

# Ru-Doped MoS<sub>2</sub> Monolayer for Exhaled Breath Detection on Early Lung Cancer Diagnosis: A First-Principles Investigation

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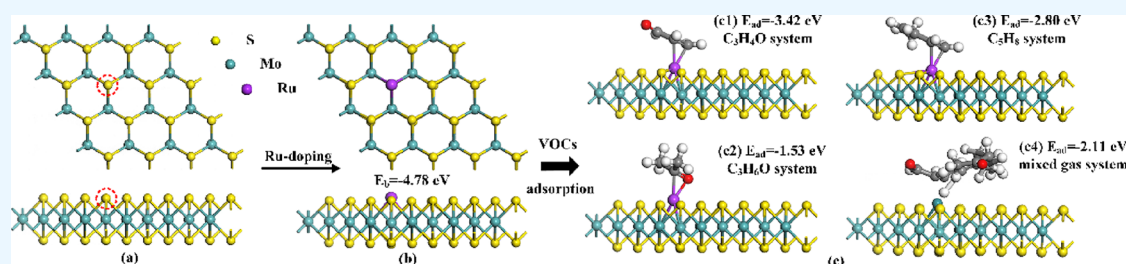
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**ABSTRACT:** Nanosensor-based patient exhaled breath detection is a practical and effective way to detect lung cancer early. In this paper, a Ru-doped MoS<sub>2</sub> monolayer (Ru-MoS<sub>2</sub>) is proposed as a promising novel biosensor based on first-principles theory for the detection of three typical early stage lung cancer exhaled volatile organic compounds, namely, C<sub>3</sub>H<sub>4</sub>O, C<sub>3</sub>H<sub>6</sub>O, and C<sub>5</sub>H<sub>8</sub>. Replacement of a S atom in the MoS<sub>2</sub> monolayer with a Ru dopant atom to form a stable Ru-MoS<sub>2</sub> monolayer with a binding energy of  $-4.78$  eV is further demonstrated by the thermostability and chemical stability analysis as well as improving the adsorption performance of the system for three VOCs. The adsorption configuration structures, adsorption properties, and electronic behavior of the Ru-MoS<sub>2</sub> monolayer are investigated by electron deformation density and density of states analysis to gain a comprehensive understanding of the physicochemical properties as sensing material. The results show that the adsorption energies of the Ru-MoS<sub>2</sub> monolayer for C<sub>3</sub>H<sub>4</sub>O, C<sub>3</sub>H<sub>6</sub>O, and C<sub>5</sub>H<sub>8</sub> are 3.42,  $-1.53$ , and  $-2.80$  eV, respectively, all of which are chemisorption with excellent adsorption performance. The sensitivities for the three VOCs could be up to 1.09, 140.50, and 5.90, respectively, and the band structure and work function further elucidate the sensing mechanism of the Ru-MoS<sub>2</sub> monolayer as a resistive gas sensor. The type and concentration of these exhaled breaths may reflect changes in the patient's physiological and biochemical status and may serve as a probe for the diagnosis of lung cancer. The results in this work could provide a guidance for researchers to explore the practical applications in the early diagnosis of lung cancer by gas sensors.

## 1. INTRODUCTION

Lung cancer is the second leading cause of death worldwide after heart disease, killing 1.6 million people each year. Histologically, lung cancer is divided into nonsmall cell lung cancer (NSCLC), which accounts for about 85% of cases, and small cell lung cancer (SCLC), which accounts for the remaining 15%. A total of 85% of lung cancer cases are diagnosed at an advanced stage, and the 5 year survival rate is only about 10–15%.<sup>1–3</sup> However, if the disease is diagnosed at stage 1, the 5 year survival rate increases dramatically to 80%. The presence and staging of disease can be confirmed using current diagnostic techniques such as blood tests, chest X-rays, computed tomography (CT), magnetic resonance imaging (MRI), and positron emission tomography (PET).<sup>4–6</sup> Genetic analysis and protein analysis are novel technologies for determining cancer risk, prognosis, and targeted (or individualized) therapy. Despite the progress made in recent years in identifying biomarkers, mutations, and genomic signatures, there are still formidable hurdles to overcome on developing

effective biomarkers, such as tumor heterogeneity, highly complex tumor–host interactions, and the complexity, diversity, and redundancy of tumor–cell signaling networks involving genetic, epigenetic, and microenvironmental effects. Furthermore, the technologies associated with these methods are often expensive, time-consuming, and unavailable in many healthcare settings and require relatively large amounts of tissue for analysis.<sup>7,8</sup> Thus, the search for diagnostic methods for early lung cancer detection is critical and urgent.

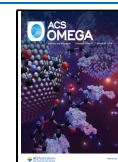
In recent years, an emerging approach through the detection of volatile organic compounds (VOCs), which can be detected in the headspace of cancer cells or blood samples or in exhaled

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breath, has come to the attention of researchers.<sup>9–11</sup> On account of the simplicity, efficiency, and low cost of the method, monitoring VOCs in breath may soon become a promising approach to traditional medical diagnostics. Tumors growing in the body have been reported to produce specific substances that release VOCs into the bloodstream and exchange with them in the lungs. Moreover, lung cancer patients have significantly higher levels of VOCs in their exhaled breath than healthy individuals, which offers the possibility of identifying possible lung cancer patients and assessing the severity of the diagnosed patients. Gordon et al.<sup>12</sup> selected a total of 22 kinds of VOCs that are considered to be characteristic levels in the exhaled breath of lung cancer patients, mainly alkanes and benzene derivatives. Among them, 2-acrylaldehyde diol (C<sub>3</sub>H<sub>4</sub>O), acetone (C<sub>3</sub>H<sub>6</sub>O), and isoprene (C<sub>5</sub>H<sub>8</sub>) are typical biomarkers that can show the severity of lung dysfunction in humans. Exhaled breath analysis tests for lung cancer using nanomaterial-based sensor arrays were reportedly proposed by Barash et al.<sup>13</sup> Chang and his colleagues<sup>3</sup> used the sensor system to analyze VOCs in exhaled breath for lung cancer diagnosis. Thus, the detection of specific VOCs in exhaled breath by highly sensitive and fast-responding chemical sensors could provide a simple, painless, and non-invasive test for early lung cancer diagnosis, which could be a promising diagnostic modality.

