

# Iron-Based Oxygen Scavengers on Mesoporous Silica Nanospheres

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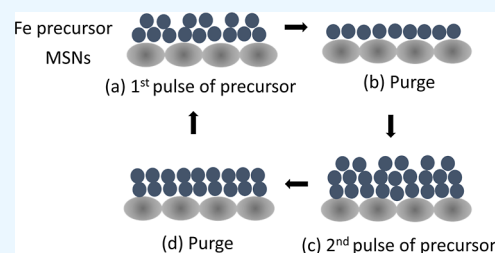


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Supporting Information

**ABSTRACT:** Iron-based materials are among the most commonly used oxygen scavengers. Here, we investigated the mesoporous silica nanosphere (MSN)-supported iron-based scavengers, such as FeO<sub>x</sub> nanoparticles and different atomic layer deposition (ALD) coatings (FeO<sub>x</sub> and Fe). We found that the scavenger performance is a result of a complex interplay between available Brunauer–Emmett–Teller surface area and the scavenger composition, with the combination of infiltrated nanoparticles and Fe-ALD coating resulting in the best performance. When the glucose-based treatment of MSN is used to further enhance oxygen scavenging capacity, Fe-ALD coating yields the best performance, with a high oxygen adsorption capacity of 126.8 mL/g. ALD deposition of Fe represents a versatile method to introduce Fe-based oxygen scavengers onto different supports, and it can facilitate the integration of scavengers with different types of packaging, as the deposition can be performed at a low temperature of 150 °C.



## INTRODUCTION

Oxygen scavengers are of significant interest for food packaging applications.<sup>1–12</sup> Oxidative degradation is one of the main mechanisms of food spoilage, and it results in undesirable changes in the food, such as the degradation of nutritional content of the food (due to the degradation of essential fatty acids, vitamins, and proteins), the degradation of odor and taste (due to rancidity), and color change (due to pigment degradation).<sup>1</sup> To reduce spoilage of the food and increase its shelf-life, oxygen scavengers are commonly used in a variety of packaged foods.<sup>1</sup> The scavengers are commonly based on iron, although other types of scavengers are also available, such as those based on other metals (cobalt, palladium, and platinum), organic acids (gallic acid and ascorbic acid), photosensitive dyes, unsaturated organic molecules, enzymes, spores, and yeasts.<sup>1,2</sup> Iron powder packed in sachets remains the most commonly used scavenger.<sup>2,3</sup> These scavengers commonly require the presence of water for scavenging action.<sup>3</sup> The oxygen scavenging capacity of an iron-based scavenger can be improved if nanoparticles instead of micron-sized particles/conventional iron powder are used.<sup>2,4</sup> The nanoparticles also have an advantage of scavenger operation under both wet and dry conditions.<sup>4</sup> Due to the increased surface area, nanosized iron particles enable fast scavenger response and increased adsorption capacity.<sup>4,5</sup> To suppress unwanted aggregation of nanoparticles, their preparation on supports such as montmorillonite clay<sup>6–9</sup> and kaolinite<sup>10</sup> has also been explored, and integration of zero-valent iron nanoparticles with clay is of significant interest for preparation of improved oxygen barrier materials.<sup>6–10</sup> In addition, mesoporous silica nanospheres (MSN) have been

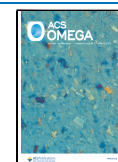
explored for oxygen scavenging applications in our previous work.<sup>11</sup>

While zero valent iron nanoparticles exhibit highly promising oxygen scavenging properties due to high reactivity of Fe<sup>0</sup> and iron-containing nanomaterials are generally considered to have low toxicity, concerns over the toxicity of nanoscale zero-valent iron has been rising in recent years.<sup>13–15</sup> As it is known that the toxicity of iron NPs decreases as their surface gets oxidized,<sup>13</sup> and iron oxide nanomaterials are generally considered non-toxic and of interest for medical applications,<sup>16,17</sup> we explored the possibility of using iron oxide nanoparticles as a more stable and less toxic alternative oxygen scavengers. The prepared nanoparticles were investigated alone and infiltrated inside the MSN. We found that infiltration of nanoparticles inside MSN enables an increase in oxygen adsorption capacity, likely due to suppression of nanoparticle aggregation. Similar performance to FeO<sub>x</sub> NPs can be obtained by atomic layer deposition (ALD) of FeO<sub>x</sub> on MSN, but combination of ALD coating and NP infiltration yields variable results (depending on the type of coating), as the coating increases the amount of iron but decreases the overall surface area available for adsorption. The effect of ALD coating composition on the oxygen scavenging performance is discussed.

