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OPEN Design the magnetic microencapsulated phase change materials with poly(MMA-MAA) @ n-octadecane modified by Fe₃O₄

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Magnetic microencapsulated phase change materials (magnetic MicroPCMs) are hotly researched for their dual-functions with phase change and magnetic properties, which provided the new applications in fields of maneuverable phase change materials and infrared electromagnetic dual shield. A series of magnetic MicroPCMs samples are synthesized by polymerization and coprecipitation method and the chemical composition contained poly(MMA-MAA) (a) n-octadecane modified by Fe₃O₆. In addition, the characterizations exhibit the excellent magnetic and phase change properties. The magnetic MicroPCMs samples present 20 emu· g^{-1} saturation magnetization with still high enthalpy of 132 J· g^{-1} , which fully illustrates that the magnetic MicroPCMs fulfill both application on thermal energy storage and magnetic control.

Phase Change Materials (PCMs) have been attracting increasing attention owing to their excellent thermal properties. Take n-octadecane for instance, a single kilogram of n-octadecane can absorb or release energy about 228 kJ as its phase change occurs isothermally, which can improve the temperature of one-kilogram water from 1 °C to 55 °C. PCMs, hence, has been widely used in many fields, such as solar energy storage²⁻⁴, construction energy conservation^{5–8}, thermal regulated clothes^{9,10}, Electro-to-Heat Conversion^{11–13}. In general, There are three categories of PCMs, organics, salt hydrates and eutectic compounds¹⁴. N-octadecane as a kind of organic PCMs has been focused more owing to its high enthalpy and near room phase change temperature (28 °C).

To overcome the mobility of liquidus PCMs, encapsulating phase change materials become the hot topic in research of PCMs. A variety of packaging process have been applied on the encapsulated phase change materials, such as PCMs composite fibers^{9,10,15}, tube encapsulation^{16,17}, PCMs in panel^{18,19}, PCMs in pouches²⁰, PCMs/ Graphite compound^{12,21-23}, and PCMs microencapsulation^{24,25}. Among these packaging technologies, microencapsulation as an important packaging technology toward PCMs has been well developed. Both of organic and inorganic materials can be used as shell materials. Organic materials such as polymer are commonly used in those materials and they are easily prepared by polymerization such as polystyrene²⁶, polyurea-formaldehyde resin²⁷, melamine-formaldehyde resin²⁸, and poly(methyl methacrylate) (PMMA)^{25,29}. As one of the polymers, poly(MMA-MAA) presents a stable reaction process and a higher microencapsulated efficiency with formaldehyde free²⁵.

Nowadays, more and more functions of microencapsulated phase change materials have been reported, such as reversible thermochromic property³⁰, antibacterial and thermoregulation³¹, and photo-thermal conversion performance³². Meanwhile, electronic device has already found the widespread application in various fields. But enormous heat and electromagnetic radiation are produced during the electronic devices usage and significantly impact the performance and the safety of user. Magnetic MicroPCMs can avoid producing enormous heat and electromagnetic radiation because of its latent heat and electromagnetic shielding. Furthermore, the magnetic MicroPCMs can be controlled by magnet or electromagnetic device when used to MicroPCMs thermal fluid. There are some publications concerning the magnetic MicroPCMs just with silica or organosilicon materials shell^{33–35}. However, poly(MMA-MAA) based magnetic MicroPCMs have been reported rarely because Fe₃O₄ could not be well contained in n-octadecane owing to the poor dispersion of Fe₃O₄ in oil. Here, we change the

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idea and encapsulated n-octade cane to poly(MMA-MAA) firstly, then ${\rm Fe_3O_4}$ was coprecipitated on the surface of MicroPCMs after SDS modified. At last, the magnetic MicroPCMs was obtained and characterized.

Methods

Materials. n-octadecane (PCMs, purity 99 wt.%), Sodium dodecyl sulfate (SDS, as dispersant. AR, 92.5–100.5 wt.%), Methyl methacrylate (MMA was used as monomer of the polymerization, purity 99.5 wt.%), Methylacrylic acid (MAA, as monomer of the polymerization, purity 98 wt.%), Benzoyl peroxide (BPO was used as an initiator, AR), Potassium persulfate ($K_2S_2O_8$ was used as a secondary initiator, ACS, purity 99.0 wt.%(RT)), Iron chloride hexahydrate (FeCl₃·6H₂O, purity 99 wt.%), Iron chloride tetrahydrate (FeCl₂·4H₂O, purity 99.95 wt.%) and ammonia solution (NH₃·H₂O, concentration 25–28 wt.%) were commercially obtained from Aladdin Industrial Corporation, Shanghai, China.

