

# SCIENTIFIC REPORTS



OPEN

## Understanding the structural transformation, stability of medium-sized neutral and charged silicon clusters

Received: 20 July 2015

Accepted: 02 October 2015

Published: 03 November 2015

Li Ping Ding<sup>1</sup>, Fang Hui Zhang<sup>1</sup>, Yong Sheng Zhu<sup>2</sup>, Cheng Lu<sup>2,4</sup>, Xiao Yu Kuang<sup>3</sup>, Jian Lv<sup>4</sup> & Peng Shao<sup>1</sup>

The structural and electronic properties for the global minimum structures of medium-sized neutral, anionic and cationic  $\text{Si}_n^\mu$  ( $n=20-30$ ,  $\mu=0, -1$  and  $+1$ ) clusters have been studied using an unbiased CALYPSO structure searching method in conjunction with first-principles calculations. A large number of low-lying isomers are optimized at the B3PW91/6-311 + G\* level of theory. Harmonic vibrational analysis has been performed to assure that the optimized geometries are stable. The growth behaviors clearly indicate that a structural transition from the prolate to spherical-like geometries occurs at  $n=26$  for neutral silicon clusters,  $n=27$  for anions and  $n=25$  for cations. These results are in good agreement with the available experimental and theoretical predicted findings. In addition, no significant structural differences are observed between the neutral and cation charged silicon clusters with  $n=20-24$ , both of them favor prolate structures. The HOMO-LUMO gaps and vertical ionization potential patterns indicate that  $\text{Si}_{22}$  is the most chemical stable cluster, and its dynamical stability is deeply discussed by the vibrational spectra calculations.

The experimental and theoretical studies of the atomic and molecular clusters are interesting topics since they constitute intermediate phases between individual atoms and bulk solids, which can be used to understand how the fundamental properties of materials evolve from isolated atoms or small molecules to a bulk phase<sup>1-8</sup>. The study of small clusters can help us to design better nanosystems with specific physical and chemical properties. Silicon is the most widely used material in the microelectronic industry. If current miniaturization trends continue, minimum device features will soon approach the size of atomic clusters. In this size regime, the structures and properties of materials often differ dramatically from those of the bulk. The study of the structures and properties of silicon clusters has been an extremely active area of current research. During the past two decades, a large number of experimental and theoretical studies have been carried out in this direction<sup>4,6-10</sup>. Much attention has been focused on understanding the structural and growth behavior of small or medium-sized silicon clusters<sup>4,10-14</sup>.

Several high-resolution photoelectron, Raman and infrared spectra experiments have been carried out to understand the atomic structure of small silicon clusters and showed that both  $\text{Si}_6$  and  $\text{Si}_{10}$  have exceptional stability<sup>11</sup>. Ion mobility measurements have revealed much of what is known about the growth behaviors of medium-sized silicon clusters<sup>14</sup>. Jarrold *et al.* have determined that anionic silicon clusters are prolate shape for  $n < 27$  and become more spherical-like geometry for larger clusters<sup>11</sup>. However, the transition from prolate to more spherical-like geometries for cationic silicon clusters was observed

<sup>1</sup>Department of Optoelectronic Science & Technology, College of Science, Shaanxi University of Science & Technology, Xian, 710021, China. <sup>2</sup>Department of Physics, Nanyang Normal University, Nanyang, 473061, China. <sup>3</sup>Institute of Atomic and Molecular Physics, Sichuan University, Chengdu, 610065, China. <sup>4</sup>Beijing Computational Science Research Center, Beijing, 100084, China. Correspondence and requests for materials should be addressed to C.L. (email: lucheng@calypso.cn) or X.Y.K. (email: scu\_kuang@163.com) or J.L. (email: lvjian@calypso.cn)

in between  $24 < n < 30^{11-14}$ . Up to now, most of spherical and compact clusters have been considered as theoretical models attempting to support this measurement. A lot of works have been carried out with results not always in agreement between authors<sup>14-20</sup>. Despite the enormous progress that has been made, the true lowest-energy structures for the silicon clusters in the size range of  $20 \leq n \leq 30$  are still debatable. The main reasons may be as follows: (i) The procedure used in the case of small clusters is not practical for larger clusters. (ii) The predicted global minima are subtle sensitivity for the selected density functional theory, or the molecular-orbital level in the *ab initio* calculations. Moreover, the determination of the true global minimum structure is also a challenging problem, because of the much increased complexity of the potential surface as well as the exponential increase of the lowest-energy structure with the number of atoms in the cluster<sup>21</sup>.

In order to systematically study the structural evolution and electronic properties of silicon clusters, we here present extensive structure searches to explore the global minimum geometric structures of medium-sized neutral and charged silicon clusters in the size range of  $20 \leq n \leq 30$ , by combining our developed CALYPSO method with the density functional theory. Our first goal of this work is to gain a fundamental understanding of the ground state geometric structures in medium-sized silicon clusters. The second one is to reexamine a number of neutral and charged low-energy isomers of  $\text{Si}_{20}\text{-Si}_{30}$  that have been reported previously by experiments or density functional calculations. Thirdly, we are also motivated to explore the physical mechanism of the growth behaviors of medium-sized silicon clusters and provide relevant information for further theoretical and experimental studies. The paper is organized as follows: the computational details are described in Section 2, results are presented and discussed in Section 3, and our final conclusions are given in Section 4.

### Computational Methods

Our cluster structure prediction is based on the CALYPSO method<sup>22-24</sup>. A global version of particle swarm optimization (PSO) algorithm<sup>25</sup> is implemented to utilize a fine exploration of potential energy surface for a given non-periodic system. The bond characterization matrix (BCM) technique is employed to eliminate similar structures and define the desirable local search spaces. This structure prediction method has been benchmarked on LJ clusters with sizes up to 150 atoms. High search efficiency is achieved, demonstrating the reliability of the current method. The significant feature of this method is the capability of predicting the stable structure with only the knowledge of the chemical composition. It has been successful in correctly predicting structures for various systems<sup>24-27</sup>. The evolutionary variable structure predictions of the neutral and charged  $\text{Si}_{20-30}$  clusters are performed. To seek low-lying structures, the computational process can be divided into two steps. Firstly, an unbiased global search is performed, using the CALYPSO method combined with density functional theory geometric optimization. Each generation contain 30 structures, 60% of which are generated by PSO. The others are new and will be generated randomly. We follow 50 generations to achieve the converged structure. Next, among the 1000–1500 isomers for the neutral and charged  $\text{Si}_n$  clusters, the top fifteen low-lying isomers are collected as candidates for the lowest-energy structure. Those isomers with energy difference from the lowest-lying isomer less than 3 eV are further optimized at B3PW91/6-311 + G\* level of functional/basis set. All the quantum chemical calculations are performed using the Gaussian 09 program package<sup>28</sup>. The convergence thresholds of the maximum force, root-mean-square (RMS) force, maximum displacement of atoms, and RMS displacement are set to 0.00045, 0.0003, 0.0018, and 0.0012 a.u., respectively. The effect of the spin multiplicity is also taken into account in the geometric optimization procedure. Meanwhile, the vibrational frequency calculations are performed at the same level of theory to make sure that the structures correspond to real local minima without imaginary frequency.

