

Article

Synthesis and Characterization of Conducting PANDB/ χ -Al₂O₃ Core-Shell Nanocomposites by In Situ Polymerization

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Abstract: Polyaniline doped with dodecylbenzenesulfonic acid/ χ -aluminum oxide (PANDB/ χ -Al₂O₃) conducting core-shell nanocomposites was synthesized via an in situ polymerization method in this study. PANDB was synthesized in the presence of dodecylbenzenesulfonic acid (DBSA), which functioned as a dopant and surfactant. The electrical conductivity of the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposite was approximately 1.7×10^{-1} S/cm when the aniline/ χ -Al₂O₃ (AN/ χ -Al₂O₃) weight ratio was 1.5. The transmission electron microscopy (TEM) results indicated that the χ -Al₂O₃ nanoflakes were thoroughly coated by PANDB to form the core-shell (χ -Al₂O₃-PANDB) structure. The TEM and field-emission scanning electron microscopy (FE-SEM) images of the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites also indicated that the thickness of the PANDB layer (shell) could be increased as the weight ratio of AN/ χ -Al₂O₃ was increased. In this study, the optimum weight ratio of AN/ χ -Al₂O₃ was identified as 1.5. The conducting PANDB/ χ -Al₂O₃ core-shell nanocomposite was then blended with water-based polyurethane (WPU) to form a conducting WPU/PANDB/ χ -Al₂O₃ blend film. The resulting blend film has promising antistatic and electrostatic discharge (ESD) properties.

Keywords: polyaniline; dodecylbenzenesulfonic acid; χ -Al₂O₃; in situ polymerization; core-shell nanocomposite



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1. Introduction

Polyaniline (PANI) is an important member of the intrinsically conducting polymer (ICP) family. Because of its unique electrochemical properties and environmental stability, PANI has been extensively studied by numerous researchers [1–6] and has been employed in many applications, such as secondary batteries [7,8], biosensors [9,10], corrosion protectors [11,12], antistatic packaging materials [13], and light-emitting diodes (LEDs) [14].

In recent years, encapsulating inorganic materials inside a PANI shell has become the most popular and interesting aspect of composites. The combination of inorganic components with conducting PANI has attracted considerable attention because of the resulting composites' novel physical and chemical properties and potential applications. These core-shell conducting composites can provide new synergistic properties that cannot be attained from the individual materials alone.

Preparing conducting core-shell PANI/inorganic composites via conventional mixing or blending in melt or solution form is difficult because PANI is insoluble, infusible, and possesses poor mechanical properties. One of the most promising methods to prepare conducting core-shell PANI/inorganic composites is through an in situ chemical polymerization of a monomer in the presence of inorganic components. Many groups have reported conducting PANI/inorganic core-shell composites such as PANI/Y₂O₃ [15], PANI/Zn_xFe_{3-x}O₄ [16], PANI/T-ZnOw [17], PANI/TiO₂ [18], PANI/MnO₂ [19], PANI/ZrO₂ [20], PANI/alumina [21–23], PANI/attapulgit (ATP) [24], and PANI/clay [25].

However, PANI doped with strong protonic acids is intractable. There have been a number of attempts towards making conducting PANI processable by doping with various dopants. The most attractive and promising system is PANI doped with dodecylbenzenesulfonic acid (DBSA) (PANDB). This is because DBSA containing a polar head and a long nonpolar chain which can function both as a surfactant and dopant. This advantage allows PANDB to not only become soluble in various organic solvents (e.g., chloroform and xylene) but also to have better compatibility with hydrophobic polymers. Therefore, conducting PANI doped with DBSA (PANDB) was adopted in this study.

The χ -Al₂O₃ nanoflake is one of the metastable polymorphs of transition alumina with a relatively high aspect ratio, thermal stability, porosity, and specific surface area. Due to its flake structure and thickness of around 20~50 nm, the χ -Al₂O₃ nanoflake can be used as an excellent reinforcement to enhance both the strength and stiffness of semicrystalline and glassy thermoplastics, elastomers, and epoxy resins.

To the best of our knowledge, no studies have reported the preparation for conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites through an in situ polymerization process in the presence of DBSA as the dopant. The goal of this study is to improve the electrical conductivity of the χ -Al₂O₃ nanoflake and let the χ -Al₂O₃ nanoflake coated with conducting PANDB form the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites. In addition, the optimum weight ratio of aniline/ χ -Al₂O₃ (AN/ χ -Al₂O₃) to synthesize the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites is also determined. Moreover, we also discuss the influences of the weight ratio of AN/ χ -Al₂O₃ on the electrical conductivity, chemical structure, and morphology of the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites. Finally, the synthesized product will be blended with water-based polyurethane (WPU) to form a conducting WPU/PANDB/ χ -Al₂O₃ blend film. The antistatic tests of the surface of the pure WPU film and the conducting WPU/PANDB/ χ -Al₂O₃ blend film will be examined and compared.

