

2D Hybrid Nanocomposite Materials (h-BN/G/MoS₂) as a High-Performance Supercapacitor Electrode

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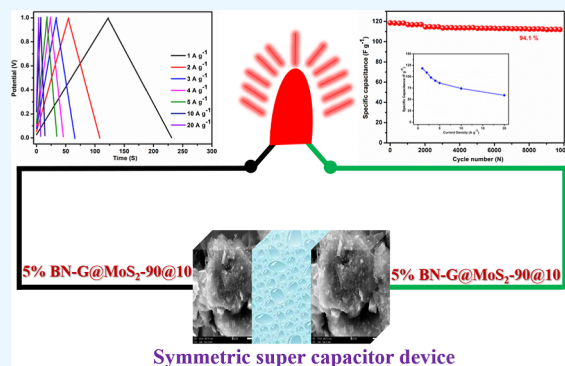


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ABSTRACT: The nanocomposites of hexagonal boron nitride, molybdenum disulfide, and graphene (h-BN/G/MoS₂) are promising energy storage materials. The originality of the current work is the first-ever synthesis of 2D-layered ternary nanocomposites of boron nitrate, graphene, and molybdenum disulfide (h-BN/G/MoS₂) using ball milling and the sonication method and the investigation of their applicability for supercapacitor applications. The morphological investigation confirms the well-dispersed composite material production, and the ternary composite appears to be made of h-BN and MoS₂ wrapping graphene. The electrochemical characterization of the prepared samples is evaluated by cyclic voltammetry and galvanostatic charge/discharge tests. With a high specific capacitance of 392 F g⁻¹ at a current density of 1 A g⁻¹ and an outstanding cycling stability with around 96.4% capacitance retention after 10,000 cycles, the ideal 5% BN_G@MoS₂_90@10 composite demonstrates exceptional capabilities. Furthermore, a symmetric supercapacitor (5% BN_G@MoS₂_90@10 composite) exhibits a 94.1% capacitance retention rate even after 10,000 cycles, an energy density of 16.4 W h kg⁻¹, and a power density of 501 W kg⁻¹. The findings show that the preparation procedure is safe for the environment, manageable, and suitable for mass production, which is crucial for advancing the electrode materials used in supercapacitors.



1. INTRODUCTION

Numerous types of electrochemical energy systems have been developed to meet the rapidly expanding demand for renewable energy in the human society.¹ Lithium-ion batteries, supercapacitors, aqueous zinc-ion batteries, and fuel cells are a few typical examples. Among them, supercapacitor (SC) is a desirable energy storage device because of its high-power density, quick charge/discharge, and long cycle life.^{2,3} SCs can be divided into electrical double-layer capacitors (EDLCs) and pseudocapacitors based on the energy storage mechanism.⁴ While pseudocapacitors primarily use electroactive conducting polymers or transition-metal oxides as electrode materials, EDLCs are often constructed of carbon materials or carbon-based hybrid materials with high surface areas and acceptable pore sizes.^{5–10}

In recent years, two-dimensional (2D) transition-metal dichalcogenides have received significant attention due to their unique graphite-like layered structure.^{11,12} These materials' chemical, electrical, thermal, and mechanical characteristics open up a wide range of application.^{13,14} Molybdenum disulfide (MoS₂), in particular, provides important indications in acting as a satisfactory electrode material that it has a place in supercapacitor applications among various binary chalcogenides, such as disulfides, diselenides, and ditellurides of transitional-metal elements with intriguing sheet-like structure.^{15–17} Its enormous surface

area and hexagonal-like structure make it a desirable material for use in supercapacitors.¹⁸ To date, some reports have been concentrated on the supercapacitor characteristics of pure MoS₂ with various morphologies.^{19,20} For instance, Huang et al. reported that MoS₂ nanosheets as electrode materials had a specific capacitance retention of 85.1% after 500 cycles and a capacitance of up to 129.2 F g⁻¹ at 1 A g⁻¹.²¹ Wang et al. created flower-like MoS₂ that, when used as electrode materials for electrochemical capacitors, demonstrated a specific capacitance of 168 F g⁻¹ at 1 A g⁻¹.²²

To the best of our knowledge, pure MoS₂ utilized as an electrode material alone still has very limited energy storage applications due to its interlayer van der Waals interactions and high surface energy and nanosheets which can rearrange, leading to poor electronic conductivity and significant volume change of the material throughout the charge/discharge process.^{23–27} Eventually, the SCs' cycling abilities substantially deteriorate. Therefore, to fill the gap in increasing electrical

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conductivity and capacitance performance, several researchers have turned to combining MoS₂ with other conducting materials.^{15,16,28–34} Due to its exceptional electrical conductivity, enormous surface area, high mechanical resilience, and excellent chemical stability, graphene, a 2D layered material with zero band gap, has been widely used in electrochemical capacitors.^{35–38} Therefore, the composites of MoS₂ and graphene would possess excellent electrochemical properties compared to that of pure MoS₂ and graphene for the application of supercapacitors.^{15,39,40}

An example of a 2D material was hexagonal boron nitride (h-BN), which was comparable to graphene and molybdenum disulfide (MoS₂). The use of h-BN as a polymer filler in a variety of applications, from electrical to biomedical, has been established. It has good thermal stability, high hydrophobicity, and high thermal conductivity.^{41,42} The electronic industry prefers using h-BN as a filler material because of its ability to absorb moisture, dielectric constant, and thermal expansion coefficient.⁴³ Application of h-BN, which includes electrolytes and electrodes, to energy storage devices like supercapacitors may present a favorable opportunity and advance the area.^{44–47} The h-BN act as a filler for the graphene and MoS₂ from the interlayer interactions and agglomeration,⁴² therefore, the combination of MoS₂ and graphene with h-BN would definitely be a promising candidate for next-generation supercapacitors.