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## RESULTS AND DISCUSSION

Figure S1 shows the transmission electron microscopy (TEM) images of  $\text{FeO}_x$  nanoparticles (NPs)<sup>18</sup> and  $\text{FeO}_x$  NPs infiltrated into MSN, while energy-dispersive X-ray (EDX) mapping of  $\text{FeO}_x$  NPs infiltrated into MSN is shown in Figure 1. In addition, samples with  $\text{FeO}_x$  and Fe coatings deposited by

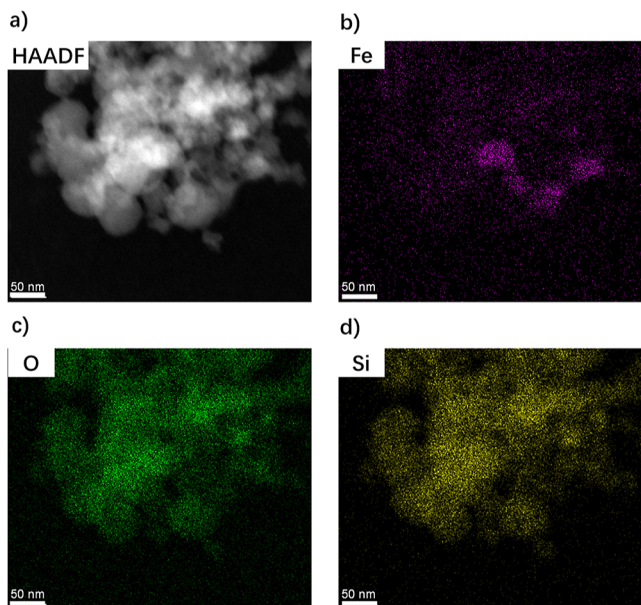


Figure 1. EDX mapping of  $\text{FeO}_x$  NPs in MSN.

ALD were prepared. TEM images and selected area electron diffraction (SAED) images of MSN with different (Fe and  $\text{FeO}_x$ ) ALD coatings,  $\text{FeO}_x$  NPs and  $\text{FeO}_x$  NP samples infiltrated into MSN, and ALD  $\text{FeO}_x$  on  $\text{FeO}_x$  NPs in MSN are shown in Figure 2. From SAED, we can observe that MSN and both types of ALD coatings are amorphous, while samples with NPs are predominantly amorphous with a small amount of the crystalline phase. To identify this phase, XRD measurements have been performed and obtained results are shown in Figure S5. We can see that all samples except samples with  $\text{FeO}_x$  NPs

are indeed amorphous, while  $\text{FeO}_x$  NPs show a weak pattern corresponding to the presence of  $\text{Fe}_2\text{O}_3$ .

To further characterize the composition and structure of the samples, EDX mapping was performed. Obtained results for  $\text{FeO}_x$  and Fe ALD-coated MSN are shown in Figures 3 and 4,

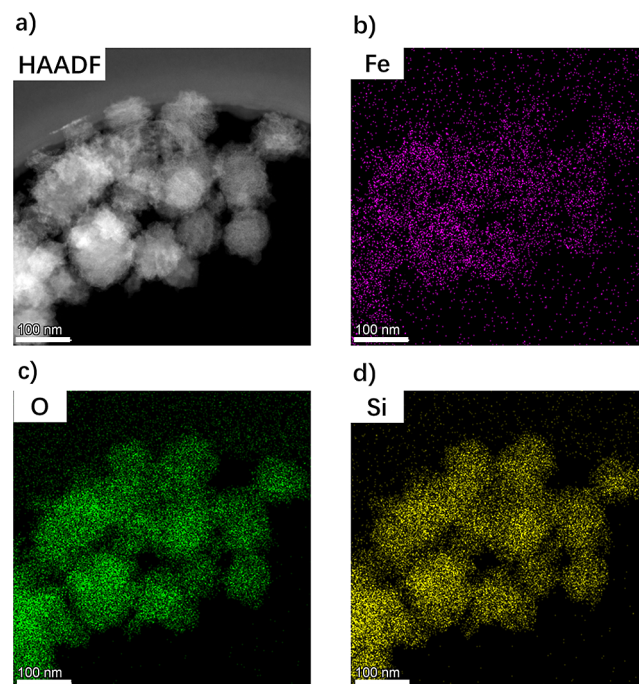


Figure 3. EDX mapping of  $\text{FeO}_x$  ALD on MSN.

