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One Step Synthesis of Tetragonal-CuBi₂O₄/ Amorphous-BiFeO₃ Heterojunction with Improved Charge Separation and Enhanced Photocatalytic Properties

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Abstract: Tetragonal CuBi₂O₄/amorphous BiFeO₃ (T-CBO/A-BFO) composites are prepared via a one-step solvothermal method at mild conditions. The T-CBO/A-BFO composites show expanded visible light absorption, suppressed charge recombination, and consequently improved photocatalytic activity than T-CBO or A-BFO alone. The T-CBO/A-BFO with an optimal T-CBO to A-BFO ratio of 1:1 demonstrates the lowest photoluminescence signal and highest photocatalytic activity. It shows a removal rate of 78.3% for the photodegradation of methylene orange under visible light irradiation for 1 h. XPS test after the cycle test revealed the reduction of Bi³⁺ during the photocatalytic reaction. Moreover, the as prepared T-CBO/A-BFO show fundamentally higher photocatalytic activity than their calcinated counterparts. The one-step synthesis is completed within 30 min and does not require post annealing process, which may be easily applied for the fast and cost-effective preparation of photoactive metal oxide heterojunctions.

Keywords: tetragonal CuBi₂O₄; amorphous BiFeO₃; visible light absorption; charge carrier recombination; photocatalytic property

1. Introduction

Bismuth-based multinary metal oxides photocatalysts have attracted wide interest because of their high visible light photocatalytic activity [1]. The valence band of bismuth-based oxides is composed of hybrid orbitals of Bi 6s and O 2p, the presence of Bi 6s slightly above O 2p endows bismuth-based oxides with reduced bandgap for visible light absorption compared to other metal oxide semiconductors [2]. Among the various bismuth-based oxides, multinary metal oxides ($Bi_xM_yO_z$, M represents other metal) have recently attracted increasing attention since they have shown fewer material limitations, higher controllability and enhanced photoactivity than their binary counterparts [3]. Specifically, CuBi₂O₄ has emerged as a promising photocatalytic material because of its suitable bandgap (1.5–1.8 eV) and high photovoltage [4–6]. Tetragonal CuBi₂O₄ possesses strong visible light absorption and



exhibits great potential for photocatalytic degradation of organic pollutants [7–9]. Perovskite BiFeO₃, with special multiferroic property for recycling, is another emerging bismuth-based multinary metal oxide showing great potential for photodegradation of organic pollutants due to its high visible light response [10–12]. However, both CuBi₂O₄ and BiFeO₃ suffer from severe carrier recombination due to their intrinsic low polaron conductivity [13,14], which significantly limits their photocatalytic activities [15–18]. One way to suppress the carrier recombination is to form heterojunctions by combining two semiconductors. Lots of efforts have been paid to combine CuBi₂O₄ with TiO₂ [19], NaTaO₃ [20], SnO₂ [21], CeO₂ [22], g-C₃N₄ [5], Fe₂O₃ [7], Bi₂WO₆ [23], Bi₂MoO₆ [24], and incorporate BiFeO₃ with Bi₂WO₆ [25], CuS [26], Ag₃PO₄ [27], BiOCl [11], CuO [28], ZnO [12], BiVO₄ [29], C₃N₄ [30]. Such semiconductor combining has been proved to be an effective method to improve the charge separation and photoactivity of CuBi₂O₄ and BiFeO₃. Nevertheless, in most cases the fabrication of the metal oxide-based heterojunction involves at least two steps: the preparation of each semiconductor photocatalyst, followed by the physical mixing and high temperature calcination [31].

The heterojunction is formed during high temperature calcination [31]. In this work, the T-CBO/A-BFO heterojunction was successfully formed within 30 min using a

one-step solvothermal treatment at 120 °C without the need of post annealing. We demonstrated that the metal oxide photocatalysts do not necessarily need to be completely crystallized for high photocatalytic property. The formation of the T-CuBi₂O₄/A-BiFeO₃ heterojunction resulted in significant improvement in visible light absorption up to 800 nm, enhanced charge separation, and consequently increased photocatalytic properties. We revealed that the oxidation and reduction of Bi³⁺ in the composite during the photocatalytic reaction highlights the importance of protection for the T-CBO/A-BFO. This work provides a facile and cost-effective method for the rapid fabrication of metal-oxides-based photocative compounds.

