



Article

Area-Scalable 10^9 -Cycle-High-Endurance FeFET of Strontium Bismuth Tantalate Using a Dummy-Gate Process

Mitsue Takahashi * and Shigeki Sakai

National Institute of Advanced Industrial Science and Technology, 1-1-1 Umezono, Tsukuba, Ibaraki 305-8568, Japan; shigeki.sakai@aist.go.jp

* Correspondence: Mitsue-takahashi@aist.go.jp

Abstract: Strontium bismuth tantalate (SBT) ferroelectric-gate field-effect transistors (FeFETs) with channel lengths of 85 nm were fabricated by a replacement-gate process. They had metal/ferroelectric/insulator/semiconductor stacked-gate structures of Ir/SBT/HfO₂/Si. In the fabrication process, we prepared dummy-gate transistor patterns and then replaced the dummy substances with an SBT precursor. After forming Ir gate electrodes on the SBT, the whole gate stacks were annealed for SBT crystallization. Nonvolatility was confirmed by long stable data retention measured for 10⁵ s. High erase-and-program endurance of the FeFETs was demonstrated for up to 10⁹ cycles. By the new process proposed in this work, SBT-FeFETs acquire good channel-area scalability in geometry along with lithography ability.

Keywords: FeFET; ferroelectric; nonvolatile; semiconductor memory; SBT



Citation: Takahashi, M.; Sakai, S. Area-Scalable 10^9 -Cycle-High-Endurance FeFET of Strontium Bismuth Tantalate Using a Dummy-Gate Process. *Nanomaterials* **2021**, *11*, 101. <https://doi.org/10.3390/nano11010101>

Received: 14 December 2020

Accepted: 29 December 2020

Published: 4 January 2021

Publisher's Note: MDPI stays neutral with regard to jurisdictional claims in published maps and institutional affiliations.



Copyright: © 2021 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (<https://creativecommons.org/licenses/by/4.0/>).

1. Introduction

Ferroelectric-gate field-effect transistors (FeFETs) comprising SrBi₂Ta₂O₉ (SBT) or Ca_xSr_{1-x}Bi₂Ta₂O₉ (CSBT) ferroelectrics have unique characteristics of high endurance against at least 10⁸ cycles of program and erase operations [1–12]. CSBT is a kind of SBT family which was derived from original SBT by Sr-site substitution with Ca. The material natures of SBT [13–32] and CSBT [33–36] have been intensively studied previously. FeFETs using CSBT with about $x = 0.2$ showed larger memory windows than those with SBT [5]. The invention of long-retention FeFET was first reported in 2002 and consisted of a metal/ferroelectric/insulator/semiconductor (MFIS) stacked-gate structure of Pt/SBT/(HfO₂)_{0.75}(Al₂O₃)_{0.25}(HAO)/Si [37]. Since then, we have investigated characteristics of (C)SBT-FeFETs [1–3,38–44], improved the device performance [4–8,45,46], and developed FeFET-integrated circuits [9–12,47–52]. For improving the single FeFET performance, we succeeded in reducing gate voltage (V_g) from the initial 6–8 [1] to 3.3 V [8]. Another progress was in shrinking gate-metal length (L_m) from the initial 10 μ m [1] to 100 nm [7].

The conventional (C)SBT-FeFETs were formed by etching the gate stacks. By decreasing the FeFET gate length, SBT etching-damage problems [29–32] on the gate-stack sidewalls became significant. Since we recognized that $L_m = 100$ nm was approaching the shortest limit by the conventional method based on etching, we changed the fabrication strategy to shape the gate stacks from etching-down to filling-up. The new (C)SBT-FeFET process is outlined as follows: Dummy-gate transistor patterns with self-aligned source- and drain regions are prepared in advance. The dummy substance is selectively removed to leave grooves which are later filled up with SBT precursor. Gate electrodes are formed. Finally, whole gate stacks of Ir/SBT/HfO₂/Si are annealed for SBT crystallization. In the new FeFET process, the (C)SBT sidewall of the gate stack is not exposed to etching plasma. The sidewall is thus free from etching damage problem [6]. Consequently, the ferroelectric becomes more controllable in terms of quality and more scalable in terms of geometry than by the etching. The new FeFET dimensions follow good lithography progress with an

adequate height of (C)SBT to show large memory windows increasing with the ferroelectric thickness [3,43]. In this work, SBT-FeFETs with gate channel lengths $L_{ch} = 85$ nm were first reported by adopting the proposed process. Excellent characteristics were demonstrated such as 10^9 cycle erase-program endurance and long stable retention for 10^5 s. The endurance and retention were as good as those of the conventional (C)SBT-FeFETs formed by the gate-stack etching [1–12].

2. Materials and Methods

2.1. Device Fabrication Process

The fabrication process (schematic drawings shown in Figure 1) in this work is as follows:

- *Step 1: Si substrate preparation.* A *p*-type Si substrate patterned with FET active areas was prepared. Local-oxidation-of-silicon (LOCOS) process was used in the patterning for device isolation. The LOCOS patterns with various channel widths (W) were designed in a sample chip. Areas for source-, drain- and substrate-contact holes on the Si were heavily ion-doped. Sacrificial SiO_2 on Si was removed with buffered hydrogen fluoride.
- *Step 2: Insulator deposition.* A 5 nm thick HfO_2 was deposited on the Si substrate by a large-area pulsed-laser deposition system (Vacuum Products Corporation, Kodaira, Tokyo, Japan) [53]. A KrF laser was irradiated on a ceramic HfO_2 target in 15.3 Pa N_2 ambient [54]. The substrate temperature was 220 °C.
- *Step 3: Lithography.* Electron-beam (EB) lithography was performed by spin-coating an organic resist, exposing 130 kV EB, and developing. Resist patterns 550 nm tall were left on the HfO_2/Si . They were later used as ion-implantation mask in *Step 4* and as dummy gates in *Step 7*.
- *Step 4: Ion implantation.* HfO_2 uncovered with resist was etched out by inductively-coupled-plasma reactive-ion etching (ICP-RIE). On the exposed Si, As^+ ions were implanted for source and drain. The energy and dose conditions were 4 keV and $5.0 \times 10^{12}/\text{cm}^2$.
- *Step 5: SiO_2 deposition.* An 830 nm thick SiO_2 was deposited to cover the resist patterns on the substrate by 300 W rf sputtering in 0.1 Pa Ar.
- *Step 6: Flattening SiO_2 .* The SiO_2 was etched back and flattened by ICP-RIE with 1.0 Pa Ar- CF_4 mixed gas until tops of the resists or dummy gates were exposed.
- *Step 7: Leaving grooves on gates.* The dummy-gate substances were selectively removed by O_2 plasma ashing. There remained grooves in a 410 nm tall SiO_2 isolation. The grooves were located on the HfO_2 with self-aligned source and drain regions prepared in *Step 4*. The whole chip was rapidly annealed at 800 °C in ambient N_2 .
- *Step 8: Ferroelectric deposition.* SBT precursor film was deposited to fill up the grooves by a metal-organic-chemical-vapor deposition (MOCVD) system (WACOM R&D, Nihonbashi, Tokyo, Japan). Sources of $\text{Bi}(\text{C}_5\text{H}_{11}\text{O}_2)_3$, $\text{Sr}[\text{Ta}(\text{OC}_2\text{H}_5)_5(\text{OC}_2\text{H}_4\text{OCH}_3)]_2$ and $\text{Ta}(\text{OCH}_2\text{CH}_3)_5$ (Tri Chemical Laboratories Inc., Uenohara, Yamanashi, Japan) were used [6]. As-deposited precursor-film thickness was estimated as 80 nm on a flat place of the substrate.
- *Step 9: Metal deposition.* Ir was deposited by rf sputtering on the SBT precursor layer. Resist mask was patterned for gate electrodes by EB lithography.
- *Step 10: Forming gate electrodes.* Ir uncovered with resist was etched out by Ar^+ ion milling. Then, the resist mask was removed by O_2 plasma ashing.
- *Step 11: FeFET completed.* SBT precursor was deposited again by MOCVD to cover the substrate [6]. The whole substrate was annealed for crystallization of the SBT to show ferroelectricity. The annealing condition was at 780 °C in an O_2 - N_2 mixed gas we investigated before [8]. Finally, contact holes for gate, source, drain and substrate were formed by ultraviolet g-line lithography and Ar^+ ion milling.

