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Au-NiCo₂O₄ supported on three-dimensional hierarchical porous graphene-like material for highly effective oxygen evolution reaction

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A three-dimensional hierarchical porous graphene-like (3D HPG) material was synthesized by a one-step ion-exchange/activation combination method using a cheap metal ion exchanged resin as carbon precursor. The 3D HPG material as support for Au-NiCo₂O₄ gives good activity and stability for oxygen evolution reaction (OER). The 3D HPG material is induced into NiCo₂O₄ as conductive support to increase the specific area and improve the poor conductivity of NiCo₂O₄. The activity of and stability of NiCo₂O₄ significantly are enhanced by a small amount of Au for OER. Au is a highly electronegative metal and acts as an electron adsorbate, which is believed to facilitate to generate and stabilize Co⁴⁺ and Ni³⁺ cations as the active centres for the OER.

Electrochemical hydrogen evolution from water splitting by coupling renewable energy devices such as wind energy and solar energy with water electrolysis has attracted more and more attention in alkaline media due to continuous consumption of fossil fuels and ever-increasing environmental problems¹. Hydrogen can be used as a fuel to get a reliable power for almost every application that fossil fuels are used. The hydrogen produced by electrolysis can be used for methanation of CO₂, combustion processes, and conversion back into electricity by fuel cells². In alkaline media, electrochemical water electrolysis consists of two half-reactions: the cathodic hydrogen evolution reaction (HER, 2H₂O + 2e = 2OH⁻ + H₂) and the anodic oxygen evolution reaction (OER, 4OH⁻ = 2H₂O + 4e + O₂). Of two half-reactions, the OER requires to form two oxygen-oxygen bonds in the four-electron redox processes by transfer protons and electrons, which results in more kinetically demand for the OER^{3,4}. So, the OER needs relatively high overpotential at the anode, which is a major cause of high energy consumption. Thus, a lot of efforts have been devoted to explore the electrocatalysts with low OER overpotential.

The rutile-type oxides of RuO₂ and IrO₂ show the lowest OER overpotential, however these oxides suffer from poor chemical stability in alkaline media and the high price and limited supply of Ru and Ir^{5,6}. So other metal oxides such as Cu oxide⁷, Mn oxide⁸ have been developed. Among of all the oxide catalysts, particular attention has been paid to the cobalt oxide^{9,10} and nickel oxide^{11,12}, due to their high abundance, low cost, small overpotential and fast kinetics of the OER. Many researchers have studied the other oxides to enhance the performance of OER for Ni oxide^{13,14} and Co oxide^{15,16}. Trotochaud and his cooperators have reported that the conductivity of Ni oxide shows a > 30-fold increase with Fe oxide addition¹³. On the other hand, the presence of Fe alters the redox properties of Ni, causing a positive shift at the potential of Ni(OH)₂/NiOOH redox reaction, a decrease in the average oxidation state of the Ni sites, and a concurrent increase in the activity of Ni cations for the OER¹⁴. The electrocatalytic synergism of mixed oxides of Co and Ni has been studied by many researchers^{17,18}. Binary NiO_x/CoO_x-modified electrodes show high catalytic activity and marked stability which far exceed that obtained at the individual oxide-modified electrodes.

The nanohybrid materials have been used as efficient electrocatalysts for OER such as CoFe₂O₄¹⁹, CoMn₂O₄²⁰, Ca₂Mn₃O₈²¹, Co₂MnO₄^{20,22}, Ca₂Mn₂O₅²³, CuCo₂O₄²⁴, ZnCo₂O₄²⁵ and CoMoO₄²⁶. The OER performance of NiCo₂O₄ spinel oxide has been studied in alkaline media and NiCo₂O₄ has high activity for OER²⁷⁻²⁹. We have reported that the activity of NiCo₂O₄ is much higher than that of NiO and Co₃O₄ for OER in 0.1 mol L⁻¹

