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# Dual-functional persistent luminescent nanoparticles with enhanced persistent luminescence and photocatalytic activity†

Ailijiang Tuerdi and Abdukader Abdukayum \*

Dual-functional nanoparticles with near-infrared (NIR) persistent luminescence and sun-light photocatalytic activity are highly desired for medical diagnosis and environmental protection. Here, we report a facile one-step method for simultaneous enhancement of persistent luminescence and photocatalytic activity of the dual-functional persistent luminescent nanoparticles (PLNPs). The Bi<sup>3+</sup>, Cr<sup>3+</sup> co-doped ZnGa<sub>2</sub>O<sub>4</sub> PLNPs, which were less than 10 nm in size, were synthesized by an ethylene glycol-assisted hydrothermal method. The persistent luminescence and the photocatalytic activity of the PLNPs were significantly and simultaneously improved *via* additional doping of Bi<sup>3+</sup> in ZnGa<sub>2</sub>O<sub>4</sub>:Cr. The prepared dual-functional PLNPs have great potential in pollutant photo-degradation and long-term imaging *in vivo*.

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## 1. Introduction

The long-lasting phosphorescence (or afterglow) of persistent luminescent materials is an optical phenomenon whereby luminescence can remain for a long time (hours to days) after the excitation source is removed.<sup>1–3</sup> In recent years, persistent luminescent materials have been drawing increasing attention in many areas, such as emergency signage, traffic signs, lighting, photocatalysis, and medicine.<sup>4–7</sup> Molecular probes based on persistent luminescent nanoparticles (PLNPs) exhibit a super-long afterglow time and do not need *in situ* excitation during biomedical imaging, so they completely avoid tissue auto-fluorescence, light scattering, and photo-toxicity originating from the excitation light.<sup>8–10</sup> PLNPs, alone or in combination with other photocatalysts such as TiO<sub>2</sub>, have great potential in photocatalysis owing to their potentially increased solar light absorption during day time and subsequent increased solar energy release at night or after the removal of illumination source.<sup>3,11</sup> Therefore, dual-functional PLNPs with near-infrared (NIR) persistent luminescence and photocatalytic activity have great potential in medical diagnosis and environmental protection.

Zinc gallate (ZnGa<sub>2</sub>O<sub>4</sub>) is a cubic normal AB<sub>2</sub>O<sub>4</sub> spinel crystal semiconductor with the *Fd3m* space group.<sup>12</sup> It is one of the excellent blue-emitting phosphors with wide band gaps (4.4 eV for ZnGa<sub>2</sub>O<sub>4</sub>) and has great potential applications in displays,

owing to its strong blue emission, better chemical and thermal stability, and excellent cathodoluminescence properties.<sup>13</sup> ZnGa<sub>2</sub>O<sub>4</sub> also acts as an excellent photo-catalyst due to its hybridized orbitals of Zn4s4p, Ga4s4p, and band gap, which can improve the mobility of photo-generated electrons and the absorption efficiency of ultraviolet (UV) light.<sup>5,14</sup>

