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# Temperature Dependent Micro-Structure of KAlF<sub>4</sub> from Solid to Molten States

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Abstract: In situ high temperature X-ray diffraction and Raman spectroscopy were used to investigate the temperature dependent micro-structure of KAlF<sub>4</sub>. Density functional theory was applied to simulate the structure of crystalline KAlF<sub>4</sub> while a quantum chemistry ab initio simulation was performed to explore the structure of molten KAlF<sub>4</sub>. Two crystal polymorphs demonstrated to be present in solid KAlF<sub>4</sub>. At the temperature below 673 K, it belongs to the tetragonal crystal system within the P4/mbm space group, while the high temperature phase is attributed to the monoclinic crystal system within the P2<sub>1</sub>/m space group. Both polymorph KAlF<sub>4</sub> phases are characterized by a layered structure consisting of K<sup>+</sup> and  $[AlF_6]^{3-}$  octahedra, each of the  $[AlF_6]^{3-}$  octahedra equivalently shares four corners with other four  $[AlF_6]^{3-}$  octahedra along the layer. The layered structure became unstable at higher temperatures and crashed when the temperature exceeded the melting point. It demonstrated that the molten KAlF<sub>4</sub> simulated by using a quantum chemistry ab initio method agreed well with the experimental Raman spectrum.

**Keywords:**  $KAlF_4$ ; in situ; layered structure;  $[AlF_6]^{3-}$  octahedra;  $[AlF_4]^-$  tetrahedra

## 1. Introduction

The physicochemical properties of molten fluoroaluminates are of considerable importance for the aluminum industry. Molten  $KAlF_4$  had been employed not only as electrolyte in producing aluminum [1] but also as a basic part of the flux in the aluminum brazing technology [2]. Hence an understanding of the molten structure of  $KAlF_4$  is essential for obtaining necessary knowledge of the properties of the electrolytes for aluminum smelters and the fluxing agent for the aluminum brazing industry.

Over the years, limited work has been carried out focusing on the structure feature of molten KAlF<sub>4</sub>. The room temperature phase [3–5], the structure from the low temperature (4 K) to room temperature [6,7] and the molten states [8] of KAlF<sub>4</sub> were extensively investigated. According to Chen's work [9,10], congruent melting occurred at 575 °C when KAlF<sub>4</sub> was heated. The KAlF<sub>4</sub> crystal consists of six-coordinated aluminum atoms, which are linked to four bridging fluorine ions, two non-bridging fluorine ions and potassium ions [7,11,12]. The main vibrational wavenumber of molten NaAlF<sub>4</sub> at about 630 cm<sup>-1</sup> was detected by Robert [13] by using high temperature Raman spectroscopy. With the consideration of the results by high temperature NMR spectra, the band at Raman wavenumber of 630 cm<sup>-1</sup> was further assigned to the symmetric stretching vibrations of the



 $[AlF_4]^-$  tetrahedron. Akdeniz concluded that the Raman peaks observed at 621 cm<sup>-1</sup> for NaAlF<sub>4</sub> are related to the Al-F bond of isolated  $[AlF_4]^-$  clusters with a bond length of 1.69 Å [14], as was also assured by molecular simulation [15]. Due to the high corrosivity of KAlF<sub>4</sub> melt, the structure evolution from room temperature to the molten states of KAlF<sub>4</sub> has not been cleared.

In the present paper, temperature dependent Raman spectroscopic studies of the micro-structure of  $KAlF_4$  up to the molten state were conducted in conjunction with high temperature X-ray diffraction to explore the structure evolution of  $KAlF_4$  from room temperature to above its melting point. Raman spectra simulations based on density functional theory (DFT) and quantum chemistry ab initio methods were also applied to make the assignments of the characteristic vibration modes of Raman spectra observed at different temperatures.