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KOH³⁰. Conductivity is an importance index for designing and developing available electrocatalysts for OER. With metallic conducting property of a conductivity of 10^{-4} S cm⁻¹, RuO₂ and IrO₂ give the best OER activity³¹. However, many oxides such as NiCo₂O₄ suffer from low electrical conductivity. So, how to improve their poor intrinsic conductivity is still challenging for oxides. Therefore, carbon materials such as carbon nanotubes have been induced into oxides to improve the electrical conductivity^{32–34}. Currently, graphene-based carbon materials including monolayer and multilayers nanosheets are highly promising materials as the new-generation supporting materials for electrocatalysts, owing to their high specific surface area, high electrical conductivity, and outstanding chemical and electrochemical stability³⁵. The graphene as support for oxides such as MnO_x³⁶, CoO_x^{16,37}, CuFe oxide³⁸, FeNi oxide³⁹, NiCo oxide⁴⁰, CoFe₂O₄¹⁹ and CuCo₂O₄²⁴ has been reported for OER. Long and his cooperators reported that a synergy between the catalytic activity of the FeNi oxide and the enhanced electron transport arising from the graphene results in superior electrocatalytic properties for the OER³⁹. The graphene supported NiCo₂O₄ has been reported for OER^{41–43}. Zhao and coworkers have prepared an active catalyst composed of porous graphene and cobalt oxide (PGE-CoO), which has demonstrated high porosity, large specific surface area and fast charge transport kinetics³⁷. The catalyst also exhibits excellent electrochemical performance towards OER with a low onset potential and high catalytic current density. The enhanced catalytic activity could be ascribed to porous structure, high electroactive surface area and strong chemical coupling between graphene and CoO nanoparticles. Moreover, this OER catalyst also shows good stability in the alkaline solution. The high performance and strong durability suggest that the porous structured composite is favorable and promising for water splitting. However, the intrinsic hydrophobic properties of graphitized basal plane structures cause a great difficulty in uniformly loading metal nanoparticles on the surface of graphene. Though the hydrophilicity of reduced graphene oxide (RGO) could be improved via introducing oxygen functional groups, their electronic conductivity is still insufficient due to their partly restored graphitic structures. Based on this fact, it is fundamental interest to develop the novel graphene-based carbon materials with high specific surface area, high electronic conductivity as well as strong affinity to foreign constituents, beyond the continuous development of hybrid architectures for electronics and various electrochemical systems³⁵. Shen and coworkers have developed a novel active three-dimensional hierarchical porous graphene-like (3D HPG) material with hierarchical pores synthesized through an efficient ion-exchange-assisted synthesis route⁴⁴. The 3D HPG material shows high electronic conductivity and strong cohesive force and distribution effects toward the catalyst nanoparticles⁴⁵. The 3D HPG material can provide a highly conductive structure in conjunction with a large surface area to contact the MnO₂ nanoparticles and effectively enhance the mechanical strength of the composite during volume changes as well as suppress the aggregation of MnO₂ nanoparticles during Li-ion insertion/extraction⁴⁶.

In recent years, a lot of efforts have been made to enhance the electrocatalytic activity catalysts and several strategies have been proposed. Among them, bifunctional mechanism, modified with highly electronegative metals, such as Pt⁴⁷, Pd⁴⁸ and Ru⁴⁹, has been demonstrated as one of the most effective methods to improve the electrocatalytic efficiency. However, the high price and limited supply of Pt, Pd and Ru are major barriers to the development of OER catalysts using Pt-based, Pd-based and Ru-based catalysts. Scientists have paid more attention to Au because it is much more abundant and more available than Pt, Pd and Ru on the earth. Gold has been used to enhance the oxide activity of OER such as Co oxide^{50–53}, Mn oxide^{54,55}. A small amount of Au nanoparticles (<5%) in α -MnO₂/Au catalysts significantly improves the catalytic activity up to 6 times compared with the activity of pure α -MnO₂ for OER⁵⁴. Bell and coworkers have developed noble metal-supported cobalt oxide and found that the OER activity of cobalt oxide deposited on Au is nearly three times higher than that of bulk Ir⁵¹. The Au/NiCo₂O₄ nanoarrays exhibit excellent OER activity, which is almost four times higher than that of Ir/C⁵⁶.

In this paper, we focused on the development of high performance Au/NiCo₂O₄ catalysts supported on the 3D HPG material for the OER. It is widely accepted that 3D HPG is an outstanding matrix as support material with high electrical conductivity, good electrochemical stability, controllable specific surface areas as well as pore structure^{46,57}.

Results

The morphology of the 3D HPG was characterized by scanning electron microscopy (SEM) as shown in Fig. 1a,b. The Fig. 1a shows a 3D interconnected porous structure with well-developed open macropores. The magnified SEM (Fig. 1b) exhibits sub-micrometer-sized pores. The thickness of the carbon sheet is about 6 nm. The degree of crystallinity of the HPG was characterized by Raman spectrum as shown in Fig. 1c. The G band peaked at 1578 cm⁻¹ is related to the in-plane bond-stretching motion of the pairs of carbon sp² atoms, which indicates the presence of crystalline graphene layers. The D band peak at 1329 cm⁻¹ is assigned to disordered carbon and highly sensitive to graphitic defects within the graphite layers⁵⁸. When the ratio of the peak intensity of D band to that of G band (I_D/I_G) is smaller, the degree of crystallinity will be higher⁵⁹. Here the value of I_D/I_G is 1.05 for HPG. The value of I_D/I_G is 0.87 for NiCo₂O₄/HPG. The value of I_D/I_G for NiCo₂O₄/HPG is a little lower than that of HPG. It may be attributable to the doping with NiCo₂O₄, which can induce defect sites and destruction in the carbon lattice, and lead to an increase in the degree of distortion⁶⁰. X-ray diffraction (XRD) patterns for the Au/HPG, NiCo₂O₄/HPG, Au-NiCo₂O₄(wt 1:5)/HPG are shown in Fig. 1d. Diffraction peak at around 26.4° observed in all the samples is assigned to (002) plane of graphene. Diffraction peaks around 31.4°, 36.7°, 44.3°, 59.1° and 64.6° are assigned to the (220), (311), (200), (511) and (440) facets of NiCo₂O₄. In the case of NiCo₂O₄/HPG and Au-NiCo₂O₄(wt 1:5)/HPG, XRD pattern peaks of NiCo₂O₄ are in good agreement with the standard card (JCPDS no. 20-0781). The strong diffraction peaks at the Bragg angles of 38.1°, 44.4°, 64.5° and 77.5° correspond to the (111), (200), (220) and (311) facets of the face-centered-cubic (fcc) crystallite Au.