ZnGa<sub>2</sub>O<sub>4</sub> is also an excellent photoluminescence host material when doped with transition metals, such as Mn-doped ZnGa<sub>2</sub>O<sub>4</sub> for green emission, and Cr<sup>3+</sup>-doped ZnGa<sub>2</sub>O<sub>4</sub> (ZnGa<sub>2</sub>O<sub>4</sub>:Cr) for red or near-infrared (NIR) emission.<sup>15</sup> In particular, ZnGa<sub>2</sub>O<sub>4</sub>:Cr is a bright NIR-emission persistent luminescence phosphor, which is considered a promising optical nanoprobe for *in vivo* bio-imaging.<sup>16–18</sup> Co-doping of ZnGa<sub>2</sub>O<sub>4</sub> or zinc gallogermanate systems with Cr<sup>3+</sup> and rare earth ions, such as Pr<sup>3+</sup> and Dy<sup>3+</sup>, can significantly improve the persistent luminescence intensity and afterglow time of PLNPs, because trivalent lanthanide ions with special 4f electron configurations can provide more efficient trap density and depth.<sup>9,19,20</sup> However, afterglow time of the Cr<sup>3+</sup>-doped ZnGa<sub>2</sub>O<sub>4</sub> is still not long enough for long-term *in vivo* imaging. Furthermore, the Cr<sup>3+</sup>-doped ZnGa<sub>2</sub>O<sub>4</sub> is hardly used on photocatalytic degradation owing to its lower photocatalytic activity, compared to un-doped ZnGa<sub>2</sub>O<sub>4</sub>. Therefore, the development of an effective one-step methodology to simultaneously improve the persistent luminescence and the photocatalytic activity of dual-functional PLNPs are highly desired for pollutant photo-degradation and long-term imaging *in vivo*. Nevertheless, to the best of our knowledge, no work on simultaneous enhancement of persistent luminescence and photocatalytic activity for Cr<sup>3+</sup> co-doped ZnGa<sub>2</sub>O<sub>4</sub> have been reported so far.

Here, we report a facile one-step method for simultaneous enhancement of persistent luminescence and photocatalytic

Xinjiang Laboratory of Native Medicinal and Edible Plant Resources Chemistry, College of Chemistry and Environmental Sciences, Kashgar University, Kashgar 844007, China. E-mail: abdukadera@sina.com

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activity of the dual-functional PLNPs. The  $\text{Bi}^{3+}$ ,  $\text{Cr}^{3+}$  co-doped  $\text{ZnGa}_2\text{O}_4$  PLNPs, which were less than 10 nm in size, were synthesized by an ethylene glycol-assisted hydrothermal method. The NIR persistent luminescence and photocatalytic activity of  $\text{Cr}^{3+}$  co-doped  $\text{ZnGa}_2\text{O}_4$  are significantly, and simultaneously, improved by additional doping with  $\text{Bi}^{3+}$ . The prepared dual-functional PLNPs are promising for *in vivo* bio-imaging and photocatalytic application.

## 2. Experimental

### 2.1 Sample preparation

$\text{Bi}^{3+}$ ,  $\text{Cr}^{3+}$  co-doped  $\text{ZnGa}_2\text{O}_4$  PLNPs were prepared by an ethylene glycol assisted hydrothermal method. To prepare the  $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$ , 10.0 mL of ethylene glycol, 3.94 mL of  $\text{Ga}^{3+}$  (0.2 mol  $\text{L}^{-1}$ ), 1 mL of  $\text{Zn}^{2+}$  (0.4 mol  $\text{L}^{-1}$ ), 0.4 mL of  $\text{Cr}^{3+}$  (0.01 mol  $\text{L}^{-1}$ ), and 0.8 mL of  $\text{Bi}^{3+}$  (0.01 mol  $\text{L}^{-1}$ ) were mixed and stirred at room temperature for 30 min. The pH of the mixture solution was then adjusted to 9 with aqueous ammonia (15 wt%) and stirred at room temperature for 1 h. The resulting mixture was transferred to a 20 mL Teflon-lined stainless-steel autoclave in ambient environment and heated at 170 °C for 24 h. The autoclave was then allowed to cool to room temperature. The purified products were separated by centrifugation and the NPs were precipitated and washed with distilled water and ethanol and dried in a vacuum oven at 60 °C for 12 h. Finally, the PLNPs were annealed in air at 1000 °C for 3 h.

