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OPEN Observation of superconducting vortex clusters in S/F hybrids

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While Abrikosov vortices repel each other and form a uniform vortex lattice in bulk type-II superconductors, strong confinement potential profoundly affects their spatial distribution eventually leading to vortex cluster formation. The confinement could be induced by the geometric boundaries in mesoscopic-size superconductors or by the spatial modulation of the magnetic field in superconductor/ ferromagnet (S/F) hybrids. Here we study the vortex confinement in S/F thin film heterostructures and we observe that vortex clusters appear near magnetization inhomogeneities in the ferromagnet, called bifurcations. We use magnetic force microscopy to image magnetic bifurcations and superconducting vortices, while high resolution scanning tunneling microscopy is used to obtain detailed information of the local electronic density of states outside and inside the vortex cluster. We find an intervortex spacing at the bifurcation shorter than the one predicted for the same superconductor in a uniform magnetic field equal to the thermodynamical upper critical field H_{c2}. This result is due to a local enhanced stray field and a competition between vortex-vortex repulsion and Lorentz force. Our findings suggest that special magnetic topologies could result in S/F hybrids that support superconductivity even when locally the vortex density exceeds the thermodynamic critical threshold value beyond which the superconductivity is destroyed.

Many of the potential applications of superconductors demand high current densities with minimal losses, a requirement that necessitates pinning of vortices. In the last decade several studies have focused on the influence of different types of pinning centers on the dynamics of superconducting vortices in type-II thin film superconductors¹⁻⁷, which resulted in better fundamental understanding of vortex physics. Compared to nano-structured inclusions made of normal metals or insulators, magnetic pinning leads to intriguing effects, such as magnetostatic interaction⁸, proximity effect⁹ and domain wall superconductivity¹⁰⁻¹³, to name a few. Moreover, magnetic pinning in magnetically coupled superconductor/ferromagnet (S/F) heterostructures, appears to be stronger than other pinning mechanisms^{4,14}. In particular, an enhanced pinning has been reported due to vortex coupling with ordered magnetic structures, such as stripes and dots^{15–17}. Random point magnetic defects can act as strong local pinning centers as well. Topological defects and singularities are often present in the microscopic structure of magnetic materials as has been shown by a variety of imaging methods: optical microscopies (magnetic Kerr and Faraday), electron-based microscopies (Lorentz, spin resolved scattering and photoemission), X-ray microscopies (transmission and circular dichroism), local probe microscopies (magnetic force and spin polarized scanning tunneling microscopy)^{18–26}. Some of these imaging techniques have been successfully used to investigate also the Abrikosov vortex distribution and vortex dynamics in superconducting thin films and S/F heterostructures²⁷⁻³³. Here, the existence of a stable mixed state results from the intrinsic repulsive interaction between superconducting vortices of the same polarity. It is known that in magnetic superconductors with easy magnetic axis, in anisotropic superconductors in tilted magnetic field and in high- κ S/F superlattices (where κ is the Ginsburg-Landau parameter, defined as the ratio of the London penetration depth λ to the coherence length ξ), attractive vortex-vortex interaction is responsible for the formation of vortex chains and vortex clusters³⁴⁻⁴¹. Nevertheless, unconventional vortex configurations such as vortex chains, vortex clusters as well as multi-vortex and giant vortex phases can also be induced by a strong magnetic or geometric confinement potential⁴²⁻⁴⁶.

¹"E.R. Caianiello" Physics Department, University of Salerno, Fisciano (SA), 84084, Italy. ²Physics Department, Temple University, Philadelphia, PA 19122, United States. ³CNR-SPIN Salerno, Fisciano (SA), 84084, Italy. ⁴Physics Department, Drexel University, Philadelphia, PA 19104, United States. ⁵Materials Science Division, Argonne National Laboratory, Argonne, IL 60439 United States. ⁶National University of Science and Technology (MISiS), Moscow, 119049, Russia. [†]Present address: University of Geneva, Department of Condensed Matter Physics, Geneva, Switzerland. Correspondence and requests for materials should be addressed to C.D.G. (email: cdigiorgio@unisa.it) The vortex confinement in peculiar superconductor/ferromagnet nanostructure formed by a superconducting microsquare with a magnetic disk on top has been theoretically addressed by Chen *et al.*⁴⁷.