Due to its extraordinary physical, chemical, optical, and mechanical qualities, the MoS<sub>2</sub> monolayer, one of the transition metal dichalcogenides (TMDs), has attracted a lot of attention for applications in gas sensors lately.<sup>14,15</sup> Three atomic layers of S–Mo–S make up the MoS<sub>2</sub> monolayer, where the Mo atom is covalently bound to six S atoms to form a sandwiched substance, with a band gap of 1.9 eV.<sup>16–19</sup> However, the MoS<sub>2</sub> monolayer is chemically inert due to the absence of defects. Over the past few years, it has been demonstrated that the chemical activity and sensitivity of MoS<sub>2</sub> monolayer substrates can be greatly improved by introducing defects and appropriate metal dopants.<sup>20–22</sup> The characteristics of 2D materials, such as their increased sensitivity to tiny molecules and increased chemical activity, can be changed by incorporating the right transition metal elements. Zhao et al.<sup>23</sup> concluded that the Cu-embedded MoS<sub>2</sub> monolayer worked the best after a thorough analysis of the effects of doping MoS<sub>2</sub> for oxygen reduction reaction with 19 different types of transition elements. Fan et al.<sup>24</sup> conducted a theoretical investigation into the adsorption of CO, NO, O<sub>2</sub>, NO<sub>2</sub>, and NH<sub>3</sub> on 10 TM-doped MoS<sub>2</sub> surfaces. The results showed that the chemical activity and adsorption performance of gas molecules on the doped MoS<sub>2</sub> surfaces were significantly improved at 3.12 eV adsorption energies as compared to those of the intrinsic MoS<sub>2</sub> monolayer. Therefore, the influence of suitable transition metal dopants on the chemical and catalytic properties of the MoS<sub>2</sub> monolayer is crucial, which may have great potential for VOC sensing.

To explore gas sensing materials with high sensitivity to VOCs and then promote the development of cutting-edge sensors for the diagnosis of early lung cancer, the density functional theory-based theoretical calculations regarding the adsorption performance of Ru-doped MoS<sub>2</sub> (Ru-MoS<sub>2</sub>) toward three typical VOCs (C<sub>3</sub>H<sub>4</sub>O, C<sub>3</sub>H<sub>6</sub>O, and C<sub>5</sub>H<sub>8</sub>) associated with lung cancer have been conducted in this work, and the feasibility of the Ru-MoS<sub>2</sub> monolayer as resistive gas sensing material for VOC detection has been discussed theoretically. The Ru-MoS<sub>2</sub> structure is produced by substituting a S atom

from the MoS<sub>2</sub> monolayer for the Ru atom, which has demonstrated excellent catalytic performance for gas inter-actions. To thoroughly examine its property as a chemical sensor, the adsorption configuration structures, adsorption properties, electronic behavior, and the sensing mechanism of the Ru-MoS<sub>2</sub> monolayer are further investigated. Our findings demonstrate that the doping of Ru could greatly enhance the adsorption performance of MoS<sub>2</sub> on VOCs and the potential of the Ru-MoS<sub>2</sub> monolayer as a viable sensor for VOC detection, opening up to further research into its application in the diagnosis of lung cancer. Our calculations provide a theoretical basis for future practical applications of gas biosensors in the field of lung cancer prognosis.

## 2. COMPUTATIONAL DETAILS

The density functional theory method was used to implement the entire calculation in the Dmol<sup>3</sup> program. To handle electron exchange and correlation, the Perdew–Burke–Ernzerhof (PBE) function with generalized gradient approximation (GGA) was used.<sup>25,26</sup> To address the relativistic effect of the Ru atom, we selected the DFT semicore pseudopotential (DSSP) approach and double numerical plus polarization (DNP) as the atomic orbital basis set. The Tkatchenko and Scheffler (TS) approach was used to gain a deeper comprehension of the van der Waals interaction.<sup>27,28</sup> For geometric optimization and electronic structure computations, the Monkhorst–Pack *k*-point mesh of 10 × 10 × 1 was taken for the Brillouin zone integration. We decided on 10<sup>−4</sup> Ha for the energy tolerance accuracy, 2 × 10<sup>−2</sup> Ha/Å for the maximum force, and 5 × 10<sup>−2</sup> Å for the displacement.<sup>29,30</sup> To provide correct findings for total energy estimates in static electronic structure calculations, a 10<sup>−6</sup> Ha self-consistent loop energy, a global orbital cutoff radius of 5.0 Å, and a smearing of 0.005 Ha were used. Each and every calculation was spin-polarized.<sup>31</sup>

A 4 × 4 × 1 MoS<sub>2</sub> monolayer supercell with a 20 Å vacuum area is created that contains 16 Mo and 32 S atoms. For additional adsorptions, it is then loosened to its most stable state. The optimized lattice parameter of the MoS<sub>2</sub> monolayer was calculated as 3.18 Å, which is fairly close to the previous report. The following formula is used to determine each gas adsorption process's adsorption energy ( $E_{\text{ad}}$ ).

$$E_{\text{ad}} = E_{\text{Ru-MoS}_2/\text{gas}} - E_{\text{Ru-MoS}_2} - E_{\text{gas}} \quad (1)$$

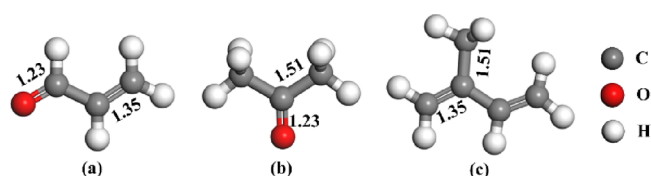
wherein  $E_{\text{Ru-MoS}_2/\text{gas}}$ ,  $E_{\text{Ru-MoS}_2}$ , and  $E_{\text{gas}}$  were the energies of the Ru-MoS<sub>2</sub> surface after gas molecule adsorption, isolated Ru-MoS<sub>2</sub>, and the individual gas molecule, respectively. An exothermic, spontaneous adsorption occurs if  $E_{\text{ad}} < 0$ ; the more dramatic the adsorption behavior, the larger the absolute value. Alternatively, the interaction cannot occur if  $E_{\text{ad}}$  is less than 0. The charge transfer ( $Q_{\text{T}}$ ) between the target molecule and the adsorbent surface, which is defined by the electron value carried by the gas molecule following adsorption, was examined by using the Hirshfeld method. The converge criterion and cutoff parameter mentioned above have been used to optimize the various structures. The initial distances between the gas molecules and the transition metal are all set at 2.0 Å.

## 3. RESULTS AND DISCUSSION

### 3.1. Analysis of the Ru-MoS<sub>2</sub> Monolayer and Gas Molecules.

Three distinct exhaled volatile organic compounds

(VOCs) of lung cancer patients are plotted in Figure 1. Calculations reveal that the bond lengths and bond angles of



**Figure 1.** Geometric structure of (a)  $C_3H_4O$ , (b)  $C_3H_6O$ , and (c)  $C_5H_8$ .

these gas molecules are almost identical to those reported previously,<sup>32</sup> which further illustrates the validity of VOC models. Therefore, a detailed analysis is not given here.

