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Enhanced photocatalytic degradation of Acid Blue dye using CdS/TiO₂ nanocomposite

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Photocatalytic degradation is essential for the successful removal of organic contaminants from wastewater, which is important for ecological and environmental safety. The advanced oxidation process of photocatalysis has become a hot topic in recent years for the remediation of water. Cadmium sulphide (CdS) nanostructures doped with Titanium oxide (CdS/TiO₂) nanocomposites has manufactured under ambient conditions using a simple and modified Chemical Precipitation technique. The nanocomposites crystal structure, thermal stability, recombination of photo-generated charge carriers, bandgap, surface morphology, particle size, molar ratio, and charge transfer properties are determined. The production of nanocomposites (CdS-TiO₂) and their efficient photocatalytic capabilities are observed. The goal of the experiment is to improve the photocatalytic efficiency of TiO₂ in the visible region by doping CdS nanocomposites. The results showed that as-prepared CdS-TiO₂ nanocomposites has exhibited the highest photocatalytic activity in the process of photocatalytic degradation of AB-29 dye, and its degradation efficiency is 84%. After 1 h 30 min of visible light irradiation, while CdS and TiO₂ showed only 68% and 09%, respectively. The observed decolorization rate of AB-29 is also higher in the case of CdS-TiO₂ photocatalyst $\sim 5.8 \times 10^{-4} \text{ mol L}^{-1} \text{ min}^{-1}$ as compared to the reported decolorization rate of CdS $\sim 4.5 \times 10^{-4} \text{ mol L}^{-1} \text{ min}^{-1}$ and TiO₂ $\sim 0.67 \times 10^{-4} \text{ mol L}^{-1} \text{ min}^{-1}$. This increased photocatalytic effectiveness of CdS-TiO₂ has been accomplished by reduced charge carrier recombination as a result of improved charge separation and extension of TiO₂ in response to visible light.

Wastewater recycling by photocatalysts could be a solution to many countries' long-standing freshwater shortages¹. Water pollution-related environmental issues have been a serious issue in recent decades, negatively affecting human health and the ecology. Access to safe drinking water has risen to the top of the priority list for long-term economic growth and societal well-being. Water is necessary for life and a valuable resource for civilization; it also plays an important role in natural ecosystems and climate regulation. Water stress is largely a problem of scarcity of water, but it can also be caused by deterioration of water quality and a lack of proper water management². Wastewater from the pulp, leather, fabric, polymer, pesticide and pharmaceutical sectors comprises a variety of organic contaminants, including dyes, that contaminate water and harmful life on the planet³⁻⁵. Separating and degrading these organic waste products has taken precedence in ensuring a nontoxic and pollution-free environment⁶. Photocatalysis an advanced oxidation process (AOP) is currently being utilized to remove colours from water without producing any hazardous by-products⁶. Fujishima and Honda⁷ established the theory of photocatalysis, which focuses on the generation of highly reactive intermediates, such as hydroxyl radicals ($\cdot\text{OH}$), which rapidly oxidize a wide range of pollutants. The photocatalyst's catalyst activity is highly dependent on the light sources; ultraviolet (UV) radiations have also been used⁸⁻¹⁰. A photocatalyst is a compound that facilitates the acceleration and enhancement of a light-induced reaction without being consumed. A photocatalyst turns solar energy into chemical energy, which can be utilized for pollution and sustainable applications such as water purification, environmental remediation, self-cleaning surfaces, hydrogen synthesis via water cleavage, and CO₂ conversion to hydrocarbon fuels. Solar energy is the purest and most abundant source of radiating energy that does not include any contaminants¹¹.

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In recent times, lot of interest in the development of novel photocatalytic technologies for a wide range of environmental applications, such as water remediation, oil spill and another pollution cleaning. Some energy-related research has focused on the utilization of photocatalysts for fuel production (e.g., H₂), specifically through water splitting¹². Solar radiation is used to convert pollutants from complicated molecules to simple and nontoxic compounds, avoiding the need for additional treatment, disposal, or the use of expensive oxidizing chemicals. Titanium oxide nanoparticle (TiO₂ NP) is widely used as a photocatalyst in the field of advanced oxidation technology due to its advantageous properties, such as exceptional optical and electronic properties, strong oxidative power, high photocatalytic efficiency, high stability, robust physical and chemical stability, low cost, low toxicity, stability in aqueous solutions, and eco-friendliness^{13,14}. It is a well-known potent photocatalyst for degrading a wide range of molecules¹⁵, including inorganic and organic compounds^{16,17}. Titanium oxide (TiO₂) photocatalytic activity is influenced not only by its bulk energy band structure, but also by the crystalline structure, textural characteristics, surface area, and particle size of the TiO₂ powder¹⁸. The photocatalytic activity of TiO₂ can be improved by changing the physicochemical nature of the nanoparticle surface to increase the adsorption of harmful compounds to be destroyed and reduce the likelihood of charge carrier recombination^{19,20}.

In the previous two decades, there has been substantial progress in the creation of effective photocatalytic materials, with a high number of research articles released each year. Advances in nanotechnology have substantially connected with improvements in the performance of photocatalytic materials. Titanium dioxide has been extensively explored for photocatalysis because it has numerous advantages such as strong photocatalytic activity, great physical and chemical stability, low cost, non-corrosive, non-toxicity, and high availability^{21,22}. When TiO₂ NP absorbs light at 385 nm, electrons (e⁻) are excited from the valence band to the conduction band, leaving a positively charged vacancy known as a hole (h⁺)²³. The hole is a strong potent oxidizing agent in and of itself, and in the presence of water and molecular oxygen, it can generate hydroxyl radicals or directly oxidize adsorbed molecules on the NP surface. TiO₂ has the advantage of being able to execute functions while using sunlight as a source of energy. Due to charge recombination, these photoinduced charges have a relatively brief lifetime, releasing the absorbed light energy as heat with no chemical consequence. These charges can also migrate to lower-energy trap sites, where they can still recombine or participate in redox processes with adsorbed species. As a result, preventing electron–hole recombination is critical for improving TiO₂ efficiency. Furthermore, TiO₂ can only absorb in the UV area of the solar spectrum, which accounts for just about 5% of total solar energy falling on the earth's surface²⁴, restricting its application. As a result, numerous attempts have been made to increase charge separation by altering the surface or bulk properties of TiO₂, such as doping²⁵, metal deposition²⁶, size reduction, and coupling of two semiconductors, consequently boosting photocatalytic activity^{27,28}. The photo-response of TiO₂ can be improved by combining it with a semiconductor that has a smaller bandgap and a higher conduction band than TiO₂. To begin with, a narrow bandgap will boost solar energy absorption efficiency by pushing it into the visible range. Second, photogenerated electrons in the narrow bandgap SC's conduction band (CB) will be injected into TiO₂'s CB, reducing charge recombination and keeping oxidation and reduction processes in different reaction sites²⁹. TiO₂ has a bandgap of 3.2 eV for the anatase phase and 3.0 eV for the rutile phase, respectively. As a result, under visible light irradiation, TiO₂ is nearly inert, making it unable to use solar energy sustainably³⁰.

On the other hand, TiO₂ has a number of disadvantages that limit its use in photocatalysis. To begin with, photogenerated electrons and holes coexist in the titania particle, with a high likelihood of recombination. As a result, the desired chemical changes occur at modest rates in relation to the absorbed light energy. Because of the very large bandgap energy (3.2 eV), photoactivation requires ultraviolet light, resulting in very low efficiency in utilizing solar radiation. When compared to visible light, UV light makes up just around 5% of the solar spectrum (45% only)^{31–33}. Furthermore, due to its non-porous nature and polar surface, titania has a poor absorption capacity for non-polar organic contaminants³⁴. The recovery of nano-sized titania particles from treated water is also a challenge, both economically and in terms of safety³⁵. Agglomeration and aggregation alter the photo-activity and light absorption of TiO₂ nanoparticles^{36–38}. To alleviate these limitations, several techniques have been proposed in the open literature. These strategies aim to increase the utilization of solar energy by extending the wavelength of photoactivation of TiO₂ into the visible region of the spectrum; preventing electron/hole pair recombination and thus allowing more charge carriers to diffuse to the surface; increasing the adsorption affinity of TiO₂ towards organic pollutants, and preventing aggregation and agglomeration of nano-titania particles while easing their recovery. Many efforts have been made to improve TiO₂ NPs immobilised on TiO₂ nanofibers (TNF), making them a promising platform for photocatalytic wastewater treatment and other applications such as sensing, photovoltaics, and photocatalytic water splitting^{39–41}. Furthermore, recent advances in the design and engineering of metal oxide–graphene–noble metal-based high-performance photocatalyst systems are developed, with an emphasis on the associated mechanisms and their applications in various photocatalytic processes^{42,43}. Several reviews on the development of ways to overcome the limits of titania photocatalysis have been published in recent years^{44–47}. Ion doping, for example, anatase–rutile phase coexisting⁴⁸, ion doping⁴⁹, or p-n heterojunction formation⁵⁰. Many elements were doped into TiO₂, such as Ag⁵¹, Fe⁵², N^{53,54}, S⁵⁵, C⁵⁶ and B⁵⁷, etc. Doping ions can introduce new energy levels into TiO₂, lowering the bandgap energy and increasing photocatalytic activity, especially when exposed to visible light. Consequently, the development of charge trapping sites by foreign ions can effectively lower the rate of electron–hole recombination⁵⁸.

