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Computational prediction of the spin-polarized semiconductor equiatomic quaternary Heusler compound MnVZrP as a spin-filter

 D. M. Hoat,^{ID}*^{ab} Duc-Quang Hoang,^{ID}^c Mosayeb Naseri,^{ID}^d J. F. Rivas-Silva,^e A. I. Kartamyshev^{fg} and Gregorio H. Coccoletzi^e

In this work, a new equiatomic quaternary Heusler (EQH) compound, MnVZrP, is predicted using first principles calculations. Simulations show the good stability of the material, suggesting experimental realization. Results show that MnVZrP is a magnetic semiconductor material, exhibiting semiconductor characteristics in both spin channels, however, with strong spin-polarization. Electronic band gaps of 0.97 and 0.47 eV are obtained in the spin-up and spin-down states, respectively. Mainly the d–d coupling regulates the electronic band structure around the Fermi level. Strain effects on the electronic properties of the proposed compound are also investigated. Simulations give the total magnetic moment of $3 \mu_B$ satisfying the Slater-Pauling rule. The main magnetic contributions are given by the Mn and V constituents. The results presented here suggest the promising applicability of EQH MnVZrP as a spin-filter. Additionally, the elastic property calculations indicate the mechanical stability and elastic anisotropy. The work may be useful in the magnetic Heusler alloys field, introducing a new member to the small group of magnetic semiconductor EQH compounds for spin-filter applications.

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1 Introduction

Recently, with the continuously increasing development of spintronic technology,^{1,2} numerous studies have focused on Heusler compounds due to their extremely intriguing physical and mechanical properties. Many members of this class of materials have been successfully applied in spintronic devices, such as spin filters,^{3,4} spin valves,^{5,6} magnetic tunnel junctions^{7,8} and spin injectors.^{9,10} It appears that the half-metallicity in the NiMnSb half-Heusler discovered by Groot *et al.*¹¹ has motivated extensive investigations into Heusler compounds of varying chemical composition. Besides the half-metallicity, novel

electronic properties have been obtained. So far, a large variety of Heusler alloys have been investigated, and they can be classified based on either chemical composition or electronic properties. Specifically,

(1) Chemical composition:

- Full-Heusler (FH) with stoichiometry 2-1-1, forming the chemical formula X_2YZ .^{12–14}
- Half-Heusler (HH) with stoichiometry 1-1-1, generating the chemical formula XYZ .^{15–17}
- Equiatomic quaternary Heusler (EQH) with stoichiometry 1-1-1-1, giving place to the chemical formula $XX'YZ$.^{13,18,19}

Herein, in most cases, X, X' and Y are transition metals or rare earth elements, however, interesting electronic and magnetic properties have been found even in those containing only sp elements.^{20–23}

(2) Electronic properties:

- Metal: both spin channels show the metallic nature (Fig. 1a).²⁴
- Half-metal: one of the spin states is metallic and the other exhibits semiconducting behavior (Fig. 1b).^{25–27}
- Semiconductor: both spin configurations are semiconductors with total symmetry, showing no magnetism (Fig. 1c).^{28,29}
- Spin-gapless semiconductor: the valence band maximum or conduction band minimum of one spin channel touches the Fermi level, and the remaining spin channel displays semiconducting characteristics (Fig. 1d).^{30,31}

^aComputational Laboratory for Advanced Materials and Structures, Advanced Institute of Materials Science, Ton Duc Thang University, Ho Chi Minh City, Vietnam. E-mail: dominhhoat@tdtu.edu.vn

^bFaculty of Applied Sciences, Ton Duc Thang University, Ho Chi Minh City, Vietnam

^cApplied Computational Civil and Structural Engineering Research Group, Faculty of Civil Engineering, Ton Duc Thang University, Ho Chi Minh City, Vietnam. E-mail: hoangducquang@tdtu.edu.vn

^dDepartment of Physics, Islamic Azad University, Kermanshah Branch, P.O. Box 6718997551, Kermanshah, Iran. E-mail: m.naseri@iauksh.ac.ir

^eBenemérita Universidad Autónoma de Puebla, Instituto de Física, Apartado Postal J-48, Puebla 72570, Mexico. E-mail: rivas@ifuap.buap.mx

^fDivision of Computational Physics, Institute for Computational Science, Ton Duc Thang University, Ho Chi Minh City, Vietnam. E-mail: kartamyshevandrey@tdtu.edu.vn

^gFaculty of Electrical and Electronics Engineering, Ton Duc Thang University, Ho Chi Minh City, Vietnam



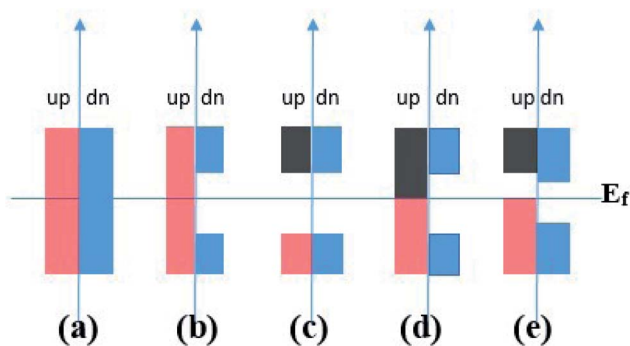


Fig. 1 Scheme of density of states of (a) metal, (b) half-metal, (c) semiconductor, (d) spin-gapless semiconductor and (e) spin-polarized semiconductor behavior.

- **Magnetic semiconductor:** the two spin states are semiconductors, however they differ from the semiconductor case by the spin-polarization, inducing important magnetic properties in the materials (Fig. 1e).³²

EQH compounds exhibit some advantageous features including small additional disorder scattering and low power dissipation.³³ These materials have been prepared and characterized experimentally. For example, the CrVTiAl with a small-moment has been successfully synthesized and characterized by Stephen *et al.*³⁴ to demonstrate its potential spin-filter applicability at room temperature. Two alloys, CoFeTiSn and CoFeTiGa, were prepared using arc-melting by Chatterjee *et al.*,³⁵ and CoFeVSi films have been grown by Yamada *et al.*³⁶ *via* molecular beam epitaxy. All of them possess extremely interesting magnetic and transport properties. Recently, CoFeV_{1-x}Mn_xSi films with positive linear magnetoresistance (PLMR) effects at low temperatures, have been prepared and characterized by Yamada *et al.*³⁷ Theoretically, EQH compounds have also been the subject of many studies, and most of them are found to have the half-metallicity feature, demonstrating the suitability for spintronic technology. Otherwise, not many EQH compounds have been reported to be magnetic spin-polarized semiconductors.