Synthesis. Synthesis of MicroPCMs. A 250 mL three-neck round-bottomed flask was equipped with a mechanical stirrer, a reflux condenser and a nitrogen gas inlet tube, which was put in a water thermostat bath as a reaction vessel. $0.25 \, \mathrm{g}$ SDS as a dispersant dissolve in $100 \, \mathrm{mL}$ distilled water, constitute the water phase. Oil phase consist of 9 g n-octadecane, 4 g MMA, 2 g MAA, and $0.1 \, \mathrm{g}$ BPO. The water phase was added to the $250 \, \mathrm{mL}$ three-neck round-bottomed flask is ready, and then was heated to $55 \, ^{\circ}\mathrm{C}$ with stirring to $500 \, \mathrm{rpm}$ for $10 \, \mathrm{min}$. Afterwards, the oil phase was poured into the water phase to form an oiliness suspension with stirring to $1200 \, \mathrm{rpm}$ for $15 \, \mathrm{min}$. And then, the suspension was heated to $85 \, ^{\circ}\mathrm{C}$ for $3.5 \, \mathrm{h}$ with rotational rate was turned down $800 \, \mathrm{rpm}$, the nitrogen gas inlet tube was opened at the same time. $0.1 \, \mathrm{g} \, \mathrm{K}_2 \mathrm{S}_2 \mathrm{O}_8$ was dissolved into $1.4 \, \mathrm{g}$ of water to form an aqueous solution which was added into the reaction suspension and continued for $1.5 \, \mathrm{h}$ with the same stirring rate after all. Ultimately, the final suspension was separated with vacuum filtration after washed by distilled water at $50 \, ^{\circ}\mathrm{C}$ to eliminate impurities and a wet cake was dried in an oven at $50 \, ^{\circ}\mathrm{C}$ for $48 \, \mathrm{h}$ to remove the residual water. Finally, white particles were obtained which like powder named MicroPCMs 25 .

Synthesis of magnetic MicroPCMs. A 250 mL three-neck round-bottomed flask was equipped equally. 2 g of the resultant MicroPCMs was added into the three-neck round-bottomed flask with 50 mL distilled water, 0.05 g SDS was added as the surfactant. After the mixture was heated to $40\,^{\circ}$ C with stirring to $300\,^{\circ}$ pm for $15\,^{\circ}$ min to obtain the homogeneous dispersions. Then, $0.88\,^{\circ}$ FeCl₂· $4H_2O$ and $0.6\,^{\circ}$ FeCl₃· $6H_2O$ was added into the dispersions with nitrogen gas inlet tube was opened until them fully dissolved. Afterward, about $20\,^{\circ}$ mL NH₃·H₂O was dropped into the dispersions at pH 10.0-11.0. Black microparticles were obtained after $10\,^{\circ}$ min. Finally, the black microparticles were carefully collected by a magnet and were washed with distilled water several times. The obtained microparticles were filtrated and dried in an oven at $50\,^{\circ}$ C for $24\,^{\circ}$ h. The final products were prepared for further characterization and testing.

Characterization. The surface morphology and elemental compositions of the magnetic MicroPCMs were presented using a Hitachi S-4800 scanning electron microscope (SEM) equipped with an energy-dispersive X-ray (EDX) spectroscopy. Tecnai F20 was chosen to characterize diameter and microstructure of fabricated Fe₃O₄ as the HRTEM. The chemical structures of the samples were investigated using a Nicolet 6700 FTIR spectrophotometer, and the simples were measured at a wave number of 4000-400 cm⁻¹ using a KBr sampling sheet. A Germany Bruker D8 Advance X-ray diffractometer with Cu-Ka radiation ($\lambda = 0.154$ nm) gave the Powder X-ray diffraction (XRD) data at a scan rate of 10°/min in the 2θ range of 5°–90°. The phase-change behaviors of the samples were investigated using a PerkinElmer Diamond Dynamic differential scanning calorimetry (DSC) scans, coupled with a thermal analysis data station. All of the measurements were carried out under a nitrogen atmosphere at a heating or cooling rate of 10 °C/min in the range of 0-60 °C, and the mass for each sample was about 7-8 mg. Thermogravimetric analysis (TGA) of the magnetic microcapsules was carried out using a TA Instruments SDT Q600 thermal gravimetric analyzer under a nitrogen atmosphere. The sample with a mass of about 7-8 mg was placed in an aluminum crucible and then was ramped from room temperature up to about 600 °C at a heating rate of 10 °C/min. A Physical Property Measurement System (PPMS-9, Quantum Design) performed the magnetic properties of the samples in applied fields up to 20,000 Oe at room temperature (~298 K). The distribution of the particle diameter of magnetic MicroPCMs was measured on the Microtrac S3500 SI particle size analyzer, with the specimens were dispersed in ethanol.

