

# InP<sub>3</sub> Monolayer as a Promising 2D Sensing Material in SF<sub>6</sub> Insulation Devices

Xin Qin,\* Chenchen Luo, Yaqian li, and Hao Cui

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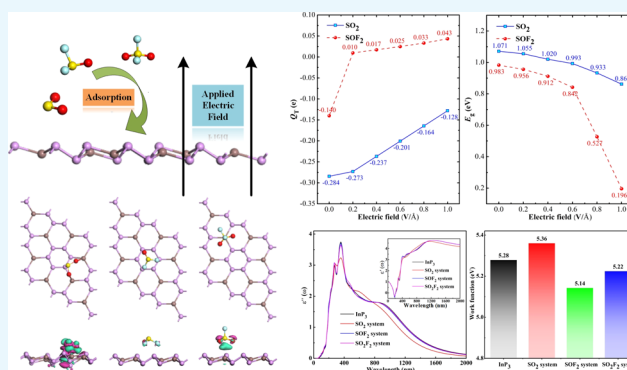
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**ABSTRACT:** In this letter, we perform a first-principles study on the adsorption performance of the InP<sub>3</sub> monolayer upon three SF<sub>6</sub> decomposed species, including SO<sub>2</sub>, SOF<sub>2</sub>, and SO<sub>2</sub>F<sub>2</sub>, to investigate its potential as a resistance-type, optical or field-effect transistor gas sensor. Results indicate that the InP<sub>3</sub> monolayer exhibits strong chemisorption upon SO<sub>2</sub> but weak physisorption upon SO<sub>2</sub>F<sub>2</sub>. The most admirable adsorption behavior is upon SOF<sub>2</sub>, which provides a favorable sensing response (−19.4%) and recovery property (10.4 s) at room temperature as a resistance-type gas sensor. A high response of 180.7% upon SO<sub>2</sub> and a poor one of −1.9% upon SO<sub>2</sub>F<sub>2</sub> are also identified, which reveals the feasibility of the InP<sub>3</sub> monolayer as a resistance-type sensor for SO<sub>2</sub> detection with recycle use via a heating technique to clean the surface. Moreover, the InP<sub>3</sub> monolayer is a promising optical sensor for SO<sub>2</sub> detection due to the obvious changes in adsorption peaks within the range of ultraviolet and is a desirable field-effect transistor sensor for selective and sensitive detection of SO<sub>2</sub> and SOF<sub>2</sub> given the evident changes of  $Q_T$  and  $E_g$  under the applied electric field.



## 1. INTRODUCTION

Two-dimensional (2D) materials, such as graphene,<sup>1</sup> phosphorene,<sup>2</sup> transition metal dichalcogenides,<sup>3</sup> tellurene,<sup>4</sup> and carbon nitrides,<sup>5</sup> raise tremendous interest in the field of gas sensing due to their unique electronic behaviors, large specific surface area, and strong chemical reactivity.<sup>6,7</sup> Although there are many reports about 2D gas sensors, the exploration of novel 2D semiconductors with high carrier mobility, good stability, and desirable sensitivity upon some noble gases is still limited,<sup>8,9</sup> which in fact is highly desired in material and engineering fields. From this point of view, computationally predicting the potential of certain 2D nanomaterials for sensing application is a workable and high-effective manner of promoting the development of related fields.<sup>10</sup>

Very recently, a series of 2D metal phosphides including GeP<sub>3</sub>,<sup>11</sup> CaP<sub>3</sub>,<sup>12</sup> and InP<sub>3</sub><sup>13</sup> have been reported with high carrier mobility. In particular, the InP<sub>3</sub> monolayer is theoretically reported with high chemical stability, desirable indirect semiconducting property, and outstanding optical property,<sup>14</sup> which allows its application as a novel nano-electronics. For example, Liu et al. reported the strong potential of the InP<sub>3</sub> monolayer to be explored as a reversible anode material for ultrafast charging.<sup>15</sup> Besides, the InP<sub>3</sub> monolayer has also been proposed as a sensing candidate for the detection of HCHO<sup>16</sup> based on the strain engineering, the n-type gas sensor for NO<sub>2</sub> sensing,<sup>13</sup> and sensing SF<sub>6</sub> decomposed species.<sup>17</sup> Therefore, the admirable electric

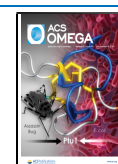
performance of the InP<sub>3</sub> monolayer with a tunable band gap and good carrier mobility provides strong potential in the gas–surface interactions for further exploration of a novel 2D semiconductor.

