

## Facile Protection of Lithium Metal for All-Solid-State Batteries

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A nanolayer of reactive propyl acrylate silane groups was deposited on a lithium surface by using a simple dipping method. The polymerization of cross-linkable silane groups with a layer of ally-ether-ramified polyethylene oxide was induced by UV light. SEM analysis revealed a good dispersion of silane groups grafted on the lithium surface and a layer of polymer of about 4  $\mu$ m was obtained after casting and reticulation. The electrochemical performance for the unmodified and modified lithium electrodes were compared in symmetrical Li/LLZO/Li cells. Stable plating/stripping and low interfacial resistance were obtained when the modified lithium was utilized, indicating that the combination of silane and polymer deposition is promising to increase Li-metal/garnet contact.

Lithium-ion batteries are a promising energy storage technology. The demand for high energy storage in commercial devices and EVs requires the use of lithium metal anodes. Unfortunately, the standard carbonate-based liquid electrolyte presents a safety risk due to high flammability of the solvent. Lithium metal can easily form dendrites that cause short circuit in liquid electrolyte.<sup>[1,2]</sup> Solid electrolytes can serve as a physical barrier, and several options were considered: a polymer<sup>[3]</sup> like polyethylene oxide (PEO):LiTFSI blend, ceramics such as garnet<sup>[4]</sup> and Nasicon,<sup>[5]</sup> and hybrid ceramic polymer electrolyte.<sup>[6]</sup> Garnet lithium lanthanum zirconate Li<sub>7</sub>La<sub>3</sub>Zr<sub>2</sub>O<sub>12</sub> (LLZO) is a promising candidate as solid electrolyte due to its high structural stability at high voltage (up to 5 V) and good ionic conductivity (around 0.5 mS cm<sup>-1</sup>). The high resistance between LLZO and lithium metal is a serious issue that must be addressed in order to develop an all-solid battery. In addition, the conversion of cubic LLZO to poorly conductive tetragonal LLZO occurs at the lithium metal interface, as reported by Ma et al.<sup>[7]</sup> Actually, several physical and chemical methods were proposed to improve the contact between LLZO and lithium metal: PEO gel,<sup>[8]</sup>

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© 2019 The Authors. Published by Wiley-VCH Verlag GmbH & Co. KGaA. This is an open access article under the terms of the Creative Commons Attribution Non-Commercial NoDerivs License, which permits use and distribution in any medium, provided the original work is properly cited, the use is non-commercial and no modifications or adaptations are made. aluminum<sup>[9]</sup> and silicon deposition. In this work, we propose a new large-scale method to protect lithium metal via silane deposition combined with polymer casting and reticulation to improve its contact with the LLZO pellet. Porcarelli *et al.* report the UV-induced (co)polymerization of PEO with tetraglyme at various lithium salt concentrations.<sup>[10]</sup> In our work, we induced a UV polymerization by using a silane-modified lithium with cross-linkable groups and Ally-ether ramified PEO. We decided to use silane because of its good interaction with lithium metal as previously reported.<sup>[11,12]</sup>

PEO-LiTFSI is well known solution as solid polymer electrolyte as reported by Armand.<sup>[13]</sup> Nakayama et al. proposed the addition of the plasticizers poly(ethylene glycol) (PEG)-borate ester and the final battery Li/boron-PEO/LiFePO4 delivered stable charge/discharge cycles (>150 cycles) at 30 °C.<sup>[14]</sup> A recent approach proposed by Goodenough and co-workers consisted of adding a cross-linked polymer between garnet and lithium metal to improve their contact and delivered stable electrochemical cycling performance.<sup>[8]</sup> Inspired by their work, Scheme 1 illustrates a procedure to reduce the Li-metal/baregarnet interfacial resistance by a simple dipping method to form nanolayers of propyl acrylate silane groups on the lithium surface. Then, a thin layer (~4 µm) of Ally-ether ramified PEO containing a lithium salt and a photoinitiator was casted on its surface. After the cross-linking reaction under UV light, a strong link between the lithium and the polymer layer was obtained. This method is versatile and can be extend to other kind of cross-linkable polymers with acrylate or epoxy functions and other type of silane with mercapto or allyl ether end groups, for instance. Finally, our method is appealing because easily adapted to a large-scale process, in contrast to Atomic Layer



Scheme 1. Schematic representation of Li metal surface modified by a thin layer of reactive propyl acrylate silane groups on which a thin (ca. 4  $\mu m$ ) layer of ally-ether-ramified PEO+LiTFSI salt is deposited. A strong contact between the lithium surface and polymer is obtained after reticulation under UV light.