In order to search for more members and provide new information for the big Heusler alloys field, we have carried out first principles calculations to predict a new EQH compound, namely MnVZrP. To the best of our knowledge, no previous studies treating this compound have been performed. Herein, our main aim is to design the structure, examine the stability, as well as investigate the electronic, magnetic and elastic properties of the proposed material. It is expected that the material designed here will be experimentally prepared soon for applications in spintronic devices. We anticipate that MnVZrP exhibits magnetic semiconducting behavior, which makes it suitable for spin-filter devices.

2 Computational details

Density functional theory (DFT)³⁸ is an efficient approach to calculate and predict properties of materials. In this work, the

study of the structural, electronic, magnetic and elastic properties of the new equiatomic quaternary Heusler compound, MnVZrP, has been carried out using the all-electron full-potential linearized augmented plane-wave (FP-LAPW) method, which is executed in the WIEN2k package.³⁹ The revised version of the Perdew–Burke–Ernzerhof generalized gradient approximation (GGA-PBESol) is adopted for the exchange–correlation. It is well known that the band structure, electron density and magnetic properties of solids, including the highly correlated systems, are accurately described by the mBJ potential,^{40–43} which has been successfully applied to study the EQH compounds.^{44,45} In addition, a much more balanced description of the band gaps can be achieved using a reparametrization of the coefficients proposed by Koller *et al.*⁴⁶ (mBJK). Therefore, the mBJK potential is also included in the calculation of electronic and magnetic properties of the compound studied here. A k -mesh size of $10 \times 10 \times 10$ is used in the Brillouin zone integrations. Electronic states are expanded in plane wave basis functions up to $R_{\text{MT}}K_{\text{max}} = 8$, R_{MT} and K_{max} indicate the magnitude of the smallest muffin-tin radius and largest wave vector in the cell. The largest quantum number of the spherical harmonics is $l_{\text{max}} = 10$. An energy tolerance of 10^{-4} Ryd is selected for the self-consistent iterations.

3 Results and discussion

3.1 Structural properties

The EQH compound MnVZrP crystallizes in a face-centered cubic (FCC) structure ascribed to the $F\bar{4}3m$ space group, in which there are four unit formulae with Wyckoff positions at A(0; 0; 0), B(0.5; 0.5; 0.5), C(0.25; 0.25; 0.25) and D(0.75; 0.75; 0.75). By fixing the P atom at the A(0; 0; 0) position, three possible atomic arrangements can be formed as listed in Table 1, and each of them forms a different structure, namely, the α -, β - and γ -structure. In Fig. 2 these structures are displayed. In order to determine the most favorable structure, we calculate the energy of the non-magnetic (NM) and ferromagnetic (FM) states for all three structures varying the cell volume. Fig. 3 shows the energy difference $\Delta E = E - E^{\beta\text{-FM}}$. Note that in all cases ΔE takes a positive value, indicating that the ferromagnetic β -structure is the most stable in terms of the energy provided, it possesses the lowest energy at a given volume. The ground state structural parameters are determined using the Birch–Murnaghan equation of state:⁴⁷

$$E(V) = E_0 + \frac{9V_0B}{16} \left\{ \left[\left(\frac{V_0}{V} \right)^{2/3} - 1 \right]^3 B' + \left[\left(\frac{V_0}{V} \right)^{2/3} - 1 \right]^2 \left[6 - 4 \left(\frac{V_0}{V} \right)^{2/3} \right] \right\} \quad (1)$$

where B and B' are the bulk modulus and its derivative, which are found by fitting the energy–volume data to eqn (1). The ground state lattice constant is optimized by minimizing this equation. The obtained bulk modulus and lattice constant for

Table 1 Possible structural configurations of the equiatomic quaternary Heusler compound MnVZrP

	Mn	V	Zr	P
α -Structure	(0.25; 0.25; 0.25)	(0.5; 0.5; 0.5)	(0.75; 0.75; 0.75)	(0; 0; 0)
β -Structure	(0.25; 0.25; 0.25)	(0.75; 0.75; 0.75)	(0.5; 0.5; 0.5)	(0; 0; 0)
γ -Structure	(0.5; 0.5; 0.5)	(0.75; 0.75; 0.75)	(0.25; 0.25; 0.25)	(0; 0; 0)

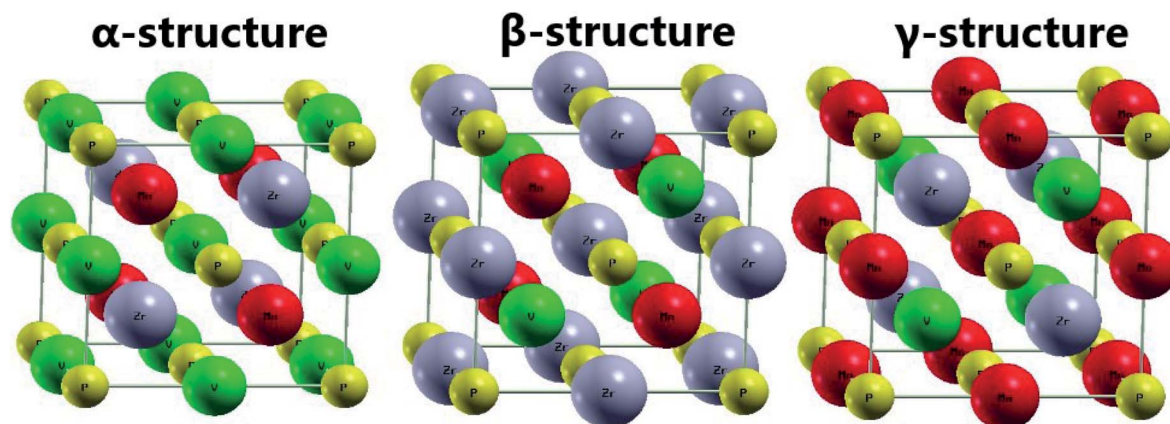
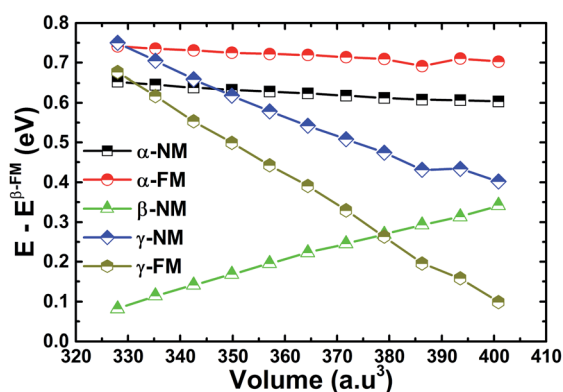


