

# Molecular Insights into the CO<sub>2</sub> Mineralization Process with Tricalcium Silicate

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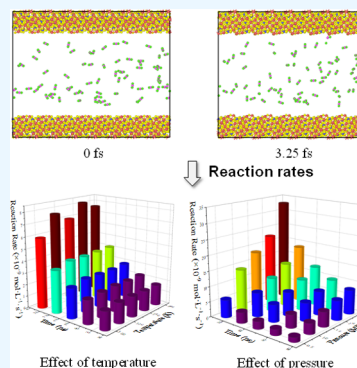
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**ABSTRACT:** CO<sub>2</sub> mineralization, a pivotal technology in CO<sub>2</sub> capture, storage, and utilization, promises to convert industrial waste into valuable industrial products. However, the intricate reaction mechanism and rate-limiting process remain inadequately elucidated. The suitable reaction conditions should be clarified when designing industrial produce. In this study, we comprehensively examined the reaction kinetics and conversion rates of C<sub>3</sub>S under varying conditions and degrees of dispersion by reactive molecular dynamics simulation. Furthermore, we set up a sophisticated model depicting C<sub>3</sub>S encased within a water film, mirroring its prevalent configuration in moist environments. The results show that the reaction is fast when the temperature is 328 K. The conversion rate tends to decrease under elevated pressure when it is higher than 1.0 MPa, while the temperature has a minimal impact. Fragmentation of solid waste can increase the degree of dispersion, disrupting the crystal structure and expanding the reaction surface area, thereby accelerating the reaction. The presence of a water film impedes mass transfer, consequently reducing the reaction speed. The present study sheds light on the reaction mechanism of the CO<sub>2</sub> mineralization process.



## 1. INTRODUCTION

Large amounts of carbon dioxide emissions exacerbate the greenhouse effect.<sup>1</sup> Thereby, many issues come into being, such as climate change, iceberg melting,<sup>2</sup> sea level rise, and so on. Carbon capture, storage, and utilization are necessary approaches to removing CO<sub>2</sub> on a large scale.<sup>3</sup> CO<sub>2</sub> mineralization technology refers to the reaction of CO<sub>2</sub> and natural mineral/industrial waste. 4.02 Gt CO<sub>2</sub> can be captured by industrial alkaline waste.<sup>4</sup> Micrometer CaCO<sub>3</sub> with high-value attachment can be produced during this process.<sup>5,6</sup> CO<sub>2</sub> mineralization technology is of great sequential potential,<sup>7</sup> and carrying out new CO<sub>2</sub> mineralization technologies can contribute to turning waste into industrial resources and achieving the goal of carbon neutrality.

Plenty of experimental efforts have been made to study CO<sub>2</sub> mineralization technologies. For example, Reddy et al.<sup>8</sup> conducted experiments on the mineralization of fly ash, a byproduct of coal combustion, using a fluid bed reaction container. The results revealed that unseparated CO<sub>2</sub> and SO<sub>2</sub> could be mineralized by fly ash, with the reaction completed within just a few minutes. Helwani et al.<sup>9</sup> discovered that in situ mineralization in a power plant was more effective than geologic sequestration for carbon capture and storage. Polettini et al.<sup>10</sup> researched the impact of temperature, pressure, and CO<sub>2</sub> concentration on conversion rate and found the reaction condition where the conversion of CO<sub>2</sub> was the highest. However, the previous works did not understand the mineralization reaction on a mechanism level. Some researchers found that CaCO<sub>3</sub> could be produced by steel

slag carbonation.<sup>11,12</sup> Besides, ammonium salt contributed to the diffusion of Ca<sup>2+</sup>, and the liquid-to-solid ratio influenced the diffusion rate greatly.<sup>12</sup> However, this experiment did not reveal mineralization reaction mechanisms. C<sub>3</sub>S is the active phase in industrial waste. The crystal structure of C<sub>3</sub>S is a tetrahedron whose center is a silicon atom coordinated by oxygen atoms.<sup>13</sup> The silicon-oxygen tetrahedrons are connected by covertex, coedge, and coface, whose structure is unstable. In the C<sub>3</sub>S crystal, the coordination number of Ca–O is large and the force between atoms is weak. These features make it easy to be ruined by CO<sub>2</sub> molecules. Thus, a mineralization reaction takes place. Reaction dynamics information can be obtained from the above experiments. Nevertheless, the experimental method can only understand the mineralization process from the macro level, cannot reveal the mechanism of mineralization reaction from the microscale, and is only confined to the limited operating conditions, which hinders the further development of mineralization technologies.

Numerical simulation provides a repeatable and cost-effective alternative to the experimental method, with the

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rapid development of numerical algorithms and computer hardware. Molecular dynamics (MD) is gradually emerging as a mainstream high-fidelity method for investigating mineralization processes, attracting widespread attention from researchers. For example, Zhao et al.<sup>14</sup> researched the carbonation process on the surface of  $\gamma$ -C<sub>2</sub>S and found that chemical adsorption was far stronger than physical adsorption. HCO<sub>3</sub><sup>-</sup> deprotonation took place, and a more stable structure formed. However, this process needed to jump over the energy barrier. Mutyasa and Kalinichev<sup>15</sup> pointed out that the mutual influence of cement and water contributed to water film formation on the solid surface and hindered CO<sub>2</sub> diffusion to the solid surface. The reaction rate depended on the structure of the cement holes and the water layer. However, constrained by the numerous computing resource demands of ab initio MD simulation, this research did not detect the solution process of the calcium salt. This drawback could be overcome by reactive molecular dynamics (RMD) simulation. Zare et al.<sup>16</sup> pointed out that the mineral surface was covered by a nanometer-scale width water layer during the mineralization process under moisture conditions. The reverse proton transfer phenomenon coincided with the results detected by in situ spectroscopy. Qin et al.<sup>17</sup> studied the calcium carbonation nucleation process during the mineralization process by reactive and classic MD simulation and found that the results of the dynamic properties computed by RMD simulation were consistent with experiment results than classic MD. Wang et al.<sup>18</sup> examined the role of H<sub>2</sub>O during the carbonation process and pointed out that water-induced adsorption of CO<sub>2</sub> in the diffusion control stage in low temperatures. The diffusion of ions and CO<sub>2</sub> molecules became more tense, and the thickness of the product layer increased with water. Dai et al.<sup>19</sup> investigated the mineralization reaction between CO<sub>2</sub> and illite and discovered the protonation process of nonbridging oxygen atoms. Cations could reach the liquid area through cube and tetrahedron space and combine with HCO<sub>3</sub><sup>-</sup> to form carbonation salt. They hold the point of view that RMD simulation as a tool to investigate mineralization mechanisms was dependable. The above simulation results can portray and reproduce the mineralization process to a certain extent, but they do not consider the effect of outer conditions and dispersion degree on the rate, as well as the effect of water film, and there are rare simulation studies on C<sub>3</sub>S, which hinders the revelation of the mineralization mechanism of C<sub>3</sub>S and the development of mineralization technology.