On the other hand, Cadmium sulphide (CdS) is an excellent photocatalyst in the visible region but has low quantum efficiency due to low stability in solution due to Cd²⁺ ion leaching. As a result, despite its relatively high photoactivity, significant efforts are being made to increase photocatalytic stability. Attempts to improve the photocatalytic efficiency of CdS have included changing the surface structure of CdS NPs by controlling morphology, doping transition metal ions into CdS, depositing CdS to Nafion membranes, graphene sheets, or carbon nanotubes to obtain a uniform, homogeneously distributed CdS QDs, and coupling CdS with another semiconductor. Combining different bandgap semiconductors to make solid solutions is an effective approach

to adjust the potential of conduction and valence bands by making successive composition changes⁵⁹. Due to the suitable band gap (2.4 eV) of CdS⁶⁰, lower CB than TiO₂, great optical property, and possible application of CdS in photo-electrochemistry, photocatalysis, and water splitting systems, coupling of TiO₂ with cadmium sulphite (CdS) has been widely researched⁶¹. Due to its low bandgap (2.4 eV), which enables its visible light response, CdS is the most important chalcogenides semiconductor as a hydrogen production catalyst^{62,63}. The restricted separation efficiency of photogenerated charge carriers can overcome either by employing CdS in the form of QDs due to a shorter transit path or by integrating CdS onto support materials, such as TiO₂^{64,65}.

CdS doped TiO₂ nanotube composites were previously synthesized by chemical bath deposition, and their light-harvesting performance was 2.9 times than that of pure TiO₂ nanotubes. Under UV light irradiation, the CdS doped TiO₂ nanotube composite had better photocatalytic activity and photodegradation efficiency than pure TiO₂ nanotube and the degradation efficiency of methyl orange was about 42 percent at a UV intensity of 32 W⁶⁶. Rao et al.⁶⁷ created CdS/TiO₂ core/shell nanorods with variable shell thickness to reduce charge carrier recombination and photo corrosion when exposed to UV-Vis light. Du et al. created the same type of composite, but with different morphology, by fabricating pyramid-like CdS nanoparticles and growing them on porous TiO₂. The H₂ generation rate of 5 mol% CdS-TiO₂ was 1048.7 mol h⁻¹ g⁻¹ under UV-Vis irradiation and without noble-metal co-catalysts, which is about six times and 1.5 times greater than pure TiO₂ and CdS, respectively⁶⁸.

In the present study, a series of TiO₂ NPs were synthesized in this study to investigate the effect of reactant concentration on size, shape, crystal structure, thermal, optical, and photocatalytic activities. According to prior research, a number of manufactured CdS NPs were used in the photocatalytic experiment, and it was discovered that the CdS NPs, has an excellent photo-response, but not suitable for photocatalytic water purification. TiO₂ is not photoactive under the visible region of the solar spectrum. Thus, in order to make CdS potentially applicable for water purification and utilization of TiO₂ in the visible region a nanocomposite of cadmium sulphide and titanium dioxide (CdS-TiO₂) was also synthesized and the effect of CdS on TiO₂ and vice versa was studied.

Materials and methods

Synthesis of titanium oxide nanoparticle (TiO₂ NP) by precipitation technique. Controlled precipitation of nanoparticles from precursors dissolved in a solution was used to make TiO₂ NP. The reductive hydrolysis of Titanium Tetra Isopropoxide (TTIP) in methanol at ambient temperature and pressure without calcinations was proposed for the manufacture of TiO₂ nanoparticles⁶⁹. By adjusting the concentration of TTIP while maintaining the amount of methanol (24.44 M) constant at 100 mL, a series of TiO₂ NP (Tma, Tmb, Tmc, Tmd, and Tme) was synthesized. Tma = 0.25 M, Tmb = 0.20 M, Tmc = 0.15 M, Tmd = 0.1 M, and Tme = 0.05 M have different TTIP concentrations. The reactions were carried out as follows: 100 mL methanol (24.44 M) was placed in a conical flask, and TTIP was added dropwise (20 drops per minute) while vigorous stirring continued for another 5 h. White precipitates observed were washed with water and acetone several times and then air-dried. The crystal structure of the produced TiO₂ was expected to be a mix of anatase, brookite, and rutile.

Chemical route for the synthesis of cadmium sulphite doped titanium oxide (CdS-TiO₂). In a reaction vessel, 100 mL aqueous Cd (NO₃)₂ (0.085 M) was added dropwise with continuous stirring, followed by 50 mL methanol (24.44 M). The reaction was then carried out for 1 min in the H₂S environment with vigorous stirring and then continued for another 2 h. The colour of the solution changed from clear to yellow. 3.53 mL TTIP (0.1 M) was added drop-by-drop to this solution (20 drops per minute). The stirring was extended for another 5 h. The hue of the solution had changed to a faint yellow.

Characterization techniques. The synthesized TiO₂ NP and CdS-TiO₂ NC were characterized by elemental, structural, optical and thermal techniques. Elemental analysis and chemical compositions were examined by energy-dispersive X-ray spectroscopy (EDS, JEOL, JSM6510LV) and Fourier Transform Infrared Spectroscopy. The structural properties were analysed by employing powder X-ray diffraction (Miniflex-TM II Benchtop, Rigaku Co-operation, Tokyo, Japan). Surface morphology and size was characterized by Scanning Electron Microscopy (JEOL, JSM6510LV) and Transmission Electron Microscopy (JEOL, JEM2100). Thermal properties were determined by Thermal Gravimetric Analysis (TGA). The optical properties were determined by employing UV-Visible Spectroscopy (Shimadzu UV-1601).

Photocatalytic experiment. The decolorization of a dye derivative Acid Blue-29 (AB-29) in the presence of UV light was used to investigate the photocatalytic activity of TiO₂ nanoparticles. The photocatalytic studies were carried out in an immersion well Pyrex glass photoreactor (inner and outer jacket) with a magnetic bar, water circulating jacket, and a molecular oxygen opening. A 125 W medium pressure mercury lamp was used to irradiate the area (Philips). The optimal catalyst dosage was established by irradiating the dye (AB-29) aqueous solution with various strengths of TiO₂ nanoparticles. 180 mL dye (AB-29) solution (0.06 mM) containing manufactured nanocatalysts (1 g L⁻¹) were magnetically swirled in the dark for at least 20 min in the presence of ambient oxygen to achieve dye (AB-29) and nanocatalyst surface adsorption-desorption equilibrium. After reaching equilibrium, the first aliquot (5 mL, 0 min) was removed and the irradiation process began. During the irradiation, other aliquots of 5 mL were taken at regular intervals and examined following centrifugation. Changes in absorption were used to track the decolorization of AB-29 using a UV-Visible spectroscopic analysis approach (Shimadzu UV-Vis 1601). The dye concentration was determined using a standard calibration curve based on the dye's absorbance at various known values. The photocatalytic activity of the CdS-TiO₂ nanocomposite was tested by examining the decolorization of AB-29 in the presence of visible light using a halogen linear lamp (500 W, 9500 Lumens) as a light source and comparable experimental conditions as described before. Photocatalytic studies were conducted for five cycles using the same batch of nanomaterial photocatalysts to

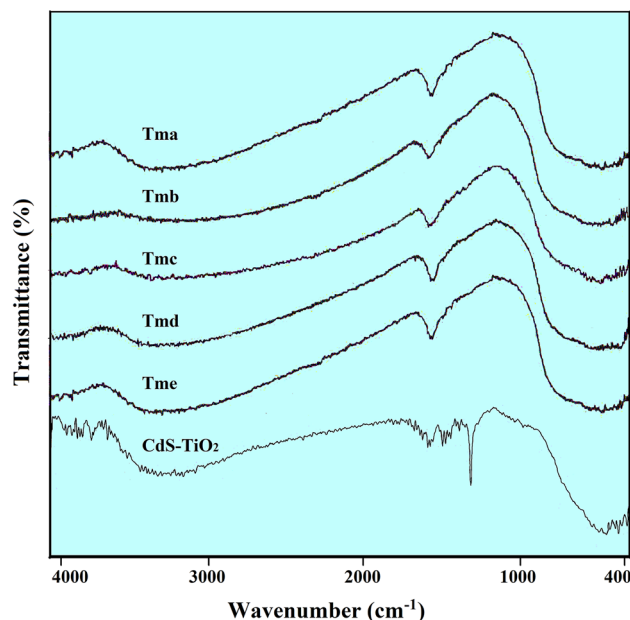


Figure 1. The FTIR Spectra of the synthesized TiO₂ nanoparticles (Tma, Tmb, Tmc, Tmd and Tme) and its nanocomposites with CdS (CdS-TiO₂).

Peak	Range	Intensity	Significance
A	(400–420)	Small and weak	Cd-S bond (CdS nanoparticles)
B	(400–800)	broadband	Ti-O bond vibration (TiO ₂ nanoparticles)
C	(570–620)	Small and weak	S-S bond (crystal S-S bond)
D	(1380–1420)	Sharp or Broad	C-H bending of CH ₃ (Acetone)
E	(1620–1740)	Small and weak	CO ₂ bending or C-H bending (Acetone)
F	(3140–3470)	Broad	Intermolecular H-bonds (Lattice water)

Table 1. Significance of peaks obtained in the FTIR spectra of the synthesized TiO₂ nanoparticles (Tma, Tmb, Tmc, Tmd and Tme) and CdS-TiO₂.

determine the reusability and recyclability of the nanomaterials as catalysts. Before each photocatalytic run, the nanocomposite catalyst was rinsed with double distilled water after each cycle and a fresh solution of AB-29 was added.

Results and discussion

Spectroscopy analysis. The purity and composition of the samples were validated by FTIR spectra as shown in Fig. 1, which revealed multiple peaks related to TiO₂ in all samples without any other elemental contamination. In the 800–400 cm⁻¹ area, bands for Ti-O and Ti-O-Ti bands were detected. The FTIR spectra of TiO₂ could be in the form of a broad band centred at 400–800 cm⁻¹ due to the Ti-O bond vibration in the TiO₂ lattice⁷⁰ or peaks centred at 760 cm⁻¹, 680 cm⁻¹, 600 cm⁻¹, 560 cm⁻¹, 500 cm⁻¹, 468 cm⁻¹, 410 cm⁻¹, 385 cm⁻¹ and 350 cm⁻¹ attributable to^{71,72}. The FTIR spectra of TiO₂ nanoparticles (Tma, Tmb, Tmc, Tmd, and Tme) and their nanocomposites with CdS are shown (CdS-TiO₂) in Fig. 1. The varied peaks generated by TiO₂ nanoparticles and their nanocomposites with CdS nanoparticles are explained in Table 1^{73,74}.