In this paper, we demonstrate that nanocomposites of h-BN with MoS₂ and graphene would be an excellent candidate for use as a supercapacitor. We employed a ball milling method for the preparation of MoS₂ and graphene composition. Following this, we have followed the sonication process for h-BN with the graphene/MoS₂ composite to get the final ternary composite (h-BN/graphene/MoS₂). In the three composites, the optimal composite (5% BN_G@MoS₂_90@10) exhibits a high specific capacitance of 392 F g⁻¹ when tested at a current density of 1 A g⁻¹ for 10,000 cycles, and the retention rate was 96.4%. Inspired by these results, a supercapacitor was fabricated using two identical (symmetric supercapacitor) 5% BN_G@MoS₂_90@10 electrodes. The assembled device yielded 94.1% of capacitance retention rate even after 10,000 cycles, an energy density of 16.4 Wh kg⁻¹, and a power density of 501 W kg⁻¹. These outstanding behaviors are attributed to the ternary nanocomposite of graphene, MoS₂, and h-BN.

2. EXPERIMENTAL SECTION

2.1. Materials. Molybdenum(IV) sulfide (~98%) and h-BN (~98%) were purchased from Graphene supermarket. Graphene powder (99.9999%) was purchased from XG sciences. Carbon black (50% compressed) and poly(vinylidene fluoride) powder were purchased from MTI Corporation. *N*-Methyl-2-pyrrolidone (~99.5%) was purchased from Merck. For cleaning, acetone (≥99.0%) and ethanol (≥99.0%) were purchased from Sigma-Aldrich.

2.2. Material Preparation. **2.2.1. Preparation h-BN/Graphene/MoS₂ Composites.** In the first step, MoS₂/graphene (50:50, 10:90, and 90:10) nanocomposites were synthesized via the ball milling and sonication process by using graphene and MoS₂ as starting materials. The graphene powder and MoS₂ powder were ground for 30 min and ball milled with different ratios (50:50, 10:90, and 90:10 ratios) [powder weight ratio with balls to material was 4:1 with the 1060 rotational speed (rpm)] for 45 min by using SPEX samplePrep P instrument [model number 8000 M MIXER/MILL using

two different sizes of the balls (four balls with 6.3 mm and two balls with 12.5 mm prepared with stainless steel)]. To get better uniformity, the above ratio powers were dispersed in ethanol (25 mL) by using ultrasonication (250 W 20 kHz, model VWR Ultrasonic cleaner USC THD instrument) for 60 min at room temperature. The yield of the MoS₂@graphene nanohybrids was dried in the vacuumed furnace at 100 °C for 6 h. Afterward, MoS₂ with graphene composites (50:50, 10:90, and 90:10 ratios) was ball milled with 5 wt % of BN for 45 min and subsequently ultrasonicated for 60 min to get better uniformity. The final products of 5% BN_G@MoS₂_50@50, 5% BN_G@MoS₂_90@10, and 5% BN_G@MoS₂_10@90 were obtained by ultrasonicated material dried in the vacuumed furnace at 100 °C for 6 h.

3. CHARACTERIZATION

3.1. Physicochemical Characterization. A Bruker D8 Advance X-ray diffractometer with a scanning range of 10–70° in 2θ and Cu Kα radiation (λ = 1.5406) was used to acquire the X-Ray diffraction (XRD) data. Surface morphology of synthesized material was investigated using Gemini scanning electron microscopy (SEM), transmission electron microscopy (TEM), and high-resolution transmission electron microscopy (HRTEM) by using FEI Quanta 200 and TEM, JEOL2100 Instruments. For sample preparation, the powder composite sample was dispersed in an ethanol–water mixture by 20 min ultrasonication, followed by drop-casting onto a fresh lacey carbon copper grid. Raman spectra of the samples were collected using WITec Apyron Raman microscope equipment and a 532 nm solid-state laser as the excitation source. X-ray photoelectron spectroscopy (JEOL JPS-9030) was utilized to study the elemental composition and oxidation states of elements by passing an energy of 20 eV.

3.2. Electrochemical Characterization. The electrochemical performance tests including cyclic voltammetry (CV), galvanostatic current charge–discharge (GCD), and electrochemical impedance spectroscopy (EIS) methods (AC voltage of 5 mV) in the three-electrode or two-electrode system were implemented by using a BioLogic SP-300 Modular electrochemical workstation. The 5% BN-G@MoS₂-10@90, 5% BN-G@MoS₂-50@50, and 5% BN-G@MoS₂-90@10 were used to fabricate the working electrodes. The composite samples (BN-G@MoS₂) were mixed with carbon black (Super-P) and poly(vinylidene fluoride) (PVDF) in a weight ratio of 85:10:5 and coated on the nickel foam. The mass loading of the three-electrode materials on the nickel foam (15 mm) was 1.0–1.5 mg. The working electrodes were obtained after drying at 60 °C overnight in a vacuum oven. The electrochemical tests of the individual electrodes were performed using 6 M aqueous KOH solution as the electrolyte in a three-electrode cell in which platinum foil and Ag/AgCl were used as counter and reference electrodes, respectively. The symmetric supercapacitors performed using the 5% BN-G@MoS₂-90@10 electrode as both the cathode and anode in a two-electrode system and the separator was Whatman filter paper. All measurements were taken at room temperature.

For three-electrode cells, the capacitance was calculated using the charge integrated from GCD curves, according to the following formula

$$C_s = \frac{I \Delta t_d}{m \Delta V} \quad (1)$$

where I is the discharge current (A), Δt is the discharge time (s), m is the mass of the active material (g), and ΔV is the discharge potential window (V).

The energy density (E , Wh kg⁻¹) at different power densities (P , W kg⁻¹) of the supercapacitor cell could be calculated from the GCD curves at different current densities according to eqs 2 and 3

$$E = \frac{0.5 \times C_{CD} \times (\Delta V)^2}{3.6} \quad (2)$$

$$P = \frac{E}{\Delta t} \quad (3)$$

where ΔV is the potential window of discharge (V) and Δt is the discharge time (s).

