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Three-dimensional graphene oxide cross-linked by benzidine as an efficient metal-free photocatalyst for hydrogen evolution†

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The use of low-cost photocatalysts to split water into H₂ fuel *via* solar energy is highly desirable for the production of clean energy and a sustainable society. Here three-dimensional graphene oxide (3DG) porous materials were prepared by cross-linking graphene oxide (GO) sheets using aromatic diamines (benzidine, 2,2'-dimethyl-4,4'-biphenyldiamine, 4,4'-diaminodiphenylmethane) that reacted with the carboxyl groups of the GO sheets at room temperature. The prepared 3DG porous materials were used as efficient metal-free photocatalysts for the production of H₂ *via* water splitting under full-spectrum light, where the photocatalytic activity was highly dependent on the cross-linker and the 3DG reduction level. It was also found that the 3DG prepared with benzidine as the linker demonstrated a significantly higher H₂ evolution rate than the 3DGs prepared using 2,2'-dimethyl-4,4'-biphenyldiamine and 4,4'-diaminodiphenylmethane as the cross-linkers. The photoactivity was further tuned by varying the mass ratio of GO to benzidine. Among the prepared 3DG materials, 3DG-3, with an intermediate C/O ratio of 1.84, exhibited the highest H₂ production rate (690 μmol g⁻¹ h⁻¹), which was significantly higher than the two-dimensional GO (45 μmol g⁻¹ h⁻¹) and the noncovalent 3DG synthesized by a hydrothermal method (128 μmol g⁻¹ h⁻¹). Moreover, this study revealed that the 3DG photocatalytic performance was favored by effective charge separation, while it could be further tuned by changing the reduction level. In addition, these results could prompt the preparation of other 3D materials and the application of new types of photocatalysts for H₂ evolution.

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Introduction

The utilization of solar energy and photocatalysts to split water into H₂ provides a low-cost and environmentally benign way of H₂ fuel production and a promising solution to solar energy storage as well.¹ In the past few decades, various metal-based semiconductor photocatalysts have been extensively studied for photocatalytic water splitting with the aim of producing H₂.^{2–7} Recently, researchers have been working to replace the expensive and rare-metal semiconductors by developing low-cost and sustainable, environment-friendly materials for efficient H₂ production. Carbon-based materials, such as graphite carbon nitride^{8–10} and carbon nanotubes,^{11,12} have been studied as metal-free semiconductor photocatalysts for water splitting. These carbon-based metal-free catalysts are expected to replace the metal-based photocatalysts owing to their wide distribution in nature.

Graphene, a two-dimensional (2D) carbon nanomaterial, has attracted much attention for the H₂ evolution *via* water splitting. Yeh *et al.*¹³ reported that graphene oxide (GO) with moderate oxidation level can be used as a photocatalyst for the production of H₂ from water. When GO is used as a semiconductor photocatalyst, the electronic properties of graphene are adjusted and the state density is altered by introducing vacancies¹⁴ or different oxidation levels,¹⁵ thereby forming a semiconductor band gap. It was found that the band gap of GO increased with increasing oxygen content, resulting in a decrease in photocatalytic activity. Putri *et al.*¹⁶ tailored GO with different oxidation degrees for the photocatalytic H₂ evolution and demonstrated that GO with an intermediate C/O ratio exhibited the highest photocatalytic activity by loading the optimum oxygen content on GO. At present, there are limited studies on the effect of the GO reduction degree on its photocatalytic performance for H₂ production.

In graphene-supported metal-containing semiconductor photocatalyst composites, graphene has been frequently used as a carrier or matrix material to promote the transfer and separation of photogenerated electron–hole pairs.^{17–20} However, conventional graphene-based photocatalysts tend to agglomerate when they are reduced.^{21–23} Three-dimensional graphene aerogel (3DGA) effectively solved this problem and has been

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widely used in the form of 3DGA-based photocatalysts in recent years. 3D graphene not only retains the large specific surface area and the good electrical conductivity of the 2D graphene nanosheets²⁴ but provides an effective multidimensional electron transport pathway, which is favorable for the separation of photogenerated electron–hole pairs from the composite photocatalysts.²⁵ Ho *et al.* used vitamin C as a reducing agent and cross-linker to synthesize a photocatalytic composite (NGH–Au), which consisted of 3D graphene hydrogel (NGH), while TiO₂ nanorods and Au nanoparticles served as the photoactive materials.²⁶ The generated composite exhibited a higher H₂ production efficiency than the corresponding 2D graphene composite (RGO–Au) under ultraviolet-visible (UV-vis) light irradiation.

Furthermore, Chen *et al.* discovered that a carbon-only 3D graphene oxide (3DG), apart from its role as an electronic reservoir accepting and transferring photogenerated electrons produced by the semiconductor,²⁷ can be directly used as a photocatalyst to generate H₂ *via* water splitting under UV-vis light.²⁸ It was also found that the calcination treatment can enhance the photoactivity of the 3DG for H₂ evolution. This photocatalyst was synthesized by a hydrothermal reduction method, which has been widely used to prepare 3D graphene materials. However, those synthetic processes usually require high temperature and high pressure conditions, which could be uncontrollable and energy consuming.

In this study, we presented a simple and effective method to obtain a covalent 3DG network using an aromatic diamine as the covalent linker to functionalize GO at room temperature. The method relied on the condensation reaction of the carboxyl groups on GO with aromatic diamines (benzidine, 2,2′-dimethyl-4,4′-biphenyldiamine (DMPDA), 4,4′-diaminodiphenylmethane (MDA)), in the presence of a carbodiimide. The photocatalytic performance of the prepared 3DG as the metal-free photocatalyst for H₂ evolution was investigated, and it was proven that 3DG-3—with benzidine as the linker—remarkably enhanced the H₂ evolution rate (690 μmol g⁻¹ h⁻¹), which was about 5.4 times higher than that of the noncovalent 3DG network synthesized by a conventional hydrothermal method (128 μmol g⁻¹ h⁻¹). Tuning the H₂ production performance of the catalyst by varying the reduction level on the prepared covalent 3DG network was also discussed.