The FTIR spectra of TiO₂ NPs (Tma, Tmb, Tmc, Tmd, and Tme) and CdS-TiO₂ NC in this study were in the shape of a large peak in the 400–800 cm⁻¹ region with multiple tiny peaks. Stretching Vibrations of hydroxyl (OH) groups of water adsorbed by the samples were ascribed to the broad peak showing at 3100–3600 cm⁻¹. Such TiO₂-OH groups are formed as a result of the hydrolysis reaction in the process. The bending modes of -OH groups of water molecules adsorbed on the surface of the catalyst are responsible for the peak at 1628 cm⁻¹⁷⁵. CO₂ deposited on the surface of the particles showed a weak absorption band at 1620–1630 cm⁻¹. Adsorption of water and CO₂ was ubiquitous for all powder samples exposed to the atmosphere, and significantly more evident for nanosized particles with large surface areas, as well known⁷⁵. In the case of CdS-TiO₂, the existence of a broad band centred at 400–800 cm⁻¹ with numerous little peaks in it proved the creation of CdS, and the presence of broadband centred at 400–800 cm⁻¹ with several small peaks in it indicated the presence of TiO₂ in the nanocomposite. The creation of a sandwich structure with CdS at the centre (core) and TiO₂ NPs surrounding CdS as a shell result in CdS stretching at 605 cm⁻¹. The FTIR spectra of the reaction product, obtained

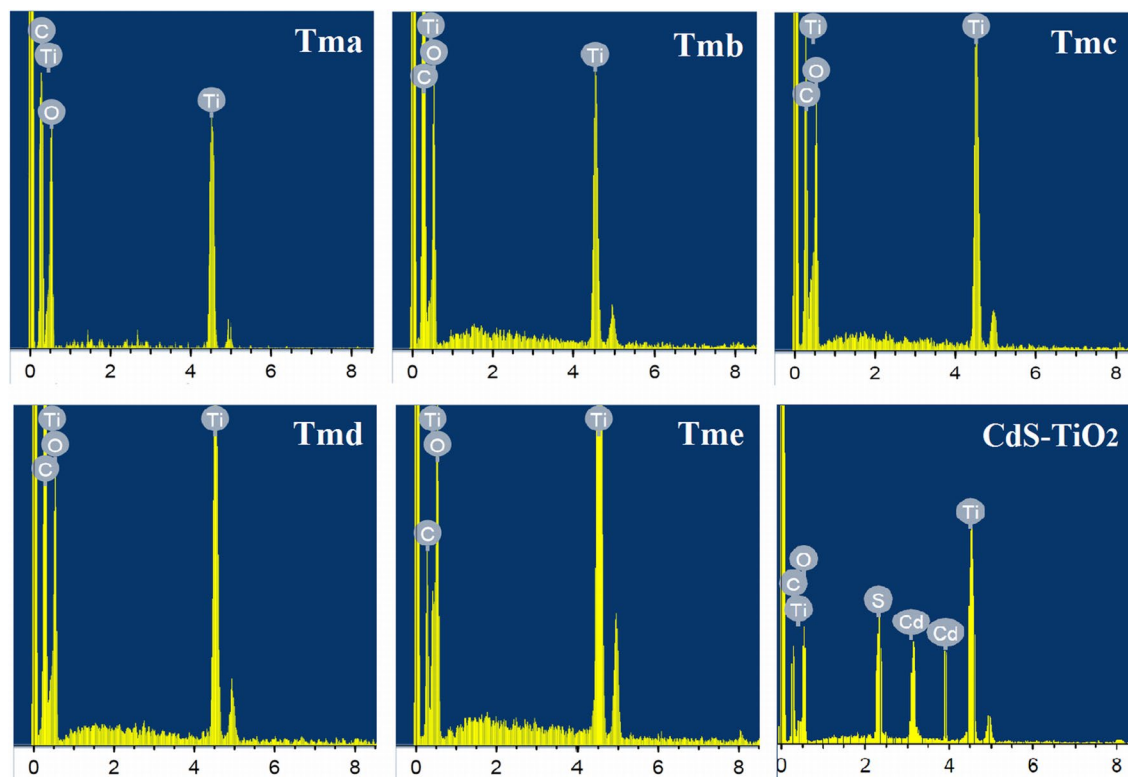


Figure 2. EDAX of the synthesized TiO₂ nanoparticles (Tma, Tmb, Tmc, Tmd and Tme) and its nanocomposites with CdS (CdS-TiO₂).

after heating the precursor at 300 °C for 2 h, reveal no absorption peak in the upper-frequency area but a sharp peak below 500 cm⁻¹, which could be attributable to the Ti–O–Ti bond stretching frequency modes in titanium dioxide⁷⁶. Individual bands at 675 cm⁻¹ may be related to the anatase phase, and are attributed to Ti–O bond stretching vibration⁷⁷. Stretching vibrations of functional groups CH, CH₂, and CH₃ are represented by the bands at 2963, 2907, and 2809 cm⁻¹. At 1600 cm⁻¹, the vibration corresponds to the C=C bond stretching vibration. The bands in the 1500–1400 cm⁻¹ range are caused by bending vibrations of CH, CH₂, and CH₃, while those in the 1300–1200 cm⁻¹ range are caused by vibrations of the methoxy group O–CH₃ and the alcohol group C–OH⁷⁸.

Energy dispersive X-ray spectroscopy (EDAX) analysis. Figure 2 shows the EDAX spectra of the prepared samples. For the samples Tma, Tmb, Tmc, Tmd, and Tme, the spectra revealed the existence of Ti and O peaks, confirming the synthesis of pure TiO₂ with no other elemental impurity. The other peaks in this graph, related to oxygen, carbon, and silicate, were caused by the sputter coating of the glass substrate on the EDS stage and not taken into account. The presence of peaks corresponding to Cd and S, as well as peaks for Ti and O, in the EDS spectra of CdS-TiO₂ confirmed the creation of the CdS-TiO₂ nanocomposite.

Structural analyses. Diffraction and microscopic examinations were used to conduct structural evaluations. X-Ray Diffraction (XRD) Spectroscopy was used to conduct diffraction research. Due to the physical nature of the materials, the diffraction patterns for TiO₂ NPs (Tma, Tmb, Tmc, Tmd, and Tme) were of poor quality. The XRD pattern of the produced TiO₂ NPs (Tma, Tmb, Tmc, Tmd, and Tme) confirmed the existence of anatase, brookite, and rutile mixes (Fig. 3). TiO₂ NPs peaks are corresponded to anatase at 2θ = 25.56° (101), 37.8° (103), 48.07° (200), 54.18° (105), 62.42° (204) and 75.2° (215), rutile at 2θ = 27.01° (110), 36.14° (101), 42.121° (111), 54.89° (211) and 68.72° (301)^{79,80} and brookite at 2θ = 30.9° (121)⁴⁹ thus confirming the presence of mixed crystal phase. Due to the presence of mixed peaks of anatase, rutile, and brookite TiO₂, estimating crystallite size for TiO₂ samples (Tma, Tmb, Tmc, Tmd, and Tme) based on the X-ray diffraction peak was not achievable. The XRD pattern of CdS-TiO₂ nanocomposites (Fig. 4) demonstrated the production of cubic CdS and anatase TiO₂ nanoparticles. The appearance of peaks at 2θ = 25.56 (101), 37.282 (103), 48.07 (200), 54.18 (105), and 62.42 (204), which corresponded only to anatase TiO₂, was linked to the synthesis of anatase TiO₂ in CdS-TiO₂^{80,81}. The appearance of peaks that corresponded only to cubic CdS at 2θ = 26.719 (111), 29.900 (200), 43.000 (220), and 51.061 (311)^{82,83} verified the production of cubic CdS. The existence of mixed peaks of cubic CdS and anatase TiO₂ made it impossible to estimate crystallite size for CdS-TiO₂ using the X-ray diffraction peak.

Scanning electron microscopy (SEM) analysis. SEM images of the synthesized TiO₂ NP are depicted in Fig. 5a–e. The creation of well-defined spherical mesoporous TiO₂ nanoclusters observed using SEM micro-

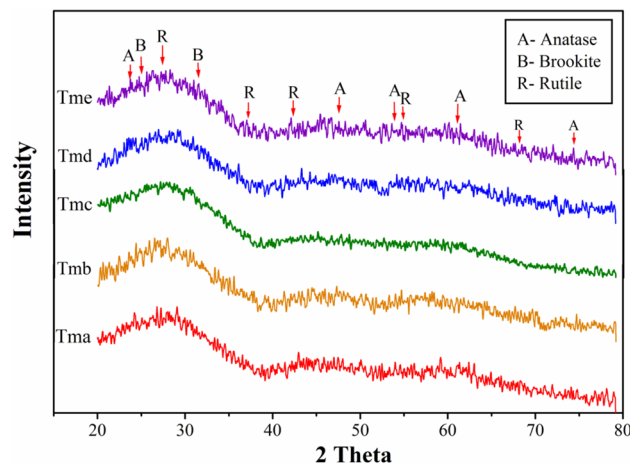


Figure 3. XRD patterns of synthesized TiO₂ nanoparticles (Tma, Tmb, Tmc, Tmd and Tme).