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**Figure 1.** SEM images of a) unmodified lithium, b) lithium@silane, and c) lithium@silane with PEO coating. The insert shows the PEO layer of about 4 µm thick on lithium surface.

Deposition (ALD) already used for the surface modification of garnet with Au,  $^{\rm [15]}$  Ge $^{\rm [16]}$  or ZnO $^{\rm [17]}$  coatings.

Unmodified and treated lithium surfaces were analyzed using scanning electron microscopy (SEM). Figure 1 shows the SEM images of a) unmodified lithium, b) lithium@silane and c) lithium@silane with PEO coating. The straight lines observed at the surface of fresh lithium (see Figure 1a) disappeared after reaction with propyl acrylate silane groups (see Figure 1b) confirming that a uniform coverage of few nm was obtained. The layer of PEO casted on lithium@silane is well observed by SEM (Figure 1c) and its thickness was estimated around 4 µm as put in evidence by the magnification in insert of Figure 1.

The effect of surface modification of lithium foil was examined using Li-metal/LLZO/Li-metal symmetric cells. Electrochemical impedance spectroscopy (EIS) was used to measure the interfacial resistance between modified or fresh lithium electrodes and the garnet electrolyte. Figure 2 shows the Nyquist plots obtained at 25 (Figure 2a) and 80 °C (Figure 2b) for unmodified lithium ( $\blacksquare$ ), lithium with PEO coating ( $\bullet$ ) and lithium@silane with PEO coating ( $\blacktriangle$ ). The Li-metal/LLZO/Li-metal cell ( $\blacksquare$ ) at 25 °C showed an obvious semicircle at high frequency (see insert in Figure 2a) that indicates a large resistance of about 550 k $\Omega$ , which is mainly attributed to the poor contact between the Li metal and the bare-garnet pellet. In fact, the garnet-structure Li<sub>7</sub>La<sub>3</sub>Zr<sub>2</sub>O<sub>12</sub> is characterized by high Li-ion conductivity (0.1–1 mS cm<sup>-1</sup>),<sup>[18]</sup> suggesting that the rate-controlling step is not in the bulk of the solid electrolyte, but at the interface between the electrode and the electrolyte material.<sup>[19]</sup> EIS data were modeled using an equivalent circuit shown in Figure 2c. Generally, a similar model is employed to depict the Li/LLZO/Li cells behavior<sup>[20]</sup> and is characterized by a succession of constant phase elements (CPE) and resistances related to the LLZO bulk, grain boundary and Li-LLZO interface.<sup>[21,22]</sup> Semicircles at high frequencies are attributable to the total resistance of the LLZO pellet and the low frequency semi-circles can be assigned to the interfacial resistance.<sup>[23]</sup> After PEO coating on lithium surface (lithium + PEO ()), the impedance has drastically diminished and in particular the resistance at the lithium-LLZO interface (low frequencies). Finally, the combination of both the silane and polymer layers (lithium@silane+PEO ( $\blacktriangle$ )) led to a total resistance of ~12 k $\Omega$ at 25 °C, very near of that obtained for unmodified lithium at 80°C (see insert in Figure 2b). This decrease in interfacial resistance is undoubtedly attributed to the greatly enhanced contact after lithium modification.

The voltage profiles for the Li/LLZO/Li symmetric cells at 80°C are presented in Figure 3, and the results are in agreement with the decreased interfacial resistance observed after surface modification of lithium. The cell assembled with fresh lithium (---) has higher voltage hysteresis for each cycling current and short-circuited after less than 130 h of cycling. In contrast, the voltage profiles for modified-Li/LLZO/modified-Li cells showed smaller voltage hysteresis of about 80 and 50 mV at 0.1 mA cm<sup>-2</sup> when lithium with PEO coating (—) and lithium@silane with PEO coating (---) were used, respectively. These cells were characterized by a long-term stable electrochemical striping/plating process (see Figure 3b). Moreover, the last one (—) displayed flat voltage plateaus during both plating and stripping. In conclusion, our work shows a new and scalable method to reduce the interfacial resistance between lithium metal and electrolyte ceramics. The symmetrical cells Li/silane-PEO/LLZO/silane-PEO/Li showed stable plating/stripping, indicating that the combination of silane and polymer deposition is promising for increasing the Li-metal/garnet contact.