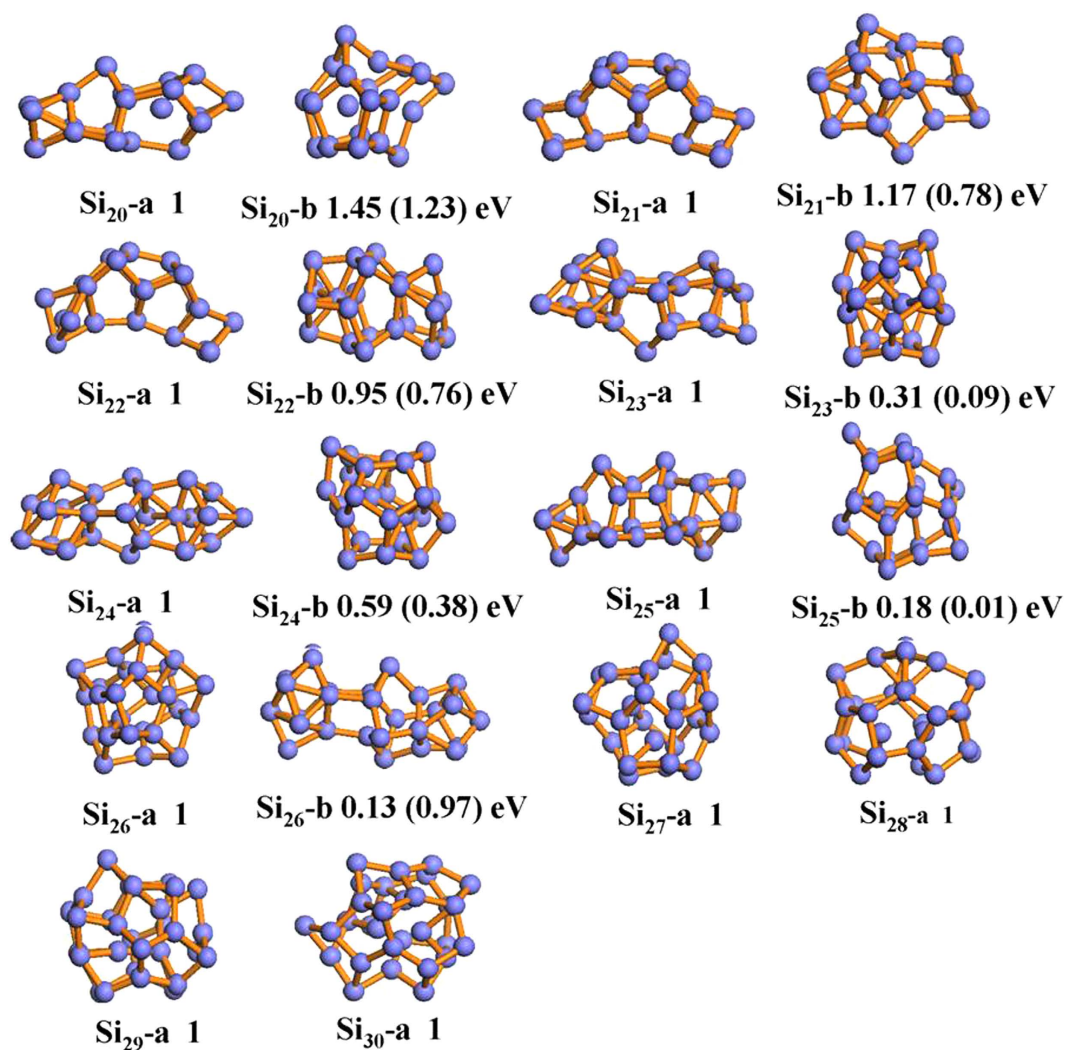
To verify the reliability of our calculations, we have calculated the neutral and charged silicon dimers ( $\text{Si}_2$ ,  $\text{Si}_2^-$  and  $\text{Si}_2^+$ ) through many different functionals (HF<sup>29</sup>, MP2<sup>30</sup>, B3LYP<sup>31,32</sup>, PW91<sup>31,33</sup>, PBE<sup>34</sup>, B3PW91<sup>31,33,35</sup>, B3P86<sup>32</sup> as well as CCSD(T)<sup>36</sup>) with 6-311 + G\* basis sets. The calculated results are summarized in Table 1. From Table 1, it is found that the results of bond length ( $r$ ) and vibrational frequency ( $\omega_e$ ) of the three-type silicon dimers based on both B3PW91 and CCSD(T) methods are in good agreement with the experimental values<sup>37-40</sup>. While the calculated dissociation energy ( $D_e$ ) of neutral  $\text{Si}_2$ , adiabatic electronic affinity energy (AEA) of anionic  $\text{Si}_2^-$  and adiabatic ionization potential (AIP) of cation  $\text{Si}_2^+$  at B3PW91 level of theory are closer to the experimental values<sup>37,40</sup>, with deviation less than 2%, 5% and 3%, respectively. So, the B3PW91/6-311 + G\* has been selected as the reasonable method for silicon clusters.

### Results and Discussion

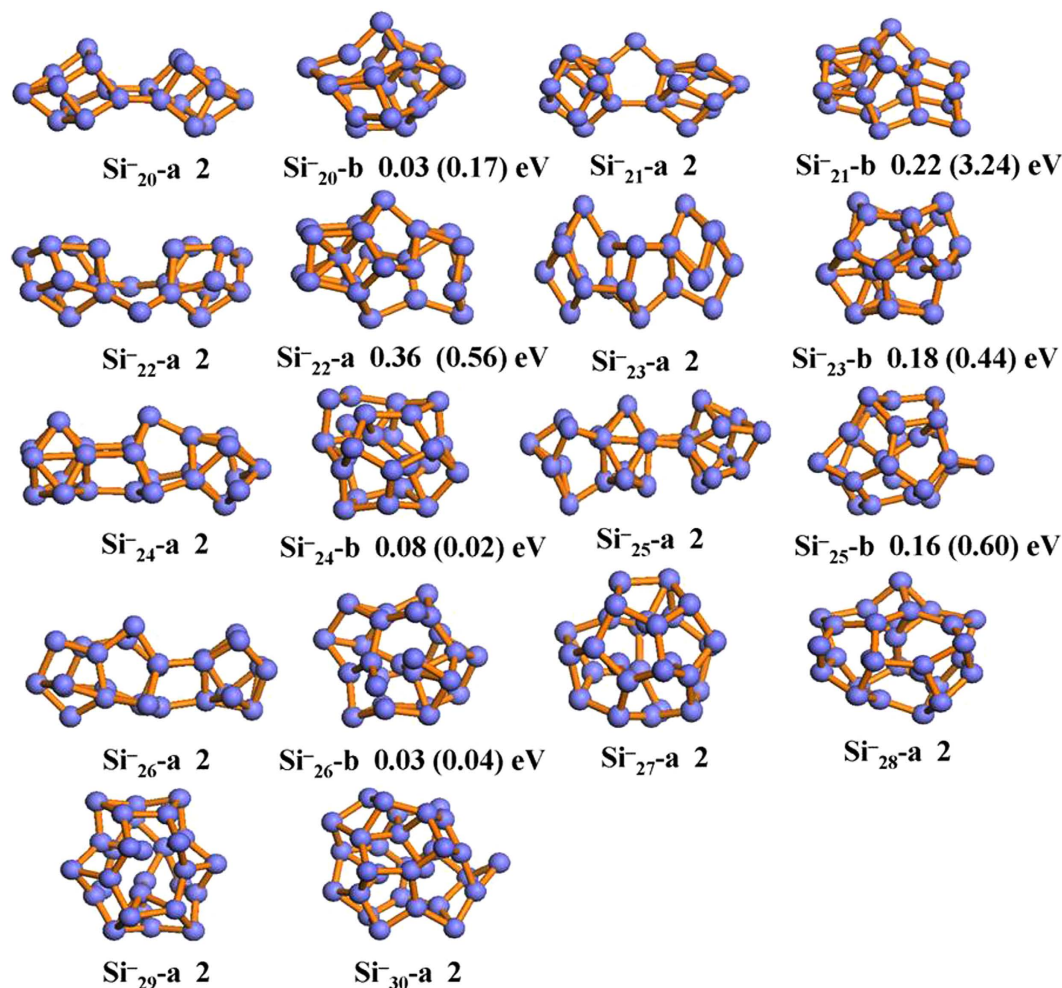
The structures found by CALYPSO searches in the range from 20 to 30 can be categorized into two kinds: prolate and spherical-like structures. All earlier known structures, experimentally and theoretically, were successfully reproduced and optimized in our current structure searches. Here, we only selected the lowest energy structures and the second low-lying isomers for neutral, anionic and cationic species and displayed them in Figs 1, 2 and 3, respectively. The other low-lying isomers of the three-type silicon clusters together with their relative energies are presented in Figures S1, S2 and S3 (see Electronic Supplementary Information). To further confirm the reliability of the present computational method, the vertical detachment energies (VDEs), adiabatic detachment energies (ADEs) and vertical ionization energies (VIPs) for large anionic and neutral silicon clusters are also calculated. The theoretical results as