The electron deformation density (EDD) of the Ru-MoS<sub>2</sub> monolayer and the most stable model of the MoS<sub>2</sub> monolayer before and after Ru atom doping are depicted in Figure 2. Since monosulfur vacancies are the most prevalent point defect structure and have a lower formation energy than Mo vacancies, the Ru atom is selected to replace the S atoms in MoS<sub>2</sub> to generate the Ru-MoS<sub>2</sub> monolayer. Ru doping causes the MoS<sub>2</sub> monolayer to undergo certain notable deformations. Upon measurement, Ru is measured to form Ru–Mo bonds with three neighboring Mo atoms with the bond length of 2.647 Å, which is slightly longer than the original Ru–Mo bond on account of the larger atomic radii of the Ru atom in comparison to the S atom. The binding force ( $E_b$ ) is determined to be  $-4.78$  eV with Ru doping on the MoS<sub>2</sub> surface, indicating a significant contact between the Ru dopant and the MoS<sub>2</sub> monolayer.

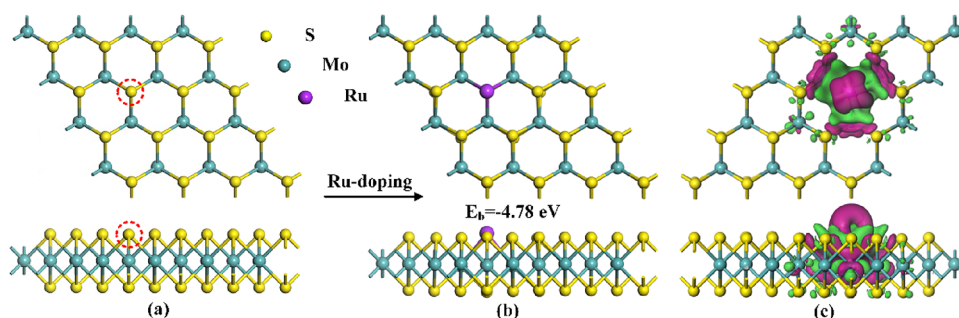
The Hirshfeld method indicates that the Ru dopant contributes 0.097 e to the MoS<sub>2</sub> surface during doping, referring to its electron-donating property on the MoS<sub>2</sub> surface. The surrounding Mo atoms are surrounded by electron accumulation, whereas the Ru atom is surrounded by electron depletion, as can be seen from the EDD distribution. The ionic character of the Ru–Mo bonds and their strong orbital interaction during Ru doping are confirmed by the obvious electron overlaps on them. Moreover, with the goal of determining if the Ru-MoS<sub>2</sub> monolayer can be stabilized at 500 K and considering the potential for their development for gas-sensitive medical devices, the thermostability is confirmed using molecular dynamic simulation for 1 ps, with a step of 1 fs. It can be found that the morphology of the Ru-MoS<sub>2</sub> monolayer is in a stable configuration after 1000 stem simulations at 500 K. Consequently, it is concluded that the

Ru-MoS<sub>2</sub> monolayer exhibits high thermostability. Additionally, the vibrational analysis reveals that the Ru-MoS<sub>2</sub> monolayer has frequencies that range from 59.8 to 455.2  $cm^{-1}$ , indicating high chemical stability in the absence of an imaginary frequency.

To further comprehend the electron property of Ru doping onto the pure MoS<sub>2</sub> monolayer, Figure 3 plots the density of state (DOS) of the Ru-MoS<sub>2</sub> system in comparison to pristine MoS<sub>2</sub> and the orbital DOS of the Ru and S atoms. As compared to pure MoS<sub>2</sub>, the result demonstrates a considerable leftward shift in the DOS curves of Ru-MoS<sub>2</sub>, especially near the Fermi energy level, where it exhibits a new peak known as the impurity state, which can be attributed to the contribution of Ru doping and its hybridization states with S and Mo atoms. In addition, the new state introduced by the Ru dopant improves the chemical reactivity and electronic activity of the Ru-MoS<sub>2</sub> monolayer to some extent. Strong orbital interactions between Ru and Mo atoms in Figure 3b are implied by the observation that the Ru 4d orbital is highly hybridized with the Mo 4d orbital from  $-6.25$  to  $2.0$  eV and  $-12.0$  to  $-13.2$  eV, which is consistent with the EDD distribution in which the Ru–Mo bond is surrounded by electron accumulation.

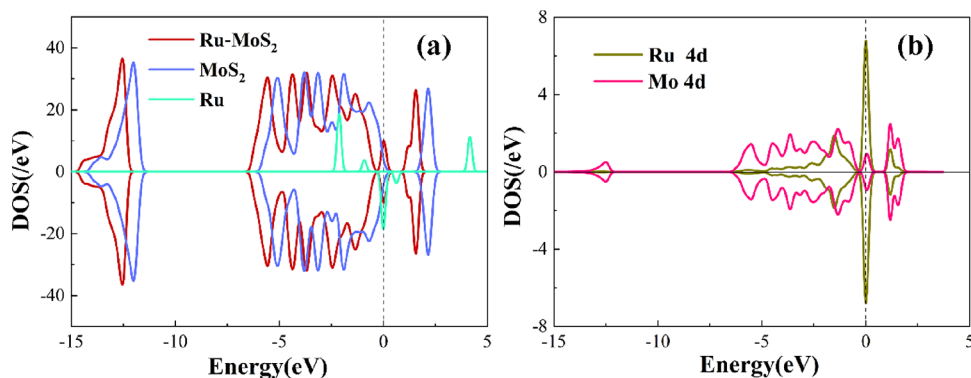
**3.2. Gas Adsorption Configurations on the Ru-MoS<sub>2</sub> Monolayer.** Adsorptions of  $C_3H_4O$ ,  $C_3H_6O$ , and  $C_5H_8$  are carried out on the surface of the Ru-MoS<sub>2</sub> monolayer with its optimal structure. Various adsorption configurations are implemented with an optimum initial adsorption distance of 2.0 Å to obtain the most stable configuration. The geometric structures and associated EDD for the three VOC systems are shown in Figure 4.

