

# **Identifying Reactive Trends in Glycerol Electro-Oxidation Using an Automated Screening Approach: 28 Ways to Electrodeposit an Au Electrocatalyst**

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comprehensively address the exploration of chemical spaces, characterization of catalyst robustness, reproducibility, and translation of results to (flow) electrolysis operation are needed. Responding to the growing interest in biomass valorization, we studied the glycerol electro-oxidation reaction (GEOR) on gold in alkaline media as a model reaction to demonstrate the efficacy of such methodology introduced here. Our platform combines individually addressable electrode arrays with HardPotato, a Python application programming interface for potentiostat control, to automate electrochemical experiments and data analysis operations. We systematically investigated the effects of reduction potential (*E*<sup>l</sup> ) and pulse width (PW) on GEOR activity during the electrodeposition (Edep) of gold, evaluating 28 different conditions in



triplicate measurements with great versatility. Our findings reveal a direct correlation between  $E_1$  and GEOR activity. Upon CV cycling, we recorded a 52% increase in peak current density and a −0.25 V shift in peak potential as *E*<sub>l</sub> varied from −0.2 to −1.4 V. We also identified an optimal PW of ∼1.0 s, yielding maximum catalytic performance. The swift analysis enabled by our methodology allowed us to correlate performance enhancements with increased electrochemical surface area and preferential deposition of Au(110) and Au(111) sites, even in disparate Edep conditions. We validate our methodology by scaling the Edep process to larger electrodes and correlating intrinsic activity with product speciation via flow electrolysis measurements. Our platform highlights opportunities in automation for electrocatalyst discovery to address pressing needs toward industrial decarbonization, such as biomass valorization.

KEYWORDS: *pulse width, reduction potential, catalytic performance, GEOR activity*

# ■ **INTRODUCTION**

Electrochemical valorization of biomass has emerged as a promising approach to furthering the global initiative of electrifying the chemical industry. This approach allows transitioning away from fossil-fuel-based processes and toward more sustainable and carbon-neutral technologies for chemical and fuel production.<sup>1−[4](#page-11-0)</sup> Among the host of biomass molecules available, glycerol, a byproduct from soap production and the burgeoning biodiesel industry<sup>[5](#page-11-0)</sup> holds particular interest due to its affordability (\$0.3/kg) and nontoxicity. The oxidation of glycerol's hydroxyl groups via the glycerol electro-oxidation reaction (GEOR) is reported to yield a plethora of value-added products<sup>[1](#page-11-0),[6,7](#page-11-0)</sup> such as tartronic acid  $(\frac{$467}{kg})$ ,<sup>8</sup> mesoxalic acid  $(1400/kg)$ ,<sup>[9](#page-11-0)</sup> and glyceric acid  $(1400/kg)$ <sup>[10](#page-11-0)</sup> as well as industrially significant chemicals such as formic  $\text{acid}_1^1$  $\text{acid}_1^1$  $\text{acid}_1^1$  glycolic acid, $11$  and oxalic acid. $12$ 

Moreover, GEOR also serves as a financially viable and energy-efficient alternative to the oxygen evolution reaction

 $(OER)$ ,<sup>[13](#page-11-0)−[16](#page-11-0)</sup> conventionally occurring at the anode of  $CO<sub>2</sub>$ reduction reaction  $(CO_2RR)$  and hydrogen evolution reaction (HER) electrolyzer cells. OER, a complex four-electron transfer process, is both energetically intensive and unprofitable due to its main byproduct $-\Omega_{2}^{17-19}$  $-\Omega_{2}^{17-19}$  $-\Omega_{2}^{17-19}$  $-\Omega_{2}^{17-19}$  $-\Omega_{2}^{17-19}$  which has limited market value. The theoretical overpotential OER is 1.23 V vs RHE, yet in practice, it requires >1.8 V to drive the reaction owing to sluggish kinetics.<sup>20</sup> Verma et al.<sup>[16](#page-11-0)</sup> reported that substituting OER with GEOR in a  $CO<sub>2</sub>RR$  electrolyzer lowers electricity consumption by 53%. Similarly, Qian et al.<sup>[21](#page-12-0)</sup> achieved a 10% reduction in energy consumption while

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<span id="page-1-0"></span>operating an  $H_2$  production electrolyzer at industrial-level current densities simply by substituting OER with GEOR.

Building on the demonstrated advantages of the GEOR as a compelling option for modern energy strategies, we hypothesized that its efficiency could be further augmented by exploring metallic electrocatalysts. Despite numerous catalyst design strategies aimed at increasing the number of active sites and enhancing intrinsic activity,  $3,22$  $3,22$  the field suffers from fragmented research and a lack of comprehensive methodologies for catalyst evaluation, with trial and error still prevailing as the guiding principle for catalyst discovery. Furthermore, discrepancies in reported performances often arise from variations in experimental conditions such as mass loading, analyte concentration, and pH, emphasizing the need for standardized electrocatalyst discovery and testing protocols.

Recognizing this, we posit a critical need for a rapid screening methodology to screen electrocatalysts. This approach would facilitate the creation of a comprehensive database and aid in predicting new materials.<sup>23</sup> Several highthroughput (HT) electrochemical (EChem) screening technologies have been proposed, most of which can be classified into three major categories−scanning probe techniques, indirect measurements using optical/spectroscopic signals, and direct current/voltage measurements.<sup>[24](#page-12-0)</sup> A few of these techniques are summarized below.

Scanning probe methods such as scanning electrochemical microscopy (SECM) have been proposed to screen catalysts for reactions such as  $CO<sub>2</sub>$  reduction,<sup>[25](#page-12-0)</sup> oxygen reduction and hydrogen oxidation,  $^{26}$  $^{26}$  $^{26}$  oxygen evolution,  $^{27}$  $^{27}$  $^{27}$  and formic acid oxidation.[28](#page-12-0) While quite powerful and information-rich, SECM-based screening methods primarily involve local delivery of species to electrodes via electro-generation, which complicates the screening possibilities for alcohol oxidation reactions. Additionally, they often require elaborate sample preparation steps that involve piezoelectric dispensing of metal salt solutions followed by reduction at elevated temperatures.<sup>[26](#page-12-0)</sup> This provides poor control over surface structure and morphology and is plagued by further complications during synthesis scale-up from microscale to practical-scale cell setups.

pH imaging screening methodologies have also been reported for the evaluation of electrocatalysts.<sup>[29](#page-12-0)</sup> In such techniques, an electrochemical reaction at the surface of the electrode leads to a change in the local pH, which is imaged using a pH-sensitive fluorescence dye. However, such techniques are typically functional only in a narrow pH window.<sup>[30](#page-12-0)</sup> Similarly, Xiang et al.<sup>30</sup> report an electrocatalyst screening technique in which bubble evolution is mapped as a function of activity; however, this method's scope is restricted to reactions involving gas evolution. While these screening techniques are rapid, most rely on indirect indicators for catalytic performance and suffer from crosstalk from materials due to the overlap of diffusional fields from adjacent electrodes. $24$  Another avenue recently explored for enabling HT electrochemical screening involves using microelectrode arrays for current and potential measurements. $31,32$  $31,32$ 

