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## Ammonia Storage by Reversible Host–Guest Site Exchange in a Robust Metal–Organic Framework

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Abstract: MFM-300(Al) shows reversible uptake of NH<sub>3</sub>  $(15.7 \text{ mmol } g^{-1} \text{ at } 273 \text{ K and } 1.0 \text{ bar}) \text{ over } 50 \text{ cycles with an}$ exceptional packing density of  $0.62 \text{ g cm}^{-3}$  at 293 K. In situ neutron powder diffraction and synchrotron FTIR microspectroscopy on ND<sub>3</sub>@MFM-300(Al) confirms reversible H/D site exchange between the adsorbent and adsorbate, representing a new type of adsorption interaction.

Approximately 150 million tonnes of NH<sub>3</sub> (ammonia) was produced in 2017, making it one of the most important base chemicals in the world.<sup>[1]</sup> As an energy resource, NH<sub>3</sub> has an excellent hydrogen density with the hydrogen density of liquid NH<sub>3</sub> and a 200-bar H<sub>2</sub> cylinder being 108 and 14 gL<sup>-1</sup>, respectively. It also has a high octane number and a low flame temperature, and its combustion to N<sub>2</sub> and H<sub>2</sub>O is potentially environmentally benign.<sup>[2]</sup> As a transportation fuel, NH<sub>3</sub> can be incorporated into existing technologies such as internal combustion engines and gas turbines, but it also holds promise for renewable energy generation through H<sub>2</sub> and NH<sub>3</sub> fuel cells.<sup>[2,3]</sup> However, under ambient conditions, NH<sub>3</sub> is a toxic and highly corrosive gas making it difficult to handle and store. For it to be transported in an energy efficient manner, NH<sub>3</sub> is often liquefied to maximise its storage density through pipelines and in storage tanks.<sup>[4]</sup> Liquid NH<sub>3</sub> can be stored at ambient pressure at 240 K, whereas smaller quantities tend to

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be stored in pressurised vessels at 16-18 bar.<sup>[4a]</sup> Reducing or eliminating the energy consumption involved in NH<sub>3</sub> storage is highly desirable, but any prospective material needs to show a high packing density that maximises NH<sub>3</sub> storage amount within a given volume. It also needs to be capable of undergoing multiple cycles whilst retaining its adsorption capabilities. Zeolites, activated carbons, mesoporous silica and organic polymers have been tested for NH<sub>3</sub> storage; however, these materials generally show low and/or irreversible uptakes.<sup>[5]</sup>

Constructed from metal ions and organic ligands, porous metal-organic frameworks (MOFs) are emerging solid sorbents for a wide variety of substrates.<sup>[6]</sup> The highly porous nature of MOFs, coupled with their large surface areas (up to  $7000 \text{ m}^2\text{g}^{-1}$ ) and high concentration of binding sites, makes them promising candidates for gas storage. Indeed, extensive research efforts have been devoted to studying their capability to serve as H<sub>2</sub>, CH<sub>4</sub> and CO<sub>2</sub> stores.<sup>[7]</sup> However, the potential of utilising MOFs for adsorption of corrosive and toxic gases remain poorly explored,<sup>[8]</sup> primarily due to the limited stability of many MOFs. Recently, a number of stable MOFs have been tested for NH<sub>3</sub> adsorption with a majority showing structural degradation on exposure or desorption.<sup>[9]</sup> Here, we examine the adsorption, binding and reversible storage of NH<sub>3</sub> in ultra-stable MFM-300(Al). At 273 K and 1.0 bar, MFM-300(Al) shows an NH<sub>3</sub> uptake of 15.7 mmol  $g^{-1}$ {corresponding to a formula of  $[Al_2(OH)_2(L)](NH_3)_{65}$  leading to a packing density of  $0.70 \text{ g cm}^{-3}$ , comparable to the liquid density of NH<sub>3</sub> (0.681 gcm<sup>-3</sup>) at 240 K. At 293 K, MFM-300(Al) also exhibits an impressive packing density of  $NH_3$  at 0.62 g cm<sup>-3</sup>, higher than leading MOFs and other stateof-the-art porous materials, with an uptake of  $13.9 \text{ mmol g}^{-1}$ . Importantly, the NH<sub>3</sub> uptake in MFM-300(Al) is fully reversible under conventional pressure-swing conditions, and no loss of storage capacity was observed after 50 cycles of adsorption-desorption at 293 K. We have also employed in situ neutron diffraction, high resolution synchrotron X-ray diffraction and micro-FTIR spectroscopy for determination of the host-guest binding interaction at the molecular level. We have characterised a novel reversible host-guest site exchange mechanism that is intermediate between traditional physisorption and chemisorption.