Up to now, low temperature magnetic force microscopy (MFM) has been a useful tool to image mesoscopic vortex configurations: vortex clusters pinned by a periodic array of magnetic dots in S/F structures^{48,49} as well as vortex-antivortex pairs and vortex chains in magnetically coupled planar S/F bilayers⁵⁰⁻⁵³. On the other hand, scanning tunneling microscopy and spectroscopy (STM/STS) has been successfully used by Karapetrov *et al.*⁵⁴ to study vortex chain formation in S/F planar heterostructures and by Cren *et al.*⁵⁵ to study the geometrical confinement effects on the stabilization of superdense multivortex and giant vortex phases in mesoscopic superconductors.

In this work we study the superconducting vortex confinement in planar S/F thin film heterostructures caused by intrinsic topological defects in the magnetic template called *bifurcations*, by low temperature magnetic force microscopy and scanning tunneling microscopy and spectroscopy. We demonstrate that a bifurcation can naturally lead to unusual vortex distribution, and eventually to the formation of vortex clusters, without any need of invasively engineering the shape of the sample via lithography or self-assembly. Moreover, we exploit the complementarities of low temperature MFM and STM/STS techniques to obtain a thorough understanding of the magnetic and electronic properties of these hybrid systems.

Indeed, MFM is sensitive to the changes in local magnetization in the superconductor on the length scale of London penetration depth λ , and it has proven to be a useful tool in studies of local magnetic behavior of ferromagnetic materials as well as imaging Abrikosov flux quanta distribution in superconductors or superconducting heterostructures. However, MFM has limited spatial resolution of few tenths of nanometers, strongly dependent on tip's lift-height. On the other hand, STM/STS is sensitive to the amplitude of the superconducting order parameter, allowing detailed measurements of the electronic density of states (DOS) with a sub-nanometer spatial resolution, at the expense of not having any true magnetic vector contrast.

Here, we use MFM to get insight into the magnetic topology of a bifurcation as well as into its confinement power for superconducting vortices induced in the adjacent superconductor. At the same time we use STM/STS to get high resolution spatial maps of the electronic DOS of the superconductor just above the bifurcation sites in the ferromagnet.

We demonstrate that vortices above a bifurcation are closer than usual, producing vortex clusters and eventually reaching an intervortex distance which is even smaller than what expected at the higher critical field H_{c2} . Such a result, never reported before in planar S/F heterostructures, is one of the main findings of the paper and can be explained as a consequence of the local enhancement of the magnetic out-of-plane stray field, occurring at the bifurcation, coupled to the peculiar bifurcation topology.

Results

Samples. Several magnetically coupled S/F heterostructures, made by Nb/Py and Pb/[Co/Pd]_n multilayers, were fabricated for MFM and STM/STS investigations. A thin insulating layer of 10 nm SiO₂ or Al₂O₃ was used to electrically decouple the superconductor from the ferromagnet. This allows to study the effect of the spatially inhomogeneous stray field on the superconductivity without any interference from proximity coupling between the superconducting and magnetic layers⁹.

The thick Py films and Co/Pd multilayers are designed to have stripe magnetic domain pattern, with local magnetization vector alternating its projection along the out-of-plane direction between the adjacent magnetic stripes. In the Py, the in-plane easy magnetization axis is such that the canting angle of the magnetization vector is very small and magnetic stripe domains are formed only above the critical thickness of t_c \approx 200–300 nm. Above this threshold, the magnetic stripe width w grows as square root of the film thickness $d_m, w \propto \sqrt{d_m}^{56,57}$. On the other hand, Co/Pd multilayers have a much stronger perpendicular anisotropy. The appearance of an out-of-plane stripe configuration of magnetic domains is a consequence of the competition between the perpendicular magnetic anisotropy and the thin-film shape anisotropy. Moreover, materials with low and medium perpendicular anisotropy, such as Py, always show twisted structure at the surface (closure domains) with an in-plane orientation of the magnetization, perpendicular to the domain wall, able to confine part of the out-of-plane magnetic flux and lower the magnetostatic energy⁵⁸. The resulting domain structure consists of partially closed magnetic loops with alternating clockwise and counterclockwise orientation. In films with a strong perpendicular magnetic anisotropy the formation of closure domains becomes energetically unfavorable. However, recent magnetic simulations, based on the measured magnetic hysteresis loops, show the appearance of closure domains in a twin Co/Pd multilayers as well⁵⁹. In general, closure domains are difficult to observe because the majority of the conventional magnetic imaging techniques rely on the detection of the out-of-plane stray fields. They could be detected by using Kerr microscopy technique, which is sensitive to the in-plane magnetic field component⁶⁰.

It is worth noticing that the magnetic domain size and configuration in both ferromagnets are insensitive to the low magnetic fields applied during the field cooling processes performed in the MFM and STM/STS experiments here.