2. Experimental Section

2.1. Materials

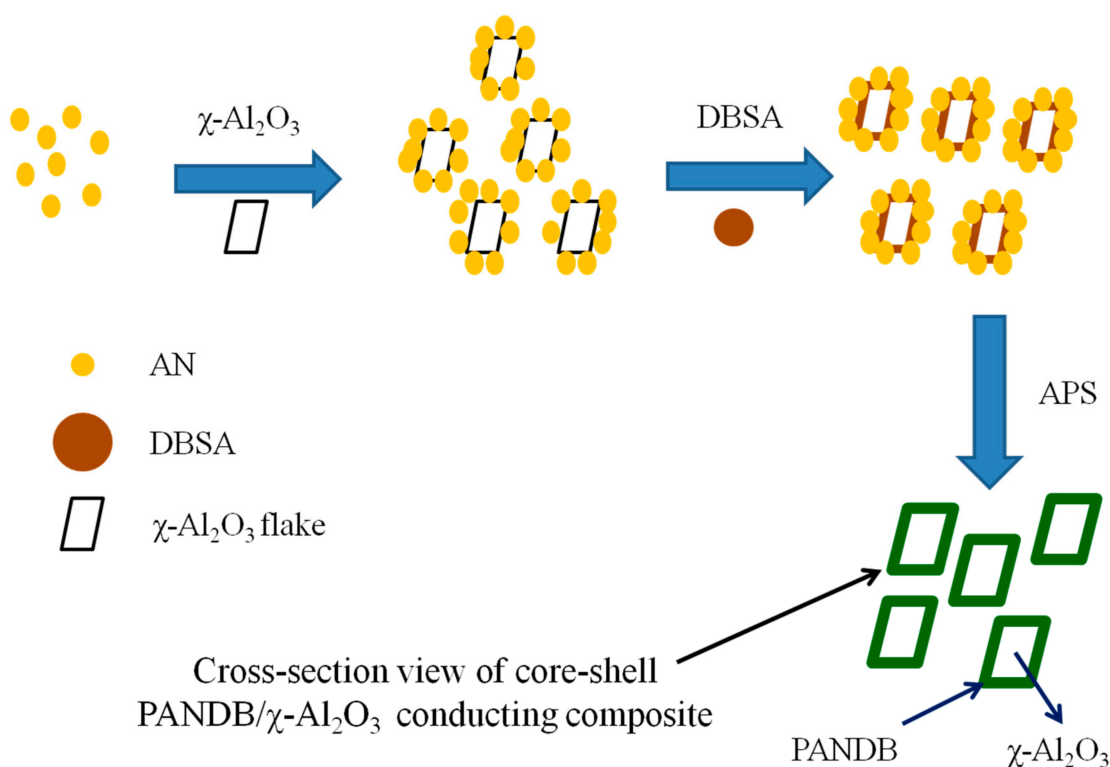
Aniline (AN) and ammonium persulfate (APS) were purchased from Merck KGaA (Darmstadt, Germany). Dodecylbenzene sulfonic acid (DBSA) was purchased from Tokyo Kasei Kogyo Co., Ltd. (Tokyo, Japan). The porous χ -Al₂O₃ nanoflake powder, which has a high porosity and high specific surface area, was supplied by the Particulate Research Center at the National Cheng-Kung University, Tainan City, Taiwan. The sectional diameter of one χ -Al₂O₃ nanoflake was in the range of 150 to 500 μ m, and the thickness of one χ -Al₂O₃ nanoflake was in the range of 20 to 50 nm. Meanwhile, the specific surface area of the χ -Al₂O₃ nanoflake was in the range of 130 to 150 m²/g. The nonionic WPU dispersion, tackiness, and curing agents for WPU were supplied by Hailsun Chemical Co., Ltd. (Taipei City, Taiwan).

2.2. Synthesis of Pure PANDB and Conducting PANDB/ χ -Al₂O₃ Core-Shell Nanocomposites

Pure PANDB was directly prepared by a chemical oxidation polymerization according to the modified procedure described in our previous study [26,27]. 8 g of aniline was mixed with 23 g of DBSA and 400 mL of distilled water to form a homogeneous milky white dispersion of anilinium-DBSA complex in a 1000 mL four-necked flat-bottom reactor under proper stirring at room temperature. The APS solution (20 g of APS dissolved in 200 mL of distilled water) was then slowly added to the reactor. The aniline:DBSA:APS molar ratio was set as 1:0.8:1. After a 2 h reaction, the darkish-green PANDB dispersion was precipitated by the addition of 600 mL of acetone. The reaction mixture was then filtered and washed several times with deionized water until the filtrate was colorless. Finally, the product was collected and dried in an oven at 60 °C for 24 h.

The conducting PANDB/ χ -Al₂O₃ core-shell nanocomposite was synthesized via an in situ chemical oxidation polymerization. 2 g of χ -Al₂O₃ was dispersed in 100 mL of an aqueous solution that contained 1 g of aniline. The resulting dispersion was then stirred

at room temperature for 10 min. The AN/ χ -Al₂O₃ weight ratio was set as 0.5. Then, 50 mL of an aqueous DBSA solution was added to the solution. After 10 min, 30 mL of an aqueous APS solution was added dropwise to the dispersion with constant stirring. The aniline:DBSA:APS molar ratio was 1: 0.8: 1. The resulting mixture was allowed to react for 2 h at room temperature. Subsequently, the product was washed with deionized water until the filtrate became colorless; the product was then dried in a vacuum oven at 60 °C for 24 h. To determine the effect of the weight ratio of AN/ χ -Al₂O₃ on the properties of the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites, products with weight ratios of 1.0, 1.5, and 2.0 were also prepared. Scheme 1 shows the synthesis of the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposite via an in situ polymerization process.



Scheme 1. Synthesis process of the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposite via an in situ polymerization.