4. RESULTS AND DISCUSSION

4.1. Physicochemical Characterization. The crystal structure and phase information on the samples (5% BN-G@MoS₂-50@50, 5% BN-G@MoS₂-10@90, and 5% BN-G@MoS₂-90@10) were obtained by X-ray diffraction spectroscopy, as shown in Figure 1. The diffraction peak located at 26.4°

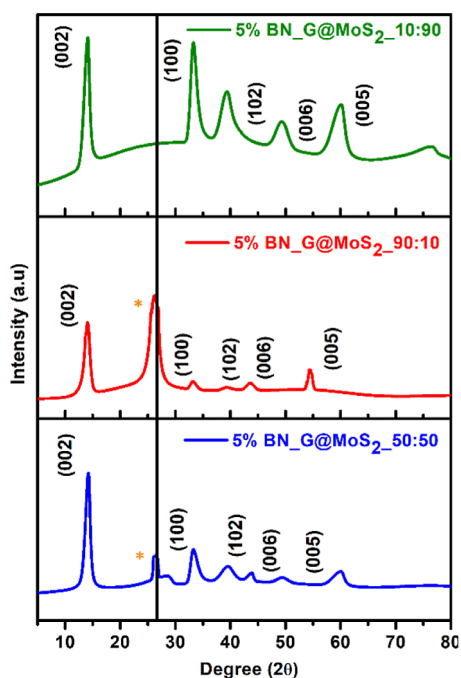


Figure 1. XRD patterns of the 5% h-BN/G/MoS₂ nanocomposites (1:9, 5:5, and 9:1).

in the 5% BN-G@MoS₂-50@50 and 5% BN-G@MoS₂-90@10 is corresponded to the (002) crystallographic plane of graphene, and this plane disappeared in the pattern of 5% BN-G@MoS₂-10@90 nanocomposites, revealing that graphene is insufficient in the composition. The diffraction peaks located at 14.1, 33.1, 39.4, 43.8, and 54.4° correspond to the (002), (100), (102), (006), and (110) crystallographic planes of MoS₂, respectively, which are in good agreement with the standard card of JCPDS no. 37-1492.^{48,49} However, 5% BN-G@MoS₂-90@10 exhibits slightly broader peaks, thus indicating nanocrystalline product formation, aided by the increased graphene content. On the other hand, 5% BN-G@MoS₂-50@50 shows a diminished (100) peak at $2\theta = 26.4^\circ$

and slightly enhanced intensity for peaks located at $2\theta = 14.1^\circ$, which is considered as a MoS₂ dominating the overall composition. As a result, the XRD of 5% BN-G@MoS₂-90@10 exhibits strong and intense peaks at $2\theta = 14.1^\circ$ and $2\theta = 26.4^\circ$ corresponding to MoS₂ and graphene would be beneficial for an easy charge transport for the application in supercapacitors.

Further analysis of the interplay between BN, MoS₂, and G during the sonication and ball-milling process was done using Raman spectroscopy, as shown in Figure 2. The characteristic

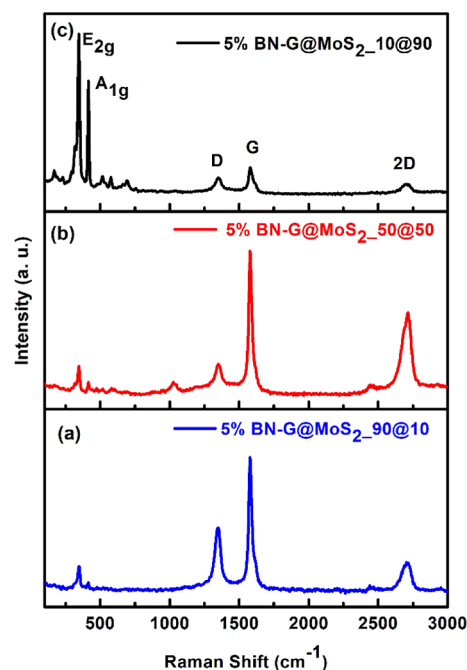


Figure 2. Raman spectra of (a) 5% BN-G@MoS₂-90@10, (b) 5% BN-G@MoS₂-50@50, and (c) 5% BN-G@MoS₂-10@90 samples.

Raman peaks of composites detected at 350, 423, 1348, 1585, and 2703 cm⁻¹ are ascribed to the in-plane ¹E_{2g}, out-of-plane A_{1g} vibrational modes, D band (disorderly defect structure), G band (sp²-bonded carbon atoms), and 2D mode, respectively.^{50,51} The peaks at 350 and 423 cm⁻¹ are Mo–S bands corresponding to in-plane ¹E_{2g} and out-of-plane A_{1g} vibrational modes for MoS₂. In the case of the 5% BN-G@MoS₂-90@10 composite with the maximum amount of MoS₂, the in-plane ¹E_{2g} and out-of-plane A_{1g} vibrational modes of MoS₂ are clearly visible with small peaks of graphene and BN. For the 5% BN-G@MoS₂-50@50 composite with an equal amount of graphene and MoS₂, the sharp peaks attributed to the D and G bands contributed by h-BN and graphene and the ¹E_{2g} and A_{1g} vibrational mode intensity was very small. However, in the case of 5% BN-G@MoS₂-90@10, the least amount of MoS₂, the ¹E_{2g} and A_{1g} vibrational modes showed less intensity, and D and G bands are of high intensity. The crystal size of the composites was calculated by using full width at half-maximum of the composites, and 5% BN-G@MoS₂-90@10 shows small size compared to the 5% BN-G@MoS₂-50@50 and 5% BN-G@MoS₂-10@90 composites. Raman analysis clearly explained that MoS₂ embedded in the graphene and BN sheets.