2. Materials and Methods

2.1. Preparation

T-CBO/A-BFO and A-BFO were synthesized by a short-time solvothermal method we have developed previously for the preparation of T-CBO. Specifically the precursor solution of A-BFO was obtained by mixing 15 mL solution of 40 mmol/L Fe(NO₃)₃·9H₂O (99.99%, Aladdin (China) Chemical Co. Ltd., Shanghai, China) in ethanol with 4.5 mL solution of 400 mmol/L Bi(NO₃)₃ (99.999%, Aladdin (China) Chemical Co. Ltd., Shanghai, China) in acetic acid. Total of 6.5 mol/L NaOH (98%, Alfa Aesar (China) Chemical Co. Ltd., Shanghai, China) solution was added to the precursor solution under magnetic stirring until pH 14 was reached; the mixture was then solvothermal treated at 120 °C in a 100 mL Teflon-lined steel autoclave for 30 min. For T-CBO/A-BFO composites, the precursor solutions were prepared by mixing Bi(NO₃)₃ in acetic acid (400 mmol/L), Cu(NO₃)₂·3H₂O (99.99%, Aladdin (China) Chemical Co. Ltd., Shanghai, China) in ethanol (40 mmol/L), and Fe(NO₃)₃·9H₂O in ethanol (40 mmol/L). Various T-CBO to A-BFO ratios (T-CBO/A-BFO (1:4), T-CBO/A-BFO (1:2), T-CBO/A-BFO (1:1), T-CBO/A-BFO (2:1), T-CBO/A-BFO (3:1), T-CBO/A-BFO (4:1)) could be obtained by adjusting the molar ratios of $Cu(NO_3)_2 \cdot 3H_2O$: $Fe(NO_3)_3 \cdot 9H_2O$: $Bi(NO_3)_3 = (1:4:6, 1:2:4, 1:1:3, 2:1:5, 3:1:7, 4:1:9)$, the details for the preparation of solvothermal solutions with various ion ratios can be found in Table S1. To investigate the effect of post-annealing, the as prepared samples were calcinated in a Muffle furnace at 450 °C for 2 h.

A Bruker D8 Advance X-ray diffractometer with Cu Kα radiation was used to analyze the crystal structure of the samples. Raman analysis was conducted using a BWS 465–785 H (BWTEK, Newark, DE, USA) spectrometer with a 785 nm laser source. Scanning electron microscopy (SEM, HITACHI, Tokyo, Japan) and a FEI Titan G2 60-300 transmission electron microscope (TEM, FEI company, Hillsboro, OR, USA) were applied to characterize the morphologies of the photocatalysts. The chemical states of the elements were tested via an ESCALAB 250XI X-ray photoelectron spectroscopy (XPS, Thermo Fisher Scientific, Waltham, MA, USA). The surface area of the samples was estimated by the TriStar II 3 flex BET model (micrometrics, Norcross, GA, USA). The light absorption properties of the samples were collected on a UV-4100 spectrophotometer (HITACHI, Tokyo, Japan). Photoluminescence (PL) measurements were carried out using a FLS1000 Spectrophotometer (Edinburgh Instruments Ltd., Edinburgh, UK) with an excitation wavelength of 325 nm, fluorescence scanning range is 340 nm to 800 nm, scanning speed is 240 nm/min.