Considering the constraints posed by current screening architectures, we introduce a novel approach in this paper leveraging HardPotato, an open-source Python API recently developed by the Rodri<sup>guez-L</sup>opez group,<sup>33,[34](#page-12-0)</sup> and individually addressable electrode (IAE) array devices. We demonstrate the applicability of this methodology to screen for the electrocatalytic performance of Au NPs exhibiting diverse physical properties. We accentuate the automatability of our approach by employing electrodeposition (Edep) as a versatile and costeffective synthesis technique. Control over deposition parameters enables tuning features such as morphology, size, and composition, which is crucial for optimizing their electrocatalytic performance. We assessed the activity of different Au electrodeposits toward GEOR, driven by the reaction's immense economic and environmental benefits.<sup>[35](#page-12-0)</sup> Although palladium is recognized for its high intrinsic activity for  $GEOR<sub>36</sub>$  $GEOR<sub>36</sub>$  $GEOR<sub>36</sub>$  it is prone to surface passivation due to the early onset of oxide formation and adsorbed carbon monoxide (CO<sub>ads</sub>) poisoning. In contrast, gold has been demonstrated to have higher  $CO<sub>ads</sub>$  resistance and attain appreciably higher currents (>5×) than platinum and palladium, albeit at higher oxidation potentials.[37,38](#page-12-0) Considering that GEOR involves the adsorption of various intermediates, we hypothesized that by tuning the Edep parameters of gold, we could adjust the catalyst surface properties, size, loading, and morphology, thereby influencing the adsorption energies of these intermediates as well as overall reactivity.<sup>[3](#page-11-0)</sup>

Using our new methodology, we explored a total of 28 different electrodeposition parameter combinations and verified them through triplicate measurements to underscore the reproducibility of the technique. We established trendlines correlating the Edep parameters to the properties of Au active sites and their GEOR performance with minimal human intervention. The validity of our findings was further confirmed through flow-electrolysis experiments to test their scaling up, illustrating our methodology's applicability to advanced electrocatalysis studies.

### ■ **MATERIALS AND METHODS**

**Chemicals and Materials.** All materials were purchased and used as received. Sodium phosphate monobasic (NaH<sub>2</sub>PO<sub>4</sub>), sodium phosphate dibasic (Na<sub>2</sub>HPO<sub>4</sub>), potassium nitrate (KNO<sub>3</sub>, 99+%), and tetrachloroauric(III) acid trihydrate trace metals basis ( $HAuCl<sub>4</sub>·3H<sub>2</sub>O$ , 99.9%) were purchased from Sigma-Aldrich. Hydroxymethyl ferrocene (FcMeOH, 97%) was purchased from Alfa Aesar. Sulfuric acid  $(H_2SO_4)$ , Trace metal grade), potassium hydroxide pellets (KOH, ACS reagent, 85%), glycerol (ACS reagent, 99.6%), and graphite rod (99.9995%) were purchased from Thermo Scientific. Ni foil (Trace Metal, 99.00%) was purchased from ChemDirect. All solutions were prepared using high-performance liquid chromatography-grade water.

**Device Microfabrication.** IAE array devices were fabricated on 4-in. glass wafers sourced from University Wafers, following a previously established workflow. Initially, the wafers were subjected to a degreasing process with acetone and isopropanol, followed by plasma treatment and RCA-1 cleaning. The metal layer for the electrodes and their circuitry was patterned by photolithography using an SPR 220−4.5/ LOR 5A bilayer photoresist followed by the sputtering of gold  $(t = 180.0 \text{ s})$ , with titanium  $(t = 60.0 \text{ s})$  as the adhesion layer, and liftoff performed with Microposit 1165 remover. After liftoff, the wafers were plasma and RCA-1 cleaned again, and a 1  $\mu$ m thick SiO<sub>2</sub> insulating layer was deposited via plasmaenhanced chemical vapor deposition (PECVD). A KMPR photoresist layer was then photolithographically patterned to define the area for the electrodes and electrical connection. The exposed  $SiO<sub>2</sub>$  was then etched with a CF<sub>4</sub> plasma. After etching, the wafer was spin-coated with a protective SPR 220− 4.5 photoresist layer for dicing and then diced into individual IAE devices by an electrical discharge machine. Prior to

making the electrical connection, the devices were cleaned by degreasing and plasma treatment (under  $O_2$  flow for 6 min). Since the devices were subjected to oxidative pretreatment, this led to the forming of an AuO*<sup>x</sup>* layer over the surface of the electrodes, which was removed by electrochemical cleaning in a phosphate buffer solution ( $pH = 7$ ).

**Electrochemical Cell Assembly.** Flex cables (Digi-Key, CA-DK05−12−30.0-A-44−88) were attached to the IAE devices by an anisotropic conductive film (3M, 3M970312- ND) followed by pressing under a manual heat press operated using a DC voltage supply set to 65−70 V, for 5.0 min. The IAE devices were then housed in a three-dimensional (3D) printed electrochemical cell sealed with an elastomer O-ring to avoid solution leaking. The cells were then connected to an 8− 1 multiplexer powered by an Arduino Uno CPU using a printed circuit board adaptor (Meccanixity FPC FFC Converter Board 12P 0.5 mm). Electrochemical measurements were performed by employing a three-electrode setup and a CHI760 potentiostat as the electrochemical workstation. The electrode array on the IAE device was used as the working electrode (WE), and a graphite counter electrode (CE) was connected via Cu tape. All measurements were made using a lab-made Ag/AgCl (3 M KCl) reference electrode (RE). The RE was introduced into the solution using an agar-based salt bridge to avoid contamination of the experiment cell with  $Ag<sup>+</sup>$ ions.

**Electrochemical Measurements.** *Cleaning and Validation.* Electrochemical cleaning of bare Au electrodes was done in a 0.1 M phosphate buffer solution ( $pH = 7$ ) by performing cyclic voltammetry (CV) measurements within the potential window of 0.2 to -1.5 V@0.1 V/s for 50 cycles. Before Edep, size validation studies were performed in 1.16 mM FcMeOH + 0.1 M  $KNO<sub>3</sub>$  on all electrode arrays reported in this study using CV measurements, which were recorded in the potential range of  $-0.1 - 0.5$  V at scan rates in the range of  $0.1 - 1.0$  V/s (100 mV/s increments) for 5 cycles at each scan rate.

*Electrodeposition of Au NPs.* For pulsed electrodeposition (PED) of Au-NPs, 1.0 mM  $HAuCl_4 + 0.5 M H_2SO_4$  was used as the precursor solution, while electrodeposition was performed at lower end potential  $(E_1)$  of  $-0.2$ ,  $-0.6$ ,  $-1.0$ , and  $-1.4$  V (vs Ag/AgCl) and at pulse widths (PW) = 0.01, 0.05, 0.2, 1.0, 5.0, 30.0, and 60.0 s. The higher-end potential  $(E_h)$  was held at a constant potential of 1.0 V (vs Ag/AgCl), and the total Edep time was fixed at 5.0 min. As a result, the number of pulses applied were 60,000, 12,000, 3000, 600, 120, 20, and 10 for their corresponding PW.

*Electrochemical Characterization of Au NPs.* To estimate the ECSA, we performed CV cycling in 0.5 M  $H_2SO_4$  within the potential window of  $-0.35 - 1.55$  V@0.1 V/s for 10 cycles. The charge passed during the reduction of the gold oxide layer is calculated using *Q*red = (area under peak/scan rate) and normalized using a factor of 390  $\mu$ C/cm<sup>2</sup> for pure Au.