SF<sub>6</sub> is widely used insulation medium in high-voltage devices to guarantee the safe operation of the power system.<sup>18</sup> However, in a long-running device, inevitable insulation defects would cause partial discharge and ionize SF<sub>6</sub>, generating the F atom and low-fluorine sulfides. In most cases, these byproducts will recombine to SF<sub>6</sub>, but in the presence of trace water and oxygen, they will continue to interact with these impurities and form several stable gas compounds such as SO<sub>2</sub>, SOF<sub>2</sub>, and SO<sub>2</sub>F<sub>2</sub>.<sup>19</sup> These noxious species would largely deteriorate the insulation behavior of SF<sub>6</sub>, therefore threatening the safe operation of related devices. In this regard, the detection of SF<sub>6</sub> decomposed species using chemical gas sensors is proposed to evaluate the operation status of SF<sub>6</sub> insulation devices and ensure the safe running of the whole power system.<sup>20,21</sup>

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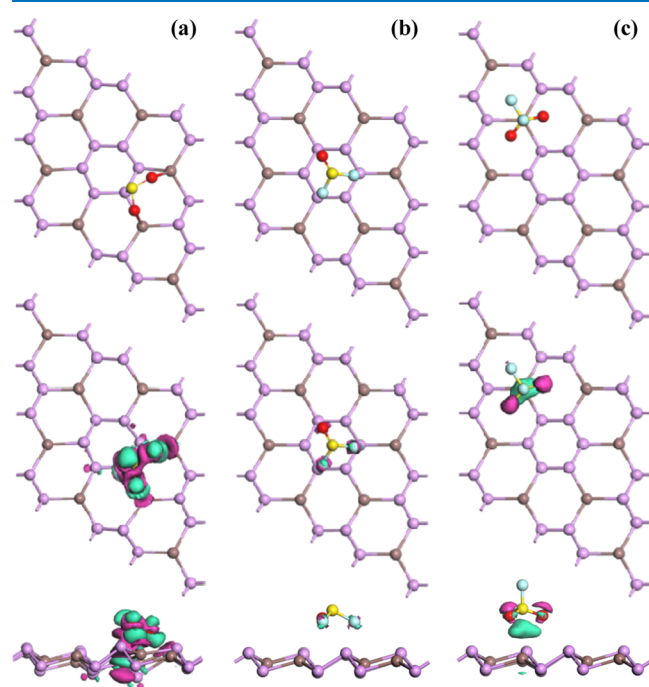


It has been investigated that the  $\text{InP}_3$  monolayer has polar bonds that would form strong dipole–dipole forces with certain polar molecules.<sup>16</sup> Given the polar behavior of the  $\text{SF}_6$  decomposed species, we assume that the  $\text{InP}_3$  monolayer can exhibit strong interactions with the typical gas species. To verify our assumption, we in this letter theoretically investigate the adsorption and sensing behaviors of the  $\text{InP}_3$  monolayer upon three  $\text{SF}_6$  decomposed products ( $\text{SO}_2$ ,  $\text{SOF}_2$ , and  $\text{SO}_2\text{F}_2$ ) using first-principles theory. Our work means to explore the possible availability of the  $\text{InP}_3$  monolayer as a sensing nanomaterial for detecting  $\text{SF}_6$  decomposed species. In the meanwhile, most theoretical studies focus on exploring the resistance-type gas-sensing application of the  $\text{InP}_3$  monolayer, while the optical and field-effect transistor sensing mechanisms of such 2D nanomaterials are less investigated. Thus, we will highlight such a perspective in this work to give detailed analysis about the  $\text{InP}_3$ -based gas sensor. It is our hope that the results in this work can stimulate further exploration of the  $\text{InP}_3$  monolayer to be a resistance-type, optical or field-effect transistor gas sensor in the near future.

## 2. RESULTS AND DISCUSSION

The most energy-favorable configurations for  $\text{SO}_2$ ,  $\text{SOF}_2$ , and  $\text{SO}_2\text{F}_2$  adsorption onto the  $\text{InP}_3$  monolayer were obtained by setting various initial adsorption structures with the initial distance of about 2 Å and selecting the one with the maximum  $E_{\text{ad}}$ . The gas molecules appear in different orientations to the possible adsorption sites of the  $\text{InP}_3$  surface, including on the top of the P-hexatomic ring and the top of the In atom.