SEM was used to examine the morphologies and microstructures of the as-prepared (5% BN-G@MoS₂-50@50, 5% BN-G@MoS₂-10@90, and 5% BN-G@MoS₂-90@10) compo-

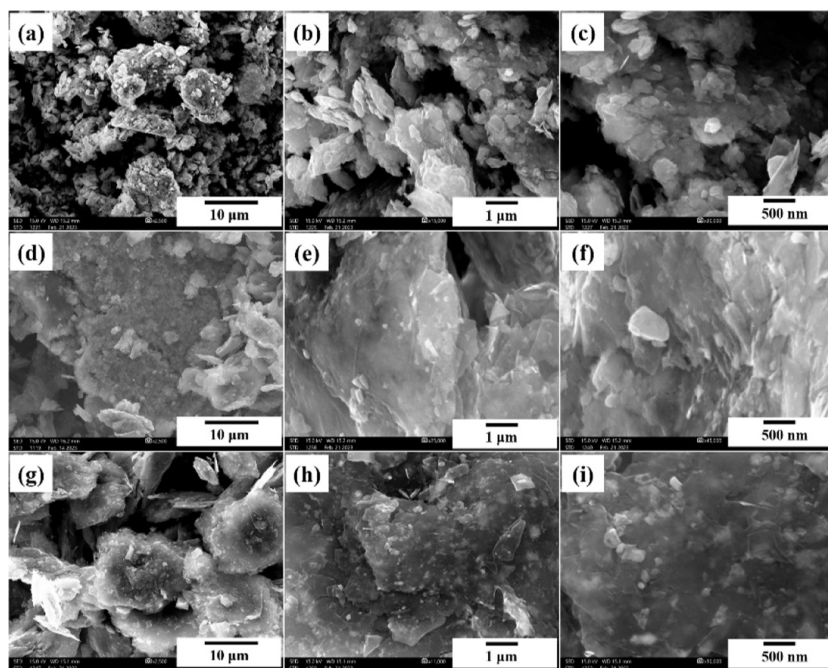


Figure 3. SEM images of the as-prepared 5% h-BN/G/MoS₂ nanocomposites. (a–c) 5% BN-G@MoS₂-10@90, (d–f) 5% BN-G@MoS₂-50@50, and (g–i) 5% BN-G@MoS₂-90@10.

site, as shown in Figure 3. Figure S1, elemental mapping study of the 5% BN-G@MoS₂-90@10 composite, shows compelling evidence of the homogeneous distribution of the elements C, O, Mo, S, B, and N. SEM images clearly suggest that the morphology is greatly influenced by the ratios of graphene and MoS₂. As shown in Figure 3a–c, the composition of 5% BN-G@MoS₂-10@90, the entire MoS₂ is not covered with a graphene substrate network and agglomerated with each other. Apart from that 5% BN-G@MoS₂-50@50 (Figure 3d–f) and 5% BN-G@MoS₂-90@10 (Figure 3g–i) composites are different; the graphene sheets are clearly visible with a sufficient amount which is in agreement with the findings from TEM and HRTEM (Figures 4 and S2). Figure 4 makes it evident that each individual MoS₂ particle has a carbon covering (gray color). The magnified view of Figure 4c,d demonstrates the existence of individual MoS₂ with a thin line of carbon coating to prevent aggregation and to constrain enable the inevitable volume increase. Especially, MoS₂ was uniformly dispersed and well decorated on the graphene sheet in the 5% BN-G@MoS₂-90@10 nanocomposite; the aggregation of the nanosheets was restrained in the composition. As shown in Figure S2a,b, both the HRTEM images reveal that MoS₂ and graphene overlap (parallel) each other with BN material, confirmed by the spacing of MoS₂ at 2.9 Å.⁵²

Elemental analysis and X-ray photoelectron spectroscopy (XPS) were also acquired to examine the as-prepared product's chemical composition and phase state of 5% BN-G@MoS₂-90@10. The XPS full survey spectrum of 5% BN-G@MoS₂-90@10 is shown in Figure S3a, where the presence of the elements Mo, S, B, N, C, and O is visible. Two distinct peaks, located at 229.4 and 233.3 eV in the high-resolution XPS spectra of MoS₂ 5% BN-G@MoS₂-90@10 (Figure S3b), are indicative of the Mo⁴⁺ 3d_{5/2} and Mo⁴⁺ 3d_{3/2} components of the compound, respectively. Another minor peak at 226.6 eV is the S 2s of MoS₂, and other peaks are seen at 162.3 and 163.5 eV,

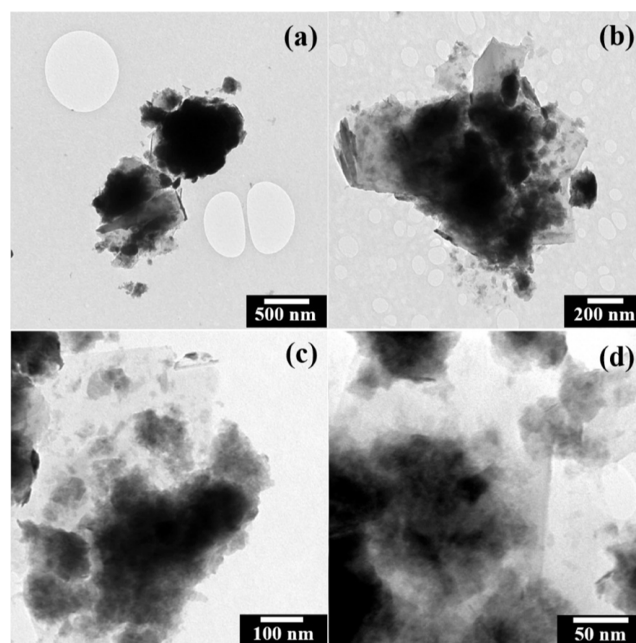


Figure 4. TEM images of (a–d) 5% BN-G@MoS₂-90@10 nanocomposite.

respectively, due to $S^{2-} - 2P_{3/2}$ and $S^{2-} - 2P_{1/2}$ (Figure S3c). The B–N peak at 190.78 eV and the N–B peak at 398.37 eV in Figure S3d and Figure S3e, respectively, are designated as the B 1s and N 1s spectra. These spectra arise from the BN of some boron and nitrogen atoms at the 5% BN-G@MoS₂-90@10 sample surface. The C 1s spectra in Figure S3f are identified as two distinct peaks with binding energies of 284.4 and 285.8 eV, attributed to sp²-hybridized graphitic carbon C=C, C–C, and C=O configurations, respectively. This finding demonstrates that the ball milling approach successfully created the 5% BN-G@MoS₂-90@10 nanocomposite. Additionally, the