*Glycerol Electro-Oxidation Reaction (GEOR).* Initially, the electrodes were conditioned by cycling in 0.02 M glycerol + 1 M KOH within the potential window of -1.0 - 0.5  $V@1V/s$  for 100 cycles. This was followed by GEOR performance evaluation through CV measurements in the same solution as above, for 40 cycles at 0.1 V/s.GEOR was also evaluated using LSV, performed in the potential window of −1.0 − 0.4 V@0.005 V/s. CVs in blank (1.0 M KOH) were also collected by cycling within the same potential window as mentioned above at 0.1 V/s for 3 cycles as background response. All electrochemical measurements are reported without *iR* compensation as electrode size, and thereby, the current scale is small enough to neglect the effect of *iR*-drop.

**Product Quantification via Flow Electrolysis.** We employed a lab-made flow electrolyzer to study product speciation under continuous flow conditions.<sup>[16](#page-11-0),[40](#page-12-0)</sup> A constant flow rate of 2 mL/min of electrolyte was maintained over both the WE  $(0.1 \text{ M}$  Glycerol + 1.0 M KOH) and CE  $(1.0 \text{ M})$ KOH) channels using a peristaltic pump (Cole Parmer Masterflex L/S). The two electrode channels were separated using an anion exchange membrane (AEM, Fumasep FAA-3- PK-75), conditioned by soaking in 1.0 M KOH for 24 h. A Ni foil coated with Pt (*t* = 8.0 min sputter) was employed as the CE. Three different WEs−one planar and two electrodeposited ones were used, where a Ni foil (conducting substrate) was coated with Au ( $t = 5.0$  min sputter), Au ( $E_1 = -0.6$  V, PW = 0.2 s), and Au  $(E_1 = -1.4 \text{ V}, \text{PW} = 1.0 \text{ s})$ . Using a T-junction, an Ag/AgCl RE was connected to the glycerol electrolyte stream running through the WE channel. All results for flow experiments are plotted versus RHE, where the potentials were converted using the formula

$$
E_{\rm RHE} = E_{\rm Ag/AgCl} + E_{\rm Ag/AgCl}^{0} + (0.0591 \times \rm pH)
$$

The electrochemical experiments were performed at constant anode potentials using a CHI 760e potentiostat. Like IAE-based experiments, the electrodes were first conditioned by cycling for 100 cycles at 1.0 V/s between  $-0.1 - 1.6$  V (vs RHE) and then cycling within the same potential window at 0.1 V/s for 10 cycles. We then performed chronoamperogram (CA) measurements at fixed potentials for 240.0 s, and products were sampled during the last 120.0 s and analyzed using high-performance liquid chromatography (HPLC). The charge passed during this sampling period was measured for Faradaic efficiency (FE) calculations, and the current density was averaged for *J*−*V* plots. The liquid products were analyzed using a Nexera 40 Series HPLC (Shimadzu Scientific Instruments) equipped with a Bio-Rad Aminex HPX-87C column. The liquid samples were neutralized with 0.5 M  $H_2SO_4$  in a 1:1 v/v ratio to maintain a pH of 1−3. A column temperature and flow rate of 60 °C and 0.6 mL/min were maintained, and a 20 *μ*L of liquid sample (0.05 M  $H<sub>2</sub>SO<sub>4</sub>$  mobile phase) was injected into the column. The products were detected using a ultraviolet−visible (UV−vis) detector with a *λ* = 210 nm. Since only the anode products were being investigated, and as glycerol majorly forms liquid products, the gaseous products were not analyzed. The concentration (mM), Faradaic efficiency (%), and relative selectivity (%) for each major product were calculated using the equations

$$
conc.prod = \frac{intensity}{slopeprod}
$$
  
FE<sub>prod</sub>(%) = 
$$
\frac{v \times t \times conc.prod}{Q/(zprod × F)} \times 100
$$

where the slopes for each product were determined by individual calibration curves,  $\nu$  = flow rate (mL/min),  $t =$ duration of electrolysis (*s*), *Q* = charge passed during each electrolysis CA, *F* = Faraday's constant (96,485 C/mol), *z* = number of electrons transferred to form one mole of the product.  $Z_{\text{oxalic}}$ ,  $z_{\text{glyceric}}$ ,  $z_{\text{glycolic}}$ ,  $z_{\text{formic}}$ , and  $z_{\text{acetic}}$  were 22:3, 4, 10:3, 8:3, and 8:3 respectively.

<span id="page-3-0"></span>

Figure 1. Schematic of electrocatalyst screening workflow. (a) Animated representation of automated electrochemical setup using an 8-electrode IAE array device powered by HardPotato. (b) Cyclic voltammetry on IAE array device for electrode size validation using electrochemical oxidation of ferrocenemethanol (1.16 mM FcMeOH + 0.1 M KNO<sub>3</sub>) at scan-rate = 0.5 V/s. Black and gray arrows indicate forward (anodic) and reverse (cathodic) potentials sweeps. (c) Chronoamperogram representing the electrodeposition of Au at two different reduction potentials: −0.2 V (purple) and −1.0 V (blue). (d) Cyclic voltammogram in 0.5 M H2SO4. (e) Cyclic voltammogram in 0.02 M Glycerol +1.0 M KOH. (f) SEM images of Au deposited at −0.2 and −1.0 V, respectively. (g) Spider graph comparing figures of merits for comprehensive evaluation of catalytic performance.

**Physical Characterization.** Morphology and particle size analysis of Au NPs was performed using the FEI Quanta FEG 450 environmental scanning electron microscopy (ESEM). Operating conditions during image acquisition were pressure =  $0.98-1.00$  Torr; dwell time = 10  $\mu$ s, accelerating voltage = 20 kV, current = 5 mA, magnification = 10k−60 kx. Crystal structure information on Au NPs was obtained by collecting Xray Diffractograms on the Rigaku MiniFlex 600 instrument with a Cu K*α* (*λ* = 1.5418 Å) X-ray source operated under 40 kV, 15 mA, and a 1.2°/min scan-speed. Denton Vacuum Desk V sputter coater was used to sputter-coat IAE devices and Ni foils using default Cr and Pt targets.

#### ■ **SCREENING METHODOLOGY**

Figure 1 depicts the experimental workflow in this manuscript, consisting of (a) device integration, (b−f) semiautomated electrodeposition and characterization, and (g) data analysis. Toward device integration, we first followed a microfabrication protocol $^{41}$  $^{41}$  $^{41}$  where we fabricated an array of individually addressable electrode (IAE) devices, each containing eight Au electrodes  $(r = 100 \ \mu m)$  on which distinct electrochemical protocols could be programmed. A concise overview of the fabrication steps, along with a workflow schematic, can be found in the [Materials](#page-1-0) and Methods Section and [Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S1. We established electrical connections to these devices using a flexible ribbon cable incorporating anisotropically conducting tape between the cable and the device. These array chips were then used in conjunction with an open-source Python API − HardPotato, enabling precise control over each electrode in the array. This control was achieved using an 8-to-1

multiplexer powered by an Arduino Uno, facilitating automated electrochemical operations at each of our electrodes via a printed circuit board adaptor depicted in Figures 1a and [S2](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf), which allowed us to switch from electrode #1 to #8 serially. This setup ensured effective cross-communication between the PC, potentiostat, multiplexer, and IAE device, leveraging the integration of the HardPotato API with the IAE array.