2.3. Fabrication of Conducting WPU/PANDB/ χ -Al₂O₃ Blend Film

WPU dispersion was placed into a beaker equipped with a mechanical stirrer. 20wt% of the synthesized conducting PANDB/ χ -Al₂O₃ core-shell nanocomposite was added to the WPU dispersion. The WPU/20wt% (PANDB/ χ -Al₂O₃) mixture was then thoroughly mixed. Next, a tackiness agent was added to the mixture under mechanical stirring until the mixture viscosity reached around 7500 cps. A curing agent of 1.5 g was added to the mixture sequentially and was completely and well mixed. The resulting mixture was degassed with a vacuum pump to eliminate air bubbles. Finally, the air bubble-free mixture was cast on a polyethylene terephthalate (PET) substrate and heated to 120 °C for 2 min, then 150 °C for 2 min to complete the curing reaction of WPU. For a comparison, the pure WPU film was also made.

2.4. Characterization

2.4.1. Electrical Conductivity and Surface Resistance Analysis

In total, 0.1 g of sample was weighed and then pressed under 3.0×10^5 psi for 2 min at room temperature. The electrical conductivity σ (S/cm) of the sample was examined by a four-point probe meter (model: LSR4-KHT200, KeithLink Technology Co., Ltd., Taipei City, Taiwan) at room temperature. Each sample was first compressed into a sample with a

diameter of 10 mm. Then, 10 different locations on the sample surface were measured and averaged. The obtained results were averaged and reported. The surface resistance (Ω/sq) of the pure WPU film and conducting WPU/20wt%(PANDB/ $\chi\text{-Al}_2\text{O}_3$) blend films was examined by the same equipment. 10 different locations on the sample surface were measured and averaged.

2.4.2. FTIR Analysis

The chemical structure of the sample was determined by Fourier-transform infrared spectroscopy (FTIR) (model Spectrum One; Perkin Elmer, Waltham, MA, USA) over the wavenumber range $400\text{--}4000\text{ cm}^{-1}$ at 32 scan/s. The powdered samples and potassium bromide (KBr) (weight ratio of approximately 1:99) were grounded together into fine powders, and the homogeneous mixture was then pressed into a pellet for analysis.

2.4.3. UV-Vis Analysis

The sample was dispersed in absolute alcohol under ultrasonic agitation for 1 h at room temperature. An ultraviolet-visible spectrophotometer (UV-Vis; model UV-2401 PC, Shimadzu Co., Ltd., Kyoto, Japan) was used to measure the optical absorbance of the sample solution at a wavelength range of 300 to 900 nm.

2.4.4. XRD Analysis

X-ray diffraction (XRD) patterns of the synthesized sample were obtained at room temperature using an X-ray diffractometer (Rigaku MultiFlex ZD3609N, Rigaku Co., Tokyo, Japan) equipped with a $\text{Cu-K}\alpha$ radiation source ($\lambda = 0.154\text{ nm}$). The diffraction patterns were recorded at a scan rate of $2^\circ/\text{min}$ over the 2θ range from 5° to 80° .

2.4.5. TEM Examination

The sample was examined by transmission electron microscopy (TEM, model JEM-1230, JEOL, Ltd., Tokyo, Japan) by casting a diluted dispersion of the sample onto a carbon-coated copper grid. The TEM was operated at an accelerating voltage of 80 kV.

2.4.6. FE-SEM Examination

The sample was dried after filtration. The sample was then fractured, and each sample was coated with a gold-palladium film. The surface morphologies of the sample were examined using a field-emission scanning electron microscope (FE-SEM, model JSM 6700F; JEOL, Ltd., Tokyo, Japan).