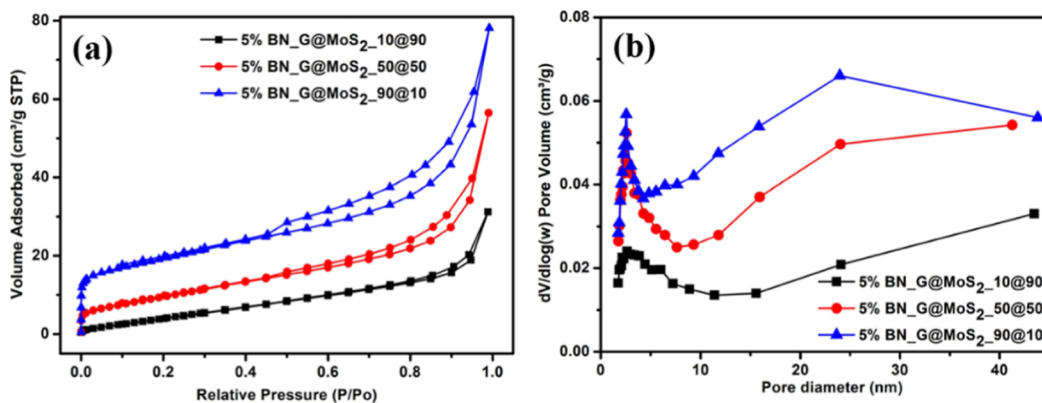


Figure 5. (a) Nitrogen adsorption/desorption isotherms and (b) the corresponding pore size distribution curves of 5% BN-G@MoS₂-10@90, 5% BN-G@MoS₂-50@50, and 5% BN-G@MoS₂-90@10.

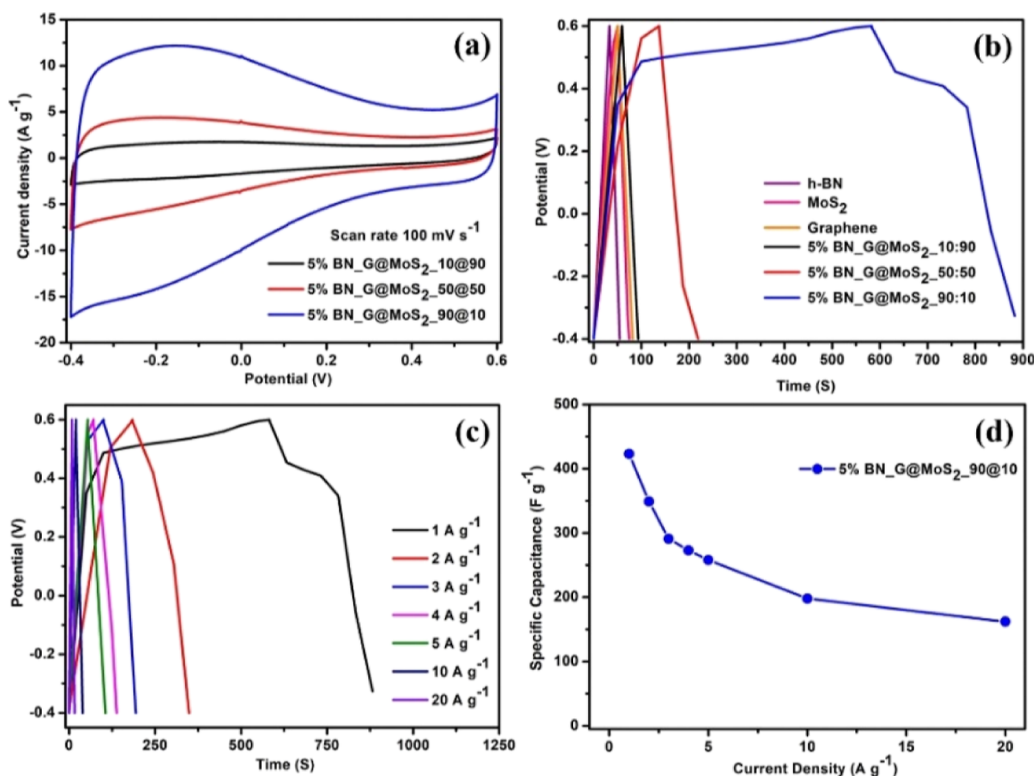


Figure 6. Electrochemical performances of 5% BN-G@MoS₂ nanocomposite (1:9, 5:5, and 9:1) electrodes in a three-electrode system: (a) CV curves at 100 mV s⁻¹. (b) GCD curves of h-BN, MoS₂, graphene, and 5% BN-G@MoS₂ nanocomposite (1:9, 5:5, and 9:1) at 1 A g⁻¹. (c) Charge/discharge profiles of 5% h-BN-G@MoS₂-90@10 at different current densities. (d) Specific capacitance of 5% h-BN-G@MoS₂-90@10 at different current densities.

nanocomposite contains certain oxygen groups that can potentially adsorb heavy metals through complexation.

The surface areas and porosities of the as-prepared 5% BN-G@MoS₂-10@90, 5% BN-G@MoS₂-50@50, and 5% BN-G@MoS₂-90@10 composite samples are characterized by N₂ adsorption/desorption isotherm measurements. The isotherm and pore size distribution curves are shown in Figure 5a,b. Three samples show type-II isotherms with hysteresis loops, suggesting the presence of the mesoporous structure. The specific surface areas of the 5% BN-G@MoS₂-10@90, 5% BN-G@MoS₂-50@50, and 5% BN-G@MoS₂-90@10 composite samples were 21.3, 36.9, and 68.2 m² g⁻¹, and the pore volume was 0.04, 0.09, and 0.11 cm³ g⁻¹, respectively. The surface area of the 5% BN-G@MoS₂-90@10 (68.2 m² g⁻¹) composite is 3.2

times higher than that of 5% BN-G@MoS₂-10@90 (21.3 m² g⁻¹) and 1.84 times higher than that of the 5% BN-G@MoS₂-50@50 (36.9 m² g⁻¹) composite sample. The average pore size was 2.6 nm in all the samples. The surface area of the samples increased with increasing concentration of graphene. High specific surface area and high pore volume (mesoporous) provide additional active sites and expand the contact space between the electrolyte solution and electrode.