Fig. 2 Three structural configurations of the equiatomic quaternary Heusler compound MnVZrP.

Fig. 3 Energy difference versus volume of the equiatomic quaternary Heusler compound MnVZrP (energy of the β -structure in the ferromagnetic state is set as the reference).

the β -structure in the ferromagnetic phase are 181.81 (GPa) and 5.99 (\AA), respectively.

In theoretical work, calculations of the formation energy E_f and cohesive energy E_c are frequently carried out using the following expressions:

$$E_f = \frac{E_t - \sum_1^n E_b(A_n)}{n} \quad (2)$$

$$E_c = \frac{E_t - \sum_1^n E_a(A_n)}{n} \quad (3)$$

herein, E_t refers to the total energy of the studied material, n denotes the number of atoms, E_b and E_a are the energy per atom

in the bulk or isolated atom of the element A_n , respectively. Provided that E_f has a negative value (-0.46 eV per atom), the experimental formation of EQH compound MnVZrP can take place. Moreover, E_c characterizes the bonding strength in materials, with a calculated E_c negative value of -5.68 (eV per atom), MnVZrP is confirmed to be structurally stable conserving its structure once it has been formed. In addition, the phonon dispersion curves are calculated using the finite displacement method in combination with the norm-conserving pseudopotential as implemented in the CASTEP package.⁴⁸ The results displayed in Fig. 4 suggest the dynamical stability of the predicted EQH compound provided that there are no imaginary phonon modes in the figure.

3.2 Electronic properties

The spin-resolved band structure of EQH compound MnVZrP within its optimized lattice parameter has been calculated using the PBESol and mBJK potentials, as illustrated in Fig. 5. Both theories predict that the valence band maximum (VBM) touches the Fermi level at Γ point in the case of the spin-up channel, while the conduction band minimum (CBM) occurs at the Γ point and along ΓX direction as determined by the PBESol and mBJK potential, respectively. The direct and indirect band gap values are 0.33 and 0.97 eV, respectively. At the same time, the PBESol functional finds the metallic nature for the spin-dn channel provided that the CBM along the $W L$ direction takes place at an energy of 0.09 eV below the Fermi level. However, this spin-dn CBM point is located above the Fermi level as determined by the mBJK potential, giving rise to a clear semiconductor nature. It is worth recalling that the PBESol functional may underestimate the electronic band gap of solids and

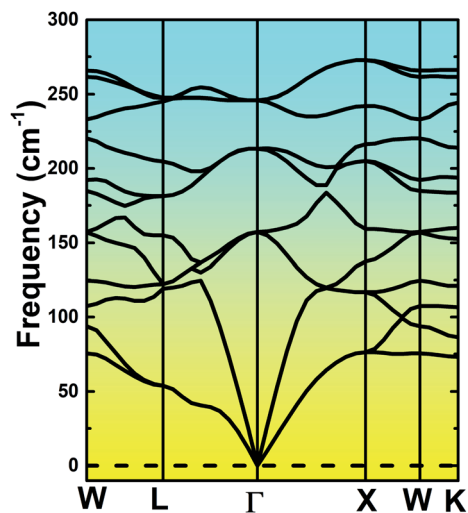


Fig. 4 Phonon dispersion curves of the equiatomic quaternary Heusler compound MnVZrP.

inefficiently treats the highly correlated electrons as those in transition metals, however, these issues are considerably overcome using the mBJK potential. Therefore, we believe that the mBJK potential results are more reliable, that is, the EQH compound MnVZrP exhibits semiconductor behavior in both of

its spin configurations. The spin-dn band gap is 0.47 eV resulting from the VBM at -0.31 eV and CBM at 0.16 eV, and both of them are situated along the WL direction. Moreover, significant magnetism can be expected in the material studied here as its band structure shows strong asymmetry between the spin-up and spin-dn states. Results show clearly the magnetic semiconducting characteristics of the studied compound, and show that prospectively it could be used as a spin-filter material in spintronic devices.

The total density of states (TDOS) and projected density of states (PDOS) yield additional information to describe electronic properties. Fig. 6 shows the TDOS and PDOS of the EQH compound MnVZrP, in which the positive DOS values represent the spin-up state, and negative DOS values indicate the spin-dn state. Note that in the energy range from -2 to 2 eV, the Mn-3d, V-3d and Zr-4d states are the most populated, while the contribution from the s and p orbitals of all four constituent atoms is quite small. It can be seen that in the spin-up channel, the Mn- $d_{t_{2g}}$ and V- $d_{t_{2g}}$ state make the most important contributions to the upper part of the valence band, whereas the Mn- d_{e_g} and V- d_{e_g} state are the main contributors to the lower part of the conduction band. Sub-bands nearest to the Fermi level are formed by the Mn- $d_{t_{2g}}$, V- $d_{t_{2g}}$ and Zr- $d_{t_{2g}}$ electrons. Results suggest that the spin-polarized semiconductor behavior of the EQH compound MnVZrP is governed mainly by the bonding and anti-bonding states generated by the d-d interactions of the