Results and Discussions

Synthetic strategy. The synthetic method of the magnetic MicroPCMs with Fe_3O_4 is shown in Fig. 1. The MicroPCMs was fabricated by n-octadecane as the core and poly(MMA-MAA) as the shell firstly in this work. After the MicroPCMs was manufactured, a little of SDS was chosen to modify the surface of MicroPCMs. Then the Fe^{2+} and Fe^{3+} were dissolved and attached to MicroPCMs due to the amphipathic character of SDS, ammonia was dropped in reaction systems as precipitating reagent. And then, the Fe_3O_4 particles were produced adopted coprecipitation method and attached to the shell of the MicroPCMs owing to physical adsorption. The SDS was used as surface modifers to prevent the Fe_3O_4 particles getting off the surface of MicroPCMs. The chemical equation for fabrication of Fe_3O_4 is shown as below:

$$Fe^{2+} + 2Fe^{3+} + 8NH_3 \cdot H_2O = Fe_3O_4 \downarrow + 8NH_4^+ + 4H_2O$$

Molar ratio of Fe²⁺ and Fe³⁺ is 1:2 in theory according to the equation. But the ratio is 2:1 or 3:1 in actual experiments because the Fe²⁺ was easy to be oxidized³⁶.

Optimum Condition of MicroPCMs' Fabrication. The dosage of copolymer poly(MMA-MAA) was determined refer to the reference²⁵. The mass ratio of n-octadecane and poly(MMA-MAA) was 9:6, as change of

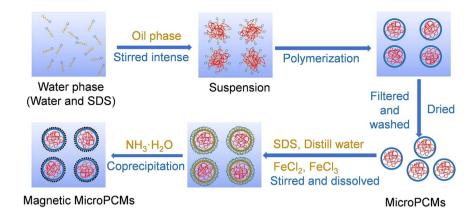


Figure 1. Schematic synthetic strategy of the magnetic MicroPCMs with an n-octadecane core and $Fe_3O_4/poly(MMA-MAA)$ shell.

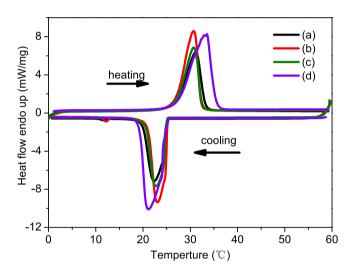


Figure 2. DSC cooling and heating thermograms of (a) SDS 0.5g, (b) SDS 0.25g, (c) SDS 0.125g, (d) pure n-octadecane.

SDS (g)	T _m (°C)	$\Delta H_m (\mathbf{J} \cdot \mathbf{g}^{-1})$	T _c (°C)	$\Delta H_c (\mathbf{J} \cdot \mathbf{g}^{-1})$	$\Delta H_a (\mathbf{J} \cdot \mathbf{g}^{-1})$	n-octadecane (wt.%)
0.5	30.3	146.7	23.5	139.4	143.0	60.1
0.25	30.3	176.4	23.5	165.6	171	71.9
0.125	30.3	145.9	23.5	141.9	143.9	60.5
0	33.3	237.2	21.1	238.6	237.9	100

Table 1. Thermal properties of MicroPCMs with various SDS dosage.

the dosage of the dispersant SDS, there are different heat characters of MicroPCMs. Therefrom the actual average enthalpy (ΔH_a) of the phase change process will be compared clearly.