When  $C_3H_4O$  is adsorbed on the Ru-MoS<sub>2</sub> surface at an angle, it is seen that the  $C_3H_4O$  molecule prefers to stand vertically on top of the Ru dopant on a slope, trapped by the Ru atom to form two new Ru–C bonds with measured diameters of 2.211 and 2.146 Å. This result implies that the binding force between Ru and C atoms is greater than the binding force between Ru and O atoms. Following  $C_3H_4O$  adsorption, the Ru–Mo bond exhibits a substantial deformation, suggesting that the Ru-MoS<sub>2</sub> monolayer is geometrically activated during gas interaction. The  $E_{ad}$  of  $-3.42$  eV in this system is categorized as chemisorption. However, this chemisorption is much stronger than that in the pristine MoS<sub>2</sub> system, in which an  $E_{ad}$  of  $-1.57$  eV is obtained as calculated, indicating that Ru doping effectively enhances the adsorption performance of the MoS<sub>2</sub> monolayer on  $C_3H_4O$ . Apart from that, compared with the previous report, the Ru-MoS<sub>2</sub> monolayer in this study shows higher adsorption energy

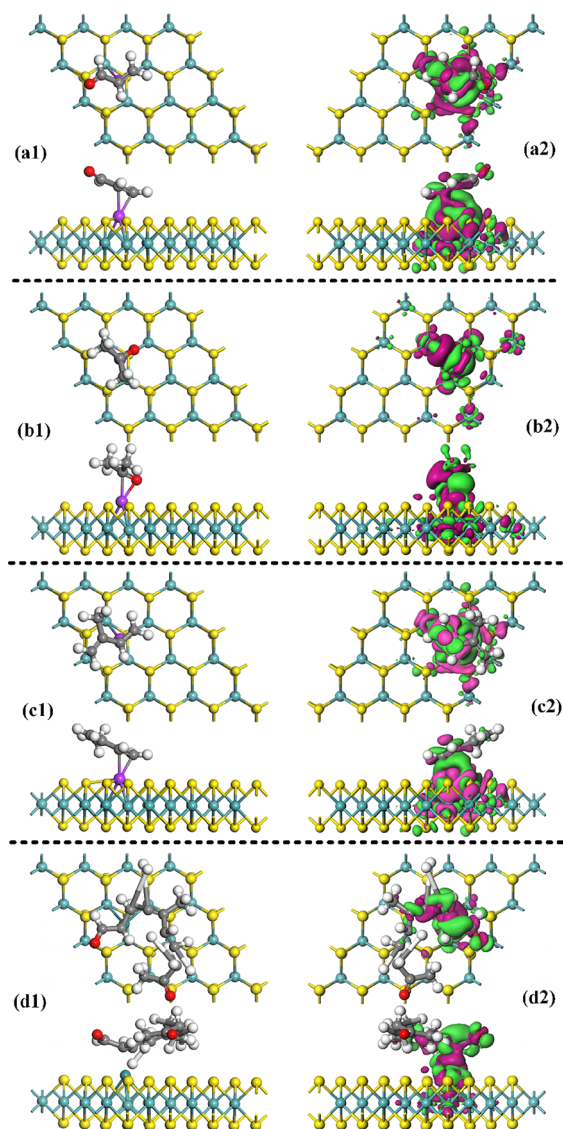


**Figure 2.** Ru doping process on the MoS<sub>2</sub> monolayer. Morphology of (a) the intrinsic MoS<sub>2</sub> monolayer, (b) Ru-MoS<sub>2</sub> monolayer, and (c) EDD of Ru-MoS<sub>2</sub> monolayer. In EDD, the green (rosy) areas indicate electron accumulation (depletion) and the isosurface is set as 0.01 eV/Å<sup>3</sup>.





**Figure 3.** (a) DOS comparison before and after doping Ru and (b) orbital DOS in the Ru-MoS<sub>2</sub> system. The dashed line in DOS is the Fermi level.



**Figure 4.** Optimized structure and EDD of the C<sub>3</sub>H<sub>4</sub>O system (a1, a2), C<sub>3</sub>H<sub>6</sub>O system (b1, b2), C<sub>5</sub>H<sub>8</sub> system (c1, c2), and mixed gas adsorption system (d1, d2). In EDD, the green (rosy) areas indicate electron accumulation (depletion).

for C<sub>3</sub>H<sub>4</sub>O than Ni-doped MoS<sub>2</sub> (−1.749 eV), indicating the superiority of Ru element doping.<sup>17</sup> The Hirshfeld study also shows that the Ru atom is assigned a positive charge of 0.130 e, and the C<sub>3</sub>H<sub>4</sub>O molecule as a whole has a negative charge of

0.225 e. These findings suggest that the Ru dopant (0.130 e) and Ru-MoS<sub>2</sub> monolayer (0.095 e) are responsible for 0.225 e that C<sub>3</sub>H<sub>4</sub>O accepts, which indicates that the Ru-MoS<sub>2</sub> surface is electron-donating, while the C<sub>3</sub>H<sub>4</sub>O molecule is electron-absorbing. The Ru dopant is surrounded by electron depletion in the EDD distribution, whereas the Ru–C and Ru–O bonds are surrounded by electron accumulation, which confirms the electron-releasing behavior of the Ru atoms and the electron hybridization on the newly formed bonds.

It is discovered that following C<sub>3</sub>H<sub>6</sub>O adsorption onto the Ru-MoS<sub>2</sub> monolayer, the C<sub>3</sub>H<sub>6</sub>O molecule is parallel to the Ru-MoS<sub>2</sub> surface and exhibits minimal geometric distortion, which is not as drastic as those in the C<sub>3</sub>H<sub>4</sub>O system, suggesting that the interactions in the C<sub>3</sub>H<sub>6</sub>O system are weaker. The Ru dopant captures a C atom and an O atom from the C<sub>3</sub>H<sub>6</sub>O molecule, forming new Ru–C and Ru–O bonds that measure 2.179 and 2.015 Å, respectively. Furthermore, the computed  $E_{ad}$  in this instance is −1.53 eV, which is less than that of C<sub>3</sub>H<sub>4</sub>O, indicating chemisorption. Nevertheless, as a result of Ru doping, the adsorption performance of the Ru-MoS<sub>2</sub> monolayer on C<sub>3</sub>H<sub>6</sub>O is noticeably improved compared to that of pristine MoS<sub>2</sub> (−0.20 eV), which indicates that the Ru dopant boosts the redistribution of the electrons and provides more adsorption sites. As can be seen from EDD, the adsorbed C<sub>3</sub>H<sub>6</sub>O molecule has a negative charge of 0.133 e, showing a relatively weak electron gain. Electron depletion is mainly concentrated on the keto group, while electron accumulation occurs mainly on the Ru–O and Ru–C bonds, which confirms the strong bonding between the Ru dopant and the O and C atoms.

In terms of the C<sub>5</sub>H<sub>8</sub> adsorption system, the optimized configuration is comparable to that of the C<sub>3</sub>H<sub>4</sub>O system, wherein the two C atoms in the C=C bond form new bonds with the Ru dopant with bond lengths of 2.145 and 2.212 Å, respectively, and the C<sub>5</sub>H<sub>8</sub> molecule is tilted in the Ru-MoS<sub>2</sub> plane. These results imply that the Ru dopant is highly chemically reactive to the C=C bond. The  $E_{ad}$  of this system is −2.80 eV, second only to that of the C<sub>3</sub>H<sub>4</sub>O system, and its adsorption efficiency surpasses that of the MoS<sub>2</sub> system (−1.36 eV). Furthermore, the adsorption energy is still larger than that of the previously reported Al-MoSe<sub>2</sub> monolayer toward C<sub>5</sub>H<sub>8</sub> (−2.00 eV), which exhibits excellent adsorption properties of the Ru-MoS<sub>2</sub> monolayer.<sup>25</sup> According to the Hirshfeld analysis, the C<sub>5</sub>H<sub>8</sub> molecule obtained a transfer of 0.120 e from the Ru-MoS<sub>2</sub> surface, while the Ru atom loses 0.11 e during this adsorption process. As can be seen from EDD, the charge source is mainly from the Ru atom of the Ru-MoS<sub>2</sub> monolayer,

which is transferred to the  $C_3H_8$  molecule mainly through two Ru–C bonds. These results clarify the charge-transfer mechanism, the synthesis of the chemical Ru–C bonds, and the localization of charge.