4.2. Electrochemical Characterization. First, by using CV and GCD measurements in an alkaline medium (6 M KOH), three electrodes were used to examine the electrochemical performance of 5% BN-G@MoS₂-50@50, 5% BN-G@MoS₂-10@90, and 5% BN-G@MoS₂-90@10. For this measurement, the potential window was for the compositions

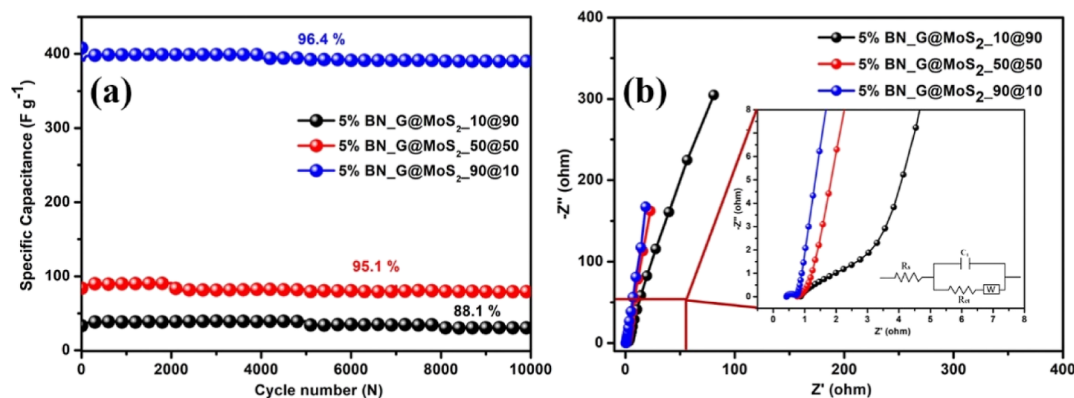


Figure 7. (a) Cycling stability at 1 A g^{-1} and (b) Nyquist plots before cycling and the fitted equivalent circuit of the 5% BN_G@MoS₂ nanocomposite (1:9, 5:5, and 9:1) electrodes.

in the voltage range of -0.4 to 0.6 V . The CV curves at different scan rates from 10 to 500 mV s^{-1} (Figure S4a–c) show the remarkable difference in electrochemical surface activity between 5% BN-G@MoS₂-50@50, 5% BN-G@MoS₂-10@90, and 5% BN-G@MoS₂-90@10 nanocomposites. For comparison, the curves of the three nanocomposites at a scan rate of 100 mV s^{-1} are shown in Figure 6a. All CV curves present a capacitive behavior with rectangular shapes and a hump in the CV curves, indicating electric double-layer capacitance (EDLC) behavior of graphene and the pseudocapacitance of MoS₂. However, the area and the current density response of 5% BN-G@MoS₂-90@10 are much higher than those of 5% BN-G@MoS₂-10@90 and 5% BN-G@MoS₂-50@50, indicating a higher specific capacitance value. To understand the state transitions during the cycling process and the reaction kinetics is needed to get additional insights into the remarkable electrochemical performance of BN-G@MoS₂ composites as a material for supercapacitors. As shown in Figure S4, we examined the CV curves of the BN-G@MoS₂ composites at different scan speeds. Usually expressed as $i = av^b$, the relationship between current (i) and scan rate (v) can be found using a $\log(i) - \log(v)$ curve. Here, parts a and b are the variable constants. A complete diffusion-controlled process is indicated by $b = 0.5$, whereas $b = 1$ denotes a capacitive process fully responsible for charge storage. The results of the calculation indicate that the electrodes 5% h-BN_G@MoS₂-10@90, 5% h-BN_G@MoS₂-50@50, and 5% h-BN_G@MoS₂-90@10 exhibit a linear relationship between scan speed and peak current, as indicated by $b = 0.84, 0.91,$ and 0.93 in Figure S5. This suggests that capacitance control, or surface drive, has a higher ratio of charge storage in comparison to diffusion control. In particular, because of the graphene proportion, 5% h-BN_G@MoS₂-90@10 exhibits a higher capacity contribution than diffusion.

The following formulas are used to determine the contribution from charge storage of the diffusive and capacitive types (eqs 4 and 5).

$$i(V) = k_1 v + k_2 v^{1/2} \quad (4)$$

$$\frac{i(V)}{v^{1/2}} = k_1 v^{1/2} + k_2 \quad (5)$$

where i is the current response at a scan rate of v and k_1 and k_2 are constants. The diffusive contribution is represented by $k_2 v^{1/2}$ and the capacitive contribution by $k_1 v$. It is possible to discern the percentage of capacitive and diffusion charge

storage by looking at the plot of $\left(\frac{i(V)}{v^{1/2}}\right)$ and $v^{1/2}$ at a given potential (V) and finding the slope k_1 and Y-intercept k_2 . Figure S6 shows the percentage contribution of the 5% h-BN_G@MoS₂-90@10 electrode at a potential of 0.2 V . The fraction of the capacitive contribution rises with an increase in scan rate.