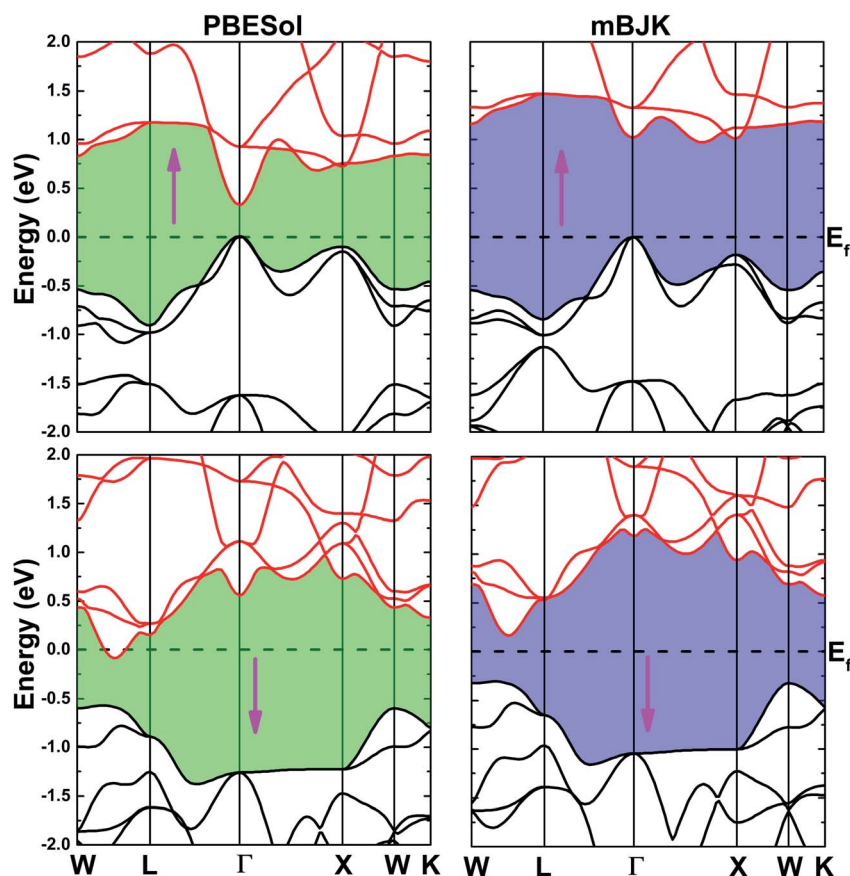


Fig. 5 Spin-resolved band structure of the equiatomic quaternary Heusler compound MnVZrP, calculated with PBESol and mBJK potentials.

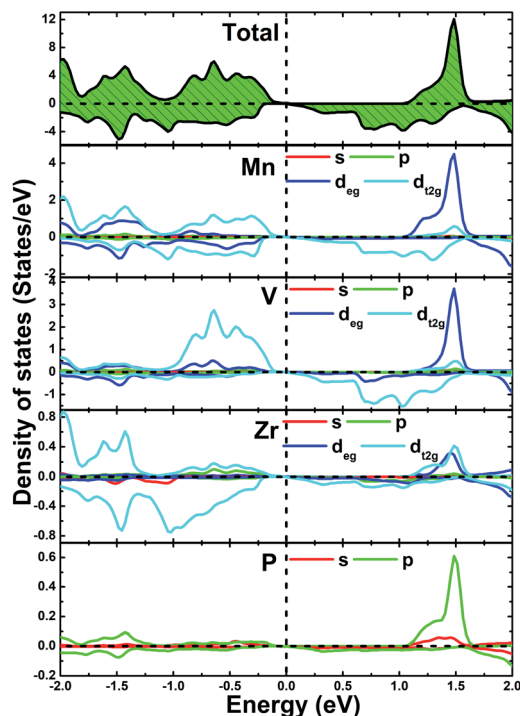


Fig. 6 Total and projected density of states of the equiatomic quaternary Heusler compound MnVZrP calculated with mBJK potential.

transition metals. Based on the DOS profiles, the mechanism of the d–d interaction, as well as d-electrons distribution in the considered compound, are indicated in Fig. 7. Specifically, the interaction of double-degenerated states Mn- d_{e_g} and V- d_{e_g} , generates the two-fold bonding state $2e_g$ and two-fold antibonding state $2e_u$, while the three-fold bonding state $3t_{2g}$ and three-fold antibonding state $3t_u$ are induced by the Mn- $d_{t_{2g}}$ and V- $d_{t_{2g}}$ states. Eight 3d electrons (5 of Mn and 3 of V) fill the spin-up channel of the $2e_g$, $3t_{2g}$ and $3t_u$ states, leaving the $2e_u$ state

totally unoccupied in the lower conduction band part. At the same time, the $3t_{2g}$ state hybridizes with the Zr- $d_{t_{2g}}$ state yielding the formation of the three-fold $3tt_{2g}$ bonding state and $3tt_u$ antibonding state. While the interaction between the $2e_g$ state with the Zr- d_{e_g} gives place to the formation of the two-fold $2ee_g$ bonding state and $2ee_u$ antibonding state. 7 electrons coming from the $2e_g$, $3t_{2g}$ and Zr- d_{e_g} occupy the last two mentioned bonding states. From the presented analysis, we can conclude that the $3t_u$ and $2e_u$ states formed the valence band upper part and conduction band lower part in the spin-up channel, while the spin-dn channel band gap is generated by the separation of the $3tt_{2g}$ and $3t_u$ states.