Method	clusters								
	Si <sub>2</sub>			Si <sub>2</sub> <sup>-</sup>			Si <sub>2</sub> <sup>+</sup>		
	r	ω	D <sub>e</sub>	r	ω	AEA	r	ω	AIP
HF	2.22	562.52	1.52	2.16	604.99	1.04	2.26	489.60	7.18
MP2	2.25	546.29	2.61	2.20	531.86	1.86	2.26	482.82	7.39
B3LYP	2.28	485.51	3.06	2.20	536.68	2.09	2.30	434.23	7.92
PW91	2.30	468.72	3.38	2.21	519.91	2.14	2.31	426.29	7.87
PBE	2.30	468.76	3.37	2.22	519.63	2.11	2.32	426.98	7.83
B3P86	2.27	498.51	3.24	2.19	550.71	2.65	2.28	451.55	8.64
<b>B3PW91</b>	<b>2.27</b>	<b>497.09</b>	<b>3.13</b>	<b>2.19</b>	<b>549.99</b>	<b>2.10</b>	<b>2.29</b>	<b>450.16</b>	<b>7.94</b>
CCSD(T)	2.26	496.93	2.52	2.20	535.97	1.70	2.28	461.55	7.42
Exp.	2.25 <sup>a</sup>	509 ± 10 <sup>a</sup>	3.21 <sup>b</sup>	2.13 <sup>a</sup>	528 ± 10 <sup>a</sup>	2.17 <sup>a</sup>	2.27 <sup>c</sup>	471.8 <sup>c</sup>	7.92 <sup>d</sup>

**Table 1.** The computed values of bond length  $r$  (Å) and frequency  $\omega$  (cm<sup>-1</sup>) for the Si<sub>2</sub>, Si<sub>2</sub><sup>-</sup> and Si<sub>2</sub><sup>+</sup> dimers at different levels together with their corresponding experimental values. In addition, the dissociation energy D<sub>e</sub> (eV) for Si<sub>2</sub>, adiabatic electron affinity AEA (eV) for Si<sub>2</sub><sup>-</sup> and adiabatic ionization potential AIP (eV) for Si<sub>2</sub><sup>+</sup> are also calculated. <sup>a</sup>ref. 37. <sup>b</sup>ref. 38. <sup>c</sup>ref. 39. <sup>d</sup>ref. 40.



**Figure 1.** The optimized structures of neutral isomers for Si<sub>n</sub> (n = 20–30) cluster at B3PW91/6-311 + G\* level. The relative energies are also listed. The calculated results based on the PBE/PBE/6-311 + G\* level are shown in the parenthesis.

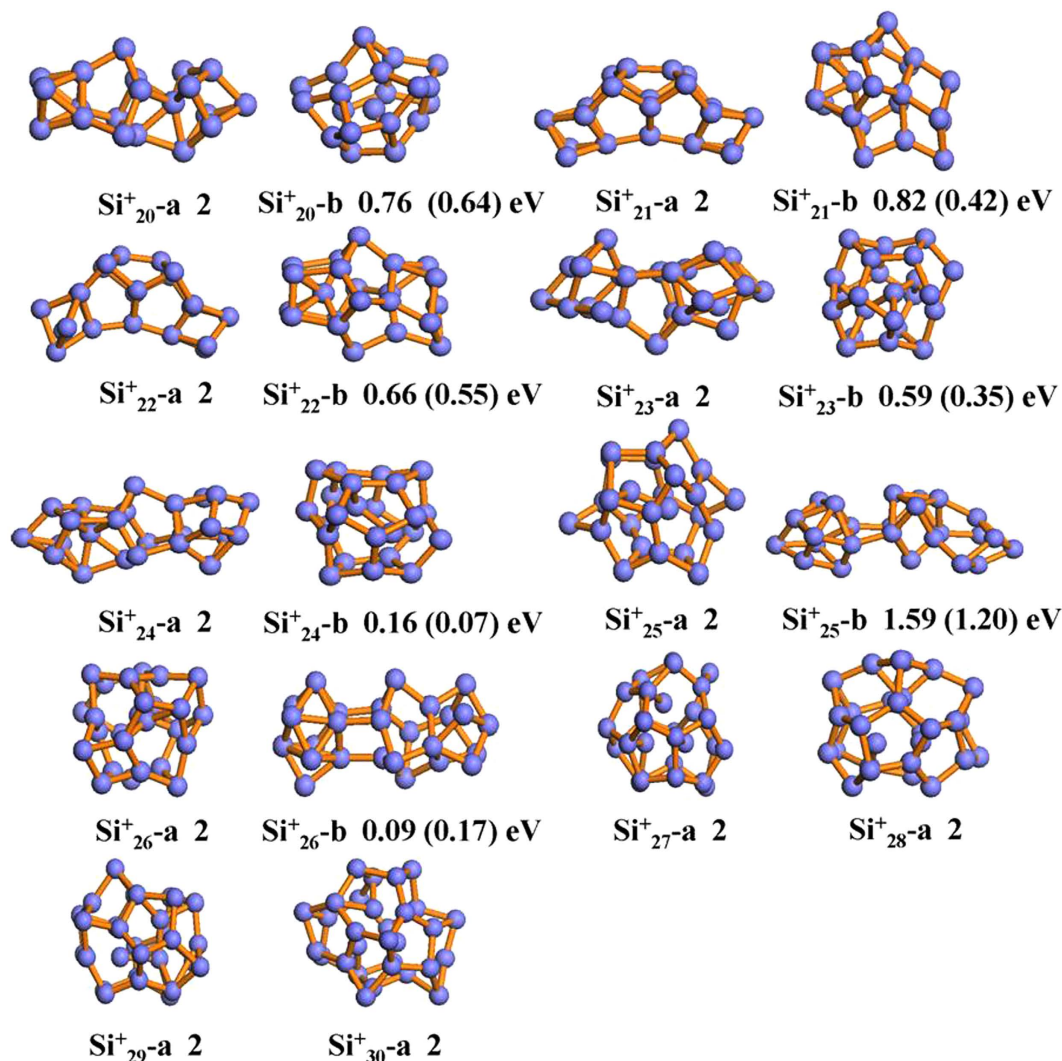


**Figure 2.** The optimized structures of anionic isomers for  $\text{Si}_n^-$  ( $n=20-30$ ) cluster at B3PW91/6-311 +  $G^*$  level. The relative energies are also listed. The calculated results based on the PBE/PBE/6-311 +  $G^*$  level are shown in the parenthesis.

well as the experimental data are listed in Table 2<sup>41</sup>. The agreement between the experimental data and the calculated results is also excellent, which shows the accuracy of the present theoretical calculations.

**Geometric structures.** In order to gain information on the growth of silicon clusters, many attempts have also been made to study the geometries of low-lying medium-sized neutral and charged silicon clusters. These include injected-ion drift-tube techniques, photoelectron spectroscopy measurements and *ab initio* calculations<sup>11,18,42-45</sup>. For neutral silicon clusters, our results indicate that the prolate structures are more stable than spherical-like structures for  $\text{Si}_n$  ( $20 \leq n \leq 25$ ) clusters, then a structural transition to more spherical-like structure occurs at  $n = 26$ . These prolate structures can be described as stacks of stable subunits. Take the  $\text{Si}_{20}$  and  $\text{Si}_{21}$  clusters for example, their structures consist of  $\text{Si}_6$  unit joined by other atoms. Our result on structural transition is in agreement with the cationic mobility experiment<sup>46</sup>, which has shown that a structural transition from prolate to more spherical-like structures may occur in between  $24 < n < 34$ . In addition, the previous theoretical studies on the medium-sized silicon clusters<sup>6,47-48</sup> also indicated that the prolate structures are more favorable for  $\text{Si}_n$  ( $n = 20-26$ ) clusters. In other words, the spherical-like isomers are expected to become more competitive energetically than the prolate isomers for larger  $\text{Si}_n$  ( $n \geq 27$ ) clusters.