 $\Delta H_a = (\Delta H_c + \Delta H_m)/2$, ΔH_c means the crystallization enthalpy of the n-octadecane, ΔH_m means the melting enthalpy of the n-octadecane.

The measured enthalpy, melting peak temperatures and crystallization peak temperatures of the MicroPCMs and pure n-octadecane are listed in Table 1 and the DSC curves are shown in Fig. 2. Compared the average enthalpy to pure enthalpy, the MicroPCMs are lower universally due to the content decrease of the n-octadecane (as Table 1 indicated). Moreover, the melting peak temperature was shifted left and the crystalline temperature was shifted right compare with pure n-octadecane, which means the thermal conductivity of the MicroPCMs is higher than pure n-octadecane's. It is observed that pure n-octadecane exhibits single peak at 21.1 °C in its solidification process. However, there are two peaks at 22.5 °C and 11.7 °C in the crystallization process of the MicroPCMs due to the impurity induced heterogeneous nucleation in the solidification process³⁷.

It's also clearly shown that there are dramatic effects of different dosage of SDS on the content n-octadecane in Fig. 2. The ΔH_a of the MicroPCMs were 143.0, 171 and 143.9 J·g⁻¹ when synthesized with the SDS mass of

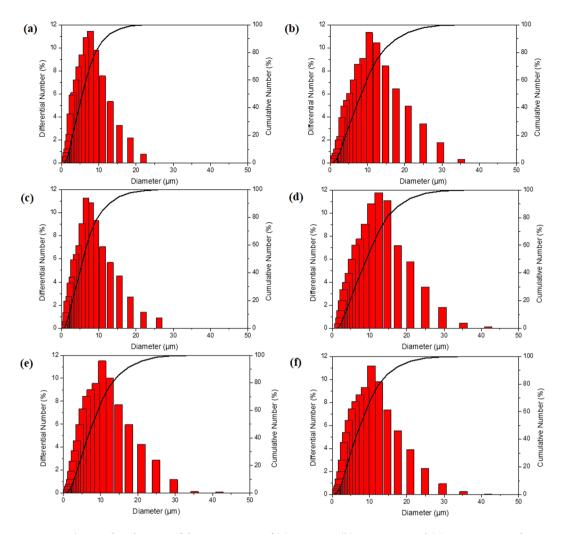


Figure 3. The size distributions of the MicroPCMs of (a) SDS 0.5g, (b) SDS 0.25g, and (c) SDS 0.125g; and magnetic MicroPCMs with Fe₃O₄/MicroPCMs mass ratios (d) 70/30, (e) 20/80, and (f) 11/89.

0.5, 0.25 and 0.125 g, respectively. The enthalpy was increased firstly and then decreased with the decreased in the mass of SDS. The distributions of the particle diameter of MicroPCMs fabricated with different masses of SDS were shown in Fig. 3(a-c). When the mass of SDS is 0.5 g, the diameters of the specimen are in the range of 0.8–22 μm, and the average diameter is about 8 μm. When the mass of SDS is decreased to 0.25 g, the diameters of the specimen are in the range of 0.9–35 µm, and the average diameter is about 11 µm. When the mass of SDS is decreased to 0.125 g, the diameters of the specimen are in the range of $0.8-26\,\mu m$, and the average is about $7\,\mu m$. The diameter ranges of the MicroPCMs with 0.5 g SDS additive as dispersant is the narrowest, but the average diameter reached the maximum when the mass of dispersant SDS is 0.25 g. The dosage of dispersant SDS influences on the sizes of droplet in the suspension, and then affects the diameter of the microcapsules after the polymerization. It has been reported that the amount of the dispersant would make a great influence on the particle diameter of the MicroPCMs. Because the thickness of shell is tiny, the size distribution of particles can be changed largely by the shell's structure³⁸. The PMMA-MAA structure could be changed by the amounts of SDS, as different amounts of surfactant were added in the same amounts of reactive monomers, the structure of copolymers will be changed in polymerization. In addition, the amounts of SDS would influence on the size of floating droplet in the suspension polymerization. And the particle diameter curve appeared inverted U-shaped with the increasing of the amount of the dispersant. As the increasing of the particle diameter, the content of the n-octadecane will increase and lead to raise of the enthalpy^{38,39}. In this paper, the magnetic MicroPCMs was fabricated from the MicroPCMs with the maximum enthalpy.

