



Numerical Simulation of S-Shaped Current–Voltage Curves Induced by Electron Traps in Inverted Organic Photovoltaics

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Abstract: Organic photovoltaics (OPVs) differ from their inorganic counterparts because of inevitable electronic disorders and structural heterogeneity. Charge carrier traps are inevitable in organic semiconductors. A common failure mechanism of OPVs is the development of an S-shaped current density–voltage characteristic (*J*-*V* curve). Herein, we focus on investigating the underlying physical mechanism of S-shaped deformation of *J*-*V* curve of the inverted organic photovoltaic devices with bulk-heterojunction, proven by experiments with the n-doped electron extraction layer and numerical simulations are quite consistent with the experimental results. In addition, the open circuit voltage induced by S-kink is exemplified to be enhanced after removing the electron traps in the interlayer by introducing a dopant of cesium carbonate.

Keywords: organic photovoltaics; current-voltage; electron traps; numerical simulation



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1. Introduction

Organic photovoltaics (OPVs) are attractive systems for harvesting photonic energy due to their potential for large-scale production of light-weight, flexible, and cost-effective thin film technologies [1–4]. By employing new materials or interface modification, the performances of OPVs have been improved steadily. As known, the device performance is directly associated with the current–voltage characteristics (*J-V* curves) obtained under illumination. Frequently, the *J-V* curves often show undesirable S-shaped deformation under illumination. This will lead to apparent decrease of fill factor (*FF*) and open-circuit voltage (V_{oc}). Reduced surface recombination for majority carriers [5,6], imbalance of charge carrier mobilities [7–9], dipoles defects at the interface [10,11], and poor quality of the interlayers between the electrode and the active layers [12,13], have been attributed as possible sources. All these phenomena result in carrier accumulation inside the device, and thus redistribution of the electric field.

In recent years, OPVs with the inverted structure have been extensively investigated by using the modified indium tin oxide (ITO) as cathode contact, due to the better longterm ambient stability compared with the regular ones [14]. The ITO cathode of the inverted OPVs is usually modified by n-type materials, such as, cesium carbonate (Cs₂CO₃) and conjugated polymer. For example, the inverted bulk-heterojunction (BHJ) photovoltaic device with an alcohol/water-soluble conjugated polymer, poly[(9,9-bis(3'-(N,Ndimethylamino)propyl)-2,7-fluorene)-alt-2,7-(9,9-dioctylfluorene)] (PFN), was frequently used as interlayer between the active layer and the cathode (ITO) [3]. However, an apparent S-shaped *J*-*V* curve was observed since the performance was very sensitive to the thickness of the poorly conductive PFN. It is supposed that the S-shape curves may be ascribed to imperfect contact caused by electron traps in PFN [15]. This is similar to that the devices containing the interfacial layers of ZnO or TiO_x were commonly dominated by electron traps in the interfacial layers [16–18]. However, investigating the effect of traps by means of electrical simulations is still not well addressed. In this contribution, we used an effective device model with the discrete mid-gap states to study the S-shape effect induced by electron traps in the interlayers between the cathode and the active layer. Besides, the wxAMPS-1D (analysis of microelectronic and photonic structures) program was used to solve the general set of equations, which described the basic phenomena in photovoltaic devices [19]. By using the proposed model, the S-shaped J-V characteristics could be qualitatively reproduced. The simulation results suggest that the discrete acceptor-like gap states can cause a dramatic change in the J-V curve, shaping the standard exponential diode curve to S-shape. Throughout this investigation, we obtained a comprehensive understanding of the underlying physics of traps induced S-shaped *J-V* curves. These findings are important and applicable to the emerging photovoltaic devices because traps are easily formed in the solution-processed devices. Furthermore, to suppress the intrinsic electron traps of PFN by trap-filling method, we introduced Cs_2CO_3 into the PFN interlayer, and consequently the S-shape curve was effectively eliminated, which led to the apparently improved FF and V_{oc} of the solar cell. The shift of V_{oc} observed in the experiment can also be rationally explained with the numerical model.