Although considerable studies have been carried out for the neutral silicon clusters, only a few studies are available for charged clusters<sup>11,41,48</sup>. For anionic silicon clusters, we have examined a number of low-energy isomers which are obtained by our structural searches. Interestingly, a clear qualitative change in the geometry of these isomers is found except the silicon clusters with  $n \geq 27$ . Further geometrical optimization for the final structures confirmed that the prolate  $\text{Si}_n^-$  isomers becomes slightly more stable than spherical-like isomers and a structural transition from prolate to more spherical-like geometries occurs at  $n = 27$ . This observation is in complete agreement with the photoelectron spectroscopy experiments and first-principles density-functional studies by Bai *et al.*<sup>40</sup>. In order to gain insight into



**Figure 3.** The optimized structures of cationic isomers for  $\text{Si}_n^+$  ( $n=20-30$ ) cluster at B3PW91/6-311 +  $G^*$  level. The relative energies are also listed. The calculated results based on the PBE/PBE/6-311 +  $G^*$  level are shown in the parenthesis.

the electronic properties of the medium-sized charged silicon clusters, the vertical detachment energies (VDEs) and adiabatic detachment energies (ADEs) of  $\text{Si}_n^-$  ( $n=20-30$ ) are calculated. The theoretical results are listed in Table 2 together with available experimental values<sup>41</sup>. It can be seen from Table 2 that the calculated VDE values of  $\text{Si}_n^-$  ( $n=20-30$ ) clusters are in good agreement with experimental values, with discrepancy in the range of 0.3% to 2.7%. In addition, we also simulated the photoelectron spectra of  $\text{Si}_n^-$  ( $n=20-30$ ) clusters and compared with the experiments<sup>49</sup>. The simulated results together with experimental photoelectron spectra are shown in Figure S4 of supplementary information. It can be seen from Figure S4, the positions and the general shape of the peaks overall agree well with experimental results. These results further give us confidence in the obtained ground-state structures for these anionic clusters. However, there is no any available experimental data to compare with our obtained ADE results for  $\text{Si}_n^-$  ( $n=20-30$ ) clusters. We hope that our results for  $\text{Si}_n^-$  ( $n=20-30$ ) clusters would provide more information for further investigation in the future.

Previous mobility measurement<sup>45</sup> has been carried out for  $\text{Si}_n^+$  ( $n=20-27$ ), which can provide information on the general shape and initial geometry of clusters. This measurement result shows that the cationic silicon clusters become spherical-like structures occurring in between  $24 < n < 34$ . Based on the unbiased global search, the prolate structures (as shown in Fig. 3) are tested to be the ground state structures for  $\text{Si}_n^+$  ( $n=20-24$ ) clusters. This result mirror well the shape transition observed in mobility measurement. In addition, the theoretical study<sup>48</sup> on  $\text{Si}_n^+$  ( $n=20-27$ ) clusters also reveals that compact  $\text{Si}_n^+$  structures lie above the prolate for  $n \leq 23$ , closely compete with them for  $n=24$  and 25, and overtake them for  $n \geq 26$  in energy. It worth mentioning that no significant difference is observed between the neutral and cation charged silicon clusters with  $n=20-24$  (see Figs 1 and 3). For  $\text{Si}_n^+$  ( $n=25-30$ ) clusters, the spherical-like structures are more stable than prolate structures. These spherical-like geometries

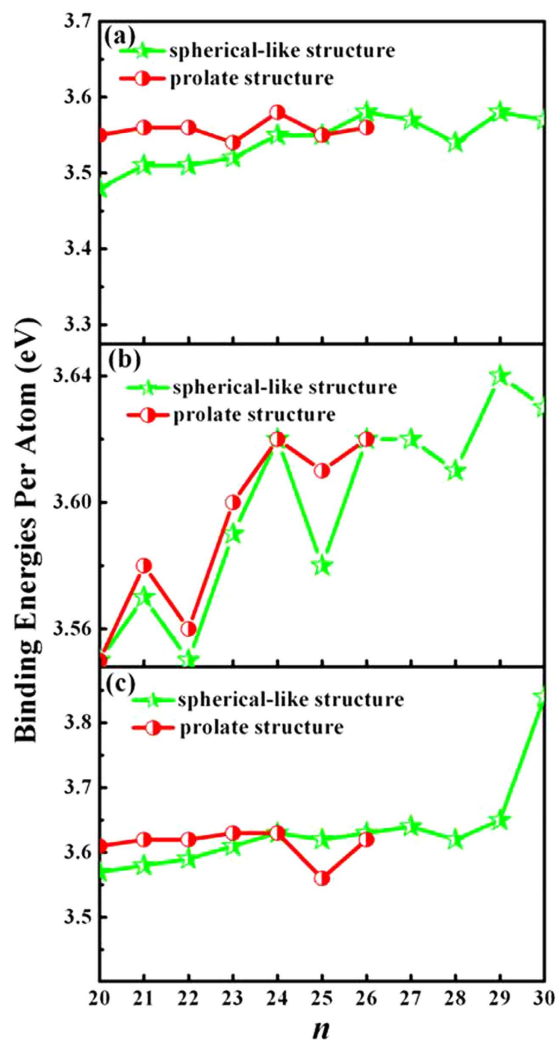
Cluster	VDE			ADE	Cluster	VIP	AIP	
	This work	Exp. <sup>a</sup>	Theor. <sup>a</sup>				This work	Exp. <sup>b</sup>
Si <sub>20</sub> <sup>-</sup>	3.65	3.57	3.587	1.52	Si <sub>20</sub>	7.06	7.05	7.46–7.53
Si <sub>21</sub> <sup>-</sup>	3.60	3.57	3.564	1.91	Si <sub>21</sub>	7.07	6.98	6.80–6.94
Si <sub>22</sub> <sup>-</sup>	3.43	3.37	3.299	1.16	Si <sub>22</sub>	7.28	6.85	5.85–5.95
Si <sub>23</sub> <sup>-</sup>	3.27	3.26	3.194	2.48	Si <sub>23</sub>	6.60	6.21	5.95–6.05
Si <sub>24</sub> <sup>-</sup>	3.56	3.66	3.597	2.43	Si <sub>24</sub>	7.03	6.81	5.95–6.05
Si <sub>25</sub> <sup>-</sup>	3.24	3.21	3.206	2.02	Si <sub>25</sub>	6.89	6.59	5.95–6.05
Si <sub>26</sub> <sup>-</sup>	3.27	3.34	3.311	3.17	Si <sub>26</sub>	6.74	6.40	5.90–5.95
Si <sub>27</sub> <sup>-</sup>	3.31	3.41	3.142	2.96	Si <sub>27</sub>	6.50	6.29	5.80–5.90
Si <sub>28</sub> <sup>-</sup>	3.43	3.38	3.271	3.20	Si <sub>28</sub>	6.56	6.38	5.80–5.90
Si <sub>29</sub> <sup>-</sup>	3.40	3.41	3.402	3.25	Si <sub>29</sub>	6.28	6.10	5.8
Si <sub>30</sub> <sup>-</sup>	3.51	3.46	3.613	3.27	Si <sub>30</sub>	6.61	6.40	5.70–5.80

**Table 2.** The theoretical calculated vertical detachment energy (VDE), compared to the experimental values and the previous calculated results at PBE/PBE/6-311 + G\* level for anions. The adiabatic detachment energy (ADE) for anions, as well as vertical and adiabatic ionization potential (VIP and AIP) for neutral cluster are also collected. All energies are in units of eV. <sup>a</sup>ref. 41. <sup>b</sup>ref. 18.

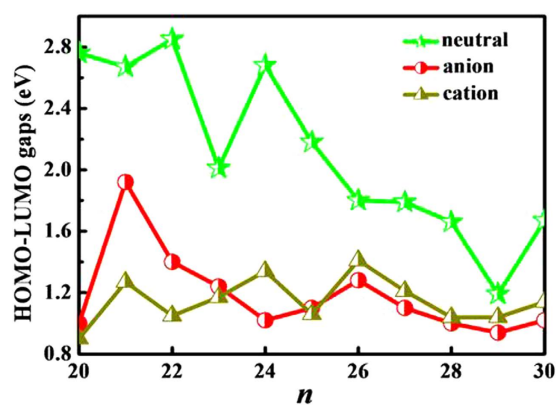
have neither the diamond-like packing of bulk silicon, nor the stuffed fullerene structure with an outer shell of pentagons and hexagons. For example, we find that the compact structure of Si<sub>22</sub><sup>+</sup> includes the tricapped trigonal prism (TTP) Si<sub>9</sub> units which are believed to appear in the prolate structures.

Considering the structural transition point from prolate to more spherical-like geometries may relate to the chosen functional, here we re-optimized the most stable prolate structures and the next low-lying isomer with spherical-like structures for three type (neutral, anion and cation) silicon clusters at the PBE level of theory. The calculated relative energies between the most stable prolate structure and the low-lying spherical-like structure at PBE level are given in parenthesis in Figs 1, 2 and 3. It can be clearly seen that although the prolate and spherical-like structures of Si<sub>23,25</sub> are almost degenerated in energy at PBE level of theory, the lowest energy structures remain unchanged for neutral, anionic and cationic silicon clusters. This suggests that the ground state structures of neutral and charged silicon clusters are independence with the used functional. Interestingly, we have also found that all the lowest-energy structures favor the low spin state.