Experimental

Materials

Natural flake graphite (99 wt% purity) with average particle diameter of 40 μm was purchased from Sinopharm Chemical Reagent Co., Ltd. Shanghai, China. All other reagents and solvents were purchased from Adamas Reagent Co., Ltd. and were used without further purification.

Characterization

The Fourier-transform infrared (FTIR) spectra were collected using a Shimadzu FTIR (IR Prestige-21) spectrometer. X-ray powder diffraction (XRD) was performed with a Rigaku Dmax

3C diffractometer using Cu K α radiation ($\lambda = 0.15408$ nm) at 40 kV and 20 mA. The UV-vis absorption spectra of the samples were recorded using a Shimadzu UV-2600 spectrophotometer. The photoluminescence (PL) spectra were recorded on a fluorospectrophotometer (Horiba Fluoromax-4) at room temperature. The chemical states of the surface elements were analyzed by X-ray photoelectron spectroscopy (XPS, Thermo Fisher ESCALAB 250Xi). The specific surface area was measured using a Micromeritics ASAP 2460 surface area and porosity analyzer. The Raman spectra were obtained on a confocal microscopy Raman spectrometer (excitation at 532 nm). The morphology of the 3DG materials was examined using scanning electron microscopy (SEM) (S-3400N).

Preparation of 3DG

GO was synthesized from natural graphite by modifying the method developed by Marcano *et al.*²⁹ To increase the oxygen-containing groups on GO, it was subjected to secondary oxidation. Double oxidized GO was synthesized using a modification of the Hummers' method.³⁰ In particular, the synthesized GO was mixed with deionized water and ultrasonically dispersed to form a 1.5 mg mL⁻¹ GO suspension. Subsequently, appropriate amounts of 1-ethyl-3-(3-dimethylaminopropyl)carbodiimide (EDC, 0.3 mmol) and *N*-hydroxysuccinimide (NHS, 0.3 mmol) were added to 100 mL of the GO suspension. The organic diamine (0.22 mmol) was dissolved in 2.0 mL DMF and then added to the GO suspension. The mixed solution was stirred at room temperature for 24 h to promote the cross-linking of GO with the diamine monomer. The cross-linked GO was washed three times, each with *N,N*-dimethylformamide (DMF) and deionized water, to remove the unreacted diamine monomer, and separated by a high speed centrifuge. The cross-linked GO powder was then freeze-dried for 2 days. The products prepared using powdery GO and benzidine, DMPDA, and MDA were labelled as GO-benzidine, GO-DMPDA, and GO-MDA, respectively. In addition, a series of five 3DG GO-benzidine composites were prepared with varying benzidine amounts, *i.e.*, 40, 60, 80, 100, and 120 mg, in DMF and mixed with the above GO suspension. The produced 3DG composites were designated as 3DG-1, 3DG-2, 3DG-3, 3DG-4, and 3DG-5, respectively.

The control sample of noncovalent 3DG (3DGA) was obtained by a hydrothermal method.³¹ Typically, 50 mL of a GO solution (2 mg mL⁻¹) was sealed in a 100 mL Teflon-lined autoclave, heated to 180 °C, and maintained at this temperature for 12 h. After cooling, the produced 3DGA was freeze-dried.

Photocatalytic measurements for H₂ evolution

The photocatalytic experiments were conducted in a gas-sealed system using a heat-resistant glass cell with a quartz window on top. Typically, a 10 mg sample was dispersed in a triethanolamine (TEOA) aqueous solution (50 mL, volume ratio of water : TEOA = 4 : 1) using a magnetic stirrer, and the resulting mixture was irradiated with a 300 W Xe lamp (CEL HXF300) under inert atmosphere. A water-cooling circulator was used to control the reaction at 25 °C. The reaction was carried out for

10 h and the generated H₂ gas was analyzed by a gas chromatograph (SHIMADZU GC-2014) equipped with a thermal conductivity detector.

Results and discussion

Preparation and characterization of 3DGs

The 3DG samples were obtained by the chemical modification of GO with different amounts of benzidine (3DG-1, 3DG-2, 3DG-3, 3DG-4, and 3DG-5), where the GO sheets were linked by the reaction of their carboxyl groups on both sides with the aromatic diamine groups of the linker (Fig. 1). The first indication of the GO reduction was the conversion of the aqueous GO solution from yellow to brown after the addition of benzidine (Fig. 1). The functional groups on GO and 3DG were determined by FTIR spectroscopy (Fig. 2a). Specifically, for GO, the absorption peaks at 3435, 1735, 1630, 1396, 1220, and 1074 cm⁻¹ corresponded to the hydroxyl group of the water molecule (O–H), carboxyl group (C=O), aromatic skeleton carbon rings (C=C), deformation vibrations of the O–H bond (CO–H), C–OH stretching, and epoxy (C–O–C) moiety, respectively. After the cross-linking of GO with benzidine, the characteristic bending vibration peak of the amide (–NH–) group appeared at 1550 cm⁻¹ (amide II band, 1460–1554 cm⁻¹),³² while the relative intensity of the C=O peak was significantly reduced compared to the starting material (GO). Thus, it could be inferred that the condensation reaction between the carboxyl group and the amino group formed the amide bond, indicating that benzidine was covalently attached to GO. Moreover, the observed peaks at 1497 and 808 cm⁻¹ could be ascribed to the C–N and N–H stretching vibrations of benzidine, respectively.