### 2.2 Characterization

X-ray diffraction (XRD) patterns were recorded on a D/max-2500 diffractometer (Rigaku, Japan) using  $\text{Cu K}\alpha$  radiation ( $\lambda = 1.5418 \text{ \AA}$ ). Transmission electron microscopy (TEM) and high-resolution TEM images were obtained on JEOL-100CX II and JEM-2100F field emission transmission electron microscopes (JEOL, Japan), respectively. The elemental composition of samples was analyzed with an EDAX Genesis XM2 attached to the S-4800 SEM (Hitachi, Japan). X-ray photoelectron spectra (XPS) were recorded on Escalab 250Xi (Thermo Fisher Scientific, USA) X-ray photoelectron spectrometer. The excitation and emission spectra and the long-lasting luminescence decay curves were obtained at room temperature using a F-4500 fluorescence spectrophotometer (Hitachi, Japan). The UV-Vis diffuse reflectance spectra were obtained using a UV-2600 UV-Vis Spectrophotometer (Shimadzu, Japan).

## 3. Results and discussion

### 3.1 Structure and morphology

Fig. 1 shows the XRD patterns of  $\text{ZnGa}_2\text{O}_4$ ,  $\text{ZnGa}_2\text{O}_4:\text{Cr}$ , and  $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$ . All the peaks are assigned to the  $\text{ZnGa}_2\text{O}_4$  sample with  $2\theta$  at 18.45, 30.28, 35.7, 37.4, 43.58, 57.56, 63.19, 65.82, and 74.72, which were indexed to the (111), (220), (311), (222), (400), (422), (511), (440), and (531) crystal planes, respectively.<sup>20</sup> All diffraction peaks are consistent with the  $\text{ZnGa}_2\text{O}_4$  spinel phase (JCPDS no. 38-1240). When the contents of Bi are 0.01 and 0.02, no characteristic peaks of other phases, such as  $\text{ZnO}$ ,  $\text{Ga}_2\text{O}_3$  and

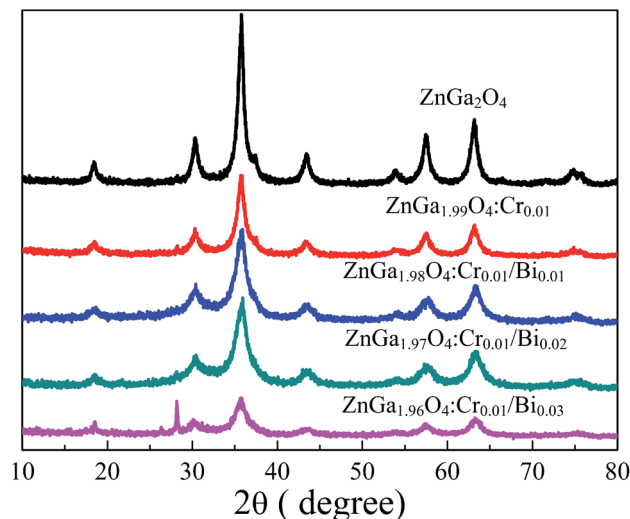


Fig. 1 XRD patterns of  $\text{ZnGa}_2\text{O}_4$ ,  $\text{ZnGa}_2\text{O}_4:\text{Cr}$  and  $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$ .

$\text{Bi}_2\text{O}_3$ , were observed, indicating the formation of a pure spinel phase zinc gallate solid solution. When the content of  $\text{Bi}^{3+}$  in  $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$  was increased from 0.02 to 0.03, we observed a strong peak at  $28.01^\circ$ , which was indexed to the rhombohedral Bi structure. The result indicates that the increasing content of  $\text{Bi}^{3+}$  influences the crystal structure of  $\text{ZnGa}_2\text{O}_4$ .

Fig. 2a shows TEM image of the as-synthesized  $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$  PLNPs and reveals that the PLNPs are less than 10 nm in size. The high-resolution TEM analysis shown in Fig. 2b indicates that the particles are single crystalline and the distance between the lattice fringes is 4.81 Å, which corresponds to the d-spacing of the spinel  $\text{ZnGa}_2\text{O}_4$  (111) lattice planes. The selected-area electron diffraction (SAED) pattern further confirmed that the crystal of as-synthesized  $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$  PLNPs is a pure spinel single crystal (Fig. 2c). The results of EDX analysis are demonstrated that the  $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$  PLNPs were composed of only Zn, Ga, Cr, Bi, and O (Fig. 2d). These results revealed that the  $\text{ZnGa}_2\text{O}_4$  crystal structure was successfully doped with Bi and Cr.