respectively, while the EDX mapping of the sample combining MSN,  $\text{FeO}_x$  NPs, and ALD deposition of  $\text{FeO}_x$  coating is shown in Figure S3. HRTEM images of different ALD coatings are shown in Figure 5. We can observe that ALD enables deposition of uniform amorphous coating, in agreement with no evidence of crystallinity obtained from SAED and XRD. From the obtained results, successful incorporation of nanoparticles, as well as coating with the Fe-containing layer by ALD (for both Fe- and  $\text{FeO}_x$ ) is evident. Sample composition and stoichiometry has been investigated by

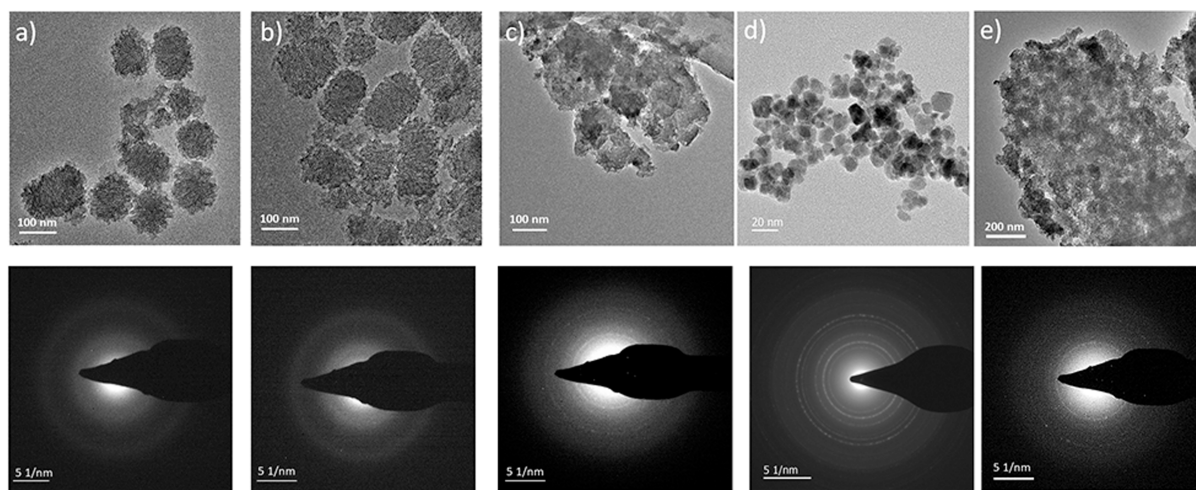


Figure 2. TEM (top) and SAED (bottom) images of (a) ALD  $\text{FeO}_x$  on MSN, (b) ALD Fe on MSN, (c)  $\text{FeO}_x$  NPs in MSN, (d)  $\text{FeO}_x$  NPs, and (e) ALD  $\text{FeO}_x$  on  $\text{FeO}_x$  NPs in MSN.

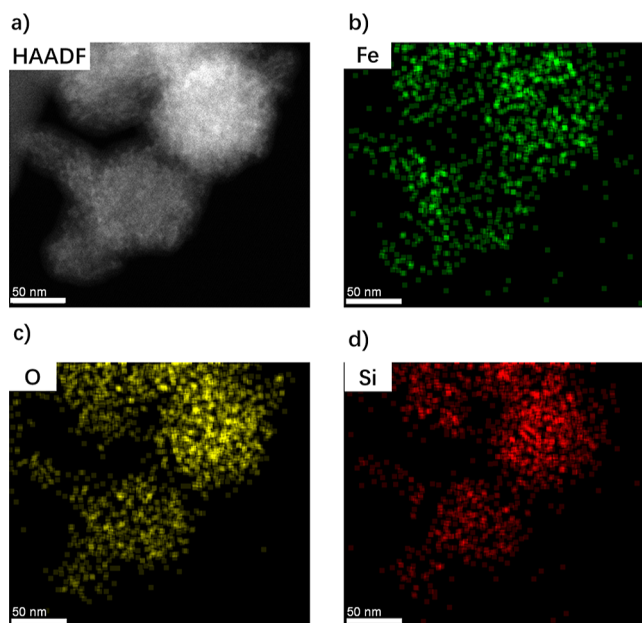


Figure 4. EDX mapping of Fe ALD on MSN.

EDX and X-ray photoelectron spectroscopy (XPS), and obtained results are summarized in Table S1. We can conclude that  $\text{FeO}_x$  NPs consist of predominant amorphous FeO phase with small  $\text{Fe}_2\text{O}_3$  content.  $\text{FeO}_x$  ALD coating likely contains a higher share of  $\text{Fe}_2\text{O}_3$  compared to FeO from a lower Fe/O ratio, while Fe ALD coating yields high Fe/O ratio, likely due to the presence of metallic Fe.

