

Engineering Long-Range Order in Supramolecular Assemblies on Surfaces: The Paramount Role of Internal Double Bonds in Discrete Long-Chain Naphthalenediimides

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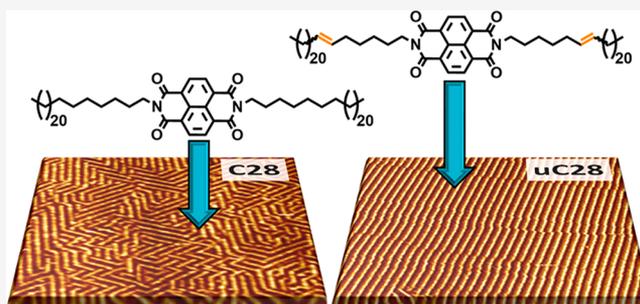


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ABSTRACT: Achieving long-range order with surface-supported supramolecular assemblies is one of the pressing challenges in the prospering field of non-covalent surface functionalization. Having access to defect-free on-surface molecular assemblies will pave the way for various nanotechnology applications. Here we report the synthesis of two libraries of naphthalenediimides (NDIs) symmetrically functionalized with long aliphatic chains (C_{28} and C_{33}) and their self-assembly at the 1-phenyloctane/highly oriented pyrolytic graphite (1-PO/HOPG) interface. The two NDI libraries differ by the presence/absence of an internal double bond in each aliphatic chain (unsaturated and saturated compounds, respectively). All molecules assemble into lamellar arrangements, with the NDI cores lying flat and forming 1D rows on the surface, while the carbon chains separate the 1D rows from each other. Importantly, the presence of the unsaturation plays a dominant role in the arrangement of the aliphatic chains, as it exclusively favors interdigitation. The fully saturated tails, instead, self-assemble into a combination of either interdigitated or non-interdigitated diagonal arrangements. This difference in packing is spectacularly amplified at the whole surface level and results in almost defect-free self-assembled monolayers for the unsaturated compounds. In contrast, the monolayers of the saturated counterparts are *globally* disordered, even though they *locally* preserve the lamellar arrangements. The experimental observations are supported by computational studies and are rationalized in terms of stronger van der Waals interactions in the case of the unsaturated compounds. Our investigation reveals the paramount role played by internal double bonds on the self-assembly of discrete large molecules at the liquid/solid interface.



INTRODUCTION

The non-covalent functionalization of surfaces has become one of the pillars of nanotechnology in the past 20 years.^{1–4} Achieving exact control over the formation of monolayers allows scientists to modulate the properties of surfaces in a predictable manner,⁵ which holds promise for relevant technological breakthroughs.^{6–8} For instance, controlling the density of nitrogen-based n-dopants on graphene via monolayer formation has proved to play a pivotal role in tuning the charge carrier concentration of the modified 2D material.⁹

Surface-supported supramolecular assemblies rely on stabilizing interactions between the adsorbed molecules and the surface, as well as favorable intermolecular interactions between the adsorbed compounds.^{10,11} Given the significant epitaxial stabilization of 64 meV (1.5 kcal/mol) *per* methylene unit that highly oriented pyrolytic graphite (HOPG) exerts at the liquid/HOPG interface, the molecular designs typically adopted in the field feature long alkyl chains—usually up to 18 carbon atoms—to favor adsorption to the substrate.^{12,13} Previous work on long-chain alkanes^{14–17} highlighted the

tendency of long aliphatic tails to form thermodynamically stable self-assembled monolayers. Moreover, a number of studies have highlighted the role played by the alkyl chains in the 2D structure.^{18–23} While a favorable interaction between the molecules and the substrate is certainly necessary, the structure and extent of order of the assemblies generated are mostly the manifestation of the intermolecular interactions between the adsorbed molecules.^{10,13} On-surface supramolecular assemblies are typically created by resorting to non-covalent forces, such as van der Waals (vdW) interactions,²⁴ hydrogen bonding (HB),^{25–32} coordination chemistries,^{33–36} and halogen bonding.^{37,38} So far, various approaches focused on limiting the number of domain boundaries and/or molecular defects to improve the organization and quality of the 2D architectures gener-

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ated.^{39–41} More recently, spatially confining the self-assembly process into nanocorrals created on HOPG afforded impressive results in terms of order.⁴² However, the defect-free engineering of surface-supported supramolecular assemblies on unconfined HOPG remains a major challenge for the whole field.⁴³

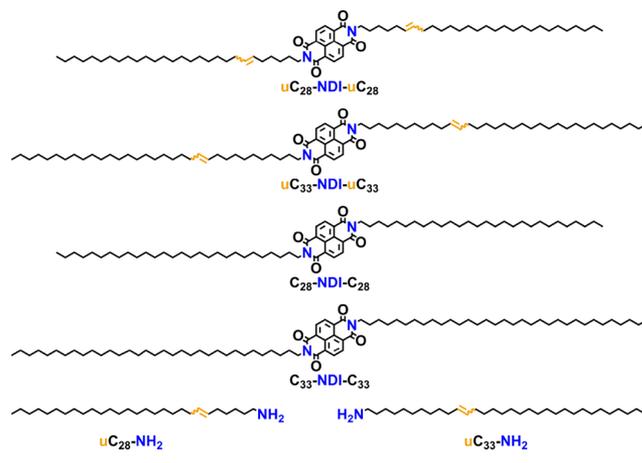
Given the high commercial availability of alkylating reagents in the C_1 – C_{22} range, a large body of work has been carried out using alkyl chain-functionalized compounds. To the best of our knowledge, however, the attention dedicated to their unsaturated analogues featuring internal double bonds has been very limited. Deng et al. compared the assemblies of *E*-oleic acid and *Z*-oleylamine at the 1-phenyloctane (1-PO)/HOPG interface.⁴⁴ Monolayers obtained from *E*-oleic acid at the 1-PO/HOPG interface were characterized by a high stability, while those deriving from the amine with *Z*-configuration were poorly stable.⁴⁴ This comparative study⁴⁴ was consistent in terms of chain length (oleyl = C_{18}) and double bond position (between carbon atoms 9 and 10), but the two structures investigated differed in double bond configurations (*E* vs *Z*) and end-group functionalities (carboxylic vs amino). The hypothesis that various parameters could play a role in the overall stability of the on-surface assemblies could not be ruled out. More recently, Shokri et al. suggested that the introduction of a *Z*-configured double bond in the side chains of bis(urea) molecules leads to the formation of long-range ordered polymers on graphite.⁴⁵ However, the study was conducted with one chain length only (C_{18}), and the influence of the internal double bond was visible only after storage of the modified surface for 1 year.⁴⁵ Although both studies independently posed the question of the influence of internal double bonds on on-surface self-assembly processes, no further investigations followed in this direction.