## 2. Materials and Methods

#### 2.1. Material Synthesis

All chemicals including AR grade  $K_2CO_3$ , Al(OH)<sub>3</sub>, and other reagents used in the present study, were obtained from Sinopharm Chemical Reagent Co., Ltd.  $K_2CO_3$  and Al(OH)<sub>3</sub> were first dried at 473 K for 2 h to remove the surface moisture. Stoichiometric amounts of  $K_2CO_3$  and Al(OH)<sub>3</sub> were then mixed with an excess amount of 40% hydrofluoric acid in a teflon cup to form a paste. The paste was dried at 373 K in a platinum crucible. After being ground in an agate mortar, the dried mixture was annealed in a platinum crucible at 723 K for 48 h [9]. The resultant product was ground for subsequent experiments.

#### 2.2. Material Characterization

In situ high temperature X-ray powder diffraction studies were conducted using the Bruker D8 Advance X-ray diffractometer (40 KV, 40 mA) with Cu/K $\alpha$ 1 radiation ( $\lambda$  = 0.15406 nm). The patterns were recorded in an inert atmosphere in the 2 $\theta$  range of 10 to 100° with a step size of 0.02° and a counting time of 0.33 s per step.

In situ high temperature Raman spectroscopic studies were carried out under a dry inert atmosphere using the LabRAM HR800 Raman spectrometer (Horiba Jobin Y'von, Paris, France) equipped with an ultraviolet pulse laser beam of 355 nm which was focused on the sample through a microprobe with a  $4 \times$  objective lens. The average laser beam power on the sample was about 60 mW. It was equipped with a microscopic heating furnace (TS1500) (Linkam, Tadworth, UK) with a small temperature deviation of about  $\pm 1$  K for investigating the microstructure of the sample at different temperatures. The platinum crucible which was applied to measure the high temperature experiment is protected by a patent [16]. A charge coupled device (CCD) detection system by an accumulated mode of  $20 \times 20$  (20 times with 20 s each time) was used to collect Raman scattering light. The sample was held at the targeted temperature for 5 min before recording Raman spectra in order to ensure the sample reaches the targeted temperature and also avoids volatilization of the sample.

#### 2.3. Computational Details

By using the functional set GGA [17] (WC [18]) version with optimized norm-conserving pseudopotentials [19], CASTEP (Cambridge Serial Total Energy Package [20,21]) based on DFT with the plane-wave pseudopotential method [22], was applied to calculate vibrational modes of KAIF<sub>4</sub>. The plane-wave cut-off energy was fixed at 780.0 eV. The other parameters which were set in the computation, included a grid of  $3 \times 2 \times 2$  k-points and a self-consistent field (SCF) convergence threshold of  $2 \times 10^{-6}$  eV/atom.

In order to examine the clusters existing in the molten KAlF<sub>4</sub>, a simulation based on a quantum chemistry ab initio method was also carried out. A series of aluminum fluoride model clusters with typical characteristic structures were proposed and optimized for geometric configuration before the simulation. The restricted Hartree-Fock (RHF) [23] calculation method with a 6-31G (d) [24] basis set was applied for geometry optimization and the calculation of the vibration frequencies of various molecules.

#### 3. Results and Discussion

## 3.1. Room Temperature Characteristics of KAlF<sub>4</sub>

The XRD pattern of the synthesized  $KAlF_4$  compound is shown in Figure 1. Both the peak position and relative intensity of main diffraction peaks, were in good agreement with the information provided in the powder diffraction file (PDF) data base for  $KAlF_4$ , confirming the dominance of the  $KAlF_4$ compound in the sample.



**Figure 1.** Room temperature XRD pattern of the synthesized KAlF<sub>4</sub> compound. The PDF card No.77-2210 was obtained from the JCPD (Joint Committee for Powder Diffraction Standards) database for synthetic KAlF<sub>4</sub>.