**Relative stability.** It is well-known that the binding energy ( $E_b$ ) of a given cluster is a measure of its thermodynamic stability. It is defined as the difference between the energy sum of all the free atoms constituting the cluster and the total energy of the cluster. The binding energies per atom for medium-sized neutral, anionic and cationic silicon clusters ( $n = 20–30$ ) are summarized in Table S1 of the supplementary material. Meanwhile, the binding energies as a function of cluster size  $n$  are plotted in Fig. 4(a–c), respectively. As is shown in Fig. 4, all the  $E_b$  values are not obviously lower than that of the silicon crystal (4.75 eV)<sup>50</sup>. In addition, the binding energies do not show a dependent behavior on the cluster size, which is in agreement with the experimental reports<sup>16,17</sup>. This shows that the structures of silicon clusters ( $n = 20–30$ ) have different growth pattern. It can be seen from Fig. 4(a) that the binding energies of prolate structures are larger than those of spherical-like types for Si <sub>$n$</sub>  ( $n = 20–25$ ), indicating that the prolate structure become more competitive energetically than the near-spherical isomers. The prolate structure of Si<sub>25</sub> cluster is almost as stable as the compact structure. Furthermore, it is found that the binding energies per atom for studied silicon clusters irrespective of prolate and spherical-like structures change in a narrow region of 3.48–3.58 eV, which is also confirmed by the experiment<sup>17</sup>. The maximum value of 3.58 eV is found at Si<sub>24</sub> with prolate structure, as well as at Si<sub>26</sub> and Si<sub>29</sub> clusters with spherical-like structures, which exhibits that these clusters are the most stable cluster in present study. Moreover, it is found that our calculated  $E_b$  (3.57 eV) of Si<sub>30</sub> agree well with the previous value (3.796 eV)<sup>19</sup>. The  $E_b$  of anionic clusters as a function of cluster size  $n$  is displayed in Fig. 5(b). From Table S1, it is found that the spherical-like and prolate Si<sub>20</sub><sup>-</sup>, Si<sub>24</sub><sup>-</sup> and Si<sub>26</sub><sup>-</sup> clusters are degenerate in binding energies, which can also be clearly seen from their relative energy difference as shown in Fig. 4(b). For the other clusters, the  $E_b$  of prolate structures are higher than those of the spherical-like structures. That is to say, the prolate structures are more energetically favorable than compact spherical-like structures. The Si<sub>29</sub><sup>-</sup> is the most stable structure among the obtained anionic clusters due to its largest binding energy. As for the cationic species (see Fig. 4(c)), we can clearly see that the prolate structures are more stable than the compact spherical-like structures for Si <sub>$n$</sub> <sup>+</sup> ( $n = 20–24$ ) clusters. This is in agreement with the result of their relative energy order. The binding energies for the prolate and compact structures increase slightly when  $n$  is smaller than 24. Furthermore, in the spherical-like structures, a sharply increasing is found at  $n = 29$

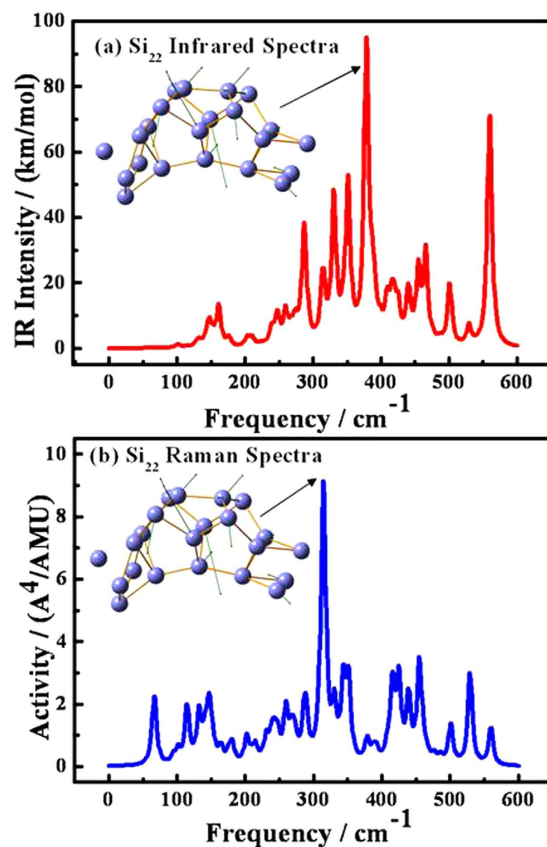


**Figure 4.** The binding energies of the spherical-like and prolate structures versus the number of atoms for neutral (a), anionic (b) and cationic (c)  $\text{Si}_n$  cluster, respectively.



**Figure 5.** The HOMO-LUMO gaps of  $\text{Si}_n^\mu$  ( $n=20-30$ ,  $\mu=0, -1$  and  $+1$ ) clusters.

and up to a maximum of 3.84 eV at  $n=30$ . Namely, the  $\text{Si}_{30}^+$  cluster is the most stable cluster within the cation charged clusters in the range of cluster size  $n=20-30$ . The small cluster  $\text{Si}_{20}^+$  is less stable compared with the other clusters for both prolate and spherical-like structures. From the above discussions, it is clear that the transition point from prolate structures to compact spherical-like structures occurs at



**Figure 6.** The gaussian broadened Raman activities and infrared intensities of  $\text{Si}_{22}$  cluster. Insets show the frequency modes corresponding to the highest activity or intensity.

$n = 26$  for neutral silicon clusters, at  $n = 27$  for anions and  $n = 25$  for cations. Therefore, the accepting or loss of an extra electron strongly affects the structures of silicon clusters.

The HOMO-LUMO energy gap between the highest occupied molecular orbital (HOMO) and the lowest unoccupied molecular orbital (LUMO) is a useful quantity for examining the kinetic stability. A large energy gap corresponds to a high energy required for electron excitation. The size-dependent energy gaps of the most stable  $\text{Si}_n^\mu$  ( $n = 20-30$ ,  $\mu = 0, -1$  and  $+1$ ) clusters are summarized in Table S1 and plotted in Fig. 5. By the comparison between the energy gaps of neutral, anionic and cationic silicon clusters, we can note that the curves of energy gaps for both anionic and cationic clusters have approximate tendency when  $n \geq 25$ . A distinct maximum occurs at charged  $\text{Si}_{26}^{+/-}$  clusters among the cluster size  $n = 25-30$ , indicating that  $\text{Si}_{26}^{+/-}$  is relatively more chemical stable in the electronic structure compared other clusters. However, within the whole studied anionic clusters,  $\text{Si}_{21}^-$  is the most chemical stable cluster. For neutral  $\text{Si}_n$  ( $n = 20-30$ ) clusters, the HOMO-LUMO energy gap results reveal that the  $\text{Si}_{22}$  is the the highest kinetic stability cluster with the largest HOMO-LUMO gap of 2.85 eV.

In order to further check the dynamical stability of  $\text{Si}_{22}$ ,  $\text{Si}_{21}^-$  and  $\text{Si}_{26}^+$  clusters, we have calculated their vibrational spectra. The infrared and Raman spectra of  $\text{Si}_{22}$  are shown in Fig. 6, and the spectra of  $\text{Si}_{21}^-$  and  $\text{Si}_{26}^+$  clusters are shown in Figure S5 of the supporting information. We have also shown the direction of motion of the ions for the frequency with the highest Raman activity or infrared intensity. In the following, we will take the infrared and Raman spectra of  $\text{Si}_{22}$  as an example to describe the dynamical stability of the neutral silicon clusters. From the insets in Fig. 6, it can be seen that Si atom localized at the outside mainly contributes to the highest peaks of infrared spectrum and Raman activity. These spectra can provide a spectroscopic fingerprint to assist experimentalists to distinguish different species and different isomers. The infrared spectra and Raman activity of  $\text{Si}_{22}$  have several peaks due to its low  $C_s$ -symmetry. The highest intensity peak of the infrared spectra is 95.02 km/mol, and the highest Raman activity is  $9.14 \text{ \AA}^4/\text{amu}$ . They are located at  $378.00$  and  $314.00 \text{ cm}^{-1}$ , respectively. The complex nature of the far-infrared region when combined with the highest intensity peak of the infrared spectra can be used as a fingerprint for identifying different  $\text{Si}_{22}$  isomer. Raman activity mainly corresponds to the breathing modes, and in these modes all the ions in clusters having high symmetry move together. The high-frequency peak for Raman spectra reflects the strong bonding of cluster  $\text{Si}_{22}$ .