Twin superconducting films have been fabricated and characterized to derive the fundamental superconducting parameters. Twin Nb films have been magneto-electrically characterized, showing $\xi_{Nb}(0) \approx 12 nm$ and $\lambda_{Nb}(0) \approx 61 nm^{50,53}$. In such a case, Py films with thickness in the range of $1 \div 1.5 \,\mu\text{m}$ (and stripe widths w $\approx 500 \div 700 \,\text{nm}^{53}$) are good candidates for low temperature MFM experiments on Nb/Py hybrids. Indeed, being $\lambda_{Nb}(6K) \approx 68 \,nm^{53}$ at the operational temperature of T = 6 K, vortices can be widely accommodated within the Py(1 $\div 1.5 \,\mu\text{m}$) stripe domains.

On the other hand, it is well known that bulk Pb exhibits type-I superconductivity (with $\xi_{Pb,bulk}(0) \approx 83 nm$ and $\lambda_{Pb,bulk}(0) \approx 38 nm^{61}$). However, due to renormalization of the coherence length, Pb thin films behave as type-II superconductors and magnetic field penetrates in form of flux quanta. Moreover, in a twin 30-nm Pb film a

 $\xi_{Pb}(0) \approx 48 nm^{11}$ has been measured using STM vortex imaging, in agreement with values of $40 \div 50$ nm reported in literature for ultrathin Pb films⁶².

Different combination of superconductor and ferromagnet pairs were used in the S/F bilayers studied by MFM and STM/STS. For the superconducting materials, when limiting the choice to the simplest elemental superconductors, Nb and Pb are the best options. Indeed, due to their low intrinsic pinning, vortex arrangement can only be addressed to the ferromagnetic template. However, Nb is not a good candidate material for STM/STS investigation due to surface oxidation and difficulty to remove the oxide layer (possible only by annealing at temperatures higher than 1000 °C, which is not compatible with the underlying ferromagnet). On the other hand, Pb is not suitable for the MFM experiments, because of its low critical temperature that is too close to the minimum operational temperature of our instrument. Another important parameter to consider is the available scanning area. While STM provides the highest spatial resolution it does not allow large scanning areas since this compromises the stability of the tip-sample gap. Therefore, the largest STM scanning area of around 500 nm × 500 nm at T = 1.5 K requires a choice of ferromagnet with small stripe width and strong stray field, such as Co/Pd multilayers, to be able to visualize at least a few magnetic stripe domains within a single scan area and allow us to properly tune the intervortex distance with the applied magnetic field. However, Co/Pd multilayers are not suitable for MFM experiments, because of their relatively high out-of-plane stray magnetic fields inducing dense vortex arrangements that overlap on scale of λ .

In the following, MFM and STM/STS experiments on five S/F hybrids, Nb(100 nm, 150 nm, 200 nm)/Py(1 μ m), Nb(150 nm)/Py(1.5 μ m) and Pb(30 nm)/[Co(2 nm)/Pd(2 nm)]₂₀₀, are being presented. The same samples have been used in our previous study of spontaneous vortex formation in the presence of regular magnetic template^{53,63}. There we showed that vortices of opposite polarities form, i.e. vortices (V) and antivortices (AV), in absence of external magnetic fields whenever the out-of-plane magnetization of the ferromagnetic layer is above the threshold value calculated using the London approach to S/F hybrids in the mixed state^{64,65}. In particular, MFM experiments proved that Nb(150 nm)/Py(1 μ m), Nb(200 nm)/Py(1 μ m) and Nb(150 nm)/Py(1.5 μ m) have magnetization that is below the threshold value, preventing the formation of spontaneous V-AV pairs. In these samples Abrikosov vortex nucleation can be induced only by cooling in the presence of external magnetic field. On the other hand, in case of Nb(100 nm)/Py(1 μ m) out-of-plane stripe magnetization exceeds the threshold value resulting in spontaneous vortex nucleation observed in low temperature MFM measurements⁵³. Similarly, the presence of spontaneous V-AV structures in Pb(30 nm)/[Co(2 nm)/Pd(2 nm)]₂₀₀ has been obtained in STM/ STS experiments⁶³. These samples are good candidates for studies of the confinement effects by magnetic bifurcations in cases of both spontaneous and external field-induced vortices.