Prior to any electrocatalytic measurements, we benchmarked our bare electrodes using a reliable, fast redox mediator (ferrocenemethanol) at all the electrodes on our array (Figure 1b). To accurately determine the real electrode radius and identify any outliers resulting from the microfabrication process, we performed Randles-Sevčik analysis using cyclic voltammetry at various scan rates. Following this step and after thorough rinsing, we electrodeposited Au catalysts using chronoamperometry by pulsing between the open-circuit potential (OCP) and a reducing potential. We systematically varied (i) the pulse width (PW) and (ii) the reduction potential (*E*<sup>l</sup> ) to deposit an array of Au catalysts, each exhibiting distinct physical and electrochemical properties. A representative electrodeposition chronoamperogram is illustrated in Figure 1c, where  $E_1$  is modulated between  $-0.2$  and −1.0 V while maintaining a constant PW of 60.0 s. Post Edep, to electrochemically characterize the Au catalyst deposited, we performed cyclic voltammetry in 0.5 M  $H_2SO_4$  (Figure 1d). This analysis provided insights into the electrochemical surface area (ECSA) and the surface roughness factor (SRF), defined as the ratio of ECSA to the geometric area of the electrode.

Finally, rigorous electrocatalytic characterization of these Au catalysts was performed in 0.02 M Glycerol +1.0 M KOH.

<span id="page-4-0"></span>[Figure](#page-3-0) 1e depicts a typical GEOR CV response on an Au catalyst. Later sections provide a more in-depth analysis of the GEOR performance. There have been several excellent reports discussing the shape of voltammograms and mechanisms for the electro-oxidation of alcohols on gold electrodes in alkali media.[42](#page-12-0)<sup>−</sup>[46](#page-12-0) Briefly, glycerol is oxidized during the forward (anodic) sweep and this process is constrained by electrode kinetics and the availability of active sites. As the potential is made more positive, gold oxide (AuO*x*) and other reaction intermediates accumulate, diminishing the activity and thus resulting in a peak labeled as Peak 1 [\(Figure](#page-3-0) 1e). On the reverse (cathodic) sweep, the potential reaches a level sufficient to reduce the gold oxide layer, refreshing the metallic sites, thus leading to another set of oxidation peaks (Peak 2) associated with the oxidation of adsorbed GEOR intermediates or nearby glycerol molecules. In addition to CVs, we also performed linear sweep voltammetry (LSV) measurements to obtain onset potentials and Tafel slopes, which are critical metrics in electrocatalysis studies.

We further leveraged our IAE array design beyond electrochemical characterization, enabling morphological characterization via scanning electron microscopy (SEM). This approach directly correlates the catalysts' electrochemical performance with their physical properties. Illustrated in [Figure](#page-3-0) 1f are SEM images of gold (Au) electrodeposits synthesized at  $E_1 = -0.2$ , and  $-1.0$  V. This capability to adjust particle size exemplifies the versatility of our method, supporting rapid and efficient synthesis and evaluation of micro/nanoscale electrocatalysts directly on a single device.

As an illustration ([Figure](#page-3-0) 1g) of how all aspects in our methodology come together, when we contrasted the GEOR performance of Au deposited at −0.2, and −1.0 V, we noted an enhancement in current density, a decrease in peak positions, a lower onset potential, and a reduced Tafel slope, all of which are markers for improvements in electrocatalytic performance. The Edep characterization and comparison were performed for all 28 unique conditions tested, including 2 additional repeats per condition, thus demonstrating the reliability and versatility of our methodology.

## ■ **RESULTS AND DISCUSSION**

**Electrode Validation with FcMeOH.** We first depict how the IAE devices provide a reproducible platform on which the geometric area and electrochemical responses of each bare electrode are comparable. Leveraging HardPotato's capabilities, we performed CVs of the one-electron redox mediator ferrocenemethanol (FcMeOH) sequentially on each electrode of the IAE array device (Figure 2a). CVs were performed by scanning between −0.2 and 0.5 V and back at varying scan rates ranging from 0.1, to 1.0 V/s at each electrode (Figures 2b and [S3](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf)). From Figure 2b, it is evident that the oxidation of FcMeOH undergoes an electrochemically reversible reaction from which analysis of the voltammetric features can help confirm the geometric area, and thus the radius, of each electrode. We conducted Randles–Ševčík (RS) by plotting the anodic peak current  $(I_{pa})$  as a function of the square root of scan rate and applied linear regression to estimate the slope, from which the accurate radius of the electrode was derived. Prior to the measurements, we determined the concentration and diffusion coefficient of the FcMeOH solution to be 1.16 mM and 7.25  $\times$  10 $^{-10}$  m $^2/s$ , respectively, through steady-state measurements at an ultramicroelectrode (UME) ([Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S4).



Figure 2. Electrode benchmarking using FcMeOH. (a) Picture of an IAE array device (left) and an optical micrograph (right) of all the 8 Au disk electrodes  $(r = 100 \ \mu m)$  available on each IAE array device. (b) CVs for FcMeOH oxidation at each electrode on the IAE array at varying scan rates (0.1, 0.2, 0.3, 0.4, 0.5, 0.6, 0.7, 0.8, 0.9, 1.0 V/s). Inset (bottom-right): Scatter plot of anodic peak currents vs scan rate $1/2$  with the linear fit. Experimental conditions: 1.16 mM FcMeOH + 0.1 M KNO<sub>3</sub>. WE: Au; CE: Graphite rod; RE: Ag/ AgCl. (c) Scatter plots of the real electrode radius (left axis) and the corresponding size error (right axis) obtained from the slope of the current vs scan rate $1/2$  plots.

Primarily, RS analysis enables us to confirm the nature of mass transport occurring at the surface of these electrodes, which is linear diffusion ([Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S5), evidenced by excellent conformity with an  $R^2 \sim 1$  at all the electrodes on the array device. Additionally, we utilized RS analysis to quantify the yield of the microfabrication process. Figure 2c illustrates the calculated radius of each electrode and the corresponding error in size relative to the nominal 100 *μ*m (radius) electrode. All eight electrodes on our device have identical feature sizes with

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Figure 3. Representative plots for pulsed electrodeposition of Au NPs. (a) Potential-time waveform employed during electrodeposition. The potential was switched from  $E_h = 1.0$  V (const.) to varying  $E_l$  values after the = 30.0 s (total = 600.0 s, i.e.,  $n_{\text{seg}} = 20$ ).  $E_l = -0.2$  V (red),  $-0.6$  V (green), −1.0 V (blue), −1.4 (purple). (b) Corresponding current−time plot generated during the application of potential waveform. (c) The total charge passed during each Edep experiment as a function of  $E_1$  Experimental Conditions: 1.0 mM HAuCl<sub>4</sub> + 0.5 M H<sub>2</sub>SO<sub>4</sub>. WE: Au (sputter), CE: Graphite rod, RE: Ag/AgCl. (d−f) SEM Images of Au particles deposited at −1.4 V (PW = 0.05 s), −0.6 V (PW = 0.05 s), and −0.6 V (PW = 5.0 s), respectively.

deviations of  $\pm 1.3$   $\mu$ m, corresponding to a size error <1.5%. We attribute this small % error in size to the swelling or erosion of insulating photoresist walls during the microfabrication stage. This analysis also highlights a key advantage of our electrochemical setup: the ability to perform reliable measurements at nA-scale using disk electrodes, with minimal limitations from heterogeneities in electrical conductivity, potential drifts due to the use of quasi-reference electrodes, and ohmic losses arising due to larger currents at larger electrodes, most which are some common limitations of HT Echem screening technologies.

While our microfabrication process does not achieve a 100% success rate, the FcMeOH measurements allow the identification of faulty electrodes. A distribution of % error in electrode radii of all electrodes used in this study is depicted in [Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S6, and any electrode with a size deviation of >10% was excluded from future measurements. Such deviations are expected to occur during the microfabrication stages due to factors such as electrode shorting, human errors, scratches on the micropatterns, or high resistivity in the electrical connections resulting from nonuniform etching of silicon oxide layers. The CV plots of faulty electrodes are shown in [Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S7, where these errors are apparent in the form of purely resistive CVs exhibiting ohmic and/or capacitive behavior or significantly higher slopes during RS analysis compared to the ones in [Figure](#page-4-0) 2.