Apart from this, the simultaneous adsorption of the three VOCs on the Ru-MoS<sub>2</sub> surface is further explored. As can be seen from Figure 4d1, after the three gases are adsorbed, the shape of the gas molecules undergoes obvious deformation and some of the chemical bonds are elongated, indicating that a strong adsorption reaction is generated. The calculations show that the adsorption energy of the Ru-MoS<sub>2</sub> monolayer for the three gas mixtures is  $-2.11$  eV, and the Ru atom behaves as an electron contributor, losing  $0.0819$  e. Since the charge transfer of the adsorption process of the Ru-MoS<sub>2</sub> monolayer for a single gas is in the same direction, the adsorption of the Ru-MoS<sub>2</sub> monolayer for the gas mixtures is synergistic. Figure 4d2 demonstrates EDD for the adsorption of mixed gases by the Ru-MoS<sub>2</sub> monolayer, from which the direction of charge transfer can be clearly seen. It can be found that a large amount of charge transfer mainly occurs near the Ru atom, indicating that the doping of the Ru atom promotes electron redistribution and increases the adsorption sites of gases.

In summary, the adsorption of the Ru-MoS<sub>2</sub> monolayer on the three VOCs is chemisorption, and the adsorption capacity is ranked as  $C_3H_4O > C_3H_8 > C_3H_6O$ . Compared with other studies on the adsorption of VOCs by gas sensing materials (Table 1), the Ru-MoS<sub>2</sub> monolayer proposed in this study has

**Table 1. Adsorption Performance of Various Gas Sensors toward VOCs**

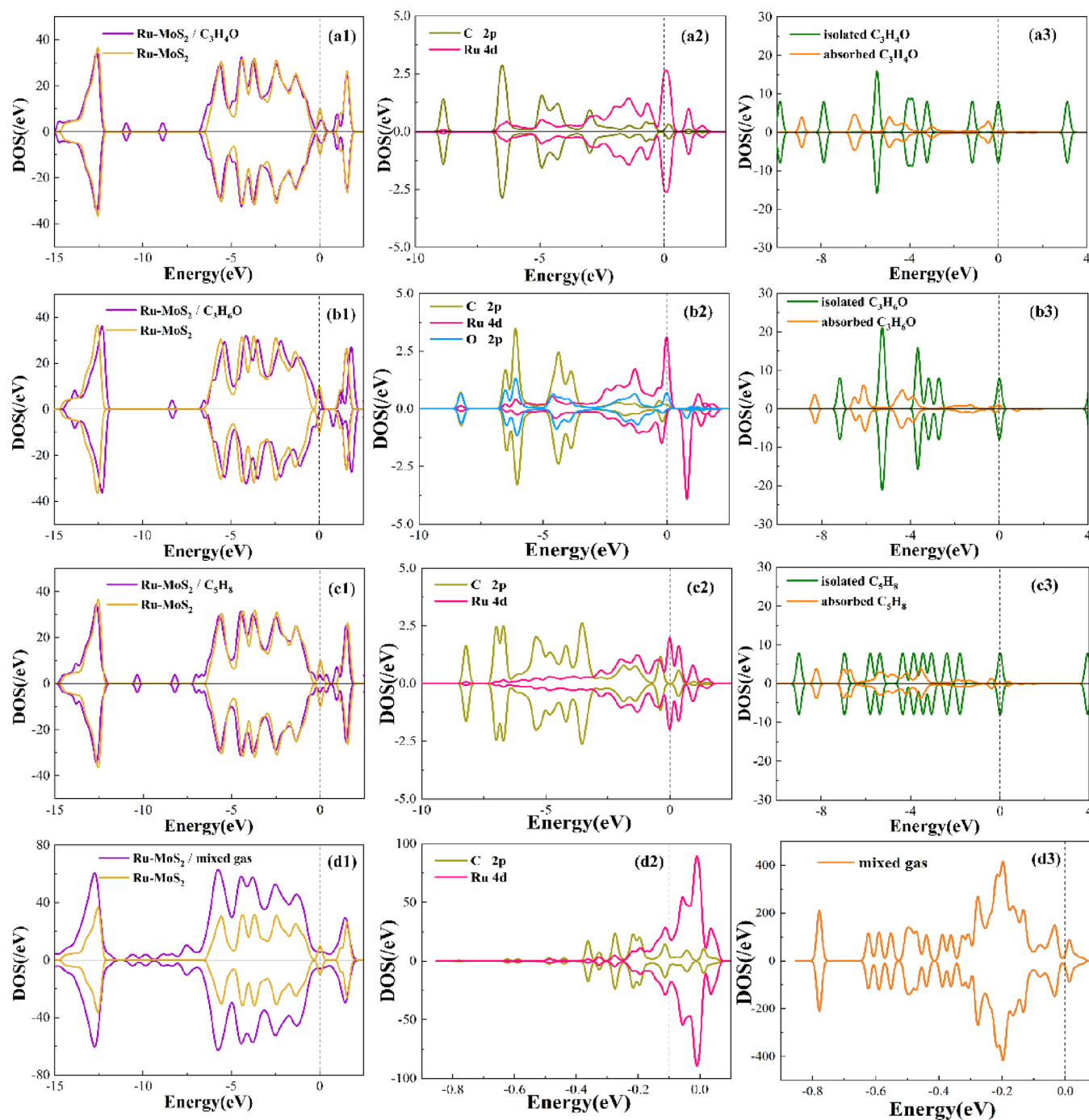
gas sensing material	target gas	$E_{ad}$ (eV)
Ni-MoS <sub>2</sub> <sup>17</sup>	$C_3H_8$	$-1.84$
	$C_3H_4O$	$-1.75$
	$C_6H_6$	$-1.52$
Al-MoSe <sub>2</sub> <sup>25</sup>	$C_3H_4O$	$-1.45$
	$C_3H_6O$	$-1.80$
	$C_3H_8$	$-2.00$
Ru-PtTe <sub>2</sub> <sup>32</sup>	$C_3H_4O$	$-1.72$
	$C_3H_6O$	$-1.12$
	$C_3H_8$	$-1.80$
Au-SnS <sub>2</sub> <sup>26</sup>	$C_3H_6O$	$-0.77$
	$C_3H_8$	$-1.31$
	$C_6H_6$	$-0.71$
Pt-SnS <sub>2</sub> <sup>27</sup>	$C_2H_4$	$-0.53$
	$C_2H_6$	$-0.31$
	$C_6H_6$	$-0.60$
Ru-MoS <sub>2</sub> in this work	$C_3H_4O$	$-3.42$
	$C_3H_6O$	$-1.53$
	$C_3H_8$	$-2.80$

significant advantages in adsorption performance. Meanwhile, the doping of the Ru atom promotes the charge transfer-induced electron redistribution, which dramatically improves the adsorption performance of the MoS<sub>2</sub> monolayer on the three gases.