Intrigued by the chance to unravel the possible influence of internal double bonds on surface-supported supramolecular assemblies, we envisioned a system based on long carbon chains featuring internal double bonds symmetrically bound at the periphery of naphthalenediimides⁴⁶ (NDIs). NDIs are electron poor⁴⁶ and have a pronounced tendency to be deposited at the liquid/HOPG interface due to a highly favorable enthalpy of interaction.^{47,48} Symmetrical NDIs functionalized with fully hydrogenated, linear alkyl chains (C_n -NDI- C_n design) with a number of carbon atoms (n) in the 3–18 range were previously investigated at the 1-tetradecane/HOPG interface.⁴⁷ Particularly relevant for the present work, alkyl chains with a number of carbon atoms equal to or greater than 13 units consistently afforded lamellar arrangements in which both the long carbon chains and aromatic cores lie flat on the surface, as visualized with scanning tunneling microscopy (STM).⁴⁷ The morphology of the obtained monolayers was explained (lamellar), but the larger long-range-ordered areas obtained represented only a limited part of the surface (50 nm × 50 nm). In order to exploit the potential of supramolecular assemblies on surfaces, ordered areas larger than 100 nm × 100 nm (at least) are highly desirable.⁴³ Relying on the consistency of the C_n -NDI- C_n design with $n > 13$ (lamellar arrangement), we hypothesized that extending the carbon chain length in the C_n -NDI- C_n design would be beneficial for expanding the order extent. Moreover, to answer the key question about the role of the internal double bonds, we envisioned a C_n -NDI- C_n system that features internal unsaturations in the carbon chain. Reducing these double bonds by catalytic hydrogenation should offer the possibility to

compare compounds that belong to a very consistent molecular platform (long-chain NDIs) but that differ by a subtle structural modification (formally two hydrogen molecules).

We present the synthesis and on-surface investigation of uC_{28} -NDI- uC_{28} and uC_{33} -NDI- uC_{33} (unsaturated NDIs) and compare them to their hydrogenated counterparts C_{28} -NDI- C_{28} and C_{33} -NDI- C_{33} (saturated NDIs). The fully extended chemical structures are shown in Chart 1. The studied NDIs

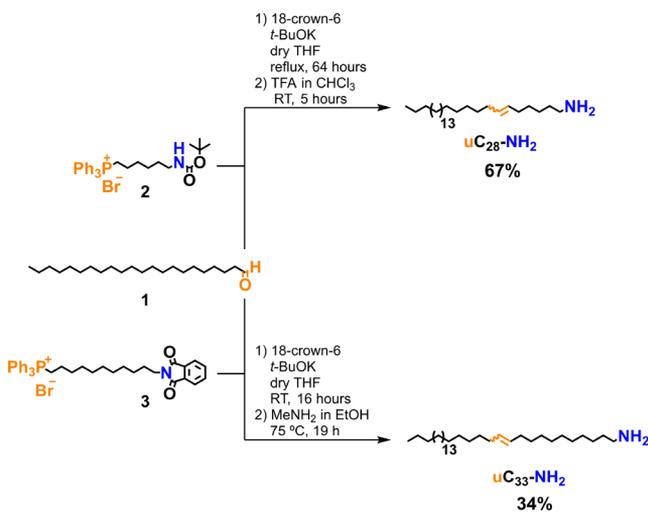
Chart 1. Fully Extended Chemical Structures of uC_{28} -NDI- uC_{28} , uC_{33} -NDI- uC_{33} , C_{28} -NDI- C_{28} , C_{33} -NDI- C_{33} , and Key Synthetic Intermediates uC_{28} -NH₂ and uC_{33} -NH₂



feature either 28 or 33 carbon atoms in the linear chain (C_{28} and C_{33} , respectively) and only differ by the presence/absence of one unsaturation in each carbon chain. The unsaturation (when present) is highlighted by the letter *u*. The key synthetic intermediates in the preparation of the final compounds were the unsaturated amines uC_{28} -NH₂ and uC_{33} -NH₂, also shown in Chart 1. We discover that the self-assembled monolayers obtained at the 1-PO/HOPG interface from the unsaturated compounds are characterized by a significantly higher degree of organization compared to their saturated counterparts, with a size difference for ordered domains corresponding to thousands of squared nanometers. The experimental results are supported by computational studies. Our results point to the establishment of the internal double bond as a counterintuitive yet key structural element for obtaining long-range order in self-assembled monolayers at the liquid/solid interface. Finally, the highly adaptive character of supramolecular assemblies at the liquid/solid interface⁴⁹ allows for the use of mixtures of *EE*, *EZ*, and *ZZ* isomers of the unsaturated NDIs, as the system selects the most stable pattern created (almost exclusively) by one stereoisomer.