The vertical ionization potential (VIP) is also an important parameter to assess the chemical stability of clusters. The VIP is defined as the energy differences between the total energy of neutral and cationic



Cluster	$\alpha$		Cluster	$\alpha$	Cluster	$\alpha$
	Present	Literature <sup>a</sup>				
Si <sub>20</sub>	4.73	4.93	Si <sub>20</sub> <sup>-</sup>	5.58	Si <sub>20</sub> <sup>+</sup>	5.32
Si <sub>21</sub>	4.87	5.09	Si <sub>21</sub> <sup>-</sup>	5.00	Si <sub>21</sub> <sup>+</sup>	4.80
Si <sub>22</sub>	4.86	4.86	Si <sub>22</sub> <sup>-</sup>	5.28	Si <sub>22</sub> <sup>+</sup>	4.82
Si <sub>23</sub>	4.78	4.99	Si <sub>23</sub> <sup>-</sup>	5.01	Si <sub>23</sub> <sup>+</sup>	4.59
Si <sub>24</sub>	4.78	4.97	Si <sub>24</sub> <sup>-</sup>	5.22	Si <sub>24</sub> <sup>+</sup>	4.80
Si <sub>25</sub>	4.97	5.13	Si <sub>25</sub> <sup>-</sup>	5.64	Si <sub>25</sub> <sup>+</sup>	4.34
Si <sub>26</sub>	4.52	4.70	Si <sub>26</sub> <sup>-</sup>	5.16	Si <sub>26</sub> <sup>+</sup>	4.31
Si <sub>27</sub>	4.47	4.64	Si <sub>27</sub> <sup>-</sup>	4.72	Si <sub>27</sub> <sup>+</sup>	4.29
Si <sub>28</sub>	4.57	4.57	Si <sub>28</sub> <sup>-</sup>	4.84	Si <sub>28</sub> <sup>+</sup>	4.45
Si <sub>29</sub>	4.45		Si <sub>29</sub> <sup>-</sup>	4.67	Si <sub>29</sub> <sup>+</sup>	4.28
Si <sub>30</sub>	4.52		Si <sub>30</sub> <sup>-</sup>	4.76	Si <sub>30</sub> <sup>+</sup>	4.38

**Table 3.** The computed mean polarizability per atom of Si<sub>n</sub><sup>μ</sup> ( $n=20-30$ ,  $\mu=0, -1$  and  $+1$ ) clusters together with the available theoretical results.  $\alpha$  is in units of  $\text{\AA}^3/\text{atom}$ . <sup>a</sup>ref. 55.

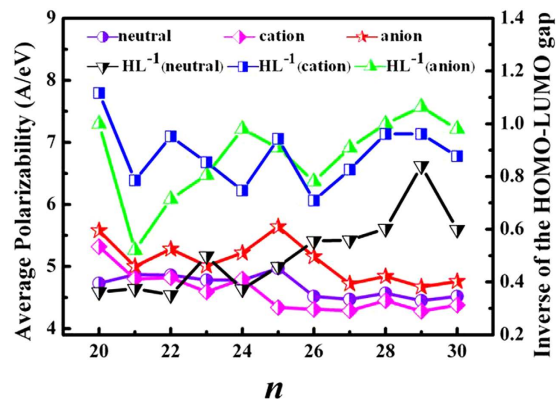
clusters with the same structure of the lowest-energy neutral state. Large VIPs indicate high chemical inertness. We have calculated the VIPs of the lowest-energy structures for the neutral silicon clusters. In addition, we have also calculated their adiabatic ionization potential (AIP), which is given by the formula  $\text{AIP} = E_{\text{optimized cation}} - E_{\text{optimized neutral}}$ . The calculated results are listed in Table 2. As shown in Table 2, our theoretical AIPs results of Si<sub>n</sub> ( $n=20-30$ ) clusters are in good agreement with the experimental data<sup>18</sup>. Si<sub>22</sub> has the largest vertical ionization potential (7.28 eV), corresponding to its higher chemical stability. The VIP values of the other clusters are less than that of Si<sub>22</sub> by 0.21 eV–1.00 eV. This suggests that Si<sub>22</sub> has the higher chemical stability than others, which is in accord with the above analysis based on HOMO-LUMO gap. For Si<sub>20</sub>, Si<sub>21</sub> and Si<sub>25</sub> clusters, our calculated VIP values (7.01 eV, 7.07 eV and 6.89 eV) are in agreement with the previous theoretical results (6.76 eV<sup>51</sup>, 6.85 eV<sup>44</sup> and 6.49 eV<sup>44</sup>), respectively.

**Polarizability.** According to the simple perturbation theory using the one-electron wave function, the value of polarizability  $\alpha$  can be given by the following sum-over-states (SOS) expression<sup>52–54</sup>:

$$\alpha_{xx} = 2 \sum_{k,l} \frac{e^2 |\langle k|x|l \rangle|^2}{\varepsilon_l - \varepsilon_k} \quad (1)$$

where the one electron matrix elements  $l$  and  $k$  are the antibonding (or unoccupied) and bonding (or occupied) orbitals, respectively.  $\varepsilon_l - \varepsilon_k$  is the corresponding HOMO-LUMO transition energy. Accurate evaluation of matrix element is not straightforward, because it involves detailed knowledge of the wave functions of orbitals  $k$  and  $l$  as well as the relative position of each atom. Luckily, when the size of a molecule is large, the matrix element part of the equation can often be considered more or less constant. In this case, the mean polarizability per atom is then given by the invariant trace  $\alpha = (\alpha_{xx} + \alpha_{yy} + \alpha_{zz})/3n$ , where  $n$  is the number of atoms in the cluster. As a benchmark test of the method, we have calculated the polarizability of isolated silicon atom. The theoretical value of  $3.71 \text{\AA}^3/\text{atom}$  agrees well with the experimental result of  $3.70 \text{\AA}^3/\text{atom}$ <sup>55</sup>. Thus, we can extend the above calculated method to the medium-sized Si<sub>n</sub> clusters in the range of  $n=20-30$ .

The calculated results for Si<sub>n</sub><sup>μ</sup> ( $n=20-30$ ,  $\mu=0, -1$  and  $+1$ ) clusters are summarized in Table 3, in which the  $\alpha$  values from the literature<sup>55</sup> are also included for comparison. From Table 3, it can be clearly seen that the calculated  $\alpha$  values of neutral clusters ( $n=20-28$ ) are in good agreement with the previous theoretical results at PBE level, with discrepancy is less than 0.25<sup>56</sup>. All the  $\alpha$  values of clusters are significantly larger than the polarizability of the bulk ( $3.17 \text{\AA}^3/\text{atom}$ )<sup>57</sup>. In addition, the polarizabilities of Si<sub>n</sub> clusters are not size sensitive and close to  $4.60 \text{\AA}^3/\text{atom}$ . However, the experimental values of Schäfer *et al.*<sup>55</sup> show much larger fluctuations as cluster size with an average value of about  $3.5 \text{\AA}^3/\text{atom}$  over the range  $n=20-28$ . This lack of disagreement may be explained by the temperature effects<sup>58</sup>. Since the average polarizability can only be directly measured in experiments if the static dipole moment of the cluster is zero. For cationic clusters, the calculated results of polarizability also show relatively small variations in the value of  $\alpha$  over the size range  $20 \leq n \leq 30$ , with all values significantly larger than the bulk limit. In contrast to anionic cluster, the calculated results indicate that the polarizabilities vary strongly and irregularly with size. From Fig. 7, it is interesting to note that there is a clear transition in the value of  $\alpha$  occurring at around  $n=27$ . The atomic polarizabilities can be related to the volume occupied by electrons. The compact spherical-like geometries have relatively fewer and shorter bonds,



**Figure 7.** The polarizability of  $\text{Si}_n^\mu$  ( $n=20-30$ ,  $\mu=0, -1$  and  $+1$ ) clusters as a function of the cluster size  $n$  together with the inverse of HOMO-LUMO gap.

binding the valence electrons tighter with a smaller spatial volume than the prolate structures. Thus, the spherical-like clusters have smaller polarizabilities than the prolate clusters. Once again the above polarizabilities transition have demonstrated that the structure transform from prolate to spherical-like geometries in anionic silicon cluster.