The capacitive performance of the electrodes was further investigated with GCD measurements, as shown in Figure 6b. The comparatively symmetrical discharging curves compared to their charging counterparts further demonstrate the electrodes' excellent capacitive qualities. In accordance with the findings of the CV curves, the 5% BN-G@MoS₂-90@10 nanocomposite exhibits a higher charge/discharge curve at current densities of 1 A g^{-1} than the other two composites. The excellent reversibility of the nanocomposite is demonstrated by the charge curves of 5% BN-G@MoS₂-90@10 at different current densities from 1 to 20 A g^{-1} , which are almost symmetrical with their corresponding discharge counterparts, as shown in Figure 6c. Figures S7a and S8a present the GCD of 5% BN-G@MoS₂-50@50 and 5% BN-G@MoS₂-10@90. The specific capacitance values of the 5% BN-G@MoS₂-90@10, 5% BN-G@MoS₂-50@50, and 5% BN-G@MoS₂-10@90 nanocomposites were calculated at various current densities (1 – 20 A g^{-1}) with voltages between -0.4 and 0.6 V using eq 1 and are shown in Figures 6d, S7b, and S8b, respectively (Table S1). The specific capacitances of 5% BN-G@MoS₂-90@10, 5% BN-G@MoS₂-50@50, and 5% BN-G@MoS₂-10@90 nanocomposites at a current density of 1 A g^{-1} are $423, 83,$ and 38 F g^{-1} , respectively, which demonstrated that the electrochemical performance of 5% BN-G@MoS₂-90@10 has been improved dramatically due to the sufficient amount of graphene and uniform network structure, which has high contact with the electrolyte. At current densities of $1, 2, 3, 4, 5, 10,$ and 20 A g^{-1} , respectively, the specific capacitances of 5% BN-G@MoS₂-90@10 are $423, 349, 291, 273, 258, 198,$ and 162 F g^{-1} , where the minimum specific capacitance of 162 F g^{-1} at a current density of 20 A g^{-1} maintains 38% of the initial capacitance at a current density of 1 A g^{-1} . Due to the inability of the ions in the electrolyte to be quickly accessed and removed inside the electrode at high current densities, the electrode uses fewer electroactive species. Additionally, the effective interaction between the ions and the electrode could be gradually reduced with the continually rising current density based on the resistance and departure from the GCD curves that were seen. The cycle stability of the supercapacitor is

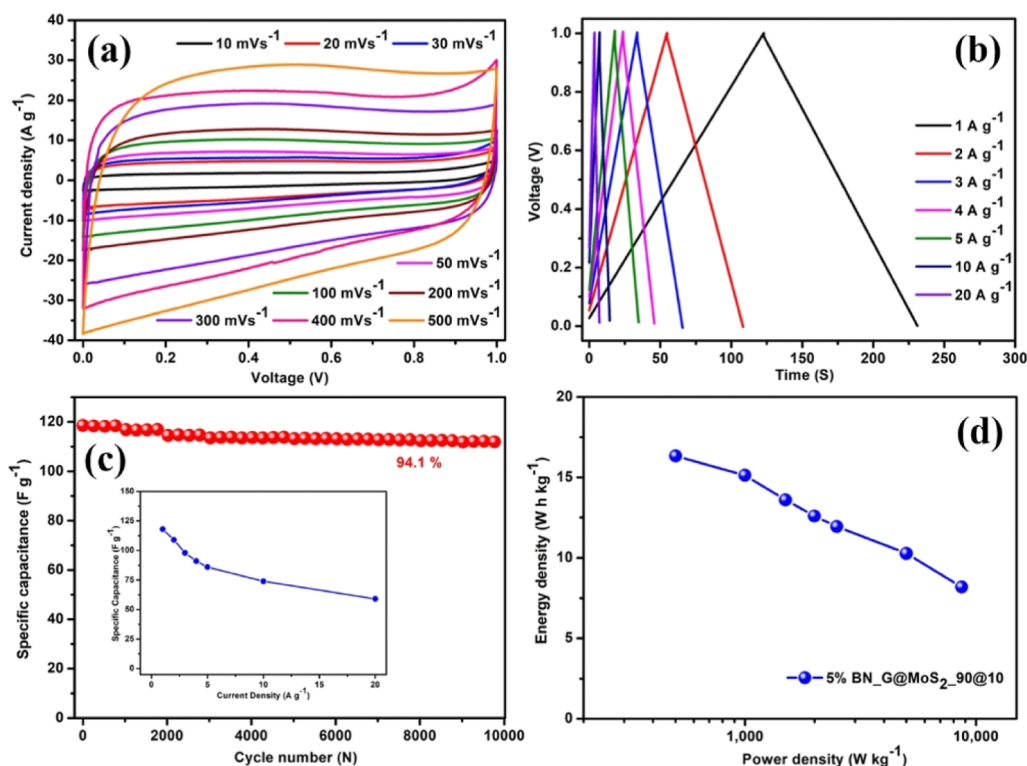


Figure 8. Electrochemical performance of symmetric supercapacitors measured in a two-electrode system. (a) CV curves at different scan rates from 10 to 500 mV s^{-1} ; (b) GCD curves at different current densities from 1 to 20 A g^{-1} ; (c) cycling stability measured at a current density of 1 A g^{-1} for 10,000 cycles; the inset shows the specific capacitances at different current densities from 1 to 20 A g^{-1} ; and (d) Ragone plot related to energy density and power density.

crucial for the nanocomposites made of 5% BN-G@MoS₂-90@10, 5% BN-G@MoS₂-50@50, and 5% BN-G@MoS₂-10@90. It was tested using a galvanostatic charge and discharge study with a current density of 1 A g^{-1} in a solution of 6.0 M KOH for 10,000 cycles, as shown in Figure 7 a. The 5% BN-G@MoS₂-90@10, 5% BN-G@MoS₂-50@50, and 5% BN-G@MoS₂-10@90 nanocomposites exhibit good charge/discharge stability after 10,000 cycles, with capacity retentions of 96.4, 95.1, and 88.1%, respectively. The volume instability of MoS₂ under external current is the root cause of the poor cycling stability of 5% BN-G@MoS₂-50@50 and 5% BN-G@MoS₂-10@90 nanocomposites, which leads to the detachment of MoS₂ nanoparticles from the scaffold network. The specific capacitance conservation for the 5% BN-G@MoS₂-90@10 active material was 96.4%, showing good cycling life that can be due to the synergistic interaction between MoS₂, graphene, and h-BN as well as the practical ion diffusion channels created by KOH activation. These findings demonstrate the 5% BN-G@MoS₂-90@10 composite's capacity to meet the requirements of a long lifetime cycle and good rate capability, which are essential for usable energy storage devices.