MFM-300(Al), [Al<sub>2</sub>(OH)<sub>2</sub>(L)], comprises of [AlO<sub>4</sub>(OH)<sub>2</sub>] moieties bridged by 3,3',5,5'-biphenyl-tetracarboxylic acid (H<sub>4</sub>L) to afford a rigid "wine-rack" framework with channels of  $\approx 6.5$  Å in diameter and hydroxyl groups pointing directly into the pore.<sup>[10]</sup> MFM-300(Al) has demonstrated exceptional adsorption and stability towards corrosive SO<sub>2</sub> and NO<sub>2</sub>.<sup>[10,11]</sup>



## **Communications**





*Figure 1.* a) Adsorption isotherms of NH<sub>3</sub> in MFM-300(Al) at 273–303 K. b) Repeated cycling of NH<sub>3</sub> up to 50 cycles in MFM-300(Al) at 293 K with percentage adsorption (red) as a function of cycle number and the corresponding desorption (blue) reached for each cycle under pressure-swing conditions. c) PXRD of in MFM-300(Al) cycling with NH<sub>3</sub>. d) Comparison of densities of stored NH<sub>3</sub> in MCM-41,<sup>[Sa]</sup> 13X zeolite,<sup>[Sa]</sup> Amberlyst 15,<sup>[Sa]</sup> MFM-300(Al), COF-10,<sup>[12]</sup> Co<sub>2</sub>Cl<sub>2</sub>BBTA<sup>[9c]</sup> and COHCC<sup>[9a]</sup> plotted against their respective gravimetric uptake compared with the density of liquid and solid NH<sub>3</sub>.

The adsorption isotherms for  $NH_3$  in MFM-300(Al) were measured at 273–303 K, where a total uptake of 15.7 and 13.9 mmol g<sup>-1</sup> was recorded at 273 and 293 K, respectively, at 1.0 bar (Figure 1). Due to the reactive nature of  $NH_3$ , many materials that have been investigated previously are unstable storage. [Co<sub>2</sub>Cl<sub>2</sub>BBTA] and COF-10 both show a loss of NH<sub>3</sub> uptake after repeated cycling of 5.6% and 4.5%, respectively, and for complete regeneration both require heating to 200°C under dynamic vacuum.<sup>[9c,12]</sup> In contrast, MFM-300(Al) is able to repeatedly adsorb from regenerated

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exposure and suffer to significant degradation (Table S7). CoHCC {Co[Co- $(CN)_{6}]_{0.60}\},^{[9a]}$  $[Co_2Cl_2-$ (BBTA)] [BBTA = 1H, 5Hbenzo(1,2-d),(4,5-d')bistriazole)],<sup>[9c]</sup> COF-10,<sup>[12]</sup> Amberlyst 15,<sup>[5a]</sup> 13X zeolite<sup>[5a]</sup> and MCM-41<sup>[5a]</sup> are the best-performing porous materials from their respective categories that are stable to repeated NH<sub>3</sub> exposure, with reported uptakes of  $21.9 \text{ mmol g}^{-1}, 18.0 \text{ mmol g}^{-1},$  $15.0 \text{ mmol g}^{-1}, 11.3 \text{ mmol g}^{-1}$  $9.30 \text{ mmol g}^{-1}$ and  $7.90 \text{ mmol g}^{-1}$ , respectively, under ambient conditions. MFM-300(Al), while not achieving the highest gravimetric uptake, supersedes all aforementioned materials in terms of the packing density of NH<sub>3</sub>,  $0.62 \text{ g cm}^{-3}$  at 293 K. Interestingly, MFM-300(Al) can liquefy NH<sub>3</sub> above its boiling point (243 K) by of reaching a density  $0.70 \text{ g cm}^{-3}$  at 273 K, eliminating the need for energy intensive liquefaction for storage.