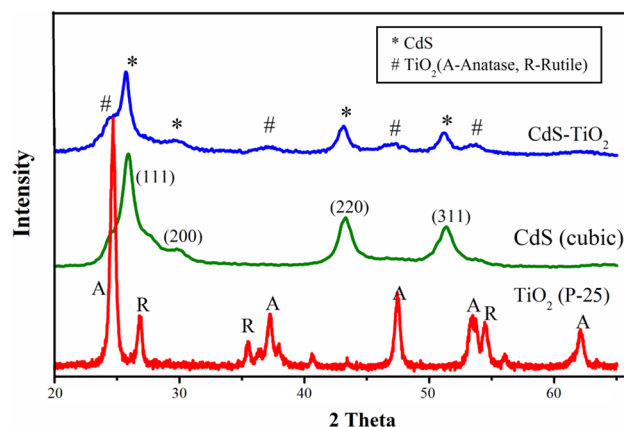


Figure 4. XRD pattern of CdS-TiO₂ nanocomposites in comparison to pure cubic CdS and Degussa P-25 TiO₂.

graphs was attributed to the high surface energy of nanosized TiO₂ particles⁸⁴. By agglomerating tiny particles, pure TiO₂ generates a layer-like structure. An amorphous mass with a very small particle structure was visible in the SEM picture of CdS-TiO₂. Due to the thin amorphous powder, no other distinct particle shape was discernible as CdS particles uniformly covered by the TiO₂ nanoparticles formed a sandwich-type structure, in which CdS is acting as a core surrounded by smaller TiO₂ NPs. SEM pictures of the synthesized CdS-TiO₂ NC are shown in Fig. 5f⁸⁵.

Transmission electron microscopy (TEM) analysis. Figure 6a–f shows a TEM micrograph of TiO₂ NP (Tma, Tmb, Tmc, Tmd, and Tme). The creation of spherical nanoclusters made up of very small TiO₂ NPs (1–2 nm) can be seen in the images. The nanocluster size has shrunk from > 500 nm (Tma) to 50 nm (Tme). In all cases, however, the individual TiO₂ nanoparticles were smaller than 2 nm.

The TEM picture of CdS-TiO₂ (Fig. 7) verifies the presence of two different-sized nanoparticles CdS and TiO₂ nanoparticles with close proximity to each other. TEM images of CdS-TiO₂ (Fig. 7a) show two different-sized nanoparticles with smaller ones surrounding the larger ones in a core-shell type fashion. Further, the presence of both CdS and TiO₂ were also confirmed by EDS and XRD test results individually. The observed two distinctly different sized particles with close proximity to each other is the direct prove of the sandwich-type model of nanoparticles with smaller particles (TiO₂) surrounding the larger particle (CdS) like a core. Figure 7b shows a TEM image of the CdS-TiO₂ composite. Table 2 shows the particle sizes acquired using TEM.

Thermal analyses. The Thermal Gravimetric Analysis (TGA) graphs were used to conduct thermal experiments. TGA findings of the synthesised TiO₂ nanoparticles (Tma, Tmb, Tmc, Tmd, and Tme) and CdS-TiO₂ nanocomposite are shown in Fig. 8. The TGA curve demonstrated great thermal stability, the lack of any impurity or intermediate complex, and a high melting point for the produced nanoparticles. The TiO₂ NPs (Tma,

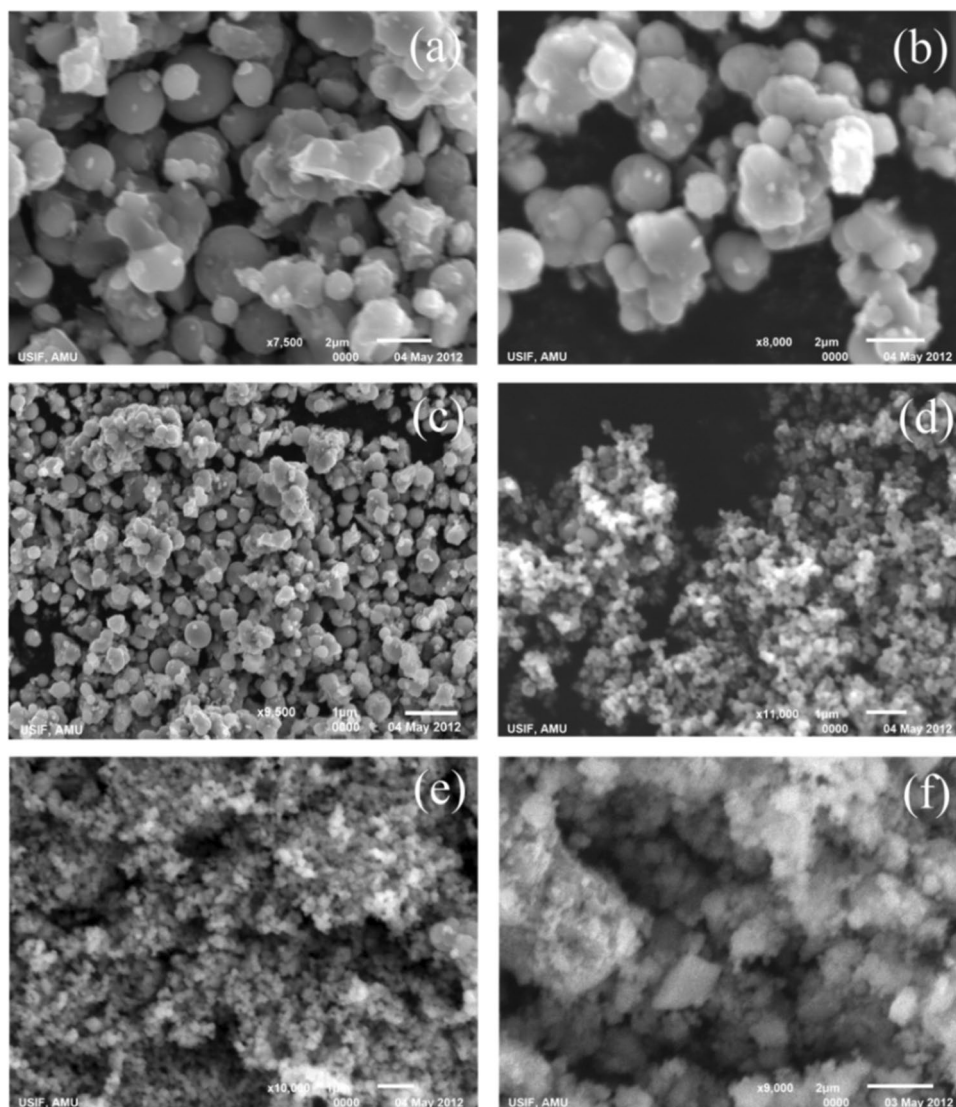


Figure 5. SEM images (a) Tma, (b) Tmb, (c) Tmc, (d) Tmd, (e) Tme and (f) CdS-TiO₂.

Tmb, Tmc, Tmd, and Tme) and CdS-TiO₂ NC were shown to be thermally stable up to 1000 °C, with a slight weight loss roughly at 100 °C, owing to the presence of moisture and other volatile solvents.

Optical analysis. The optical properties were determined using the UV-Visible absorption spectra. Figure 9a shows the absorption spectra of TiO₂ nanoparticles synthesized in the range of 350 nm to 600 nm range of the light spectrum. The absorption edge in bulk TiO₂ for anatase is 387 nm (3.2 eV), rutile is 410 nm (3.02 eV), and brookite is 381 nm (3.25 eV)^{86,87}. The absorption edges of produced TiO₂ nanoparticles (Tma, Tmb, Tmc, Tmd, and Tme) were found to be in the wavelength range 382–403 nm, implying the presence of mixed phases or a blue shift from the bulk rutile phase. This blue shift was detected with the decrease of TiO₂ NP particle sizes from Tma to Tme, and was in good agreement with previous findings. This change was linked to QSE or the presence of TiO₂ mixed-phase⁸⁸. XRD results given in the study indicated the presence of mixed-phase TiO₂. The values of the absorption edges of the different TiO₂ NPs are listed in Table 3.

Figure 9b shows the bandgap energy of TiO₂ NP. The bandgap curve was plotted $(\alpha h\nu)^{1/2}$ vs $h\nu$ based on Tauc relation revealed an indirect bandgap⁸⁹. As the size of TiO₂ NP reduced from Tma to Tme, a greater band gap was observed.

Table 3 lists the corresponding band gaps of the various TiO₂ NPs. The bandgap of TiO₂ can also be calculated using the Eq. (1):

$$E_g = 1240/\lambda_{\text{onset}} \quad (1)$$

where E_g denotes the bandgap energy, and onset denotes the absorption edge as determined by the absorption spectra⁹⁰. The results of the above formula were in good agreement with the Tauc relation curves (Table 3).