The UV-vis absorption spectra of 3DG in ethanol are depicted in Fig. 2b. Two main characteristic peaks were observed for the GO solution: one at 230 nm, caused by the π – π^* transition of the aromatic structure, and a shoulder around 300 nm due to the n – π^* transitions of the carbonyl groups.³³ For the 3DG-1 solution, the absorption peak shifted from 230 to 282 nm. By increasing the mass of benzidine to 120 mg, the same peak further shifted to 287 nm, indicating the reduction of GO and the appearance of electronic conjugation in the graphene sheets due to the covalent interactions.^{34,35} In addition, the inset of Fig. 2b displays the square of the absorption energy (Ahv^2 , where A is the absorbance) against the photon energy (hv). Moreover, the Kubelka–Munk plots^{36,37} implied that the band gap energy value of GO was 2.4–4.3 eV for the semiconductor photocatalyst.¹³ As the reduction degree increased, the 3DG-1, 3DG-2, 3DG-3, 3DG-4, and 3DG-5 samples displayed decreasing band gap energies of 3.65, 3.65, 3.60, 3.57, and 3.50 eV, respectively.

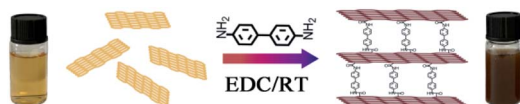


Fig. 1 Preparation scheme of the 3DG materials with benzidine as the cross-linker.

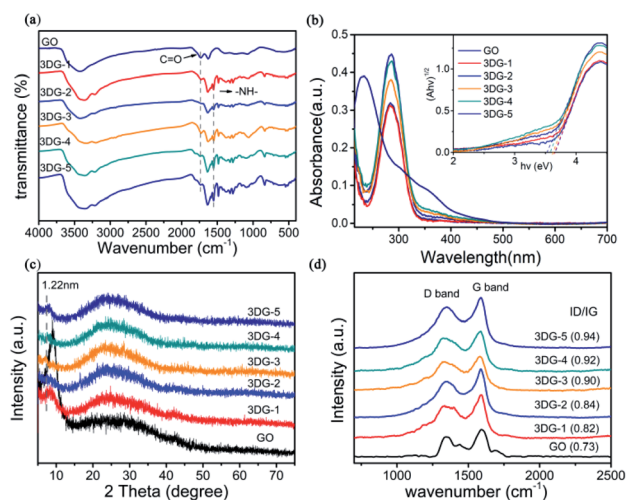


Fig. 2 Spectroscopic characterization of GO and 3DG (3DG-1, 3DG-2, 3DG-3, 3DG-4, and 3DG-5) with successively increasing reducing properties. (a) FTIR spectra (b) UV-vis absorption spectra recorded in ethanol. The inset shows the $(Ahv)^2$ vs. hv plots for GO and the five 3DGs. (c) XRD patterns. (d) Raman spectra. The numbers in brackets represent the D/G intensity ratio of the samples.

This result was consistent with the relationship between the oxygen density and the energy band of the reduced GO.^{38,39}

Furthermore, the covalent functionalization of GO with benzidine could also be observed by the corresponding XRD patterns (Fig. 2c). The characteristic sharp peak of GO at 9° corresponded to an interlayer spacing of 0.98 nm, while the prepared 3D functionalized graphene-based networks exhibited diffraction peaks at smaller diffraction angles (8.4°, 8.2°, 7.2°, 8°, and 7.8° for 3DG-1, 3DG-2, 3DG-3, 3DG-4, and 3DG-5, respectively). According to the Bragg equation, the d_{001} spacing of 3DG-1, 3DG-2, 3DG-3, 3DG-4, and 3DG-5 was calculated to be 1.07, 1.05, 1.22, 1.10, and 1.13 nm, respectively, suggesting an increasing enlargement in the interlayer distance compared to GO, which could be attributed to the cross-linking of benzidine in the GO sheets by the amide formation reaction.^{40,41} In addition, a broad peak (002) appeared at 24.6°, indicating the efficient reduction of GO by benzidine.³¹

The reduction degrees of the five 3DG materials were further compared by Raman spectroscopy (Fig. 2d), which is a well-known, powerful tool for characterizing graphite materials. The common characteristic Raman bands of GO are the D band (1350 cm⁻¹), resulting from the sp³ carbon atom in the defect structure, and the G band (1591 cm⁻¹) corresponding to the in-plane vibration mode of the sp² hybridized carbon. Based on Fig. 2d, all five 3DG samples revealed the D- and G-bands of the GO skeleton structure. Interestingly, the Raman spectra of GO cross-linked with various benzidine amounts exhibited higher D/G intensity ratios than GO (0.73), while the D/G intensity ratio increased by increasing the benzidine mass (40, 60, 80, 100, and 120 mg) from 0.82 to 0.94 for 3DG-1 to 3DG-5, respectively. The increase in the D/G intensity ratio in the Raman spectrum of the prepared 3DGs also suggested that the number of the sp³ domains increased.⁴²

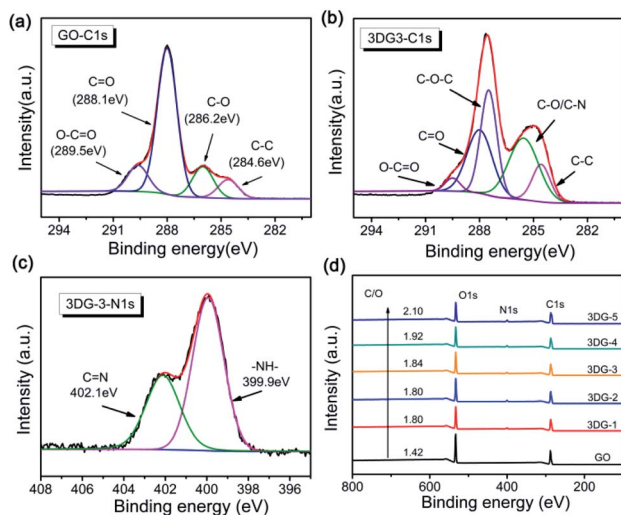


Fig. 3 C 1s XPS spectra of (a) GO and (b) 3DG-3. (c) N 1s XPS spectra of 3DG-3. (d) XPS survey spectrum of GO and the 3DG samples.