Figure 1 shows the most stable configurations and related electron density difference (EDD) for gas adsorptions, while Table 1 lists the adsorption parameters. From Figure 1, one can see that the  $\text{SO}_2$  molecule prefers to be adsorbed on the top of a P atom with S being trapped, and two O atoms are



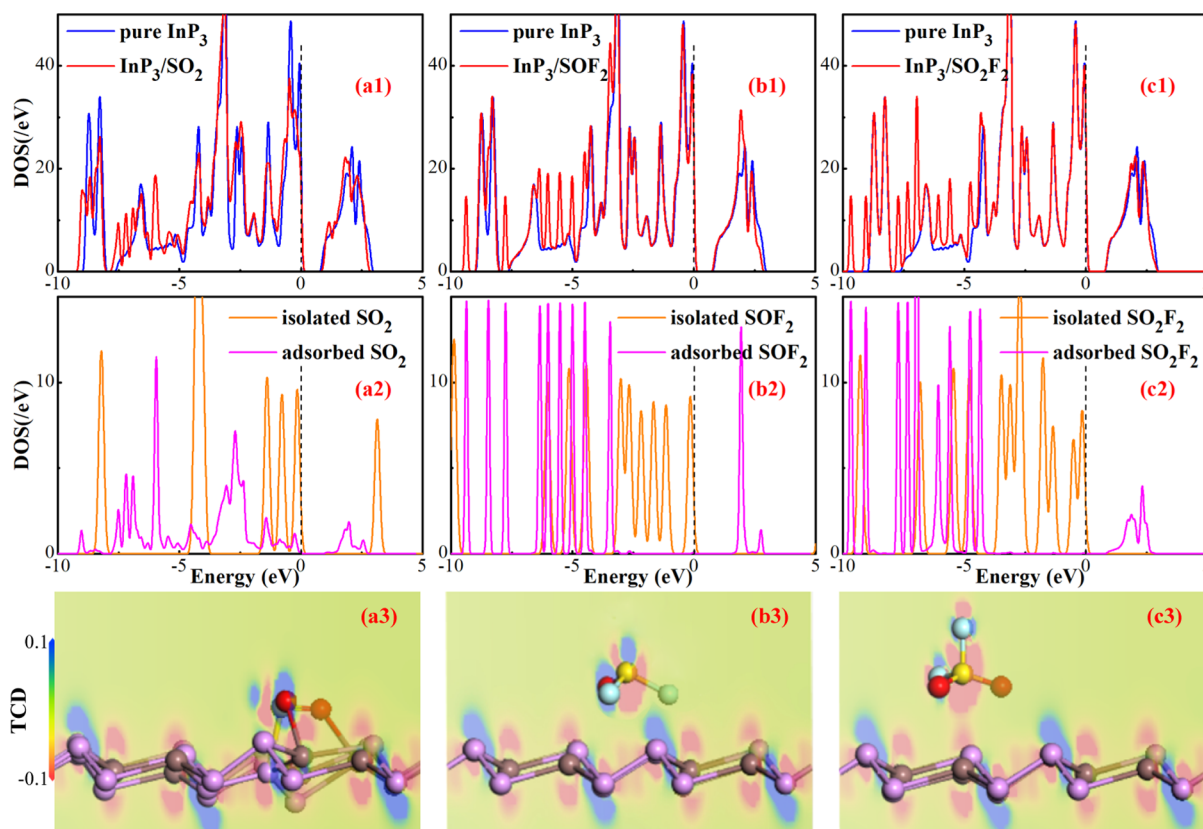
**Figure 1.** Most stable configurations and related EDD for (a)  $\text{SO}_2$ , (b)  $\text{SOF}_2$ , and (c)  $\text{SO}_2\text{F}_2$  adsorption. In EDD, the cyan (rosy) areas are electron accumulation (depletion). The isosurface is  $0.008 \text{ e}/\text{Å}^3$ .

**Table 1.** Adsorption Parameters in Three Systems

systems	$E_{\text{ad}}$ (eV)	$Q_{\text{T}}$ (e)	$D$ (Å)
$\text{SO}_2$	−1.54	−0.28	2.30
$\text{SOF}_2$	−0.77	−0.12	2.73
$\text{SO}_2\text{F}_2$	−0.31	0.05	3.51