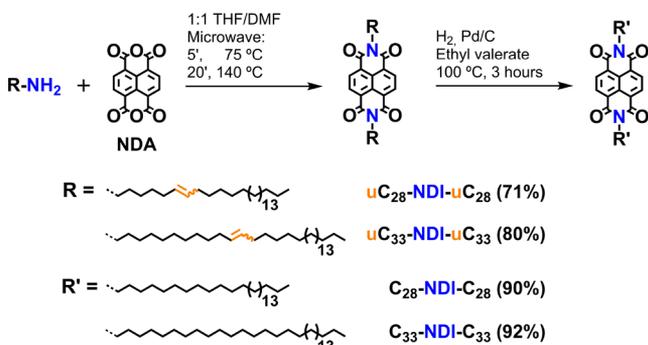
RESULTS AND DISCUSSION

Synthesis and Characterization. Compounds uC_{28} -NH₂ and uC_{33} -NH₂ (Chart 1) were the key intermediates in the preparation of the target NDIs. They were synthesized from building blocks 1,⁵⁰ 2, and 3 in 67% and 40% yield, respectively, applying a strategy based on Wittig olefination (Scheme 1). Details on the preparation of 2 and 3 are presented in the Supporting Information (SI). The amines were obtained as an ~84:16 mixture of non-separable *Z* and *E* isomers, respectively (assigned by integration of the ¹³C NMR spectra, see SI). The preference for the *Z*-configuration of the

Scheme 1. Synthesis of $uC_{28}\text{-NH}_2$ and $uC_{33}\text{-NH}_2$ 

double bond is in line with the Wittig olefination conditions adopted, especially with the use of non-stabilized phosphonium ylides.⁵¹ The position of the unsaturation along the two carbon chains (between C_6 and C_7 in uC_{28} , and C_{11} and C_{12} in uC_{33}) was exactly engineered, as will be corroborated by our STM study (*vide infra*). The choice of the base-induced Wittig reaction for the elongation step implied a careful choice of protecting groups for the amino moieties on the phosphonium salts. We opted for *tert*-butyloxycarbonyl- (Boc) and phthalimide- (Phth) protected **2** and **3** for $uC_{28}\text{-NH}_2$ and $uC_{33}\text{-NH}_2$, respectively, after an initial screening of the reaction conditions. A related approach for obtaining discrete oligoethylenes ($C_n \leq 400$) was previously reported in the effort to build model compounds to study the crystallization of polyethylene.^{52,53} Being complementary, our synthesis allows for the introduction of functional groups in the linear moieties, expanding the applicability of these long aliphatic chains. The cleavage of the -Boc and -Phth protecting groups was carried out with trifluoroacetic acid (TFA) and methylamine solution in ethanol (33 wt%), respectively (experimental details in SI).

The unsaturated amines were subsequently coupled to commercially available naphthalenedianhydride (NDA) via a modified microwave assisted protocol (Scheme 2).^{54–56} The unsaturated NDIs $uC_{28}\text{-NDI-}uC_{28}$ and $uC_{33}\text{-NDI-}uC_{33}$ were obtained in 71% and 80% yield, respectively, as non-resolvable mixtures of *ZZ:ZE:EE* isomers ($\sim 70.5:27:2.5$, based on the possible combinations of the two reacting amines) after

Scheme 2. Synthesis of $uC_{28}\text{-NDI-}uC_{28}$, $uC_{33}\text{-NDI-}uC_{33}$, $C_{28}\text{-NDI-}C_{28}$, and $C_{33}\text{-NDI-}C_{33}$ 

chromatographic purification. The fully saturated analogs $C_{28}\text{-NDI-}C_{28}$ and $C_{33}\text{-NDI-}C_{33}$ were prepared from their alkenyl counterparts by palladium-on-carbon (Pd/C)-catalyzed hydrogenation in ethyl valerate at 100 °C (Scheme 1) and purified by Soxhlet extraction (see SI).

Self-Assembly on HOPG. We started our investigation by studying the self-assembly of saturated $C_{28}\text{-NDI-}C_{28}$ and $C_{33}\text{-NDI-}C_{33}$ at the 1-PO/HOPG interface. Solutions of the two NDIs (0.4 mg/mL in 1-PO) were drop-cast at 100 °C onto freshly cleaved HOPG substrates and subsequently imaged. The saturated compounds spontaneously self-assembled into ordered lamellae immediately after deposition. In the STM images, the aromatic cores appear as bright protrusions and the alkyl chains as dark regions (Figure 1a–d). The lamellar