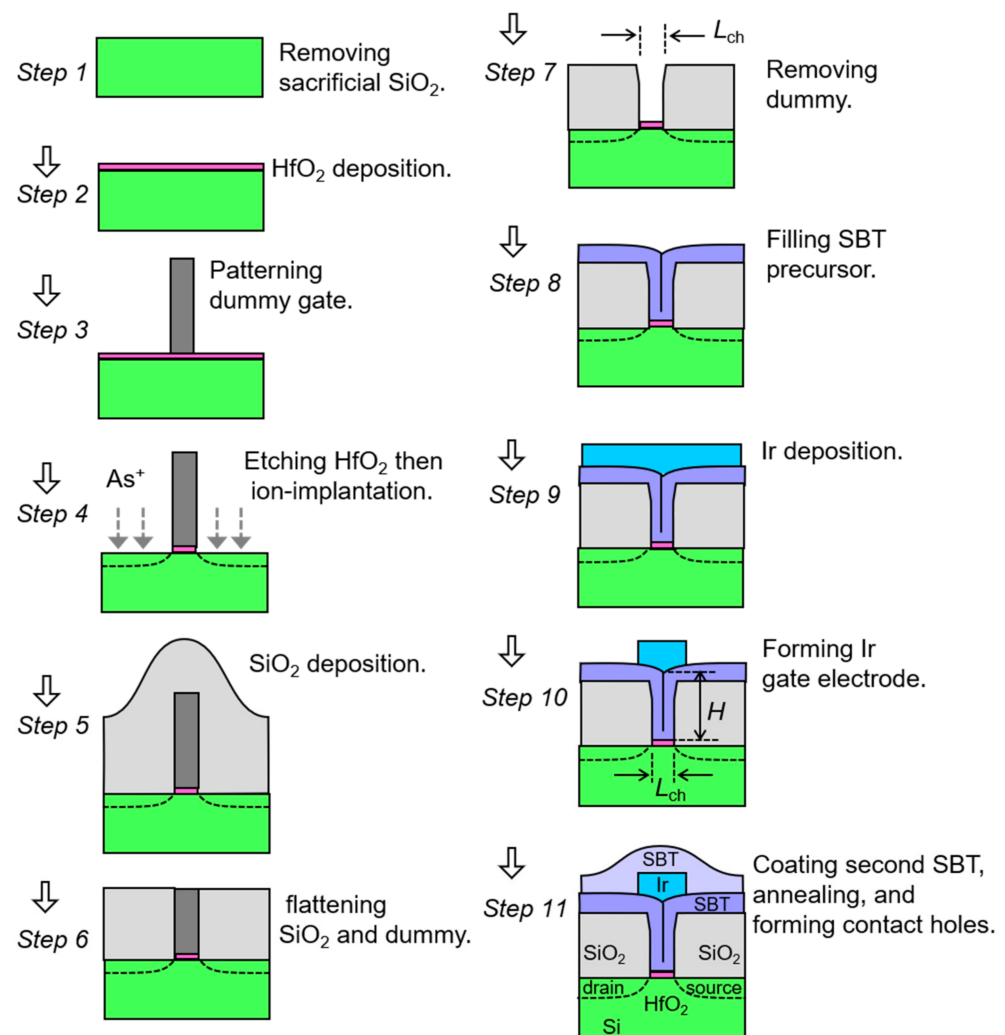


Figure 1. New fabrication process of Strontium bismuth tantalate (SBT)-ferroelectric-gate field-effect transistors (FeFETs) demonstrated in this work.

2.2. Reason for Using SBT in FeFET

The gate stack of MFIS should be regarded as MFI(IL)S, as shown in in Figure 2a, where F, I, IL, S are connected in series. The IL is an interfacial layer between I and S which is formed during the ferroelectric crystallization annealing process of FeFETs [8,39,55–57]. The main component of IL is silicon dioxide with an electric permittivity (ϵ_{IL}) of $\epsilon_{IL} = 3.9$. In the MFI(IL)S, $|P_F| \approx \epsilon_0 \cdot \epsilon_I \cdot |E_I| = \epsilon_0 \cdot \epsilon_{IL} \cdot |E_{IL}| = |Q_S|$ is satisfied in any time. The P_F is ferroelectric polarization. E_I and E_{IL} are electric fields in the I and the IL. The Q_S is charge area density in the semiconductor surface. The ϵ_I is a relative permittivity of the I. The ϵ_0 is the vacuum dielectric constant of $\epsilon_0 = 8.85 \times 10^{-12}$ F/m. For a simplified explanation, we assumed a virtual equivalent circuit of series capacitance as drawn in Figure 2a which is expressed by $|P_F| \approx |Q_I| = |Q_{IL}| = |Q_S|$ with virtual charges Q_I and Q_{IL} on I and IL, respectively. In MFI(IL)S, the IL suffers from a stress of field $|E_{IL}| \approx |P_F| / (\epsilon_0 \cdot \epsilon_{IL}) = 8.7$ MV/cm even at a small $|P_F| = 3 \mu\text{C}/\text{cm}^2$. For example, real IL thickness is 2.6 nm [8] or about 1 nm [55–57]. Electric-field-assisted tunnel current through such a thin SiO_2 [58,59] brings charge injection into the gate stack from S across IL. In erase-and-program operations, a large E_{IL} derived from a large P_F swing induces significant trapped-charge accumulation which accelerates endurance degradations [2,52]. According to our experience [43,52,60], $|P_F|$ should normally be less than $2.5 \mu\text{C}/\text{cm}^2$ all the time and should not exceed $2.0 \mu\text{C}/\text{cm}^2$ for further high-endurance requirements of the FeFET.

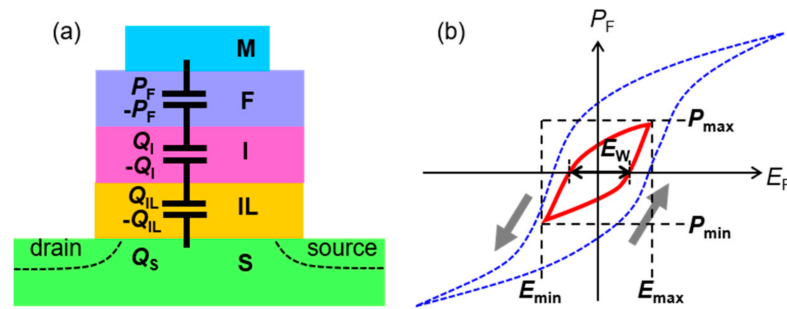


Figure 2. (a) Schematic cross-section of a FeFET with an equivalent circuit of MFI(IL)S gate stack. For convenience of explanation, the circuit is represented using virtual capacitances instead of a strict physical explanation by the electric flux density continuity, D . (b) Schematic drawings of P_F versus E_F . All P_F - E_F loops are drawn in counter-clockwise directions. The inner loop (red solid) is a minor loop corresponding to unsaturated P_F discussed in Section 2.2. Outer loop (blue broken) is a major loop for saturated P_F added as a reference. Every loop has its P_{\max} at E_{\max} and P_{\min} at E_{\min} .

Ferroelectric materials show P_F versus E_F hysteresis loops as illustrated in Figure 2b. The E_F is the electric field across the F. We defined E_{\max} as the positive maximum E_F and P_{\max} as the P_F at $E_F = E_{\max}$. Similarly, E_{\min} and P_{\min} are the negative minimum E_F and the P_F at $E_F = E_{\min}$. The loop is called “major” loop when the E_{\max} and $|E_{\min}|$ are strong enough to force P_F saturated, whereas it is called “minor” loop when P_F is unsaturated by moderate E_F swing. In SBT-FeFETs, restrictions of $P_{\max} \leq 2.5 \mu\text{C}/\text{cm}^2$ corresponding to the minor loops are used during all operations as we emphasized in early works [39,43,52,60].

Regarding a ferroelectric hidden in MFI(IL)S, an exact symmetric swing maximum, i.e., $P_{\max} = |P_{\min}|$ or $E_{\max} = |E_{\min}|$, is difficult because $|Q_S|$ versus Φ_S is very asymmetric [61,62]. The Q_S is the charge area density of the semiconductor surface and Φ_S is the surface potential. Presence of the flat-band voltage V_{fb} makes the symmetric swing further difficult. However, to simplify the physical explanation, $P_{\max} = |P_{\min}|$ and $E_{\max} = |E_{\min}|$ are assumed as shown in Figure 2b with $V_{fb} = 0\text{V}$. In every P_F - E_F loop, the E_F width at $P_F = 0$ is defined as E_w being related with a voltage memory window (V_w) by an approximate expression $E_w = 2E_c = V_w/d_F$, where the E_c is a coercive field and d_F is ferroelectric thickness. According to a method we proposed before [43], an important characteristic E_{\max} of the ferroelectric can be evaluated which has not been measurable by direct probing on a FeFET. If P_{\max} is provided, a gate voltage V_g to achieve a target memory window $V_w = E_w \cdot d_F$ can be estimated as a sum of $E_{\max} \cdot d_F$, $E_I \cdot d_I$, $E_{IL} \cdot d_{IL}$ and Φ_S at $Q_S = P_{\max}$. An exact discussion can be found in the paper [43].