**3.3. Electronic Property of the Ru-MoS<sub>2</sub> Monolayer upon Gas Adsorption.** For the purpose of understanding the electronic behavior of the Ru-MoS<sub>2</sub> monolayer following gas adsorption, the DOS distributions of various VOC systems are shown in Figure 5. The findings demonstrated that there are varying degrees of change in the overall DOS distributions of the Ru-MoS<sub>2</sub> monolayer-adsorbed target molecules. The

electronic behavior is evident in the total DOS distribution for the  $C_3H_4O$  molecular adsorption system, as depicted in Figure 5a1. It can be noticed that the peaks at the Fermi energy level become weaker, while several new peaks appear near  $-11.2$ ,  $-8.9$ , and  $1.8$  eV compared to the total DOS of the Ru-MoS<sub>2</sub> monolayer alone. It is hypothesized that these deformations are caused by the interaction between Ru 4d and C 2p. The DOS peak of Ru 4d in the  $C_3H_4O$  system is noticeably less than the partial DOS in the Ru-MoS<sub>2</sub> system in Figure 3, which can further support the above findings. Furthermore, following the adsorption of  $C_3H_4O$ , a change in DOS is caused by the strong hybridization of Ru 4d with C 2p between  $-8.9$  and  $1.8$  eV. The electronic behavior of the  $C_3H_8$  adsorption system, as depicted in Figure 5c1, is comparable to that of the  $C_3H_4O$  system. However, fewer splitting peaks and recombination peaks of the total DOS are produced, further demonstrating a marginally weaker chemical interaction between the  $C_3H_8$  molecule and the adsorbent surface. In the  $C_3H_6O$  system, the overall DOS curve after adsorption of the target gas is shifted to the right, which is attributed to the electron-contributing property of the Ru-MoS<sub>2</sub> surface. During the gas adsorption process, electrons are transferred from the Ru-MoS<sub>2</sub> surface to the gas molecule, which reduces the carrier density of the surface and thus reduces the conductivity of Ru-MoS<sub>2</sub>. In addition, the highly activated C 2p and O 2p orbitals of  $C_3H_6O$  are responsible for the notable enhancement of the Fermi energy level peaks and the new states at  $-8.3$  and  $-6.5$  eV, which show a major shift in its electronic behavior. Moreover, the overlapping region in the partial DOS at  $-6.8$  to  $2.0$  eV suggests certain orbital hybridizations between the C and O atoms of the  $C_3H_6O$  molecule and Ru-MoS<sub>2</sub> monolayer. Figure 5d1–d3 demonstrates the related DOS of the Ru-MoS<sub>2</sub> monolayer adsorbing three gas mixtures, from which the changes in the electronic properties of the Ru-MoS<sub>2</sub> monolayer upon adsorption of the gas mixtures can be seen. As can be seen from Figure 5d1, the adsorption performance of the Ru-MoS<sub>2</sub> monolayer toward the three gas mixtures results in a great enhancement of the electronic density of states of the Ru-MoS<sub>2</sub> monolayer, while the general profile does not change, indicating that the effect of the gas mixtures on the Ru-MoS<sub>2</sub> monolayer is synergistic and isotropic and that it does not change the properties of the Ru-MoS<sub>2</sub> monolayer. Figure 5d2 shows the partial DOS of the Ru atom and C atom that interact most with the Ru atom. The comparison of Figure 5a2–c2 reveals that the Fermi energy levels of the Ru atom are greatly enhanced by adsorption of mixed gases, which is due to the synergistic effect of the three gases on the charge transfer of the Ru-MoS<sub>2</sub> monolayer, and the result is in perfect agreement with Figure 5d1 and the EDD of mixed gas adsorption.

**3.4. Band Structure (BS) Analysis.** The electrical conductivity of sensing materials through band structure calculations can be predicted. Thus, the altered Ru-MoS<sub>2</sub> monolayer band gap following gas adsorption illustrates the change in its electrical conductivity in the presence of a specific gas. Increased conductivity is correlated with a narrower band gap and reduced conductivity with a broadened band gap. The band structures of the isolated Ru-MoS<sub>2</sub> monolayer, each VOC, and the mixed gas adsorption system are shown in Figure 6. It can be seen that the band gap of the Ru-MoS<sub>2</sub> monolayer ( $0.082$  eV) is much smaller than that of the pristine MoS<sub>2</sub> monolayer of  $1.9$  eV, indicating the substantially enhanced conductivity of the Ru-MoS<sub>2</sub> monolayer. The band gaps of  $C_3H_4O$ ,  $C_3H_6O$ ,  $C_3H_8$ , and mixed gas adsorption



**Figure 5.** DOS of (a1–a3) the  $C_3H_4O$  system, (b1–b3)  $C_3H_6O$  system, (c1–c3)  $C_5H_8$  system, and (d1–d3) mixed gas adsorption system. The dashed line is the Fermi level.

systems are 0.184, 0.767, 0.348, and 0.314 eV, respectively, which increased to varying degrees compared to the band gaps before adsorption, making it harder for electrons to be excited and decreasing the overall conductivity. The band gap change of the mixed gas adsorption system is the result of the combined action of three VOCs. It is discovered that, in contrast to the order of  $E_{ad}$  ( $C_3H_6O < C_5H_8 < C_3H_4O$ ), the rising order of the band gaps in the three systems is  $C_3H_6O > C_5H_8 > C_3H_4O$ . The stronger Ru–C bond than the weaker Ru–O bond is thought to be the potential cause of the greater  $E_{ad}$ . Furthermore, the  $C_5H_8$  molecule is larger in size than the  $C_3H_4O$  molecule, resulting in a lower  $E_{ad}$  due to the fact that it