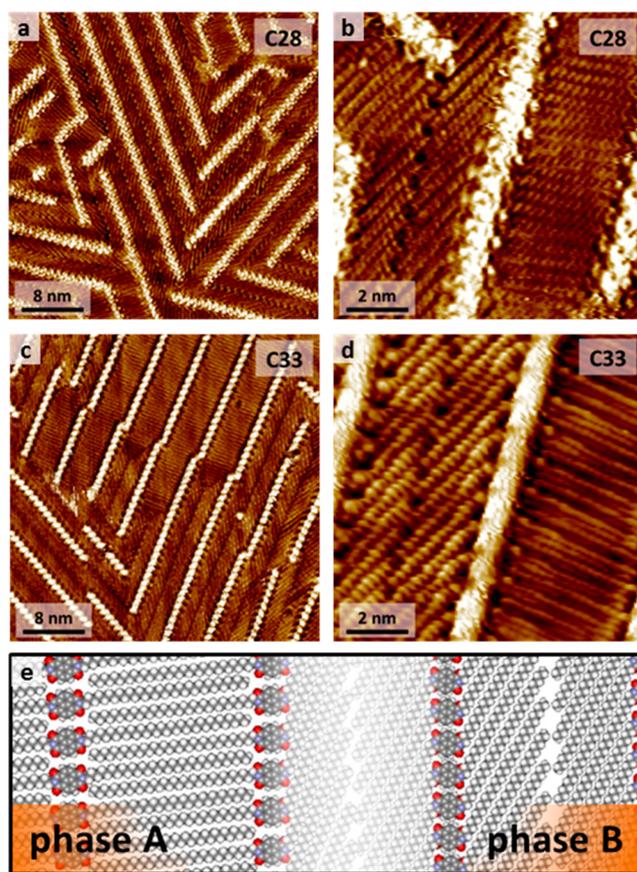


Figure 1. Self-assembly of $C_{28}\text{-NDI-}C_{28}$ and $C_{33}\text{-NDI-}C_{33}$ at the 1-phenyloctane/HOPG interface. (a) STM image of $C_{28}\text{-NDI-}C_{28}$ (40 nm \times 40 nm, $V_{\text{tip}} = 1$ V, $I_{\text{set}} = 50$ pA). (b) STM image of $C_{28}\text{-NDI-}C_{28}$ showing the two arrangements of alkyl chains (phase A and phase B) (10 nm \times 10 nm, $V_{\text{tip}} = 1$ V, $I_{\text{set}} = 50$ pA). (c) STM image of $C_{33}\text{-NDI-}C_{33}$ (40 nm \times 40 nm, $V_{\text{tip}} = -0.6$ V, $I_{\text{set}} = 50$ pA). (d) STM image of $C_{33}\text{-NDI-}C_{33}$ showing the two arrangements of alkyl chains (phase A and phase B) (10 nm \times 10 nm, $V_{\text{tip}} = 0.6$ V, $I_{\text{set}} = 150$ pA). (e) Schematic representation of lamellar phase A (with interdigitation of the alkyl chain) and phase B (no interdigitation, diagonal organization of the alkyl chains).

packings are consistent with the aromatic cores lying flat and next to each other on the surface, while the alkyl chains are straight and parallel to each other and modulate the distance between the NDI cores (Figure 1a,c).⁴⁷ The arrangement of the individual alkyl chains was determined from high-resolution STM images. We could identify two different

Table 1. Unit Cell Parameters for the Supramolecular Arrangements of C_{28} -NDI- C_{28} , uC_{28} -NDI- uC_{28} , C_{33} -NDI- C_{33} , and uC_{33} -NDI- uC_{33} at the 1-PO/HOPG Interface^a

compound	<i>a</i> [nm]	<i>b</i> [nm]	γ [deg]	lamellar phase	domain size average [nm ²]	domain size median [nm ²]	disordered areas [%]
C_{28} -NDI- C_{28}	4.45 ± 0.24	0.88 ± 0.08	85.21 ± 3.39	A and B	949	737	26 ± 5
uC_{28} -NDI- uC_{28}	4.53 ± 0.08	0.86 ± 0.10	87.33 ± 1.78	A	6764	2923	–
C_{33} -NDI- C_{33}	5.29 ± 0.49	0.99 ± 0.10	84.10 ± 5.28	A and B	1268	540	24 ± 8
uC_{33} -NDI- uC_{33}	5.27 ± 0.08	0.94 ± 0.06	84.93 ± 1.80	A	8026	3684	–

^aThe lengths of the unit cell vectors are labeled *a* and *b*, and the internal angle is specified by γ (see also Figure 5a,b).

packing modes for the aliphatic chains of C_{28} -NDI- C_{28} (Figure 1b) and C_{33} -NDI- C_{33} (Figure 1d): an interdigitated mode, hereby defined as “lamellar phase A”, and a non-interdigitated diagonal mode, denominated “lamellar phase B”. A pictorial representation of both lamellar phases A and B is given in Figure 1e. The lamellae are rotated by 60° with respect to each another. The observation of the two different packing modes of the aliphatic chains is in line with previous reports on C_n -NDI- C_n , with $13 \leq n \leq 18$.⁴⁷ In this respect, extending the length of the alkyl chains did not result in significant differences compared to previous studies.

The unit cell parameters determined for C_{28} -NDI- C_{28} and C_{33} -NDI- C_{33} are listed in Table 1, while for a visualization of the unit cell we refer to Figure 5 and the discussion further on. Although the two lamellar assemblies differ in the orientation of the aliphatic chains, the unit cell parameters do not differ for a fixed alkyl chain length. The measured values are $a = 4.45 \pm 0.24$ nm, $b = 0.88 \pm 0.08$ nm, and $\gamma = 85.21 \pm 3.39^\circ$ for C_{28} -NDI- C_{28} , and $a = 5.29 \pm 0.49$ nm, $b = 0.99 \pm 0.10$ nm, and $\gamma = 84.10 \pm 5.28^\circ$ for C_{33} -NDI- C_{33} .

Next, we focused on unsaturated uC_{28} -NDI- uC_{28} and uC_{33} -NDI- uC_{33} at the 1-PO/HOPG interface under similar experimental conditions. Exemplary images are shown in Figure 2. Assemblies similar to the ones obtained for the

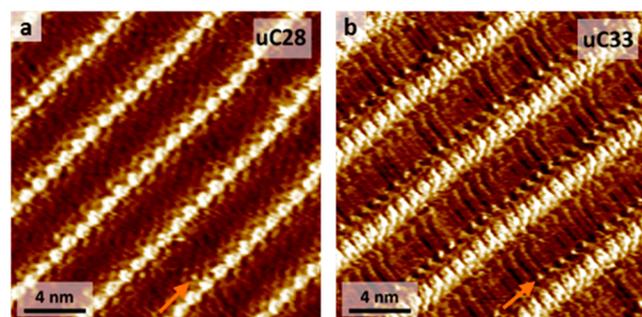


Figure 2. Self-assembly of uC_{28} -NDI- uC_{28} and uC_{33} -NDI- uC_{33} at the 1-phenyloctane/HOPG interface. (a) STM image of uC_{28} -NDI- uC_{28} (20 nm × 20 nm, $V_{\text{tip}} = 1$ V, $I_{\text{set}} = 100$ pA). (b) STM image of uC_{33} -NDI- uC_{33} (20 nm × 20 nm, $V_{\text{tip}} = 1$ V, $I_{\text{set}} = 90$ pA). The double bonds appear as bright protrusions next to the bright NDI cores (orange arrows). Both unsaturated molecules assemble in an interdigitated fashion (phase A).