For instance, Pt/SBT/HAO/Si FeFETs showed $E_w = 18 \text{ kV}/\text{cm}$ at $P_{\max} = 2.0 \mu\text{C}/\text{cm}^2$ and $E_{\max} = 25 \text{ kV}/\text{cm}$ [43]. By adopting an advanced process [8], Ir/CSBT/HfO₂/Si FeFETs had the best improved values of $E_w = 65 \text{ kV}/\text{cm}$ at $P_{\max} = 2.0 \mu\text{C}/\text{cm}^2$ and $E_{\max} = 140 \text{ kV}/\text{cm}$ [3,43]. A good reason for using (C)SBT in Si-based FeFETs is the (C)SBT ferroelectric nature of a convenient minor P_F - E_F loop [14,17,20] which has E_w available and is controllable in a restricted P_F range of $P_{\max} \leq 2 \mu\text{C}/\text{cm}^2$ with $E_{\max} \leq 140 \text{ kV}/\text{cm}$.

There are some other ferroelectric materials also intensively studied for applications in Si-based MFIS FeFETs. Regarding Pb₅Ge₃O₁₁ (PGO), attempts to develop replacement-gate-type Pt/PGO/ZrO₂/Si FeFETs were reported [63] but the erase-program-test results of the FeFETs were not found although the ferroelectric itself showed a good potential P_{\max} - E_{\max} and $E_w - E_{\max}$ judging from hysteresis loops of the PGO metal/ferroelectric/metal capacitors [64]. Regarding another candidate, the ferroelectric HfO₂ family [55–57,65–70], the intrinsic material nature may not be suitable for applying to Si-based FeFETs. Informative minor hysteresis loops were reported on Y-doped HfO₂ in which E_w seemed nearly equal to 0 V/cm at $P_{\max} = 2.0 \mu\text{C}/\text{cm}^2$, although it was as large as about 1 MV/cm at $P_{\max} = 10 \mu\text{C}/\text{cm}^2$ [66]. Operation of the FeFETs under the restriction of $P_{\max} \leq 2 \mu\text{C}/\text{cm}^2$ may be difficult. Some reports suggested that HfO₂-FeFETs cannot help using a large P_{\max} ($\gg 2 \mu\text{C}/\text{cm}^2$) [52,55]. The large P_{\max} may induce significant charge injection into the gate

stack. As far as we know, fair works on HfO₂-FeFETs have not cleared 10⁸ cycles endurance in spite of using sophisticated production facilities [56,67–70].

3. Results and Discussion

3.1. Device Dimensions

A cross-sectional scanning-electron-microscope photograph of an Ir/SBT/HfO₂/Si FeFET fabricated by the new proposed process is shown in Figure 3a. Figure 3b shows the same picture added with support lines to clarify the material boundaries. The schematic drawing of the FeFET was assigned with four terminals of gate, drain, source and substrate (Figure 3c). The gate-channel length (L_{ch}) was $L_{ch} = 85$ nm. The gate-channel width was $W = 100$ μm depending on the initial LOCOS pattern designed in Step 1 in Section 2.1. The metal-gate length L_m was 150 nm which could be shorter but was not the focus in this work. The SBT precursor film thickness was about 80 nm measured on a flat place. By filling gate grooves with SBT precursor (Step 8 in Section 2.1.), the effective SBT height (H) was finally about 450 nm which was a distance between Ir and HfO₂. Area scalability of the new FeFET was equivalent to that of the dummy gates which are organic resist patterns made by lithography. From the viewpoint of Si transistor technology, $L_{ch} = 10$ nm is expected to be the critical limit [71]. A significant Curie-temperature decrease in SBT started when particle were sizes of around 20 nm [25]. Thus, the prospective shortest limit of L_{ch} by our proposed FeFET process may be around 20 nm.

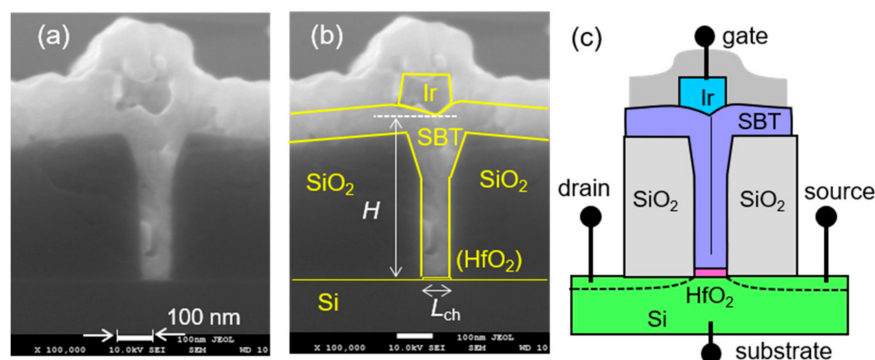


Figure 3. Cross-section of a FeFET with $L_{ch} = 85$ nm fabricated in this work. (a) Original photo by SEM observation and (b) the photo with supporting lines added to clarify material boundaries. (c) Schematic drawing assigned with gate, drain, source and substrate terminals for electrical characterizations.

3.2. Electrical Characterizations

In this study, memory windows, endurance and retention of FeFETs were investigated at room temperature. A semiconductor parameter analyzer (4156C, Keysight Technologies, Santa Rosa, CA, USA) was used for measuring static drain current versus gate voltage (I_d - V_g) curves of the FeFETs. A pulse generator (81110A, Keysight Technologies, Santa Rosa, CA, USA) was used to apply V_g pulses. The instruments were computer-controlled using programs written by the language of LabVIEW (ver. 10, National Instruments, Austin, TX, USA).

3.2.1. Memory Windows

As an elementary test of the FeFETs, I_d - V_g hysteresis loops were investigated (Figure 4). The I_d was measured by V_g increments and decrements with 0.1 V steps. The V_g sweeping ranges were $V_g = 1 \pm 4$ V, 1 ± 5 V and 1 ± 6 V. Drain voltage (V_d), source voltage (V_s) and substrate voltage (V_{sub}) were fixed to $V_d = 0.1$ V and $V_s = V_{sub} = 0$ V during the measurements. The I_d - V_g showed hysteresis loops drawn in counter-clockwise directions because the FeFET was an n -channel-type one. In an I_d - V_g curve, threshold voltage (V_{th}) was defined as a V_g value at $I_d/W = 1 \times 10^{-7}$ A/cm. Two V_{th} values were extracted from the left- and right-side curves in an I_d - V_g hysteresis loop. A memory window was defined

as the V_{th} difference. In this work, we call this a *static* memory window (V_w) because V_g sweep by 4156C is slow. The static V_w was, for instance, 1.0 V by sweeping V_g from -5 to 7 V then back to -5 V, or at $V_g = 1 \pm 6$ V as expressed in Figure 4. During the measurement of a wide-range I_d from 10^{-12} to 10^{-4} A as indicated in Figure 4, V_g sweep speed depends on the current range. Therefore, an I_d - V_g hysteresis curve only gives reference information that is not suitable for accurate discussion.

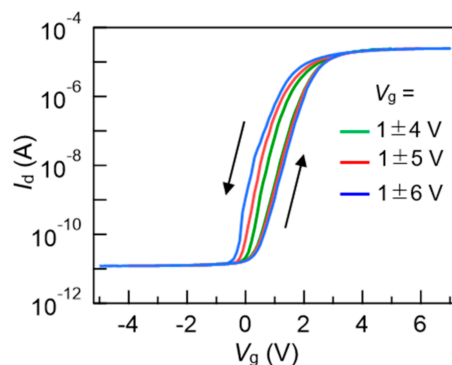


Figure 4. Static static drain current versus gate voltage (I_d - V_g) curves of a FeFET with $L_{ch} = 85$ nm. The channel width was $W = 150$ μ m. V_g ranges were $V_g = 1 \pm 4$ V, 1 ± 5 V and 1 ± 6 V.