The composition of various industrial wastes is complex<sup>20</sup> and their impact on the mineralization mechanism is not valid nowadays. The rate of the reaction of CO<sub>2</sub> mineralization is influenced by many factors. Hence, to adjust the mineralization reaction overall, revealing the microscopic mineralization mechanism needs to be done. Due to this reason, the C<sub>3</sub>S composite in industrial waste is taken as a mineralization resource to reveal the mineralization mechanism by RMD simulation in this work. The article is organized as follows: Section 2 provides a pithy description of RMD and force field validation results. Section 3 presents the simulation steps and parameters. The impact of outer conditions, dispersion degree, and water film on mineralization reaction dynamics and mechanism are expounded in Section 4. Conclusions are drawn in Section 5.

## 2. MATHEMATICAL MODEL

**2.1. RMD Model.** MD denotes the movement of molecules and atoms by Newton's Second Law. However, it is unable to model chemical reactions<sup>21</sup> and lacks the required accuracy.<sup>22</sup> On the other hand, quantum chemistry computation is rooted in the Schrödinger equation, which offers high accuracy but demands substantial computational resources. In contrast, RMD can represent the process of bond breaking and forming in the form of bond order.<sup>23</sup> It balances and transfers charge punctuation by the electron negativity. The equation is given by:

$$E_{\text{system}} = E_{\text{bond}} + E_{\text{over}} + E_{\text{angle}} + E_{\text{tors}} + E_{\text{vdWaals}} + E_{\text{Coulomb}} + E_{\text{Specific}} \quad (1)$$

where  $E_{\text{system}}$  is the total energy.  $E_{\text{bond}}$  is the bond forming energy between atoms as a continuous function of the distance between atoms.  $E_{\text{angle}}$  and  $E_{\text{tors}}$  are three-atom and four-atom dihedral energy, respectively.  $E_{\text{over}}$  is the revision term based on the atomic valence rule.  $E_{\text{Coulomb}}$  and  $E_{\text{vdWaals}}$  are the static charge energy and dispersion energy, respectively.  $E_{\text{Special}}$  is the term added to the definition for special systems.

The bond order of ReaxFF is calculated from an empirical formula.<sup>24</sup>

$$\begin{aligned} \text{BO}_{ij} &= \text{BO}_{ij}^{\sigma} + \text{BO}_{ij}^{\pi} + \text{BO}_{ij}^{n\pi} \\ &= \exp\left[p_{\text{bo1}}\left(\frac{r_{ij}}{r_o^{\sigma}}\right)^{p_{\text{bo2}}}\right] + \exp\left[p_{\text{bo3}}\left(\frac{r_{ij}}{r_o^{\pi}}\right)^{p_{\text{bot}}}\right] \\ &\quad + \exp\left[p_{\text{bo5}}\left(\frac{r_{ij}}{r_o^{n\pi}}\right)^{p_{\text{bob}}}\right] \end{aligned} \quad (2)$$

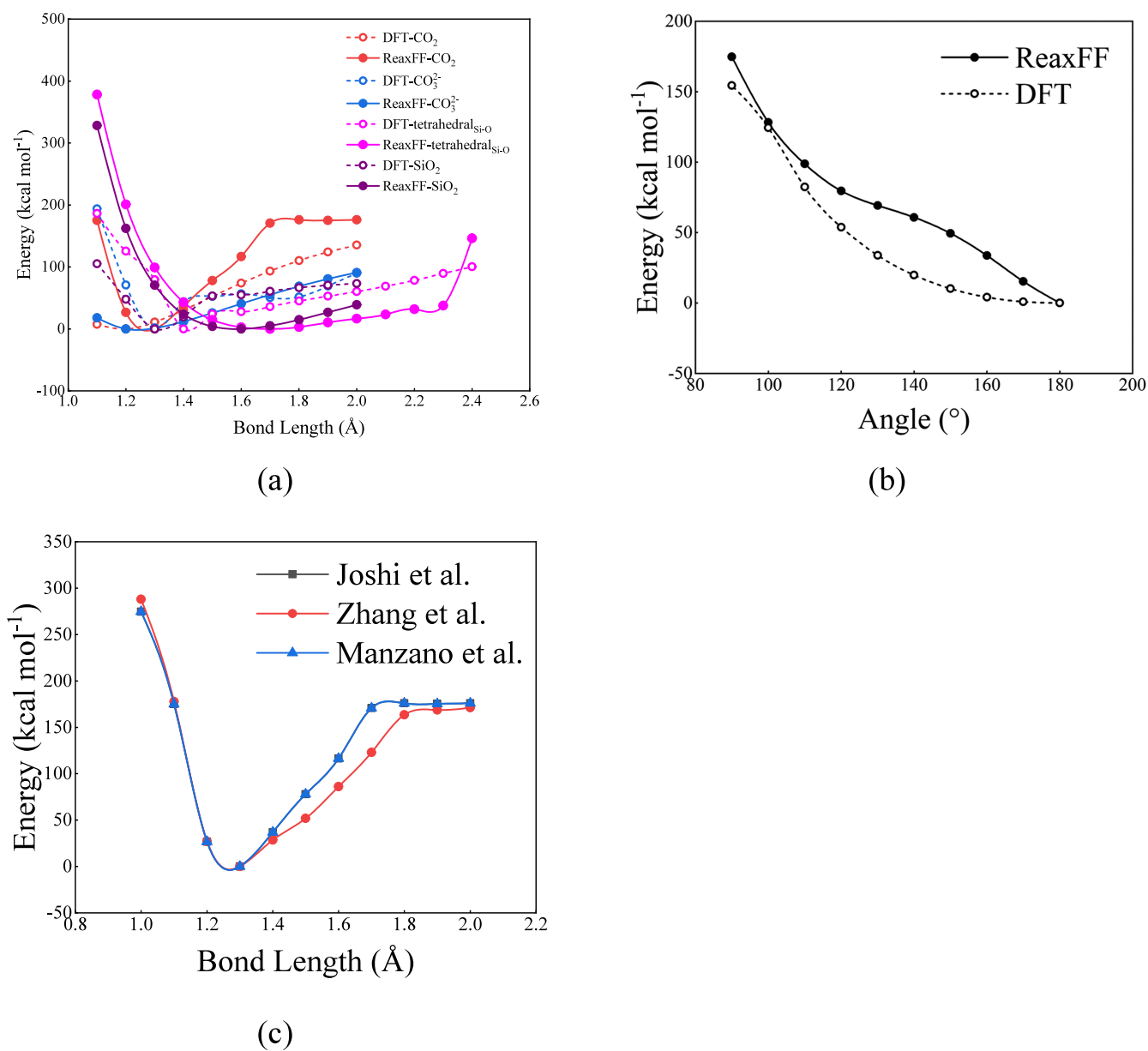
where  $\text{BO}_{ij}$  is the bond order between atoms  $i$  and  $j$ ,  $r_{ij}$  is the distance between atoms  $i$  and  $j$ ,  $r_o$  terms are the equilibrium bond lengths, and  $p_{\text{bo}}$  terms are empirical parameters.