Chemical bonding states in the Au-NiCo₂O₄(wt 1:5)/HPG were analyzed by X-ray photoelectron spectroscopy (XPS) as shown in Fig. 2. A survey spectrum of Au-NiCo₂O₄(wt 1:5)/HPG is shown in Fig. 2a and the peaks are corresponding to existence of Au 4f, C 1s, O 1s, Co 2p and Ni 2p. The binding energy values of XPS spectrum of Au 4f are 82.8 eV and 86.5 eV, which can be assigned to the Au 4f_{5/2} and Au 4f_{7/2} as shown in Fig. 2b, which are

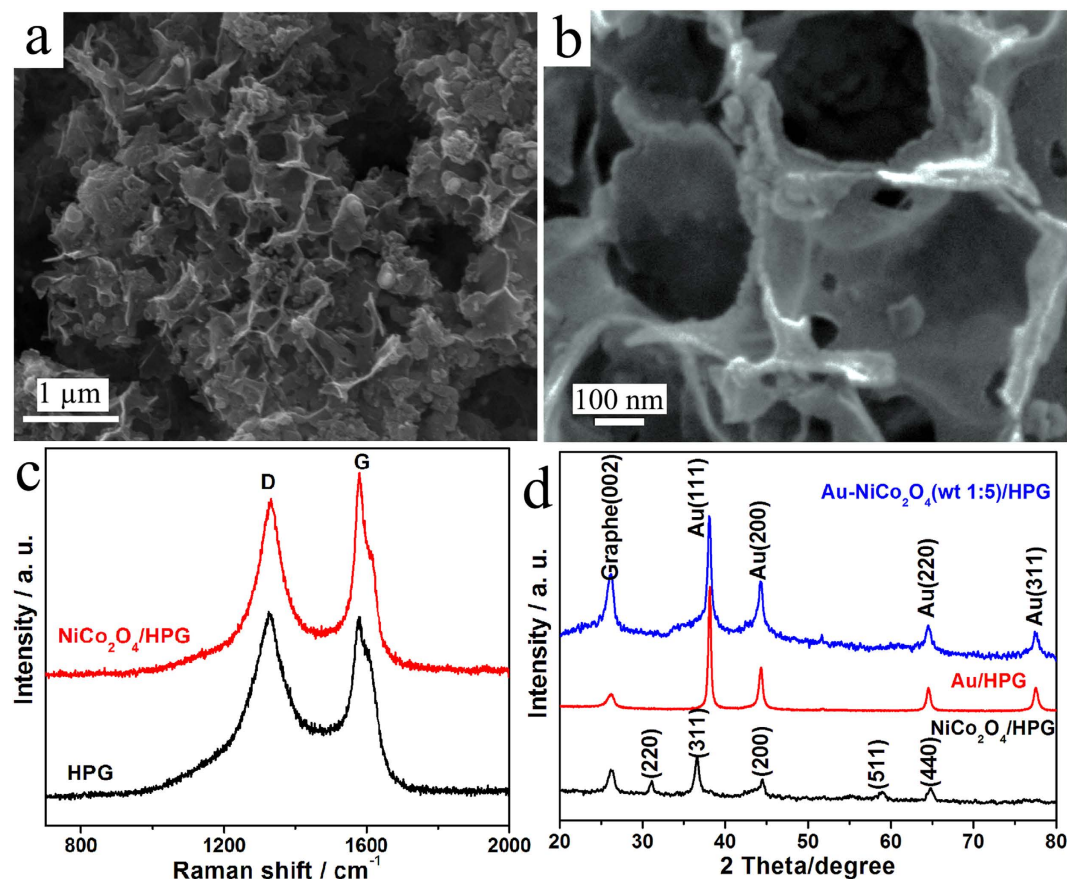


Figure 1. (a,b) SEM image for 3D HPG. (c) Raman spectra for HPG and NiCo₂O₄/HPG. (d) XRD patterns for NiCo₂O₄/HPG, Au/HPG and Au-NiCo₂O₄(wt 1:5)/HPG.