MFM observation of vortex clusters. Low temperature MFM measurements were performed on Nb/Py bilayers. Figure 1 shows the effect of the bifurcations on the vortex configurations in different situations. MFM maps acquired above and below the Nb critical temperature ($T_s = 8.9K^{53}$) in samples with three different Nb layer thicknesses (150 nm, 200 nm, 100 nm) while keeping the Py layer 1 μ m thick, are shown in Fig. 1. Above T_s , at T = 13 K, the images show the stripe-like domain pattern of Py, with a stripe width w \approx 500 nm, each of them containing dislocations of the regular magnetic structure. The significant increase of the magnetic contrast at the core of the bifurcation is representative of a stronger local stray field. Below the Nb T_s , superconducting vortices are favored to nucleate at the bifurcation. Moreover, vortices always appear on a partially shielded magnetic stripe contrast. Indeed, the Nb thin films do not completely shield Py magnetic stray field, allowing the MFM visualization of the magnetic stripe domains even below T_s . Nevertheless, a suppression of the magnetic contrast of the Py stripes due to the Nb diamagnetism, is confirmed by the comparison of the frequency span scale, above and below T_s .

Figure 1(d) shows the MFM image acquired on Nb(150 nm)/Py(1 μ m) at T = 6 K after a field cooling in the magnetic tip's field, which is a non-uniform local dipole field (as shown in the Supplementary Fig. S1). This sample's stripe magnetization is below the threshold for spontaneous formation of vortices and antivortices. Therefore, an external magnetic field is needed to facilitate nucleation of Abrikosov vortices. While cooling down below T_s, the tip was kept in the bottom-left corner of the image, spaced by only a few nanometers from the sample surface. In such a case, vortices of the same magnetic polarity as the tip are induced within few microns around the tip's apex. As shown in Fig. 1(d), a vortex is present in the middle of the dislocation, separated from its nearest neighbors by a distance sensibly smaller than the spacing between the vortices in the upper-right area of the image. Indeed, an average intervortex distance of about 1.2 μ m is measured between flux quanta around the dislocation, whereas the intervortex spacing on the top of the bifurcation is reduced by 35% to about 780 nm. We infer that such effect is driven by the bifurcation topology and by the local enhancement of the Py stray field.

On the other hand, MFM maps acquired at T = 6 K in Nb(200 nm)/Py(1 µm) (Fig. 1(e)) and Nb(100 nm)/ Py(1 µm) (Fig. 1(f)) show a strong magnetic contrast at the bifurcation site, surrounded by individual vortices nucleated in the presence of applied field of H = 30 Oe and in zero field cooling respectively. Line profile analysis performed on Nb(200 nm)/Py(1 µm) below T_s (Fig. 1(e)), confirms that superconducting vortices induced by the external field are mostly separated by a distance of (920 ± 105)nm, close to the expected value of 890 nm, derived by using $H = \frac{2}{\sqrt{3}} \frac{\Phi_0 66}{d^2}$, where H is the applied field, Φ_0 is the flux quantum and d is the distance between vortices. Far from the dislocation area, a vortex cluster formed by two separate vortices, spaced by a distance of around 150 nm, appears in the map in Fig. 1(e) (outlined by a dark blue dotted circle). We infer that in this specific case the intrinsic pinning in the Nb layer causes a non-uniform vortex distribution.

On the other hand, spontaneous V/AV formation in Nb(100 nm)/Py(1 μ m) is observed in Fig. 1(f), with vortices and antivortices induced by the stray fields of the Py stripes. The vortices are confined in blue and white dotted circles corresponding to vortex and antivortex confined to stripes with opposite magnetization polarity. Besides the vortices at the dislocation area, few spontaneous vortices populate the imaged area, being the



Attraction Frequency Span 0.6Hz Repulsion

Attraction Frequency Span 0.7Hz Repulsion

Attraction Frequency Span 2.0 Hz Repulsion

Figure 1. Vortex configurations at the bifurcation. (a)–(b)–(c) MFM maps acquired at T = 13 K and in zero applied magnetic field H = 0 on (a) Nb(150 nm)/Py(1 µm) at tip-sample distance h = 110 nm, (b) Nb(200 nm)/Py(1 µm) at h = 120 and (c) Nb(100 nm)/Py(1 µm) at h = 130. MFM maps acquired at T = 6 K and on (d) Nb(150 nm)/Py(1 µm) field cooled in the magnetic tip's field, (e) Nb(200 nm)/Py(1 µm) field cooled in H = 30 Oe, (f) Nb(100 nm)/Py(1 µm) zero field cooled. Each map is $3.8 \mu m \times 3.8 \mu m$ in size. In (e) the blue dotted circle points at a cluster of two vortices probably induced by a local Nb pinning. In (f), instead, blue and white dotted circles point at vortices and anti-vortices induced by the ferromagnet stray field.

out-of-plane magnetization of Py(1 μ m) very close to the threshold value for spontaneous flux quanta nucleation in Nb(100 nm)/Py(1 μ m)⁵³. Careful analysis of the magnetic contrast at the dislocation is required to distinguish whether the observed magnetic profile represents a vortex clusters or a giant vortex.