**Pulsed Electrodeposition of Au NPs.** In this section, we showcase our screening methodology's application toward probing the influence of driving force and time scales on the Edep rates and, consequently, the morphology of Au deposits. We employed pulsed electrodeposition (PED), a chronoamperometric technique that alternates short pulses between a higher potential  $(E_{\mathrm{h}})$  and lower potentials  $(E_{\mathrm{l}})$  across multiple segments.[38,47](#page-12-0),[48](#page-12-0) We controlled the driving force and time scales by manipulating the deposition potential and pulse width (PW), respectively. Figure 3a illustrates the Au Edep potential waveform, pulsing between  $E<sub>h</sub> = 1.0$  V and  $E<sub>l</sub> = -0.2$ , −0.6, −1.0, and −1.4 V for 30.0 s at each potential, repeated across 20 segments  $(n_{\text{seg}})$ , with the corresponding chronoamperogram (CA) depicted in Figure 3b. For simplicity, we maintained the higher-end potential constant in our experiments, fixing it at  $E_h = 1.0$  V, which closely approximates the cell's open-circuit potential, facilitating the replenishment of Au precursor after each reduction segment. Holding the total electrodeposition time constant  $(t_{\text{Edep}} = 5.0 \text{ min})$ , we systematically varied the PW from 60.0 to 30.0, 5.0, 1.0, 0.2, 0.05, and 0.01 s, respectively, at each  $E_1$ . With each combination of unique deposition parameters−a total of 28 conditions−and each condition done in triplicate on different days and random devices to track the reproducibility of each result, we conducted 84 full experiments overall.

The CAs in Figure 3b represent a typical Edep process characterized first by an initial exponential current decay due to double-layer capacitance (DLC), followed by a gradual rise due to continuous nucleation of fresh Au site, and finally trailed by a Cottrell decay. The effect of  $E<sub>1</sub>$  (driving force) on the rate of electrodeposition is evident in Figure 3c, where we measure the total reductive charge (*Q*red) passed in each CA.  $Q_{\text{red}}$  increases exponentially from 0.4 to 2553  $C/mm^2$  when the *t*Edep and PW are held constant at 5.0 min and 0.05 s, respectively, and  $E_1$  changes from  $-0.2$ , to  $-1.4$  V. Comparative SEM analysis of electrodeposits at different *E*<sup>l</sup> (Figure 3d,3e) reveals a significant increase in particle nucleation density at the more negative potential attributed to a substantially larger driving force applied during deposition. In addition to changes in particle density, a considerable



Figure 4. Electrochemical surface area estimation of selected results from CV in 0.5 M H<sub>2</sub>SO<sub>4</sub>. (a) Representative CVs performed at Au deposited at varying *E*<sup>l</sup> , holding PW fixed at 0.2 s depicting distinct features present at different Edep conditions (b, c) Surface Roughness Factor plotted as a function of  $E_1$  (fixed PW = 0.2 s), and PW (fixed  $E_1 = -1.4$  V) respectively.

increase in particle size from  $117 \pm 21.2$ , to  $280 \pm 66.5$  nm ([Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S8) was observed, and the morphology also shifts from a predominantly spherical shape to a cuboidal/tetragonal nanoparticle (NP), with higher agglomeration.

Unlike [Figure](#page-5-0) 3c, the variation of Q<sub>red</sub> with PW is more complicated, as the electrodeposition trends vary based on applied reduction potential ( $Figure S9$ ). At the more negative *E*l , a peak-shaped plot emerged with a maximum at ∼ PW = 0.2 s. The influence of PW on Au morphology can be understood by analyzing the SEM images of the deposits. As illustrated in [Figure](#page-5-0) 3e,[3f](#page-5-0), the morphology transitions from spherical NPs  $(117 \pm 21.2 \text{ nm})$  with uniform coverage to large needle-shaped microclusters  $(4.43 \pm 2.31 \ \mu m)$  with poor electrode coverage when the PW changes from 0.05 to 5.0 s. Shorter PW favors nucleation over particle growth, resulting in high particle density and smaller sizes. Conversely, longer deposition times lead to more pronounced particle growth and coalescence due to extended durations of applied driving force, resulting in a noticeable increase in particle size.[49](#page-12-0) Moreover, the decreasing particle density with increasing diameter is likely due to electrochemically enhanced Ostwald ripening, as reported by Redmond et al., occurring during the experimental time scale, resulting in these larger Au particles.<sup>[50](#page-12-0)</sup> These results underscore how the interplay of a vast space of electrodeposition variables and factors condition the formation of significantly different structures, which, as we will show in the following section, have distinct electrochemical properties. Our approach swiftly enables the semiautomated deposition and analysis of these structures, including repeats, to ensure reproducibility.

**Electrochemical Characterization.** We determined the electrochemical surface area (ECSA) and the surface roughness factor (SRF) of the gold (Au) electrodeposits through AuO<sub>x</sub> stripping analysis performed in 0.5 M  $H_2SO_4$ , a well-established technique for estimating ECSA.<sup>[51](#page-12-0)</sup> We performed CVs by sweeping the potential between −0.35 V and 1.55@0.1 V/s (Figures 4a and [S10\)](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf). A series of peaks were observed on the anodic sweep from 0.95 to 1.45 V, corresponding to the oxidation of surface gold sites, with each peak typically associated with specific Au surface orientations. Conversely, we observed a sharp Gaussian-shaped peak at approximately 0.86 V on the cathodic sweep, corresponding to the reduction of surface AuO*x*. This peak was the basis for calculating the ECSA ([Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S11) and, subsequently, the SRF. To facilitate the analysis of large data sets, we employed a custom-written Python script for batch processing of all CVs. This involved automatic plotting, baseline fitting of the reduction peak, and estimation of *Q*<sub>AuOx</sub> − the charge passed stripping of oxide peak (shaded region under peak @0.86 V in [Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S10). The scripts also enabled the subsequent plotting of SRF (obtained from *Q*AuOx) as a function of the Edep parameters, as depicted in Figures 4b,4c, and [S12](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf). Note that these CVs were performed for over 5 cycles on each electrode as repeated cycling is known to induce surface restructuring and electrochemically roughen the electrode (refer to [Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf)  $S13$ ).<sup>52</sup>

As demonstrated in the previous section and corroborated by existing literature, increasing deposition potentials lead to the formation of higher-density nanoparticles (NPs) with rougher surfaces. According to the Butler−Volmer kinetic model, the nucleation rate is expected to increase exponentially with an increase in potential.<sup>39</sup> Baker and co-workers have illustrated that controlling the  $E<sub>l</sub>$  can effectively manipulate the kinetics of metal ion reduction and, consequently, the nucleation rate.<sup>[53](#page-13-0)</sup> They report a lower particle yield at less negative *E*<sup>l</sup> values, resulting in more spherical nanostructures that evolve into cubic nanoparticles with sharper edges and significantly higher particle yields at more negative  $E<sub>1</sub>$  values. A similar effect manifests in our study as well, albeit the structural changes are not as evident due to the absence of chemical additives. However, the effects on particle yield are striking, as confirmed by SEM images [\(Figure](#page-5-0) 3c,[3d](#page-5-0)) and ECSA measurements. Specifically, there is a significant increase in the surface roughness factor (SRF) from 2.3 to 37.5 as the *E*<sup>l</sup> changes from  $-0.2$ , to  $-1.4$  V (vs Ag/AgCl), as illustrated in Figure 4b.