In the next step, we examine the strain effect on the MnVZrP compound's electronic structure. The strain is defined by means of a small lattice distortion as follows: $\delta = \frac{a - a_0}{a_0} \times 100\%$, the equilibrium and strained states are represented by their corresponding lattice parameter a and a_0 , respectively. The VBM and CBM positions calculated at different strains from -6% to 6% are listed in Table 2, and the results are plotted in Fig. 8 for visualization. In the case of the spin-dn channel, both VBM and CBM increase in terms of energy in most of the considered strain range when switching the strain from compressive to tensile, with exception of the CBM from a tension strength of 4%. Clearly, the MnVZrP spin-dn state retains the semiconductor nature for the strain ranging from -4% to 5% with the band gap increasing from compressive to tensile strain, beyond this it becomes metallic as either the CBM is found below the Fermi level or the VBM is above the Fermi level. Passing to the spin-up channel, the CBM approaches the Fermi level once the strain is applied to the lattice, it appears that the CBM is more sensitive to the tensile strains than to the compressive strain. Negligible variation is observed for the VBM, which is found at 0 eV in the strain range -4% to 4%. From a compression and tension strength of $\pm 5\%$, the studied material exhibits a metallic spin-up channel. It is worth mentioning that the half-metallicity is achieved at a tensile

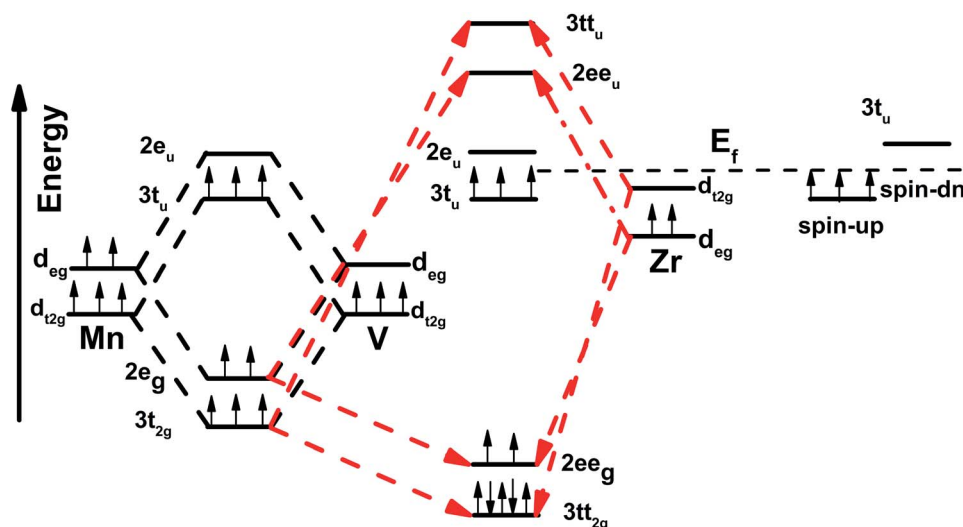


Fig. 7 Mechanism of d–d interactions and d-electrons distribution in the equiatomic quaternary Heusler compound MnVZrP.

Table 2 Calculated band edges (eV) and spin magnetic moments (μ_B) of the equiatomic quaternary Heusler compound MnVZrP at different strains

	Band edges				Magnetic moments				
	dn-VBM	dn-CBM	up-VBM	up-CBM	Total	Mn	V	Zr	P
−6%	−0.84	−0.20	0.03	0.49	2.85	1.10	1.58	−0.03	0.00
−5%	−0.70	−0.10	0.01	0.55	2.91	1.13	1.62	−0.03	0.00
−4%	−0.57	0.01	0.00	0.70	2.95	1.14	1.67	−0.04	−0.01
−3%	−0.50	0.06	0.00	0.79	2.96	1.11	1.69	−0.03	−0.01
−2%	−0.42	0.11	0.00	0.87	2.99	1.15	1.73	−0.05	−0.01
−1%	−0.38	0.12	0.00	0.93	3.00	1.16	1.74	−0.06	−0.02
0%	−0.31	0.16	0.00	0.97	3.00	1.18	1.76	−0.08	−0.03
1%	−0.25	0.18	0.00	0.80	3.00	1.22	1.77	−0.11	−0.03
2%	−0.21	0.19	0.00	0.55	3.00	1.33	1.77	−0.17	−0.04
3%	−0.16	0.21	0.00	0.33	3.00	1.27	1.80	−0.14	−0.04
4%	−0.11	0.23	0.00	0.12	3.00	1.29	1.81	−0.16	−0.05
5%	−0.08	0.05	0.00(1)	−0.10	3.00	1.29	1.82	−0.17	−0.05
6%	0.01	−0.11	0.02	−0.02	3.00	1.37	1.79	−0.21	−0.06

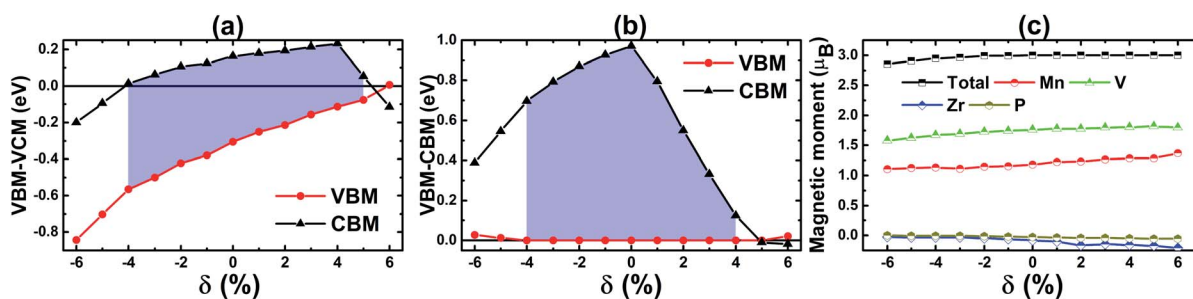


Fig. 8 Valence band maximum (VBM) and conduction band minimum (CBM) of the (a) spin-dn and (b) spin-up channels (colored areas indicate the semiconductor nature); and (c) total and atomic spin magnetic moments of the equiatomic quaternary Heusler compound MnVZrP calculated with mBJK potential.

strain of 5%, at the same time the spin-dn and spin-up configurations show metallic and semiconductor behavior, respectively.

3.3 Magnetic properties

As analyzed above, the asymmetric spin-resolved band structure may lead to significant magnetism in the MnVZrP compound. Since the MnVZrP has 21 valence electrons Z_t (7 of Mn atom, 5 of V atom, 4 of Zr atom and 5 of P atom), a total spin magnetic moment M_t of 3 μ_B can be expected when using the Slater-Pauling rule:⁴⁹

$$M_t = Z_t - 18 \quad (4)$$

At equilibrium, FP-LAPW results do obey eqn (4), that is, a total magnetic moment of 3 μ_B . The magnetism is produced mainly by the 3d transition metals with the atomic magnetic moments of Mn and V being 1.18 and 1.76 μ_B , respectively. It is important to clarify that these positive values evidence the ferromagnetic coupling between Mn and V atoms in the studied EQH compound. Moreover, the Zr- and P-contribution is quite small, with local atomic moments of −0.08 and −0.03 μ_B , respectively.