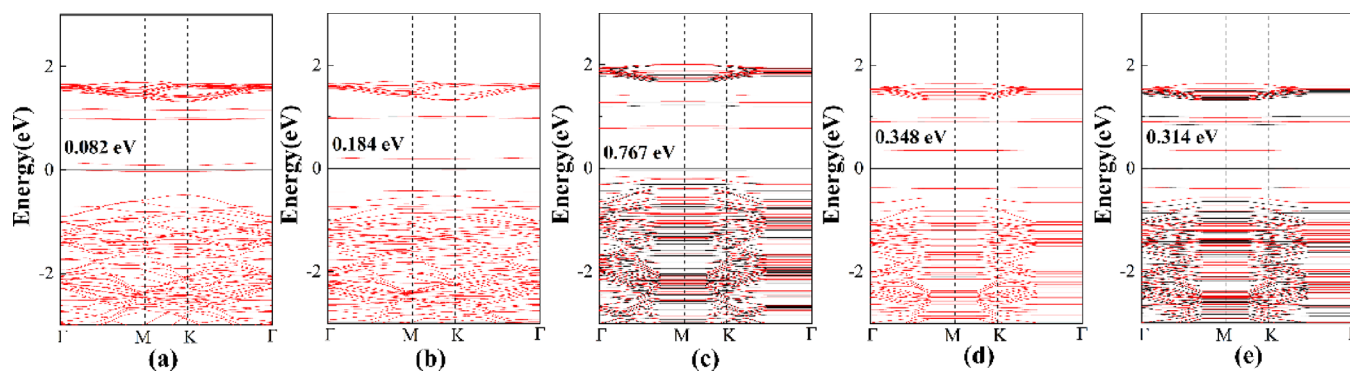
is easier to be trapped to produce the Ru–C bond. Thus, the band gap shift in the gas adsorption system might not be directly proportional to the  $E_{ad}$  value.

The conductivity ( $\sigma$ ) and sensitivity ( $S$ ) are crucial factors for resistive gas sensors. The characteristic gas species to be detected is assessed using the sensed response, which is based on the resistance changes before and after gas adsorption. The formulas for the calculation are given below:<sup>33,34</sup>

$$\sigma = A \cdot e^{(-E_g/2kT)} \quad (2)$$

$$S = (\sigma_{gas}^{-1} - \sigma_{pure}^{-1}) / \sigma_{pure}^{-1} = (\sigma_{pure} - \sigma_{gas}) / \sigma_{gas} \quad (3)$$

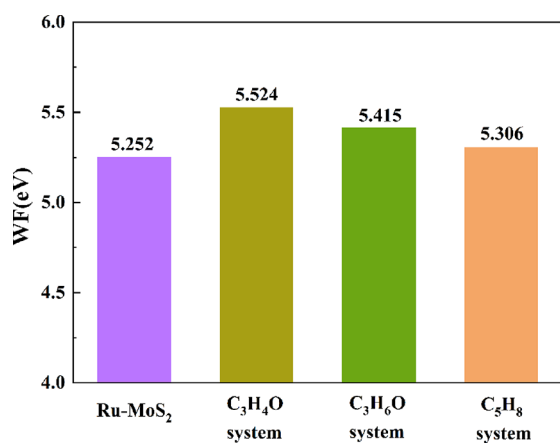




**Figure 6.** Band structure of the isolated Ru-MoS<sub>2</sub> monolayer and adsorbed systems. (a) Pristine Ru-MoS<sub>2</sub> monolayer; (b) C<sub>3</sub>H<sub>4</sub>O system; (c) C<sub>3</sub>H<sub>6</sub>O system; (d) C<sub>3</sub>H<sub>8</sub> system; (e) mixed gas system. The black values are the band gap of a related system.

where  $A$ ,  $B_g$ ,  $k$ , and  $T$  mean the certain constant, band gap, Boltzmann constant, and working temperature in formula 2, respectively. In formula 3,  $\sigma_{\text{gas}}$  and  $\sigma_{\text{pure}}$  represent the conductivity of the adsorption system and isolated Ru-MoS<sub>2</sub> monolayer, respectively. The sensitivity reflects the reaction degree of the resistive gas sensor to the measured gas by monitoring the change of resistance. The computed findings demonstrate the sensitivity of the Ru-MoS<sub>2</sub> monolayer for C<sub>3</sub>H<sub>4</sub>O, C<sub>3</sub>H<sub>6</sub>O, and C<sub>5</sub>H<sub>8</sub> to be 1.09, 140.50, and 5.90 at room temperature (298 K), respectively, which shows that the gas sensing material is quite sensitive to the target gas, and the resistance change can be intuitively generated. These results demonstrate the superior gas-sensitive performance for these gases, particularly the comparatively high sensitivity for C<sub>3</sub>H<sub>6</sub>O. The significant changes in the conductivity of Ru-MoS<sub>2</sub> monolayers caused by gas adsorption are sufficient for their detection. Therefore, the Ru-MoS<sub>2</sub> monolayer will be quite promising for clinical lung cancer diagnostic applications as a novel resistive sensing material for detecting the exhaled characteristic gases of lung cancer, C<sub>3</sub>H<sub>4</sub>O, C<sub>3</sub>H<sub>6</sub>O, and C<sub>5</sub>H<sub>8</sub>.

**3.5. Work Function (WF) Analysis.** To further understand the adsorption behavior of the doped system for each VOC gas, the work function ( $\Phi$ ) of each adsorption system is calculated, as shown in Figure 7. The variation of the work function is an important parameter for studying the gas-sensitive properties of materials, as is the variation of the electronic structure. The work function is the minimum energy required to move an electron from the Fermi energy level to



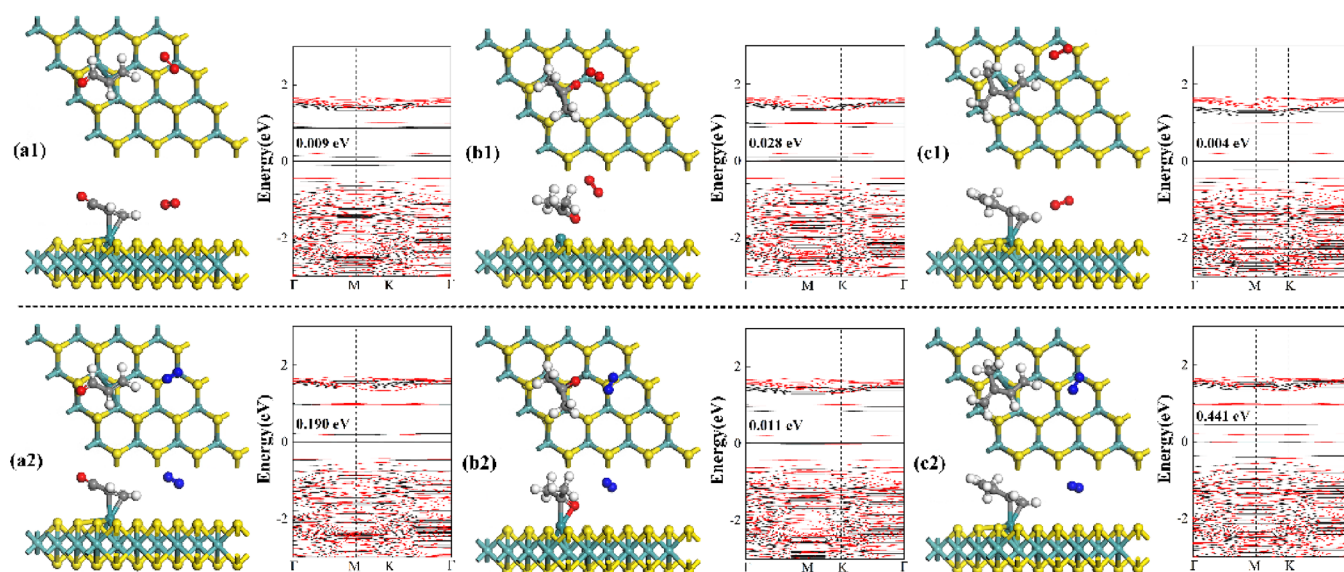
**Figure 7.** Work function of the isolated Ru-MoS<sub>2</sub> monolayer and various adsorption systems.