**Figure 2.** Electrochemical impedance spectroscopy measurements recorded at a) 25 °C and b) 80 °C for Li/LLZO/Li symmetric cells assembled with: unmodified lithium ( $\blacksquare$ ), lithium with PEO coating ( $\blacklozenge$ ), and lithium@silane with PEO coating ( $\blacktriangle$ ). The inserts show the Nyquist plot when unmodified lithium ( $\blacksquare$ ) is used. C) Equivalent circuit used for modeling the EIS data represented by the red lines ( $\frown$ ).



**Figure 3.** a) Galvanostatic stripping-plating cycling at 80 °C at different current densities for Li/LLZO/Li symmetric cells assembled with: unmodified lithium (—), lithium with PEO coating (—) and lithium@silane with PEO coating (—). b) Long-term galvanostatic cycling at 0.1 mA cm<sup>-2</sup> for cells assembled with modified lithium electrodes.

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## **Experimental Section**

#### **Preparation of Polymer Solution**

Allyl-ether-ramified polyethylene oxide (provided by Hydro-Québec) was dispersed in a small amount of anhydrous tetrahydrofuran (THF from Sigma-Aldrich) that was mixed until a viscous solution was obtained. Lithium bis(trifluoromethanesulfonyl)imide salt (LiTF-SI from Sigma-Aldrich) corresponding to a O:Li molar ratio of 20:1 was then added to the polymer solution. After mixing for several hours, the polymer solution was diluted with THF to obtain 26.8 wt.% of solid (polymer and salt). Finally, 0.4 wt.% of Omnirad 651 photoinitiator (from IGM resins) was dispersed in the polymer solution, and then the mixture was stored in the dark.

# Surface Modification of Lithium Foil with 3-(Trimethoxysilyl) propyl Acrylate

One side of a lithium foil (39  $\mu$ m thick, provided by Hydro-Québec) was reacted for 2 minutes with 0.25 mL cm<sup>-2</sup> of a 10<sup>-1</sup> M 3-(Trimethoxysilyl)propyl acrylate in THF solution under a fume hood in a dry-room chamber. After the solvent was evaporated, the lithium foil was rinsed two times with fresh THF and placed under vacuum for at least 15 h.

#### **Polymer Casting on Lithium Foil**

The diluted polymer solution was first spread on a polypropylene (PP) sheet to estimate the thickness of the deposited film after drying and reticulation. Wet polymer casting corresponding to 1 mil (~25  $\mu$ m), 2 mils (~50  $\mu$ m) and 3 mils (~75  $\mu$ m) produced dry films of about 3–4, 6 and 9  $\mu$ m thickness, respectively. The fresh and silane-modified lithium foils (18 cm×4.5 cm) were then coated with the polymer solution (~20  $\mu$ m wet) and directly placed in an oven at 50 °C for 10 minutes. Subsequently, the lithium foils were introduced in a Büchner glass assembly and put under continuous vacuum for 2 minutes followed by exposure to UV light (300 WPI) at a distance of 50 cm for 5 minutes. After crosslinking, the lithium was kept under vacuum in an oven at 80 °C for at least 15 h before utilization.

#### Surface Morphology of Lithium Foil

Li metal surfaces were analyzed using scanning electron microscope (SEM) Lyra 3 by TESCAN. Samples were prepared in a dry chamber and inserted in the SEM using an air-tight transfer holder to minimize external contamination. The micrograph and X-ray map were acquired at an accelerating voltage of 5 kV, a probe current of 500 pA and a working distance of 9 mm.

#### Synthesis and Densification of LLZO Pellets

Ga-doped LLZO was synthesized following a solid-state method.<sup>[24]</sup> The LLZO pellets were prepared by annealing 1-mm thick pellets at 1100 °C for 16 h in an alumina crucible surrounded by sacrificial powder of LLZO. The final pellets were polished in a glove box.

### **Electrochemical Characterization**

Unmodified or modified lithium electrodes (10-mm diameter) were hot-pressed on each side of a LLZO pellet (13-mm diameter) at 80 °C under vacuum for 3 min. The components were assembled in a coin cell in a glove box filled with argon ( $O_2 < 10$  ppm,  $H_2O <$ 

10 ppm). The coin cells were electrochemically characterized with a BioLogic VMP3 potentiostat.

Electrochemical impedance measurements of the cells were performed at 25 and 80 °C with an AC amplitude of 10 mV and a frequency range of 10 MHz to 10 mHz. Galvanostatic stripping-plating cycling of the symmetric cells was recorded at 80 °C with different current densities ranging from 0.001 to 0.1 mA cm<sup>-2</sup>.

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## Conflict of interest

The authors declare no conflict of interest.

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