To get a clear insight of the correlation between polarizability and HOMO-LUMO gap, the  $\alpha$  values and the inverse of HOMO-LUMO gaps are plotted as a function of the cluster size  $n$  in Fig. 7. As is shown in Fig. 7, the curves of  $\alpha$  values is dissimilar to the  $(\text{HOMO-LUMO gap})^{-1}$  lines. This reveals that there is no such correlation between the polarizability and HOMO-LUMO gap among these clusters, which is consistent with the conclusions of Jackson *et al.*<sup>59</sup> and Deng *et al.*<sup>56</sup>. For example, the neutral  $\text{Si}_{25}$ , anion  $\text{Si}_{25}^-$  and cation  $\text{Si}_{20}^+$  have the largest polarizability (4.97, 5.64 and 5.32 Å<sup>3</sup>/atom, respectively) in respective species. If the inverse relationship between polarizability and HOMO-LUMO gaps is true, their  $(\text{HOMO-LUMO gap})^{-1}$  should be the largest values. Namely, these clusters should have the smallest HOMO-LUMO gaps as well. However, that is not the case in neutral and anionic  $\text{Si}_{25}$  cluster (see Table S1). This lost correlation between the polarizability and the HOMO-LUMO gap can be attributed to the vanishing matrix element  $|\langle k|x|l\rangle|^2$  between the HOMO and LUMO in certain clusters of high symmetry. Take neutral  $\text{Si}_{22}$  cluster as an example, the wave functions of the HOMO and LUMO can be of completely different symmetry (Figure S6). This incompatibility symmetry causes that the HOMO-LUMO transition is forbidden. Consequently, all the matrix elements between these occupied and unoccupied orbitals are zero.

## Conclusions

The following conclusions emerge from the present combined CALYPSO structure searching method and density-functional theory study of medium-sized neutral, anionic and cationic silicon clusters.

- (i) For each cluster size, an extensive search of the lowest-energy structure has been conducted by considering a number of isomers. The binding energies, HOMO-LUMO energy gaps, vertical ionization potentials, adiabatic detachment energies, polarizability including Raman activities, and infrared intensities are predicted at the B3PW91/6-311 + G\* level.
- (ii) Our structural optimizations indicate that an appreciable structural transition from prolate to spherical-like geometries occur at  $n=26$  for neutral  $\text{Si}_n$  clusters,  $n=27$  for anions and  $n=25$  for cations. This is in agreement with the previous experimental observations and theoretical predictions. In addition, the growth pattern of both neutral and cationic  $\text{Si}_n$  ( $n=20-24$ ) clusters shows a similar behavior.
- (iii) For neutral and cationic  $\text{Si}_n$  ( $n=20-30$ ) clusters, the structural stabilities between prolate and spherical-like structures were further verified by binding energies. Moreover, the relative stability analysis is carried out by calculating HOMO-LUMO gaps and vertical ionization potential, which shows that  $\text{Si}_{22}$  cluster has higher stability than the neighboring clusters.
- (iv) Based on the simple perturbation theory, we have discussed the relationship between the polarizability and HOMO-LUMO gaps. The results indicate that the inverse relationship between them does not hold in general by comparing the curves of polarizability and  $(\text{HOMO-LUMO gap})^{-1}$ . How to explain this phenomenon is still an open topic.

## References

1. Gao, J. F. & Zhao, J. J. Initial geometries, interaction mechanism and high stability of silicene on Ag (111) surface. *Sci. Rep-uk.* **2**, 861–868 (2012).
2. Limaye, M. V. *et al.* Understanding of sub-band gap absorption of femtosecond-laser sulfur hyperdoped silicon using synchrotron-based techniques. *Sci. Rep-uk.* **5**, 11466–11478 (2015).