trapped by the neighboring In atoms, whereas the  $\text{SOF}_2$  and  $\text{SO}_2\text{F}_2$  molecules tend to be trapped on the top of the P-hexatomic ring and the top of the In atom respectively, without the formation of new bonds. Then, newly formed P–S and In–O bonds are measured to be 2.36 and 2.30 Å, respectively. The shorter atomic distance between the  $\text{InP}_3$  monolayer and the  $\text{SO}_2$  molecule in comparison with that of  $\text{SOF}_2$  (2.73 Å) and  $\text{SO}_2\text{F}_2$  (3.51 Å) indicates the strong interaction in the former system. The calculated  $E_{\text{ad}}$  in the  $\text{SO}_2$  system is −1.54 eV, exactly two times −0.77 eV in the  $\text{SOF}_2$  system and about five times −0.31 eV in the  $\text{SO}_2\text{F}_2$  system. These results manifest the strong chemisorption in the  $\text{SO}_2$  system and favorable physisorption in the  $\text{SOF}_2$  system; however, a quite weak interaction is identified in  $\text{SO}_2\text{F}_2$  systems.<sup>22</sup> Such a finding, in our opinion, may attribute to the strong polar effect of the  $\text{SO}_2$  molecule that can enhance the dipole–dipole force in the gas adsorption process. According to the Hirshfeld analysis,  $\text{SO}_2$  and  $\text{SOF}_2$  molecules are electron acceptors, receiving 0.28 and 0.12 e from the  $\text{InP}_3$  monolayer, respectively, while the  $\text{SO}_2\text{F}_2$  molecule is positively charged by only 0.05 e, suggesting its weak electron-donating behavior. These parameters consistently imply that the adsorption behavior of the  $\text{InP}_3$  monolayer upon three gases is in the order  $\text{SO}_2 > \text{SOF}_2 > \text{SO}_2\text{F}_2$ . From EDD, the  $\text{SO}_2$  molecule is surrounded by electron accumulation, and a dense electron overlap is seen on the P–S and In–O bonds, which suggests the orbital hybridization and illustrates the strong binding force between bonding atoms. On the contrary, such a phenomenon is unobvious in the  $\text{SOF}_2$  and  $\text{SO}_2\text{F}_2$  systems, indicating relatively weaker performance of the  $\text{InP}_3$  monolayer upon their adsorptions.

Further analysis of the electronic behavior of the  $\text{InP}_3$  monolayer upon gas adsorption is exhibited in Figure 2, in which the density of states (DOS) and total charge density (TCD) are plotted. From the total DOS, it is seen that the states around the Fermi level are somewhat deformed in the  $\text{SO}_2$  system, especially at the top of the valence band and the bottom of the conduction band, while a little change is seen around the Fermi level in the  $\text{SOF}_2$  and  $\text{SO}_2\text{F}_2$  systems. Although remarkable deformations occur below −5.0 eV in three systems, we assume that these novel states at the deep valence band may not exert a really remarkable impact on the electronic behavior of the whole system.<sup>23</sup> In fact, the deformations of the total DOS attribute to the contribution from the activated DOS of adsorbed gas molecules. After adsorption, the DOS of  $\text{SO}_2$  splits into several small peaks standing around the Fermi level from −2.5 to 2.5 eV, which leads to the formation of the total DOS near the Fermi level and therefore causes a change in the electronic behavior largely. On the other hand, the DOS of  $\text{SOF}_2$  and  $\text{SO}_2\text{F}_2$  after adsorption is left-shifted by about 3.0 eV, exerting small contribution to the states around the Fermi level and thus leading to a slight change in the electronic property of the whole system accordingly. From the TCD, the electron accumulation could be found at  $\text{SO}_2$  and  $\text{SOF}_2$  molecules, while electron depletion at the  $\text{SO}_2\text{F}_2$  molecule is identified.



**Figure 2.** DOS and TCD of gas adsorption systems. The dash line is the Fermi level. (a1–a3) SO<sub>2</sub>; (b1–b3) SOF<sub>2</sub>; and (c1–c3) SO<sub>2</sub>F<sub>2</sub>.

These agree with the Hirshfeld analysis well. Also, the electron accumulation could be seen on the P–S and In–O bonds, confirming the strong orbital hybridization between these atoms.