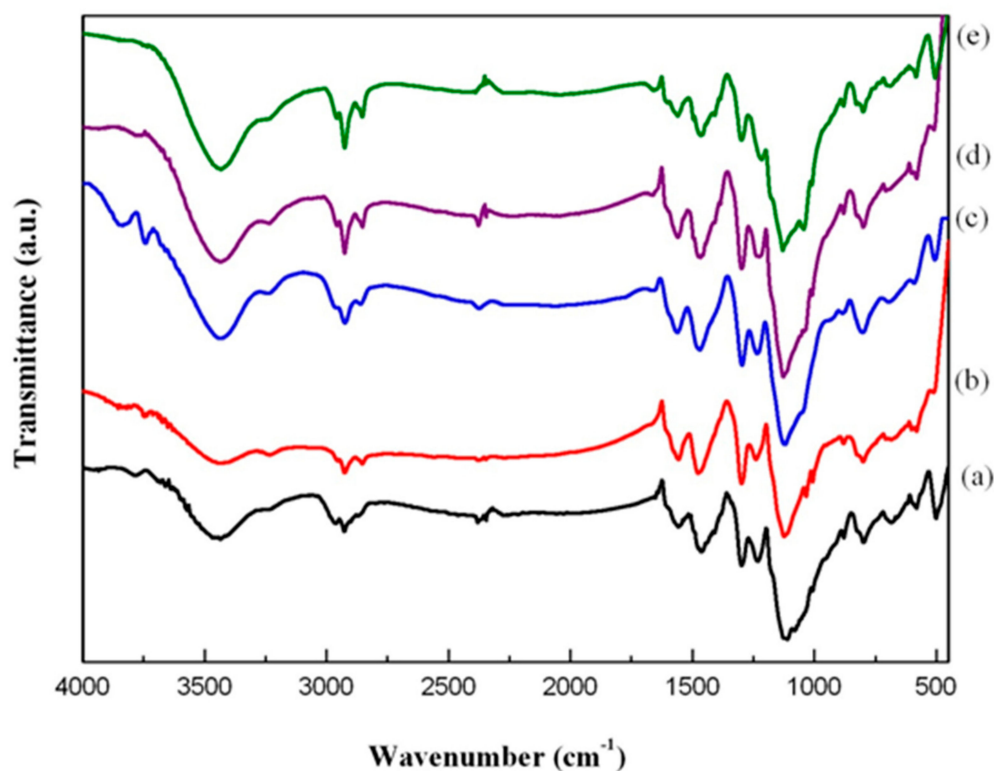
3. Results and Discussion

Table 1 shows the electrical conductivities of the pure PANDB and the conducting PANDB/ $\chi\text{-Al}_2\text{O}_3$ core-shell nanocomposites. The electrical conductivity of the pure PANDB is approximately $6.3 \times 10^{-1}\text{ S/cm}$. Table 1 indicates that the electrical conductivity of the conducting PANDB/ $\chi\text{-Al}_2\text{O}_3$ core-shell nanocomposites is increased as the weight ratio of AN/ $\chi\text{-Al}_2\text{O}_3$ is increased. When the weight ratio of AN/ $\chi\text{-Al}_2\text{O}_3$ is equal to 1.5, the conductivity of the conducting PANDB/ $\chi\text{-Al}_2\text{O}_3$ core-shell nanocomposite is approximately $1.7 \times 10^{-1}\text{ S/cm}$. This result implies that when the weight ratio of AN/ $\chi\text{-Al}_2\text{O}_3$ is higher, the amount of PANDB generated is higher. Therefore, the surface of $\chi\text{-Al}_2\text{O}_3$ can be covered more completely. As a result, the conducting path of the product increases, and the electrical conductivity increases. However, when the weight ratio of AN/ $\chi\text{-Al}_2\text{O}_3$ is increased to 2.0, the conductivity of the conducting PANDB/ $\chi\text{-Al}_2\text{O}_3$ core-shell nanocomposite remains at approximately $1.7 \times 10^{-1}\text{ S/cm}$. This result indicates that the optimum weight ratio of AN/ $\chi\text{-Al}_2\text{O}_3$ is equal to 1.5 in this study.

Table 1. Influence of the weight ratio of AN/ χ -Al₂O₃ on the electrical conductivity of the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites.

Weight Ratio of AN/ χ -Al ₂ O ₃	Conductivity (S/cm)
Pure PANDB	6.3×10^{-1}
0.5	0.6×10^{-1}
1.0	1.1×10^{-1}
1.5	1.7×10^{-1}
2.0	1.7×10^{-1}

Figure 1 shows the FTIR spectra of the pure PANDB and the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites. The absorption peak of the $-\text{CH}_2-$ stretching vibration (resulting from the DBSA molecules) was observed at 2926 cm^{-1} for all of the samples. For the pure PANDB (Figure 1(a)), the characteristic peaks at 1561 and 1467 cm^{-1} are due to the stretching vibrations of the $\text{N}=\text{Q}=\text{N}$ ring and the $\text{N}-\text{B}-\text{N}$ ring, respectively. The characteristic peak at 1300 cm^{-1} is attributed to the $\text{C}-\text{N}$ stretching vibration of the secondary amine in the PANDB main chain. The peaks at 998 – 1040 cm^{-1} are due to the asymmetric and symmetric $\text{O}=\text{S}=\text{O}$ stretching vibrations of DBSA. The characteristic peaks at 1100 – 1200 cm^{-1} are due to the $\text{B}-\text{NH}-\text{Q}$ bond or the $\text{B}-\text{NH}-\text{B}$ bond and the in-plane bending vibration of benzenoid or quinonoid $\text{C}-\text{H}$ (where B represents benzoic-type rings and Q represents quinonic-type rings). Because these absorption peaks were all located in the range of 1000 – 1200 cm^{-1} , a broad peak was observed instead of each single peak in the FTIR spectrum. The peaks at 700 – 800 cm^{-1} are attributed to a characteristic feature of the $\text{B}-\text{NH}-\text{Q}$ bond or the $\text{B}-\text{NH}-\text{B}$ bond and to the out-of-plane bending vibration of benzenoid or quinonoid $-\text{C}-\text{H}$ and $-\text{N}-\text{H}$ bonds. Additionally, the peaks at 2900 – 2950 and 3267 cm^{-1} are due to the stretching vibration mode of the $-\text{C}-\text{H}$ ($-\text{CH}_3$ or $-\text{CH}_2-$) and $-\text{N}-\text{H}$ bonds, respectively. These characteristic peaks are in good agreement with the data reported in the literature [28,29].

**Figure 1.** FTIR spectra of (a) pure PANDB and PANDB/ χ -Al₂O₃ composites synthesized at AN/ χ -Al₂O₃ weight ratios of (b) 0.5, (c) 1, (d) 1.5, and (e) 2.0.

For the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites (Figure 1(b)–(e)), as the weight ratio of AN/ χ -Al₂O₃ is increased, the peak at 1467 cm⁻¹, which is attributed to the N–B–N ring stretching vibration mode, is slightly shifted to 1472 cm⁻¹. This slight shift is due to the interaction between PANDB and the surface of the χ -Al₂O₃ nanoflakes. The other characteristic peaks of the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites are similar to those of pure PANDB.