saturated NDIs were observed with uC_{28} -NDI- uC_{28} and uC_{33} -NDI- uC_{33} . The lamellar arrangements correspond to parallel NDI cores flat on the surface (bright protrusions) and the interdigitating aliphatic chains that tune the distance between them (dark regions) (Figure 2a for uC_{28} -NDI- uC_{28} , and Figure 2b for uC_{33} -NDI- uC_{33}). In stark contrast with the saturated NDIs, additional bright protrusions were observed in the STM images of uC_{28} -NDI- uC_{28} and uC_{33} -NDI- uC_{33} (orange arrows

in Figure 2a,b). They appeared symmetrically with respect to the aromatic cores, and their distance to the aromatic cores changed upon extending the chain length. These features were less evident in the case of uC_{28} -NDI- uC_{28} (Figure 2a), while they appeared more separated and resolved in the case of uC_{33} -NDI- uC_{33} (Figure 2b). We attribute these additional bright protrusions to the double bonds present in the unsaturated chains. As a general remark, the imaging of the double bonds was in general easier for uC_{33} -NDI- uC_{33} than uC_{28} -NDI- uC_{28} . Such behavior is attributed to the structural differences between the two molecules and corroborates the more remote position of the double bond with respect to the NDI core in uC_{33} -NDI- uC_{33} (between C_{11} and C_{12}) compared to uC_{28} -NDI- uC_{28} (between C_6 and C_7).

The determined unit cell parameters for uC_{28} -NDI- uC_{28} and uC_{33} -NDI- uC_{33} are reported in Table 1. The values are very similar to those obtained for the saturated counterparts, pointing to an apparent similarity between the assemblies of saturated and unsaturated NDIs. The close resemblance of the unit cell parameters of the NDIs with the same chain length (C_{28} or C_{33}) strongly suggests that the self-assembled monolayers are mostly formed by all-*E*-configured molecules. The *E*-configured carbon chains are expected to assume zigzag conformations on HOPG in a very similar fashion to alkyl chains and hence cover distances comparable to their saturated counterparts (C_{28} and C_{33}). The *Z*-configured chains, instead, should differ in distance, as the *Z*-configuration forces a bending of the carbon which cannot be compensated by a rotation around the double bond (forbidden in this case). As an indicative example, the carbon chain of *Z*-oleylamine shows this bending as a consequence of the fixed configuration of the double bond.⁴⁴ The deposition of mainly *EE*-isomer is remarkable, since this isomer is calculated to be roughly 2.5% of the whole population of unsaturated NDIs (based on the ¹³C NMR analysis of $uC_{28}NH_2$ and $uC_{33}NH_2$ and the binomial distributions of the two amines). The remaining 97.5% of the material, which accounts for the *EZ*- and *ZZ*-isomers, remains in the overlying liquid phase and is not imaged. We conclude that our long-chain NDIs system at the 1-PO/HOPG interface is highly dynamic and adaptive. Such characteristic allows for the use of *EE*-, *EZ*-, and *ZZ*-isomers mixtures because the system autonomously selects the isomer that forms the most stable pattern on the surface—the *EE*-isomer in this case. A similar concept has been recently reported by Samori, Lehn, et al. with on-surface bisimine formation.⁴⁹ It should be noted that some *Z*-configured double bonds were present in the monolayer and we speculate that these are responsible for the tiny defects and irregularities observed in the monolayers.

Pivotal Role of the Internal Double Bonds in the 2D Crystallization. The results presented so far have apparently revealed only minor differences in the self-assembly of both

saturated and unsaturated NDIs at the 1-PO/HOPG interface. However, a very important difference arises in the organization of the aliphatic chains: the fully saturated ones simultaneously arrange in either phase A or B, while the unsaturated chains only pack in the phase A fashion. This difference does not alter the *local* ordering of the self-assembled monolayer, but has dramatic repercussions on the *global* ordering of the 2D architectures. The presence of just one type of self-assembly arrangement (phase A) for the carbon chains of $\text{uC}_{28}\text{-NDI-uC}_{28}$ and $\text{uC}_{33}\text{-NDI-uC}_{33}$ results in considerably increased domain sizes and thus, in a reduction of the number of domains per area compared to those created by their saturated counterparts. The contrast is striking: for large-scale images, very large domains and significantly less defects are observed in the STM images of $\text{uC}_{28}\text{-NDI-uC}_{28}$ and $\text{uC}_{33}\text{-NDI-uC}_{33}$ (Figure 3b,d, respectively) compared to those of $\text{C}_{28}\text{-NDI-C}_{28}$ and $\text{C}_{33}\text{-NDI-C}_{33}$ (Figure 3a,c, respectively).