To further evaluate the potential of different samples for oxygen scavenging applications Brunauer–Emmett–Teller (BET) surface areas and oxygen scavenging performance were determined, as summarized in Table 1 (corresponding isotherms are shown in Figures S4–S6). For comparison, reported performances of various iron-based scavengers in the literature are summarized in Table S2. We can observe that  $\text{FeO}_x$  NPs exhibit oxygen adsorption capacity comparable to that of commercial scavengers (23.5–37.2 mL/g, measured under 53–57% RH). The oxygen adsorption capacity increases after infiltration into MSN, which are inert with respect to oxygen. The  $\text{FeO}_x$  coating of MSN by ALD<sup>11,19–23</sup> yields similar oxygen scavenging performance to  $\text{FeO}_x$  NPs. We can observe that the surface area of ALD coated samples is significantly higher than that of  $\text{FeO}_x$  NPs in MSN, but the oxygen scavenging performance is worse despite higher surface area. This is likely due to the fact that oxygen adsorption capacity is dependent both on the surface area and sample

Table 1. BET Surface Areas and Oxygen Adsorption Capacity of Different Samples

sample	BET surface area (m <sup>2</sup> /g)	absorbed oxygen (mL/g)
MSNs	841.6	0.0
$\text{FeO}_x$ NPs	134.7	28.0
$\text{FeO}_x$ NPs in MSNs	549.7	37.4
$\text{FeO}_x$ ALD on MSNs	779.2	28.3
$\text{FeO}_x$ ALD on $\text{FeO}_x$ NPs in MSNs	504.2	34.4
Fe ALD on MSNs	760.4	55.3
Fe ALD on $\text{FeO}_x$ NPs in MSNs	730.1	58.7
MSN-glucose	673.7	119.7
Fe ALD on MSNs-glucose	617.4	126.8
$\text{FeO}_x$ ALD on MSNs-glucose	615.9	105.5
$\text{FeO}_x$ ALD on $\text{FeO}_x$ NPs in MSNs-glucose	430.6	101.9
$\text{FeO}_x$ NPs in MSNs-glucose	504.6	108.6
Fe ALD on $\text{FeO}_x$ NPs in MSNs-glucose	447.9	114.9

composition, i.e., availability of surface Fe which can be further oxidized. Both  $\text{FeO}_x$  ALD coating and  $\text{FeO}_x$  NPs contain FeO and  $\text{Fe}_2\text{O}_3$ , but in different proportions as evident from differences in sample stoichiometry (Table S1). The higher  $\text{Fe}_2\text{O}_3$  content of  $\text{FeO}_x$ -ALD coating would result in decreased oxygen scavenging capacity despite the increased surface area. The combination of NPs and ALD coating results in reduced oxygen scavenging capacity compared to  $\text{FeO}_x$  NPs in MSNs and increased oxygen scavenging capacity compared to  $\text{FeO}_x$  ALD on MSNs, likely a result of reduced surface area compared to  $\text{FeO}_x$  NPs in MSNs (and possibly increased  $\text{Fe}^{3+}$  content due to ALD coating) and increased  $\text{Fe}^{2+}$  content compared to  $\text{FeO}_x$  ALD on MSNs due to the presence of  $\text{FeO}_x$  NPs.

The samples with Fe-ALD coating exhibit significant improvements in oxygen scavenging capacity, as expected since the oxygen adsorption capacity for oxidation of  $\text{Fe}^0$  into  $\text{Fe}^{2+}/\text{Fe}^{3+}$  is higher than that for oxidation of  $\text{Fe}^{2+}$  to  $\text{Fe}^{3+}$ . In this case, the combination of  $\text{FeO}_x$  NPs and ALD coating results in further enhancement of the performance.