infinity, and the variation of the work function directly affects the conductivity of the material, which can be calculated as<sup>35,36</sup>

$$\Phi = V_{\text{vac}} - E_f \quad (4)$$

where  $\Phi$  is the work function,  $V_{\text{vac}}$  is the electrostatic potential at the vacuum level, and  $E_f$  is the Fermi energy. The work function for the Ru-MoS<sub>2</sub> monolayer is calculated to be 5.252 eV, and the work function after adsorption of C<sub>3</sub>H<sub>4</sub>O, C<sub>3</sub>H<sub>6</sub>O, and C<sub>5</sub>H<sub>8</sub> has varying increases to 5.524, 5.415, and 5.306 eV, respectively. Due to the transfer of charge, the Fermi energy level and work function change accordingly, and when the electrons are drawn out, the Fermi energy level of the system drops while the work function rises. The  $Q_T$  values of C<sub>3</sub>H<sub>4</sub>O, C<sub>3</sub>H<sub>6</sub>O, and C<sub>5</sub>H<sub>8</sub> adsorption systems are  $-0.225 e$ ,  $-0.133 e$ , and  $-0.120 e$ , respectively, which are consistent with the change of the work function.

In view of the fact that the gas sensor will be exposed to the atmosphere in the practical application, the influence of O<sub>2</sub> and N<sub>2</sub> on the adsorption of VOCs by the Ru-MoS<sub>2</sub> monolayer is considered in this work. The gas mixture adsorption configurations of VOCs and O<sub>2</sub> (or N<sub>2</sub>) molecule are carried out and BSs are examined to understand their impacts in this study, as shown in Figure 8. It can be seen that the presence of an O<sub>2</sub> molecule considerably weakens the band gap of the adsorption system, whereas the presence of N<sub>2</sub> molecules has almost no effect on the band gap of the Ru-MoS<sub>2</sub> monolayer for the gas mixture adsorption systems of C<sub>3</sub>H<sub>4</sub>O/N<sub>2</sub> and C<sub>5</sub>H<sub>8</sub>/N<sub>2</sub>, with the exception of C<sub>3</sub>H<sub>6</sub>O/N<sub>2</sub>. It is also worth noting that in the three adsorption systems containing O<sub>2</sub>, the band gap is almost zero and the systems exhibit metallic properties. The presence of the O<sub>2</sub> molecule will promote strong n-type doping in the mixed adsorption system of O<sub>2</sub> and VOCs, and the electrons will be released back into the conduction band of the Ru-MoS<sub>2</sub> monolayer, leading to a reduction in the thickness of the electron depletion region and an increase in the overall conductivity of the Ru-MoS<sub>2</sub> monolayer, which will have a contributing effect on the sensing response for gas detection. Based on the above analysis, it can be seen that atmospheric O<sub>2</sub> and N<sub>2</sub> will not hinder the adsorption of the three VOCs on the Ru-MoS<sub>2</sub> monolayer. Therefore, it can be hypothesized that the Ru-MoS<sub>2</sub> monolayer is expected to be used as a resistive gas sensor for the detection of exhaled breath for early lung cancer screening.



**Figure 8.** Gas mixture adsorption configuration and BS. (a1)  $C_3H_4O/O_2$  system; (a2)  $C_3H_4O/N_2$  system; (b1)  $C_3H_6O/O_2$  system; (b2)  $C_3H_6O/N_2$  system; (c1)  $C_5H_8/O_2$  system; (c2)  $C_5H_8/N_2$  system.

#### 4. CONCLUSIONS

Based on first principles, the gas sensing properties of the transition metal Ru-doped  $MoS_2$  monolayer for  $C_3H_4O$ ,  $C_3H_6O$  and  $C_5H_8$  gases have been systematically investigated to explore their potential as a resistive gas sensor for the early diagnosis of lung cancer. In this study, the adsorption energy, charge transfer, electronic behavior, band structure, and work function of the adsorption system are calculated. In addition, the sensitivity of the Ru- $MoS_2$  monolayer to each gas is investigated to fully understand its application characteristic properties as a gas sensor. The following are the primary conclusions:

(1) A Ru dopant replacement of a S atom forms a stabilized Ru- $MoS_2$  monolayer with a binding energy of  $-4.78$  eV, and the thermostability and chemical stability are further confirmed. In addition, the doping of Ru could improve the adsorption performance for the three VOCs as calculated.

(2)  $C_3H_4O$ ,  $C_3H_6O$ , and  $C_5H_8$  can be stably adsorbed on the Ru- $MoS_2$  surface with  $E_{ad}$  values of  $-3.42$ ,  $-1.53$ , and  $-2.80$  eV, respectively, all of which are chemisorption. EDD and DOS are used to further understand the adsorption behavior of gas molecules on Ru- $MoS_2$  monolayers.

(3) The band structure and work function analysis reveal the conductivity change and sensing mechanism of gas adsorption by Ru- $MoS_2$  monolayers, and the sensitivities for three gases at room temperature are obtained as 1.09, 140.50, and 5.90, respectively.

In this study, the strong potential of Ru- $MoS_2$  monolayers as novel 2D TMD gas sensing material for the detection of exhaled characteristic gases of lung cancer has been explored, which provides theoretical guidance for the application in practical lung cancer diagnosis. Nevertheless, the Ru- $MoS_2$  monolayer discussed in this work still has certain defects, such as weak selectivity for specific VOCs. Therefore, it is necessary to solve the problem of cross interference of mixed gases in the practical application of gas sensors in the later period, such as using neural network algorithms. On the other hand, continuing to explore sensing materials with high selectivity and high sensitivity for specific VOCs is also a challenge to be overcome in the future.

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