The initial configuration was constructed from randomly placed molecules. In fact, the smaller the energy, the more stable the structure is. Therefore, the energy minimization process was conducted to make the model coordinate with the real situation. The equation of energy minimization is given by:

$$\begin{aligned} E(r_1, r_2, \dots, r_N) &= \sum_{ij} E_{\text{pair}}(r_i, r_j) + \sum_{ij} E_{\text{bond}}(r_i, r_j) \\ &\quad + \sum_{ijk} E_{\text{angle}}(r_i, r_j, r_k) + \\ &\quad \sum_{ijkl} E_{\text{dihedral}}(r_i, r_j, r_k, r_l) \\ &\quad + \sum_{ijkl} E_{\text{improper}}(r_i, r_j, r_k, r_l) + \sum_i E_{\text{fix}}(r_i) \end{aligned} \quad (3)$$

where  $r_1, r_2, \dots, r_N$  are position vectors of atoms from 1 to  $N$ .  $E_{\text{pair}}$  is nonbonded pairwise bond interactions.  $E_{\text{angle}}$  is the bond angle interactions.  $E_{\text{dihedral}}$  is a dihedral interactions.  $E_{\text{improper}}$  is the improper dihedral interactions.  $E_{\text{fix}}$  is the extra energy term due to extra force to atoms. The left-hand side of the equation is the total energy of the system.

**2.2. Model Validation.** The RMD model must be validated before simulation. The force field parameters utilized were sourced from Joshi et al.<sup>25</sup> This study primarily focuses on validating bond length energy and angle energy, particularly

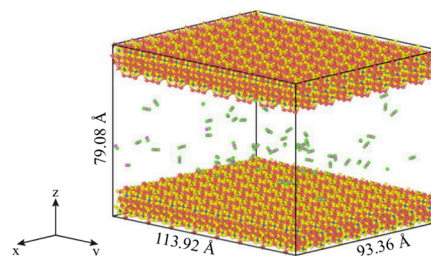


**Figure 1.** Model validation of force field: (a) bond length, (b) angle, and (c) different force fields. Dashed lines represent DFT results while solid lines represent ReaxFF results.

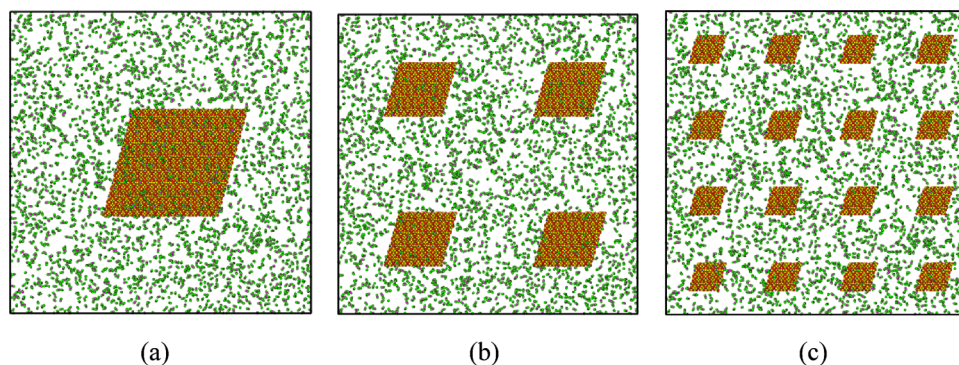
about CO<sub>2</sub> and carbonates. As depicted in Figure 1a, the DFT and ReaxFF results exhibit corresponding trends, and the location of the curve's extremal point is consistent between the two methods. It is worth noting that the bond energy calculated by ReaxFF deviates from that calculated by DFT when the bond length is less than the equilibrium length. However, this discrepancy is inconsequential, as it has a minimal impact on the simulation results. Figure 1b illustrates the O=C=O angle as a function of the angle, comparing the results between DFT and ReaxFF. The values at the same angle are very close to each other, and both methods yield minimum values at 180°. Figure 1c presents the results of the bond energy from different force fields.<sup>25–27</sup> These three types of force fields were applied in separate systems. The simulation results indicate that CO<sub>2</sub> bond energy is consistent among the three sources, suggesting that different force fields have a minimal impact on the simulation.

### 3. SIMULATION SETUPS

Figure 2 presents the simulation setup. The crystal cell is from the crystal database.<sup>28</sup> The crystal was built by repeating 8, 8,



**Figure 2.** Model setup, where the blue sphere represents the silicon atom, the yellow sphere represents oxygen atoms of C<sub>3</sub>S, the red sphere represents calcium atoms, the green sphere represents oxygen atoms of CO<sub>2</sub>, and the magenta represents carbon atoms.