The composition and chemical status of GO and 3DGs were also analysed by XPS. The C 1s XP spectrum of GO (Fig. 3a) could be decomposed into four peaks that corresponded to the unoxidized graphite carbon skeleton (C–C, 284.6 eV), epoxy/hydroxyl carbon (C–O, 286.2 eV), carbonyl (C=O, 288.1 eV), and carboxylate functional group (O–C=O, 289.5 eV).⁴³ After cross-linking with benzidine, a new signal corresponding to the C–N group of 3DG-3 appeared at 285.6 eV, while the peak intensity of the C=O and O–C=O groups decreased substantially (Fig. 3b). The cross-coupling reaction between the carboxyl group of GO and the benzidine amine to form amide bonds was also confirmed by the C 1s spectra of the other 3DG samples (ESI, Fig. S5†).⁴⁴ Different N-containing species in the 3DGs could be revealed by the N 1s XPS spectra (Fig. 3c), where the broad peak could be divided into two components with binding energies at 399.9 and 402.1 eV that were assignable to the N-atom of the amide (O=C–N) and imine groups (C=N), respectively. The presence of the (O=C–N) peak confirmed the chemical reaction between GO and benzidine, which has been suggested by the C 1s XP spectrum as well. Furthermore, the results of the N 1s XPS analysis of the other 3DG samples were consistent to the above results (ESI, Fig. S6†). Moreover, the C/O atomic ratio increased from 1.42 for GO to 1.80 for 3DG-1, indicating the reduction of GO by benzidine (Fig. 3d), while the C/O ratio increased gradually from 3DG-1 to 3DG-5, highlighting that the reduction degree of the 3DGs increased by increasing the benzidine amount during the reaction.

Morphology characterizations

The morphology of GO and the 3DGs (3DG-1 to 3DG-5) were revealed by SEM images (Fig. 4). 2D GO displayed a very large anisometric shape with face-to-face stacking of the sheets, resulting in the interlayer aggregation with narrow interlayer spacing. As shown in Fig. 4a, the surface of the sheet of GO was quite smooth. In contrast, the surface of the benzidine cross-linked GO samples was rougher and possessed highly porous

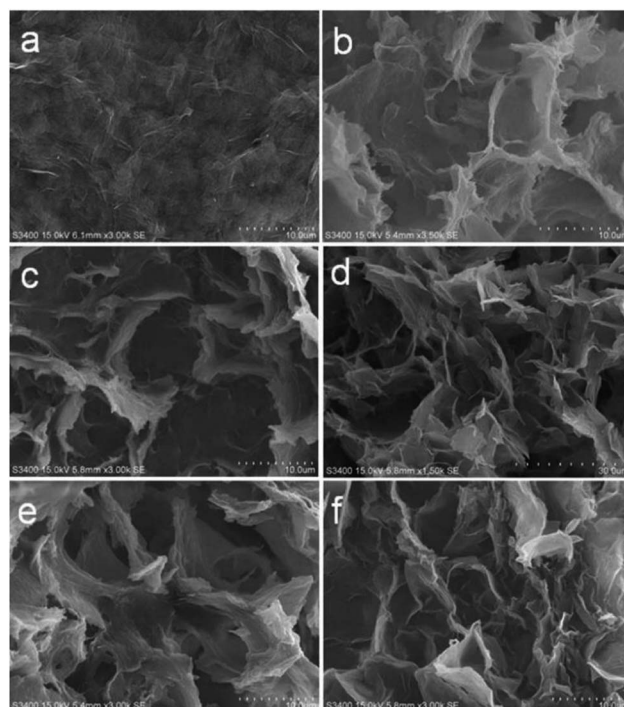


Fig. 4 SEM images of (a) GO, (b) 3DG-1, (c) 3DG-2, (d) 3DG-3, (e) 3DG-4, and (f) 3DG-5.

surfaces (Fig. 4b–f), indicating that treating 2D GO flakes with rigid benzidine molecules hinders the face-to-face stacking. However, no significant differences in the morphology of the five samples were observed.

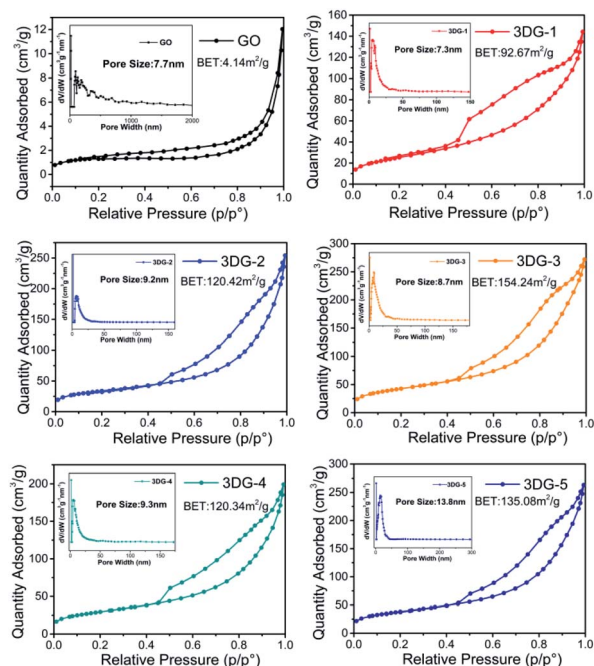


Fig. 5 N₂ adsorption–desorption isotherm pore size distribution of the prepared 3DG samples.