To understand the mechanism of oxygen scavenging of different samples, XPS measurements before and after exposure to atmosphere for oxygen adsorption were conducted. Obtained results are shown in Figure 6 and summarized in Table 2, where the ratios of  $\text{Fe}^0$ ,  $\text{Fe}^{2+}$ , and  $\text{Fe}^{3+}$  in different samples were determined from fitting the Fe 3p spectra.<sup>24–27</sup> We can observe that for both  $\text{FeO}_x$  NPs and  $\text{FeO}_x$  coating by ALD, oxygen scavenging occurs due to oxidation of  $\text{Fe}^{2+}$  to  $\text{Fe}^{3+}$ . The higher starting  $\text{Fe}^{2+}/\text{Fe}^{3+}$  ratio of

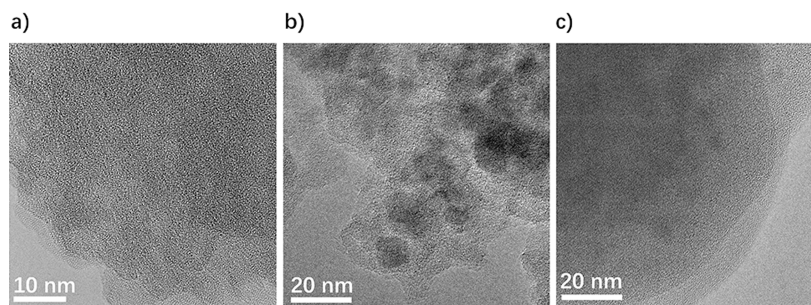
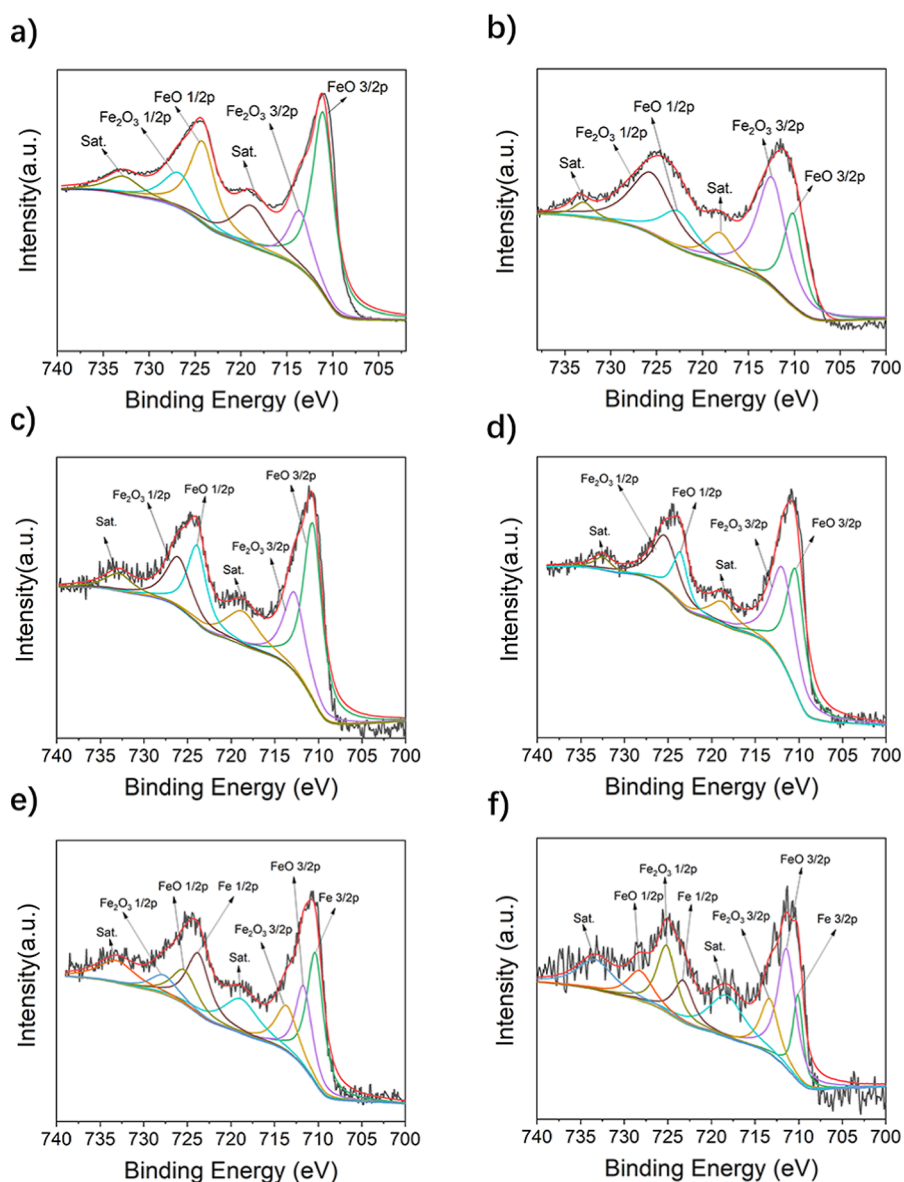


Figure 5. HRTEM images of (a) Fe ALD on MSN; (b)  $\text{FeO}_x$  ALD on MSN with  $\text{FeO}_x$  NPs; and (c)  $\text{FeO}_x$  ALD on MSN.