For an accurate understanding, the FeFET performance, a pulsed V_g with a controlled time width, was applied to the FeFETs for the erase (*Ers*) or program (*Prg*) operation. The V_g pulse heights with the time widths were (V_E , t_E) for *Ers*, and (V_P , t_P) for *Prg*, respectively. For the n -channel-type FeFET, the V_E was negative ($V_E < 0$ V) and V_P was positive ($V_P > 0$ V) [9]. The pulse time widths t_E and t_P were the same with each other in this work ($t_E = t_P = t_{EP}$). After, *Ers* and *Prg*, I_d - V_g curves were individually measured with a small common V_g range for *Read*. Two V_{th} values were defined in the I_d - V_g curves as the V_g at $I_d/W = 1 \times 10^{-7}$ A/cm. They were expressed as V_{thE} after *Ers* and V_{thP} after *Prg*. The V_{thE} was larger than the V_{thP} [9]. The V_{th} difference of $\Delta V_{th} = V_{thE} - V_{thP}$ was defined as a memory window obtained by read operation after erase-and-program pulse applications. The memory window ΔV_{th} is normally smaller than the above-mentioned static V_w , because slow switching components in a ferroelectric do not respond to short pulses [27,72,73]. The V_{thE} and V_{thP} were investigated by repeating a series of operations: *Ers*, *Read*, *Prg*, *Read*, in this order (Figure 5a). In *Ers*, a pulsed V_g of (V_E , t_{EP}) was applied with keeping $V_d = V_s = V_{sub} = 0$ V. In *Read* after *Ers*, a V_{thE} was extracted from an I_d - V_g curve drawn by narrow-range varying V_g from 0 to 1.1 V at $V_d = 0.1$ V and $V_s = V_{sub} = 0$ V. In *Prg*, a pulsed V_g of (V_P , t_{EP}) was applied, keeping $V_d = V_s = V_{sub} = 0$ V. In *Read* after *Prg*, a V_{thP} was extracted from an I_d - V_g curve drawn under exactly the same conditions as those in *Read* after *Ers*.

Figure 5b shows V_{thE} and V_{thP} by *Read* after *Ers* and *Prg* for three sets of (V_E , t_{EP}) and (V_P , t_{EP}) of $|V_E| = V_P = 6, 7$ and 8 V. Every marker corresponds to the measured V_{thE} and V_{thP} . Memory windows, $\Delta V_{th} = V_{thE} - V_{thP}$, as a function of pulse height $|V_E| = |V_P|$ (Figure 5c) and width t_{EP} (Figure 5d) can be seen in Figure 5b, where the V_{thE} and V_{thP} results (not shown in Figure 5b) of other $V_P (= |V_E|)$ conditions were also used. Short V_g pulses of $t_{EP} = 50$ ns were available for *Ers* and *Prg* of the FeFET. Memory windows of $\Delta V_{th} > 0.7$ V were obtained using 8 and 8.5 V pulses.

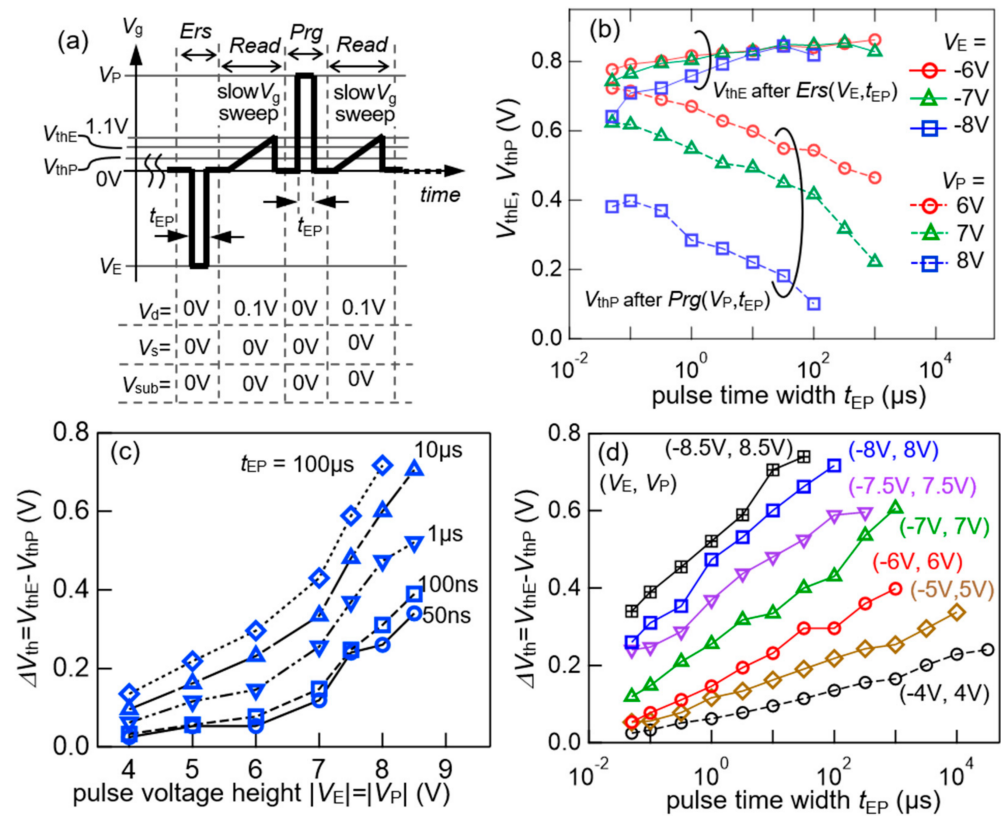


Figure 5. Investigation of V_{thE} and V_{thP} by applying V_g pulses to a FeFET with $L_{ch} = 85$ nm. The channel width was $W = 100 \mu m$. (a) The measurement procedure; (b) measured original V_{thE} and V_{thP} ; (c) pulse-height dependence of $\Delta V_{th} = V_{thE} - V_{thP}$ and (d) pulse width dependence of ΔV_{th} .

Figure 5c,d show a clear monotonic ΔV_{th} increases when raising either the pulse height or width. Good analog V_{thE} and V_{thP} controllability was suggested by smooth and linear ΔV_{th} growths with raising $\log(t_{EP})$ as shown in Figure 5d. The similar tendencies of ΔV_{th} and t_{EP} have already been reported in our previous works [3,5,7,9,52]. In the prior FeFETs, poly-crystallized ferroelectrics were visualized by electron backscatter diffraction (EBSD) [44]. The EBSD indicated that the (C)SBT consisted of multi-grains with various crystal orientations in the FeFETs. The poly-crystallized ferroelectrics may bring the analog V_{thE} and V_{thP} controllability to the FeFETs. In the present FeFET, there must be numerous grains in channel-width direction with $W = 100 \mu m$ whereas a single grain or a few were expected in channel-length with $L_{ch} = 85$ nm which was smaller than average diameters of SBT grains freely grown in-plane [44].

In a preferable geometry of the replacement-gate FeFET in the future, only the channel area $L_{ch} \times W$ will be intensively scaled down with remaining the height H . The H is decided by the gate-groove depth in Step 7 in Section 2.1 and Figure 1. The ΔV_{th} in this report was not yet at its best ability considering the ferroelectric height $H = 450$ nm. In the vertical direction of FeFET, a gate stack by filling SBT should be essentially the same as a large L_{ch} conventional one by etching SBT. Therefore, potential ΔV_{th} will become the same as that of conventional FeFETs by improving the details in the fabrication process in Section 2.1. An immediate target for the present FeFET will be realizing $\Delta V_{th} = 0.7$ V by *Ers* of $(-6V, 10 \mu s)$ and *Prg* of $(6V, 10 \mu s)$ for $H = 190$ nm as demonstrated before using Pt/CSBT/HfO₂/Si FeFETs [7].