To improve our understanding in the morphology of the five prepared materials, N_2 adsorption–desorption isotherms were recorded (Fig. 5). The specific surface area of the synthesized GO was $4.14 \text{ m}^2 \text{ g}^{-1}$, which was significantly lower than the theoretical specific surface area of graphene ($2600 \text{ m}^2 \text{ g}^{-1}$). This common problem was caused by the face-to-face aggregation of the GO sheets during the wet processing. Instead, the specific surface area of the 3DGs increased from 3DG-1 ($92.67 \text{ m}^2 \text{ g}^{-1}$) to 3DG-3 ($154.24 \text{ m}^2 \text{ g}^{-1}$) and then declined to 3DG-5 ($135.08 \text{ m}^2 \text{ g}^{-1}$). The obvious increase in the surface area of the 3DGs indicated that the face-to-face stacking was effectively limited by the rigid cross-linkers derived from benzidine. Thus, with the appropriate amount of benzidine in the 3DGs, the benzidine molecules could serve as pillaring agents, thus separating the graphene sheets to form a porous network between them. On the contrary, benzidine excess leads to the graphene sheets stacking through π - π interactions. Consequently, as the amount of benzidine increased, the specific surface area of the 3DGs gradually increased and then decreased, while 3DG-3 exhibited the highest specific surface area.

Photocatalytic H_2 evolution

The photocatalytic H_2 evolution properties of the 3DGs prepared with different diamines (GO-benzidine, GO-DMPDA, GO-MDA) were evaluated under full-spectrum irradiation in the presence of a TEOA solution as the sacrificial agent. The H_2 evolution rate of the three 3DG samples bearing a unique 3D network structure were significantly enhanced relative to the control GO, especially for GO-benzidine (Fig. S3†). The enhanced specific surface area of these three samples (Fig. S4†) apparently increased their adsorption capacity and the number of active sites, which in turn improved the photocatalytic performance. Surprisingly, GO-MDA that had the largest specific surface area among the three samples did not yield the highest H_2 evolution rate. For the synthesized GO-DMPDA and GO-MDA, they all showed larger specific surface area than GO-benzidine (Fig. S4†), but their photocatalytic activity was lower than GO-benzidine. This suggests that the specific surface area of 3DGs does not play decisive role on the photocatalytic H_2 production.

While, the reduction degree of GO was found to be an important factor for the H_2 evolution in the photoinduced water splitting process with the GO-based catalyst.¹⁵ The aromatic diamine molecules of MDA, DMBPDA and benzidine were employed both as a pillar to separate the GO sheets and as a reducing agent to reduce GO to prepare the 3DGs. Thus, the reduction degree of the prepared 3DGs varied with the diamine linker, among which benzidine turned out to be the most suitable cross-linker with appropriate reducibility, rendering the GO-benzidine 3DG with the highest H_2 evolution rate. Tuning the reducibility by variation of the structure of *p*-phenylene diamine was previously noticed as well.⁴⁷

Next, the reduction level of GO in the 3DGs was then further regulated by altering the mass ratio of GO to benzidine, which afforded a series of five 3DGs (3DG-1 to 3DG-5) with increasing benzidine amounts. The results from the above C/O ratio and

the band gap analysis revealed that the reduction degree of the 3DGs was positively correlated with the increase in the benzidine mass. To exclude the effect of the specific surface area on the H_2 production rate, the evolution rate ($\mu\text{mol g}^{-1} \text{ h}^{-1}$) was further divided by each sample's specific surface area value. 3DG-5 with relative large specific surface area was found to show the lowest H_2 production rate ($3.25 \mu\text{mol g}^{-1} \text{ m}^{-2} \text{ h}^{-1}$) among the five 3DGs. And the photocatalytic activity of the H_2 evolution followed the order of 3DG-3 > 3DG-2 > 3DG-1 > 3DG-4 > 3DG-5 (Table 1).

For the prepared GO-benzidine samples, 3DG-3 exhibited the best photocatalytic activity, suggesting the best oxygen content loaded with C/O ratio of 1.84 that is consistent with the previous observation.¹⁶ 3DG-3 exhibited an H_2 evolution rate of $690 \mu\text{mol g}^{-1} \text{ h}^{-1}$, which was remarkably higher than that of GO ($45 \mu\text{mol g}^{-1} \text{ h}^{-1}$) and 3DGA ($125 \mu\text{mol g}^{-1} \text{ h}^{-1}$) that was hydrothermally synthesized under the same experimental conditions (Fig. 6a). Instead, the most reduced sample (3DG-5), which had the lowest energy band (3.50 eV) and absorbed the most light, exhibited the poorest photocatalytic performance. The excessive reduction of the oxygen-containing groups on the GO surface led to a small exposed area on 3DG-5, thus inhibiting the access of the hydrophilic water molecules, which did not favor the evolution of the photocatalytic H_2 production. Nevertheless, the least reduced sample (3DG-1) also did not show the best photoactivity. Although 3DG-1 had the largest energy band gap (3.65 eV) among the prepared 3DGs, it only absorbed high-energy light in the short wavelength range. In addition, because of the high electronegativity of the oxygen groups, it would have a negative effect when used as recombination center.⁴⁸ Therefore, higher oxygen coverage on the GO sheets also inhibited H_2 production. It was demonstrated that the covalent cross-linking method could effectively regulate the structure of the 3DG catalyst, thus achieving the optimum photocatalytic activity for 3DG-3. The high efficient H_2 evolution rate as observed with 3DG-3 is comparable to those excellent metal-free photocatalysts reported (Table S1†).