typical binding energy values of metallic Au⁰ species⁶¹. These data show that Au species attached to the surface of 3D HPG exist in the form of Au⁰. The binding energy of C 1s peak locates at 284.1 eV which is related to the graphitic carbon in 3D HPG as shown in Fig. 2a. The binding energy values of XPS spectrum of Ni 2p are 854.9 eV and 872.5 eV for Au-NiCo₂O₄(wt 1:5)/HPG as shown in Fig. 2c can be assigned to the Ni2p_{3/2} and Ni2p_{1/2}, which can be assigned to Ni²⁺. The binding energy values of XPS spectrum of Co 2p are 780.1 eV and 795.9 eV for Au-NiCo₂O₄(wt 1:5)/HPG as shown in Fig. 2d can be assigned to the Co 2p_{3/2} and Co2p_{1/2}, which can be assigned to Co³⁺. The relatively narrow peak width, the 2p_{3/2} to 2p_{1/2} separation of 15.9 eV, and the absence of any shake-up peak all reveal that no Co²⁺ exists in the NiCo₂O₄ phase⁵⁶. It means that the Co cation is composed of lots of Co³⁺.

The typical high-resolution transmission electron microscopy (HRTEM) images of the Au-NiCo₂O₄(wt 1:5)/HPG are shown in Fig. 3. It can be observed that the catalyst particles are well dispersed on the surface of HPG with a narrow size distribution of 3–8 nm (Fig. 3a) and the average particle size is around 5.9 nm (Fig. 3b). The parallel fringe with a spacing of 0.203 nm is corresponding to the (400) plane of NiCo₂O₄ and the parallel fringe with a spacing of 0.235 nm is corresponding to the (111) plane of Au as shown in Fig. 3c,d. The nanoparticles of NiCo₂O₄ and Au contact each other or exit in one particle. The element mapping images (Fig. 4) show homogeneous C (Fig. 4a), Au (Fig. 4b), Ni (Fig. 4c), Co (Fig. 4d) and O (Fig. 4e) distributions in the Au-NiCo₂O₄(wt 1:5)/HPG (Fig. 3a). The element mapping images also shows that the catalyst particles are well dispersed on the surface of HPG.

In order to illustrate the advantages of the Au-NiCo₂O₄(wt 1:5)/HPG electrocatalyst, the OER activity of Au/HPG, NiCo₂O₄/HPG and Au-NiCo₂O₄(wt 1:5)/HPG is evaluated through linear sweep voltammetry (LSV) curves in 0.1 mol L⁻¹ KOH with a sweep rate of 0.001 V s⁻¹ as shown in Fig. 5a. The sum of loading for Au and NiCo₂O₄ on the electrodes was accurately controlled at 0.1 mg cm⁻². The onset potential (E_{onset}) of OER is 0.589 V on the Au/HPG electrode, 0.520 V on the NiCo₂O₄/HPG electrode, and 0.512 V on the Au-NiCo₂O₄(wt 1:5)/HPG electrode. The value of E_{onset} on the Au-NiCo₂O₄(wt 1:5)/HPG electrode is 8 and 77 mV lower than that on the NiCo₂O₄/HPG and Au/HPG electrodes. When the value of E_{onset} is lower, the OER happens easier. OER has the lowest value of E_{onset} on the Au-NiCo₂O₄(wt 1:5)/HPG electrode, so OER happens the most easily on the Au-NiCo₂O₄(wt 1:5)/HPG electrode. The values of current density at 0.7 V ($j_{0.7V}$) are 1.4, 6.8 and 9.1 mA cm⁻² on the Au/HPG, NiCo₂O₄/HPG and Au-NiCo₂O₄(wt 1:5)/HPG electrodes. The Au is poorly active for OER and the NiCo₂O₄ as a major electrocatalyst shows a considerable activity. The current density on the NiCo₂O₄/HPG improves remarkably by adding a small amount of Au. There is a synergistic effect between Au and NiCo₂O₄ because the value of E_{onset} is lower and the current density is higher on the Au-NiCo₂O₄(wt 1:5)/HPG electrode than that on the Au/HPG and NiCo₂O₄/HPG electrodes. The values of E_{onset} and $j_{0.7V}$ of LSV on the Au-NiCo₂O₄/