Significantly, adsorption of NH<sub>3</sub> in MFM-300(Al) is highly reversible with no loss of uptake capacity or crystallinity, and no broadening of Bragg peaks (Figure S4 in the Supporting Information) after 50 observed was adsorption-desorption cycles at 293 K (Figure 1). CoHCC is able to undergo four cycles with no loss of NH<sub>3</sub> uptake; however, this necessitates reactivation at 150°C under dynamic vacuum for 24 hrs.<sup>[9a]</sup> Furthermore, each isotherm adsorption point requires 170 minutes for equilibration, thus restricting its potential for portable NH<sub>3</sub> MOF to complete saturation in 6 mins and completely desorb (saturation to full release) within  $\approx 13.5$  mins using a standard pressure-swing over 50 cycles, making it an ideal candidate for NH<sub>3</sub> storage.

The binding domains for adsorbed NH<sub>3</sub> molecules within MFM-300(Al) have been elucidated by in situ neutron powder diffraction (NPD). Structural analysis via Rietveld refinement of NPD data for 1.5 ND<sub>3</sub>/Al-loaded MFM-300(Al) identified three distinct binding sites (I, II and III) in  $[Al_2(OH)_2(L)](ND_3)_3$  (Figure 2). Site I [occupancy = 0.736-(6)] is 1.76(2) Å {O  $\rightarrow$  N1 = 2.84(1) Å} from the bridging  $\mu_2$ -OH moiety in a hydrogen bonding pocket formed by the pore wall (Figure 2). The sub 2 Å distance and the high isosteric heat of adsorption  $(Q_{st} > 40 \text{ kJ mol}^{-1})$  are indicative of a strong binding mode being present between NH3 and MFM-300(Al).<sup>[13]</sup> Sites II [occupancy = 0.236(3)] and III [occupancy = 0.213(5)] lie at a distance of 2.68(1) Å {N1}  $\rightarrow$ N2 = 3.63(1) Å and  $2.29(3) \text{ Å} \{N2 \rightarrow N3 = 3.09(1) \text{ Å}\}$  from site I and site II, respectively. This cooperative network of ND<sub>3</sub> molecules propagates down the length of the 1D channel, anchored in place by site I. Bond distances for sites II…III and a slightly lengthened site I…II are similar to



**Figure 2.** View of the structure of  $1.5ND_3/Al-loaded MFM-300(Al)$  determined by in situ NPD studies. a) View down the c-axis; b) side on view of three binding sites in relation to the OH functionality; c) view of binding of ND<sub>3</sub> to the framework. Framework hydrogen  $\rightarrow$  Site I (orange) = 1.76(2) Å {O  $\rightarrow$  N1 = 2.84(1) Å}. Linker ring  $\rightarrow$  Site I (yellow) = 3.841(7) Å. Linker H1  $\rightarrow$  Site I (red) = 2.94(1) Å. Linker H2  $\rightarrow$  Site I (indigo) = 2.994(9) Å. Site I  $\rightarrow$  Site II (green) = 2.68(1) Å {N1  $\rightarrow$  N2 = 3.63(1) Å}. Site II  $\rightarrow$  Site III (purple) = 2.29(3) Å {N2  $\rightarrow$  N3 = 3.09(1) Å}.

a typical inter-molecular bond between ND<sub>3</sub> molecules in the solid state at 2 K [N···D = 2.357(2) Å],<sup>[14]</sup> whereas the bond between the framework  $\mu_2$ -OH and site I is significantly shorter. The structure for ND<sub>3</sub>-loaded MFM-300(Al) has also been determined at a loading of 0.5 ND<sub>3</sub>/Al and 1.0ND<sub>3</sub>/Al and these have shown similar binding sites as discussed above. With increased loading from 0.5 ND<sub>3</sub>/Al to 1.5 ND<sub>3</sub>/Al, we observed an overall shortening of the framework  $\mu_2$ -OH···site I and sites I···II and an increase in the site II···III.