The changed electronic behavior of the InP<sub>3</sub> monolayer ascribes to the adsorption of gas species, which results in the electron redistribution for the adsorbed systems,<sup>13</sup> and in this case, the band gap of the InP<sub>3</sub>/gas system will differ from that of the pure InP<sub>3</sub> surface. According to our band structure calculations, the band gap of the primitive InP<sub>3</sub> monolayer is 1.018 eV, close to 1.14 eV reported previously.<sup>16</sup> Besides, the band gap of the SO<sub>2</sub> system rises to 1.071 eV, while that of SOF<sub>2</sub> and SO<sub>2</sub>F<sub>2</sub> systems reduces to 0.983 and 1.017 eV, respectively. While the conductivity ( $\sigma$ ) of materials is related to its band gap as  $\sigma = \lambda e^{(-E_g/2kT)}$ , wherein  $\lambda$  is a constant,  $E_g$  is the band gap,  $k$  is the Boltzmann constant, and  $T$  is temperature (set as 298 K in this work), the gas response ( $S$ ) could be formulated as  $S = (\sigma_{\text{gas}}^{-1} - \sigma_{\text{pure}}^{-1})/\sigma_{\text{pure}}^{-1}$ , wherein  $\sigma_{\text{gas}}$  and  $\sigma_{\text{pure}}$ , respectively, mean the conductivity of the InP<sub>3</sub> monolayer after and before gas adsorption. Based on such two formulas, the responses for sensing SO<sub>2</sub>, SOF<sub>2</sub>, and SO<sub>2</sub>F<sub>2</sub> molecules are calculated to be 180.7, –19.4, and –1.9%, respectively. In other words, the InP<sub>3</sub> monolayer could realize SO<sub>2</sub> detection with high sensitivity and exhibit a negative response upon SOF<sub>2</sub> and SO<sub>2</sub>F<sub>2</sub> sensing with a detectable response upon SOF<sub>2</sub> and limited response upon SO<sub>2</sub>F<sub>2</sub>. We should note that in the experimental research studies, there are some other parameters that can impact the electrical conductivity of the sensing materials, such as carrier mobility and surface decoration. However, in this work, we focus on the exploration of the pristine InP<sub>3</sub> monolayer as a sensing material; thus, the surface decoration manner is not

mentioned, and if possible, some further research studies could be conducted to highlight such an important parameter. Besides, since the band gap is the most highlighted parameter to impact the electrical conductivity of the sensing materials applied in the theoretical research,<sup>24</sup> we herein use such a manner as well to evaluate the electrical conductivity of the InP<sub>3</sub> monolayer to analyze its sensing response in an easier and more effective manner.

The recovery time is another important parameter for estimating the ability for gas desorption from the sensor's surface, which is calculated by the van't Hoff–Arrhenius expression<sup>25</sup>

$$\tau = A^{-1} e^{(-E_a/K_B T)} \quad (1)$$

where  $A$  is the attempt frequency ( $10^{12} \text{ s}^{-1}$ ),<sup>26</sup>  $T$  is temperature, and  $K_B$  is the Boltzmann constant [ $8.318 \times 10^{-3} \text{ kJ}/(\text{mol K})$ ].  $E_a$  is the potential barrier for gas desorption, which is assumed to be equal to  $E_{\text{ad}}$  here.<sup>27,28</sup> Equation 1 indicates that the larger the  $E_{\text{ad}}$ , the harder it is for gas desorption, and a higher temperature could help.

Figure 3 displays the recovery time of the InP<sub>3</sub> monolayer toward three gases at various temperatures. At ambient temperature (298 K), one can see that desorption for SO<sub>2</sub> is somewhat unrealistic and for SOF<sub>2</sub> is desirable, while the quite short time for SO<sub>2</sub>F<sub>2</sub> desorption identifies its feeble interaction on the InP<sub>3</sub> surface. Via increasing the working temperature to 498 K, desorption of the stably adsorbed SO<sub>2</sub> from the InP<sub>3</sub> surface becomes feasible. Combining the sensing performance of the InP<sub>3</sub> monolayer, it could be inferred that the InP<sub>3</sub> surface is an admirable sensing material for SO<sub>2</sub> detection at room temperature, and its recycle use could be realized through heating at 498 K. Besides, the desirable sensing and

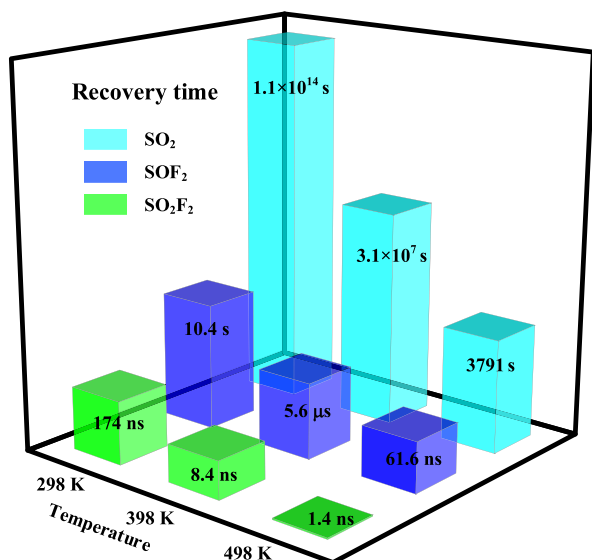


Figure 3. Recovery time of the InP<sub>3</sub> monolayer for gas desorption.