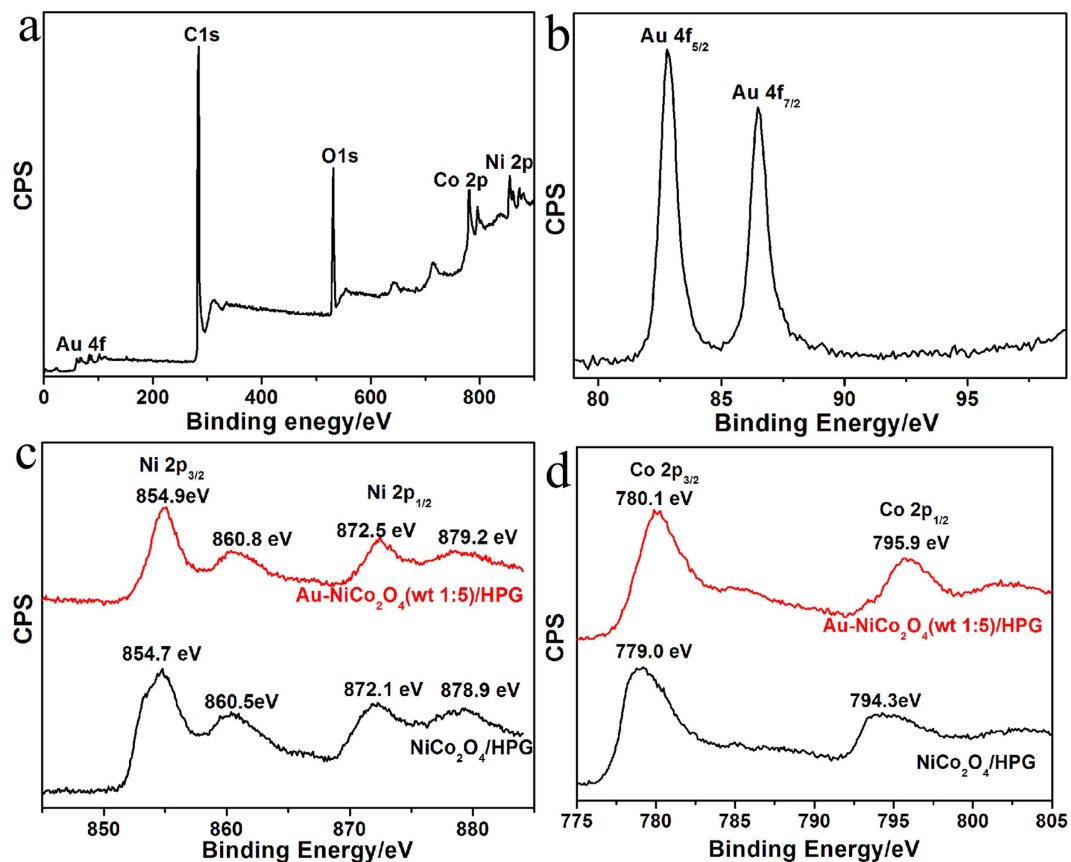


Figure 2. XPS spectra for Au-NiCo₂O₄(wt 1:5)/HPG. (a) survey, (b) Au 4f, (c) Ni 2p and (d) Co 2p.

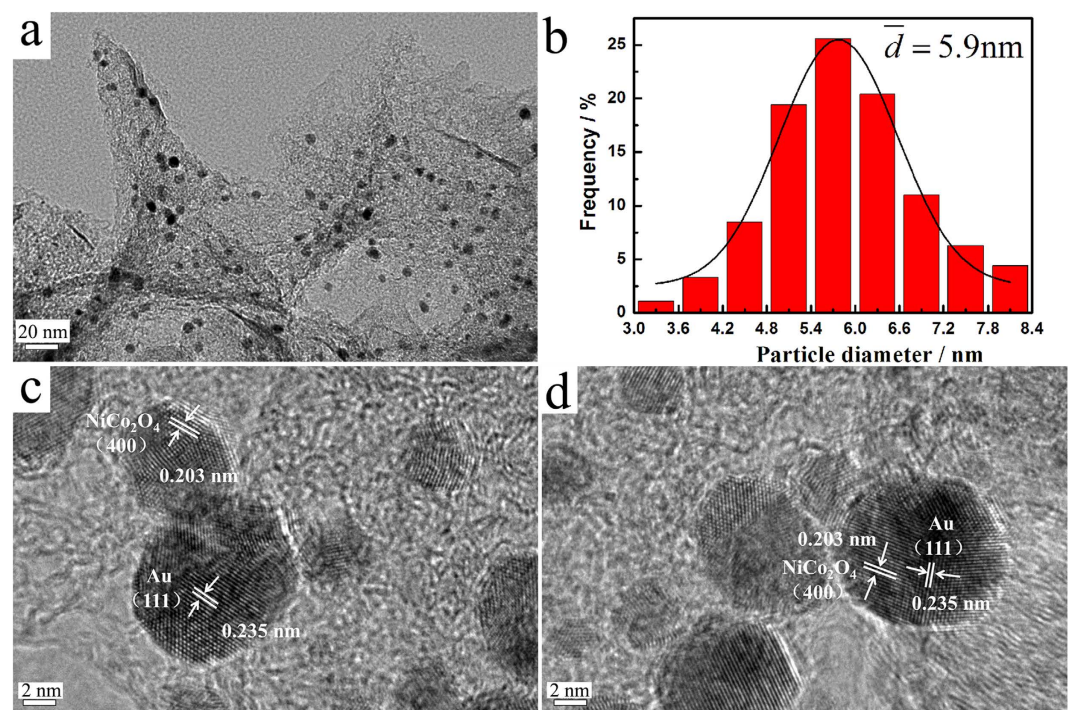


Figure 3. (a) TEM image, (b) particle size distribution histogram, (c) and (d) HRTEM images for Au-NiCo₂O₄(wt 1:5)/HPG.