Refinement of the NPD data for ND<sub>3</sub>-loaded MFM-300(Al) revealed an interesting observation: as the loading of ND<sub>3</sub>/ $\mu_2$ -OH was increased, the hydrogen on hydroxyl groups underwent a reversible site exchange with the deuterium from guest ND<sub>3</sub> molecules residing at Site I in the pore. This exchange is very distinct in the analysis of NPD data owing to the significant difference on neutron scattering of hydrogen and deuterium. We noted as the loading of ND<sub>3</sub> increased from 0 to 0.5 ND<sub>3</sub>/ $\mu_2$ -OH, the occupancy of the hydrogen of the  $\mu_2$ -OH group decreased from 1.0 to 0.794(7) (Table S6). As the loading of ND<sub>3</sub> was further increased to 1.0 and 1.5 ND<sub>3</sub>/ $\mu_2$ -OH, the occupancy of the hydrogen on the  $\mu_2$ -OH group decreased to 0.416(10) and the site was replaced by a deuterium to give a  $\mu_2$ -OD [D occupancy=0.468(9) and 0.584(10), respectively].

We sought to examine the reversibility of this H–D exchange via in situ synchrotron FTIR micro-spectroscopy by monitoring the vO-H stretching vibration at 3692 cm<sup>-1</sup> as a function of ND<sub>3</sub> loading at 293 K (Figure 3). A rapid depletion of this band was observed on adsorption of ND<sub>3</sub> under flow conditions, accompanied by the growth of a new band at 2720 cm<sup>-1</sup> assigned to the vO-D stretching mode, thus confirming  $H \rightarrow D$  exchange. Once the  $H \rightarrow D$  exchange is completed to give  $[Al_2(OD)_2(L)]$  with no residual O-H stretching band observed, the material was charged with a flow of NH<sub>3</sub> at 293 K. Interestingly, the vO-D stretching



**Figure 3.** Reversible switching of framework hydroxyl hydrogen from  $H \rightarrow D \rightarrow H$ . The v(OH) stretching vibration is at 3692 cm<sup>-1</sup> and the v(OD) stretching vibration at 2720 cm<sup>-1</sup>. a) Bare MFM-300(Al) (black); b) ND<sub>3</sub> exposed MFM-300(Al) (red) and c) regenerated hydroxyl H functionalised MFM-300(Al) after exposure of (b) to NH<sub>3</sub> (blue).

band at 2720 cm<sup>-1</sup> disappeared and the vO-H band at  $3692 \text{ cm}^{-1}$  returned, indicating that the H $\rightarrow$ D exchange is completely reversible. It is worth noting that such H–D reversible exchange does not lead to any detectable structural degradation of the long range order of the framework (Figure S8). Traditionally, chemisorption and physisorption is distinguished on the basis of host–guest binding interaction and the formation of adsorbate–adsorbent bonds at the interface. Significantly, the adsorption of ND<sub>3</sub> in MFM-300(Al) revealed a new type of adsorption where adsorbent and adsorbates undergo rapid site-exchange via reversible formation and cleavage of O–H and O–D chemical bonds.

In summary, MFM-300(Al) shows excellent  $NH_3$  adsorption capacity with the intrinsic ability to achieve liquefaction of  $NH_3$  under near ambient conditions, and outperforms the state-of-the-art porous materials in terms of  $NH_3$  packing density, reversibility and stability. MFM-300(Al) offers unparalleled repeatable uptake characteristics and, coupled with the pseudo-chemisorption binding mechanism, is a promising  $NH_3$  storage material for portable applications.

CCDC 1856081, 1856082, 1856083 and 1856084 contain the supplementary crystallographic data for this paper. These data can be obtained free of charge from The Cambridge Crystallographic Data Centre. Correspondence and requests for materials should be addressed to S.Y. (Sihai.Yang@manchester.ac.uk) and M.S. (M.Schroder@manchester.ac.uk).

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## **Conflict of interest**

The authors declare no conflict of interest.

**Keywords:** ammonia  $\cdot$  metal–organic framework  $\cdot$  MFM-300  $\cdot$  neutron diffraction  $\cdot$  storage materials

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- [1] J. A. Ober, *Mineral Commodity Summaries*, Reston, VA, **2018**, p. 204.
- [2] A. Klerke, C. H. Christensen, J. K. Nørskov, T. Vegge, J. Mater. Chem. 2008, 18, 2304–2310.