3.2.2. Retention

Retention of a FeFET was measured by the procedures as shown in Figure 6a,b. After program (*Prg*), *Retain* and *Read* were repeated during the scheduled time. In *Prg*, a V_g pulse of (V_P, t_{EP}) was applied with $V_d = V_s = V_{sub} = 0$ V. In *Retain*, all the terminals were kept at

zero as $V_g = V_d = V_s = V_{sub} = 0$ V. In *Read* at a certain time t , an I_d - V_g curve was drawn by varying V_g in a narrow range from 0 to 1.0 V at $V_d = 0.1$ V and $V_s = V_{sub} = 0$ V. A V_{thP} was extracted from the I_d - V_g and plotted with a marker at t as shown in Figure 6c. After completing the V_{thP} - t , V_{thE} - t started to be measured. In *erase* (*Ers*), a V_g pulse of (V_E , t_{EP}) was applied with $V_d = V_s = V_{sub} = 0$ V. After *Ers*, *Retain* and *Read* were repeated during the scheduled time. The *Retain* and *Read* conditions for V_{thE} - t were the same as those for V_{thP} - t . In the *Read* at a certain time t , an extracted V_{thE} was plotted with a marker at t as shown in Figure 6c. In this work, $V_P = 8$ V, $V_E = -8$ V and $t_{EP} = 10$ μ s. The retention was measured for 10^5 s in each of V_{thP} - t and V_{thE} - t . At $t = 10^5$ s, they were still distinguishable with a difference $\Delta V_{th} = 0.26$ V. When $t > 10^3$ s, as shown in Figure 6c, the gradient of the V_{thP} - $\log(t)$ and V_{thE} - $\log(t)$ curves appeared to be nearly zero. A possible ten-year retention was suggested by extrapolation lines drawn on the last three markers in each branch. The present $L_{ch} = 85$ nm FeFET showed a good retention to the same extent as those of the conventional (C)SBT FeFETs [1–9,11,12,37–40,42,45,46,52].

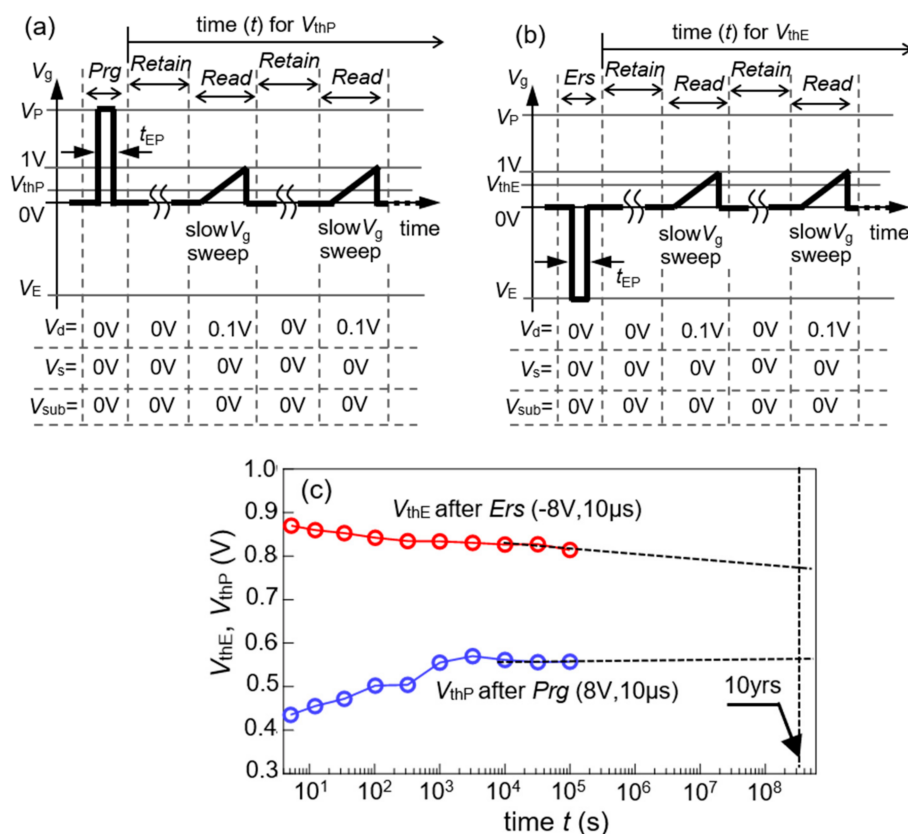


Figure 6. Retention investigation after applying V_g pulses to a FeFET with $L_{ch} = 85$ nm. The channel width was $W = 100$ μ m. The measurement procedures for the retentions of (a) V_{thP} after *Pr*g (V_P , t_{EP}) and (b) V_{thE} after *Ers* (V_E , t_{EP}). (c) The measured retentions for 10^5 s each. Dashed lines are extrapolations of V_{thP} - $\log(t)$ and V_{thE} - $\log(t)$ for estimating V_{thP} and V_{thE} after ten years.

3.2.3. Endurance

Endurance of a FeFET was measured by the procedure shown in Figure 7a. After imposing endurance cycles on FeFETs, pairs of V_{thE} and V_{thP} were obtained. The endurance cycles consisted of periodic bipolar V_g pulses for an alternate *Ers* (V_E , t_{EP}) and *Pr*g (V_P , t_{EP}) with $V_d = V_s = V_{sub} = 0$ V. The endurance-cycle application was interrupted at certain scheduled cycle numbers (N). After the N cycle application, V_{thE} and V_{thP} were read as follows: a series operation of *Ers*, *Read*, *Pr*g, and *Read*, in this order was performed. In *Ers*, a single V_g pulse of (V_E , t_{EP}) was applied with $V_d = V_s = V_{sub} = 0$ V. In *Read* after *Ers*, an I_d - V_g was measured by varying V_g in a narrow range from 0 to 1.5 V at $V_d = 0.1$ V

and $V_s = V_{sub} = 0$ V. A V_{thE} was extracted from the I_d-V_g and plotted with a marker at N as shown in Figure 7b. In *Prgr*, a single V_g pulse of (V_P, t_{EP}) was applied with $V_d = V_s = V_{sub} = 0$ V. In *Read* after *Prgr*, an I_d-V_g was measured under the same conditions with *Read* after *Ers*. The obtained V_{thP} was plotted with a marker at N as shown in Figure 7b.

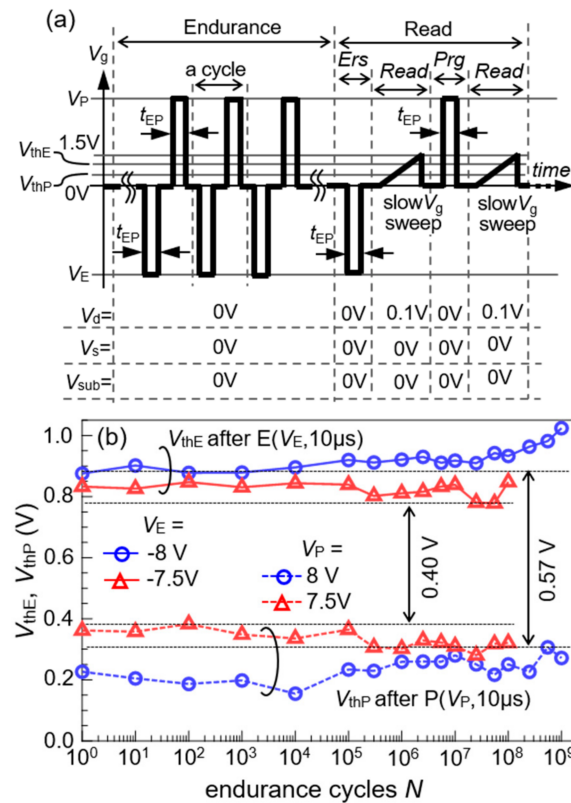


Figure 7. Endurance of a FeFET with $L_{ch} = 85$ nm. The channel width was $W = 80$ μ m. (a) The measurement procedures of applying endurance cycles and reading V_{thE} and V_{thP} . (b) Endurances were measured up to $N = 10^8$ cycles for 7.5 V V_g pulse heights and $N = 10^9$ cycles for 8 V.

As shown in Figure 7b, the *Ers* of $(-7.5$ V, 10 μ s) and *Prgr* of $(7.5$ V, 10 μ s) were first applied for an endurance up to $N = 10^8$ cycles. Next, a stronger input of $(-8$ V, 10 μ s) and $(8$ V, 10 μ s) was applied to the same FeFET up to $N = 10^9$ cycles. No significant shifts of V_{thE} and V_{thP} were observed throughout the measurements. By taking the minimum of the V_{thE} and the maximum of the V_{thP} in the endurance test, $\Delta V_{th} = 0.40$ V for $|V_E| = V_P = 7.5$ V and $\Delta V_{th} = 0.57$ V for $|V_E| = V_P = 8$ V were obtained. These were margins for distinguishing V_{thE} from V_{thP} as indicated in Figure 7b. In spite of using the rather complicated dummy-gate process, the $L_{ch} = 85$ nm FeFET fabricated showed high endurance up to $10^8 \sim 10^9$ cycles. This is the same as the endurance level that (C)SBT-FeFETs inherently have [1–12].