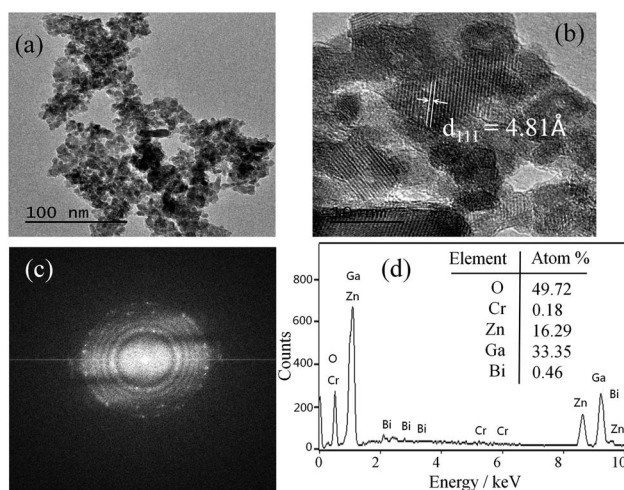


Fig. 2 (a–c) TEM, HR-TEM, SAED images and (d) EDX patterns of  $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$ .

### 3.2 Persistent luminescence of PLNPs

Fig. 3 shows the phosphorescence emission spectra of  $\text{ZnGa}_2\text{O}_4:\text{Cr}$  and  $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$  PLNPs at room temperature at an excitation wavelength of 254 nm.  $\text{ZnGa}_2\text{O}_4:\text{Cr}$  produced an intensive NIR emission band peaking at 698 nm (assigned to  ${}^2\text{E} \rightarrow {}^4\text{A}_2$  transition of  $\text{Cr}^{3+}$  ions) that superimposes on a broad emission band at 600–800 nm (assigned to  ${}^4\text{T}_2 \rightarrow {}^4\text{A}_2$  transition of disordered  $\text{Cr}^{3+}$  ions).<sup>2</sup> The emission spectra of  $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$  are similar to  $\text{ZnGa}_2\text{O}_4:\text{Cr}$ , but the emission peaks of  $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$  were slightly red-shifted due to the crystal field strength of  $\text{Cr}^{3+}$  centres decreased by additional doping of  $\text{Bi}^{3+}$  in  $\text{ZnGa}_2\text{O}_4:\text{Cr}$ .<sup>21</sup> Fig. S1 (ESI<sup>†</sup>) shows the phosphorescence excitation spectra of  $\text{ZnGa}_2\text{O}_4:\text{Cr}$  and  $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$ . There are four main absorption bands. The strong band peak around 260 nm is probably the combination of the  $\text{ZnGa}_2\text{O}_4$  host excitation band and the O–Cr charge transfer band.<sup>9</sup> The bands peaked at approximately 411 nm, 471 nm, and 554 nm and originated from the  ${}^4\text{A}_2 \rightarrow {}^4\text{T}_1(\text{te}^2)$  transition, the  ${}^4\text{A}_2 \rightarrow {}^4\text{T}_1(\text{t}^2\text{e})$  transition, and the  ${}^4\text{A}_2 \rightarrow {}^4\text{T}_2$  transition originated from the 3d intrashell transitions of  $\text{Cr}^{3+}$ , respectively.<sup>2</sup>