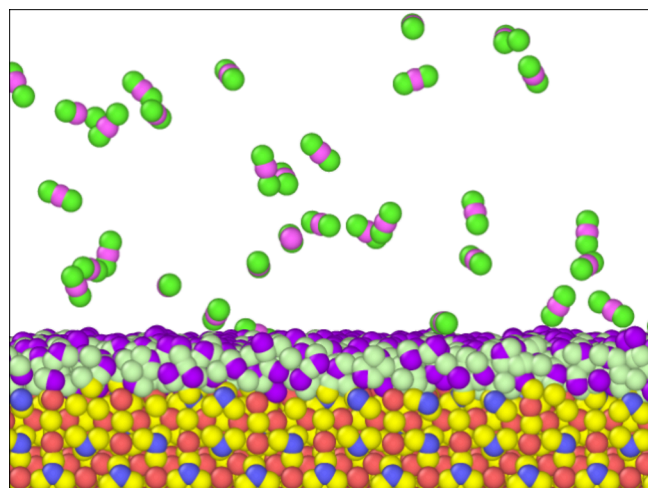


**Figure 3.** Model of different types of dispersion degree: (a) Compaction, (b) accumulation, and (c) spread.

and 2 times along the  $x$ ,  $y$ , and  $z$  directions, respectively. The length along the  $z$  direction is 26.36 Å. The (001) surface<sup>29</sup> and (00 $\bar{1}$ ) surface reacted with CO<sub>2</sub>. The number of CO<sub>2</sub> molecules was calculated by volume and density as follows:

$$N = \frac{\rho V \times 10^3}{M} \times N_A \quad (4)$$

The models of different types of C<sub>3</sub>S dispersion degrees are shown in Figure 3. The length of each periodic cubic box is 300 Å, and the number of CO<sub>2</sub> molecules is 3162. The mass of solids is equal to each other. The dispersion degree is represented as three kinds, spread, compaction, and accumulation. Dispersion degree decreases successively. The model of the C<sub>3</sub>S surface covered by a water film is shown in Figure 4.



**Figure 4.** Model that the C<sub>3</sub>S surface covered by the water film. Light green spheres represent hydrogen atoms and purple spheres represent oxygen of water.

Table 1 shows the simulation parameters. When the temperature is 298 K and the pressure is 0.5 MPa, the number of CO<sub>2</sub> molecules is 70. Energy minimization was conducted before formal simulation under a microcanonical ensemble (NVE). The size of the box was set to 93.36 Å × 113.92 Å × 79.08 Å. To research the impact of different parameters on reaction dynamics, the pressure range was set to 0.5–2.5 MPa and the temperature range to 298–358 K.

**Table 1. Simulation Settings**

simulation parameters	values
periodic conditions	$x, y, z$
box length along $x, y, z$	93.36 Å, 113.92 Å, 79.08 Å
temperature	298 K, 313 K, 328 K, 343 K, 358 K
pressure	0.5 MPa, 1.0 MPa, 1.5 MPa, 2.0 MPa, 2.5 MPa
ensemble	microcanonical ensemble (NVE)
temperature coupling algorithm	Berendsen
time step	0.25 fs
neighbor cut off for bond calculation	4.5 Å
H-bond cut off	6 Å

## 4. RESULTS AND DISCUSSION

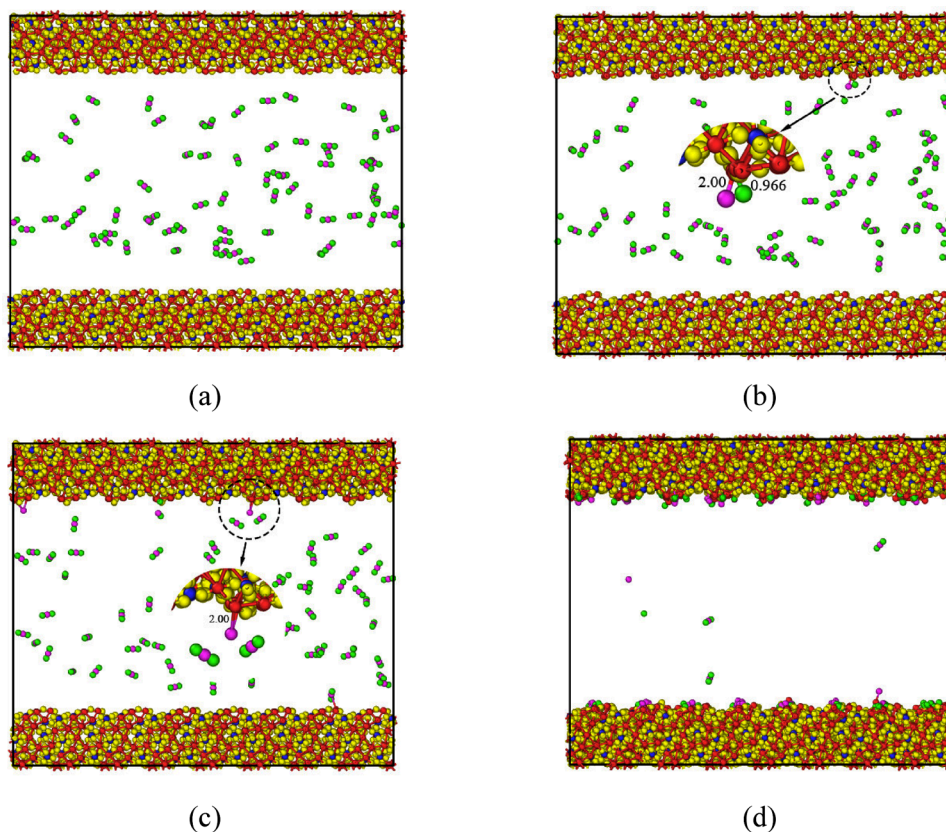
**4.1. CO<sub>2</sub> Mineralization Process.** Figure 5 shows the mineralization process. The snapshots show the reaction of carbon dioxide molecules with C<sub>3</sub>S. After the collision of activated CO<sub>2</sub> molecules with C<sub>3</sub>S, the C=O bonds break up. C=O free radicals or O atoms bind to the tricalcium silicate crystal and form CaCO<sub>3</sub> and SiO<sub>2</sub>. Meanwhile, the carbon atoms bind first in a few cases. Then, the left free radicals and atoms are adsorbed by C<sub>3</sub>S. The ab initio MD of cleaved quartz carbonation revealed that CO<sub>2</sub> reacted with under-coordinated Si and nonbridging O atoms, thus forming CO<sub>3</sub>, and was bonded to a solid surface by Si–O bonds.<sup>30</sup> When the time is 0.325 ps, the bond order between C and Ca is 2 and that between O and Ca is 0.966. When time is 0.775 ps, the bond order between C and Ca is 2. The bond order evolution of C=O is shown in Figure 6. Before the reaction, the bond order is near 2. When the time is 0.325 ps, the C=O breaks and the bond order is 0.