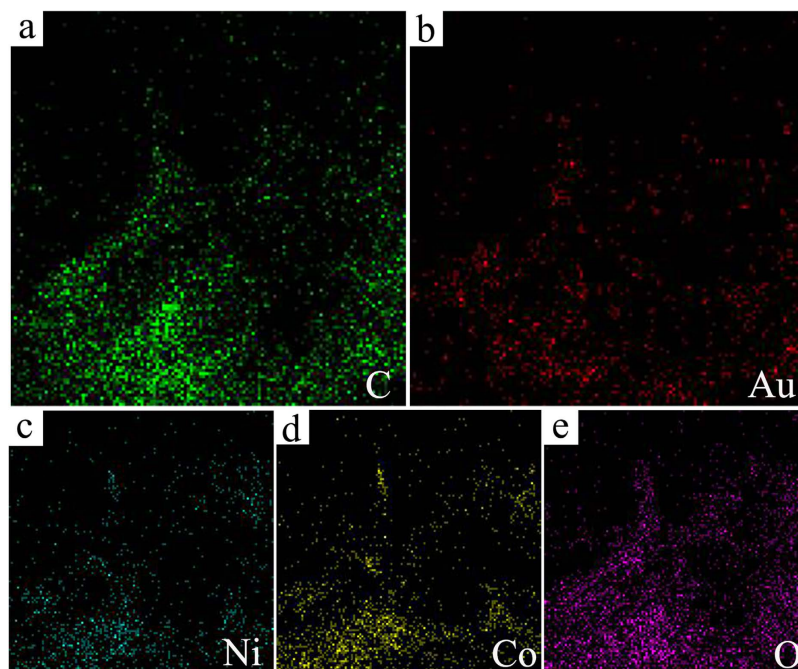


Figure 4. The element mapping images in Au-NiCo₂O₄(wt 1:5)/HPG.

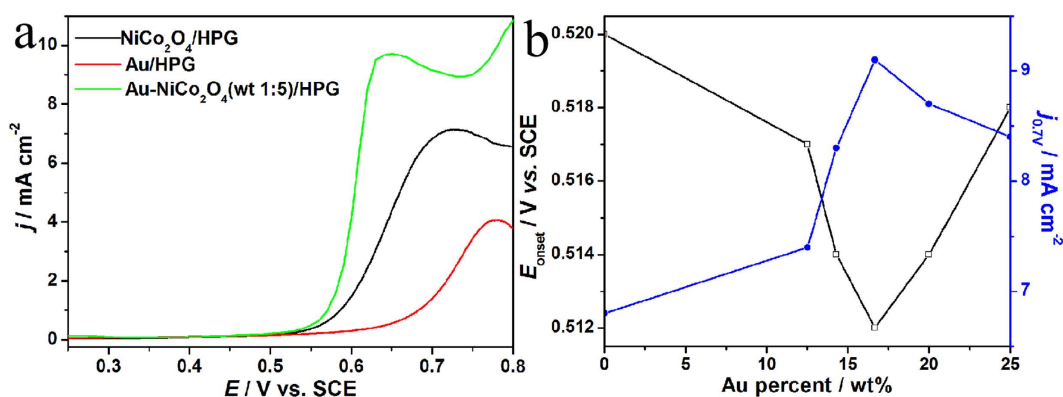


Figure 5. (a) LSV curves on NiCo₂O₄/HPG, Au/HPG and Au-NiCo₂O₄(wt 1:5)/HPG electrodes in 0.1 mol L⁻¹ KOH with a sweep rate of 0.001 V s⁻¹. (b) Plots of E_{onset} and $j_{0.7V}$ in LSV curves as a function of the Au weight percent in Au-NiCo₂O₄/HPG with a total loading of 0.1 mg cm⁻² on the electrodes.

HPG catalysts with different weight percent of Au are compared as shown in Fig. 5b. The sum of loading for Au and NiCo₂O₄ on the electrodes was accurately controlled at 0.1 mg cm⁻² in the Au-NiCo₂O₄/HPG electrode. The lowest value of E_{onset} is 0.512 V when weight ratio for Au to NiCo₂O₄ is 1: 5. The highest value of $j_{0.7V}$ is 9.1 mA cm⁻² when weight ratio for Au to NiCo₂O₄ is 1: 5. It can be seen that Au-NiCo₂O₄(wt 1:5)/HPG shows the highest activity for OER.