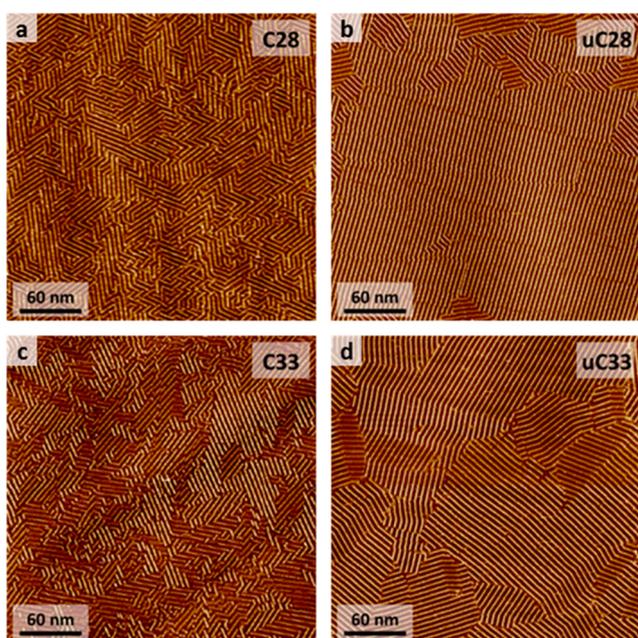


Figure 3. Large-scale STM images ($300\text{ nm} \times 300\text{ nm}$) of different NDIs at the 1-PO/HOPG interface: (a) $\text{C}_{28}\text{-NDI-C}_{28}$ ($V_{\text{tip}} = 1\text{ V}$, $I_{\text{set}} = 100\text{ pA}$), (b) $\text{uC}_{28}\text{-NDI-uC}_{28}$ ($V_{\text{tip}} = 1\text{ V}$, $I_{\text{set}} = 80\text{ pA}$), (c) $\text{C}_{33}\text{-NDI-C}_{33}$ ($V_{\text{tip}} = 1\text{ V}$, $I_{\text{set}} = 100\text{ pA}$), and (d) $\text{uC}_{33}\text{-NDI-uC}_{33}$ ($V_{\text{tip}} = 1\text{ V}$, $I_{\text{set}} = 100\text{ pA}$).

The different position of the double bonds in $\text{uC}_{28}\text{-NDI-uC}_{28}$ and $\text{uC}_{33}\text{-NDI-uC}_{33}$ does not seem to play a role, considering the very similar behavior (Figure 3b,d). Results obtained on a positional isomer of $\text{uC}_{28}\text{-NDI-uC}_{28}$ with the double bond located between carbon atoms 11 and 12 also rule out a positional influence of the unsaturation (see SI). In contrast, the overview STM images of the saturated NDIs are characterized by relatively small domains accompanied by disordered areas. The lack of a clear preference for either lamellar phase A or B arrangements seems to cause the existence of disordered regions and welter areas (Figure 3a,c; see SI for the assignment of disordered areas).

We conducted a statistical analysis on the domain sizes for the different NDIs to support the qualitative observation on the dramatic influence of the internal double bonds. For a detailed description on the assignment of the domain size and further experimental observations upon scanning see the SI.

The results on the domain size distributions for $\text{C}_{28}\text{-NDI-C}_{28}$ and $\text{uC}_{28}\text{-NDI-uC}_{28}$, and $\text{C}_{33}\text{-NDI-C}_{33}$ and $\text{uC}_{33}\text{-NDI-uC}_{33}$, are summarized by the two histograms shown in Figure 4. The

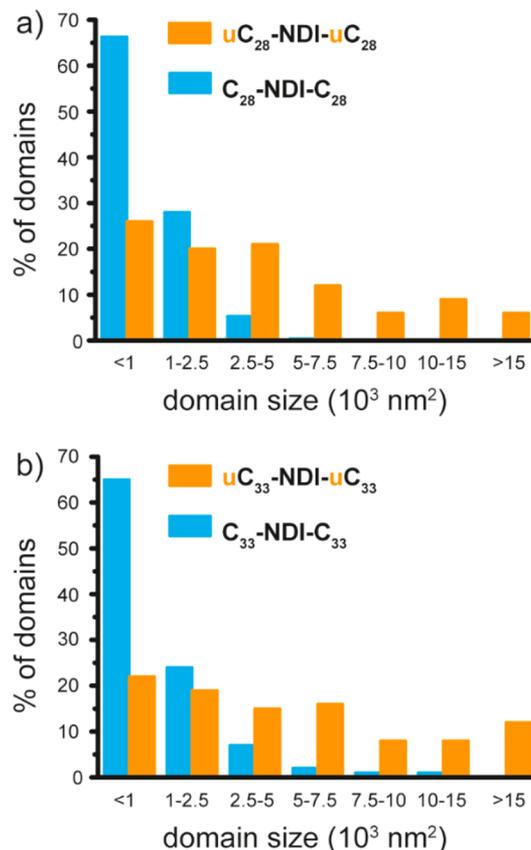


Figure 4. Domain size distribution for (a) $\text{C}_{28}\text{-NDI-C}_{28}$ (blue) and $\text{uC}_{28}\text{-NDI-uC}_{28}$ (orange), and (b) $\text{C}_{33}\text{-NDI-C}_{33}$ (blue) and $\text{uC}_{33}\text{-NDI-uC}_{33}$ (orange). Y-axis: percentage of ordered domains (% of domains); X-axis: domain size (10^3 nm^2).

saturated NDIs mainly arrange in relatively small domains ($\leq 1000\text{ nm}^2$) (Figure 4a and 4b, blue columns). Moreover, on roughly 24% of the surface, the molecules do not arrange in an ordered way resulting in disordered areas. On the other hand, the images of the unsaturated NDIs show only a marginal amount of disordered areas. The observed domains reach much larger extensions, with a significant population larger than 15000 nm^2 (Figure 4a,b, orange columns).