Figure 2 presents the UV–Vis absorption spectra of the pure PANDB and the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites. Three characteristic absorption peaks are clearly observed in the UV–Vis spectrum of pure PANDB at approximately 350, 440, and 830 nm (Figure 2(a)). The absorption peak at approximately 350 nm is attributed to the π – π^* transition of the benzenoid rings, whereas the peaks at approximately 440 and 830 nm are attributed to the polaron– π^* transition and π –polaron transition, respectively [30]. The results indicate that the synthesized PANDB is an emeraldine salt (ES) PANI. Three characteristic absorption peaks also appear in the spectrum of the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposite because of the presence of PANDB (Figure 2(b)–(e)). Moreover, the intensity of the characteristic absorption peaks of the spectrum of the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposite is increased as the weight ratio of AN/ χ -Al₂O₃ is increased. This result is attributed to more PANDB being synthesized and coated onto the surface of the χ -Al₂O₃ nanoflakes.

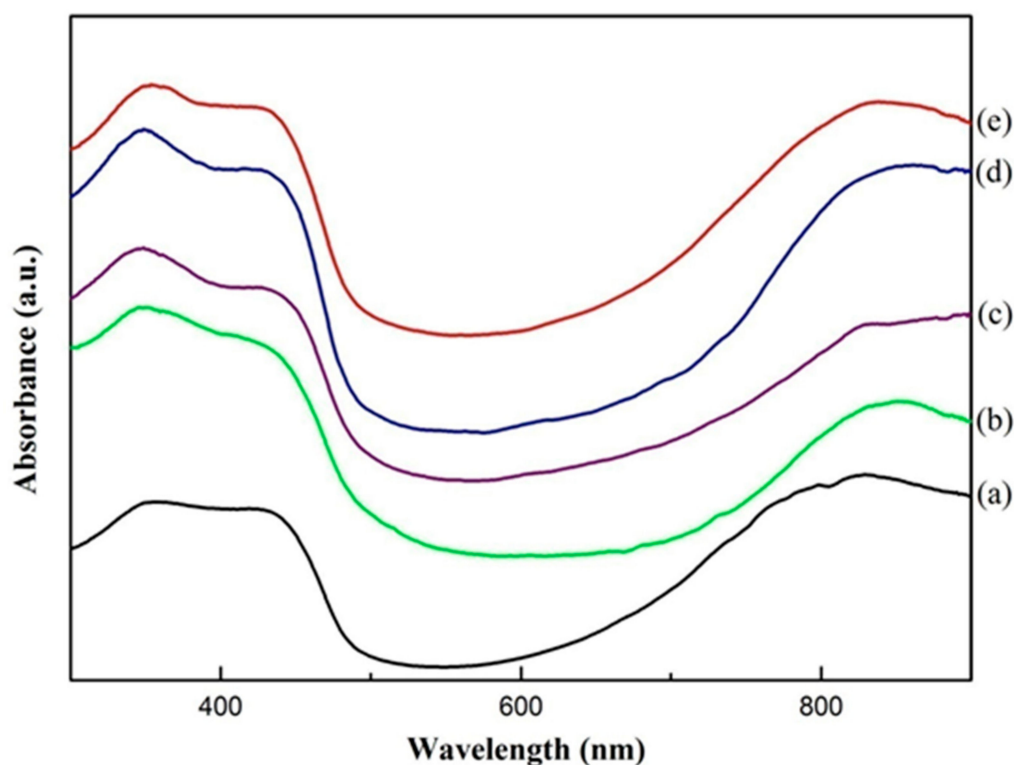


Figure 2. UV–Vis spectra of (a) pure PANDB and conductive PANDB/ χ -Al₂O₃ nanocomposites synthesized at AN/ χ -Al₂O₃ weight ratios of (b) 0.5, (c) 1, (d) 1.5, and (e) 2.0.

Figure 3 presents the XRD patterns of the χ -Al₂O₃ nanoflakes (Figure 3(a)), pure PANDB (Figure 3(b)), and the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites (Figure 3(c)–(f)). The pattern of the χ -Al₂O₃ nanoflake shows three characteristic peaks centered at $2\theta = 37.4^\circ$, 43.1° , and 67.6° . The spectrum of the pure PANDB shows two characteristic peaks centered at $2\theta = 20.0^\circ$ and 25.8° , which are attributed to the periodicity parallel and perpendicular to the polymer chain of PANDB, respectively [31]. It can be found that the patterns of the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites contain not only the two broad peaks centered at $2\theta = 20^\circ$ and 25.8° but also all of the

peaks associated with the χ -Al₂O₃ nanoflake, which indicate that the χ -Al₂O₃ nanoflakes are coated by the PANDB. It can also be found that when the weight ratio of AN/ χ -Al₂O₃ increases, the characteristic peaks of PANDB are more obvious (peaks 20.0° and 25.8°), while the characteristic peaks of χ -Al₂O₃ decrease (peaks 37.4°, 43.1°, and 67.6°). This is because when the weight ratio of AN/ χ -Al₂O₃ increases, more PANDB is synthesized and there is more PAND coating on the surface of the χ -Al₂O₃ nanoflake.