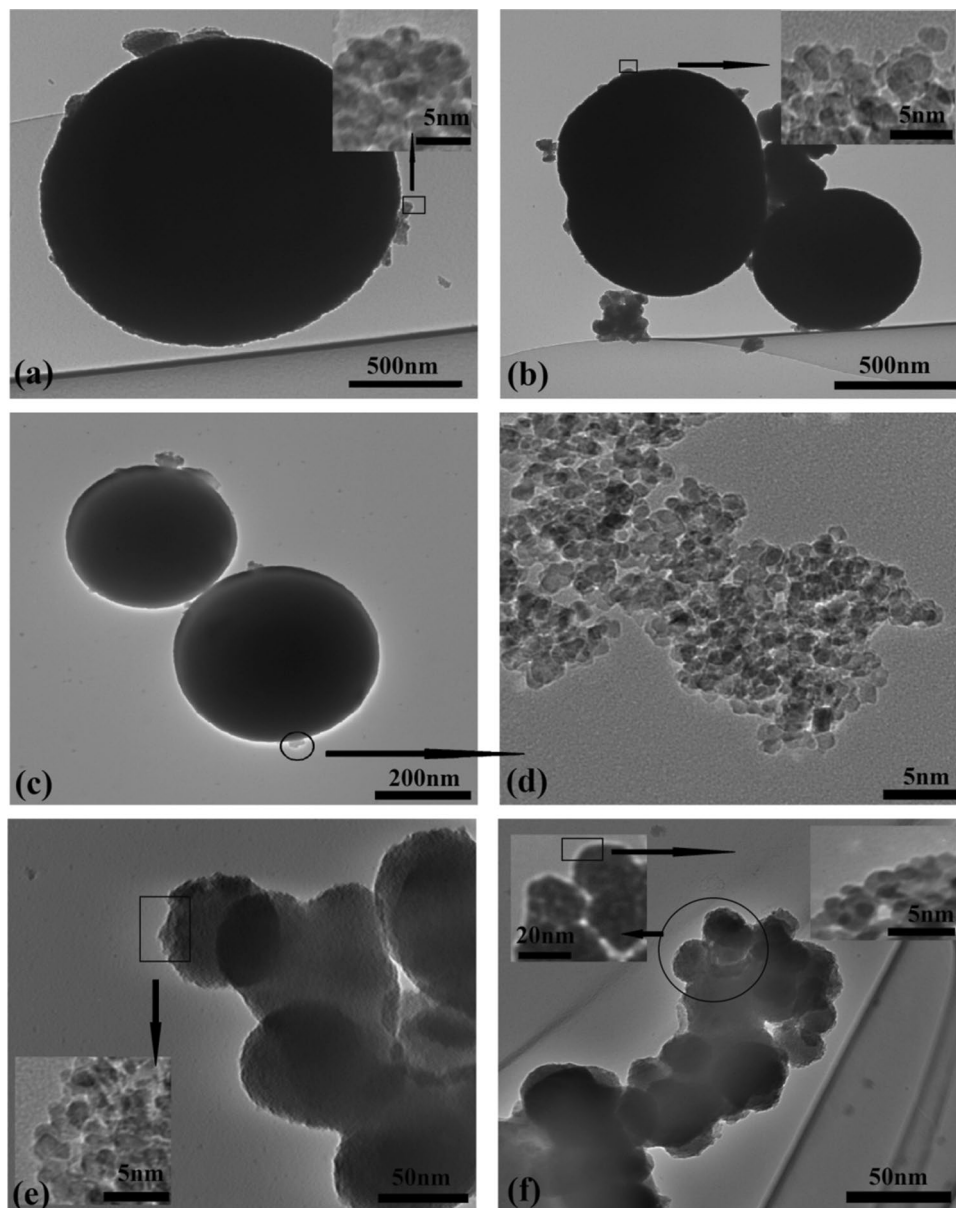


Figure 6. TEM images of (a) Tma, (b) Tmb, (c) Tmc, (d) magnified portion of Tmc, (e) Tmd, (f) Tme, and in the inset are the magnified portion of the corresponding images.

In Fig. 9c, the absorption spectra of CdS-TiO₂ are compared to those of bulk CdS and TiO₂ (anatase). Bulk CdS exhibited a sharp edge at 515 nm, while anatase TiO₂ exhibited one at 388 nm. The spectrum of CdS-TiO₂ NC displayed a mixture of these two spectra, with an absorption edge at 487 nm that was blue-shifted from that of pure CdS NP (515 nm) and at 388 nm corresponded to that of pure anatase TiO₂ NP. The synthesis of anatase TiO₂ in CdS-TiO₂ was shown to be good in accordance the XRD results. The creation of a solid solution at the interfaces as a result of close contact between CdS and TiO₂ caused this shift, and this behaviour determined the optical properties of the final nanostructure⁹¹. Which attributes to electronic semiconductor-support interaction (SEMSI) by several researchers^{92,93}. The UV area is used to excite (bulk) TiO₂ (bandgap = 3.2 eV), whereas the visible region is used to excite CdS (bandgap 2.42 eV). As a result of visible light absorption, electrons in CdS nanoparticles can be stimulated from the valence band to the conduction band, forming electron-hole pairs that are then trapped by the surface state⁹⁴. These electrons can transfer from the CdS to the TiO₂ conduction band. They can then migrate across the TiO₂ conduction band and contribute to the reduction of species like oxygen molecules or adsorbed pollutants⁹⁵. In Fig. 9d, the bandgap curve of CdS-TiO₂ is compared to bulk CdS and TiO₂ (anatase) using the Tauc relation. Bulk CdS had a bandgap of 2.42 eV, while bulk TiO₂ had a bandgap of 3.2 eV. The bandgap curve of CdS-TiO₂ exhibited a combination of these two, with a bandgap of 3.2 eV for anatase TiO₂ NP and a bandgap of 2.55 eV for CdS NP.

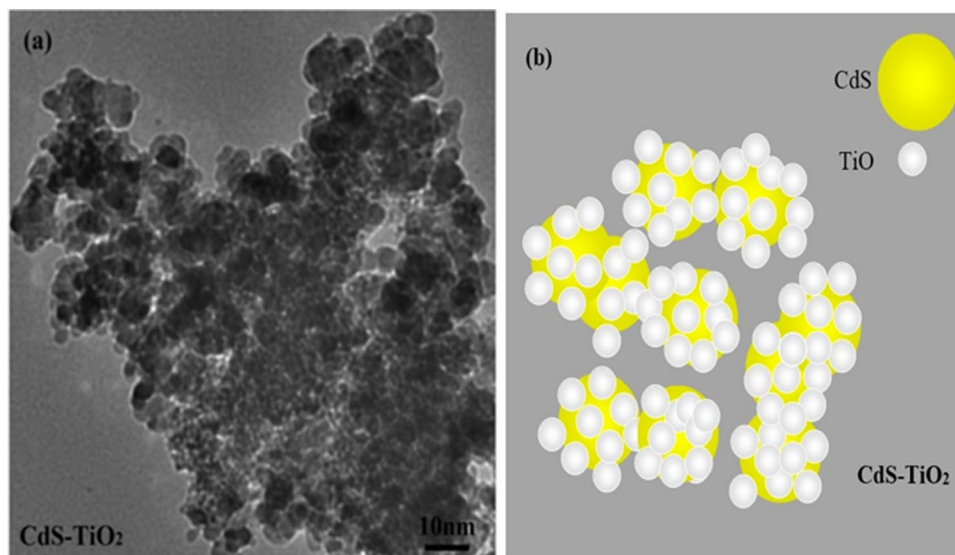


Figure 7. (a) The TEM image of the synthesized CdS-TiO₂ nanocomposite (b) A representative diagram of the synthesized CdS-TiO₂ nanocomposite.

Nanoparticles	Tma	Tmb	Tmc	Tmd	Tme	CdS-TiO ₂
Particle size (nm)	1.5–2.0	1.5–1.7	1.2–1.5	1–1.2	0.8–1.0	5–6 (CdS), (TiO ₂)

Table 2. Average particle sizes of the synthesized TiO₂ NP (Tma, Tmb, Tmc, Tmd and Tme) and CdS-TiO₂ NC obtained by TEM.

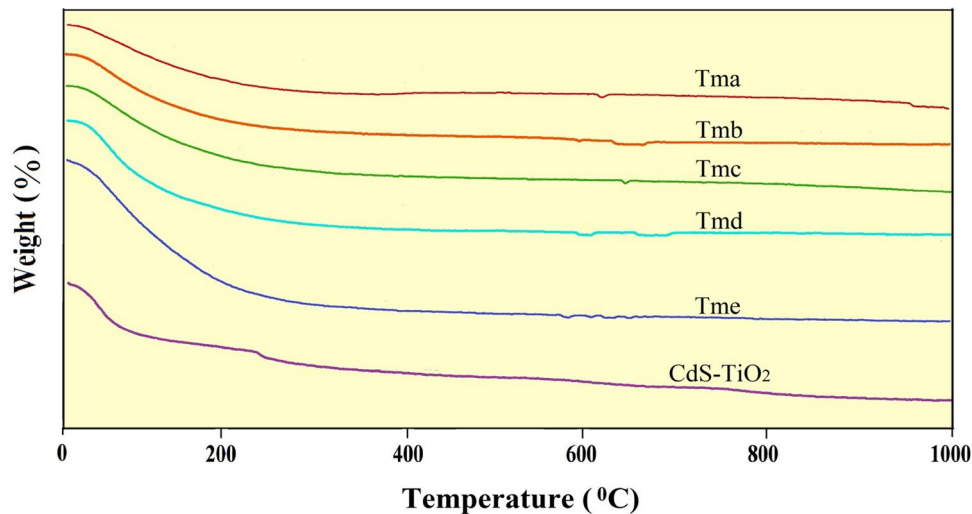


Figure 8. TGA graphs of synthesized TiO₂ nanoparticles (Tma, Tmb, Tmc, Tmd and Tme) and CdS-TiO₂ nanocomposite.

Photocatalytic activity. The effect of samples synthesized in five distinct modes as catalysts on the removal of dye AB-29 was investigated, and the results are shown in Fig. 10a a relative change in AB-29 concentration (C/C_0) as a function of time in the presence and absence of photocatalysts. The activity of all five samples was compared in order to get superior photocatalytic activities of TiO₂ nanoparticles. UV light was used to conduct the photo-degradation investigations. Blank experiments were also separately carried out in presence of the photocatalyst under dark conditions and absence of the photocatalyst under irradiation. In both the cases, analysis of the samples did not show any appreciable loss of the dye (AB-29). The percentage of relative decolorization (C/C_0) of the dye AB-29 was determined in the following order: Tma (71%) Tmb (74%) Tmc (79%) Tmd (83%)

Nanoparticle	Absorption edge (nm)	Band gap (eV) (from Tauc curve)	Band gap (eV) ($E_g = 1240/\lambda_{onset}$)	Particle size (nm)
Tma	403	3.07	3.074	0.522
Tmb	393	3.14	3.155	0.208
Tmc	390	3.17	3.179	0.188
Tmd	388	3.2	3.195	0.161
Tme	382	3.25	3.246	0.125
CdS-TiO ₂	487, 388	2.55, 3.2	2.546, 3.195	5.8, 0.161

Table 3. Absorption peaks and bandgap energy of the synthesized TiO₂ NP (Tma, Tmb, Tmc, Tmd and Tme) and CdS-TiO₂.