At the ambient temperature and pressure, the KAlF<sub>4</sub> phase is characterized by a layered structure consisting of K<sup>+</sup> and  $[AlF_6]^{3-}$  octahedra, each  $[AlF_6]^{3-}$  octahedron equivalently shares four corners with other four  $[AlF_6]^{3-}$  octahedra within the layer. The  $[AlF_6]^{3-}$  octahedron consists of a six-coordinated aluminum atom which is linked to six fluorine ions including four bridging fluorine ions and two non-bridging fluorine ions. Figure 2 illustrates a unit cell structure of the room temperature KAlF<sub>4</sub> crystal, drawn by the VESTA software [25]. The room temperature KAlF<sub>4</sub> unit cell is tetragonal with dimensions a = 5.122 Å, b = 5.122 Å, c = 6.288 Å,  $\alpha = \beta = \gamma = 90^{\circ}$ , V = 164.999 Å<sup>3</sup>, belonging to the P4/mbm space group [3].



**Figure 2.** The unit cell structure of the room temperature  $\text{KAlF}_4$  crystal. The purple balls represent the K atoms while the dark cyan octahedral structures represent the  $[\text{AlF}_6]^{3-}$  octahedra.

The room temperature Raman spectrum was calculated by CASTEP for the KAIF<sub>4</sub> crystal and is compared in Figure 3 with the experimental spectrum obtained for the synthesized KAIF<sub>4</sub> compound. The small shift in wavenumbers between the experimental and calculated spectra is due to the precision and limitation of the theoretical simulation methods. In order to compare with the experimental spectrum, the calculated spectrum was correct by a factor  $k_1 = 0.000426x + 0.80$ , where the factor  $k_1$  is a function of wavenumber *x*.



**Figure 3.** The room temperature Raman spectrum of the KAlF<sub>4</sub> crystal is compared with the calculated (using CASTEP) followed by a Raman shift correction with a factor of  $k_1 = 0.000426x + 0.80$  and the Lorentzian smearing is 10 cm<sup>-1</sup>.

In Figure 3, there are two well defined peaks in the calculated spectrum, which locates at 227.6 and 547.4 cm<sup>-1</sup>. The calculation results show that the band at 227.6 cm<sup>-1</sup> relates to the shearing vibrations of  $AI^{VI}$ - $F_{nb}$  ( $F_{nb}$  represents non-bridging fluorine) while the band at 547.4 cm<sup>-1</sup> corresponds to the symmetric stretching vibrations of  $AI^{VI}$ - $F_{nb}$ . According to the group theory analysis, the vibrational modes are distributed among the following irreducible representations:  $4A_{2u} + 16E_u + 2B_{2u} + 2A_{1g} + 6E_g + 1B_{2g} + 2A_{1u} + 2A_{2g} + 1B_{1g}$ , where only  $A_{1g}$ ,  $E_g$ ,  $B_{2g}$  and  $B_{1g}$  are Raman-active modes based on the selection rules. Similarly, two peaks were observed at 228 and 547 cm<sup>-1</sup> in the wavenumber range of 200–600 cm<sup>-1</sup> of the experimental spectrum at room temperature. Comparing with the calculated results of the room temperature KAIF<sub>4</sub> crystal, the band at 228 cm<sup>-1</sup> is believed to be due to the shearing vibrations of  $AI^{VI}$ - $F_{nb}$ , while the band at 547 cm<sup>-1</sup> results from the symmetric stretching vibration peaks. Therefore, it is feasible to apply DFT to calculate Raman spectra of fluoroaluminate systems [26,27]. The major vibration modes of the room temperature phase of KAIF<sub>4</sub> are listed in Table 1. And all of the calculated Raman modes of the room temperature phase of KAIF<sub>4</sub> are shown in Table S1 (in Supplementary Materials).

 Table 1. The assignment of major room temperature vibration modes of crystalline KAlF<sub>4</sub>.