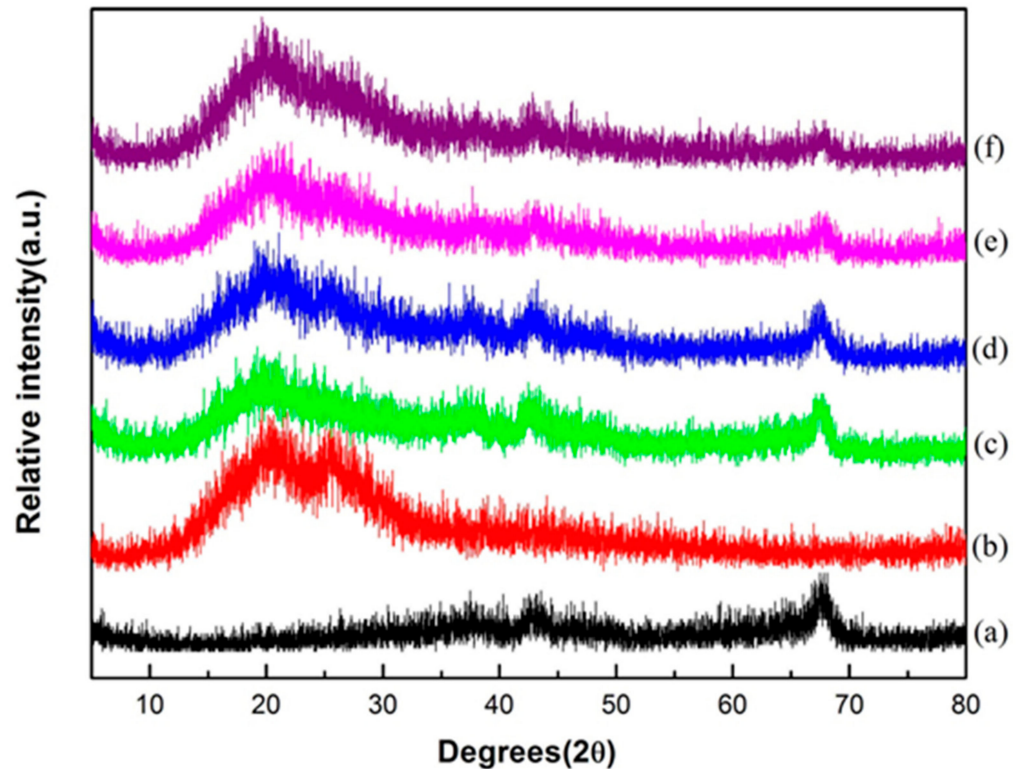


Figure 3. XRD patterns of (a) χ -Al₂O₃, (b) pure PANDB, and PANDB/ χ -Al₂O₃ composites synthesized at AN/ χ -Al₂O₃ weight ratios of (c) 0.5, (d) 1, (e) 1.5, and (f) 2.0.

Figure 4 shows the TEM images of pure PANDB and the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites. The morphology of the pure PANDB is fiber-like. Its diameter is approximately 0.2 μ m, and its length is approximately 1–3 μ m (Figure 4a). For the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites (Figure 4b–e), it can be found that the surface of the χ -Al₂O₃ nanoflake is coated with more PANDB as the weight ratio of AN/ χ -Al₂O₃ is increased from 0.5 to 2.0. Meanwhile, the electrical conductivity of the PANDB/ χ -Al₂O₃ core-shell nanocomposite is also increased from 0.6×10^{-1} S/cm to 1.7×10^{-1} S/cm (Table 1). If the weight ratio of AN/ χ -Al₂O₃ is 1.5 or 2.0, the synthesized products have a similar electrical conductivity at approximately 1.7×10^{-1} S/cm. This result indicates that the surface of the χ -Al₂O₃ nanoflake can be completely coated when the weight ratio of AN/ χ -Al₂O₃ is higher than 1.5 in this study. When the weight ratio of AN/ χ -Al₂O₃ is increased, more PANDB is synthesized, and then the more conductive paths are generated.

Figure 5 shows the FE-SEM images of pure PANDB (Figure 5a) and the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites (Figure 5b–d). The surface of pure PANDB exhibits a dense morphology. For the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites, the χ -Al₂O₃ flakes were more thoroughly coated by PANDB when increasing the weight ratio of AN/ χ -Al₂O₃. The surface morphologies of the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposite synthesized with AN/ χ -Al₂O₃ weight ratios of 1.5 and 2.0 (Figure 5d,e) are similar to that of pure PANDB. This means that when the AN/ χ -Al₂O₃ weight ratio is equal to 1.5, PANDB can cover χ -Al₂O₃ completely and form the conduct-

ing core-shell nanocomposite. Therefore, the optimum weight ratio of AN/ χ -Al₂O₃ is identified as 1.5 in this study.

The surface resistances of pure WPU film and WPU/20wt% (PANDB/ χ -Al₂O₃) blend films are around 6.0×10^{10} and $1.5 \times 10^4 \Omega/\text{sq.}$, respectively. It can be found that adding the 20wt% (PANDB/ χ -Al₂O₃) conducting nanocomposite into the WPU matrix can significantly reduce the surface resistance of the WPU film from 6.0×10^{10} to $1.5 \times 10^4 \Omega/\text{sq.}$ This is due to the formation of conducting paths through the WPU film by containing 20wt% conducting PANDB/ χ -Al₂O₃ nanocomposite. Figure 6 shows an antistatic comparison between the pure WPU film (Figure 6a) and the WPU/20wt%(PANDB/ χ -Al₂O₃) blend film (Figure 6b). The surfaces of the two films are first abraded and then placed among foamed polystyrene (PS) spheres. Clearly, many PS spheres are adsorbed on the pure WPU film, yet are not found on the WPU/20wt% (PANDB/ χ -Al₂O₃) blend film. This phenomenon indicates that the WPU/20wt% (PANDB/ χ -Al₂O₃) conducting blend film can function as a promising antistatic or electrostatic discharge material.