4. Summary

A new fabrication process of a FeFET was proposed and demonstrated. Dummy-gate patterns with self-aligned sources and drains were prepared on a Si substrate. HfO_2 with a thickness of 5 nm was inserted in advance between the dummy-gate substance and the Si substrate. The dummy substance was selectively removed to form a self-aligned groove on the gate. A thin SBT precursor film was deposited to fill up the groove. After forming the Ir gate electrode on the SBT, the whole gate stack was annealed for the SBT crystallization. The finished FeFET of Ir/SBT/ HfO_2 /Si had a channel length $L_{ch} = 85$ nm. The FeFET exhibited a 10^9 cycle-high endurance and long stable retentions measured for 10^5 s. By adopting the replacement-gate process, area-scalable SBT-FeFETs with the high endurance and long retention were successfully produced.

Author Contributions: Conceptualization followed by numerous improvements with respect to the device structure and processing, M.T.; the anneal and MOCVD processes, S.S.; SEM observation, M.T.; creation of PC-controlled measurement programs, S.S.; electrical measurement of the devices, S.S.; data analysis and discussion, S.S. and M.T.; writing—original draft preparation, M.T.; writing—review and editing, M.T. and S.S. All authors have read and agreed to the published version of the manuscript.

Funding: This research received no external funding.

Institutional Review Board Statement: Not applicable.

Informed Consent Statement: Not applicable.

Data Availability Statement: Not applicable.

Acknowledgments: This work was partially supported by WACOM R&D Corporation. We used the EB lithography system in the NPF of AIST, supported by MEXT, Japan.

Conflicts of Interest: The authors declare no conflict of interest.

References

1. Sakai, S.; Ilangoan, R. Metal-ferroelectric-insulator-semiconductor memory FET with long retention and high endurance. *IEEE Electron Device Lett.* **2004**, *25*, 369–371. [[CrossRef](#)]
2. Yan, K.; Takahashi, M.; Sakai, S. Electrical properties of ferroelectric-gate FETs with SrBi₂Ta₂O₉ formed using MOCVD technique. *Appl. Phys. A Mater. Sci. Process.* **2012**, *108*, 835–842. [[CrossRef](#)]
3. Sakai, S.; Zhang, W.; Takahashi, M. Dynamic analog characteristics of 10⁹ cycle-endurance low-voltage nonvolatile ferroelectric-gate memory transistors. In Proceedings of the 2017 IEEE 9th International Memory Workshop, Monterey, CA, USA, 14–17 May 2017; pp. 95–98. [[CrossRef](#)]
4. Hai, L.V.; Takahashi, M.; Sakai, S. Fabrication and characterization of sub-0.6-μm ferroelectric-gate field-effect transistors. *Semicond. Sci. Technol.* **2010**, *25*, 115013. [[CrossRef](#)]
5. Zhang, W.; Takahashi, M.; Sakai, S. Electrical properties of Ca_xSr_{1-x}Bi₂Ta₂O₉ ferroelectric-gate field-effect transistors. *Semicond. Sci. Technol.* **2013**, *28*, 085003. [[CrossRef](#)]
6. Hai, L.V.; Takahashi, M.; Zhang, W.; Sakai, S. Novel process for widening memory window of sub-200nm ferroelectric-gate field-effect transistor by ferroelectric coating the gate-stack sidewall. *Semicond. Sci. Technol.* **2015**, *30*, 015024. [[CrossRef](#)]
7. Hai, L.V.; Takahashi, M.; Zhang, W.; Sakai, S. 100-nm-size ferroelectric-gate field-effect transistor with 10⁸-cycle endurance. *Jpn. J. Appl. Phys.* **2015**, *54*, 088004. [[CrossRef](#)]
8. Zhang, W.; Takahashi, M.; Sasaki, Y.; Kusuhara, M.; Sakai, S. 3.3 V write-voltage Ir/Ca_{0.2}Sr_{0.8}Bi₂Ta₂O₉/HfO₂/Si ferroelectric-gate field-effect transistors with 10⁹ endurance and good retention. *Jpn. J. Appl. Phys.* **2017**, *56*, 04CE04. [[CrossRef](#)]
9. Sakai, S.; Takahashi, M.; Takeuchi, K.; Li, Q.-H.; Horiuchi, T.; Wang, S.; Yun, K.-Y.; Takamiya, M.; Sakurai, T. Highly scalable Fe(ferroelectric)-NAND cell with MFIS(metal-ferroelectric-insulator-semiconductor) structure for sub-10nm tera-bit capacity NAND flash memories. In Proceedings of the 2008 Joint Non-Volatile Semiconductor Memory Workshop and International Conference on Memory Technology and Design, Opio, France, 18–22 May 2008; pp. 103–105. [[CrossRef](#)]
10. Zhang, X.; Miyaji, K.; Takahashi, M.; Takeuchi, K.; Sakai, S. 0.5V bit-line-voltage self-boost-programming in ferroelectric-NAND flash memory. In Proceedings of the 2011 3rd IEEE International Memory Workshop, Monterey, CA, USA, 22–25 May 2011; pp. 155–158. [[CrossRef](#)]
11. Zhang, X.; Takahashi, M.; Takeuchi, K.; Sakai, S. 64 kbit ferroelectric-gate-transistor-integrated NAND flash memory with 7.5 V program and long data retention. *Jpn. J. Appl. Phys.* **2012**, *51*, 04DD01. [[CrossRef](#)]
12. Takahashi, M.; Zhang, W.; Sakai, S. High-endurance ferroelectric NOR flash memory using (Ca,Sr)Bi₂Ta₂O₉ FeFETs. In Proceedings of the 2018 IEEE 10th International Memory Workshop, Kyoto, Japan, 13–16 May 2018; pp. 58–61. [[CrossRef](#)]
13. De Araujo, C.A.P.; Cuchiari, J.D.; McMillan, L.D.; Scott, M.C.; Scott, J.F. Fatigue-free ferroelectric capacitors with platinum electrodes. *Nature* **1995**, *374*, 627–629. [[CrossRef](#)]
14. Noguchi, T.; Hase, T.; Miyasaka, Y. Analysis of the dependence of ferroelectric properties of strontium bismuth tantalate (SBT) thin film on the composition and process temperature. *Jpn. J. Appl. Phys.* **1996**, *35*, 4900–4904. [[CrossRef](#)]
15. Lettieri, J.; Jia, Y.; Urbanik, M.; Weber, C.I.; Maria, J.-P.; Schlom, D.G.; Li, H.; Ramesh, R.; Uecker, R.; Reiche, P. Epitaxial growth of (001)-oriented and (110)-oriented SrBi₂Ta₂O₉ thin films. *Appl. Phys. Lett.* **1995**, *66*, 221–223. [[CrossRef](#)]
16. Ishikawa, K.; Funakubo, H. Electrical properties of (001)- and (116)-oriented epitaxial SrBi₂Ta₂O₉ thin films prepared by metalorganic chemical vapor deposition. *Appl. Phys. Lett.* **1999**, *75*, 1970–1972. [[CrossRef](#)]
17. Ishikawa, K.; Funakubo, H.; Saito, K.; Suzuki, T.; Nishi, Y.; Fujimoto, M. Crystal structure and electrical properties of epitaxial SrBi₂Ta₂O₉ films. *J. Appl. Phys.* **2000**, *87*, 8018–8023. [[CrossRef](#)]
18. Lee, H.N.; Visinoini, A.; Senz, S.; Harnagea, C.; Pignolet, A.; Hesse, D.; Gösele, U. Structural and electrical anisotropy of (001)-, (116)-, and (103)-oriented epitaxial SrBi₂Ta₂O₉ thin films on SrTiO₃ substrates grown by pulsed laser deposition. *J. Appl. Phys.* **2000**, *88*, 6658–6664. [[CrossRef](#)]