**Figure 6.** XPS Fe spectra of (a,b)  $\text{FeO}_x$  NPs fresh and 24 h; (c,d)  $\text{FeO}_x$  ALD on MSN fresh and 24 h; and (e,f) Fe ALD on MSN fresh and 24 h.

**Table 2.** XPS Fitting Results for  $\text{Fe}^0$ ,  $\text{Fe}^{2+}$ , and  $\text{Fe}^{3+}$  Content in Different Samples

samples	$\text{Fe}^{2+}/\text{Fe}^{3+}$	$\text{Fe}/\text{Fe}^{3+}$	$\text{Fe}/\text{Fe}^{2+}$
$\text{FeO}_x$ NPs-fresh	3.20	0	0
$\text{FeO}_x$ NPs 24 h	0.75	0	0
$\text{FeO}_x$ ALD on MSNs-fresh	2.22	0	0
$\text{FeO}_x$ ALD on MSNs-24 h	0.78	0	0
Fe ALD on MSNs-fresh	1.77	2.86	1.61
Fe ALD on MSNs-24 h	2.2	1.59	0.72

NPs is likely responsible for higher oxygen adsorption capacity of these samples. The oxygen adsorption capacity is a result of interplay between surface area and iron content in an oxidation state where further oxidation is possible ( $\text{Fe}^0$  or  $\text{Fe}^{2+}$ ). Consequently, samples with lower surface area and higher  $\text{Fe}^{2+}$  content ( $\text{FeO}_x$  NPs) and samples with higher surface area and lower  $\text{Fe}^{2+}$  content ( $\text{FeO}_x$  ALD on MSNs) can exhibit a similar oxygen scavenging performance. For Fe-coated MSN, we can observe a significantly different mode of action. In this

case,  $\text{Fe}^0$  gets oxidized to  $\text{Fe}^{2+}$ , and then,  $\text{Fe}^{2+}$  gets oxidized to  $\text{Fe}^{3+}$ . Consequently, after 24 h, the  $\text{Fe}^{2+}/\text{Fe}^{3+}$  ratio is increased, while  $\text{Fe}/\text{Fe}^{2+}$  and  $\text{Fe}/\text{Fe}^{3+}$  ratios are both decreased. From the fact that the samples still contain unreacted  $\text{Fe}^0$  after 24 h exposure to ambient atmosphere, we can conclude that the reactivity of Fe ALD-coated MSN is not very high and that these samples likely represent a good compromise between reactivity and oxygen adsorption capacity.

To investigate the possibility of further enhancement of oxygen adsorption capacity, we considered glucose-treated MSN samples, as these samples can exhibit high oxygen scavenging capacity.<sup>11</sup> Successful coating with a carbon-containing layer is evident from the TEM images (Figure S7) and EDX mapping of the samples (Figure S8). The glucose treatment enables significant oxygen scavenging capacity, which was attributed to loosely bound oxygen on the surface. To investigate the mechanism of oxygen adsorption, FTIR measurements were conducted, and the obtained results are shown in Figure S9. We can observe a significant enhancement in the intensity, as well as a change in

the position and shape of a broad feature in the spectral range 3200–3700  $\text{cm}^{-1}$ , which corresponds to O–H stretching vibrations,<sup>29</sup> as well as the vibration at  $\sim 1640 \text{ cm}^{-1}$  corresponding to molecular water.<sup>29</sup> No changes in prominent feature in the range 1000–1120  $\text{cm}^{-1}$  corresponding to Si–O–Si vibrations<sup>29</sup> is observed. The shift in the O–H vibrations peak position and the change in the peak shape likely originate from a change in proportions of free O–H and hydrogen-bonded (intra- and inter-molecular) O–H vibrations.<sup>29</sup> The observed changes are consistent with the increase in loosely bound surface oxygen. In the case of glucose-modified MSN, the use of  $\text{FeO}_x$  NP infiltration results in significant reduction of the available surface area in all cases, and consequently, no enhancement of the oxygen adsorption capacity is obtained. For ALD coating only, there is a smaller loss of BET surface area. However,  $\text{FeO}_x$  ALD coating results in lower oxygen adsorption capacity compared to glucose-treatment only, while Fe ALD coating is the only condition resulting in the enhancement of oxygen adsorption capacity. Considering the versatility of ALD deposition, the preparation method should be compatible not only with different MSN-based samples investigated in this work but also generally applicable, for example, as a part of multilayer oxygen barriers in packaging. Unlike nanoparticle samples incorporated into extruded polymers, ALD deposited coatings are expected to have excellent mechanical integrity and are unlikely to result in leakage of nanomaterial into the environment.