**4.2. Effect of Temperature on Mineralization.** Mean square displacements (MSD) have been widely employed to identify Brownian motion and random walk, which is a function of simulation time. The MSD is defined by<sup>18</sup>:

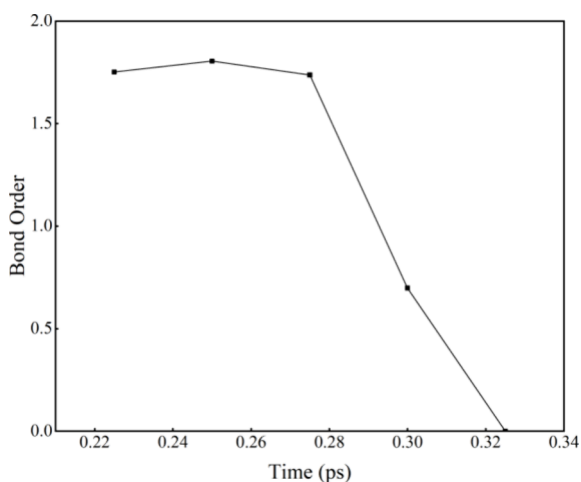
$$\text{MSD} = \langle |r(t) - r(0)|^2 \rangle \quad (5)$$

where  $r(t)$  represents the position coordination vector at time  $t$ ;  $r(0)$  represents the position coordination vector at initial time.

Figure 7 shows the effect of the temperature on the mineralization reaction rate. Within the time range of 0–9.7 ps, the MSD of 343 K is larger than that of all other temperatures. After 9.7 ps, the MSD of 358 K is bigger than 343 K. The MSD of 313 K is the lowest. Figure 7b shows the CO<sub>2</sub> molecules as a function of time at different temperatures.



**Figure 5.** Mineralization process: Snapshots of (a) 0, (b) 0.325, (c) 0.775, and (d) 58.6 ps. The temperature is 313 K.



**Figure 6.** Bond order evolution of the C=O bond in the CO<sub>2</sub> molecule which is shown in Figure 5b.

It can be concluded from Figure 7b that the reaction of 298 K is slow, while that of other values is similar. CO<sub>2</sub> would adsorb on the C<sub>3</sub>S surface and react with it during mineralization. High temperatures contributed to CO<sub>2</sub> getting away from the C<sub>3</sub>S surface as a reverse process of adsorption. When the temperature was 313 K, the MSD of CO<sub>2</sub> reached the lowest because numerous CO<sub>2</sub> molecules were adsorbed on a solid surface, which constrained their free movement. It can be found from Figure 7b that the reaction is fast when the temperature is 328 K in 18–29 ps. In the late stage of the reaction, the MSD of 358 K is far higher than that of other temperatures because getting rid of adsorption becomes the

main factor at this stage. It can be found from Figure 7b that the remaining CO<sub>2</sub> molecule number is just lower than that of 298 K within 14.8–31.9 ps. The MSD values are close to each other when the temperature is 298, 328, and 343 K.

The relationship between MSD and diffusion coefficient can be obtained by Einstein equation.<sup>18</sup>

$$D = \frac{\lim_{t \rightarrow \infty} \langle |r_i(t) - r_i(0)|^2 \rangle}{6t} \quad (6)$$

When the slope of the MSD curve is large, the diffusion coefficient is large. At each temperature, the slope of the MSD curves decreases as the reaction continues, which means diffusion coefficients are decreasing. Wang et al.<sup>18</sup> found that high temperatures contribute to a large slope. In this perspective, the situation is more complex. When the temperature is 358 K, the decreasing scale of the diffusion coefficient is big as time goes by. This is because many CO<sub>2</sub> molecules had been reacted in the late reaction stage and only a small quantity of them was adsorbed on the surface.

In the early stage of the reaction, the higher the temperature, the faster the mineralization reaction rate, which can also be concluded from Figure 8. However, the CO<sub>2</sub> molecules remain stable after 45 ps. The remaining numbers of CO<sub>2</sub> molecules at different temperatures are very close. Therefore, it contributes little to the mineralization conversion rate by raising the temperature. Nevertheless, temperature affects product form.<sup>31</sup>

**4.3. Effect of Pressure on Mineralization.** In the early stage of the reaction (0–29.0 ps), MSD decreases as pressure increases, except for that the values of 2.0 and 2.5 MPa are close. Because when pressure increases, in constant volume molecule number increases, the mean free range decreases. In

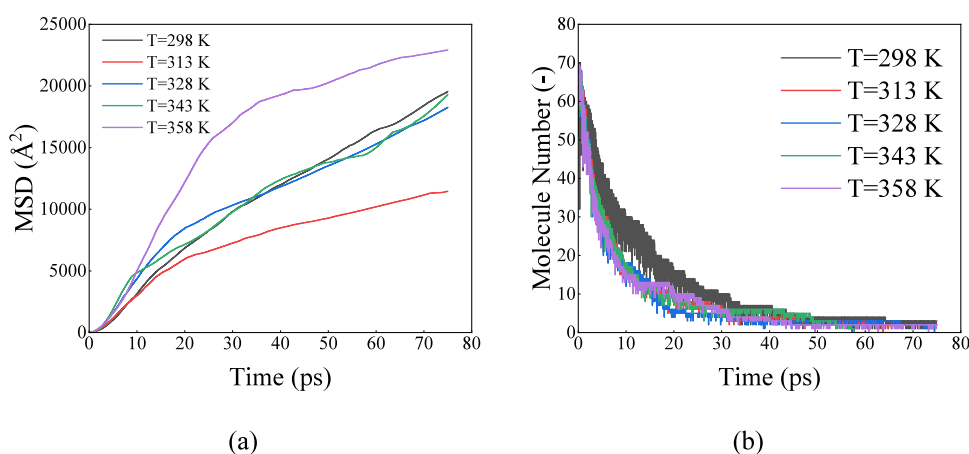


Figure 7. Effect of temperature on mineralization process: (a) CO<sub>2</sub> mean square displacement; (b) CO<sub>2</sub> molecule number.