An interesting relationship emerges when correlating ECSA with the PW during electrodeposition. Holding other variables constant ( $t_{\text{Edep}}$ ,  $E_{\text{L}}$  and  $E_{\text{L}}$ ), we study the ECSA-PW trend as shown in Figure 4c. We hypothesize that the Edep process is constrained by the dominance of DL discharge at very short pulses. To maintain the same  $t_{\text{Edep}}$ , the number of segments  $(n_{\text{seg}})$  varied with PW—for example, at PW = 0.05 s,  $n_{\text{seg}}$  is 12,000, compared to just 10 segments at  $PW = 60.0$  s. Thus, at very short PW (0.01 and 0.05 s), DL discharge primarily

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Figure 5. Evaluating GEOR performance as a function of Au Edep parameters. (a) Representative GEOR CVs at different *E*<sup>l</sup> and fixed PW (1.0 s). (b, c) Peak current density ( $I_f$ ) and peak position ( $V_f$ ) during the forward sweep of the CV are presented as a function of  $E_1$  when PW = 1.0 s and as a function of PW when *E*<sup>l</sup> = −1.4 V, respectively. (d) Representative GEOR slow-scan LSV at different *E*<sup>l</sup> when PW = 1.0 s. (e, f) GEOR characterization by Onset Potential (orange) and *V*<sup>10</sup> (yellow) as a function of *E*<sup>l</sup> and PW, respectively.

contributes to the currents measured in each reduction segment as opposed to the Edep process. This hypothesis is supported by a lower SRF of 1.54 recorded at a PW of 0.01 s (close to a typical 1.0−1.2 for a planar electrode) compared to 37.5 when PW reaches 0.2 s. However, the SRF declines to 11.2 as PW extends to 1.0 s and further decreases to 6.4 at 60.0 s. This trend of variation in PW remains consistent across different electrodeposition potentials, as seen in [Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S12. This significant decline at longer PWs can be attributed to drastic morphological changes (dendritic growth, [Figure](#page-5-0) 3d), leading to the formation of microstructures and, consequently, a reduced electroactive surface area.

However, our triplicate measurements reveal noticeable discrepancies in the ECSAs measured for gold deposited under varying conditions. As illustrated in [Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S14, the inconsistency in Au loading is influenced by both the  $E_1$  as well as the PW. Notably, when  $E_1$  is set to  $-0.6$ ,  $-1.0$ , and  $-1.4$ V, there is a visible deviation in measured ECSA at short PWs (0.05 and 0.20 s). We speculate that this may be due to larger hydrogen evolution reaction (HER) rates occurring at these potentials. The generation of hydrogen bubbles during HER likely obstructs the electrode surface, leading to irregularities in gold deposits.[54](#page-13-0) This variance in ECSA occurs despite no significant deviation in the charge passed during the reduction segments ([Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S9). This suggests that the Coulombic efficiency split between HER and metal ion reduction is inconsistent at these shorter PWs. The ability to capture reactive trends and determine the electrodeposits' robustness is

a feature conveniently enabled by our semiautomated approach.

**Glycerol Electro-Oxidation Reaction.** The electrochemical study of glycerol oxidation was primarily carried out by performing CVs within the range of -1.0 -0.5 V (V vs Ag/ AgCl) in 0.02 M glycerol +1.0 M KOH at 0.1 V/s (Figure 5a). We performed our experiments in alkaline conditions owing to the higher reactivity of glycerol in these conditions, as pH can influence current density and product distribution.<sup>6,[55](#page-13-0),[56](#page-13-0)</sup> The fast kinetics in alkali media are attributed to the base-catalyzed glycerol deprotonation step leading to the generation of glyceroxide species, whose adsorption on the active site is the rate-determining step in the oxidation reaction.<sup>[46](#page-12-0)</sup> Similarly, other reports discuss the influence of increasing glycerol concentration on the electrochemical performance. Furthermore, while the peak current densities have been reported to rise due to an increase in reactive species in solution, the peak positions and the onset potentials shift toward more positive potentials explained by the Langmuir−Hinshelwood-like mechanism due to competition adsorption between glycerol and OH<sup>−</sup>. [56](#page-13-0) We found that a concentration of 0.02 M glycerol afforded an easily measurable response while preventing the blocking of Au sites toward OH<sup>−</sup> adsorption.

Prior to GOR measurements, CV at 1.0 V/s was performed in the same electrolyte as a conditioning step to allow for surface reconstruction and stabilization of active sites [\(Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) [S15\)](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf).<sup>[57](#page-13-0)</sup> A comprehensive compilation of all GEOR CVs performed in this study is shown in [Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S16, with each color palette representing each *E*<sup>l</sup> employed during deposition and

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Figure 6. Relative no. of reactive Au sites as a function of (a)  $E_l$  and (b) PW. Surface facets were identified based on peak assignments from literature[.59](#page-13-0)<sup>−</sup>[61](#page-13-0) Note: Sputtered Au recorded to have exclusively Au(111) sites, with a relative no. of sites (∼100%).

each independent CV within the column representing the GEOR CV of the Au catalyst deposited at a different PW. To thoroughly test the robustness of our approach, the error associated with each measurement is shown as a shaded region representing the standard deviation around the mean, from *n* = 3 different catalyst deposits. As illustrated in [Figure](#page-7-0) 5b[,5](#page-7-0)c, the electrocatalytic performance can be evaluated using several CV features, but we focused on the peak current density  $(I_{\mathrm{f}})$  on the anodic forward sweep and its corresponding position  $(V_{\rm f})$ .

One common oversight in GEOR literature is the lack of discourse on the GEOR CVs over multiple cycles, as most catalysts see activity decay upon cycling due to surface passivation, reconstruction, or metal dissolution. Addressing this oversight, we performed our GEOR CVs for over 20 cycles and have reported the CV features of the last cycle in [Figure](#page-7-0) [5](#page-7-0)b[,5c](#page-7-0). We also report the  $I_f$  and  $V_f$  values for the first and the 10th cycles of CVs in [Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S17 to visualize trends in activity as a function of cycling. Additionally, we also performed linear sweep voltammogram (LSV) measurements ([Figure](#page-7-0) 5d) at a scan rate of 0.005 V/s to measure the onset potential, defined as the potential needed to achieve 1.0 mA/cm $^2$ , as well as the potential needed to achieve a current density of 10.0 mA/cm $^2\!$ , which are denoted by  $V_{\text{Onset}}$  and  $V_{10}$  respectively. [Figures](#page-7-0) 5e[,5f](#page-7-0), and [S18](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) depict the  $V_{\text{Onset}}$  and  $V_{10}$  plotted as a function of PW and *E*<sup>l</sup> , while the corresponding LSVs from which this data was obtained can be found in [Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S19. Furthermore, we generated Tafel plots to determine Tafel slopes from these LSVs using basic data processing scripts, as shown in [Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) [S20.](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf)

We first validated our technique by exploring the dependence of GEOR on the deposition potential, as dependencies on this parameter are well established in the literature for alcohol electro-oxidation.[38,48,](#page-12-0)[58](#page-13-0) [Figure](#page-7-0) 5a depicts GEOR CVs of Au deposited at varying *E*<sup>l</sup> values, maintaining PW at 1.0 s. Studying the peak features of this CV as a function of  $E<sub>l</sub>$  reveals a direct correlation between deposition potential and GEOR activity ([Figure](#page-7-0) 5b). We observed a direct correlation between deposition potential and GEOR activity, with a 52% increase in  $I_f$  and a 0.250 V reduction in  $V_f$  as  $E_l$  changes to a more negative potential. This improvement in catalytic activity is further corroborated by the measured trends in  $V_{\text{Onset}}$  and  $V_{10}$ , as −0.208 and −0.205 V shifts were recorded, respectively. While we did not rely on the Tafel slope as a critical tool for measuring catalytic activity, basic Tafel analysis ([Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S20e− [h](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf)) further reinforces our findings by indicating a 13 mV/Dec reduction in the Tafel slope.