2. Results

2.1. Device Performance

The experimental *J*-*V* curves of the devices (schematically shown in Figure 1) under illumination with the pristine PFN and Cs_2CO_3 doped PFN (*n*-PFN) are compared in Figure 2. The PFN interlayers with and without Cs_2CO_3 were very uniform (see Figure S1). However, the distinct S-shaped behaviors were observed in the devices with the pristine PFN (see Figures 2, S2 and S3). After doping Cs_2CO_3 into the PFN layer, the V_{oc} was increased from 0. 59 V to 0. 76 V, the *FF* from 49% to 76%, and the short-circuit current density (J_{sc}) from 13.0 mA/cm² to 13.9 mA/cm². As the initial S-shape characteristics in our OPVs can be eliminated by adding Cs_2CO_3 into PFN, the S-shape *J*-*V* characteristics should originate from the PFN layer itself. It is known that traps can be either extrinsic or intrinsic. Intrinsic traps can arise from the energetic and structural disorders in organic semiconductors and are more difficult to be eliminated compared with extrinsic traps. Yan et al. found that trap density can be effectively decreased by trap-filling with 2,3,5,6-tetrafluoro-7,7,8,8-tetracyanoquinodim-ethane(F_4 -TCNQ) in the donor polymer [20]. Thus, we infer that, after introducing Cs_2CO_3 into PFN layer, the electron traps may be passivated by the addition of electrons introduced by molecular n-doping (see Figure 3) [21–23].



Figure 1. Schematic diagrams of the device structure (**a**) and energy level alignment along with the chemical structure of PFN (**b**). The energy levels of organic materials were mainly cited from [15].



Figure 2. Measured *J*-*V* characteristics of the devices with the pristine PFN and Cs_2CO_3 doped PFN (*n*-PFN) under illumination.



Figure 3. (**a**) Schematic diagram of trapping of an electron by a gap state. (**b**) Gap state filled by the additional dopant.

2.2. Numerical Model

In order to understand the electrical behavior in organic devices and interpret the experimental data, we need the flexible device models. The common device models based on a drift-diffusion approach have been successfully used to describe the electrical behavior of OPV [24–26]. Three coupled differential equations must be solved in a macroscopic simulation, which is based on the Poisson equation (see below).

$$\frac{d^2\varphi}{dx^2} = \frac{q}{\varepsilon_0\varepsilon_r} \left[n(x) - p(x) + N_A^-(x) - N_D^+(x) + n_t(x) - p_t(x) \right]$$
(1)

where in the metal–insulator–metal model, $N_A^-(x) = N_D^+(x) = 0$.

The drift diffusion equations respectively for the electrons and holes are listed below.

$$-\frac{1}{q}\frac{dJ_n(x)}{dx} = -D_n\frac{d^2n(x)}{dx^2} - F\mu_n\frac{dn(x)}{dx} = G(x) - R(x, n, p)$$
(2)

$$\frac{1}{q}\frac{dJ_p(x)}{dx} = -D_p\frac{d^2p(x)}{dx^2} + F\mu_p\frac{dp(x)}{dx} = G(x) - R(x, n, p)$$
(3)

In the above equations, q is the elementary charge, φ is the potential barrier, ε_0 is the permeability in free space, ε_r is the relative dielectric permeability, n and p are the concentrations of the free electrons and holes, n_t and p_t are the concentrations of trapped electrons and holes, and N_D^+ and N_A^- are the concentrations of the ionized donors and acceptors, respectively. D_n and D_p are the diffusion coefficients of electrons and holes, respectively. J_n and J_p are the electron and hole current densities. F denotes the electrical field, G and R are the generation and recombination rates of free charge carriers, and μ_n and μ_p are the charge carrier mobilities, respectively for electrons and holes. Here, we only consider Shockley–Read–Hall recombination. Therefore, we find that bimolecular recombination merely changes the S-Kink in the simulated *J*-*V* curves.

The boundary condition for the electrical potential is described as below.

$$p(d) - \varphi(0) = V_{bi} - V \tag{4}$$

 $\varphi(0)$ and $\varphi(d)$ are the energy barriers at the cathode contact and the anode contact, respectively. V_{bi} is the built-in voltage and V is the applied voltage.