Wavenumber		171	Tune of Vibratian
$v_{exp}$ (cm <sup>-1</sup> )	$v_{cal}$ (cm <sup>-1</sup> )	vibrational Modes	Type of vibration
228	227.6	$E_g$	$\gamma$ (Al <sup>VI</sup> -F <sub>nb</sub> ) <sup>1</sup>
547	547.4	$A_{1g}$	$v_{s}$ (Al <sup>VI</sup> -F <sub>nb</sub> ) <sup>1</sup>

 $^{1}$   $\gamma$  and  $\upsilon$  represent the shearing and stretching vibrations, respectively. Subscript s represents the symmetric vibration. Al<sup>VI</sup> denotes a six-coordinated Al atom.

## 3.2. The Phase Transformation of KAlF<sub>4</sub>

The temperature dependent X-ray diffraction spectra of the synthesized  $KAlF_4$  compound were recorded while the sample was heated from room temperature to 823 K. The sample was heated at about 10 K/min under an argon atmosphere and held at each temperature point for five minutes prior to measurement to ensure that the sample reached the designated temperature. As shown in Figure 4, the sample maintained in its room temperature tetragonal structure until 673 K and then transformed to the high temperature phase as the sample temperature was further raised from 673 to

723 K. The high temperature phase of KAlF<sub>4</sub> is characterized by a monoclinic structure, belonging to the P2<sub>1</sub>/m space group, with its unit cell dimensions a = 6.542 Å, b = 7.195 Å, c = 7.177 Å,  $\alpha$  = 90°,  $\beta$  = 108.478°,  $\gamma$  = 90°, V = 320.415 Å<sup>3</sup>, which is similar to the structure in the literature [7].



**Figure 4.** Temperature dependent X-ray diffraction spectra collected from room temperature up to 823 K for the synthesized KAIF<sub>4</sub> compound. No.77-2210 and 77-2211 were obtained from the JCPD (Joint Committee for Powder Diffraction Standards) database for the tetragonal and monoclinic KAIF<sub>4</sub> unit cell structures, respectively.

Figure 5 shows a monoclinic unit cell structure of the high temperature phase of KAlF<sub>4</sub> which consists of potassium ions and a six-coordinated Al atom forming an octahedron. The high temperature phase was also characterized by a layered structure consisting of K<sup>+</sup> ions and  $[AlF_6]^{3-}$  octahedra, where each  $[AlF_6]^{3-}$  octahedron shares four corners with other  $[AlF_6]^{3-}$  octahedra within the layer. The potassium ions between the  $[AlF_6]^{3-}$  octahedron layers were used to keep the charge balanced. Comparing with the room temperature phase, the structure of the high temperature phase is more distorted due to the temperature impact, resulting in a more complicated XRD spectrum, as evidenced in Figure 4.



**Figure 5.** The unit cell structure of high temperature phase of KAlF<sub>4</sub>. The purple balls represent K ions, while the dark cyan octahedral structures represent  $[AlF_6]^{3-}$  octahedra.

The Raman spectrum of the high temperature phase of  $KAlF_4$  was calculated (based on its cell structure in Figure 5) by CASTEP and is shown in Figure 6 together with the experimental spectrum obtained at 773 K. The calculated wavenumber values of the high temperature phase of

KAlF<sub>4</sub> were corrected with a factor  $k_2 = -0.000835x + 1.41$  to account for the shifts for comparing with the experimental spectrum. Overall, there were very little distinction between the simulated and experimental spectra for the main vibrational peaks. As shown in Figure 6, four peaks, located at 193.9, 335.9, 474.5 and 539.5 cm<sup>-1</sup>, respectively, were identified from the calculated spectrum. Based on simulation, the bands at 193.9, 335.9 and 474.5 cm<sup>-1</sup> are assigned to the shearing vibrations of Al<sup>VI</sup>-F<sub>nb</sub>, while the Raman band at 539.5 cm<sup>-1</sup> is assigned to the symmetric stretching vibrations of Al<sup>VI</sup>-F<sub>nb</sub>. According to the group theory analysis, the vibrational modes are distributed among the following irreducible representations:  $13B_g + 19A_u + 23B_u + 17A_g$ , where only  $A_g$  and  $B_g$  are Raman-active modes based on the selection rules. The major vibration modes calculated for the high temperature phase of