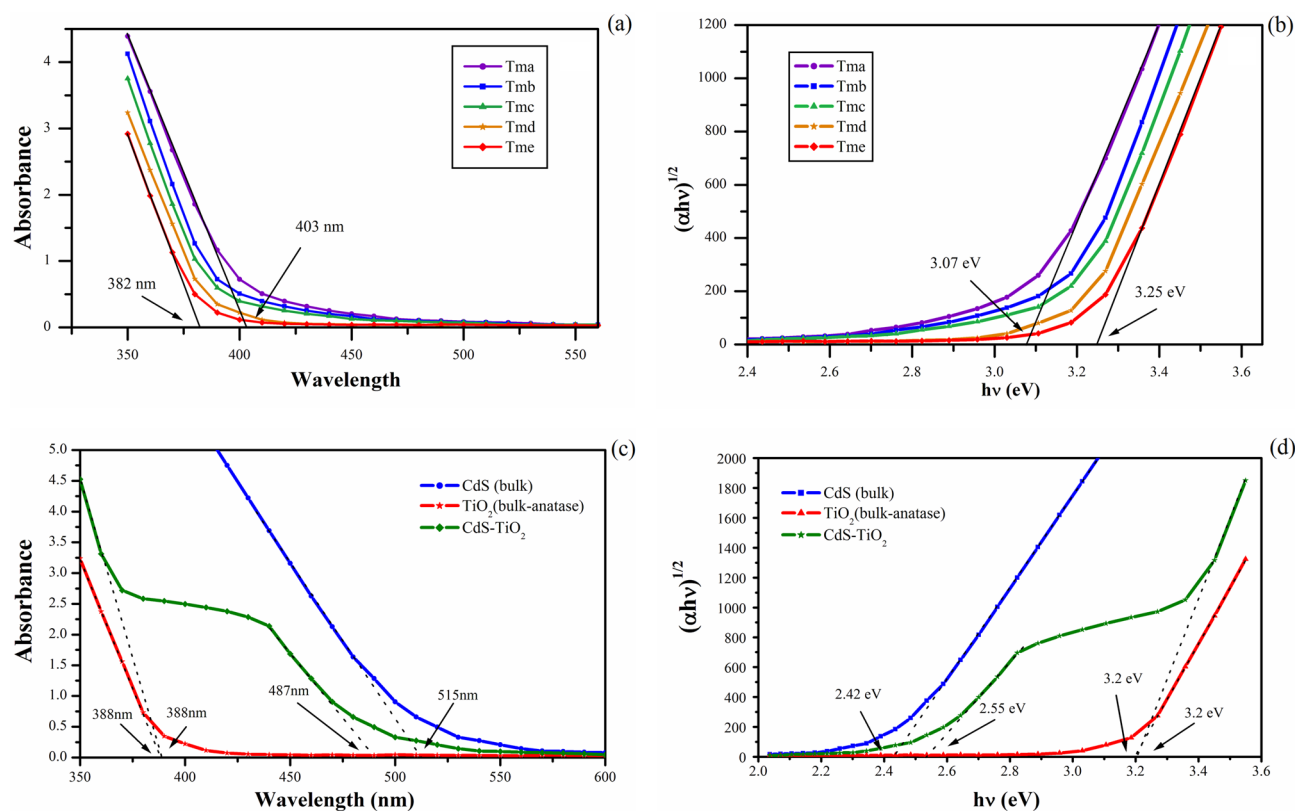


Figure 9. (a) The UV-Visible absorption spectrum of synthesized TiO₂ nanoparticles (Tma, Tmb, Tmc, Tmd and Tme) shows a slight blue shift in absorption edge. (b) The bandgap energy of the synthesized TiO₂ nanoparticles (Tma, Tmb, Tmc, Tmd and Tme). (c) The absorption spectra of CdS-TiO₂ with respect to pure CdS bulk and TiO₂ bulk. (d) Band gap energy curve of CdS-TiO₂ with respect to pure CdS and TiO₂ bulk.

Tme (86%). In the absence of photocatalyst, however, there was no discernible decrease in dye concentration. Multiple factors may have contributed to sample Tme's maximum photocatalytic activity.

Most dyes undergo photocatalytic decolorization according to the Langmuir-Hinshelwood kinetic model⁹⁶⁻⁹⁸, which can be summarised as (2):

$$-dC/dt = kKC/(1 + KC) \quad (2)$$

where k represents the reaction rate constant (mMmin^{-1}), K represents the reactant's adsorption coefficient (mM^{-1}), and C represents the reactant concentration (mM). When C is very low, KC is minimal in comparison to unity, allowing Eq. (2) to be simplified to apparent pseudo-first-order kinetics⁹⁹.

$$-dC/dt = kKC = k_{app}C \quad (3)$$

The decolorization curve (Fig. 10a), emerged as an exponential decay curve which represents pseudo-first-order kinetics reasonably well. The rate constant was obtained for each experiment by plotting the natural logarithm of dye concentration as a function of irradiation time¹⁰⁰. The following is a representation of the equation:

$$\ln(C_0/C) = k_{app}t \quad (4)$$

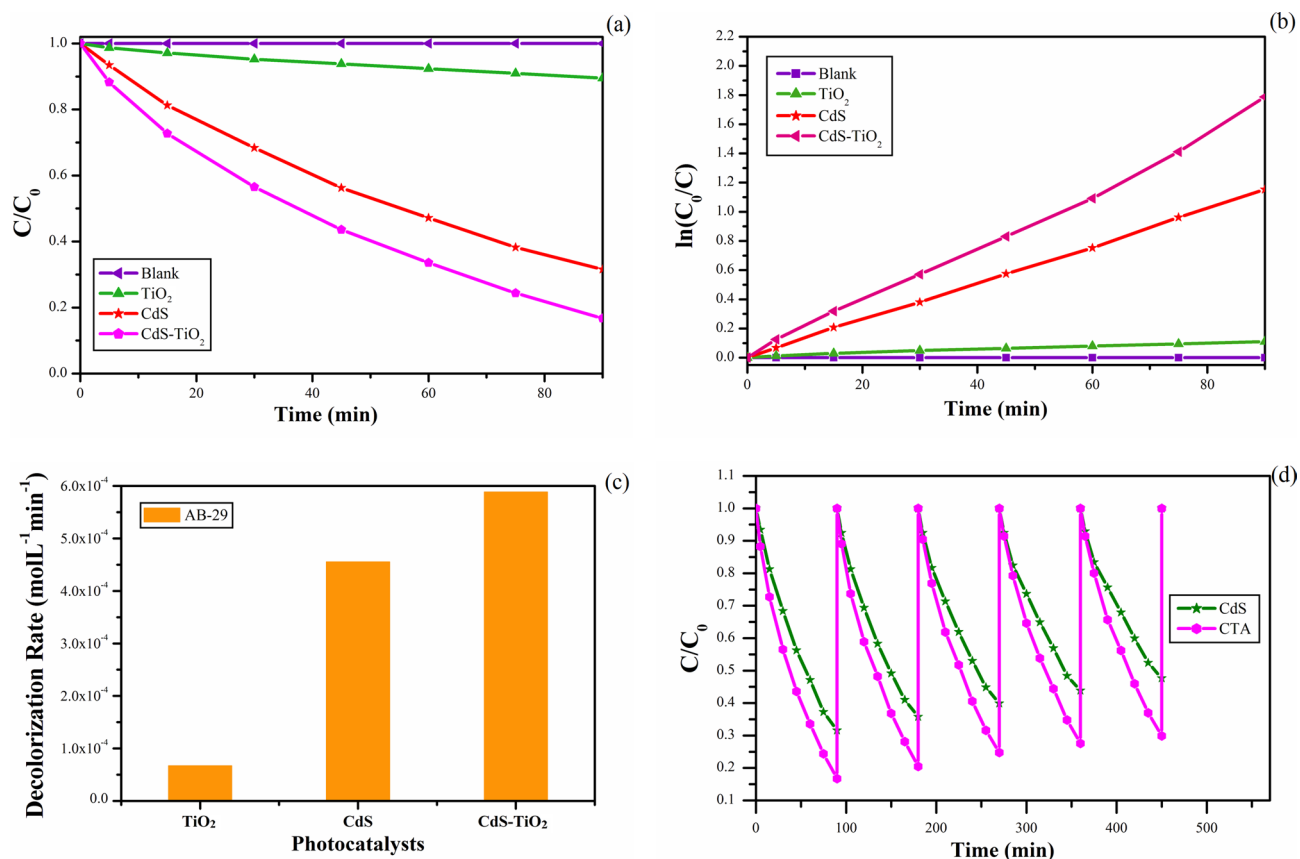


Figure 10. (a) Change in concentration of AB-29 with time in the presence and absence of synthesized TiO₂ nanoparticles (Tma, Tmb, Tmc, Tmd and Tme). (b) Change in concentration of AB-29 with time in the presence and absence of synthesized TiO₂ nanoparticles (Tma, Tmb, Tmc, Tmd and Tme). (c) The decolorization rate of AB-29 in the presence of different synthesized TiO₂ nanoparticles (Tma, Tmb, Tmc, Tmd and Tme). (d) Stability and recycle of TiO₂ nanocomposites (Tma, Tmb, Tmc, Tmd and Tme) for five consecutive cycles.

C_0 represents the starting reactant concentration (mM), C represents the reactant concentration (mM) at time “ t ”, and k_{app} represents the apparent pseudo-first-order rate constant (min^{-1}).

The data in Fig. 10a were in good agreement with the pseudo-first-order reaction for our experimental conditions, as shown in Fig. 10b by plotting $\ln(C_0/C)$ versus irradiation time. For all of the experiments, the correlation constant (R^2) for the fitted lines was calculated to be 0.99.