19. Amanuma, K.; Hase, T.; Miyasaka, Y. Preparation and ferroelectric properties of SrBi₂Ta₂O₉ thin films. *Appl. Phys. Lett.* **1995**, *66*, 221–223. [CrossRef]
20. Atsuki, T.; Soyama, N.; Yonezawa, T.; Ogi, K. Preparation of Bi-based ferroelectric thin films by Sol-Gel method. *Jpn. J. Appl. Phys.* **1995**, *34*, 5096–5099. [CrossRef]
21. Robertson, J.; Chen, C.W.; Warren, W.L.; Gutleben, C.D. Electronic structure of the ferroelectric layered perovskite SrBi₂Ta₂O₉. *Appl. Phys. Lett.* **1996**, *69*, 1704–1706. [CrossRef]
22. Harnagea, C.; Pignolet, A.; Alexe, M.; Hesse, D.; Gösele, U. Quantitative ferroelectric characterization of single submicron grains in Bi-layered perovskite thin films. *Appl. Phys. A* **2000**, *70*, 261–267. [CrossRef]
23. Kalinin, S.V.; Gruverman, A.; Bonnell, D.A. Quantitative analysis of nanoscale switching in SrBi₂Ta₂O₉ thin films by piezoresponse force microscopy. *Appl. Phys. Lett.* **2004**, *85*, 795. [CrossRef]
24. Amorín, H.; Shvartsman, V.V.; Kholkin, A.L.; Costa, M.E.V. Ferroelectric and dielectric anisotropy in high-quality SrBi₂Ta₂O₉ single crystals. *Appl. Phys. Lett.* **2004**, *85*, 5667. [CrossRef]
25. Yu, T.; Shen, Z.X.; Toh, W.S.; Xue, J.M.; Wang, J. Size effect on the ferroelectric phase transition in SrBi₂Ta₂O₉ nanoparticles. *J. Appl. Phys.* **2003**, *94*, 618–620. [CrossRef]
26. Gruverman, A. Scaling effect on statistical behavior of switching parameters of ferroelectric capacitors. *Appl. Phys. Lett.* **1999**, *75*, 1452–1454. [CrossRef]
27. Tamura, T.; Arimoto, Y.; Ishiwara, H. A New Circuit Simulation Model of Ferroelectric Capacitors. *Jpn. J. Appl. Phys.* **2002**, *41*, 2654–2657. [CrossRef]
28. Pan, B.; Yu, H.; Wu, D.; Zhou, X.H.; Liu, J.-M. Dynamic response and hysteresis dispersion scaling of ferroelectric SrBi₂Ta₂O₉ thin films. *Appl. Phys. Lett.* **2003**, *83*, 1406–1408. [CrossRef]
29. Lee, W.-J.; Cho, C.-R.; Kim, S.-H.; You, I.-K.; Kim, B.W.; Yu, B.-G.; Shin, C.H.; Lee, H.C. Etching Behavior and Damage Recovery of SrBi₂Ta₂O₉ Thin Films. *Jpn. J. Appl. Phys.* **1999**, *38*, L1428–L1431. [CrossRef]
30. Asami, K.; Koiwa, I.; Yamanobe, T. Effects of Ion Etching and Annealing in O₂ Atmosphere Following Ion Etching on Properties and Chemistry of Sr_{0.9}Bi_{2.1}Ta₂O_{9+α} Thin Films. *Jpn. J. Appl. Phys.* **1999**, *38*, 5423–5427. [CrossRef]
31. Stafford, L.; Margot, J.; Delprat, S.; Chaker, M.; Pearton, S.J. Influence of redeposition on the plasma etching dynamics. *J. Appl. Phys.* **2007**, *101*, 083303. [CrossRef]
32. Efremov, A.M.; Kim, D.-P.; Kim, C.-I. Investigation of thin films etching mechanisms in plasma. *J. Vsc. Sci. Technol. A* **2003**, *21*, 1017–1023. [CrossRef]
33. Shimakawa, Y.; Kubo, Y.; Nakagawa, Y.; Goto, S.; Kamiyama, T.; Asano, H.; Izumi, F. Crystal structure and ferroelectric properties of ABi₂Ta₂O₉ (A = Ca, Sr, and Ba). *Phys. Rev. B* **2000**, *61*, 6559–6564. [CrossRef]
34. Noguchi, Y.; Shimizu, H.; Miyayama, M.; Oikawa, K.; Kamiyama, T. Ferroelectric properties and structure distortion in A-site-modified SrBi₂Ta₂O₉. *Jpn. J. Appl. Phys.* **2001**, *40*, 5812–5815. [CrossRef]
35. Das, R.R.; Bhattacharya, P.; Pe´rez, W.; Katiyar, R.S. Ferroelectric properties of laser-ablated Sr_{1-x}A_xBi₂Ta₂O₉ thin films (where A = Ba, Ca). *Appl. Phys. Lett.* **2002**, *80*, 637–639. [CrossRef]
36. Das, R.R.; Bhattacharya, P.; Pe´rez, W.; Katiyar, R.S. Influence of Ca on structural and ferroelectric properties of laser ablated SrBi₂Ta₂O₉ thin films. *Jpn. J. Appl. Phys.* **2003**, *42*, 162–165. [CrossRef]
37. Development of the 1T FeRAM: Towards the Realization of the Ultra-Gbit Next-Generation Semiconductor Memory. AIST Research Results Archive, 24 October 2002. Available online: https://www.aist.go.jp/aist_e/list/latest_research/2002/20021024/20021024.html (accessed on 14 December 2020).
38. Sakai, S.; Takahashi, M.; Ilangoan, R. Long-retention ferroelectric-gate FET with a (HfO₂)_x(Al₂O₃)_{1-x} buffer-insulating layer for 1T FeRAM. In Proceedings of the 2004 IEDM Technical Digest. IEEE International Electron Devices Meeting, San Francisco, CA, USA, 13–15 December 2004; pp. 915–918. [CrossRef]
39. Sakai, S.; Ilangoan, R.; Takahashi, M. Pt/SrBi₂Ta₂O₉/Hf-Al-O/Si field-effect-transistor with long retention using unsaturated ferroelectric polarization switching. *Jpn. J. Appl. Phys.* **2004**, *43*, 7876–7878. [CrossRef]
40. Li, Q.H.; Sakai, S. Characterization of Pt/SrBi₂Ta₂O₉/Hf-Al-O/Si field-effect transistors at elevated temperatures. *Appl. Phys. Lett.* **2006**, *89*, 222910. [CrossRef]
41. Li, Q.H.; Takahashi, M.; Horiuchi, T.; Wang, S.Y.; Sakai, S. Threshold-voltage distribution of Pt/SrBi₂Ta₂O₉/Hf-Al-O/Si MFIS FETs. *Semicond. Sci. Technol.* **2008**, *23*, 045011. [CrossRef]
42. Li, Q.H.; Horiuchi, T.; Wang, S.Y.; Takahashi, M.; Sakai, S. Threshold voltage adjustment of ferroelectric-gate field effect transistors by ion implantation. *Semicond. Sci. Technol.* **2009**, *24*, 025012. [CrossRef]
43. Sakai, S.; Zhang, W.; Takahashi, M. Method for disclosing invisible physical properties in metal-ferroelectric-insulator-semiconductor gate stacks. *J. Phys. D Appl. Phys.* **2017**, *50*, 165107. [CrossRef]
44. Zhang, W.; Takahashi, M.; Sakai, S. Investigation of ferroelectric grain sizes and orientations in Pt/Ca_xSr_{1-x}Bi₂Ta₂O₉/Hf-Al-O/Si high performance ferroelectric-gate field-effect-transistors. *Materials* **2019**, *12*, 399. [CrossRef]
45. Takahashi, M.; Sakai, S. Self-aligned-gate metal/ferroelectric/insulator/semiconductor field-effect transistors with long memory retention. *Jpn. J. Appl. Phys.* **2005**, *44*, L800–L802. [CrossRef]
46. Horiuchi, T.; Takahashi, M.; Li, Q.H.; Wang, S.Y.; Sakai, S. Lowered operation voltage in Pt/SBi₂Ta₂O₉/HfO₂/Si ferroelectric-gate field-effect transistors by oxynitriding Si. *Semicond. Sci. Technol.* **2010**, *25*, 055005. [CrossRef]