2.3. Photocatalytic Activity Measurement

Total of 20 mg photocatalyst was added to 50 mL methyl blue (MB) or methyl orange (MO) solution, and then magnetically stirred in the dark for 30 min to ensure adsorption/desorption equilibrium was achieved. Then, a trace amount of 50 μ L H₂O₂ (30%) was added to the solution, and the photocatalytic degradation experiment was carried out at 26 °C under the illumination of a LED lamp ($\lambda = 400~900$ nm, 100 mW/cm²). Total of 2 mL aliquots were extracted from the solution and centrifuged at certain time intervals (5 to 20 min), and the absorption of the aliquots at 654 nm (maximum absorption wavelength of methylene blue) and 464 nm (maximum absorption wavelength of methylene orange) was measured on a UV-vis spectrophotometer. The standard curves were obtained by measuring standard solutions with different dye concentrations, from which the correlation between the absorption and dye concentration was determined. For the stability test, the MO solution was adjusted to the initial concentration and the separated photocatalyst was washed and reused after each cycle of the photocatalytic experiment. The degradation rate was calculated by Equation (1):

Degradation rate (%) =
$$1 - C_t / C_0$$
 (1)

where C_0 is the concentration of the initial dye solution, C_t is the remaining concentration of dye solution at a reaction time of t.

3. Results and Discussions

3.1. Crystal Structure, Morphology, and Composition

The XRD patterns of the as prepared CBO, BFO, and CBO/BFO composites were shown in Figure 1a. For CBO, diffraction peaks at 20.6°, 27.7°, 29.3°, 30.8°, 32.9°, 34.2°, 37.3°, 46.1°, 53.0°, 55.1°, 59.7°, and 65.8° were observed, which can be assigned to the (200), (211), (220), (002), (310), (112), (202), (411), (213), (332), (521), and (413) lattice planes of tetragonal CuBi₂O₄ (JCPDF 42-0334), respectively [9,20,22,33–35]. For the BFO, only one broad peak appeared at 32.07°, which might be attributed to the (110) peak of perovskite BiFeO₃ [36,37]. This suggested that the as prepared BiFeO₃ was mainly amorphous. The CBO/BFO with a CBO to BFO ratio of 1:2 showed a sharp peak at 27.7° in addition to a broad peak at 32.07°, which can be attributed to the (211) peak of tetragonal CuBi₂O₄ and the (110) peak of perovskite BiFeO₃. As the CBO to BFO ratio increased to 1:1 and 2:1, all the peaks of tetragonal CBO appeared, while no characteristic peak of BFO was presented. This suggested that CBO/BFO composites showed the main structure of CBO, and the BFO remained amorphous, this is why they were denoted as T-CBO/A-BFO. To investigate the effect of annealing on the crystal structure, the as-prepared samples were calcined in a muffle furnace at 450 °C for 2 h and the XRD patterns of the samples before and after annealing were displayed in Figure S1a,1b, respectively. The T-CBO

maintained the tetragonal structure after the calcination. For the A-BFO, the diffraction peaks at 22.42°, 31.75°, 32.07°, 39.48°, 45.75°, 51.31°, 51.74°, 56.97°, and 67.07° appeared after calcination, which can be assigned to the (012), (104), (110), (202), (024), (116), (122), (214), and (220) lattice planes of BiFeO₃ (JCPDF 86-1518), respectively [36,37]. In addition, the diffraction peaks at 28.2°, 31.6°, 33.1°, 44.1°, 46.8° , 49.1° , and 53.5° were obviously observed, which can be attributed to the (400), (420), (332), (532), (541), (444), and (721) lattice planes of Bi₂₅FeO₄₀ (JCPDF 86-0368), respectively [36]. It has been reported that pure phase BiFeO₃ was difficult to prepare via hydrothermal methods, and usually multiphase was obtained (e.g., Bi₂₅FeO₄₀, Bi₂Fe₄O₉) [36]. However, after high temperature calcination, the T-CBO/A-BFO composites showed CBO and BFO diffraction peaks without the appearance of Bi₂₅FeO₄₀ or Bi₂Fe₄O₉. This implied that rapid grown CBO could provide nucleation center for BFO. Raman spectroscopy was carried out to investigate the vibration modes of the samples, and the results are shown in Figure 1b. No obvious peak was found in the Raman spectra of A-BFO, which further confirmed that the as prepared A-BFO was amorphous as demonstrated in the XRD results. For T-CBO, three main Raman bands of tetragonal CuBi₂O₄ at 76, 121, and 256 cm⁻¹ were observed. The band at 76 cm⁻¹ corresponds to the B_{2g} mode of the in-plane bending vibration of the Bi rhombohedra. The strong peak at 123 cm⁻¹ can be assigned to the A_{1g} mode of the translational vibration of the CuO₄ plane along the Z-axis. The broad peak at 256 cm^{-1} is attribute to the rotation of two stacked CuO₄ squares in opposite directions [38-40]. The T-CBO/A-BFO (1:1) showed main peaks of CuBi₂O₄ at the same position as T-CBO, indicating that the T-CBO/A-BFO (1:1) maintained the main structure of T-CBO.