Fig. 4 shows the persistent luminescence emission decay curves of  $\text{ZnGa}_2\text{O}_4:\text{Cr}$  and  $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$  PLNPs, monitored at 698 nm after being excited by 254 nm UV light for 5 min. The data represent the persistent luminescence intensity ( $I$ ) as a function of time ( $t$ ) and the data recording lasted for 20 min. The persistent luminescence intensity increased as the content of Bi increased from 0 to 0.02, but decreased with a further increase from 0.02 to 0.03. The most intense persistent luminescence was obtained for  $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$ . The decay curves of the persistent luminescence can be well fitted by the three-exponential eqn (1) as follows:<sup>9</sup>

$$I(t) = I_0 + A_1 \exp\left(-\frac{t}{\tau_1}\right) + A_2 \exp\left(-\frac{t}{\tau_2}\right) + A_3 \exp\left(-\frac{t}{\tau_3}\right) \quad (1)$$

where  $I(t)$  is the intensity of persistent luminescence at time  $t$ ;  $I_0$  is the initial intensity of persistent luminescence;  $A_1$ ,  $A_2$ , and  $A_3$  are constants;  $\tau_1$ ,  $\tau_2$ , and  $\tau_3$  are derived lifetimes for the exponential components indicating the three different decay processes. The fitting results according to the above formula are

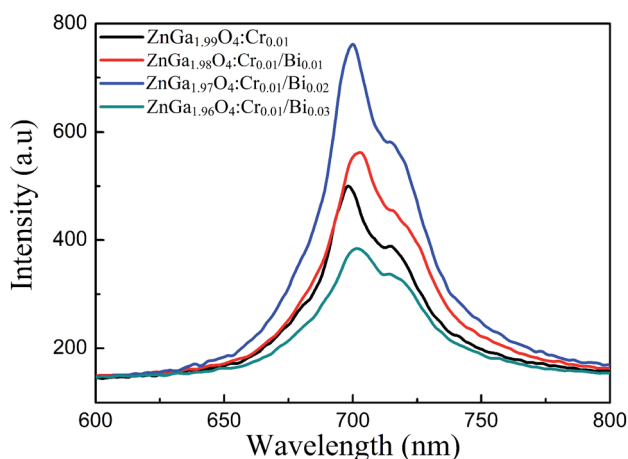


Fig. 3 Phosphorescence emission spectra of  $\text{ZnGa}_2\text{O}_4:\text{Cr}$  and  $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$  at an excitation wavelength of 254 nm.

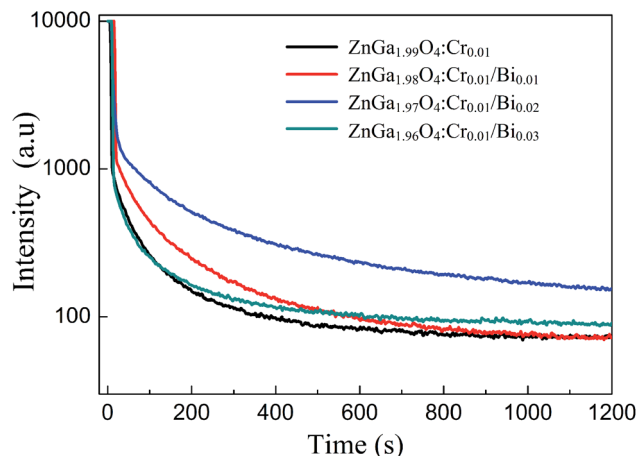


Fig. 4 Afterglow decay curve of  $\text{ZnGa}_2\text{O}_4:\text{Cr}$  and  $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$  after irradiation with a 254 nm UV lamp for 5 min.