The stability of OER on the all electrodes is investigated by chronoamperometry. The chronoamperometry curves for OER in 0.1 mol L⁻¹ KOH solution under a potential of 0.7 V was shown in Fig. 6. The current has a wave because the oxygen evolution on the NiCo₂O₄/HPG and Au-NiCo₂O₄(wt 1:5)/HPG electrodes, which also exhibits that NiCo₂O₄/HPG and Au-NiCo₂O₄(wt 1:5)/HPG catalysts have good activity for OER. When the OER happens, the oxygen bubbles will form on the surface of electrode and block the active sites on the surface of electrode, then the current of water oxidation will decrease. As the reaction is proceeding, oxygen bubbles grow up gradually, and the current of water oxidation also decreases. When the oxygen bubbles grow up enough to separate from the surface of electrode, the current of water oxidation will increase significantly due to the active sites on the surface of electrode are released for further electrochemical reaction. Until the end of the experiment, the oxidation current density on the Au-NiCo₂O₄(wt 1:5)/HPG electrode is 2.2 mA cm⁻², which is 1.7 times as bigger as that on the NiCo₂O₄/HPG electrode (1.3 mA cm⁻²). The result shows that OER on the Au-NiCo₂O₄(wt 1:5)/HPG electrode has a higher current density than that on the NiCo₂O₄/HPG electrode with the same potential.

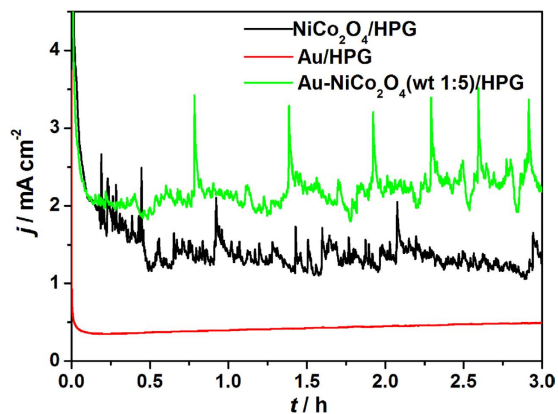


Figure 6. Chronoamperometry curves on NiCo₂O₄/HPG, Au/HPG and Au-NiCo₂O₄(wt 1:5)/HPG electrodes in 0.1 mol L⁻¹ KOH at a potential of 0.7 V.

For further understanding of the intrinsic reaction of the OER performance on the Au-NiCo₂O₄(wt 1:5)/HPG catalyst, the XPS data of pure NiCo₂O₄/HPG and Au-NiCo₂O₄(wt 1:5)/HPG catalysts were compared as shown in Fig. 2c,d. The XPS data show that the binding energy value of Ni 2p has a 0.2~0.4 eV positively shift and that of Co 2p has a 1.1~1.6 eV positively shift after loading with Au. Au is a highly electronegative metal and acts as an electron adsorbate, which generates and stabilizes cobalt and nickel ions at higher oxidation states (e.g. Co⁴⁺ and Ni³⁺). Such a big shift in binding energy will promote the formation of Co⁴⁺ and Ni³⁺ cations. A general understanding is that the Co⁴⁺ and Ni³⁺ cations are the active centres for the OER. The presence of strong electrophilic Co⁴⁺ and Ni³⁺ cations will accelerate to form the OOH via nucleophilic reaction with O⁶². Depend on electrochemical oxidation, progressive oxidation from Co³⁺ to Co⁴⁺ and Ni²⁺ to Ni³⁺ is supposed as rate-limiting step, so the increased population of Co⁴⁺ and Ni³⁺ cations results in enhanced OER performance. Similar results were reported in OER using metal oxides (Co and Ni) with noble metals, where the noble metals generate and stabilize metal ions at higher oxidation states (e.g. Co⁴⁺ and Ni³⁺). Casella and his cooperators⁶³ and Yeo and his cooperators⁶⁴ demonstrated that the growth of Ni hydroxide on a gold electrode favors the oxide states of Ni³⁺ over Ni²⁺. Yeo and his cooperators also noted that the cobalt oxide deposited on Au electrodes exhibits a high occurrence of Co⁴⁺ species on the surface⁵². The enhanced activity was correlated to the electronegativity of noble metals.