Total and atomic spin magnetic moments at different strains are listed in Table 2 and are plotted for visualization in Fig. 8c. Note that the integer total magnetic moment of 3 μ_B is retained in the strain range from −1% to 6%, in contrast stronger compressive strains induce a slight reduction of this parameter. The local atomic moment of the 3d transition metals increases when switching from lattice compression to tension. However, the opposite behavior is observed for Zr and P atoms. It is well known that a magnetic moment is the difference of the spin-up charge and spin-dn charge. Results suggest that the total spin-dn charge increases slightly with compressive strain, which can be attributed mainly to the stronger interaction of the d-d orbitals in the considered material being favored by the interatomic distance.

3.4 Elastic properties

In practical applications, materials suffer stress frequently which may generate structural deformations, and their ability to return to their original form after removing the stress is measured using elastic constants. Therefore, these constants are also indicative of the materials mechanical and physical properties. In this work, we calculate three different elastic constants C_{11} , C_{12} and C_{44} of MnVZrP, which are sufficient to

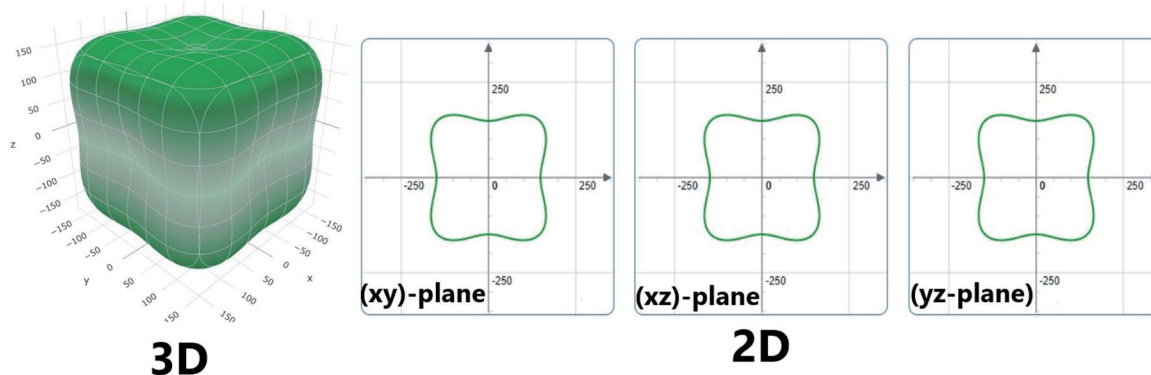


Fig. 9 Directional dependence of Young's modulus of the equiatomic quaternary Heusler compound MnVZrP.

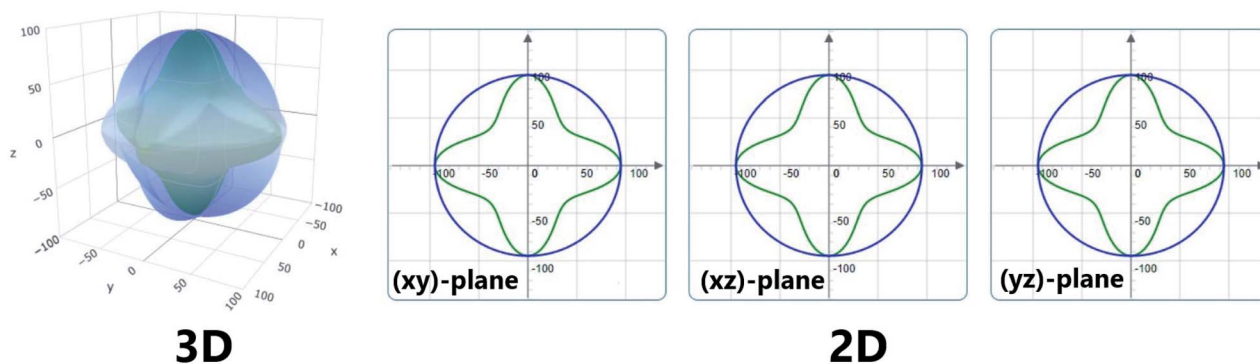


Fig. 10 Directional dependence of the shear modulus of the equiatomic quaternary Heusler compound MnVZrP.

investigate the elastic properties of materials adopting a cubic structure. To achieve this goal, the energy is calculated when small orthorhombic, cubic and tetragonal strains are applied to the lattice, and the elastic constants are determined by the second derivative of the energy with respect to strain (see Appendix). Our calculated elastic constants are: $C_{11} = 253.01$ GPa, $C_{12} = 143.88$ GPa and $C_{44} = 94.53$ GPa.

Mechanical stability Born criteria:⁵⁰ $C_{11} > C_{12}$; $C_{11} > 0$; $C_{44} > 0$ and $C_{11} + 2C_{12} > 0$ have been widely employed for solids, here we adopt these criteria. Note that for the EQH compound MnVZrP, elastic constants satisfy all the mentioned criteria, indicating that the material is mechanically stable. Moreover, greater resistance to the longitudinal deformation in comparison to that of the shear deformation is also demonstrated provided that the C_{11} is considerably larger than C_{44} . Further, various important mechanical parameters including bulk modulus, shear modulus, and Young's modulus can be calculated from elastic constants (see Appendix). In general, bulk modulus and shear modulus provide important information on the hardness of materials, while their stiffness is characterized by the Young's modulus. Calculations give the values of the bulk modulus, shear modulus, and Young's modulus as 180.26, 75.83 and 199.51 GPa, respectively.