$$n(0) = N_c \exp\left(-\frac{\varphi_1}{kT}\right) \tag{5}$$

$$p(0) = N_v \exp\left(-\frac{E_{gap} - \varphi_1}{kT}\right) \tag{6}$$

$$n(d) = N_c \exp\left(-\frac{E_{gap} - \varphi_2}{kT}\right) \tag{7}$$

$$p(d) = N_v \exp\left(-\frac{\varphi_2}{kT}\right) \tag{8}$$

where φ_1 and φ_2 are injection barriers respectively for electrons and holes at the cathode and anode contacts. n(0) and p(0) represent the density of charge carriers at the interface of between the active layer and the interlayer. Meanwhile, n(d) and p(d) denotes the density of charge carriers at the interface between the active layer and the anode if no interlayer is introduced and d is the thickness of the active layer. E_{gap} is the effective energy offset inside the active layer. N_c and N_v are the intrinsic densities of states at the conduction and valence bands, respectively. *k* is the Boltzmann constant and *T* is the absolute temperature. Using Equations (1)–(8), we can calculate the charge carrier concentrations and electrical potential. By integrating Equations (2) and (3), *J-V* curve under illumination or in the dark can be simulated. Here, we use the wxAMPS-1D program, an open software for solar-cell simulation software that is an enhanced version of AMPS-1D [27,28], to solve the above set of equations. WxAMPS-1D consists of the optimized algorithms, including Gummel and Newton methods, and the simulation stability is enhanced. Some of the parameters are adapted from the reference data available at the open source [29]. It is also possible to customize the data which are fit to the devices constructed in this investigation. For simplicity, we adapted the buffer layer for qualitative study of the S-shape effect in the *J*-V curves. The device structure diagram used in our simulation is shown in Figure 4, and the parameters used in the simulation are listed in Table 1. The numerically simulated *J*-V curves with the discrete acceptor-like gap states (a concentration of 10^{18} cm⁻³) and without any gap states in the buffer layer were illustrated in Figure 5. With the parameters given in Table 1, we retrieved a V_{oc} of 0.59 V as for the device with traps, which was increased to 0.76 V as for the device without traps. The calculated J_{sc} was slightly increased from 14.6 to 15.5 mA/cm², and the FF increased from 49% to 70%, respectively, when no trapped electrons were involved in the electron extraction layer.



Figure 4. Schematic diagram of energy level alignment with electron traps and injection barriers used in the numerical simulation.

Table 1. Parameters used in the simulation.

Parameter	Symbol	Numerical Value
Effective band gap	E_g^{DA}	1.2 eV
Electron mobility	μ_n	$2 imes 10^{-3} m cm^2 V^{-1} s^{-1}$
Hole mobility	μ_p	$2 imes 10^{-4}~{ m cm^2V^{-1}s^{-1}}$
Relative static permittivity	ε_r	3.4
Injection barrier at cathode side	$arphi_1$	0.1 eV
Injection barrier at anode side	φ_2	0 eV
Density of states at conduction band	N_C	$2 imes 10^{18}~\mathrm{cm}^{-3}$
Density of states at valence band	N_V	$2 imes10^{19}~\mathrm{cm}^{-3}$
Distance to cathode	L_1	$10^{-6} { m cm}$
Distance to anode	L_2	$10^{-5} { m cm}$
Surface recombination velocity of electrons (cathode side)	S _{n,cat}	$10^{7} { m cm s^{-1}}$
Surface recombination velocity of holes (cathode side)	S _{p,cat}	$10^{7} { m cm s^{-1}}$
Surface recombination velocity of electrons (anode side)	S _{n,an}	$10^7 { m ~cm~s^{-1}}$
Surface recombination velocity of holes (anode side)	S _{p,an}	$10^{7} { m cm s^{-1}}$
Discrete acceptor-like gap state density	$n_{t,a}$	$10^{18} { m cm}^{-3}$
Capte cross section of electrons	σ_n ,	$10^{-10} { m cm}^2$
Capture cross section of holes	σ_p	10^{-12} cm^2



Figure 5. Simulated *J-V* characteristics of the devices with and without traps.

3. Discussion

The deformation of the *J*-*V* curves originates from the internal energetic band structure and charge carrier distributions. The energetic band structures of the devices with/without acceptor-like defect states in the interlayer are shown in Figure 6a,b, respectively for the case under the applied voltages of 0.2 V and 0.7 V. The energy levels of the lowest unoccupied molecular orbital (LUMO) (E_{LUMO}) and the highest occupied molecular orbital HOMO (E_{HOMO}) of the device with traps are shown in thick solid red lines, while E_{LUMO} and E_{HOMO} of device without any traps are shown in thick solid black lines. The quasi-Fermi levels E_{Fn} and E_{Fp} of the device with traps are shown in thin dash magenta lines and those of the device without traps are indicated in thin dash blue lines.



Figure 6. Energy distributions of the devices with/without traps (a) under 0.2 V and (b) under 0.7 V.

Figure 7a shows the carrier distribution of the device with traps at 0.2 V and 0.7 V, respectively. The black solid and dash lines indicate the electron and hole densities at 0.2 V, respectively. In contrast, the red solid and dash lines indicate the electron and hole densities at 0.7 V, respectively. Besides, the density of acceptor-like states that have trapped electrons are shown in magenta and blue lines respectively for the cases under 0.2 V and 0.7 V. For comparison, carrier distributions of the device without traps is shown in Figure 7b. As for the device with traps under the voltages below V_{oc} , i.e., under extracting condition, the generated charges are extracted from the active layer into the electrodes, a region of the trapped space charge is formed near the cathode. As the applied voltage approaches to V_{oc} , the effective internal voltage dropping over the trapped space charge region starts to decrease and thus this negatively charged region begins to narrow down. The rate of the traps to capture electrons is proportional to the number of free electrons, and the rate of the captured holes is proportional to the number of holes [30,31].