The morphology, crystallinity and composition of magnetic MicroPCMs. There are four samples of different mass ratio of $Fe_3O_4/MicroPCMs$ as controlling the amount of Fe^{2+} and Fe^{3+} in Table 2. And the SEM micrographs of the surface morphology of MicroPCMs and the magnetic MicroPCMs were presented in Fig. 4. Figure 4(a) shows the microscopic morphology of the MicroPCMs, they are excellent coated by Fe_3O_4 and the surface looks smooth and flat. The surface morphology of the magnetic MicroPCMs is different to MicroPCMs without magnetic particles, and it's clearly shown in Fig. 4(b-d) that there are lots of nanoparticles on the MicroPCMs shell. Moreover, the magnetic MicroPCMs destroyed with ultrasounds still exhibits nanoparticles coated shell

Sample No.	m(g, FeCl ₂ ·4H ₂ O)	m(g, FeCl ₃ ·6H ₂ O)	m(g, Fe ₃ O ₄)	mass ratios of Fe ₃ O ₄ /MicroPCMs
1	2.99	1.35	4.67	70:30
2	0.88	0.6	0.5	20:80
3	0.44	0.3	0.25	11:89
4	0	0	0	0:100

Table 2. Mass ratios of Fe₃O₄/MicroPCMs.

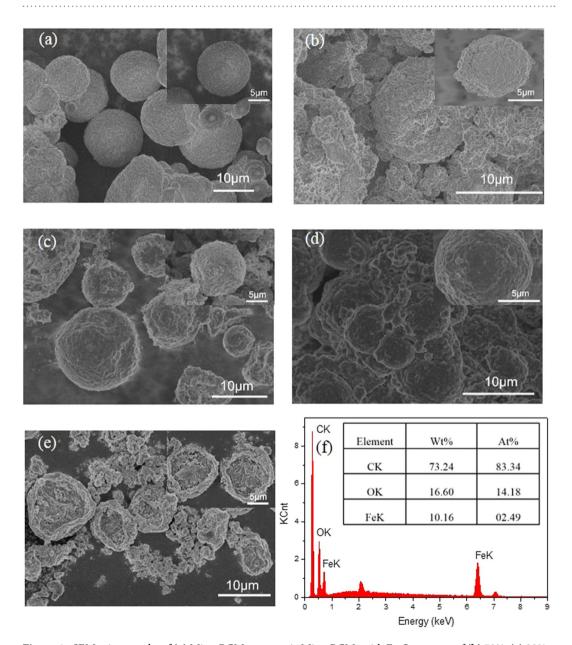


Figure 4. SEM micrographs of **(a)** MicroPCMs, magnetic MicroPCMs with Fe₃O₄ content of **(b)** 70%, **(c)** 20%, **(d)** 11%, **(e)** magnetic MicroPCMs destroyed with ultrasounds, and **(f)** EDX spectra of magnetic MicroPCMs.

in Fig. 4(e). Therefore, the nanoparticles Fe_3O_4 could attach the surface of the MicroPCMs and the thin Fe_3O_4 presented better surface appearance. In addition, the EDX spectra of the magnetic MicroPCMs samples show peaks clearly corresponding to C, O and Fe elements in Fig. 4(f), confirming that the magnetic MicroPCMs shell consists of Fe_3O_4 .

The diameter and microstructures of Fe_3O_4 of the magnetic MicroPCMs obtained in this study were examined using HRTEM, and the resulting TEM micrographs were presented in Fig. 5. As is shown in Fig. 5(a,b), the border of the magnetic MicroPCMs with 20% Fe_3O_4 is not to distinguish clearly because the organic shell's carbon

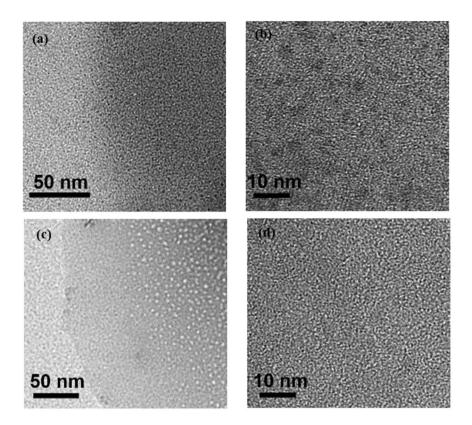


Figure 5. TEM micrographs of magnetic MicroPCMs of with Fe₃O₄/MicroPCMs mass ratios of (**a**) and (**b**) 20/80, and (**c**) and (**d**) 11/89.