KAlF<sub>4</sub> are listed in Table 2. And all of the calculated Raman modes of the high temperature phase of KAlF<sub>4</sub> are shown in Table S2 (in Supplementary Materials). Similar peaks were observed at 229, 319, 481 and 539 cm<sup>-1</sup> in the 773 K experimental spectrum. Comparing with the calculated spectrum of the high temperature phase of KAlF<sub>4</sub>, the bands at 229, 319 and 481 cm<sup>-1</sup> are believed to represent the shearing vibration of Al<sup>VI</sup>-F<sub>nb</sub> and the band at 539 cm<sup>-1</sup> is relating to the symmetric stretching vibrations of Al<sup>VI</sup>-F<sub>nb</sub>. Two new bands which are located at 319 and 481 cm<sup>-1</sup>, were observed at 773 K, are related to the transformation of the unit cell structure from tetragonal to monoclinic.



**Figure 6.** The Raman spectrum of the high temperature phase of KAIF<sub>4</sub> calculated by CASTEP (using a Raman shift correction with a factor of  $k_2 = -0.000835x + 1.41$ ) and the experimental Raman spectrum of the same phase at 773K. The Lorentzian smearing is 40 cm<sup>-1</sup>.

Wavenumber		Vibrational Madea	Type of Vibration
$v_{exp}$ (cm <sup>-1</sup> )	$v_{cal}$ (cm <sup>-1</sup> )	vibrational wodes	Type of vibration
229	193.9	$B_g$	$\gamma (Al^{VI}-F_{nb})^{1}$
319	335.9	$A_g$	$\gamma$ (Al <sup>VI</sup> -F <sub>nb</sub> ) <sup>1</sup>
481	474.5	$A_g$	$\gamma$ (Al <sup>VI</sup> -F <sub>nb</sub> ) <sup>1</sup>
539	539.5	$A_g^{\circ}$	$v_{\rm s}$ (Al <sup>VI</sup> -F <sub>nb</sub> ) <sup>1</sup>

Table 2. The assignment of major vibration modes of high temperature phase of KAlF<sub>4</sub>.

 $^1$   $\gamma$  and  $\upsilon$  represent the shearing and stretching vibrations, respectively. Subscript s represents the symmetric vibration. Al^{VI} denotes a six-coordinated Al atom.

Figure 7 presents the Raman spectra of the KAlF<sub>4</sub> crystal recorded when the sample was heated (at a heating rate of 10 K/min) from room temperature to 823 K and at a heating rate of 5 K/min afterwards. The high temperature spectra were recorded 5 min after the sample reached the target temperature. Figure 7 includes the calculated Raman spectrum of  $[AlF_6]^{3-}n$  clusters (by ab initio method) which refers to our previous work [28]. As the sample temperature increased, the Raman band at 547 cm<sup>-1</sup> shows a clear evidence of band broadening and redshifts in wavenumber.



Wavenumber/cm<sup>-1</sup>

**Figure 7.** Temperature dependent Raman spectra of  $\text{KAlF}_4$  and the Raman spectra calculated by DFT and an ab initio method. The value of n is equal to 7 in the calculated  $[\text{AlF}_4]^-{}_n$  by ab initio. The calculated  $[\text{AlF}_6]^{3-}{}_n$  by ab initio is referred by our previous work [28].

Two new Raman bands at 319 and 481 cm<sup>-1</sup> started to appear at 673 K and were gradually enhanced as the sample temperature increased from 673 to 823 K. As discussed earlier, the bands at 319 and 481 cm<sup>-1</sup> were attributed to the shearing vibrations of  $Al^{VI}$ - $F_{nb}$  of the  $[AlF_6]^{3-}$  octahedra with the high temperature phase. As clearly shown in Figure 4, the sample transformed from the tetragonal to monoclinic structure as the sample temperature was further raised from 673 to 723 K. Table 3 summarizes the structure parameters of the room and high temperature phases. Figure 8 illustrates an intermediate KAlF<sub>4</sub> structure as it evolves from the room to high temperature structure. The room temperature and high temperature phase are drawn in a simple diagram. In Figure 8, the grey unit cell represents the room temperature phase while the chromatic unit cell constitutes the high temperature phase. As KAlF<sub>4</sub> turns from the tetragonal to monoclinic structure approximately at 723 K, the space group changes from P4/mbm to P2<sub>1</sub>/m. The cell length becomes much longer, and one of the cell angles increases from 90.000 to 108.478°. As a consequence, the distribution of the bond length and bond angle of KAlF<sub>4</sub> becomes more complicated.