presented in Table S1 (ESI<sup>†</sup>). These average photoluminescence lifetimes ( $\tau_{\text{av}}$ ) indicate that the  $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$  PLNPs produced a much longer photoluminescence lifetime (32.41 s) than the single  $\text{Cr}^{3+}$ -doped  $\text{ZnGa}_2\text{O}_4$  PLNPs (9.35 s). The above experimental results indicated that the additional doping of  $\text{Bi}^{3+}$  cannot cause remarkable changes in the excitation and emission spectra of  $\text{ZnGa}_2\text{O}_4:\text{Cr}$  but can significantly improve the intensity and time of persistent luminescence for  $\text{ZnGa}_2\text{O}_4:\text{Cr}$  PLNPs. Previous work reported that additional doping with  $\text{Bi}^{3+}$  in  $\text{ZnGa}_2\text{O}_4:\text{Cr}$  can also enhance the persistent luminescence of PLNPs.<sup>22,23</sup> The  $\text{Cr}^{3+}$  prefers to substitute for  $\text{Ga}^{3+}$  (with a similar ionic radius) in distorted octahedral coordination in  $\text{ZnGa}_2\text{O}_4:\text{Cr}$ . The  $\text{Bi}^{3+}$  substituting for  $\text{Ga}^{3+}$  can cause a higher periodic lattice distortion in  $\text{ZnGa}_2\text{O}_4:\text{Cr}$ , because the radius of  $\text{Bi}^{3+}$  (103 pm) is bigger than that of  $\text{Ga}^{3+}$  (62 pm). The higher periodic lattice distortion could generate more defects in the  $\text{ZnGa}_2\text{O}_4$  host.<sup>24</sup> Doping with  $\text{Bi}^{3+}$  probably increases the afterglow time of  $\text{ZnGa}_2\text{O}_4:\text{Cr}$  PLNPs by producing more hole traps.<sup>19</sup>

### 3.3 Photocatalytic activity of PLNPs

Fig. 5 shows the UV-Vis diffuse reflectance spectra of  $\text{ZnGa}_2\text{O}_4$ ,  $\text{ZnGa}_2\text{O}_4:\text{Cr}$ , and  $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$ . The  $\text{Bi}^{3+}$ ,  $\text{Cr}^{3+}$  co-doped  $\text{ZnGa}_2\text{O}_4$  PLNPs have a broad absorption spectra from UV to visible light (670 nm), and the UV-Vis light absorption is remarkably enhanced, compared to  $\text{ZnGa}_2\text{O}_4$  and  $\text{ZnGa}_2\text{O}_4:\text{Cr}$ . The broad absorption band of the  $\text{Bi}^{3+}$ ,  $\text{Cr}^{3+}$  co-doped  $\text{ZnGa}_2\text{O}_4$  PLNPs at 450 nm is associated with the charge transfer band from  $\text{Bi}^{3+} 6s^2$  to  $\text{Bi}^{5+} 6s^0$ .<sup>25,26</sup> We carried out the XPS experiments to verify the existence of  $\text{Bi}^{5+}$ . The Bi 4f XPS spectrum shows two distinct peaks which were fitted into three peaks at 158.87 eV, 160.23 eV and 163.92 eV, respectively (Fig. S2 (ESI<sup>†</sup>)). The peaks of Bi 4f<sub>7/2</sub> at 158.87 eV and Bi 4f<sub>5/2</sub> at 163.92 eV are corresponding to Bi(III). The peak of Bi 4f<sub>7/2</sub> at 160.23 eV is ascribed to Bi(V), which indicates existence of the  $\text{Bi}^{5+}$  in the  $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$ .<sup>27,28</sup> Band gaps ( $E_g$ ) of  $\text{ZnGa}_2\text{O}_4$ ,  $\text{ZnGa}_2\text{O}_4:\text{Cr}$ , and  $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$  were calculated to be approximately 4.38 eV, 4.45 eV, and 2.75 eV, respectively (Fig. S3 (ESI<sup>†</sup>)), according to the following eqn (2):<sup>29</sup>