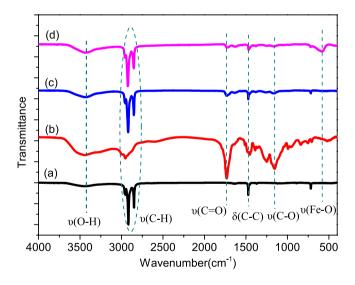


Figure 6. FTIR spectra of the (a) n-octadecane, (b) PMMA-MAA, (c) MicroPCMs, and (d) MicroPCMs-Fe₃O₄.

element of the MicroPCMs is the same as micro grid of the sample preparation devices. Nevertheless, the Fe_3O_4 particles of the magnetic MicroPCMs can be discriminated distinctly with homogeneous distribution, and the size of Fe_3O_4 is at the nanoscale. The TEM micrographs of magnetic MicroPCMs with 11% Fe_3O_4 is also demonstrated as Fig. 5(c,d), the magnetic particles are more clear and uniform due to the less mass of Fe_3O_4 . Meanwhile, the size and microstructure of Fe_3O_4 is same as 20% Fe_3O_4 of magnetic MicroPCMs.

The chemical structures and compositions of the magnetic MicroPCMs were characterized by FTIR spectroscopy and the resulting spectra are shown in Fig. 6, in which the infrared absorption spectra of the n-octadecane, ploy(MMA-MAA), and MicroPCMs are also presents. It is clearly known that 3430 cm⁻¹ is ascribed to the O-H stretching vibrations peaks, three intensive absorption peaks at 2960 cm⁻¹, 2920 cm⁻¹ and 2850 cm⁻¹ corresponding to the alkyl C-H stretching vibrations of methyl and methylene groups. Meanwhile, the absorption peak of

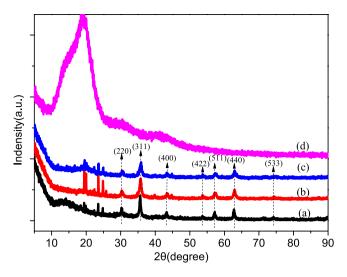


Figure 7. XRD patterns of the Fe₃O₄/MicroPCMs mass ratios of (a) 70/30, (b) 20/80, (c) 11/89, (d) 0/100.

 $1732\,\mathrm{cm^{-1}}$ is assigned to the C=O stretching vibrations of ester group, the peak at $1472\,\mathrm{cm^{-1}}$ denote C-H bending vibrations of methyl group, the peak at $1150\,\mathrm{cm^{-1}}$ is attributed to C-O stretching vibrations of ester group, and the peak Fe-O vibration appears at $575\,\mathrm{cm^{-1}}$ in the infrared spectra³⁴. Compared to the three infrared absorption curves in Fig. 6(a-c), it is clearly noted that the vibration peaks of curve (a) and curve (b) can all be found in curve (c), which means that MicroPCMs is composed of the shell ploy(MMA-MAA) and the core n-octadecane. In addition, the curve (d) exhibits stronger Fe-O vibration peak than that in curve (c), it indicates that magnetic MicroPCMs has extra Fe₃O₄ than MicroPCMs. It proves that the magnetic MicroPCMs consists of n-octadecane, ploy(MMA-MAA) and Fe₃O₄.

The crystalline structure of the magnetic MicroPCMs shell was characterized by XRD, and the corresponding patterns were presented in Fig. 7. It is evidently shown the diffraction peaks at $2\theta = 30.2^{\circ}$, 35.6°, 43.4°, 53.8°, 57.2°, 63.1°, and 74.2° in Fig. 7(a–c). According to the XRD standard PDF card no. 65-3107, these diffraction peaks are well attributed to the (220), (311), (400), (422), (511), (440), and (533) crystal planes of Fe₃O₄. However, the curve of Fig. 7(d) have not been found the diffraction peaks, it exactly explains the truth that the MicroPCMs shell poly(MMA-MAA) is amorphous. This well illustrated that the surface of magnetic MicroPCMs is coated by Fe₃O₄ nanoparticles.