In conclusion, a 3D HPG material was synthesized by a one-step ion-exchange/activation combination method using a cheap metal ion exchanged resin as carbon precursor. The 3D HPG material as support for Au-NiCo₂O₄ gives good activity and stability for OER. The 3D HPG material is induced into NiCo₂O₄ as conductive support to increase the specific area and improve the poor conductivity of NiCo₂O₄. The activity of and stability of NiCo₂O₄ significantly are enhanced by a small amount of Au for OER. The Au-NiCo₂O₄(wt 1:5)/HPG shows the highest activity for OER. The value of E_{onset} on the Au-NiCo₂O₄(wt 1:5)/HPG electrode is 8 and 77 mV lower than that on the NiCo₂O₄/HPG and Au/HPG electrodes. The values of $j_{0.7\text{V}}$ are 1.4, 6.8 and 9.1 mA cm⁻² on the Au/HPG, NiCo₂O₄/HPG and Au-NiCo₂O₄(wt 1:5)/HPG electrodes. Benefiting from the synergistic effect, the as-prepared Au-NiCo₂O₄/HPG catalyst shows significantly higher activity and better stability than NiCo₂O₄/HPG catalyst. The XPS data show that the binding energy value of Ni 2p has a 0.2~0.4 eV positively shift and that of Co 2p has a 1.1~1.6 eV positively shift after loading with Au. Au is a highly electronegative metal and acts as an electron adsorbate, which is believed to facilitate to generate and stabilize Co⁴⁺ and Ni³⁺ cations as the active centres for the OER.

Methods

Materials synthesis. The 3D HPG was synthesized by a one-step ion-exchange/activation combination method using a cheap metal ion exchanged resin as carbon precursor according to the Li method⁴⁴. Firstly, the pretreated macroporous acrylic type cation exchange resin was impregnated with targeting ions of nickles in 0.05 mol L⁻¹ nickel acetate solution for 6 h. Secondly, the exchange resin was washed with deionized water and dried at 333 K for 12 h. And then the exchanged resin was added into a 400 mL KOH/ethanol solution which contained 40 g KOH and stirred at 353 K for 6 h. After that, the mixture solution was dried at 343 K for 48 h and smashed by a disintegrator. Finally, the mixture was heated at 1123 K for 2 h in N₂ atmosphere. When the resulted sample cooled down to room temperature, 3 mol L⁻¹ HCl solution was added in it with a specific volume for more than 12 h with magnetic stirring. After that, the sample was repeated washed until the pH value was 7 and dried at 343 K. NiCo₂O₄/HPG was prepared through a typical heterogeneous reaction method. 1 mmol of Ni(NO₃)₂·6H₂O, 2 mmol of Co(NO₃)₂·6H₂O and 0.4811 g HPG were added into H₂O (40 mL), followed by the addition of 5 mmol NH₄F and 12 mmol urea. After being stirred for 1 h, the obtained mixture was transferred to a Teflon-lined stainless steel autoclave and heated to 393 K for 6 h. The resultant precipitate was washed several times with deionized water until the pH of the filtrate became 7 before being dried in a vacuum oven for 12 h. Finally the obtained powder was then annealed at 673 K for 2 h in air. The Au-NiCo₂O₄/HPG electrocatalysts were prepared by reduction of HAuCl₄ solution on the NiCo₂O₄/HPG powders using an excess 0.01 mol L⁻¹ NaBH₄ solution.

Electrode preparation. The electrocatalyst powders were dispersed in deionized water with 5 wt% PTFE (polytetrafluoroethylene) on the surface of a graphite rod with a geometric area of 0.33 cm². The loading of carbon black and PTFE on the electrodes was accurately controlled at 0.23 mg cm⁻² and 0.1 mg cm⁻². The total loading of amount of Au and NiCo₂O₄ in the catalysts on the electrodes was accurately controlled at 0.1 mg cm⁻².

Characterization. XRD was carried out using a Panalytical X'Pert powder X-ray diffractometer with Cu K α radiation ($\lambda = 0.15418$ nm). SEM images were obtained using a Quanta 400 FEG microscope (FEI Company). Transmission electron microscopy (TEM) images were carried out on a JEOL JEM-2010 (JEOL Ltd.). Raman spectroscopic measurements were carried out on a Raman spectrometer (Renishaw Corp., UK) using a He-Ne laser with a wavelength of 514.5 nm. XPS measurements were performed in an ESCALAB 250 spectrometer under vacuum (about 2×10^{-9} mbar). All electrochemical measurements were carried out in 0.1 mol L⁻¹ KOH solution by using the solartron 1287 electrochemical workstation using a standard three-electrode cell at 298 K. Solutions were freshly prepared before each experiment. A platinum foil (3.0 cm²) was used as counter electrode. All the potentials were measured versus a saturated calomel electrode (SCE, 0.241 V versus SHE) electrode. A salt bridge was used between the cell and the reference electrode.

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Author Contributions

C.-W.X. designed the experiments. W.-Y.X. and N.L. performed the experiments, Q.-Y.L. supervised the experiments, C.-W.X. and K.-H.Y. collected and analyzed the data, and wrote the paper. Q.-Y.L. gave suggestion to revise the manuscript. All authors analyzed data, discussed the results, and reviewed the manuscript. All the authors have equal contribution in this work.

Additional Information

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