In Fig. 2, the 3D zoom-in of the MFM maps acquired on the top of the bifurcations in Nb(200 nm)/Py(1 μ m) (Fig. 2 (a)–(b)) and Nb(100 nm)/Py(1 μ m) (Fig. 2 (d)–(e)), above and below T_s, are compared. Keeping the same tip-sample separation, the magnetic contrast of Nb(200 nm)/Py(1 μ m) above and below the Nb superconducting transition appears quite different (Fig. 2(a,b)). A comparison of line profiles extracted from Fig. 2(a,b), above and below T_s is presented in Fig. 2(c). One observes significant change in the magnetic modulation. Indeed, two maxima of the magnetic roughness are clearly present below T_s, while two more kinks can be observed closer to the borders (marked by arrows), which might suggests the presence of a more complex vortex cluster. The average intervortex spacing is about 200 nm, whereas the average distance far from the dislocation is about (920 ± 105) nm. Similar analysis performed on Nb(100 nm)/Py(1 μ m), (Fig. 2(e)) does not show any drastic change in the magnetic roughness at the bifurcation core, compared to Fig. 2(d), acquired above T_s at the same tip-sample separation. The line profile comparison, reported on Fig. 2(f) and extracted from the white dotted profiles of Fig. 2(d)–(e), only confirms a reduction of the frequency shift below T_s is determined from its interaction with the MFM tip when it moves closer to the Nb surface (Supplementary Fig. S2).

The MFM experiments performed on a S/F sample having a thicker Py layer, Nb(150 nm)/Py(1.5 μ m), did not show any occurrence of spontaneous V-AV formation, proving that the equilibrium value of Py(1.5 μ m) out-of-plane magnetization has to be lower than the threshold required to nucleate spontaneous vortices in



Figure 2. Magnetic contrast at magnetic stripe bifurcation. (a)–(b) 3D zoom-in of MFM maps acquired at T = 13 K and T = 6 K on Nb(200 nm)/Py(1 μ m), focusing on the stripe bifurcation; (c) magnetic profile taken along white dotted lines of (a) and (b). Blue and black dots show the profiles above and below T_c, respectively. Arrows point at possible vortex position inside the cluster. (d)–(e) 3D zoom-in of MFM maps acquired at T = 13 K and T = 6 K on Nb(100 nm)/Py(1 μ m), focusing on the bifurcation; (f) magnetic profile taken along white dotted lines of (d) and (e). Blue and black dots draw the profiles above and below T_c, respectively. Each map is 2μ m × 2μ m in size.



Figure 3. Local enhancement of stray field at the bifurcation. Patchwork of two MFM images acquired at T = 6 K and at a tip-sample distance of h = 180 nm, after a zero field cooling, in adjacent and partially overlapping locations on Nb(150 nm)/Py(1.5 μ m) surface. Each map is $3.8 \,\mu$ m × $3.8 \,\mu$ m in size.

Nb(150 nm)^{53,64,65}. However, a strongly localized superconducting vortex nucleation can still be induced at the dislocation site, as shown in Fig. 3. Here, we present a patchwork made of two MFM maps of Nb(150 nm)/ Py(1.5 μ m) bilayer at T = 6 K after zero field cooling. In addition to the vortex nucleation on the top of the bifurcation, a vortex of opposite polarity (antivortex) is induced on the adjacent oppositely magnetized stripe and only a third red vortex appears in proximity of the dislocation. No other vortices (or antivortices) populate the imaged areas, confirming the magnetic influence of the dislocations on vortex nucleation. Indeed, not only that



Figure 4. Topological influence of the dislocation on vortex configuration. Patchwork of four MFM images acquired at T = 6 K and h = 110 nm, after a field cooling in H = 19 Oe, in adjacent and partially overlapping locations on Nb(150 nm)/Py(1 μ m) surface. The total size of the strip is $12 \,\mu$ m × $3.6 \,\mu$ m.

the vortex nucleation is favored at the bifurcation whenever the sample is over-threshold (Fig. 1(f)) or it is field cooled (Fig. 1(d)-(e)), but vortices could be locally induced by the dislocation even when no other spontaneous vortices are present. This also proves that the out-of-plane magnetization is under threshold everywhere except for the dislocation site, where a local enhancement of the stray field always occurs.