The experimental results were rationalized by means of a computational study (for computational details see SI). For consistency with our experimental observations on the self-assembled monolayers, we studied only alkenes with *E*-configurations. Our working hypothesis focused on the increasing strength of van der Waals interchain interactions upon introducing internal double bonds in the carbon chains. Initial studies on shorter carbon chains (C_6) in the gas phase showed a promising trend in this respect (see SI). Periodic energy decomposition analysis (PEDA)⁵⁷ revealed that the interaction energy between neighboring chains becomes more favorable upon introducing the internal double bonds (Figure S41 and Table S1). Encouraged by these results, we focused on both $\text{C}_{28}\text{-NDI-C}_{28}$ and $\text{uC}_{28}\text{-NDI-uC}_{28}$ in the lamellar A organization. In the calculated molecular arrangements, the NDI cores lay flat and next to each other while the carbon

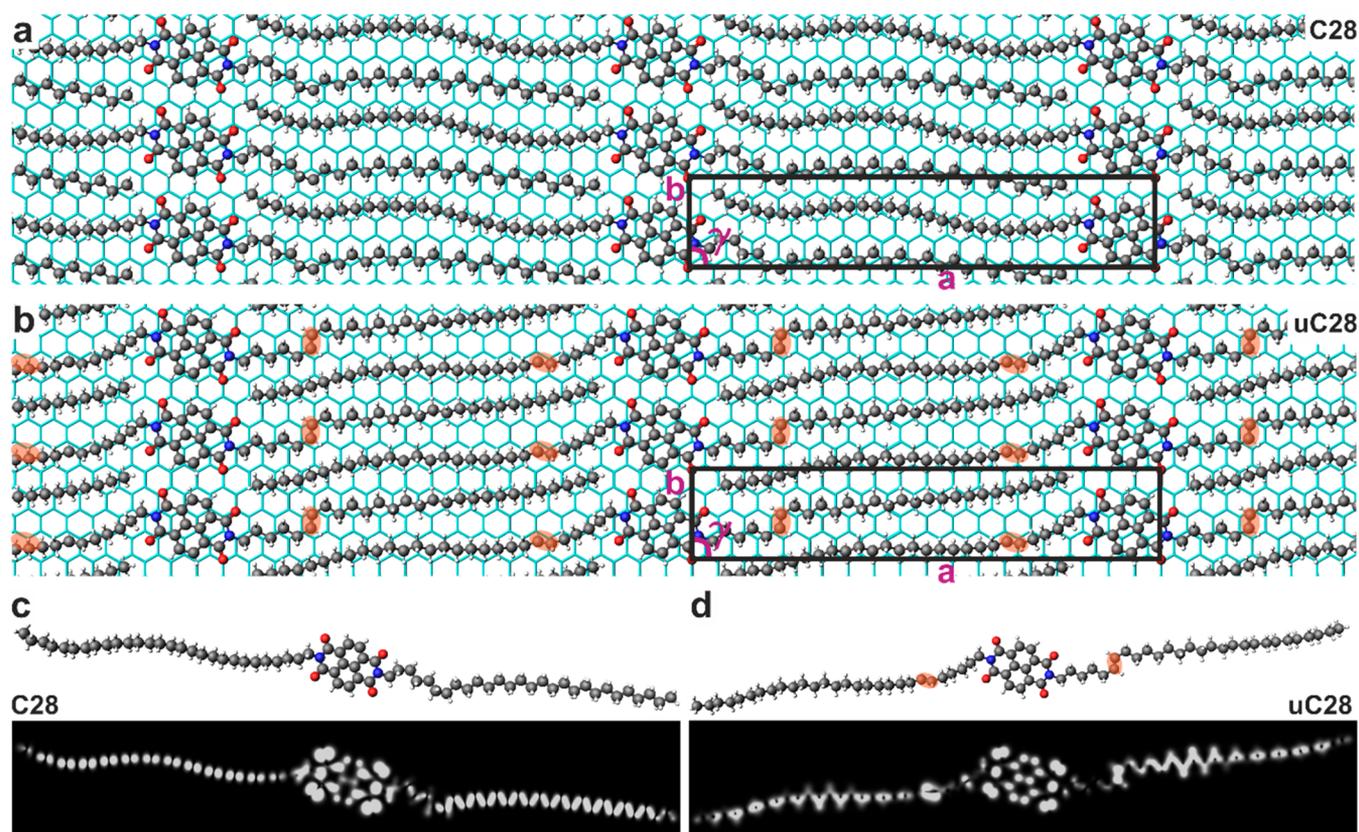


Figure 5. (a) Optimized geometries for phase A of (a) C₂₈-NDI-C₂₈ and (b) uC₂₈-NDI-uC₂₈ adsorbed on a graphene surface. The black rectangle shows the unit cells. The orange ellipses show the positions of the double bonds within the NDIs. Hydrogen, carbon, oxygen, and nitrogen atoms are shown in white, gray, red, and blue, respectively. The graphene layer is shown in cyan. The unit cell parameters *a*, *b*, and *γ* are marked in magenta. (c) Single molecule in lamellar phase A geometry (top) and simulated STM image (bottom, black and white image) at -1 V for C₂₈-NDI-C₂₈. (d) Single molecule in lamellar phase A geometry (top) and simulated STM image (bottom, black and white image) at -1 V for uC₂₈-NDI-uC₂₈.