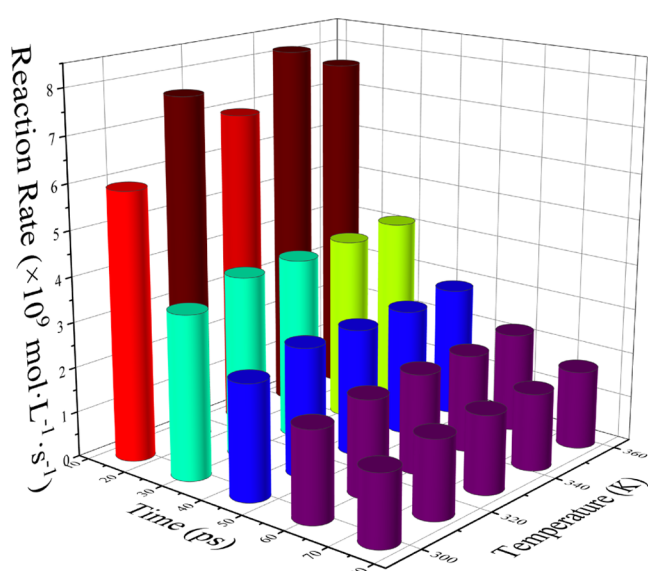


Figure 8. Average reaction rate as a function of the temperature at several time points.

the late stage of the reaction (29.0–71.9 ps), the MSD of 1.0 MPa is greater than that of 0.5 MPa. When pressure is 1.0 MPa, active CO<sub>2</sub> molecules can react with an active solid surface, and the left CO<sub>2</sub> molecules are not adsorbed.

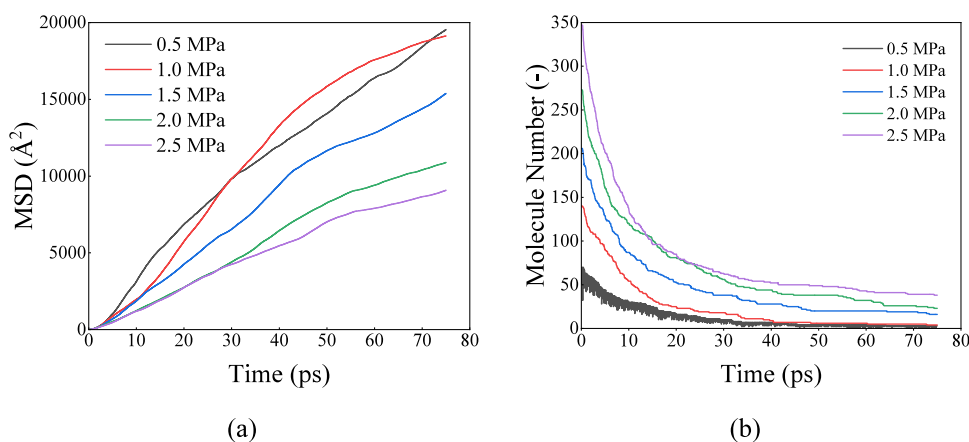


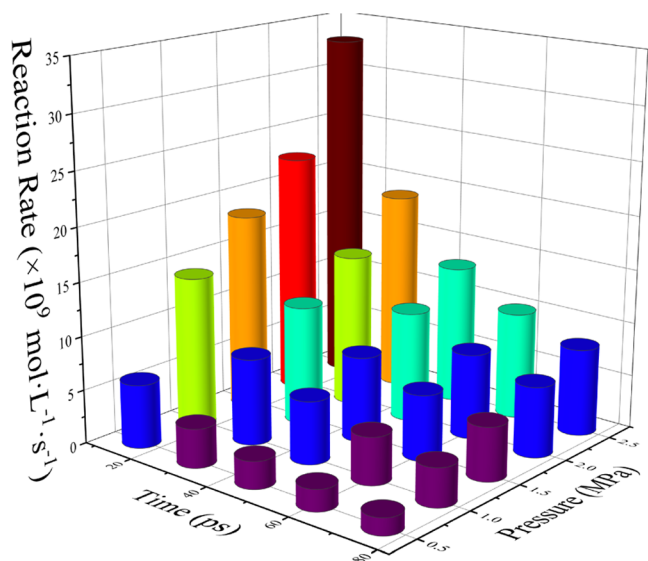
Figure 9. Pressure impact on mineralization process: (a) CO<sub>2</sub> MSD and (b) CO<sub>2</sub> molecule number as a function of simulation time.

Therefore, the MSD is big. Figure 9b shows the CO<sub>2</sub> molecule number as a function of the simulation time. It can be found in Figure 9b that the conversion rate reaches the maximum when the pressure is 1.0 MPa, which is 96.6%.

The reaction rate increases as the pressure becomes higher, as seen in Figure 10. After 60 ps, the CO<sub>2</sub> molecule number reaches a stable value. However, when pressure is higher than 1 MPa, the mineralization conversion rate decreases. When pressure is 1.5, 2.0, and 2.5 MPa, the conversion rate decreases one by one.

**4.4. Effect of Dispersion Degree on Mineralization.** Ashraf and Olek<sup>32</sup> pointed out that dispersing solid particles could contribute to gas reactions in the gap. The impact of dispersion degree on the mineralization reaction rate on the mechanism level is explained in this section. By simulation data that CO<sub>2</sub> molecule number changes as a function of time, it can be concluded that mineralization speed becomes fast as the dispersion degree increases.