Finally, in Fig. 4, the topological influence of the bifurcation on vortex distribution is investigated. A strip of $12 \,\mu\text{m} \times 3.6 \,\mu\text{m}$ in size, made by a patchwork of MFM images acquired at T = 6 K in four adjacent and partially overlapping locations on Nb(150 nm)/Py(1 μ m) surface, is presented. The sample is field cooled in H = 19 Oe that corresponds to the first matching field $H_m = \frac{2}{\sqrt{3}} \frac{\Phi_0 66}{d^2}$ with intervortex distance *d* depending on the period of magnetization modulation in the ferromagnetic film (i.e. $d = \frac{2w}{\cos(30^\circ)} = \frac{4}{\sqrt{3}} w$). Indeed, after the field cooling, vortices nucleate on the underlying magnetic domains of the same polarity as the external field. Wherever the magnetic stripe pattern is straight and regular, as in the left half-image of Fig. 4, the hexagonal arrangement of vortex lattice occurs with an average intervortex spacing of $1.2 \,\mu\text{m}$, close to the expected value of $1.12 \,\mu\text{m}$. In the right half-image the regularity of the hexagonal array is lost and an inhomogeneous distribution of flux quanta with a higher vortex density is observed. The intervortex spacing in this area is reduced by 40% to 700 nm. Such irregular vortex arrangement is driven by the presence of two dislocations of the magnetic stripe structure. The curvature of two stripes converging at the bifurcation acts as a vortex accumulation area increasing the local vortex density.

STM/STS study of vortex clusters. Scanning tunneling microscopy and spectroscopy measures the local electronic density of states and could be used to discriminate between presence of Abrikosov vortex cluster and giant vortex. Despite the lack of magnetic field sensitivity, STM/STS allows vortex imaging with high lateral resolution up to very high magnetic fields. With STM one can infer the domain pattern in ferromagnetic layer in an indirect way: the nucleation sites of superconducting flux quanta after field-cooling the sample below T_s in opposite polarity magnetic fields, provides a unique fingerprint of the magnetic domains in the underlying ferromagnet¹¹.

Low temperature STM/STS experiments were performed on Pb(30 nm)/[Co(2 nm)/Pd(2 nm)]₂₀₀. As shown in Fig. 5(a) room temperature MFM on $[Co(2 nm)/Pd(2 nm)]_{200}$ confirms a stripe-like configuration of magnetic domains with occasional dislocation defects. Figure 5(b) shows the position of spontaneous superconducting vortices in Pb(30 nm)/[Co(2 nm)/Pd(2 nm)]₂₀₀ superimposed to a cartoon of the magnetic domains. This is a patchwork of five conductance maps of 438 nm × 438 nm in size, acquired at the Fermi energy at T = 1.5 K after zero field cooling. Vortices and antivortices appear as red spots with higher zero bias conductance (ZBC) compared to the superconducting background, which is here masked by the cartoon of the magnetic texture. An agglomeration of spontaneous flux quanta appears at the bifurcation core, where the vortex nucleation is favored because of the local stray field enhancement. The conductance map at the dislocation, in the green dotted square, at T = 1.5 K and in zero field cooling, is enlarged in Fig. 5(c). Three hot spots of conductance are present inside the cluster, which is surrounded by other two individual and isolated vortices. This strongly inhomogeneous vortex distribution is an evidence of the strong influence of magnetic defects on vortex arrangement.

In Fig. 6(a),(b) we show ZBC maps acquired at T = 1.5 K in the yellow dotted area of Fig. 5(b) (just below the bifurcation) after a field cooling in H = -300 Oe and H = 300 Oe, respectively. A conductance distribution of individual and well-separated vortices appears in Fig. 6(a), while, in the opposite field antivortices agglomerate at the end point of the interrupted stripe (Fig. 6(b)). Here, spectroscopic analysis confirms the nucleation of a vortex cluster, formed by three individual flux quanta. The plot of Fig. 6(c) shows the evolution of the superconducting density of states (DOS) along the black dotted line of Fig. 6(b). The tunneling spectra gradually evolve while approaching the vortex cluster and they are characterized by an increase in the zero bias conductance and a decrease of the coherence peaks' height. Clearly the superconductivity is fully suppressed in three separate locations, where a zero bias peak is observed (red dotted curves in Fig. 6(c)). The presence of a pronounced zero-bias peak in the conductance spectra at the vortex core is a clear evidence of the superconducting clean limit regime^{67,68}. Finally, superconducting DOS features are completely recovered outside of the cluster. The plot of the normalized ZBC as a function of position, reported in Fig. 6(d), shows three ZBC peaks inside the cluster (indicated as A, B and C), separated by 105 nm (A and B) and 109 nm (B and C). Surprisingly, the resulting intervortex separation is about 1.4 times smaller than the minimum value possible, achievable at the second critical field