Another important property that shall be examined is the ductile and brittle character of materials. The Pugh's index (B/G) and Poisson's ratio (see Appendix) with values larger than their critical points of 1.75 and 0.26, respectively, are indicative

of the ductility, and otherwise the brittleness is suggested. In the case of MnVZrP, these parameters take values of 2.38 and 0.32, respectively. These results in combination with the positive Cauchy pressure ($C_{12} - C_{44}$) indicate that the considered EQH compound is mechanically ductile.

The crack formation and intrinsic defects distribution within materials are linked to the elastic anisotropy, an important parameter that should be taken into account during material design. An elastically isotropic material will have a unity value of the elastic anisotropy. The calculated value of the MnVZrP compound is 1.73, significantly deviating from unity, indicating that the material studied here is elastically anisotropic. Additionally, we also examine the elastic anisotropy by calculating the directional dependent Young's modulus and shear modulus, their 3D surface construction and corresponding 2D projections are visualized by ELATE code,⁵¹ and are exhibited in Fig. 9 and 10. The totally spherical surfaces indicate the elastic isotropy, while the greater the deviation from a sphere form, the more anisotropic the material. Clearly, MnVZrP is classified as an elastically anisotropic material provided that non-spherical surfaces are obtained, in particular, stronger deviation is observed in the shear modulus.

4 Conclusions

In conclusion, the structural, electronic, magnetic and elastic properties of the new EQH compound MnVZrP have been

comprehensively investigated using first principles calculations based on DFT. The most significant results of the work are:

- MnVZrP crystallizes in the FCC structure (space group $F\bar{4}3m$), where Mn, V, Zr and P are located at the 4c, 4d, 4b and 4a Wyckoff positions, respectively.

- Thermodynamic and structural stability are confirmed by the negative formation and cohesive energies, respectively, suggesting the experimental realization of the material at hand.

- Semiconductor behavior is found in both spin channels, and is generated by the partially occupied hybridized states resulting from the d–d interactions of the transition metals.

- At equilibrium, Mn and V are the main contributors to the magnetism with atomic spin moments of 1.18 and 1.76 μ_B , respectively.

- Study of the strain effect shows that the material retains magnetic semiconductor characteristics in the range from –4% to 4%, in contrast the half-metal is induced with a tensile strain of 5%.

- The MnVZrP compound is mechanically stable with greater resistance to longitudinal deformation than to shear deformation. The elastic ductility and anisotropy of the material are also demonstrated.

We hope that these results motivate experimental work to synthesize and characterize the MnVZrP compound for spin-filter applications, proving that spin-polarized semiconductor characteristics are found in this material making it suitable for practical applications.

Appendices

The energy is expressed according to Hooke's law as follows:

$$E(\varepsilon) = E(0) + \sum_{i=1}^6 \frac{\partial E}{\partial \varepsilon_i} \varepsilon_i + \frac{1}{2} \sum_{i,j=1}^6 \frac{\partial^2 E}{\partial \varepsilon_i \partial \varepsilon_j} \varepsilon_i \varepsilon_j \quad (5)$$

Elastic constants are calculated through the second derivative of energy:

$$C_{ij} = \frac{1}{V_0} \frac{\partial^2 E}{\partial \varepsilon_i \partial \varepsilon_j} \quad (6)$$

Derivation of related parameters from the calculated C_{11} , C_{12} and C_{44} constants of the cubic structure:

Bulk modulus B :

$$B = \frac{C_{11} + 2C_{12}}{3} \quad (7)$$

Shear modulus G :

$$G = \frac{G_V + G_R}{2} \quad (8)$$

here the Voigt and Reuss components:

$$G_V = \frac{C_{11} - C_{12} + 3C_{44}}{5} \quad (9)$$

$$G_R = \frac{5(C_{11} - C_{12})C_{44}}{4C_{44} + 3(C_{11} - C_{12})} \quad (10)$$

Young's modulus:

$$E = \frac{9BG}{3G + B} \quad (11)$$

Elastic anisotropy A :

$$A = \frac{2C_{44}}{C_{11} - C_{12}} \quad (12)$$

Poisson's ratio:

$$\gamma = \frac{3B - 2G}{6B + 2G} \quad (13)$$

Conflicts of interest

No conflicts of interest to declare.

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Notes and references

- 1 I. Žutić, J. Fabian and S. D. Sarma, *Rev. Mod. Phys.*, 2004, **76**, 323.
- 2 V. K. Joshi, *Eng. Sci. Technol.*, 2016, **19**, 1503–1513.
- 3 I. Galanakis, K. Özdoğan and E. Şaşıoğlu, *J. Phys.: Condens. Matter*, 2014, **26**, 086003.
- 4 I. Galanakis, K. Özdoğan and E. Şaşıoğlu, *AIP Adv.*, 2016, **6**, 055606.
- 5 H. Sukegawa, S. Kasai, T. Furubayashi, S. Mitani and K. Inomata, *Appl. Phys. Lett.*, 2010, **96**, 042508.
- 6 K. Kodama, T. Furubayashi, H. Sukegawa, T. Nakatani, K. Inomata and K. Hono, *J. Appl. Phys.*, 2009, **105**, 07E905.
- 7 T. Ishikawa, T. Marukame, H. Kijima, K.-I. Matsuda, T. Uemura, M. Arita and M. Yamamoto, *Appl. Phys. Lett.*, 2006, **89**, 192505.
- 8 Z. Gercsi, A. Rajanikanth, Y. Takahashi, K. Hono, M. Kikuchi, N. Tezuka and K. Inomata, *Appl. Phys. Lett.*, 2006, **89**, 082512.
- 9 S. Chadov, T. Graf, K. Chadova, X. Dai, F. Casper, G. H. Fecher and C. Felser, *Phys. Rev. Lett.*, 2011, **107**, 047202.
- 10 R. Farshchi and M. Ramsteiner, *J. Appl. Phys.*, 2013, **113**, 7_1.
- 11 R. De Groot, F. Mueller, P. Van Engen and K. Buschow, *Phys. Rev. Lett.*, 1983, **50**, 2024.
- 12 F. Dahmane, Y. Mogulkoc, B. Doumi, A. Tadjer, R. Khenata, S. B. Omran, D. Rai, G. Murtaza and D. Varshney, *J. Magn. Magn. Mater.*, 2016, **407**, 167–174.