As the sample temperature increased further, the sample was fully melted at 893 K and is characterized by the molten structure as evidenced in Figure 7.



Figure 8. Intermediate KAIF<sub>4</sub> structure as it evolves from the room temperature to high temperature structure.

Parameter		Compound	
		Room Temperature Phase	High Temperature Phase
Crystal System Space Group		tetragonal P4/mbm	monoclinic P2 <sub>1</sub> /m
Cell Length/Å	а	5.122	6.542
	b	5.122	7.195
	с	6.288	7.177
Cell Angle/°	α	90.000	90.000
	β	90.000	108.478
	γ	90.000	90.000
Bond Length/Å	Al <sup>VI</sup> -F <sub>b</sub>	1.733	1.829, 1.800
	Al <sup>VI</sup> -F <sub>nb</sub>	1.674	1.769, 1.768
Bond Angle/°	∠F <sub>b</sub> -Al <sup>VI</sup> -F <sub>b</sub>	90.000	89.307, 90.693, 90.307, 89.693
	$\angle F_b$ -Al <sup>VI</sup> - $F_{nb}$	90.000	90.958, 91.871, 88.129, 89.042, 91.204, 92.520, 87.480, 88.796
	∠Al <sup>VI</sup> -F <sub>b</sub> -Al <sup>VI</sup>	167.476	170.788(2), 159.189, 158.912

Table 3. Structure parameters of room and high temperature phases of KAlF<sub>4</sub>.

## 3.3. Characteristics of Molten KAlF<sub>4</sub>

According to the literatures [29,30],  $[AIF_4]^-$  is the main species in molten KAIF<sub>4</sub>. Figure 9 illustrates a series of possible  $[AIF_4]^-_n$  model clusters presented in molten KAIF<sub>4</sub>. After being optimized for geometry configuration, these model clusters were used for a quantum chemistry ab initio method to calculate the Raman spectra. The characteristic vibrational wavenumber of the Al-F symmetric stretching vibrational modes of the  $[AIF_4]^-_n$  clusters were then estimated accordingly.



**Figure 9.** Model clusters used for quantum chemistry ab initio methods:  $[AlF_4]^-$  denotes an ionic molecule consisting of a four-coordinated Al atom and four F atoms while n (=1,2,3,4) stands for the number of  $[AlF_4]^-$  molecules in the clusters. The purple, beige and cyan balls represent K, Al and F atoms, respectively. The spatial configurations of the central atoms of  $[AlF_4]^-$  in the cluster are point, linear, triangular and tetrahedronal respectively from n = 1 to 4.