Figure 5. STM observation of vortex clusters at the bifurcation. (a) Room temperature MFM of $[Co(2 \text{ nm})/Pd(2 \text{ nm})]_{200}$ multilayer; (b) spontaneous superconducting vortices in Pb(30 nm)/ $[Co(2 \text{ nm})/Pd(2 \text{ nm})]_{200}$ superimposed onto a cartoon of the magnetic domains. The vortices have been mapped by acquiring five conductance maps at the Fermi energy, each map being of 438 nm × 438 nm in size (T = 1.5 K after a zero field cooling); (c) conductance map at the dislocation (the location is the green dotted square in (b)), acquired at the Fermi energy and at T = 1.5 K in zero field cooling.

Figure 6. Vortex Cluster at the stripe endpoint revealed by STM. (a)–(b) Zero bias conductance maps acquired around the end point of an interrupted stripe at T = 1.5 K after field cooling in H = -300 Oe and H = 300 Oe, respectively; (c) plot of the spatial evolution of the electronic DOS along the dotted line in (b). Red dotted curves show pronounced zero bias peaks; (d) normalized ZBC vs position along the dotted line in (b).

 $d(H_{c2}) \approx 2.8 \times \xi$ for fully separated vortices in bulk superconductors. Moreover, a rough estimate of $\xi(1.5K)$ can be deduced by the vortex profiles (red dotted arrow of Fig. 6(d)) resulting in $\xi(1.5K) \approx 50$ *nm*, which is in good agreement with previous estimates on our twin samples¹¹.

It is worth noticing that the size of the cluster that occupies the top of the bifurcation (Fig. 5(c)) is fully consistent with an agglomeration of three flux quanta, each of them having $\xi(1.5K) \approx 50 nm$ (Supplementary Fig. S3).

Figure 7. Schematics of the closure domains. (a)–(b) Front view and 3D schematics of an ideal magnetic stripe arrangement in the presence of closure domains; (c) schematics of the magnetic configuration around the bifurcation.

Discussion

Direct observation of superconducting vortex cluster and vortex accumulation as a consequence of topological defects of the stripe magnetic structure in S/F thin film hybrids, made by Nb/Py and Pb/ $[Co/Pd]_n$, has been carried out by using MFM and STM/STS techniques.

In general in S/F hybrids, with S in the mixed state, the position-dependent Lorentz force due to the screening currents above the magnetic domains, pushes the vortices to the center of the magnetic domains, thus minimizing the interaction energy between the superconducting vortices and the magnetic template^{65,69}. In the presence of more than one flux quantum, the vortex-vortex repulsive interaction $U_{VV} = \sum_{i,j} \frac{\Phi_0^2}{8\pi^2 |\vec{r_i} - \vec{r_j}|}$ needs to be taken into

account. In bulk superconductors vortices distribute in a hexagonal lattice, keeping a constant intervortex spacing given by $d = \sqrt{\frac{2}{\sqrt{3}} \frac{\Phi_0}{H}}$. On the other hand in S/F hybrids with F in the stripe regime, the strong confinement exerted by the magnetic template leads to a chain-like distribution of vortices that eventually exhibit zig-zag displacement at high fields in wide stripes⁵⁰. In such a scenario, vortex clusters can be explained as a consequence of several factors: a local stray field enhancement, a local reduction of the intervortex repulsive force as well as a local change in the Lorentz force action.

MFM measurements clearly show that a bifurcation, where two magnetic stripe domains converge and coalesce in a single one, leads to a local enhancement of the out-of-plane stray field. In Fig. 7(a),(b) a front view and a 3D sketch of an ideal magnetic stripe arrangement is shown. Blue and red stripes are representative of magnetic domains with opposite out-of-plane polarization. Closure domains in the shape of triangular structures with an in-plane orientation of the magnetization (yellow dotted arrows) are at the sample surface depicted in purple. Such a magnetization distribution is consistent with the magnetic configuration of Co/Pd multilayers, whereas in Py an additional strong in-plane component of the magnetization along the stripes needs to be considered. Magnetic configuration around the bifurcation is presented in Fig. 7(c). The closure domains, indicated by the yellow dotted arrows, are forced to follow the domain wall curvature. As a consequence, an in-plane magnetic flux surplus (resulting from closure domains in Co/Pd multilayers and from closure domains and in-plane stripe magnetization in Py) needs to be expelled by flowing out from the sample surface via the out-of-plane stray field. This gives rise to the enhancement of the out-of-plane magnetization at the bifurcation core which leads to the enhancement of the local stray field that in-turn induces vortex clusters, as demonstrated by MFM and STM experiments.