## CONCLUSIONS

We have investigated the oxygen scavenging performance of  $\text{FeO}_x$  NPs and iron-based coatings deposited by ALD. We found that ALD  $\text{FeO}_x$  coating results in lower  $\text{Fe}^{2+}/\text{Fe}^{3+}$  ratios compared to  $\text{FeO}_x$  NPs, while Fe ALD coating results in increased oxygen adsorption capacity due to oxidation of  $\text{Fe}^0$  into  $\text{Fe}^{2+}/\text{Fe}^{3+}$ . For combining the  $\text{FeO}_x$  NPs and ALD-coated MSN for increasing overall oxygen adsorption capacity, there is a complex interplay between the coating composition and reduced surface area after deposition. Thus, for  $\text{FeO}_x$  NPs infiltrated inside MSN, an increase in oxygen adsorption capacity can be obtained with Fe-ALD coating, but not with  $\text{FeO}_x$ -ALD coating. In addition, Fe-ALD coating is generally applicable to different samples and can be used to enhance oxygen adsorption capacity of glucose-treated MSN, with the highest oxygen adsorption capacity of 126.8 mL/g obtained. Fe-ALD coating is a versatile method for enhancing the oxygen adsorption capacity and also has a potential promise for integration with the packaging as a part of oxygen barriers.

## EXPERIMENTAL SECTION

**Materials.** Tetraethyl orthosilicate (TEOS, purity 99.9%) and cetyl trimethylammonium bromide (CTAB, purity 98%) were obtained from Alfa Aesar. Ethanol (purity 99.8%),  $\text{NH}_3 \cdot \text{H}_2\text{O}$  (ACS, 28–30%), and 1,3,5-trimethylbenzene (TMB, purity 98%) were obtained from Energy Chemical. ALD precursor ferrocene (purity 99.999%) was obtained from Sigma-Aldrich. All materials were used as received without further purification.

**Synthesis of  $\text{FeO}_x$  NPs.** The NPs were synthesized following a previously reported procedure<sup>18</sup> with minor modifications. Briefly, 12 mL of 0.2 M  $\text{FeCl}_3$  aqueous solution was poured into 12 mL of 0.5 M  $\text{NH}_4\text{OH}$  solution at room temperature and allowed to sonicate for 2 min. 6 mL of

aqueous 0.2 M  $\text{FeCl}_2$  was then added under sonication and the mixture was poured into 36 mL of 0.5 M  $\text{NH}_4\text{OH}$  aqueous solution. The solution was then swirled and stirred for 1 min. The solution was then kept still for 15 min, then magnetically separated, and washed with 200 mL of DI water each time at least five times. Afterward, 1.5 mL of 0.1 M sodium citrate was added under sonication, followed by dropwise addition of 1 mL of 5% sodium hypochlorite solution. The washing process was then repeated to collect the product. The final product was then washed once more under centrifugation at 9000 rpm for 5 min, the supernatant was disposed of, and the colloid was left in vacuum at 50 °C to dry overnight. The resulting big clumps were broken into a coarse powder using a metal spatula to obtain the final product.

**Synthesis of MSN and Glucose-Treated MSN.** The synthesis of MSNs was reported in our previous work.<sup>11</sup> Briefly, 16 mL of deionized water, 6 mL of ethanol, 0.5 mL of  $\text{NH}_3 \cdot \text{H}_2\text{O}$ , and 0.6 g CTAB were mixed and sonicated for 15 min. After the sonication, 0.225 mL of ammonia was added to adjust pH to  $\sim 10$  to 11 and the solution was homogenized for 30 min. TMB (0.9 mL) and TEOS (2 mL) were then added with rapid stirring at 800 rpm. After 2 h reaction, the white precipitates were centrifuged at 10,000 rpm for 10 min and purified by ethanol three times. Then, the as-prepared precipitates were dried for 16 h in air at 60 °C. Finally, MSNs obtained by calcination at 550 °C for 10 h in a muffle furnace with a heating rate of 1 °C/min. For the glucose treatment of MSN,<sup>11</sup> 100 mg of MSNs and 60 mg glucose were dispersed into 30 mL of deionized water and stirred for 30 min, followed by ultra-sonication for 10 min and then stirring for another 10 min. The solution was then transferred into a 50 mL Teflon lined stainless steel autoclave and kept at 180 °C for 4 h. The reacted solution was centrifuged and washed by ethanol several times. Prepared samples were then ground and dried at 40 °C under high vacuum.