- [3] a) F. R. Westlye, A. Ivarsson, J. Schramm, Fuel 2013, 111, 239–247; b) D. R. Jenkins, Hypersonics before the shuttle: A concise history of the X-15 research airplane, NASA, 2000; c) F. Schüth, R. Palkovits, R. Schlögl, D. S. Su, Energy Environ. Sci. 2012, 5, 6278–6289; d) R. Lan, J. T. S. Irvine, S. Tao, Int. J. Hydrogen Energy 2012, 37, 1482–1494; e) T. Per, H. Christian, A. Serina, Environ. Prog. Sustainable Energy 2014, 33, 1290–1297.
- [4] a) "Ammonia, 1. Introduction": M. Appl in Ullmann's Encyclopedia of Industrial Chemistry, Wiley, Hoboken, 2011; b) G. Thomas, Potential roles of ammonia in a hydrogen economy, U.S. Department of Energy, 2006.
- [5] a) J. Helminen, J. Helenius, E. Paatero, I. Turunen, J. Chem. Eng. Data 2001, 46, 391–399; b) J. F. Van Humbeck, T. M. McDonald, X. Jing, B. M. Wiers, G. Zhu, J. R. Long, J. Am. Chem. Soc. 2014, 136, 2432–2440.
- [6] S. Kitagawa, Chem. Soc. Rev. 2014, 43, 5415-5418.
- [7] a) Y. He, W. Zhou, G. Qian, B. Chen, *Chem. Soc. Rev.* 2014, 43, 5657–5678; b) L. J. Murray, M. Dincă, J. R. Long, *Chem. Soc. Rev.* 2009, 38, 1294–1314; c) K. Sumida, D. L. Rogow, J. A. Mason, T. M. McDonald, E. D. Bloch, Z. R. Herm, T. H. Bae, J. R. Long, *Chem. Rev.* 2012, 112, 724–781.
- [8] E. Barea, C. Montoro, J. A. Navarro, Chem. Soc. Rev. 2014, 43, 5419-5430.
- [9] a) A. Takahashi, H. Tanaka, D. Parajuli, T. Nakamura, K. Minami, Y. Sugiyama, Y. Hakuta, S. Ohkoshi, T. Kawamoto, J. Am. Chem. Soc. 2016, 138, 6376–6379; b) A. J. Rieth, Y. Tulchinsky, M. Dincă, J. Am. Chem. Soc. 2016, 138, 9401–9404; c) A. J. Rieth, M. Dincă, J. Am. Chem. Soc. 2018, 140, 3461–3466.
- [10] a) S. Yang, J. Sun, A. J. Ramirez-Cuesta, S. K. Callear, W. I. David, D. P. Anderson, R. Newby, A. J. Blake, J. E. Parker, C. C. Tang, M. Schröder, *Nat. Chem.* 2012, *4*, 887–894; b) S. Yang, A. J. Ramirez-Cuesta, R. Newby, V. Garcia-Sakai, P. Manuel, S. K. Callear, S. I. Campbell, C. C. Tang, M. Schröder, *Nat. Chem.* 2015, *8*, 121–129; c) X. Han, H. G. W. Godfrey, L. Briggs, A. J. Davies, Y. Cheng, L. L. Daemen, A. M. Sheveleva, F. Tuna, E. J. L. McInnes, J. Sun, C. Drathen, M. W. George, A. J. Ramirez-Cuesta, K. M. Thomas, S. Yang, M. Schröder, *Nat. Mater.* 2018, *17*, 691–696.
- [11] M. Savage, Y. Cheng, T. L. Easun, J. E. Eyley, S. P. Argent, M. R. Warren, W. Lewis, C. Murray, C. C. Tang, M. D. Frogley, G. Cinque, J. Sun, S. Rudić, R. T. Murden, M. J. Benham, A. N. Fitch, A. J. Blake, A. J. Ramirez-Cuesta, S. Yang, M. Schröder, *Adv. Mater.* 2016, 28, 8705–8711.
- [12] C. J. Doonan, D. J. Tranchemontagne, T. G. Glover, J. R. Hunt, O. M. Yaghi, *Nat. Chem.* **2010**, *2*, 235–238.
- [13] a) G. A. Jeffrey, G. A. Jeffrey, An introduction to hydrogen bonding, Vol. 12, Oxford University Press, New York, 1997;
  b) G. R. Desiraju, T. Steiner, The weak hydrogen bond: in structural chemistry and biology, Vol. 9, International Union of Crystallography, 2001.
- [14] A. W. Hewat, C. Riekel, Acta Crystallogr. Sect. A 1979, 35, 569– 571.

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