While the dependence of GEOR activity on  $E_1$  on Au has been previously reported, our approach allowed us to swiftly explore the impact of PW, as well. Across all four *E*<sub>l</sub> potentials examined; we identified an optimal GEOR activity at around PW = 1.0 s. This optimized PW duration minimized DLC effects and was sufficiently brief to induce preferential nucleation of NPs with highly active sites while preventing particle agglomeration at much longer PWs (>1.0 s). While observing PW–GEOR performance trends in the case of *E*<sub>l</sub> = −1.4 V, as depicted in [Figure](#page-7-0) 5c, we noticed a 36% increase in  $I_f$  and a 0.22 V reduction in  $V_f$  as PW varied from 0.01 to 1.0 s. Unlike reactive trends with  $E_{\rm b}$  the  $V_{\rm onset}$  and  $V_{\rm 10}$  changes are less drastic with PW variation within this range [\(Figure](#page-7-0) 5f).

As  $E_1$  is made more negative, there is an increase in mass loading of Au on the electrocatalyst, and this parameter is correlated to a higher activity, as grasped from [Figure](#page-7-0) 5b[,5](#page-7-0)e. However, CV analyses in 0.5 M  $H_2SO_4$  allow us to infer surface properties that are likely more relevant to electrocatalysis, such as the faceting of Au deposited under varying conditions. Kim et al. reported that the presence of  $Au(110)$  sites significantly enhances electro-oxidation activity.<sup>[59](#page-13-0)</sup> Similarly, there have been other reports that explore the facet-dependent activity of Au(100), Au(111), and Au(110) for alcohol oxidation, underscoring the importance of these facets in reducing the energy barrier for  $Au(OH)_x$  formation.<sup>60,[62,63](#page-13-0)</sup> Figures 6 and [S21](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) illustrate how CV data can be analyzed to estimate the presence of specific Au facets at various electrodeposition conditions. Although this analysis only estimates the type of surface site, it is quite revealing to observe great diversity in the voltammetric profiles of this  $Au(OH)_x$  formation region. A common trend observed in cases of varying *E*<sup>l</sup> (Figure 6a) and PW (Figure 6b) is the correlation between step/kink edges and Au(100) sites with GEOR activity, where a lower relative number of these sites results in enhanced activity. Conversely, the presence of Au(111) + Au(110) sites results in higher activity. One important test for this analysis is to provide insight into stark differences in reactivity. For example, a sharp peak at around 1.0 V in the case of  $E_1 = -1.4$  V and PW = 60.0 s could likely correspond to a higher index faceted site such as Au(210) or Au(310) formed upon extensive electroreduction.

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Figure 7. Glycerol electro-oxidation via flow-electrolysis. (a) Picture of lab-made flow-electrolyzer cell. (b) GEOR CV performed at 0.1 V/s. The 20th cycle is represented in the plot. (c) *J*−*V* curves representing mean current density obtained from running a 240.0 s CA. (d) Concentration and (e) Faradaic Efficiencies of all major products quantified via HPLC when *E* = 1.4 V vs RHE. (f) X-ray Diffractograms of Planar Au (peach), Au(*E*<sup>l</sup> = −0.6 V) (blue), and Au (*E*<sup>l</sup> = −1.4 V) (violet). Note: Peaks labeled with <sup>⧫</sup> correspond to the underlying Ni foil.

It is precisely this catalyst that showed a sudden increase in catalytic performance as PW increased from 30.0 to 60.0 s in [Figure](#page-7-0) 5c. Interestingly, such comparisons hold even for underperforming catalysts produced at greatly different PW. Consider the surface and activity comparison of catalysts obtained at  $E_1 = -1.4$  V and PW 0.01 and 30.0 s. These two catalysts display a nearly identical surface feature distribution ([Figures](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S21a,f, and [6](#page-8-0)b) as per our analysis, and they also exhibit the same lowest catalytic GEOR activity evident from [Figure](#page-7-0) 5c.

**Scale-Up Flow Electrolyzer Testing.** Finally, this section addresses the translations of our observations from the semiautomated electrocatalyst screening to a flow reactor. A critical challenge most electrocatalyst screening architectures face is their scalability to large-scale, real-world electrolyzer systems. To address this, we validated the results obtained from our screening methodology by scaling up synthesis from  $\mu$ m-scale electrodes to electrodes of 3 cm<sup>2</sup> using a custommade cell holder, as shown in [Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S22. This allowed for the electrodeposition of Au nanoparticles (NPs) on a Ni foil. Reports have indicated that employing flow-electrolyzers to study product speciation and Faradaic efficiencies (FE) is an effective strategy for evaluating the practical electrochemical performance of catalysts.<sup>[12](#page-11-0),[40](#page-12-0),[64,65](#page-13-0)</sup> To this end, we used a labmade flow electrolyzer (Figure 7a) cell to study the performance of GEOR of the top-performing candidates from IAE screening experiments.<sup>1</sup>

Collating the entire range of GEOR performance metrics across all *E*<sup>l</sup> and PWs provides valuable insights into activity, reproducibility, and stability. While  $Au(E_1 = -1.0 V, PW = 0.2$ s) did yield an initial  $I_f$  > −40.0 mA/cm<sup>2</sup> during the first cycle ([Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S17g,i), The  $I_f$  and  $V_f$  fluctuated greatly across the 20 cycles, indicating poor catalyst stability. Additionally, this Edep condition also displayed poor consistency during ECSA measurements ([Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S14), suggesting poor reproducibility of this condition. In contrast, two other top candidates demonstrated strong activity, reproducibility, and stability: Au( $E_1$  = −0.6 V, PW = 0.2 s) and Au( $E_1$  = −1.4 V, PW = 1.0 s). Thus, these conditions were scaled up to  $cm<sup>2</sup>$ -scale electrodes and employed in flow-electrolysis setups. We also performed flow-electrolysis studies on a planar Au (sputtered) electrode as a benchmark. A detailed discussion regarding the flowelectrolysis setup, high-performance liquid chromatography (HPLC) measurements, and Faradaic efficiency calculations can be found in the [Materials](#page-1-0) and Methods Section.

Similar to our screening methodology using IAEs, we electrochemically conditioned our electrodes in 0.1 M Glycerol + 1.0 M KOH and recorded CVs at 0.1 V/s to identify potentials of interest (Figure 7b). CAs were subsequently performed for 240.0 s at these selected potentials [\(Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) [S23a](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf)−c), and the liquid samples were collected for HPLC measurements ([Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S23d−f). Product concentrations were quantified using these chromatographs by performing calibration curves for each product [\(Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S24). We also performed

<span id="page-10-0"></span>similar flow experiments in 0.02 M Glycerol +1.0 M KOH to study product speciation and Echem performance under the same conditions as in IAE screening experiments [\(Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S25).