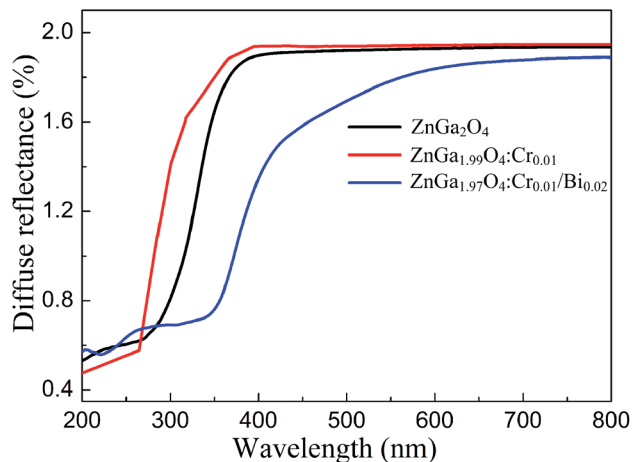


Fig. 5 UV-Vis diffuse reflectance spectra of  $\text{ZnGa}_2\text{O}_4$ ,  $\text{ZnGa}_2\text{O}_4:\text{Cr}$  and  $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$ .

$$\alpha h\nu = A(h\nu - E_g)^2 \quad (2)$$

where  $\alpha$  is the absorption coefficient,  $h\nu$  is the photon energy, and  $A$  is a constant. The experimental results shows that band gap of the  $\text{ZnGa}_2\text{O}_4$  was slightly increased by doping  $\text{Cr}^{3+}$ , probably due to the self-reduction effect on  $\text{Cr}^{3+}$  ions reduce into lower valence states and the absence of localized states in the band gap.<sup>30,31</sup> In contrast, the additional doping of  $\text{ZnGa}_2\text{O}_4:\text{Cr}$  with  $\text{Bi}^{3+}$  can significantly decrease the band gap of  $\text{ZnGa}_2\text{O}_4$ , owing to the electronic structure of Bi with a unique 6s orbital,<sup>27,32</sup> which has the benefit of increasing solar light absorption in applications of photocatalysis degradation.

Rhodamine B (RhB) was used as the probe molecule to evaluate the photocatalytic performance of the  $\text{Bi}^{3+}$ ,  $\text{Cr}^{3+}$  co-doped  $\text{ZnGa}_2\text{O}_4$  PLNPs. The photo-degradations efficiency of  $\text{ZnGa}_2\text{O}_4$ ,  $\text{ZnGa}_2\text{O}_4:\text{Cr}$ , and  $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$  for RhB are shown in Fig. 6. The as-prepared samples were analysed first, to establish an adsorption-desorption equilibrium between the catalyst and the dye molecules. The suspension was stirred in darkness for 40 min, and the samples were then placed under UV light irradiation. The  $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$  PLNPs exhibited an excellent photocatalytic activity with a degradation efficiency of 99.2% after 100 min irradiation with UV light. Experimental results indicate that the photocatalytic activity of the pure  $\text{ZnGa}_2\text{O}_4$  is lower than that of the  $\text{Bi}^{3+}$ ,  $\text{Cr}^{3+}$  co-doped  $\text{ZnGa}_2\text{O}_4$  and higher than that of the single  $\text{Cr}^{3+}$  doped  $\text{ZnGa}_2\text{O}_4$  (Fig. 6). The photocatalytic activity of  $\text{ZnGa}_2\text{O}_4$  was suppressed by the doping with  $\text{Cr}^{3+}$ , because the  $\text{Cr}^{3+}$  ions in  $\text{ZnGa}_2\text{O}_4$  act as recombination centres, which can reduce the number and lifetime of the electron-hole pairs.<sup>33</sup> By contrast, additional doping of  $\text{Bi}^{3+}$  in  $\text{ZnGa}_2\text{O}_4:\text{Cr}$  not only enhanced the UV-Vis light absorption of  $\text{ZnGa}_2\text{O}_4$ , but also produced more hole trap energy levels, which resulted in more photo-generated electron-holes being captured by traps, thus inhibiting the recombination efficiency of the photo-generated electron-holes.<sup>24</sup> As a consequence, the lifetime of the photo-generated electron-holes and the afterglow time were prolonged.