It has been proved that the magnetic nanoparticles are coated on the shell of MicroPCMs from the SEM micrographs, EDX spectra, FTIR absorption spectra, and XRD pattern as mentioned above. This paper discussed the influences on the performance of MicroPCMs with different amounts of Fe_3O_4 below.

The performance of MicroPCMs with different amounts of Fe₃O₄. The SEM micrographs of the surface morphology of the magnetic MicroPCMs with different dosages of Fe₃O₄ are illustrated in Fig. 4, and the corresponding sample number is shown in Table 2. It's clearly to be found that the surface of the sample 1 is coated by lots of aggregative nanoparticles in Fig. 4(b), even they are easy to drop out from the surface, so that there are many nanoparticles fall out shown in Fig. 4(b). It is presented in Fig. 4(c) that the sample 2 have a complete coverage of the surface. Meanwhile, there are little nanoparticles out of the surface, it means that it is the proper dosage of the Fe₃O₄ on the sample 2. There are many magnetic MicroPCMs adhered to each other in the sample 3 was shown in Fig. 4(d). However, they are flatter on the surface than sample 2. Therefore, the little amounts of Fe₃O₄ could cover the surface of the magnetic MicroPCMs and make the surface smooth like the MicroPCMs. In addition, the distributions of the particles diameter with different mass of Fe₃O₄ were given in Fig. 3(d-f). And the average diameter of the magnetic MicroPCMs with 70% Fe₃O₄ has a great increase from 11 to 13 μ m. However, the average diameter of the sample 2 and 3 just have a little increase. These prove that the small amount of Fe₃O₄ could attach with shell of MicroPCMs densely. Furthermore, the diameters of the magnetic MicroPCMs confirm that Fe₃O₄ particles covered the surface of MicroPCMs uniformly. To conclude, 20 percent of Fe₃O₄ in the magnetic MicroPCMs can present excellent microscopic surface morphology.

Phase change characteristics and thermal stability performance. The phase change behaviors were investigated by DCS and the resulting DSC thermograms are presented in Fig. 8. Table 3 lists the thermal properties of MicroPCMs with various Fe_3O_4 dosage, in which ΔH_{ac} means the calculate value of the enthalpy with different amounts of Fe_3O_4 by $\Delta H_{ac} = \Delta H_{aMicroPCMs} \times w_{MicroPCMs}$ and the $w_{MicroPCMs}$ means the content of MicroPCMs in magnetic MicroPCMs. Compared to the four curves in Fig. 8, there are not great influence on melting peaks and the crystalline peaks, and they all have the melting peaks at about 30 °C and the crystalline peaks at about 24 °C. Meanwhile, the average enthalpy ΔH_a decrease with increase of the Fe_3O_4 amounts. It is clearly known that the ΔH_a is related to the content of the n-octadecane, and the increased amounts of Fe_3O_4 must lead to the decrease of the n-octadecane's content in the magnetic MicroPCMs. In addition, the ΔH_a are pretty close to the ΔH_{ac} of the sample 2 and sample 3, it means that the results is corresponding to the anticipation. However, the sample 1 presents the great difference between ΔH_a and ΔH_{ac} , the value ΔH_a is greater than ΔH_{ac} in Table 3. It indicates

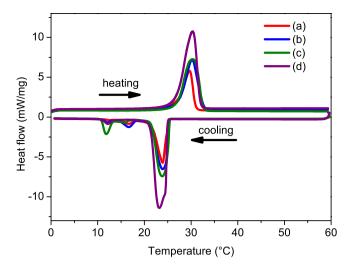


Figure 8. DSC cooling and heating thermograms of the Fe₃O₄/MicroPCMs mass ratios of (**a**) 70/30, (**b**) 20/80, (**c**) 11/89, (**d**) 0/100.

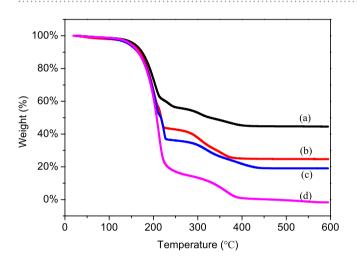


Figure 9. TGA thermograms of the Fe₃O₄/MicroPCMs mass ratios of (a) 70/30, (b) 20/80, (c) 11/89, (d) 0/100.