It has been reported that the specific surface area is an important factor affecting the catalytic activity.⁴⁵ But there are also some other factors, such as the oxidation degree of graphene,¹⁶ the distribution of active sites⁴⁶ and so on. For the photocatalytic H_2 evolution catalysed by 3DGs, we found that the reduction degree of 3DGs, the proper C/O ratio, rather than its surface area plays a major role in H_2 production.

The catalytic stability of 3DG-3 was additionally evaluated by measuring repeated cycles of photocatalytic H_2 production under the same reaction conditions. After four cycles of continuous exposure to full-spectrum light, no significant decline in photoactivity was observed over time (Fig. 6b).

Plausible mechanism of the photocatalytic H_2 production

The significant enhancement of the photocatalytic H_2 production by 3DG-3 compared to the 2D GO was due to the greater absorption of UV-vis light by the 3DGs,⁴⁹ while 3DG-3 had a more efficient separation of the photoinduced charge carriers under UV light excitation based on PL spectroscopic and

Table 1 Structure parameters and H₂ evolution activities of the five 3DG samples

Sample	BET (m ² g ⁻¹)	I _D /I _G	Band gap (eV)	C/O (%)	H ₂ evolved (μmol g ⁻¹ h ⁻¹)	H ₂ evolved (μmol m ⁻² h ⁻¹)
3DG-1	92.67	0.82	3.65	1.80	360	3.88
3DG-2	120.42	0.84	3.65	1.80	480	3.98
3DG-3	154.24	0.90	3.60	1.84	690	4.49
3DG-4	120.34	0.92	3.57	1.92	400	3.32
3DG-5	135.08	0.94	3.50	2.10	440	3.25

photochemical measurements. The PL emission spectrum of 3DG-3 revealed a very weak fluorescence, while GO showed a strong intrinsic fluorescence emission peak at 500 nm (Fig. 6c), indicating that the barrier of charge recombination in 3DG-3 was larger than that in GO.⁵⁰ Furthermore, 3DG-3 showed a much smaller arch radius than GO in the EIS spectrum (Fig. 6d), suggesting that its interface charge transfer rate was faster.⁵¹ Moreover, compared to GO, the photocurrent density of the 3DG-3 electrode was significantly increased, indicating its more efficient photogenerated charge carrier separation,⁵² which may be due to the unique porous structure. The 3D interconnected network of the porous nanosheets promoted the mass transfer and shortened the diffusion length of the charge transfer, which in turn promoted the electron repositioning at the edge of the surface, thereby hindering the charge recombination.⁴⁵

Although it has been reported that the GO band gap changes with the oxygen concentration, studies have shown that GO with different degrees of oxidation have similar conduction band edge levels of -0.75 eV (*vs.* Ag/AgCl).¹⁵ As shown in Fig. 2b, the energy gap (E_g) of 3DG-3 was 3.60 eV for an indirect transition. Based on the above results and literature reports, we could infer the valence band position of 3DG-3 (3.60 eV) and propose

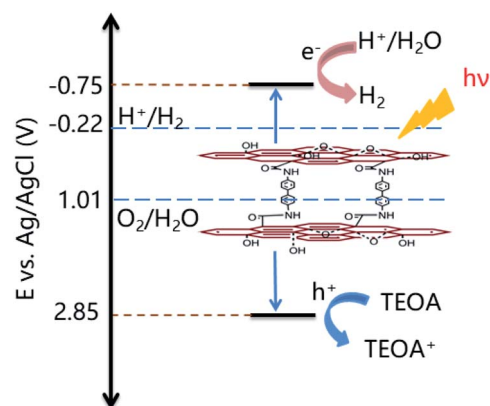


Fig. 7 Plausible mechanism for the photocatalytic H₂ production catalyzed by the 3DG-3 porous material.

a plausible mechanistic diagram of the photocatalytic reaction (Fig. 7). In particular, under irradiation, the GO vacancies are excited. Then, water molecules are reduced to H₂ by the photoinduced electrons (e^-), while the holes (h^+) are captured by the sacrificial agent (TEOA) on the 3DG surface.

Conclusions

In summary, we presented a facile method to prepare covalently linked 3DG network composites using aromatic diamines as the cross-linker, while the prepared 3DG were applied as metal-free photocatalyst for the efficient evolution of H₂. It was proven that the photocatalytic performance of the 3DGs was highly dependent on the linker and the reduction degree of GO. The benzidine-linked 3DGs exhibited much higher H₂ evolution rate than the DMPDA and MDA-linked 3DG catalysts. Furthermore, tuning the reduction degrees of GO in the 3DG catalyst was achieved by varying the amount of the introduced benzidine. Thus, the optimized 3DG-3 catalyst with an intermediate C/O ratio of 1.84 exhibited the highest H₂ evolution rate (690 μmol g⁻¹ h⁻¹) among the prepared 3DGs, which was about 15.3 higher than that of the 2D GO (*ca.* 45 μmol g⁻¹ h⁻¹) and 5.4 times higher than that of the noncovalent 3DG (128 μmol g⁻¹ h⁻¹) synthesized by a conventional hydrothermal method under similar experimental conditions. This work thus provided a new strategy for improving the photocatalytic activity of 3DG using the benzidine linker both as a pillar and a reducing agent to covalently assemble GO layers in the corresponding 3DG networks.