Figure 7. (a) Carrier densities of the device with traps at 0.2 V and 0.7 V. (b) Carrier distribution of the device without traps.

When the net rate of the captured holes is larger than that of electrons, the density of the acceptor-like states that have trapped electrons will decrease, which will cause the narrowed trapped space charge region. Therefore, we can observe that the hole density in the interlayer with traps under 0.7 V is increased while the electron density is reduced, in contrast to that of the device under 0.2 V, which is consistent with the narrowing of the trapped space charge region. At the interface between the active layer and the interlayer, fewer trapped space charges, fewer free electrons and more holes are observed. Consequently, the electric field across the interface between the active layer and the interlayer is reduced and will even change signs, as shown in Figure 8. The transition between both cases creates an S-kink around V_{oc} .

As shown in Figures 2 and 5, the S-shape J-V curves lead to the loss of V_{oc} . For the device without traps under the V_{oc} of 0.76 V, the energy band diagrams are calculated and shown in Figure 9a. Moreover, Figure 9b illustrates the case of the device with traps.



Figure 8. Electric field distribution of device with traps under 0.2 V and 0.7 V.



Figure 9. Calculated energy alignment referred to V_{oc} of the devices (a) without and (b) with traps.

Utilizing the metal–insulator–metal concept, we can express the V_{OC} as below [32].

$$qV_{OC} = E_g^{DA} - \max(\varphi_2, \delta_p) - \max(\varphi_1, \delta_n)$$
(9)

where $\delta_n = E_{LUMO} - E_{Fn}$ ($\delta_p = E_{Fp} - E_{HOMO}$) at x_{max} which corresponds to the case in the active layer where the product *np* is the largest. According to Figure 9, the *V*_{OC} can be given by

$$qV_{OC} = E_g^{DA} - \delta_p - \varphi_1 \tag{10}$$

As in the case of the acceptor-like traps in the interlayer, the magnitude of the band bending from the immobile trapped space charge can be estimated as below.

Δ

$$=\frac{q^2 n_t d_c^2}{2\varepsilon_0 \varepsilon_r} \tag{11}$$

where n_t is the trap density mentioned previously, and d_c is the thickness of the negatively charged region. Thus, the effective electron injection barrier ($\varphi_{1,eff}$) can be rewritten as $\varphi_{1,eff} = \varphi_1 + \Delta$ and the effective built-in voltage ($V_{bi,eff}$) is changed to $V_{bi,eff} = qV_{bi} - \Delta/q$. From the energy band diagram of Figure 9b, we can estimate the thickness $d_c \approx 8$ nm. Therefore, according to Equation (11), $\Delta \approx 0.17$ eV. Consequently, the shift of V_{oc} between the devices with and without traps will be equal to -0.17 V, according to Equation (10), which is well consistent with the experiments.

4. Materials and Methods

The device structure was ITO/PFN/PTB7:PC₇₁BM/MoO_x/Ag. The donor material PTB7—i.e., poly(thieno [3,4-*b*]-thiophene benzodithiophene)—and the acceptor material PC₇₁BM, i.e., [6,6]-phenyl-C₇₁-butyric acid methyl ester, were purchased from Lumtec and used as received. The interlayer material PFN (PLT105081B), purchased from Xi'an P-OLED Corp., was dissolved in methanol in the presence of a small amount of acetic acid (2 μ L/mL) and its solution (2 mg/mL) was spin-coated on top of the prepatterned ITO substrate at 6000 r.p.m. for 10 min. As for the device with the so-called n-doping, PFN:Cs₂CO₃ (5:1, w/w) layer, ca. 10 nm, were spin-coated from the solution in methanol (2 mg/mL) at 6000 r.p.m. for 10 min. The PTB7:PC₇₁BM (1:1.5, w/w) active blend layer (80 nm) was prepared by the solution (25 mg/mL) with the mixed solvent of chlorobenzene:1,8-Diiodoctane

9 of 10

(97:3 by volume) spin-coated at 1200 r.p.m. for 2 min. A 10 nm-thick M_0O_x and 50 nm-thick Ag were thermally evaporated, subsequently.

5. Conclusions

In summary, we investigated the trapped electrons of the interfacial layer in the inverted organic photovoltaics with the common bulk heterojunction, which contributed the S-shaped *J-V* curves verified by both experiments and numerical simulation. By analyzing the calculated energetic band structure and carrier distributions, we assign the changing of the trapped space charge region which will induce a change of the electric field distribution as the origin of the characteristic deformation. The S-kink led to the loss of V_{oc} can be reasonably explained. A thorough understanding of the electron trapping effects is critical for designing high-performance organic photovoltaics where traps are very commonly found.

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