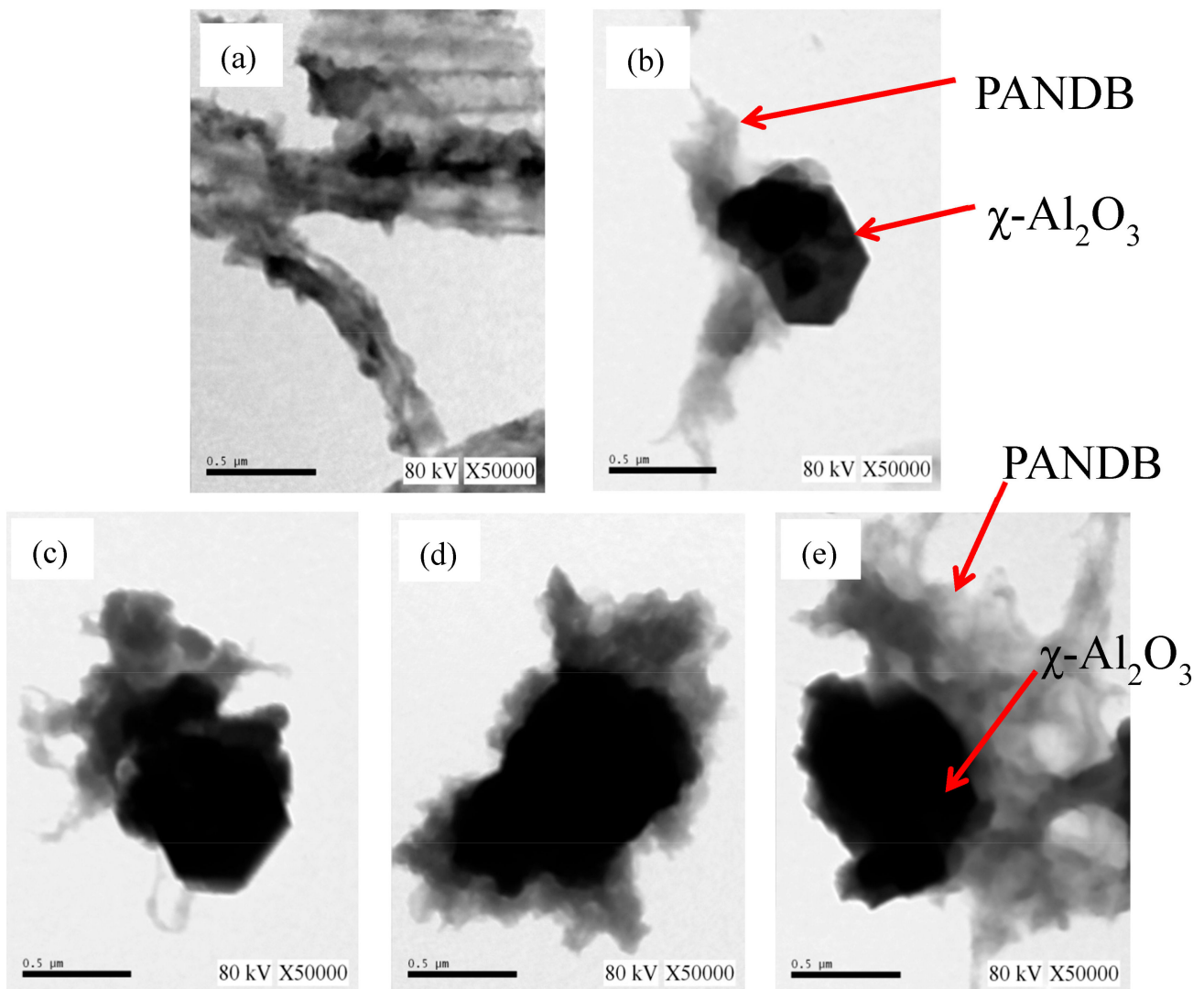


Figure 4. TEM images of (a) pure PANDB and conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites synthesized at AN/ χ -Al₂O₃ weight ratios of (b) 0.5, (c) 1, (d) 1.5, and (e) 2.0.