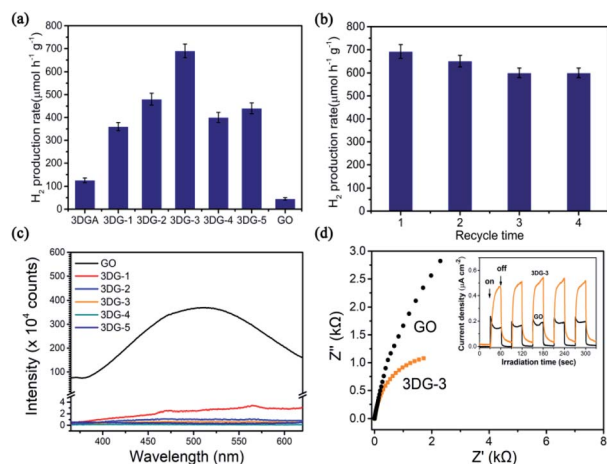


Fig. 6 (a) Photocatalytic H₂ evolution by water splitting over a series of 3DG samples synthesized at different conditions, GO, and 3DGA. (b) Cycling measurement of H₂ evolution over 3DG-3. (c) Fluorescence spectra ($\lambda_{\text{ex}} = 325$ nm) of GO and 3DG-1, 3DG-2, 3DG-3, 3DG-4 and 3DG-5. (d) Electrochemical research-impedance (EIS) plot and photocurrent-time dependence of GO and 3DG-3.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

- 1 X. Li, J. Yu, J. Low, Y. Fang, J. Xiao and X. Chen, *J. Mater. Chem. A*, 2015, **3**, 2485–2534.
- 2 X. Chen, S. Shen, L. Guo and S. S. Mao, *Chem. Rev.*, 2010, **110**, 6503–6570.
- 3 Q. Lu, Y. Yu, Q. Ma, B. Chen and H. Zhang, *Adv. Mater.*, 2016, **28**, 1917–1933.
- 4 M. Murdoch, G. I. N. Waterhouse, M. A. Nadeem, J. B. Metson, M. A. Keane, R. F. Howe, J. Llorca and H. Idriss, *Nat. Chem.*, 2011, **3**, 489–492.
- 5 X. Zhou, Z. Luo, P. Tao, B. Jin, Z. Wu and Y. Huang, *Mater. Chem. Phys.*, 2014, **143**, 1462–1468.
- 6 J. Zhang, J. Yu, Y. Zhang, Q. Li and J. R. Gong, *Nano Lett.*, 2011, **11**, 4774–4779.
- 7 J. Suntivich, K. J. May, H. A. Gasteiger, J. B. Goodenough and Y. Shao Horn, *Science*, 2011, **334**, 1383–1385.
- 8 X. Wang, K. Maeda, A. Thomas, K. Takanabe, G. Xin, J. M. Carlsson, K. Domen and M. Antonietti, *Nat. Mater.*, 2009, **8**, 76–80.
- 9 J. Liu, Y. Liu, N. Liu, Y. Han, X. Zhang, H. Huang, Y. Lifshitz, S. Lee, J. Zhong and Z. Kang, *Science*, 2015, **347**, 970–974.
- 10 W. J. Ong, L. L. Tan, Y. H. Ng, S. T. Yong and S. P. Chai, *Chem. Rev.*, 2016, **116**, 7159–7329.
- 11 W. Cui, Q. Liu, N. Cheng, A. M. Asiri and X. Sun, *Chem. Commun.*, 2014, **50**, 9340–9342.
- 12 K. Xie, H. Wu, Y. Meng, K. Lu, Z. Wei and Z. Zhang, *J. Mater. Chem. A*, 2015, **3**, 78–82.
- 13 T. F. Yeh, J. M. Syu, C. Cheng, T. H. Chang and H. Teng, *Adv. Funct. Mater.*, 2010, **20**, 2255–2262.
- 14 Y. Wu, N. Yi, L. Huang, T. Zhang, S. Fang, H. Chang, N. Li, J. Oh, J. A. Lee, M. Kozlov, A. C. Chipara, H. Terrones, P. Xiao, G. Long, Y. Huang, F. Zhang, L. Zhang, X. Lepro, C. Haines, M. D. Lima, N. P. Lopez, L. P. Rajukumar, A. L. Elias, S. Feng, S. J. Kim, N. T. Narayanan, P. M. Ajayan, M. Terrones, A. Aliev, P. Chu, Z. Zhang, R. H. Baughman and Y. Chen, *Nat. Commun.*, 2015, **6**, 6141–6150.
- 15 T. F. Yeh, F. F. Chan, C. T. Hsieh and H. Teng, *J. Phys. Chem. C*, 2011, **115**, 22587–22597.
- 16 L. K. Putri, B. J. Ng, K. H. Tan, F. S. Lim, W. J. Ong, W. S. Chang and S. P. Chai, *Catal. Today*, 2018, **315**, 93–102.
- 17 Q. Li, B. Guo, J. Yu, J. Ran, B. Zhang, H. Yan and J. R. Gong, *J. Am. Chem. Soc.*, 2011, **133**, 10878–10884.
- 18 I. V. Lightcap, T. H. Kosel and P. V. Kamat, *Nano Lett.*, 2010, **10**, 577–583.
- 19 Q. Xiang, J. Yu and M. Jaroniec, *J. Am. Chem. Soc.*, 2012, **134**, 6575–6578.
- 20 K. Iwashina, A. Iwase, Y. H. Ng, R. Amal and A. Kudo, *J. Am. Chem. Soc.*, 2015, **137**, 604–607.
- 21 Y. Zhang, N. Zhang, Z. R. Tang and Y. J. Xu, *J. Phys. Chem. C*, 2014, **118**, 5299–5308.
- 22 W. Wan, Y. Lin, A. Prakash and Y. Zhou, *J. Mater. Chem. A*, 2016, **4**, 18687–18705.
- 23 Y. Zhu, S. Murali, W. Cai, X. Li, J. W. Suk, J. R. Potts and R. S. Ruoff, *Adv. Mater.*, 2010, **22**, 3906–3924.
- 24 B. Wang, L. Si, J. Geng, Y. Su, Y. Li, X. Yan and L. Chen, *Appl. Catal., B*, 2017, **204**, 316–323.
- 25 R. Zhang, W. Wan, D. Li, F. Dong and Y. Zhou, *Chin. J. Catal.*, 2017, **38**, 313–320.
- 26 M. Gao, C. K. N. Peh, W. L. Ong and G. W. Ho, *RSC Adv.*, 2013, **3**, 13169–13177.
- 27 R. Wang, K. Q. Lu, F. Zhang, Z. R. Tang and Y. J. Xu, *Appl. Catal., B*, 2018, **233**, 11–18.
- 28 Y. Lu, B. Ma, Y. Yang, E. Huang, Z. Ge, T. Zhang, S. Zhang, L. Li, N. Guan, Y. Ma and Y. Chen, *Nano Res.*, 2017, **10**, 1662–1672.
- 29 D. C. Marcano, D. V. Kosynkin, J. M. Berlin, A. Sinitskii, Z. Sun, A. Slesarev, L. B. Alemany, W. Lu and J. M. Tour, *ACS Nano*, 2010, **4**, 4806–4814.
- 30 B. J. Hong, O. C. Compton, Z. An, I. Eryazici and S. T. Nguyen, *ACS Nano*, 2012, **6**, 63–73.
- 31 Y. Xu, K. Sheng, C. Li and G. Shi, *ACS Nano*, 2010, **4**, 4324–4330.
- 32 Q. Y. Cheng, D. Zhou, Y. Gao, Q. Chen, Z. Zhang and B. H. Han, *Langmuir*, 2012, **28**, 3005–3010.
- 33 L. Li, B. Song, L. Maurer, Z. Lin, G. Lian, C. C. Tuan, K. S. Moon and C. P. Wong, *Nano Energy*, 2016, **21**, 276–294.
- 34 J. I. Paredes, S. Villar Rodil, A. Martinez Alonso and J. M. D. Tascon, *Langmuir*, 2008, **24**, 10560–10564.
- 35 X. Zhao, Q. Zhang, D. Chen and P. Lu, *Macromolecules*, 2010, **43**, 2357–2363.
- 36 M. A. Khan, S. Mutahir, F. Wang, W. Lei, M. Xia and S. Zhu, *J. Hazard. Mater.*, 2019, **367**, 293–303.
- 37 M. A. Khan, S. Mutahir, F. Wang, J. W. Zhou, W. Lei and M. Xia, *Sol. Energy*, 2019, **186**, 204–214.
- 38 J. Ito, J. Nakamura and A. Natori, *J. Appl. Phys.*, 2008, **103**, 113712.
- 39 H. Huang, Z. Li, J. She and W. Wang, *J. Appl. Phys.*, 2012, **111**, 054317.
- 40 H. Yang, C. Shan, F. Li, D. Han, Q. Zhang and L. Niu, *Chem. Commun.*, 2009, 3880–3882.
- 41 W. Li, X. Z. Tang, H. B. Zhang, Z. G. Jiang, Z. Z. Yu, X. S. Du and Y. W. Mai, *Carbon*, 2011, **49**, 4724–4730.
- 42 Z. Wang, X. Zhou, J. Zhang, F. Boey and H. Zhang, *J. Phys. Chem. C*, 2009, **113**, 14071–14075.
- 43 W. S. Hung, C. H. Tsou, M. De Guzman, Q. F. An, Y. L. Liu, Y. M. Zhang, C. C. Hu, K. R. Lee and J. Y. Lai, *Chem. Mater.*, 2014, **26**, 2983–2990.
- 44 B. Song, J. I. Choi, Y. Zhu, Z. Geng, L. Zhang, Z. Lin, C. C. Tuan, K. S. Moon and C. P. Wong, *Chem. Mater.*, 2016, **28**, 9110–9121.