To investigated the role of reactive oxygen species (ROS) in the degradation process, we carried out the ROS scavenging

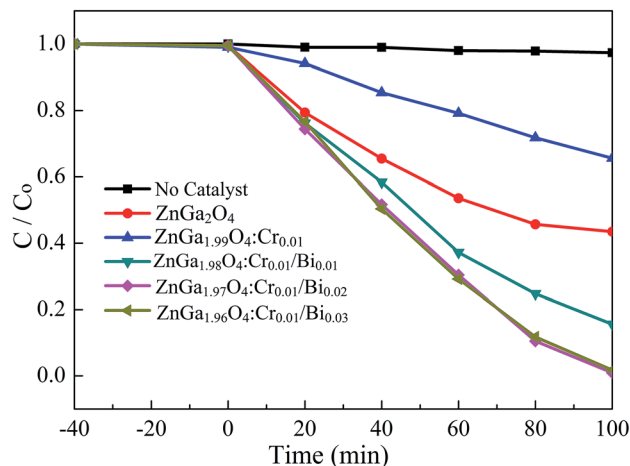
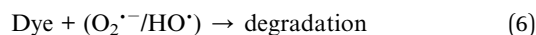
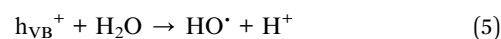
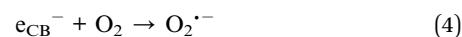
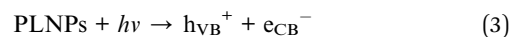


Fig. 6 The photo-degradation efficiency of  $\text{ZnGa}_2\text{O}_4$ ,  $\text{ZnGa}_2\text{O}_4:\text{Cr}$  and  $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$  for RhB.

experiments with different scavengers, such as *p*-benzoquinone (BQ) as  $\text{O}_2^{\cdot-}$  scavenger, isopropyl alcohol (IPA) as  $\text{HO}^{\cdot}$  scavenger, and ammonium oxalate (AO) as the hole scavenger.<sup>34</sup> The experimental results shows that the photo-degradation of RhB is dramatically decreased by adding AO in the RhB solution which indicates that the holes are the most dominant species in the photocatalytic degradation process (Fig. S4 (ESI<sup>+</sup>)). Compared to AO and BQ, the addition of IPA result in slight decrease for degradation of RhB, which indicates the  $\text{HO}^{\cdot}$  radicals are not main contributor for degradation process. The electrons and holes are generated on the surface of PLNPs under light UV radiation. The photogenerated electron-hole pairs reacted with oxygen and water to generate peroxide ( $\text{O}_2^{\cdot-}$ ) and hydroxyl radicals ( $\text{HO}^{\cdot}$ ).<sup>35</sup> The resulting radicals cause degradation of the organic molecules. This photocatalytic reaction can be represented as follows:



## 4. Conclusions

The  $\text{Bi}^{3+}$ ,  $\text{Cr}^{3+}$  co-doped  $\text{ZnGa}_2\text{O}_4$  PLNPs were synthesized by an ethylene glycol assisted hydrothermal method. The particle size of the as-prepared PLNPs was less than 10 nm. The photo-degradation efficiency of  $\text{Bi}^{3+}$ ,  $\text{Cr}^{3+}$  co-doped  $\text{ZnGa}_2\text{O}_4$  PLNPs for RhB was 99.2% after 100 min irradiation with UV light. Compared to  $\text{ZnGa}_2\text{O}_4:\text{Cr}$  and un-doped  $\text{ZnGa}_2\text{O}_4$ , the additional doping of  $\text{Bi}^{3+}$  in  $\text{ZnGa}_2\text{O}_4:\text{Cr}$  not only significantly enhanced the NIR persistence luminescence of the PLNPs, but also remarkably improved the photocatalytic degradation efficiency of RhB. Addition of  $\text{Bi}^{3+}$  in  $\text{ZnGa}_2\text{O}_4:\text{Cr}$  may produce

more trap energy levels, which prolongs the lifetime of the photo-generated electron–holes and the afterglow time. The novel dual-functional PLNPs are promising for *in vivo* bio-imaging and photocatalytic applications.

## Conflicts of interest

The authors declare no conflict of interest.

## Acknowledgements

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