The dye's degradation rate was estimated using the formula below¹⁰¹:

$$-(d[C])/dt \rightarrow k[C]^n \quad (5)$$

k = rate constant, C = concentration of the dye, n = order of reaction.

Tme exhibited the maximum activity and nearly fully decolorized the solution in only 90 min, according to the Kinetic findings. The decolorization rate of AB-29 in the presence of several photocatalysts (Tma, Tmb, Tmc, Tmd, and Tme) demonstrated that the decolorization of AB-29 continued faster as the diameters of the TiO₂ nanoparticles dropped (Fig. 10c). The decolorization rate followed the order; Tma (4.6×10^{-4}) < Tmb (4.8×10^{-4}) < Tmc (5.0×10^{-4}) < Tmd (5.3×10^{-4}) < Tme ($5.5 \times 10^{-4} \text{ mol L}^{-1} \text{ min}^{-1}$). Increased catalyst surface area is responsible for the rise in photocatalytic efficiency when particle size decreases¹⁰².

In addition, when particle size decreases, band gap energy rises, reducing charge carrier recombination. It is widely assumed that a bigger band gap equates to greater redox capacity¹⁰³. As a result of its large surface area, enhanced bandgap, and strong redox capability with low photo corrosion, Tme had the highest photocatalytic activity. A photocatalytic reaction using TiO₂ is essentially a redox reaction including photogeneration, migration, trapping, and recombination of reactants adsorbed on its surface¹⁰⁴. The process of photocatalysis over titanium dioxide (TiO₂) can be explained as follows: photocatalysis was initiated by the absorption of a photon with energy equal to or greater than the bandgap of TiO₂ (3.2 eV), leading to photo-excitation, producing electron-hole (e^-/h^+) pairs (Eq. 6)¹⁰⁵. As a result, the TiO₂ particle operated as an electron donor or acceptor for molecules in the surrounding medium after irradiation. The photoexcited electron and hole took part in redox reactions with adsorbed species like water, hydroxide ions (OH⁻), organic compounds, and oxygen. The photoexcitation of TiO₂ under UV light irradiation is depicted in Fig. 11 as an example scheme.

The holes (h^+) oxidized water (H₂O) Eq. (7) or hydroxyl anion (OH⁻) Eq. (8) in the valence band to form the hydroxyl radical ($\cdot\text{OH}$), a highly potent and indiscriminate oxidant. Similarly, the electron (e^-) reduced the

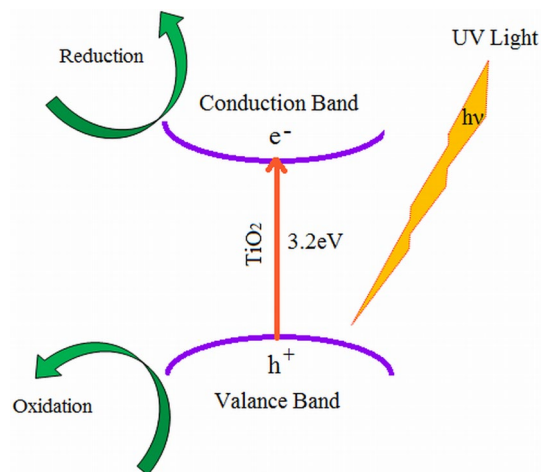


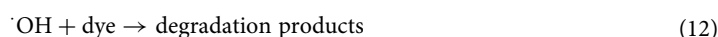
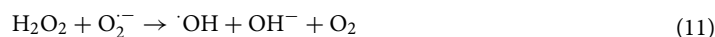
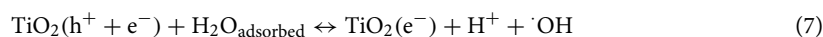
Figure 11. Schematic diagram of the photoexcitation of TiO₂ under UV light irradiation.

Cycle	Tma	Tmb	Tmc	Tmd	Tme
I	71.4	74.2	78.6	83.3	85.9
II	70.1	72.7	76.9	80.6	84.8
III	69.0	71.4	74.3	78.5	83.4
IV	67.8	70.4	72.7	76.8	82.1
V	66.7	68.9	71.4	74.4	80.5

Table 4. Decolorization rates of all the TiO₂ nanocatalysts (Tma, Tmb, Tmc, Tmd and Tme) for five consecutive cycles under UV light irradiation.

adsorbed O₂ to form superoxide radical anion (O₂^{•-}) Eq. (9) and hydrogen peroxide (H₂O₂) Eq. (10), which interacted to produce the ·OH radical Eq. (11). ·OH radicals quickly attacked contaminants on the surface, as well as in solution Eq. (12). This prevented the electron from recombining with the hole and resulted in a concentration of oxygen radical species, which aided in the attack on pollutants^{106,107}.

The reactions can be expressed as follows:



After 90 min of reaction time, the decolorization rates employing TiO₂ nanocatalysts (Tma, Tmb, Tmc, Tmd, and Tme) for the 5-cycling reuse are shown in Fig. 10d. Table 4 shows the results of five successive cycles of decolorization rates for all TiO₂ nanocatalysts (Tma, Tmb, Tmc, Tmd, and Tme). The catalytic activity of TiO₂ nanocatalysts (Tma, Tmb, Tmc, Tmd, and Tme) dropped marginally after the first cycles, according to the findings. Tme showed the most stability among them when compared to other TiO₂ nanocatalysts, which could be owing to Tme's small size.

Because the differences in relative stability among the TiO₂ NPs were not significant, it can be concluded that all of the produced TiO₂ NPs have good photocatalytic activity and UV light irradiation stability. However, because UV radiation accounts for just 4–6% of the total solar spectrum, it is vital to investigate TiO₂'s applicability in the visible zone.

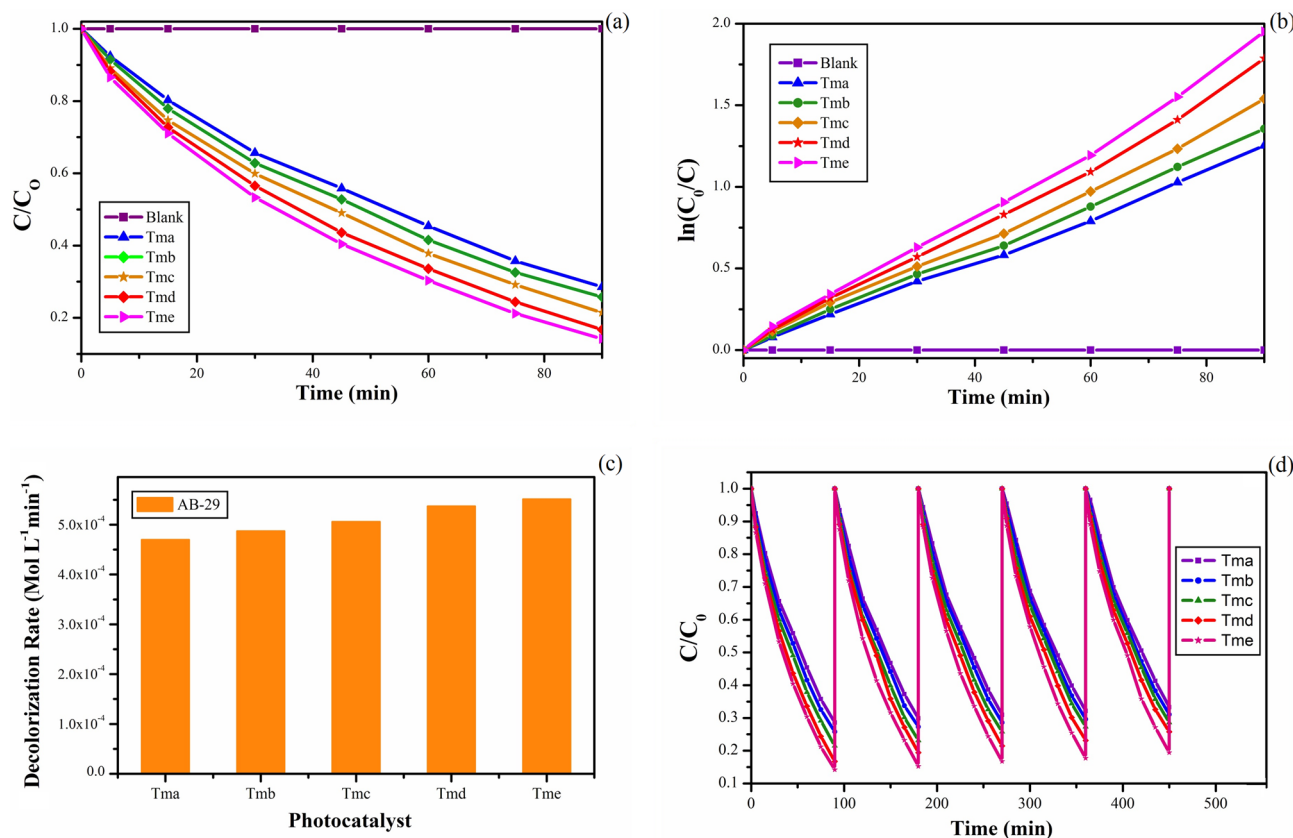


Figure 12. (a) Change in concentration of AB-29 with time in the presence and absence of synthesized CdS-TiO₂ nanoparticles in comparison to pure CdS and TiO₂ (b) Change in concentration of AB-29 with time in the presence and absence of synthesized CdS-TiO₂ nanoparticles in comparison to pure CdS and TiO₂. (c) The decolorization rate of AB-29 in the presence of different photocatalysts (TiO₂, CdS and CdS-TiO₂). (d) Stability and recycle of CdS-TiO₂ nanocomposite in comparison to pure CdS nanocomposite for five consecutive cycles.