One of the primary techniques for assessing the fundamental characteristics of electrode materials is EIS, which analyzes ion diffusion properties, internal resistance, and charge-transfer kinetics. EIS was conducted on the electrode materials 5% BN-G@MoS₂-90@10, 5% BN-G@MoS₂-50@50, and 5% BN-G@MoS₂-10@90 in the frequency range of 10 kHz–1 Hz before and after cycles in Figures 7b and S9. The depressed semicircle and the linear feature region in the high- and low-frequency regions of all materials, respectively, were clearly visible in the Nyquist plots. The intercepts on the x -axis displayed equivalent series resistance (R_s) matched with the internal

resistance of the electrode active material, the electrolytic resistance, and the resistance at the interface between the current collector and the active material. The charge-transfer resistance (R_{ct}) and the high-frequency semicircle's diameter were compatible.⁵³ The R_s values of 5% BN-G@MoS₂-90@10, 5% BN-G@MoS₂-50@50, and 5% BN-G@MoS₂-10@90 before and after cycles are 0.42, 0.75, 0.86, and 0.45, 0.80, and 0.90 Ω , respectively. The R_{ct} values of the nanocomposite's electrodes before and after cycles are 0.72, 0.85, and 1.61 and 0.40, 0.91, and 2.85 Ω (Table S2), respectively, by the calculation with the Nyquist plots. We can find that both R_s and R_{ct} of 5% BN-G@MoS₂-90@10 are lower than those of 5% BN-G@MoS₂-50@50 and 5% BN-G@MoS₂-10@90, indicating that the electrical conductivity of MoS₂ has been improved after combination with the excellent conductive graphene. The adequate percentage of graphene in 5% BN-G@MoS₂-90@10 composition blocks the aggregation of MoS₂ nanoparticles and increases the efficient diffusion of electrolyte ions within the electrode.

Experiments were conducted by utilizing a two-electrode symmetric arrangement to assess the possible use of 5% BN-G@MoS₂-90@10 in electrochemical supercapacitors. As shown in Figure 8, the electrochemical performance was examined at room temperature. Figure 8 shows the CV profile of the as-prepared symmetric cell with a 5% BN-G@MoS₂-90@10 electrode at a potential range of 0–1.0 V in 6 M KOH electrolyte. The CV curves of the symmetric supercapacitor at different scan rates exhibit quasi rectangular shapes. The CV profile clearly shows that the 5% BN-G@MoS₂-90@10 electrode is stable at all scan rates and that EDLC, rather than pseudo capacitance, is the major source of capacitance. The GCD curves at various current densities (Figure 8 b)

show symmetrical triangle curves during charging and discharging in addition to good linear voltage/time profiles, illustrative of the device's exceptional capacitive performance and quick, reversible Faradaic reactions. The specific capacitances of the supercapacitor were calculated from its GCD curves, as shown in Figure 8c (inset). The supercapacitor delivered high capacitances of 118 F g⁻¹ at a current density of 1 A g⁻¹ and 59 F g⁻¹ at a current density 20 A g⁻¹ and maintains 50% of the initial capacitance at a current density 1 A g⁻¹. As shown in Figure 8c, the stability of the device was evaluated using the galvanostatic charge/discharge process at a current density of 1 A g⁻¹. After 10,000 cycles, the device still has roughly 94.1% of its initial capacitance. Graphene and h-BN are responsible for the material's exceptional electrochemical stability, which provides structural support during charging and discharging.

The energy and power densities of SCs are two crucial factors because they show how effectively they can be used as a power source. Equations 2 and 3 were used to get the associated energy density (*E*) and power density (*P*) based on GCD curves of various current densities, as illustrated in Figure 8d. 5% BN-G@MoS₂-90@10 device and those of available energy storage devices reported in the literature are compared in a Ragone diagram.^{18,20,50,51,54–56} The synergistic effect of graphene, h-BN, and MoS₂ nanoparticles, where graphene serves as the conductive network, h-BN as the adhesion layer, and MoS₂ nanoparticles provide active sites for charge storage, may be responsible for the excellent electrochemical performance of the 5% BN-G@MoS₂-90@10 electrode. The maximum energy density of the 5% BN-G@MoS₂-90@10 supercapacitors is 16.33 W h/kg (power density is 501 W/kg), and the power density is 8671.76 W/kg.

5. CONCLUSIONS

In conclusion, 5% BN-G@MoS₂-90@10, 5% BN-G@MoS₂-50@50, and 5% BN-G@MoS₂-10@90 composites were prepared using the ball milling procedure and the precursor materials h-BN, MoS₂, and graphene. The supercapacitor's electrode material, 5% BN-G@MoS₂-90@10, has a high specific capacitance of 423 F g⁻¹ at a current density of 1 A g⁻¹ in 6 M KOH aqueous electrolyte and a capacitance retention of 96.4% after 10,000 cycles. A composite made of graphene and MoS₂ nanoparticles has outstanding cycling stability and rate capabilities. High specific capacitance (118 F g⁻¹), outstanding cycle performance (the capacity retains 94.1% after 10,000 cycles), and high energy and power densities are all advantages of the symmetric SC that was made with the 5% BN-G@MoS₂-90@10 composite. The synergistic interaction of MoS₂, graphene, and h-BN was said to be responsible for the superior electrochemical performances. According to the findings, the 5% BN-G@MoS₂-90@10 electrode has a wide range of potential applications in the field of energy storage devices.

■ ASSOCIATED CONTENT

SI Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acsomega.3c09877>.

EDX mapping analysis, SEM image, and different percentages of atoms in the composite; HRTEM images; XPS spectra; CV profiles; log vs log(*v*) curves; separation of capacitance and diffusion capacity; GCD

curves; Nyquist plots; specific capacitance; and variation of impedance (PDF)

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Notes

The authors declare no competing financial interest.

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