**Synthesis of  $\text{FeO}_x$  NP Infiltrated MSN.** Based on a previously reported procedure,<sup>28</sup> 0.6 g of CTAB is added to a mixture of deionized water,  $\text{FeO}_x$  NP solution and ethanol (8 mL/8 mL/4 mL). To adjust the pH (pH  $\sim 11$ –12) of the mixture, 0.3 mL of ammonia solution is added. The solution is then sonicated for 15 min to give a homogeneous mixture. TMB (0.9 mL) and TEOS (2 mL) were added to the mixture and stirred for 2 h at 800 rpm. The gel obtained is then centrifuged at 4000 rpm for 30 min and washed with ethanol twice at 4000 rpm for 10 min. The obtained gel is then dried in vacuum at 70 °C overnight. The dried product is ground to give a fine powder and calcined at 550 °C for 5 h at a ramping rate of 1 °C/min to yield  $\text{FeO}_x$  NPs incorporated in MSN.

**ALD Coating of MSN.** ALD deposition of Fe and  $\text{FeO}_x$  coatings were performed on the pre-synthesized MSN powder using a Savannah S200 ALD chamber from Ultratech. The Fe deposition was carried out at a low temperature of 150 °C and ferrocene was used as a precursor heated at 423 K. The pulse time for the ferrocene precursor was 0.5 s, and 10 sccm  $\text{N}_2$  was purged in between two consecutive ferrocene pulses for 20 s. 100 cycles of Fe were coated on the MSNs. For the deposition of  $\text{FeO}_x$  coating, ferrocene and water ( $\text{H}_2\text{O}$ ) were used as precursors.<sup>11</sup> The deposition was carried out at 573 K and the ferrocene precursor was heated up to 423 K to maintain the adequate vapor pressure in the bubbler. One cycle consisted of a 0.01 s pulse of ferrocene with 5 s purge nitrogen and a 0.015 s pulse of water, with 15 s purge nitrogen.

**Characterizations.** The sample morphologies and EDX mapping were recorded using TEM equipped with EDX analyzer (FEI, Talos) operated at 300 kV. SAED patterns were also acquired at 300 kV. EDX analyses were performed using NOVA Nano SEM450 from FEI. A voltage of 5 kV and high beam current was used for composition analysis. The EDX measurements for overall Fe content determination were carried out on the NP powders and ALD coating samples on MSN powders. Same ALD coatings were deposited on Al substrates (purity 99.999%) for Fe/O ratio detection (since MSN samples contain oxygen).

X-ray photoelectron spectra (XPS) were recorded using an ESCALAB 250Xi from Thermo Fisher equipped with Al K $\alpha$  (1486.69 eV) X-ray source. The C 1s binding energy of 284.8 eV was used as energy reference. In order to obtain a sufficiently strong signal, one spectrum was obtained under vacuum with an accumulation time of 100 s and five scans. The BET measurements were performed using a Micromeritics ASAP2020 physisorption analyzer with pure nitrogen at 77 K. The FTIR measurements were conducted by a Bruker VERTEX 80v spectrometer. Samples were well mixed and ground with dried potassium bromide (KBr, Sigma-Aldrich). The final concentrations of samples/KBr were ~1% (w/w). The crystal structures of the samples were examined using X-ray diffraction (Rigaku, SmartLab; operated at 40 kV and 150 mA, Cu K $\alpha$  source).

Oxygen adsorption capacity was determined as described previously.<sup>11</sup> Briefly, measurements carried out using an MapXpert from JC Instruments with a measurement range of 0–100% and a resolution of 0.001%. The sample with the weight of 250 mg was placed in a 350 mL bottle and sealed. The humidity was controlled at 57% in the bottle. 3 mL of the gases were purged within 5 s for testing with an interval of 24 h.

## ■ ASSOCIATED CONTENT

### SI Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acsomega.3c01242>.

Nitrogen adsorption isotherms for the determination of BET surface area and EDX mapping images (PDF)

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## Author Contributions

A. M. C. N. and A. B. D. designed the whole study, analyzed and interpreted results, and wrote the manuscript. Y. H. conducted ALD deposition and various sample characterizations. A. K. synthesized iron oxide nanoparticles, and P. S. L. and I. A. prepared MSN samples. All co-authors provided input to the final version of the manuscript.

## Notes

The authors declare no competing financial interest.

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