- 45 Q. Liang, Z. Li, X. Yu, Z. H. Huang, F. Kang and Q. H. Yang, *Adv. Mater.*, 2015, **27**, 4634–4639.
- 46 W. D. Wei, X. Y. Liu, S. C. Cui and J. G. Liu, *RSC Adv.*, 2017, **7**, 25650–25656.
- 47 J. R. Quinn, F. W. Foss Jr, L. Venkataraman and R. Breslow, *J. Am. Chem. Soc.*, 2007, **129**, 12376–12377.
- 48 K. Bothe and J. Schmidt, *J. Appl. Phys.*, 2006, **99**, 013701–013711.
- 49 Y. S. Jun, E. Z. Lee, X. Wang, W. H. Hong, G. D. Stucky and A. Thomas, *Adv. Funct. Mater.*, 2013, **23**, 3661–3667.
- 50 A. Du, S. Sanvito, Z. Li, D. Wang, Y. Jiao, T. Liao, Q. Sun, Y. H. Ng, Z. Zhu, R. Amal and S. C. Smith, *J. Am. Chem. Soc.*, 2012, **134**, 4393–4397.
- 51 G. Zhang, M. Zhang, X. Ye, X. Qiu, S. Lin and X. Wang, *Adv. Mater.*, 2014, **26**, 805–809.
- 52 Y. Hou, F. Zuo, A. P. Dagg, J. Liu and P. Feng, *Adv. Mater.*, 2014, **26**, 5043–5049.