It can be concluded from Figure 11a that the CO<sub>2</sub> molecule movement and diffusion are fast around the solid surface when dispersion degree is low. However, collision frequency decreases. The bigger the dispersion degree, the more access there is by CO<sub>2</sub> molecules. By raising the dispersion degree, the mineralization reaction rate can be raised. Berndsen<sup>33</sup> also pointed out that accessible area is one of the rate-limiting factors in mineralization.



**Figure 10.** Average reaction rate as a function of the pressure at several time points.

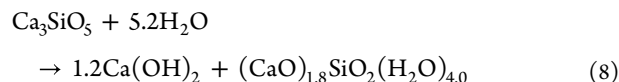
Figure 11b shows the radial distribution function (RDF) of Ca and Si under different dispersion degrees, which can be calculated as follows.

$$g(r) = \frac{dN}{4\pi r^2 \rho dr} \quad (7)$$

$dN$  is the atom number in the range from  $r$  to  $r + dr$ .  $\rho$  is the number density of atoms. RDF represents the distribution of the number density of one type of atom relative to another. The peak height represents the strength of the mutual interaction. It can be found that the peak of compact  $C_3S$  is the highest, while that of accumulated is in the middle. The peak height of spread  $C_3S$  is just 2/3 of compact since spread  $C_3S$  is taken by grinding solid and crystal structure is disrupted. The interaction strength between Ca and Si decreases, which contributes to  $CO_2$  being adsorbed and enhances the mineralization reaction. The experiment of natural and industrial alkaline solids carbonation also illustrated that atomic topology is the rate-controlling step.<sup>34</sup> This is the same as the conclusion for accessible areas.

**4.5. Effect of Water Film on Mineralization.** In real mineralization reactions, there is usually a water film covering a

solid in moist conditions.<sup>15–17,35</sup> Salah Uddin and Midendorff<sup>36</sup> pointed out that water tessellation was observed on the (001) surfaces of  $C_3S$ . The opposite surface (00 $\bar{1}$ ) has no spontaneous reactive site.<sup>37</sup> Thus, based on Figure 2, the model with water film covered on (001) surface was constructed, which is shown in Figure 4. The hydration process of  $C_3S$  can be expressed as<sup>38</sup>:

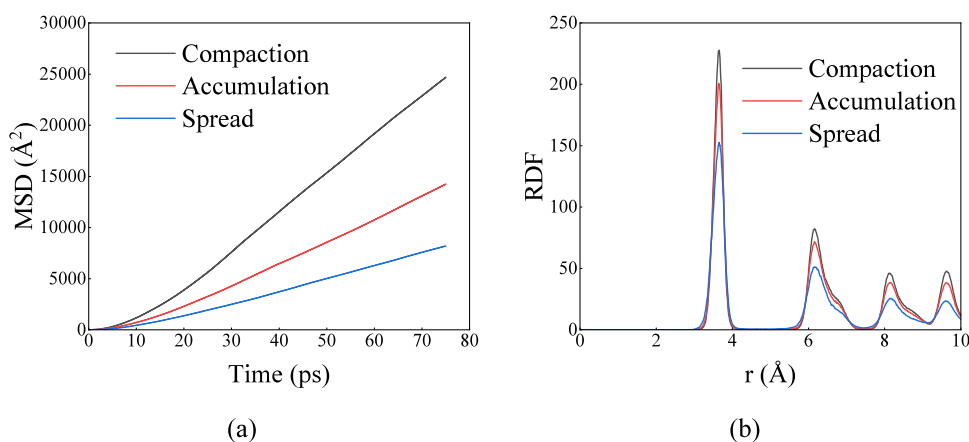


The difference in reaction speed with and without a water film is obvious, as shown in Figure 12a. Without water, the  $CO_2$  conversion rate is 95.7%, while it is only 84.3% with water. Without water, the molecular number of  $CO_2$  stays constant after 45.8 ps. From the initial state to this point, the average reaction speed of  $CO_2$  is  $4.27 \times 10^9$  mol/(L s), while it is  $3.82 \times 10^9$  mol/(L s) with water. Water film covered in the solid surface has a negative influence on mass transfer during mineralization, and the reaction rate and conversion rate will decrease, which is in coordination with others' work simulated by the ab initio MD method.

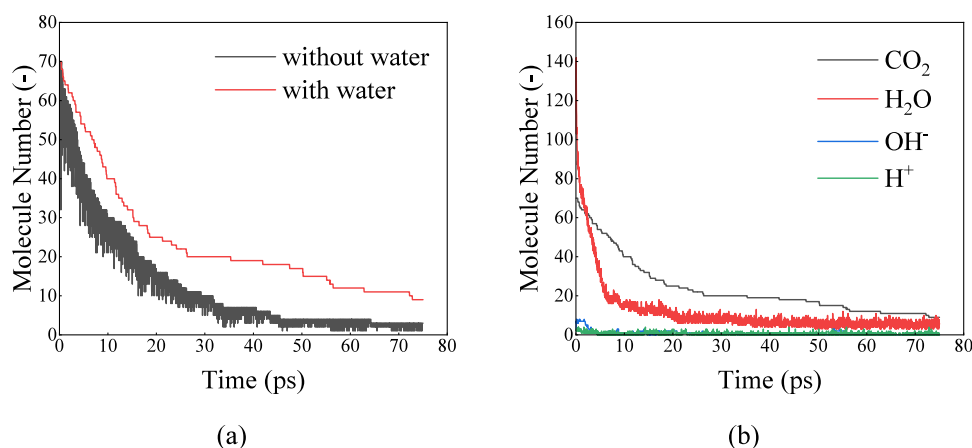
Figure 12b demonstrates the reaction path of mineralization with water. In the early stage (0–3.1 ps),  $C_3S$  hydration occurs, and numerous  $Ca(OH)_2$  is produced. The dissolution of  $OH^-$  can provide an alkaline environment for mineralization. This is usually called the alkali activation phenomenon.  $Ca(OH)_2$  dissolves in water and much  $OH^-$  comes into being. However,  $OH^-$  is run out quickly. Water molecules decrease very quickly since  $C_3S$  is the active hydration solid phase and the water film is directly covered on its surface. The number of water molecules is 142 at first and decreases to 18 at 7.0 ps. About 87.3% of water molecules participate in the reaction. Water molecules decrease to 8 at 21.1 ps and remain stable afterward.