While the  $I_f$  and  $V_f$  measurements from CVs under flow conditions do not exactly match the values from screening experiments, this is expected as IR-drop and complex mass transport of glycerol species manifest at these large-scale electrodes due to increased size and convection due to flow, respectively. However, as seen in [Figure](#page-9-0) 7b,[7](#page-9-0)c, the trends in catalytic performance do translate. Like the screening experiments, even under flow conditions,  $Au(E_1 = -1.4 \text{ V}, \text{PW} = 1.0$ s) achieves higher *I*<sub>f</sub> at a lower *V*<sub>f</sub> value than  $Au(E_1 = -0.6 \text{ V})$ . However, it is noteworthy that the planar Au outperforms the Au( $E_1$  = -0.6 V, PW = 0.2 s) electrode, indicating a poor translatability of this deposition condition to practical scales. *J*−*V* curves also bolster these observations ([Figure](#page-9-0) 7c). Upon comparing average current density  $(J_{\text{avg}})$  at a fixed *E* (1.2 V vs RHE), Au( $E_1 = -1.4$  V, PW = 1.0 s) achieves  $J_{avg}$  of  $-49.5$ mA/cm<sup>2</sup>, significantly larger than  $J_{\text{avg}}$  for Au( $E_{\text{l}}$  =  $-0.6$  V, PW = 0.2 s) and planar Au, which were  $-13.0$  and  $-19.1$  mA/cm<sup>2</sup>, respectively, thus showing the value of our exploration in identifying truly robust Edep conditions for electrocatalysis. From [Figure](#page-9-0) 7d, it is evident that the product yield is also higher in the case of  $Au(E_1 = -1.4 \text{ V}, \text{PW} = 1.0 \text{ s})$ , as compared to Au( $E_1$  = −0.6 V, PW = 0.2 s) and planar Au, in line with  $J_{\text{avg}}$ measurements as shown in [Figure](#page-9-0) 7c. A more detailed potential-dependent product speciation study is found in [Figures](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S26 and S27.

Correlating product speciation trends to the nature of the gold electrode, we observed a notable difference: lower Faradaic efficiency (FE) contributions [\(Figure](#page-9-0) 7e) toward glyceric acid and higher combined contributions toward glycolic and formic acids in both the electrodeposited samples compared to the planar gold electrode. This difference is expected, as glyceric acid (4e<sup>−</sup> transfer step) is a C3 product formed simply via oxidation of glycerol's terminal OH group, while formic  $(C1)$  and glycolic  $(C2)$  acids are formed by further oxidation of glyceric acid via another 2e<sup>−</sup> transfer process. It has been commonly reported that cleavage of the C − C at the gold site is responsible for the generation of formic and glyceric acid, and this ability is often considered an indicator of catalytic activity.<sup>40,42,[66](#page-13-0)</sup>

As illustrated in [Figure](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S22, our deconvoluted CVs did reveal a high fraction of Au $(110)$  and Au $(111)$  sites on the electrodeposited samples. Furthermore, comparing XRDs of Au( $E_1 = -1.4$  V, PW = 1.0 s) and Planar Au, [Figure](#page-9-0) 7f reveals higher intensity peaks with smaller fwhms recorded at 64.6 and 81.8°, corresponding to the Au(220) and Au(222) peaks. This indicates the presence of large, well-ordered crystals with a higher fraction of  $Au(110)$  and  $Au(111)$  surface sites. These higher active sites, especially Au(110), are expected to increase the reactivity of glyceric acid to formic and glycolic acid, facilitated by enhanced OH adsorption.<sup>[59](#page-13-0)</sup>

At lower operating potentials (∼0.9 V vs RHE), the glycerol oxidation pathway proceeds mainly through a 4e<sup>−</sup> transfer process. As potential increases, glycerol undergoes further oxidation to generate formic and glyceric acid via a 5e<sup>−</sup> and 8e<sup>−</sup> transfer process, respectively. As seen in [Figures](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) S26 and [S27,](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf) the production of these C2 and C1 products increases significantly with applied potential as  $C - C$  bond cleavage is accelerated at more oxidative potentials. This increase is even more pronounced in the case of  $Au(E_1 = -1.4 \text{ V}, \text{PW} = 1.0 \text{ s})$ than in planar Au, which further reinforces the role of these active sites generated via Edep. These flow electrolysis results highlight the apparent dependence of GEOR reactivity and product formation on the nature of the electrocatalyst, applied potentials, and concentration of glycerol as well. In the next stage of our screening methodology, we intend to further improve GEOR reactivity by exploring Au−Pd−Pt trimetallic systems $\frac{6}{x}$  and increasing the methodology's efficiency by automating solution handling.<sup>[34](#page-12-0),[68](#page-13-0)</sup>

#### ■ **CONCLUSIONS**

In this study, we introduced a semiautomated approach using IAE devices and HardPotato for the deposition, screening, and analysis of heterogeneous electrocatalysts. Utilizing this platform, we conducted an in-depth examination of gold catalysts for glycerol electro-oxidation, demonstrating that electrodeposition parameters−particularly reduction potential  $(E_1)$  and pulse width (PW) significantly dictate catalytic performance. By rigorously optimizing these parameters, we identified optimal conditions of  $E_1 = -1.4$  V and PW = 1.0 s, which maximized activity and minimized issues like particle agglomeration, instabilities, and reproducibility inconsistencies. Notably, while establishing a direct correlation between  $E_1$  and GEOR activity, we discovered a peak-shaped relationship between PW and GEOR activity, which was particularly significant in identifying the superior electrocatalyst. Furthermore, our analysis elucidated the critical influence of specific surface facets in augmenting glycerol electro-oxidation, determined by the electrodeposition conditions. Finally, the scalability of our approach was validated by transitioning from *μ*m-scale to cm-scale electrodes, which demonstrated enhanced product speciation and a stronger tendency for C− C bond cleavage in electrodeposited Au compared to planar Au electrodes, as evidenced by detailed product speciation studies.

#### ■ **ASSOCIATED CONTENT**

#### $\bullet$  Supporting Information

The Supporting Information is available free of charge at [https://pubs.acs.org/doi/10.1021/acscatal.4c04190.](https://pubs.acs.org/doi/10.1021/acscatal.4c04190?goto=supporting-info)

Further information regarding the microfabrication process, experimental methodology, such as electrodeposition results, electrocatalytic and other relevant electrochemical measurements, Tafel plots, SEM images, particle size analysis, flow-electrolyzer and HPLC measurements, and COMSOL Simulations ([PDF\)](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_001.pdf)

Photomask design for IAE device, sample Python scripts employed to run electrodeposition and other relevant electrochemical measurements, Arduino code to control the multiplexer, Python scripts for data analysis of electrochemical experiments, and COMSOL Simulation Report [\(ZIP\)](https://pubs.acs.org/doi/suppl/10.1021/acscatal.4c04190/suppl_file/cs4c04190_si_002.zip)

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#### **Author Contributions**

All authors provided significant contributions to this work in various ways. J.R.L. conceived the project, and R.G. designed the methodology, conducted experiments, performed data analysis, and wrote the paper. Z.W. performed the microfabrication of IAE device arrays. Y.L. and E.R.G. assisted in electrodeposition experiments and electrocatalytic measurements. Y.L. also assisted in SEM image acquisition and XRD measurements. L.C.H. assisted in flow-electrolyzer experiments and the acquisition of HPLC data. M.A.P. assisted in the design of experiments and microfabrication of IAE devices. P.J.A.K., A.A.G., and J.R.L. provided supervision, oversaw project administration, acquired funding for the project, and contributed to the discussion. All authors have given approval to the final version of the manuscript.

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### **Notes**

The authors declare no competing financial interest.

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