Figure 1. (a) XRD patterns of the T-CBO, A-BFO, and T-CBO/A-BFO composites with different composite ratios; (b) XRD patterns of the CuBi₂O₄, BiFeO₃, and CuBi₂O₄/BiFeO₃ composite photocatalyst with different composite ratios and after calcination.

SEM was performed to investigate the morphologies of T-CBO, A-BFO, and T-CBO/A-BFO composites, the images are shown in Figure 2. It can be observed that pure T-CBO photocatalysts (Figure S2) were formed by the self-assembling of nanorod arrays with a dumbbell-like morphology, more details of the T-CBO can be found in our previous report [9]. The A-BFO photocatalyst (Figure 2a) showed an irregular granular morphology with small particle size. Figure 2a–d showed the morphologies of T-CBO/A-BFO composites with different composite ratios. The T-CBO/A-BFO composites showed similar irregular granular morphology to A-BFO, and the particle size was obviously smaller than that of pure T-CBO.



Figure 2. SEM images of (a) A-BFO; (b) T-CBO/A-BFO (1:2); (c) T-CBO/A-BFO (1:1); (d) T-CBO/A-BFO (2:1).

The fine structure of T-CBO/A-BFO (1:1) composite was characterized by HRTEM. The nanoparticles of T-CBO/A-BFO (1:1) composites were observed in Figure 3a. The well-ordered lattice fringes were presented in Figure 3b, in which an interplanar spacing of 0.318 nm and 0.240 nm were observed, corresponding to the (211) and (202) plane of T-CBO, respectively [7,41,42]. No obvious A-BFO lattice fringes was found, which is consistent with the XRD and Raman results. This further confirmed the amorphous nature of BiFeO₃ in the T-CBO/A-BFO (1:1) composite. The microstructure and elemental distribution of the T-CBO/A-BFO (1:1) composites photocatalyst were further investigated by HAADF-STEM and energy dispersive-ray spectroscopy (EDS). In the mapping area (Figure 3c), the Cu, Fe, and Bi were uniformly distributed in the T-CBO/A-BFO (1:1) composite, as shown in Figure 3d–f.

The chemical composition and elemental valence of T-CBO, A-BFO, and T-CBO/A-BFO (1:1) composite were analyzed by XPS, and the results are shown in Figure 4. The XPS survey spectra for the Cu, Bi, Fe, and O are displayed in Figure S3. All the XPS spectra were calibrated by the binding energy of C 1s at 284.8 eV. As shown in Figure 4a, T-CBO and T-CBO/A-BFO (1:1) showed two main peaks at around 933.8 eV and 953.6 eV and shake-up satellites at approximately 941.7 eV and 961.8 eV, which could be assigned to Cu $2p_{3/2}$ and Cu $2p_{1/2}$ binding energies of Cu²⁺, respectively [38,42–44]. In the A-BFO, two peaks at 710.3 eV and 724.8 eV were observed (Figure 4b), corresponding to the Fe $2p_{3/2}$ and Fe $2p_{1/2}$ peaks of Fe³⁺. In addition, two satellite peaks at 716.4 eV and 729.4 eV appeared in the A-BFO, which can be assigned to the Fe $2p_{3/2}$ and Fe $2p_{1/2}$ satellite of Fe²⁺, respectively. This suggests the co-existence of Fe^{3+} and Fe^{2+} in the A-BFO. Whereas in the T-CBO/A-BFO (1:1) composite, the peaks at 710.7 eV and 724.2 eV correspond to Fe $2p_{3/2}$ and Fe $2p_{1/2}$ of Fe³⁺, respectively, and the small satellite peak at 718.4 eV further confirmed the existence of Fe^{3+} [7,37]. No satellite peaks of Fe²⁺ was found in the T-CBO/A-BFO (1:1) composite, suggesting different valence bond mechanisms of A-BFO when it was co-synthesized with the T-CBO. In Figure 4c, the peak for O-mental bounds appeared at 529.4, 529.6, and 529.8 for the A-BFO, T-CBO, and T-CBO/A-BFO (1:1) respectively. The positive shift in the T-CBO/A-BFO (1:1) was due to the increase in binding energy of O-mental in the heterojunction. The peaks at around 531.5 eV can be assigned to the chemisorbed oxygen [45,46], and the peak at 532.7 eV in the A-BFO suggested the formation of metal-OH groups on the surface [47]. In Figure 4d, the T-CBO, A-BFO, and T-CBO/A-BFO (1:1) all showed peaks at around 158.7 eV and 164.0 eV, which can be assigned to Bi 4f 7/2 and Bi 4f 5/2 of Bi³⁺, respectively. The positive shift in the