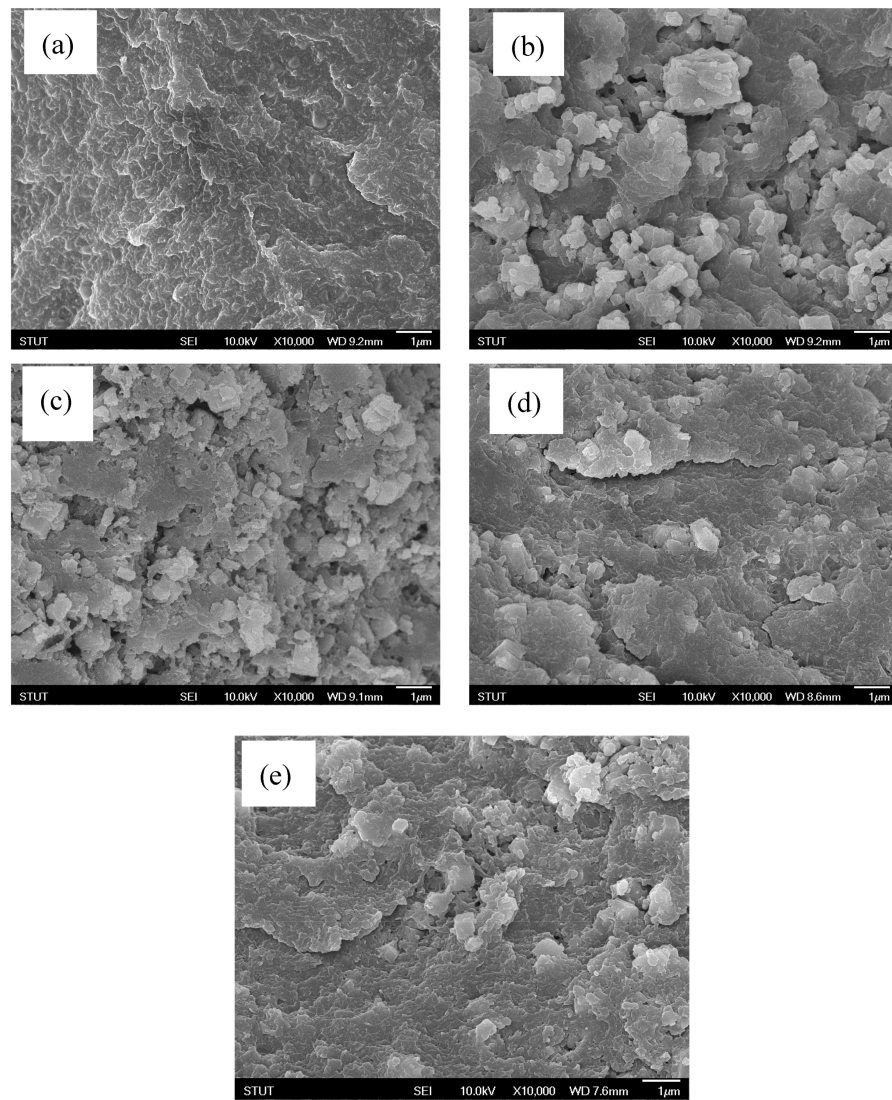


Figure 5. FE-SEM images of (a) pure PANDB and PANDB/ χ -Al₂O₃ composites synthesized at AN/ χ -Al₂O₃ weight ratios of (b) 0.5, (c) 1, (d) 1.5, and (e) 2.0.

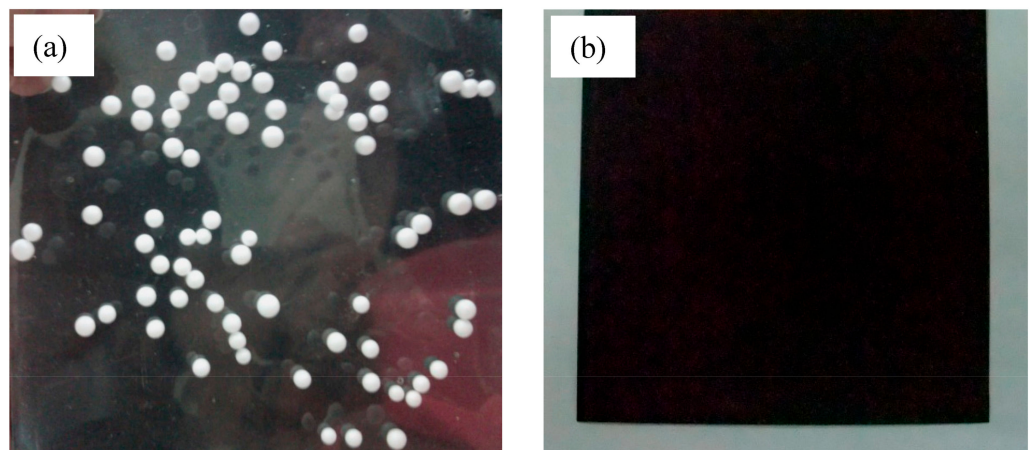


Figure 6. Antistatic tests: (a) pure WPU film, and (b) WPU/20wt% (PANDB/ χ -Al₂O₃) conducting blend film.

4. Conclusions

In this study, an in situ polymerization method was used to synthesize conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites. For the synthesized conducting core-shell nanocomposites, the core is a χ -Al₂O₃ nanoflake and the shell is conducting PANDB. The maximum electrical conductivity of the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposite is 1.7×10^{-1} S/cm when the AN/ χ -Al₂O₃ weight ratio is higher than 1.5. The TEM and FE-SEM images show that the χ -Al₂O₃ nanoflakes have been thoroughly coated by PANDB to form a core-shell structure when the weight ratio of AN/ χ -Al₂O₃ is higher than 1.5. Furthermore, the TEM images of the conducting PANDB/ χ -Al₂O₃ core-shell nanocomposites indicate that the thickness of the PANDB layer can be increased as the weight ratio of AN/ χ -Al₂O₃ is increased. It can be found that the optimum weight ratio of AN/ χ -Al₂O₃ is identified as 1.5 in this study. It can be found that adding a 20wt% (PANDB/ χ -Al₂O₃) conducting nanocomposite into the WPU matrix can significantly reduce the surface resistance of the WPU film from 6.0×10^{10} to 1.5×10^4 Ω /sq. The result indicates that the WPU/20wt% (PANDB/ χ -Al₂O₃) conducting blend film can function as a promising antistatic or electrostatic discharge material. In future work, the synthesized conducting PANDB/ χ -Al₂O₃ core-shell nanocomposite can be used as a novel and multiple functional reinforcement and be blended with semicrystalline and glassy thermoplastics, elastomers, or epoxy resins. Thus, PANDB is expected to provide electrical conductivity, and χ -Al₂O₃ nanoflakes are expected to enhance the mechanical properties of the polymer matrix. Meanwhile, the synthesized product can be used as antistatic and static discharge material and shows good potential for use in chemical sensors, biosensors, and anticorrosion.

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Conflicts of Interest: The authors declare no conflict of interest.

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