Sample No.	Fe ₃ O ₄ : MicroPCMs	T _m (°C)	T _c (°C)	ΔH_a $(\mathbf{J} \cdot \mathbf{g}^{-1})$	ΔH_{ac} (J·g ⁻¹)	n-octadecane (wt.%)
1	70:30	30.3	23.5	88	51.3	37.0
2	20:80	30.3	23.5	132	136.8	55.5
3	11:89	30.3	23.5	154	152.2	64.7
4	0:100	30.3	23.5	171	171	71.9

Table 3. Thermal properties of MicroPCMs with various Fe₃O₄ dosage.

that there are some Fe_3O_4 nanoparticles lost in the fabrication process of the magnetic MicroPCMs, so that the actual quality of Fe_3O_4 is lower than the expected value. Nonetheless, the 20 percent content of Fe_3O_4 nanoparticles in the magnetic MicroPCMs could coated the MicroPCMs well and still keep high enthalpy.

The thermal stability of the magnetic MicroPCMs was investigated by TGA and the resulting thermograms are presented in Fig. 9. The MicroPCMs starts to loss the weight at about 180 °C in Fig. 9(d), it is clearly shown that high temperature make the MicroPCMs swell, n-octadecane flows out of the MicroPCMs and begins to decompose. The second weight loss happens at nearly 300 °C, because the polymer shell starts to decompose with the molecular chain ruptured. The weight loss curves of magnetic MicroPCMs are presented in Fig. 9(a-c). There is the second weight loss at around 200 °C differs from MicroPCMs, it is obviously known that the Fe₃O₄ nanoparticles coated on the surface lead to high temperature of the weight loss for some n-octadecane. The first and third weight loss of the magnetic MicroPCMs are the same temperature as the first and second weight loss of MicroPCMs. Hence, compare with MicroPCMs, the magnetic MicroPCMs have a certain degree of ascension. In

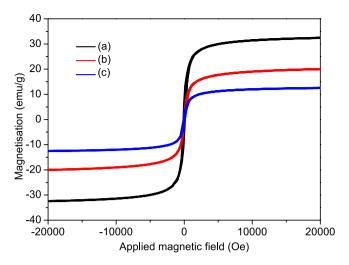


Figure 10. Magnetic hysteresis loops of the Fe₃O₄/MicroPCMs mass ratios of (a) 70/30, (b) 20/80, (c) 11/89.

addition, because the pure n-octade cane decomposition temperature (T_d) is just 128 °C⁴⁰, the thermal stability of magnetic MicroPCMs has a substantial increase.

Magnetic properties. The magnetic properties of the magnetic MicroPCMs were characterized by PPMS-9 at room temperature (\sim 300 K), and the resulting magnetic hysteresis loops are shown in Fig. 10. It is obviously shown that three samples exhibit low residual intensity of magnetization and coercivity. It means that the magnetic MicroPCMs possess super paramagnetic behavior. Furthermore, the saturation magnetization of three magnetic MicroPCMs samples were measured to be 32.5 emu·g⁻¹, 20 emu·g⁻¹, 12.5 emu·g⁻¹ when the Fe₃O₄/MicroPCMs mass ratios of 70/30, 20/80, and 11/89, correspondingly. It is clear that the saturation magnetization improved with the increase of Fe₃O₄ content. This further proves that the Fe₃O₄ could well coated on the surface of the MicroPCMs with various amounts.

Conclusions

In this work, we coated the Fe $_3$ O $_4$ nanoparticles on the surface of the MicroPCMs and fabricated the functional phase change materials with magnetic property. A series of products were prepared with differ amounts of Fe $_3$ O $_4$ and characterized by particle size analyzer for size distributions, SEM and TEM for microscopic morphology, FTIR for chemical structure, XRD for crystalline structure, DSC for phase change behaviors, TGA for thermal stability, and PPMS for magnetic properties. After these characterizations, the magnetic MicroPCMs exhibits excellent magnetic and phase change properties. For example, the 11 percent of content of Fe $_3$ O $_4$ in the magnetic MicroPCMs still present 20 emu·g $^{-1}$ saturation magnetization with 132 J·g $^{-1}$ enthalpy. This will provide new thinking on the applications of the phase change materials on electromagnetic protection clothing with thermostat for soldiers, gravida, or other special operations personnel, infrared-electromagnetic dual shield for stealth aircraft, and thermal manipulation in real production.

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Author Contributions

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Additional Information

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