T-CBO/A-BFO (1:1) was attributed to the increasing bounding energy of Bi^{3+} with the surrounding O atoms due to the formation of the heterojunction. The A-BFO, in addition, had two peaks at 159.4 eV and 165.8 eV, which could be attributed to the Bi 4f 7/2 and Bi 4f 5/2 of Bi^{5+} [48]. This implied the co-existence of Bi^{3+} and Bi^{5+} in the A-BFO.



Figure 3. (**a**) TEM image; (**b**) HRTEM image; (**c**) TEM image of the selected mapping area and the corresponding elemental mapping of (**d**) Cu, (**e**) Fe, (**f**) Bi of the T-CBO/A-BFO (1:1). The arrows pointed to the interplanar spacing for different CBO planes.



Figure 4. XPS high-resolution spectra of (**a**) Cu 2p, (**b**) Fe 2p, (**c**) O 1s, and (**d**) Bi 4f for the T-CBO, A-BFO, and T-CBO/A-BFO (1:1). The arrow pointed to the curves of different samples, and specific XPS peaks in case of overlapping.

In general, photocatalysts with high specific surface area and large pore size are expected to provide more photocatalytic active sites, which is beneficial for improving the photocatalytic activity. To elucidate the effect of specific area, BET measurements were carried out. The N₂ adsorption-desorption curves with T-CBO, A-BFO, and T-CBO/A-BFO with different composite ratios were displayed in Figure 5. It can be seen that all the samples had isotherms of type IV, indicating the presence of mesopores (2–50 nm). The isotherms exhibited H3 hysteresis loops at a high relative pressure range from 0.8 to 1.0, suggesting the presence of slit-like pores.



Figure 5. N₂ adsorption-desorption isotherms of the T-CBO, A-BFO, T-CBO/A-BFO (2:1), T-CBO/A-BFO (1:1), and T-CBO/A-BFO (2:1) composites.

The BET surface area, pore volume, and average pore size of the samples are summarized in Table 1. The instrument for BET test (micrometrics, TriStar II 3 flex, THE UNITED STATES OF AMERICA) could extend specific surface area, pore volume, and average pore size measurements to as low as 0.001 m²/g, 4×10^{-6} cm³/g, and 0.01 nm, respectively. The uncertainties of the BET results were obtained by repeating the measurements three times, from which the standard deviation was calculated. The T-CBO and A-BFO showed a BET surface area of 11.60 m²/g and 80.39 m²/g, respectively. The surface area of the T-CBO/A-BFO composite was in between T-CBO and A-BFO, and increased from 38.95 m²/g to 74.18 m²/g as the T-CBO to A-BFO ratio decreased from 2:1 to 1:2. Specifically, the T-CBO/A-BFO (1:1) showed a surface area of 58.86, which is relatively higher compared to the reported values (7.2–28.1 m²/g) for BFO and its compounds [49,50].

Table 1. BET surface area, pore volume, and average pore size of the samples.