chains interdigitate, in line with the experimental observations (Figure 5a for C₂₈-NDI-C₂₈, and Figure 5b for uC₂₈-NDI-uC₂₈). The distance between the hydrogen atoms of the aromatic C-H and the oxygen atoms of the neighboring imide moieties amounts to 2.5 Å for both C₂₈-NDI-C₂₈ and uC₂₈-NDI-uC₂₈, in line with the literature.⁴⁷ This allows for unconventional hydrogen bonding interactions between adjacent NDI cores, which stabilize the molecular arrangement. Unconventional hydrogen bonding may additionally occur between the oxygen atoms of the imide moieties and the terminal methyl groups of the interdigitating chains from the adjacent row of NDIs for both C₂₈-NDI-C₂₈ and uC₂₈-NDI-uC₂₈. The H-O distance varies in the 2.6–3 Å range in this case. The calculated unit cell values are *a* = 44.8 Å, *b* = 8.5 Å, and *γ* = 90° for C₂₈-NDI-C₂₈, and *a* = 44.5 Å, *b* = 8.5 Å, and *γ* = 90° for uC₂₈-NDI-uC₂₈, nicely matching with the experimental values (Table 1). This further confirmed the accuracy of the computational study. Finally, we compared the adsorption energies for both C₂₈-NDI-C₂₈ and uC₂₈-NDI-uC₂₈ in the lamellar phase A arrangement on graphene. Assemblies of C₂₈-NDI-C₂₈ adsorbed on graphene were 0.166 eV (3.83 kcal/mol) per molecule energetically more favorable than those of uC₂₈-NDI-uC₂₈. However, the experimental observation of improved long-range order with uC₂₈-NDI-uC₂₈ compared to C₂₈-NDI-C₂₈ and the stronger van der Waals interchain interactions between unsaturated C₆ carbon chains (Figure S41 and Table S1) clearly point to more favorable

intermolecular interactions in the case of uC₂₈-NDI-uC₂₈. Considering that the unconventional hydrogen bonding occurring in the calculated lamellar phases of C₂₈-NDI-C₂₈ and uC₂₈-NDI-uC₂₈ should be very similar, if not identical, from the energetic point of view, we can only ascribe such “more favorable intermolecular interactions” to van der Waals forces.

We simulated an STM image for an individual molecule at a bias voltage of -1 V from the calculated lamellar phases of both C₂₈-NDI-C₂₈ and uC₂₈-NDI-uC₂₈. The individual molecules, as well as the corresponding simulated STM images at bias -1 V for C₂₈-NDI-C₂₈ and uC₂₈-NDI-uC₂₈, are shown in Figure 5c,d. Both C₂₈-NDI-C₂₈ and uC₂₈-NDI-uC₂₈ showed some level of distortion from a linear geometry of the carbon chains (Figure 5c,d, top part). Interestingly, the two *E*-configured double bonds (encircled by an orange ellipse in Figure 5d) were rotated by almost 90° with respect to the imaginary line that connects the nitrogen atoms of the NDI core in uC₂₈-NDI-uC₂₈ (Figure 5d, top part). As expected, the simulated STM images of C₂₈-NDI-C₂₈ and uC₂₈-NDI-uC₂₈ are almost identical with respect to the aromatic cores (Figure 5c,d, top parts). The main difference concerns the long carbon chains, with the clear presence of the internal double bonds in uC₂₈-NDI-uC₂₈ (Figure 5d). The two internal double bonds appear as bright spots, suggesting the presence of two localized areas of higher electronic densities along the carbon chains (Figure 5d). In stark contrast, the distribution of the electronic

density along the carbon chains of C_{28} -NDI- C_{28} is more homogeneous and points to a discrete series of single bonds (Figure 5c). Consistently with the on-graphene optimized structure of uC_{28} -NDI- uC_{28} , the internal double bonds are rotated by almost 90° , also in the simulated STM image (Figure 5d). This peculiar feature may account for a different visualization of the internal double bond by STM. Such hypothesis seems to be consistent with the experimental STM images reported in Figure 2, in which one of the two double bonds appears more visible than the other one for both uC_{28} -NDI- uC_{28} and uC_{33} -NDI- uC_{33} . This difference is more evident in the case of uC_{33} -NDI- uC_{33} and it is probably due to an increased distance for the internal double bonds from the NDI core, which ultimately facilitates the imaging. Hence, the gratifying agreement between the calculations on uC_{28} -NDI- uC_{28} and the experimental STM images on both uC_{28} -NDI- uC_{28} and uC_{33} -NDI- uC_{33} allowed us to generalize the conclusions to both unsaturated molecular systems.

CONCLUSIONS

We presented the synthesis and self-assembly at the 1-PO/HOPG interface of C_{28} -NDI- C_{28} , C_{33} -NDI- C_{33} , uC_{28} -NDI- uC_{28} , and uC_{33} -NDI- uC_{33} . The molecular structures only differ by the presence/absence of precisely positioned internal double bonds in their molecular skeletons. These compounds self-assembled into lamellar arrangements characterized by parallel aromatic cores that lay flat on the surface, and aliphatic chains that modulate the distance between such cores. The longer the chain, the larger the distance, implying that the entire molecular system lays flat on the surface. The packing of the long carbon chains results into two different arrangements: one in which the tails are interdigitated (lamellar phase A), and a second one where the long tails arrange diagonally, without interdigitation (lamellar phase B). We find the presence/absence of the simple double bonds to be the critical parameter for the selection of the chain arrangements. The fully saturated compounds present a combination of both self-assembly motifs, whereas the unsaturated molecules are capable of selecting the fully interdigitated arrangement. Such difference is magnified and reflected on the long-range order of the generated monolayers, with the unsaturated compounds forming much larger domains (in some cases larger than $15\,000\text{ nm}^2$). This contrasts starkly with the *locally* ordered, yet *globally* disordered, monolayers of the saturated compounds. The experimental results were also corroborated by computational studies, which suggest stronger van der Waals interactions between unsaturated carbon chains as a possible explanation. Showing the paramount role played by internal double bonds in the self-assembly of long carbon chain derivatives on surfaces, our results point to the use of "simple" internal double bonds as a critical structural parameter for obtaining long-range order in surface-supported supramolecular processes. We envision the application of our findings toward post-functionalization of non-covalently functionalized surfaces and highly dynamic and smart functional substrates.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/jacs.0c00765>.

Synthetic details, characterization of the new isolated compounds, in situ STM imaging at the liquid/solid interface, and computational details (PDF)

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Notes

The authors declare no competing financial interest.

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