desorbing performances toward SOF<sub>2</sub> allow the application of the InP<sub>3</sub> monolayer as a reusable sensor under normal conditions. Moreover, such superior behavior of the InP<sub>3</sub> monolayer upon SO<sub>2</sub> and SOF<sub>2</sub> adsorption makes it applicable as a valid gas adsorbent to scavenge these noxious gases from SF<sub>6</sub> insulation devices, guaranteeing their safe operation effectively. However, the InP<sub>3</sub> monolayer is not suitable for SO<sub>2</sub>F<sub>2</sub> sensing or removal due to the weak interaction in adsorption.

Figure 4 displays the dielectric functions and work function (WF) of the InP<sub>3</sub> monolayer upon gas adsorption. From Figure 4a, we can see two major adsorption peaks (277 and 356 nm) for the pure InP<sub>3</sub> surface, which within the range of ultraviolet is promising for gas detection due to its high energy.<sup>29</sup> After SO<sub>2</sub> adsorption, the peaks at 356 nm are remarkably weakened, and the peak at 277 nm suffers from a red shift to 285 nm. These findings suggest the sensitive detection of SO<sub>2</sub> using an InP<sub>3</sub> monolayer optical sensor. On the contrary, the slight change in adsorption intensity in SOF<sub>2</sub> and SO<sub>2</sub>F<sub>2</sub> systems reveals the unfeasibility for their detection. From Figure 4b, one can see that the WF of the InP<sub>3</sub> monolayer suffers from a great increase after SO<sub>2</sub> adsorption and a more remarkable

decrease after SOF<sub>2</sub> adsorption, while an unobvious decrease is determined in the SO<sub>2</sub>F<sub>2</sub> system. The WF reveals the minimum energy required to release an electron from the surface, which impacts the performance of certain nanomaterials to be explored as a field-effect transistor gas sensor.<sup>30</sup> Based on these results, one can identify the potential of the InP<sub>3</sub> monolayer as a field-effect transistor sensor for sensitive and selective detection of SO<sub>2</sub> and SOF<sub>2</sub>, given their obviously opposite changes of WF on the InP<sub>3</sub> surface.<sup>31</sup> However, both methods are not suitable for SO<sub>2</sub>F<sub>2</sub> detection due to the weak response of the InP<sub>3</sub> monolayer upon gas adsorption.

In addition, the variations of  $Q_T$  and  $E_g$  in the SO<sub>2</sub> and SOF<sub>2</sub> systems tuned by the applied electric field are studied to further analyze the modulated sensitivity for their detection.<sup>32</sup> Such an electric field is vertical to the InP<sub>3</sub> surface pointing to the two gas species, with results shown in Figure 5. It is found that as the strength of the electric field increases from 0.2 to 1.0 V/Å, the  $Q_T$  is decreased in the SO<sub>2</sub> system but is increased in the SOF<sub>2</sub> system with the opposite charge-transferring path. In other words, the applied electric field can weaken the charge transfer from the InP<sub>3</sub> monolayer to the SO<sub>2</sub> molecule, while it accelerates the charge transfer from the SOF<sub>2</sub> molecule to the InP<sub>3</sub> surface. Such results are in good accordance with the WF analysis, wherein the opposite change of WF accounts for the different path of charge transfer when the extra electric field is applied. Associating with the change of  $Q_T$ , the  $E_g$  decreases sharply in the SOF<sub>2</sub> system, while it is slightly decreased in the SO<sub>2</sub> system. These findings manifest the desirable gas-sensing response of the InP<sub>3</sub> monolayer for SO<sub>2</sub> and SOF<sub>2</sub> detection as a field-effect transistor gas sensor,<sup>33</sup> especially for SOF<sub>2</sub> when the applied electric field is higher than 0.6 V/Å.