Samples	Т-СВО	T-CBO/A-BFO (2:1)	T-CBO/A-BFO (1:1)	T-CBO/A-BFO (1:2)	A-BFO
BET (m^2/g)	11.60 ± 0.39	38.95 ± 0.47	58.86 ± 0.48	74.18 ± 0.66	80.39 ± 0.63
Pore volume (cm ³ /g)	0.02 ± 0.01	0.19 ± 0.03	0.27 ± 0.08	0.41 ± 0.02	0.57 ± 0.01
Average pore size (nm)	6.82 ± 0.94	17.56 ± 2.25	21.78 ± 2.60	17.59 ± 1.19	28.22 ± 1.05

3.2. Optical and Electronic Properties

The optical absorption properties of the T-CBO, A-BFO, and T-CBO/A-BFO (1:1) were investigated by the UV-vis diffusive reflectance analyzer, and the spectra are shown in Figure 6a. UV-vis spectra for T-CBO/A-BFO composites with other T-CBO to A-BFO ratios were provided in Figure S4a, and the corresponding Tauc plots could be found in Figure S4b.



Figure 6. (**a**) UV-vis diffuse reflectance spectra; (**b**) XPS VB spectra of the T-CBO, A-BFO and T-CBO/ A-BFO (1:1); (**c**) schematic energy band diagram.

The A-BFO could effectively absorb light under 500 nm, while the T-CBO exhibited wide visible light absorption up to 800 nm. The T-CBO/A-BFO composites showed significant higher absorption and extended absorption region compared with the A-BFO. The bandgap of the T-CBO and A-BFO were estimated to be 1.75 eV, 1.95 eV from the Tauc plot (Figure S4b), and their valence band edge was determined to be 1.29 V and 0.95 V from the XPS VB spectra (Figure 6b). Based on the valence band positions and bandgaps of the T-CBO and A-BFO, we have tried to sketch an energy band diagram of the T-CBO/A-BFO composite, as shown in Figure 6c. The band edges of A-BFO are located slightly higher than those of T-CBO, such band alignment is beneficial for forming T-CBO/A-BFO heterojunction with improved charge separation efficiency. In the T-CBO/A-BFO heterojunction, the photo-generated electrons on the CB of A-BFO tend to move onto the CB of T-CBO, while the photo-generated holes on the VB of T-CBO can be effectively transferred to the VB of A-BFO.

To further investigate the carrier recombination and charge transfer properties, the T-CBO, A-BFO, and T-CBO/A-BFO (1:2), T-CBO/A-BFO (1:1), T-CBO/A-BFO (2:1) were evaluated by photoluminescence (PL) with an excitation wavelength of 325 nm and the results are shown in Figure 7. All the samples showed emission peak at about 420 nm. Both T-CBO and A-BFO exhibited strong emission peak, while a decrease in emission signal was observed in all the T-CBO/A-BFO composites, indicating inhibited carrier recombination in the composites. This confirmed that the formation of T-CBO/A-BFO heterojunction benefited the charge transfer between the T-CBO and A-BFO, as has been demonstrated in Figure 6c. The PL emission intensity decreased significantly as the T-CBO to A-BFO ratio increased from 1:2 to 1:1, further increasing the ratio to 2:1 resulted in a sharp increase in PL emission. This implied that

recombination of photo-generated electron-hole pairs was effectively prohibited in the T-CBO/A-BFO composites and the optimal T-CBO to A-BFO ratio is 1:1.



Figure 7. PL spectra of T-CBO, A-BFO, T-CBO/A-BFO (1:2), T-CBO/A-BFO (1:1), T-CBO/A-BFO (2:1).