Figure 13 shows the coordination number of Si atoms around Ca. The coordination is calculated from the surface layer of  $C_3S$  with a depth of 5 Å. At the initial state, the coordination number is around 8 and then decreases. Li et al.<sup>39</sup> also pointed out that the connectivity between Ca and Si can be up to eight. As the reaction progresses,  $CaSiO_3$  turns into  $CaCO_3$ . Thus, the coordination number of Si around Ca decreases as time goes by.

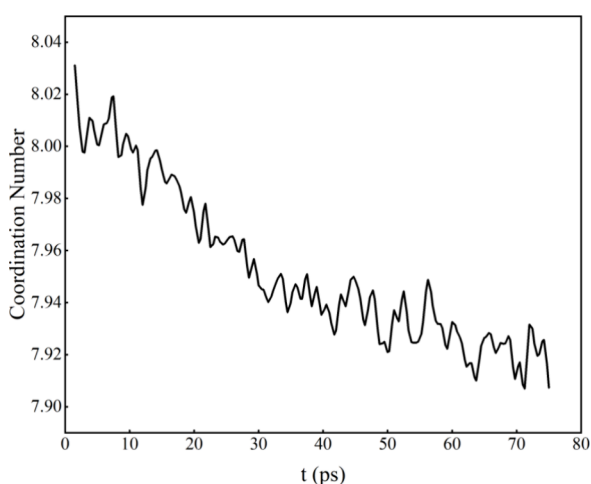
In conclusion, although the water film contributes to the alkali activation phenomenon and  $OH^-$  is produced,  $OH^-$  runs



**Figure 11.** Comparison among different types of solid dispersion degree. (a) MSD. (b) RDF of Ca and Si.



**Figure 12.** Molecule number as a function of time: (a) CO<sub>2</sub> molecule number with and without water and (b) molecule number during reaction.



**Figure 13.** Coordination number of Si atoms around Ca varies with time.

out within just 3.1 ps. The negative effect of water that hinders the diffusion of CO<sub>2</sub> to a solid surface is during the process, which takes more effect. In a word, water film decreases the mineralization reaction rate and conversion rate. Falzone<sup>40</sup> also pointed out that the drying process is necessary before and during mineralization.

**4.6. Discussion.** CO<sub>2</sub> mineralization reactions are affected by many factors, including inner and outer ones. This article focuses on temperature and pressure as outer factors and dispersion degree as inner factors. Temperature has two opposite effects on mineralization reaction dynamics. On the one hand, high temperature activates reactants and accelerates reaction. On the other hand, high temperature contributes to adsorbed CO<sub>2</sub> molecules getting rid of the C<sub>3</sub>S surface, which is not conducive to reaction dynamics. This work takes the active solid phase components, C<sub>3</sub>S to carry out research. In reality, industrial wastes contain many crystalline phases, which makes the reaction speed not as fast as this work. The temperature of 328 K is the best choice for industrial wastes whose main reactive solid phase is C<sub>3</sub>S. Poletini et al.<sup>10</sup> also pointed out that the optimum temperature for BOF slag is 50 °C by experiment, which is very close to the results in this article. High pressure made the reaction fast. However, higher pressures require reaction vessels with higher pressure capacities. In this way, the right pressure needs to be selected

to save production costs. Grinding and pulverizing industrial solid waste can increase the dispersion degree. High dispersion degree not only activates the solid phase but also increases the effective specific surface area for the gas–solid reaction. When mineralization is under moisture conditions, solid waste is usually covered by a layer of water film. Water promotes the phenomenon of alkali excitation in the early stage. However, the hydroxide ions produced are consumed in a very short period. The water film limits the direct contact of CO<sub>2</sub> molecules with C<sub>3</sub>S. This effect is throughout the whole reaction. Thus, the reaction rate and conversion rate decrease in general. RMD saves computational costs for simulating mineralization reactions compared with the DFT and AIMD methods. Thus, it can simulate a long period and reveal many aspects of mineralization. However, its accuracy relies on the force field. The model built in this article simulates direct mineralization. Usually, indirect mineralization is widely used in industry. Because the reaction condition is mild, the energy cost is low. In the future, indirect mineralization should be considered to be researched. The reaction dynamics investigated in this review can guide industrial practice.

## 5. CONCLUSIONS

By RMD simulation, the CO<sub>2</sub> mineralization reaction dynamics is investigated in this article. The model is composed of a C<sub>3</sub>S solid and CO<sub>2</sub> gas. The influence of temperature and pressure is investigated. Also, the water film is added in one section to show its role in the reaction. This research work is expected to provide theoretical guidance for quantitative regulation of industrial production. Based on the results, several conclusions can be drawn:

1. Temperature has a dual effect on the kinetics of the mineralization reaction. Elevating the temperature increases the reactant activity but also promotes the detachment of carbon dioxide molecules from the surface of the C<sub>3</sub>S solid phase. In terms of increasing the reaction rate, 328 K is the best choice.
2. Increasing pressure increases collision frequency and enhances reaction. The higher the pressure, the faster the reaction rate. In actual industrial production, the cost increase of increasing the pressure needs to be taken into account to determine the optimal pressure.
3. Grinding and pulverizing solid waste are often employed in engineering. The dispersion degree of the solid phase is classified into three types: compaction, accumulation,



and spread in this article. By raising the dispersion degree, the crystal structure of the solid phase is ruined and reactivity is enhanced. Moreover, the effective access area is increased. Therefore, increasing the dispersion degree can reinforce the mineralization reaction.

- In moisture conditions, the  $C_3S$  solid surface is usually covered with a layer of water film. The coordination number of Si surrounding Ca on the surface of  $C_3S$  was analyzed. During the reaction,  $C_3S$  turned into  $CaCO_3$ . Thus the coordination number decreased. Water facilitates the alkali excitation phenomenon and drives  $C_3S$  to produce a large amount of  $OH^-$  in a short period. However, the  $OH^-$  ions are consumed shortly, and the negative effect of water film that prevents direction access of  $CO_2$  and  $C_3S$  is throughout the entire process. Generally, the reaction rate and conversion rate decrease with water film. In the real mineralization process, keeping the environment dry can contribute to the process of reaction.

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### Notes

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