Based on the abovementioned analysis, further experimental research studies should be conducted to further verify our results, which would be the following work of our group. It is worth noting that to prepare the InP<sub>3</sub>-based gas sensor, the interaction between the InP<sub>3</sub> nanomaterial and the substrate is very significant.<sup>34</sup> The main impact parameters include the texture of the substrate, deposition temperature, and exposure time and humidity in the air,<sup>35</sup> which are significant to ensure that the sensing materials can be doped on the substrates with the best-sensing performance using a proper thermal treatment technique. We are hopeful that such a highlight is beneficial to

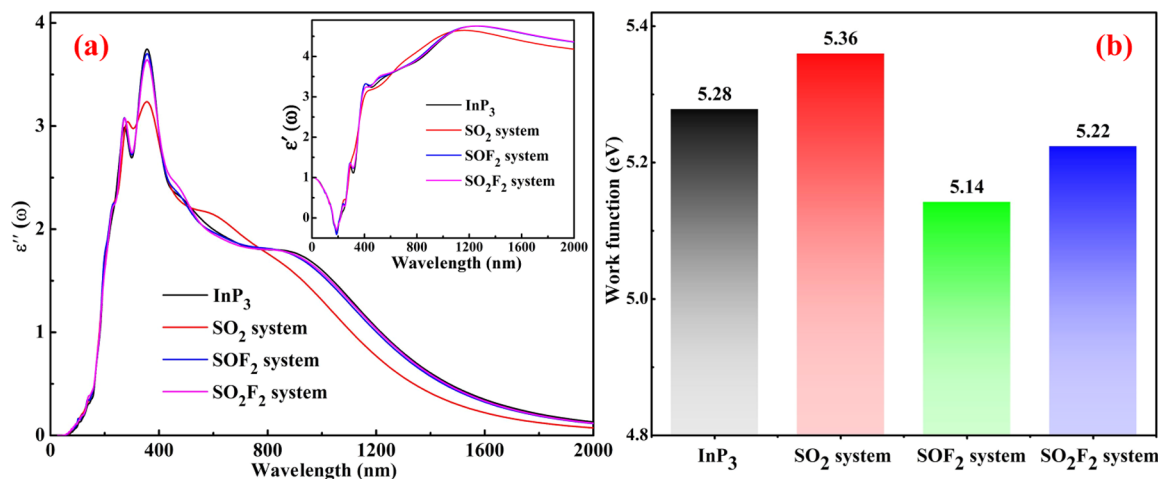
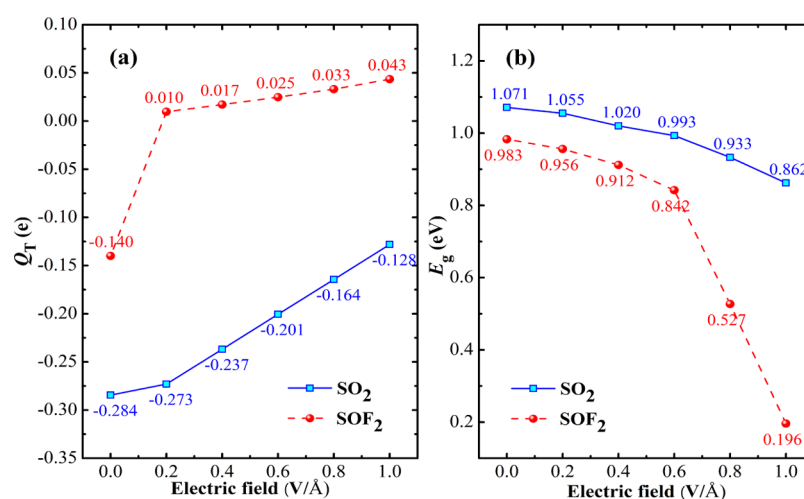


Figure 4. Dielectric functions (a) and WF (b) of the InP<sub>3</sub> monolayer upon gas adsorption.



**Figure 5.** Variations of  $Q_T$  (a) and  $E_g$  (b) with the applied electric field.

help the experimental researchers to prepare an InP<sub>3</sub>-based gas sensor with high gas sensitivity.

