (or difference) pair distribution function analysis; DRIFTS, diffuse reflectance infrared Fourier transform spectroscopy; EXAFS, extended X-ray absorption fine structure; fcc, face-centered cubic; SOMC, surface organometallic chemistry; STEM-EDX, scanning transmission electron microscopy–element-dispersive X-ray spectroscopy; TMP, thermolytic molecular precursor; TOS, time-on-stream; XANES, X-ray absorption near-edge structure; XAS, X-ray absorption spectroscopy

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