- 13 D. Hoat, J. Rivas-Silva and A. M. Blas, *J. Comput. Electron.*, 2018, **17**, 1470–1477.
- 14 A. O. Oliynyk, E. Antono, T. D. Sparks, L. Ghadbeigi, M. W. Gaultois, B. Meredig and A. Mar, *Chem. Mater.*, 2016, **28**, 7324–7331.
- 15 C. S. Birkel, W. G. Zeier, J. E. Douglas, B. R. Lettiere, C. E. Mills, G. Seward, A. Birkel, M. L. Snedaker, Y. Zhang, G. J. Snyder, *et al.*, *Chem. Mater.*, 2012, **24**, 2558–2565.
- 16 D. Hoat, *Comput. Mater. Sci.*, 2019, **159**, 470–477.
- 17 T. Djaafri, A. Djaafri, A. Elias, G. Murtaza, R. Khenata, R. Ahmed, S. B. Omran and D. Rached, *Chin. Phys. B*, 2014, **23**, 087103.
- 18 D. Hoat and M. Naseri, *Phys. B*, 2020, **583**, 412058.
- 19 Y. Han, Y. Wu, T. Li, R. Khenata, T. Yang and X. Wang, *Materials*, 2018, **11**, 797.
- 20 G. Gao, K. Yao, E. Şaşıoğlu, L. Sandratskii, Z. Liu and J. Jiang, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2007, **75**, 174442.
- 21 G. Gao, K. Yao, Z. Liu, J. Jiang, L. Yu and Y. Shi, *J. Phys.: Condens. Matter*, 2007, **19**, 315222.
- 22 D. Hoat, J. Rivas-Silva and A. M. Blas, *Chin. J. Phys.*, 2018, **56**, 3078–3084.
- 23 K. Kenmochi, V. Ann Dinh, K. Sato, A. Yanase and H. Katayama-Yoshida, *J. Phys. Soc. Jpn.*, 2004, **73**, 2952–2954.
- 24 M. Ilkhani, A. Boochani, M. Amiri, M. Asshabi and D. P. Rai, *Solid State Commun.*, 2020, **308**, 113838.
- 25 V. Alijani, J. Winterlik, G. H. Fecher, S. S. Naghavi and C. Felser, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2011, **83**, 184428.
- 26 E. Şaşıoğlu, L. Sandratskii and P. Bruno, *J. Appl. Phys.*, 2005, **98**, 063523.
- 27 M. Meinert, C. Friedrich, G. Reiss and S. Blügel, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2012, **86**, 245115.
- 28 S. A. Khandy, I. Islam, D. C. Gupta, R. Khenata and A. Laref, *Sci. Rep.*, 2019, **9**, 1–8.
- 29 I. H. Bhat, T. M. Bhat and D. C. Gupta, *J. Phys. Chem. Solids*, 2018, **119**, 251–257.
- 30 I. Galanakis, K. Özdoğan, E. Şaşıoğlu and S. Blügel, *J. Appl. Phys.*, 2014, **115**, 093908.
- 31 G. Xu, E. Liu, Y. Du, G. Li, G. Liu, W. Wang and G. Wu, *EPL*, 2013, **102**, 17007.
- 32 X. Wang, Z. Cheng, G. Liu, X. Dai, R. Khenata, L. Wang and A. Bouhemadou, *IUCrJ*, 2017, **4**, 758–768.
- 33 L. Bainsla and K. Suresh, *Appl. Phys. Rev.*, 2016, **3**, 031101.
- 34 G. M. Stephen, I. McDonald, B. Lejeune, L. H. Lewis and D. Heiman, *Appl. Phys. Lett.*, 2016, **109**, 242401.
- 35 S. Chatterjee, S. Das, S. Pramanick, S. Chatterjee, S. Giri, A. Banerjee and S. Majumdar, *J. Magn. Magn. Mater.*, 2019, **478**, 155–160.
- 36 S. Yamada, S. Kobayashi, F. Kuroda, K. Kudo, S. Abo, T. Fukushima, T. Oguchi and K. Hamaya, *Phys. Rev. Mater.*, 2018, **2**, 124403.
- 37 S. Yamada, S. Kobayashi, A. Masago, L. Kumara, H. Tajiri, T. Fukushima, S. Abo, Y. Sakuraba, K. Hono, T. Oguchi, *et al.*, *Phys. Rev. B*, 2019, **100**, 195137.
- 38 W. Kohn and L. J. Sham, *Phys. Rev.*, 1965, **140**, A1133.
- 39 K. Schwarz and P. Blaha, *Comput. Mater. Sci.*, 2003, **28**, 259–273.
- 40 A. Becke and E. Johnson, *J. Chem. Phys.*, 2006, **124**, 221101.
- 41 F. Tran, P. Blaha and K. Schwarz, *J. Phys.: Condens. Matter*, 2007, **19**, 196208.
- 42 F. Tran and P. Blaha, *Phys. Rev. Lett.*, 2009, **102**, 226401.
- 43 D. Koller, F. Tran and P. Blaha, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2011, **83**, 195134.
- 44 H. Kara, M. U. Kahaly and K. Özdoğan, *J. Alloys Compd.*, 2018, **735**, 950–958.
- 45 S. A. Khandy and J.-D. Chai, *J. Appl. Phys.*, 2020, **127**, 165102.
- 46 D. Koller, F. Tran and P. Blaha, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2012, **85**, 155109.
- 47 F. Birch, *J. Geophys. Res.: Solid Earth*, 1978, **83**, 1257–1268.
- 48 S. J. Clark, M. D. Segall, C. J. Pickard, P. J. Hasnip, M. I. Probert, K. Refson and M. C. Payne, *Z. Kristallogr. - Cryst. Mater.*, 2005, **220**, 567–570.
- 49 S. Skaftouros, K. Özdoğan, E. Şaşıoğlu and I. Galanakis, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2013, **87**, 024420.
- 50 F. Mouhat and F.-X. Coudert, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2014, **90**, 224104.
- 51 R. Gaillac, P. Pullumbi and F.-X. Coudert, *J. Phys.: Condens. Matter*, 2016, **28**, 275201.