### 3. CONCLUSIONS

In this work, we provide a first insight into the InP<sub>3</sub> monolayer as a resistance-type, optical or field-effect transistor gas sensor upon three SF<sub>6</sub> decomposed species, namely, SO<sub>2</sub>, SOF<sub>2</sub>, and SO<sub>2</sub>F<sub>2</sub>, in order to evaluate the operation status of SF<sub>6</sub> insulation devices. The adsorption performances of the InP<sub>3</sub> monolayer upon three gases are in the order SO<sub>2</sub> > SOF<sub>2</sub> > SO<sub>2</sub>F<sub>2</sub>, in which strong chemisorption in the SO<sub>2</sub> system and weak physisorption in the SOF<sub>2</sub> system are identified. From the aspect of a resistance-type sensor, the InP<sub>3</sub> monolayer could realize sensitive detection of SO<sub>2</sub> with recycle use with the heating technique at 498 K. For SOF<sub>2</sub> sensing, the detectable response and desirable desorption property allow the working conditions of the InP<sub>3</sub> monolayer at room temperature. Moreover, the remarkable change in ultraviolet peaks of the InP<sub>3</sub> monolayer after adsorption of SO<sub>2</sub> indicates its optical sensing application, and the evident changes of WF of the InP<sub>3</sub> monolayer in two opposite trends illustrate the sensitive and selective detection between SO<sub>2</sub> and SOF<sub>2</sub>. However, the InP<sub>3</sub> monolayer is not suitable for SO<sub>2</sub>F<sub>2</sub> sensing due to the weak response in adsorption.

### 4. COMPUTATIONAL DETAILS

Spin-polarized calculations were implemented in this paper within the DMol<sup>3</sup> package.<sup>36</sup> The Perdew–Burke–Ernzerhof function within the generalized gradient approximation was employed to deal with the electron exchange–correlation terms.<sup>37</sup> The DFT-D2 method developed by Grimme was employed<sup>38</sup> to better understand the van der Waals force and long-range interactions. Double numerical plus polarization was selected as the atomic orbital basis set.<sup>39</sup> The Monkhorst–Pack  $k$ -point mesh of  $12 \times 12 \times 1$  was sampled for supercell calculations of geometric optimization and electronic properties.<sup>40</sup> The energy tolerance accuracy, maximum force, and displacement were set as  $10^{-5}$  Ha,  $2 \times 10^{-3}$  Ha/Å, and  $5 \times 10^{-3}$  Å, respectively.<sup>41</sup> For static electronic calculations, a self-consistent loop energy of  $10^{-6}$  Ha, global orbital cutoff radius of 5.0 Å, and smearing of 0.005 Ha were used to ensure the accuracy of total energy.<sup>42</sup>

We established a  $2 \times 2 \times 1$  primitive InP<sub>3</sub> monolayer supercell with a vacuum region of 15 Å to prevent possible interactions between adjacent units.<sup>43</sup> The lattice constant of the fully optimized InP<sub>3</sub> monolayer was obtained as 7.535 Å, equal to the previous theoretical work,<sup>44</sup> and the frequency ranges from 80.48 to 962.55 cm<sup>-1</sup>, indicating the good stability of the InP<sub>3</sub> structure. The adsorption energy ( $E_{ad}$ ) is calculated by

$$E_{ad} = E_{\text{InP}_3/\text{gas}} - E_{\text{InP}_3} - E_{\text{gas}} \quad (2)$$

where  $E_{\text{InP}_3/\text{gas}}$ ,  $E_{\text{InP}_3}$ , and  $E_{\text{gas}}$  represent the total energies of the gas-adsorbed system, pure InP<sub>3</sub> monolayer, and isolated gas molecule, respectively. Moreover, the Hirshfeld method was considered to analyze the charge transfer ( $Q_T$ ) in the gas adsorption systems, wherein the positive values indicated the electron-losing property of the adsorbed gas species.

### ■ AUTHOR INFORMATION

#### Corresponding Author

**Xin Qin** – Academics Working Station, Changsha Medical University, Changsha 410219, China; Hunan Key Laboratory of the Research and Development of Novel Pharmaceutical Preparations, Changsha Medical University, Changsha 410219, China; [orcid.org/0000-0001-6275-4738](https://orcid.org/0000-0001-6275-4738); Email: [qinxin@cqu.edu.cn](mailto:qinxin@cqu.edu.cn)

#### Authors

**Chenchen Luo** – Maintenance Branch of State Grid Zhejiang Electric Power Limited Liability Company, Hangzhou 311232, China

**Yaqian li** – Academics Working Station, Changsha Medical University, Changsha 410219, China

**Hao Cui** – State Key Laboratory of Power Transmission Equipment & System Security and New Technology, Chongqing University, Chongqing 400044, China; College of Artificial Intelligence, Southwest University, Chongqing 400715, China; [orcid.org/0000-0002-9410-6345](https://orcid.org/0000-0002-9410-6345)

Complete contact information is available at: <https://pubs.acs.org/10.1021/acsoomega.1c04185>

#### Notes

The authors declare no competing financial interest.

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