3.3. Photocatalytic Properties

The photocatalytic performance of the as-prepared samples was evaluated for the degradation of MB and MO under visible light. As shown in Figure 8, the MO concentration obviously decreased under dark, especially for the dye solution with A-BFO added as the photocatalyst. This is probably due to the strong physical absorption of dye on the surface of BiFeO₃, which has been reported previously [51]. Among all our samples, A-BFO showed the highest BET surface area (80.39 m²/g, see Table 1). Previous study demonstrated that larger surface area could lead to higher absorption of dye in the dark [52]. After 30 min of visible light irradiation, the degradation rates of MB by the T-CBO and A-BFO photocatalysts were 56% and 19%, respectively. Obviously, the T-CBO/A-BFO composites showed increased photocatalytic property than the T-CBO and A-BFO alone. Based on the aforementioned results, we tried to elucidate the correlation between the crystal structure, specific surface area, light absorption property, carrier transfer of the samples with their photocatalytic properties. To investigate the effect of crystal structure, the as-prepared samples were calcined in a muffle furnace at 450 °C for 2 h and the photocatalytic property was tested. Tetragonal CBO and perovskite BFO were obtained after calcination (Figure S1b), which were commonly regarded as the photoactive phases. However, in our results the well crystalized tetragonal-CBO/perovskite-BFO composite after calcination showed substantially lower photoactivity than the as-prepared T-CBO/A-BFO (Figure S5), indicating that better crystallinity did not lead to higher photoactivity. With respect to the surface area, in our study the A-BFO processed the highest BET surface area (Table 1), but it exhibited the lowest photoactivity (Figure 8a). Therefore, specific surface area could not be the main determinant for the photocatalytic performance. Herein, the improved photoactivity of the CBO/A-BFO composite was mainly attributed to the extended visible light absorption (Figure 6a) and improved charge transfer of the T-CBO/A-BFO heterojunction (Figure 6). The photoactivity of the CBO/A-BFO composite increased as the T-CBO to A-BFO ratio increases from 1:2 to 1:1. Further increasing the ratio from 1:1 to 2:1 resulted in a decrease in photoactivity, which is probably due to the sharp increase in carrier recombination as demonstrated in Figure 7. The same trend in photoactivity was also found for the degradation of MO, as shown in Figure 8b. Among the T-CBO/A-BFO composites with various T-CBO to A-BFO ratios, the T-CBO/A-BFO (1:1) composite exhibited the best photocatalytic performance and it could photodegrade 97% MB within 30 min and 80% MO within 80 min. An interesting observation about the T-CBO/A-BFO 1:2 and 2:1 was that they showed PL signal almost as high as the T-CBO or A-BFO alone, whereas their photocatalytic activity were close to that of T-CBO/A-BFO 1:1. This suggested that other factors (e.g., light absorption, surface area) played an important role in the photocatalytic activity of the T-CBO/A-BFO 1:2 and 2:1, even though they might not be the main determinants. Specifically, T-CBO/A-BFO 2:1 possessed considerably higher visible light absorption in the range of 520–800 nm compared to T-CBO/A-BFO 1:1, as shown in Figure S4a. T-CBO/A-BFO 1:2 exhibited much higher specific surface area (74.18 m²/g) than the T-CBO/A-BFO 1:1 (58.86 m²/g).



Figure 8. photocatalytic degradation of (**a**) methylene blue; (**b**) methyl orange. C_0 is the concentration of the initial dye solution, C_t is the remaining concentration of dye solution at a reaction time of t, C_t/C_0 is the residual ratio of the dye in the solution, the degradation rate equals $1 - C_t/C_0$.

Moreover, we also prepared T-CBO and A-BFO composite by simply physically mixing T-CBO and A-BFO at a ratio of 1:1, and the photocatalytic performance was tested (Figure S6). The experimental results showed that the degradation rate of MO by the physically mixed T-CBO/A-BFO was slightly higher than that of T-CBO or A-BFO alone, but still much lower than that of the one-step synthesized T-CBO/A-BFO (1:1) composite. This indicated that a solid heterojunction T-CBO/A-BFO (1:1) was formed during the solvothermal reaction, leading to improved charge carrier separation and enhanced photocatalytic performance.