As a result, TiO₂ NPs were combined with CdS NPs (CdS-TiO₂) and their photoactivity was investigated. A photo-degradation experiment using a dye derivative AB-29 in the presence of visible light was used to investigate the photocatalytic activity of CdS-TiO₂ nanoparticles. The photocatalytic experiment was conducted in the same manner as the TiO₂ NP experiment. The absorption edge of CdS-TiO₂ falls well inside visible radiation, resulting in enhanced photodegradation when exposed to visible light. In the presence of CdS-TiO₂ nanocomposite and ambient oxygen, irradiation of the dye under examination resulted in the decrease in absorption intensity as a function of irradiation time. In the presence and absence of CdS-TiO₂ photocatalysts, the relative change in the concentration of AB-29 (C/C_0) as a function of time is shown in Fig. 12a. The results were compared to nanoparticles of TiO₂ (Degussa P-25) and CdS⁵⁵. CdS-TiO₂ demonstrated 84% decolorization of AB-29 after 90 min of visible light irradiation, whereas CdS and TiO₂ showed only 68% and 09%, respectively, as shown in Fig. 12a. In the absence of photocatalyst, however, there was no discernible decrease in dye concentration. In the visible area, it was confirmed that CdS-TiO₂ NC had better photocatalytic activity than individual CdS and TiO₂ NP. The observations were consistent with a pseudo-first-order response, as illustrated in Fig. 12b by plotting $\ln(C_0/C)$ with irradiation time. A visualization of the natural logarithm of dye concentration as a function of irradiation duration Eq. (4) yielded the rate constant. The fitted lines' correlation constant (R^2) was calculated to be 0.99.

Using the Eq. (5), the dye degradation rate was estimated. The decolorization rate of AB-29 in the presence of CdS-TiO₂ photocatalyst as shown in Fig. 12c demonstrated that CdS-TiO₂ decolorized AB-29 was faster ($5.8 \times 10^{-4} \text{ mol L}^{-1} \text{ min}^{-1}$) than CdS ($4.5 \times 10^{-4} \text{ mol L}^{-1} \text{ min}^{-1}$) or TiO₂ ($0.67 \times 10^{-4} \text{ mol L}^{-1} \text{ min}^{-1}$). The increased photocatalytic effectiveness of CdS-TiO₂ in the visible area was owing to reduced charge carrier recombination as a result of better charge separation and TiO₂ extension in response to visible light.

Figure 12d depicts the photodegradation of AB-29 by CdS-TiO₂ NC and CdS NP over a five-cycle period. After 90 min of response time, the relative decolorization utilising CdS-TiO₂ for the 5-cycling reuse was 83.3%, 79.6%, 75.4%, 72.4%, and 70.1%, respectively.

The catalytic activity of CdS-TiO₂ was declined after the first cycles but at a lower rate than that of pure CdS NP, which reduced at a faster rate (68.4%, 64.3%, 60.1%, 56.1% and 52.3% respectively for 5 consecutive cycles). As a result, CdS-TiO₂ appears to be a superior photocatalyst to pure CdS NP, with increased activity and stability. However, photo corrosion of CdS, which forms cadmium cations, may be the cause of the decrease in CdS-TiO₂ stability during photocatalytic degradation events.

Under the instance of CdS/TiO₂, the narrow band-gap allowed CdS/TiO₂ to absorb more photons, increasing TiO₂'s photocatalytic efficiency in the sun. The absorption of a photon by CdS with energy equal to or greater

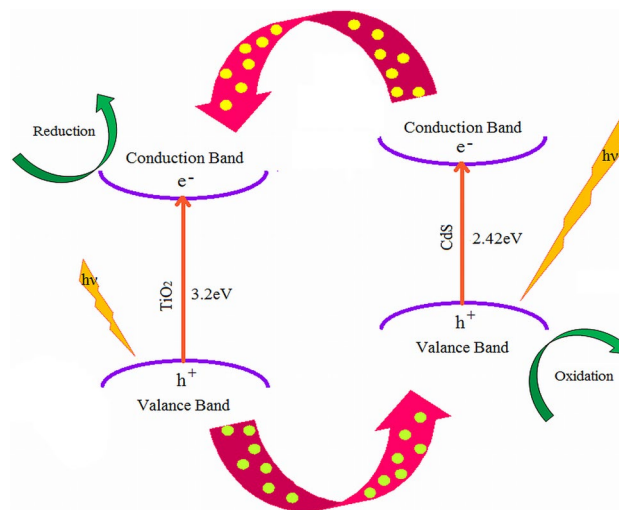
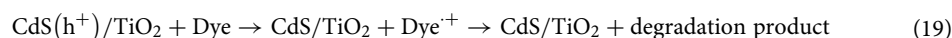
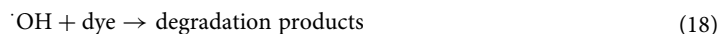
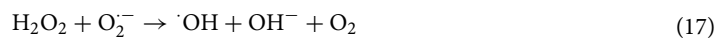
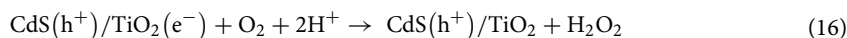
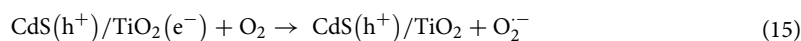
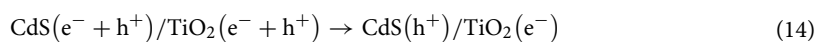
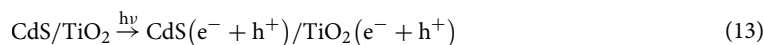


Figure 13. Schematic representation of photocatalytic mechanism followed by CdS-TiO₂.

than the bandgap of CdS (2.42 eV) (515 nm) caused excitation of electrons (e⁻) from VB to CB of CdS, leaving a positive vacancy (hole, h⁺) Eq. (13). TiO₂ also absorbed only a little quantity of visible light, causing photo-generated electrons and holes to occur in the CB and VB of TiO₂ Eq. (13). Because the conduction band (CB), valence band (VB), and band-gap of the two semiconductors were incompatible and overlapped, photo-generated electrons were directed from the CB of CdS to the CB of TiO₂, while photo-generated holes were directed from the VB of TiO₂ to the VB of CdS Eq. (14). Because holes flow in the opposite direction as electrons, they became trapped in the CdS. As a result, charge separation has improved, and recombination has decreased. A proposed mechanism for the degradation of contaminants on CdS coupled TiO₂ catalyst under visible light irradiation is given in Fig. 13 based on literature findings^{13,19,79,89,91,92} and our experiment results.

In oxygen-equilibrated environments, the photoexcited and transmitted electrons in TiO₂'s CB were scavenged by molecular oxygen O₂, yielding the superoxide radical anion O₂⁻ Eq. (15) and hydrogen peroxide H₂O₂ Eq. (16). The interaction of these intermediates resulted in the formation of the hydroxyl radical ·OH Eq. (17). The oxidative breakdown of AB-29 eq was then triggered by the ·OH radical Eq. (18)⁹¹. The photo-generated and transmitted hole in the VB of CdS cannot form hydroxyl radicals by oxidizing hydroxyl groups and H₂O molecules, but it can oxidize dye molecules to reactive intermediates and then to final products Eq. (19).

The reactions can be expressed as follows:



Because photo-generated holes in CdS nanocrystals are unable to convert hydroxyl groups to hydroxyl radicals due to their valence band potential, photo corrosion of CdS occurs, resulting in the formation of cadmium cations^{108–111}. The decrease in photo-stability of CdS-TiO₂ in the recycle experiment (Fig. 12d) also indicated leaching of cadmium cations. Table 5 provides additional examples of reports for the degradation performance of CdS and other nanocomposites for AB-29 azo dye as organic pollutants at the indicated experimental conditions.

Conclusion

Under ambient conditions, Titanium Dioxide nanoparticles (Tma, Tmb, Tmc, Tmd, and Tme) were effectively produced using a single pot chemical precipitation approach. The size of the TiO₂ nanocluster was reduced as the concentration of the Ti precursor dropped, as seen by the prepared TiO₂ NP. In all TiO₂ samples, a mixed crystalline phase was detected. The micrographical analysis demonstrated the production of spherical clusters

Catalyst	D (%)	Refs
Montmorillonite K10-Cu(II)ethylenediamine (MMTK10-Cu(en)2)	90	112
CdS	98	60
CdS with Sulphide ions	98	113
Fe (II) doped CdS	98.75	114
ZnFe ₂ O ₄	98.83	115
CdS-TiO ₂	84	Present work

Table 5. Comparative degradation of AB-29 azo dye by various nanocomposites.

whose diameters shrank drastically as the Ti precursor concentration dropped. With the decrease in Ti precursor concentration, the absorption spectra indicated a minor blue shift and, as a result, a slight rise in bandgap energies of TiO₂ NP was observed. Finally, under UV light, the photocatalytic response for the breakdown of the organic dye AB-29 was improved as TiO₂ had photoactive under UV light and CdS found as photocorroded during photocatalytic processes, an effort was made to combine the beneficial qualities of both CdS and TiO₂ while minimizing their disadvantages. A sandwich-type nanocomposite (CdS-TiO₂) of CdS with TiO₂ was also successfully synthesized. The CdS-TiO₂ showed good elemental purity and thermal stability. Cubic CdS and anatase TiO₂ were discovered in the XRD spectrum. Because of the presence of TiO₂, the absorption edge of CdS in CdS-TiO₂ shifted slightly blue, but the absorption edge of TiO₂ was identical to that of pure anatase TiO₂, indicating an improvement in the crystal structure. TiO₂'s optical sensitivity was moved towards the visible range in CdS-TiO₂, allowing it to be photocatalytically active in visible light as well. With the decrease in Ti precursor concentration, the absorption spectra indicated a minor blue shift and, as a result, a slight rise in bandgap energies of TiO₂ NP.

Received: 27 October 2021; Accepted: 2 March 2022

Published online: 06 April 2022

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Competing interests

The authors declare no competing interests.

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