The restricted Hartree-Fock calculation method using a 6-31G (d) [31,32] basis set and was applied for geometry optimization and simulation of Raman vibrational modes of possible model clusters in the melt. When the calculated Raman spectra are compared with the experimental results, corrections are often needed for the intensity and frequency due to the effect of experimental temperature and the laser wavelength [33]. The wavenumbers of the calculated Raman spectra in Figure 10 have been corrected using a scaling factor of 0.90 [34] to better represent the experimental Raman spectrum. The characteristic wavenumber of Al-F symmetric stretching vibrational modes was further estimated from Figure 10 and compared with the experimental and reported values in Table 4. The local structure environment of a crystal is believed to have brought an impact on the structure in of the melt when it is melted. Therefore it was assumed that there are six  $[AlF_4]^-$  tetrahedra in the neighbor of an isolated  $[AlF_4]^-$  tetrahedron in the melt according to the crystal KAlF<sub>4</sub> structure in KAlF<sub>4</sub>. In Figure 11, the characteristic wavenumber of the symmetric stretching vibrational modes of a particular model cluster was plotted in a log scale as a liner function of the number of molecules in the  $[AlF_4]^-n$  cluster. As the number of molecules in the clusters increased, the characteristic wavenumber of the Al-F symmetric stretching vibrational mode of  $[AlF_4]^-n$  appeared to increase. The symmetric stretching vibrational wavenumber of the model cluster with seven molecules of  $[AlF_4]^-$  was therefore estimated to be 625.6 cm<sup>-1</sup>, which was in good agreement with the values observed in the experiment and reported in the literature [15,35–37].



Table 4. Calculated, experimental and reported Raman wavenumber of [AlF<sub>4</sub>]<sup>-</sup> clusters.

**Figure 10.** Calculated Raman spectra of various model clusters (shown in Figure 9), corrected for frequency shift with a factor of 0.90, and experimental spectrum of molten KAIF<sub>4</sub> (Spectrum A).



**Figure 11.** Calculated characteristic Raman wavenumber of Al-F symmetric stretching vibrational modes as a function of log n (the number of molecules in  $[AlF_4]^{-}_n$  cluster). The reference point A in the figure represents the experimental characteristic wavenumber of molten KAlF<sub>4</sub>.

As shown in Figure 7, the Raman band of 628 cm<sup>-1</sup>, which was attributed to the symmetric stretching vibrations of  $Al^{IV}$ - $F_{nb}$  of the  $[AlF_4]^-$  tetrahedron, appeared at 863 K. Simultaneously, the intensity of the Raman band of 628 cm<sup>-1</sup> increased while the intensity of the Raman band of 536 cm<sup>-1</sup> decreased. The six-coordinated Al atom began to turn into four-coordinated. At 893 K, KAlF\_4 had been completely melted, revealing the symmetric stretching vibrations of  $Al^{IV}$ - $F_{nb}$  of the  $[AlF_4]^-$  tetrahedron and little symmetric stretching vibrations of  $Al^{VI}$ - $F_{nb}$  of the  $[AlF_4]^-$  octahedron were in the melt.

## 4. Conclusions

The structure evolution of  $KAlF_4$  was investigated by in situ high temperature X-ray diffraction and Raman spectroscopy. The major vibrational modes of temperature dependent Raman spectra of  $KAlF_4$  were assigned based on the calculated spectra by both density functional theory and quantum chemistry ab initio methods.

Below the melting point, solid KAlF<sub>4</sub> was found to present in two different polymorph phases at room and high temperature. Both polymorph KAlF<sub>4</sub> phases are characterized by a layered structure consisting of K<sup>+</sup> and  $[AlF_6]^{3-}$  octahedra, where each  $[AlF_6]^{3-}$  octahedron equivalently shares four corners with other four  $[AlF_6]^{3-}$  octahedra along the layer. While the room temperature phase is in the tetragonal cell (space group P4/mbm), the high temperature phase undergoes a monoclinic distortion (space group P2<sub>1</sub>/m) together with an ordering of the K atoms.

The layer structure decomposed with increasing temperature and finally turned to  $[AlF_4]^-$  tetrahedra when the temperature exceeds the melting point. It was hence indicated that in molten KAlF<sub>4</sub>, there is a principal amount of  $[AlF_4]^-$  and a small amount of  $[AlF_6]^{3-}$ . This was further confirmed by quantum chemistry ab initio simulations.

**Supplementary Materials:** The following are available online at http://www.mdpi.com/1996-1944/11/10/1846/ s1, Table S1: The assignment of all calculated room temperature Raman vibration modes of crystalline KAlF<sub>4</sub>, Table S2: The assignment of all calculated Raman vibration modes of high temperature phase of KAlF<sub>4</sub>.

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