3.4. Stability of the T-CBO/A-BFO (1:1) Composite

Eventually, the stability of the T-CBO/A-BFO (1:1) composite was evaluated by repeating photodegradation experiments under visible light irradiation for 5 cycles. As illustrated in Figure 9a, after five cycles of experiments, the decomposition rate of MO by the T-CBO/A-BFO (1:1) slightly decreased by 8.4%, demonstrating the relatively high stability of the as-prepared T-CBO/A-BFO (1:1) composite photocatalyst. We note that the decrease may be partially due to the loss of the sample powder during the separation and cleaning processes. XPS of Fe, Cu, and Bi was performed before and after the stability test, as shown in Figure 9b–d. No significant change was found for Fe 2p (Figure 9b) and Cu 2p (Figure 9c) before and after the stability test. However, in the Bi 4f spectra, two new peaks at lower binding energy of 157.7 and 163.0 appeared after the cycle experiments, suggesting the reduction of Bi³⁺ into lower valence states such as Bi^{2.75} [53], Bi²⁺ [54] during the photocatalytic test. Previously we have demonstrated that the T-CBO was stable after 5 cycles photodegradation test [9], therefore the reduction of Bi³⁺ was presumably taking place in the A-BFO of the T-CBO/A-BFO composite. This may be part of the reason for the decreased photocatalytic property in the cycle experiment. Our results highlight the importance of preventing Bi³⁺ from reduction to further improve the stability of the T-CBO/A-BFO composite.



Figure 9. (a) Five-cycle test of T-CBO/A-BFO (1:1) composite photocatalyst for the degradation of methyl orange; (b) Fe 2p, (c) Cu 2p, and (d) Bi 4f XPS spectra of the T-CBO/A-BFO (1:1) composite before and after the stability test. The arrows pointed to the curves of pristine and used samples and specific XPS peaks in case of overlapping.

4. Conclusions

In this study, a T-CBO/A-BFO composite was prepared using a facile one-step method. A 30-min solvothermal treatment at 120 °C resulted in T-CBO/A-BFO composites with extended visible light absorption region, lower charge recombination and consequently higher photocatalytic performance compared to T-CBO or A-BFO alone. The photoluminance intensity of the CBO/A-BFO composite decreased as the T-CBO to A-BFO ratio increases from 1:2 to 1:1. Further increasing the ratio from 1:1 to 2:1 resulted in a sharp increase in photoluminance intensity. The CBO/A-BFO (1:1) composite showed the lowest photoluminance intensity, indicating suppressed carrier recombination and improved the charge transfer in the heterojunction. The CBO/A-BFO (1:1) composite had a high BET surface area of 58.86 m²/g, and exhibited wide visible light absorption up to 800 nm. It could degrade 95.6% MB within 15 min, and 78.3% MO within 1 h under visible light illumination. The reduction of Bi³⁺ was observed after the photocatalytic stability test. The CBO/A-BFO (1:1) composite prepared by the one-step method showed significantly higher photocatalytic activity than the physically mixed T-CBO and A-BFO with a ratio of 1:1, which demonstrates the advantage of the one-step synthesized T-CBO/A-BFO heterojunction. Moreover, the as prepared CBO/A-BFO (1:1) composite exhibited substantially higher photocatalytic performance than its post annealed counterpart. The short-time synthesis method was carried out at mild reaction conditions and high temperature calcination was not required. This work provides an easy and cost-effective route for the fast preparation of multinary metal oxide-based heterojunctions for photocatalytic applications.

Supplementary Materials: The following are available online at http://www.mdpi.com/2079-4991/10/8/1514/s1. Figure S1: XRD patterns of the T-CBO, A-BFO, and T-CBO/A-BFO composites (**a**) before annealing; (**b**) after annealing at 450 °C for 2 h. Figure S2: SEM images of the T-CBO. Figure S3: XPS survey spectra of Cu 2p, Fe 2p, O 1s, Bi 4f of the T-CBO, A-BFO and T-CBO/A-BFO (1:1). Figure S4: (**a**) UV-vis diffuse reflectance spectra; (**b**) corresponding Tauc plots of the T-CBO, A-BFO, and T-CBO/A-BFO composites. Figure S5: Photodegradation of MB by post annealed CBO, BFO, and CBO/BFO composites. Figure S6: Photodegradation of MO by T-CBO/A-BFO (1:1) and physical mixed T-CBO and A-BFO at a ratio of 1:1.

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