3. Pittaway, F. *et al.* Theoretical studies of palladium-gold nanoclusters: Pd-Au clusters with up to 50 atoms. *J. Phys. Chem. C* **113**, 9141–9152 (2009).
4. Zhu, X. & Zeng, X. C. Structures and stabilities of small silicon clusters: Ab initio molecular-orbital calculations of Si<sub>7</sub>-Si<sub>11</sub>. *J. Chem. Phys.* **118**, 3558–3570 (2003).
5. Yoo, S., Zhao, J., Wang, J. L. & Zeng, X. C. Endohedral silicon fullerenes Si<sub>N</sub> (27 ≤ N ≤ 39). *J. Am. Chem. Soc.* **126**, 13845–13849 (2004).
6. Yoo, S. & Zeng, X. C. Motif transition in growth patterns of small to medium-sized silicon clusters. *Angew. Chem. Int. Ed.* **44**, 1491–1495 (2005).
7. Yoo, S. & Zeng, X. C. Structures and stability of medium-sized silicon clusters. III. reexamination of motif transition in growth pattern from Si<sub>15</sub> to Si<sub>20</sub>. *J. Chem. Phys.* **123**, 164303 (2005).
8. Yoo, S. *et al.* Structures and relative stability of medium-sized silicon clusters. V. low-lying endohedral fullerene-like clusters Si<sub>31</sub>-Si<sub>40</sub> and Si<sub>45</sub>. *J. Chem. Phys.* **124**, 164311 (2006).
9. Yoo, S., Shao, N. & Zeng, X. C. Structures and relative stability of medium and large-sized silicon clusters. VI. fullerene cage motif for low-lying clusters Si<sub>39</sub>, Si<sub>40</sub>, Si<sub>50</sub>, Si<sub>60</sub>, Si<sub>70</sub> and Si<sub>80</sub>. *J. Chem. Phys.* **128**, 104316 (2008).
10. Arnold, C. C. & Neumark, D. M. Study of Si<sub>4</sub> and Si<sub>4</sub><sup>-</sup> using threshold photodetachment (ZEKE) spectroscopy. *J. Chem. Phys.* **99**, 3353–3362 (1993).
11. Jarrold, M. F. & Constant, V. A. Silicon cluster ions: evidence for a structural transition. *Phys. Rev. Lett.* **67**, 2994–2997 (1991).
12. Jarrold, M. F. Nanosurface chemistry on sized-selected silicon clusters. *Sci.* **252**, 1085–1092 (1991).
13. Jarrold, M. F. & Bower, J. E. Mobilities of silicon cluster ions: the reactivity of silicon sausages and spheres. *J. Chem. Phys.* **96**, 9180–9190 (1992).
14. Shvartsburg, A. A., Hudgins, R. R., Dugourd, P. & Jarrold, M. F. Structural information from ion mobility measurements: applications to semiconductor clusters. *Chem. Soc. Rev.* **30**, 26–35 (2001).
15. Kaxiras, E. & Jackson, K. Shape of small silicon clusters. *Phys. Rev. Lett.* **71**, 727–730 (1993).
16. Jarrold, M. F. & Honea, E. C. Dissociation of large silicon clusters: the approach to bulk behavior. *J. Phys. Chem.* **95**, 9181–9185 (1991).
17. Bachelis, T. & Schafer, R. Binding energies of neutral silicon clusters. *Chem. Phys. Lett.* **324**, 365–372 (2000).
18. Fuke, K., Tsukamoto, K., Misaizu, F. & Sanekata, M. Near Threshold photoionization of silicon clusters in the 248–146 nm region: ionization potentials for Si<sub>n</sub>. *J. Chem. Phys.* **99**, 7808–7812 (1993).
19. Ma, L. *et al.* Lowest-energy endohedral fullerene structures of Si<sub>N</sub> (30 ≤ N ≤ 39) clusters by density functional calculations. *Phys. Rev. A* **73**, 063203 (2006).
20. Wang, J., Zhao, J., Ma, L. & Wang, G. First-principles study of structural evolution of medium-sized Si<sub>N</sub> clusters (41 ≤ N ≤ 50) within stuffed fullerene cages. *Eur. Phys. J. D* **45**, 289–294 (2007).
21. Yoo, S. & Zeng, X. C. Structures and relative stabilities of medium-sized silicon clusters. IV. Motif-based low-lying clusters Si<sub>21</sub>-Si<sub>30</sub>. *J. Chem. Phys.* **124**, 054304–054306 (2006).
22. Wang, Y. C., Lv, J., Zhu, L. & Ma, Y. M. Crystal structure prediction via particle-swarm optimization. *Phys. Rev. B* **82**, 094116 (2010).
23. Wang, Y. C., Lv, J., Zhu, L. & Ma, Y. M. CALYPSO: A method for crystal structure prediction. *Comput. Phys. Commun.* **183**, 2063–2070 (2012).
24. Wang, Y. C. *et al.* An effective structure prediction method for layered materials based on 2D particle swarm optimization algorithm. *J. Chem. Phys.* **137**, 224108 (2012).
25. Lv, J., Wang, Y. C., Zhu, L. & Ma, Y. M. Particle-swarm structure prediction on clusters. *J. Chem. Phys.* **137**, 084204 (2012).
26. Zhu, L. *et al.* Reactions of xenon with iron and nickel are predicted in the Earth's inner core. *Nature Chem.* **6**, 644–648 (2014).
27. Lu, S. H. *et al.* Self-assembled ultrathin nanotubes on diamond (100) surface. *Nature Commun.* **5**, 3666–3672 (2014).
28. Frisch, M. J. *et al.* *Gaussian 09 (Revision C.0)*. Gaussian, Inc., Wallingford, CT, 2009.
29. Cheeseman, J. R., Trucks, G. W., Keith, T. A. & Frisch, M. J. A comparison of models for calculating nuclear magnetic resonance shielding tensors. *J. Chem. Phys.* **104**, 5497–5509 (1996).
30. Adamo, C., Matteo, A. di & Barone, V. Tuning of structural and magnetic properties of nitronyl nitroxides by the environment. A combined experimental and computational study. *Adv. Quantum. Chem.* **36**, 45–47 (1999).
31. Becke, A. D. Density-functional thermochemistry. III. the role of exact exchange. *J. Chem. Phys.* **98**, 5648–5652 (1993).
32. Lee, C. T., Yang, W. T. & Parr, R. G. Development of the colle-salveti correlation-energy formula into a functional of the electron density. *Phys. Rev. B* **37**, 785–789 (1988).
33. Perdew, J. P. & Wang, Y. Accurate and simple analytic representation of the electro-gas correlation energy. *Phys. Rev. B: Condens. Matter.* **45**, 13244–13249 (1992).
34. Perdew, J. P., Burke, K. & Ernzerhof, M. Generalized gradient approximation made simple. *Phys. Rev. Lett.* **77**, 3865–3868 (1996).
35. Perdew, J. P., Ziesche, P. & Eschrig, H. *Electronic Structure of Solids*, ed. Akademie Verlag, Berlin, 1991.
36. Barlett, R. J. & Musia, M. Coupled-cluster theory in quantum chemistry. *Rev. Mod. Phys.* **79**, 291–352 (2007).
37. Kitsopoulos, T. N., Chick, C. J., Zhao, Y. & Neumark, D. M. Study of the low-lying electronic states of Si<sub>2</sub> and Si<sub>2</sub><sup>-</sup> using negative ion photodetachment techniques. *J. Chem. Phys.* **95**, 1441–1448 (1991).
38. Huber, K. P. & Herzberg, G. *Molecular Spectra and Molecular Structure, Constants of Diatomic Molecules*, vol. IV, Van Nostrand Reinhold, New York, 1979.
39. Dixon, D. A., Feller, D., Peterson, K. A. & Gole, J. L. The molecular structure and ionization potential of Si<sub>2</sub>: The role of the excited states in the photoionization of Si<sub>2</sub>. *J. Phys. Chem. A* **104**, 2326–2332 (2000).
40. Marijnissen, A. & Ter Meulen, J. J. Determination of the adiabatic ionization potentials of Si<sub>2</sub> and SiCl by photoionization efficiency spectroscopy. *Chem. Phys. Lett.* **263**, 803–810 (1996).
41. Bai, J. *et al.* Structural evolution of anionic silicon clusters Si<sub>N</sub><sup>-</sup> (20 ≤ N ≤ 45). *J. Phys. Chem. A* **110**, 908–912 (2006).
42. Zhu, X. L., Zeng, X. C., Lei, Y. A. & Pan, B. Structures and stability of medium silicon clusters. II. ab initio molecular orbital calculations of Si<sub>12</sub>-Si<sub>20</sub>. *J. Chem. Phys.* **120**, 8985–8965 (2004).
43. Nigam, S., Majumder, C. & Kulshreshtha, S. K. Structural and electronic properties of Si<sub>n</sub>, Si<sub>n</sub><sup>+</sup>, and AlSi<sub>n</sub> (n = 2–13) clusters: theoretical investigation based on ab initio molecular orbital theory. *J. Chem. Phys.* **121**, 7756–7763 (2004).
44. Yoo, S., Zeng, X. C., Zhu, X. & Bai, J. Possible lowest-energy geometry of silicon clusters Si<sub>21</sub> and Si<sub>25</sub>. *J. Am. Chem. Soc.* **125**, 13318–13319 (2003).
45. Ho, K. M. *et al.* Structures of medium-sized silicon clusters. *Nature* **392**, 582–585 (1998).
46. Hudgins, R. R., Imai, M. & Jarrold, M. F. High-resolution ion mobility measurements for silicon cluster anions and cations. *J. Chem. Phys.* **111**, 7865–7870 (1999).
47. Rata, I. *et al.* Single-parent evolution algorithm and the optimization of Si clusters. *Phys. Rev. Lett.* **85**, 546–549 (2000).
48. Jackson, K. A. *et al.* Unraveling the shape transformation in silicon clusters. *Phys. Rev. Lett.* **93**, 013401–013404 (2004).
49. Hoffmann, M. A. *et al.* Ultraviolet photoelectron spectroscopy of Si<sub>4</sub><sup>-</sup> to Si<sub>1000</sub><sup>-</sup>. *Eur. Phys. J. D* **16**, 9–11 (2001).
50. Pisani, C. *Quantum Mechanical Ab Initio Calculation of the Properties of Crystalline Materials*. Springer, Berlin, 1996, p. 183.
51. Liu, B. *et al.* Ionization of medium-sized silicon clusters and the geometries of the cations. *J. Chem. Phys.* **109**, 9401–9409 (1998).

52. Brieger, M., Renn, A., Sodeik, A. & Hese, A. Ionization of medium-sized silicon clusters and the geometries of the cations. *J. Chem. Phys.* **75**, 1–8 (1983).
53. Brieger, M. The Origin of the infrared multiphoton induced luminescence of chromyl chloride. *J. Chem. Phys.* **89**, 275–295 (1984).
54. Bégué, D., Mérawa, M. & Pouchan, C. Dynamic dipole and quadrupole polarizabilities for the ground  $2^1S$  and the low-lying  $3^1S$  and  $3^3S$  states of Be. *Phys. Rev. A* **57**, 2470–2473 (1998).
55. Schäfer, R., Schlecht, S., Woenckhaus, J. & Becker, J. A. Polarizabilities of isolated semiconductor clusters. *Phys. Rev. Lett.* **76**, 471–474 (1996).
56. Deng, K., Yang, J. & Chan, T. Polarizabilities of isolated semiconductor clusters. *Phys. Rev. A* **61**, 025201 (2000).
57. Pouchan, C., Bégué, D. & Zhang, D. Y. Calculated polarizabilities of small Si clusters. *J. Chem. Phys.* **121**, 4628–4634 (2004).
58. Jackson, K., Pederson, M., Wang, C. Z. & Ho, K. M. Calculated polarizabilities of intermediate-size Si clusters. *Phys. Rev. A* **59**, 3685–3687 (1999).
59. Jackson, K. A., Yang, M., Chaudhuri, I. & Frauenheim, Th. Shape, polarizability, and metallicity in silicon clusters. *Phys. Rev. A* **71**, 033205 (2005).

## Acknowledgements

This work is supported by the the National Natural Science Foundation of China (Nos. 11274235 and 11304167), the Doctoral Education Fund of Education Ministry of China (Nos. 20100181110086 and 20111223070653), Postdoctoral Science Foundation of China (No. 20110491317) and Open Project of State Key Laboratory of Superhard Materials (No. 201405)

## Author Contributions

C.L. conceived the idea. L.P.D., C.L. and J.L. performed the calculations. L.P.D., F.H.Z., Y.S.Z., X.Y.K. and P.S. wrote the manuscript and all authors contributed to revisions.

## Additional Information

**Supplementary information** accompanies this paper at <http://www.nature.com/srep>

**Competing financial interests:** The authors declare no competing financial interests.

**How to cite this article:** Ding, L. P. *et al.* Understanding the structural transformation, stability of medium-sized neutral and charged silicon clusters. *Sci. Rep.* **5**, 15951; doi: 10.1038/srep15951 (2015).



This work is licensed under a Creative Commons Attribution 4.0 International License. The images or other third party material in this article are included in the article's Creative Commons license, unless indicated otherwise in the credit line; if the material is not included under the Creative Commons license, users will need to obtain permission from the license holder to reproduce the material. To view a copy of this license, visit <http://creativecommons.org/licenses/by/4.0/>