47. Takahashi, M.; Horiuchi, T.; Li, Q.H.; Wang, S.Y.; Sakai, S. Basic operation of novel ferroelectric CMOS circuits. *Electron. Lett.* **2008**, *44*, 467–468. [[CrossRef](#)]
48. Takahashi, M.; Wang, S.Y.; Horiuchi, T.; Sakai, S. FeCMOS logic inverter circuits with nonvolatile-memory function. *IEICE Electron. Express* **2009**, *6*, 831–836. [[CrossRef](#)]
49. Wang, S.Y.; Takahashi, M.; Li, Q.H.; Takeuchi, K.; Sakai, S. Operational method of a ferroelectric (Fe)-NAND flash memory array. *Semicond. Sci. Technol.* **2009**, *24*, 105029. [[CrossRef](#)]
50. Miyaji, K.; Noda, S.; Hatanaka, T.; Takahashi, M.; Sakai, S.; Takeuchi, K. A 1.0 V power supply, 9.5 GByte/sec write speed, Single-Cell Self-Boost program scheme for Ferroelectric NAND Flash SSD. In Proceedings of the 2010 IEEE International Memory Workshop, Seoul, South Korea, 16–19 May 2010; pp. 1–4. [[CrossRef](#)]
51. Zhang, X.; Takahashi, M.; Sakai, S. FeFET logic circuits for operating a 64 kb FeNAND flash memory array. *Integr. Ferroelectr.* **2012**, *132*, 114–121. [[CrossRef](#)]
52. Sakai, S.; Zhang, X.Z.; Hai, L.V.; Zhang, W.; Takahashi, M. Downsizing and memory array integration of Pt/SrBi₂Ta₂O₉/Hf-Al-O/Si ferroelectric-gate field-effect transistors. In Proceedings of the 2012 12th Annual Non-Volatile Memory Technology Symposium, Singapore, 31 October–2 November 2012; pp. 55–59. [[CrossRef](#)]
53. Sakai, S.; Takahashi, M.; Motohashi, K.; Yamaguchi, Y.; Yui, N.; Kobayashi, T. Large-area pulsed-laser deposition of dielectric and ferroelectric thin films. *J. Vac. Sci. Technol. A* **2007**, *25*, 903–907. [[CrossRef](#)]
54. Sakai, S. Semiconductor-Ferroelectric Storage Devices and Processes for Producing The Same. U.S. Patent 7,226,795, 2007.
55. Böske, T.S.; Müller, J.; Bräuhäus, D.; Schröder, U.; Böttger, U. Ferroelectricity in hafnium oxide: CMOS compatible ferroelectric field effect transistors. In Proceedings of the 2011 IEDM Technical Digest. IEEE International Electron Devices Meeting, Washington, DC, USA, 5–7 December 2011; pp. 24.5.1–24.5.4. [[CrossRef](#)]
56. Müller, J.; Yurchuk, E.; Schlösser, T.; Paul, J.; Hoffmann, R.; Müller, S.; Martin, D.; Slesazek, S.; Polakowski, P.; Sundqvist, J.; et al. Ferroelectricity in HfO₂ enables nonvolatile data storage in 28 nm HKMG. In Proceedings of the 2012 Symposium on VLSI Technology, Honolulu, HI, USA, 12–14 June 2012; pp. 25–26. [[CrossRef](#)]
57. Ali, T.; Polakowski, P.; Riedel, S.; Büttner, T.; Kämpfe, T.; Rudolph, M.; Pätzold, B.; Seidel, K.; Löhr, D.; Hoffmann, R.; et al. Silicon doped hafnium oxide (HSO) and hafnium zirconium oxide (HZO) based FeFET: A material relation to device physics. *Appl. Phys. Lett.* **2018**, *112*, 222903. [[CrossRef](#)]
58. Fukuda, M.; Mizubayashi, W.; Kohno, A.; Miyazaki, S.; Hirose, M. Analysis of Tunnel Current through Ultrathin Gate Oxides. *Jpn. J. Appl. Phys.* **1998**, *37*, L1534–L1536. [[CrossRef](#)]
59. Shrenk, A.; Heiser, G. Modeling and simulation of tunneling through ultra-thin gate dielectrics. *J. Appl. Phys.* **1997**, *81*, 7900–7908. [[CrossRef](#)]
60. Takahashi, M.; Sakai, S. 2.3.3 Requirements to the F Layer, Chap.2 Development of high-endurance and long-retention FeFETs of Pt/Ca_ySr_{1-y}Bi₂Ta₂O₉/(HfO₂)_x(Al₂O₃)_{1-x}/Si gate stacks. In *Ferroelectric-Gate Field Effect Transistor Memories*, 2nd ed.; Park, B.-E., Ishiwara, H., Okuyama, M., Sakai, S., Yoon, S.-M., Eds.; Springer: Singapore, Singapore, 2020; pp. 35–37. [[CrossRef](#)]
61. Kingston, R.; Neustadter, S.F. Calculation of the Space Charge, Electric Field, and Free Carrier Concentration at the Surface of a Semiconductor. *J. Appl. Phys.* **1955**, *26*, 718–720. [[CrossRef](#)]
62. Sze, S.M. Chap.7, MIS Diode and Charge-Coupled Device. In *Physics of Semiconductor Devices*, 2nd ed.; John Wiley & Sons: New York, NY, USA, 1981; pp. 366–369.
63. Zhang, F.; Hsu, S.T.; Ono, Y.; Ulrich, B.; Zhuang, W.; Ying, H.; Stecker, L.; Evans, D.R.; Maa, J. Fabrication and characterization of sub-micron metal-ferroelectric-insulator-semiconductor field effect transistors with Pt/Pb₅Ge₃O₁₁/ZrO₂/Si structure. *Jpn. J. Appl. Phys.* **2001**, *40*, L635–L637. [[CrossRef](#)]
64. Li, T.; Hsu, S.T.; Ulrich, B.; Ying, H.; Stecker, L.; Evans, D.; Ono, Y.; Maa, J.; Lee, J.J. Fabrication and characterization of a Pb₅Ge₃O₁₁ one-transistor-memory device. *Appl. Phys. Lett.* **2001**, *79*, 1661–1663. [[CrossRef](#)]
65. Müller, J.; Böske, T.S.; Schröder, U.; Mueller, S.; Bräuhäus, D.; Böttger, U.; Frey, L.; Mikolajick, T. Ferroelectricity in simple binary ZrO₂ and HfO₂. *Nano Lett.* **2012**, *12*, 4318–4323. [[CrossRef](#)]
66. Shimizu, T.; Katayama, K.; Kiguchi, T.; Akama, A.; Konno, T.J.; Sakata, O.; Funakubo, H. The demonstration of significant ferroelectricity in epitaxial Y-doped HfO₂ film. *Sci. Rep.* **2016**, *6*, 32931. [[CrossRef](#)] [[PubMed](#)]
67. Yurchuk, E.; Müller, J.; Hoffmann, R.; Paul, J.; Martin, D.; Boschke, R.; Schlösser, T.; Müller, S.; Slesazek, S.; Bentum, R.; et al. HfO₂-Based Ferroelectric Field-Effect Transistors with 260 nm Channel Length and Long Data Retention. In Proceedings of the 2012 IEEE International Memory Workshop, Milan, Italy, 20–23 May 2012; pp. 1–4. [[CrossRef](#)]
68. Trentzsch, M.; Flachowsky, S.; Richter, R.; Paul, J.; Reimer, B.; Utess, D.; Jansen, S.; Mulaosmanovic, H.; Müller, S.; Slesazek, S.; et al. A 28nm HKMG super low power embedded NVM technology based on ferroelectric FETs. In Proceedings of the 2016 IEEE International Electron Devices Meeting, San Francisco, CA, USA, 3–7 December 2016; pp. 11.5.1–11.5.4. [[CrossRef](#)]
69. Mulaosmanovic, H.; Breyer, E.T.; Mikolajick, T.; Slesazek, S. Ferroelectric FETs with 20-nm-Thick HfO₂ Layer for Large Memory Window and High Performance. *IEEE Trans. Electron. Devices* **2019**, *66*, 3828–3833. [[CrossRef](#)]
70. Zhou, H.; Ocker, J.; Mennenga, M.; Noack, M.; Müller, S.; Trentzsch, M.; Dünkler, S.; Beyer, S.; Mikolajick, T. Endurance and targeted programming behavior of HfO₂-FeFETs. In Proceedings of the 2020 IEEE International Memory Workshop, Dresden, Germany, 17–20 May 2020; pp. 1–4. [[CrossRef](#)]
71. Wong, H.; Iwai, H. On the scaling issues and high-κ replacement of ultrathin gate dielectrics for nanoscale MOS transistors. *Microelectron. Eng.* **2006**, *83*, 1867–1904. [[CrossRef](#)]

-
72. Tagantsev, A.K.; Stolichnov, I.; Setter, N.; Cross, J.S.; Tsukada, M. Non-Kolmogorov-Avrami switching kinetics in ferroelectric thin films. *Phys. Rev. B* **2002**, *66*, 214109. [[CrossRef](#)]
 73. Jo, J.Y.; Yang, S.M.; Kim, T.H.; Lee, H.N.; Yoon, J.-G.; Park, S.; Jo, Y.; Jung, M.H.; Noh, T.W. Nonlinear dynamics of domain-wall propagation in epitaxial ferroelectric thin films. *Phys. Rev. Lett.* **2009**, *102*, 045701. [[CrossRef](#)]