Moreover, STS measurements on Pb/[Co/Pd]_n provided evidence for the existence of a vortex cluster spaced by a distance shorter than the minimum value achievable at the second critical field $d(H_{c2}) \approx 2.8 \times \xi$. We infer that such a condition can be explained by taking into account the role of the bifurcation as topologically induced confinement. Indeed, while each vortex inside an infinite chain would feel the same net repulsive force, which leads to a constant intervortex distance, an unbalanced force is felt by vortices close to magnetic channel interruptions (stripe endpoint or bifurcation core). For instance, the vortex at the stripe endpoint feels a long-range repulsive interaction due to the semi-infinite vortex chain on one side, while on the other side only the Lorentz force would keep it away from the domain wall. In such a case a reduction of the intervortex distance close to the magnetic stripe endpoint is expected.

The bifurcation topology indirectly affects the vortex distribution at the nearest neighbor domains as well. As clearly shown in Fig. 4, hexagonal vortex lattice with an average lattice parameter of about $1.2 \,\mu\text{m}$ is achieved at the matching field⁶⁶ wherever the stripes are straight and regular. On the other hand, around the dislocations the intervortex distance is affected by the stripe curvature, leading to a vortex-vortex spacing of about 700 nm. In the presence of straight magnetic domains, the Lorentz force vectors, being always normal to the domain walls where

supercurrents flow, push vortices to the middle of the stripes. In such a case, the vortex distribution along the stripe is set by their repulsive interaction producing uniform intervortex spacing determined by the net magnetic field present - sum of the ferromagnet stray field and the external applied field. Instead, whenever there is a curvature of the domain walls, the Lorentz force vectors are locally not parallel, causing modulations of intervortex distance. At the bifurcation region the local bending of the stripes is extreme, leading to strong bending of the Lorentz force vectors that results in effective topological vortex confinement.

In conclusion, we have shown that randomly distributed defects in underlying magnetic structures, such as bifurcations, act as strong pinning sites that induce the nucleation of vortex clusters. Such dislocations strongly affect superconducting vortex distribution because of the magnetic confinement power coming from the local enhancement of the stray field and magnetic domain intrinsic topology. The topology of the magnetization at the bifurcation points in the ferromagnetic layer facilitates creation of vortex clusters in the superconductor, with vortices confined by the strong local Meissner currents while at the same time experiencing mutual repulsion.

Methods

Sample preparation. Nb/SiO₂/Py were fabricated *ex-situ* by sputtering deposition and moved to the UHV chamber where the MFM experiments were performed. Py films were deposited by dc sputtering from a Ni₈₀Fe₂₀ target onto a Si substrate at a base pressure of 1.5×10^{-7} Torr, followed by a 10-nm SiO₂ layer, in order to decouple the F from the S layer, suppressing proximity effects⁹. Nb films were deposited by dc sputtering at room temperature in a dedicated system with a base pressure of 2×10^{-8} Torr.

Samples for STM studies were fabricated in a different way to preserve their high surface quality. The $[Co/Pd]_n$ multilayers (where *n* is the number of bilayers in the stack) were deposited *ex-situ* on Si substrates by dc sputtering in a dedicated system and in the presence of an applied in-plane magnetic field, favoring a stripe-like magnetic domain pattern. A 10-nm Al₂O₃ film was made by RF sputtering deposition from an Al target, in order to insulate the F from the S layer and suppress the proximity effect⁹. The 30-nm Pb film was deposited *in-situ* via e-beam evaporation at low temperature (120 K) and base pressure of 10⁻¹¹ Torr, followed by a room temperature annealing. This procedure guaranties flat and clean surfaces, suitable for STM studies. The UHV chamber, where Pb films were made is linked to the STM chamber where the experiments were performed, in order to avoid surface contamination due to the exposure of the films to the air.

Magnetic Force Microscopy. MFM experiments on Nb/Py where performed at T = 6 K using an Omicron cryogenic ultra-high vacuum (UHV) scanning force microscope, operating in frequency modulation-MFM mode. A commercial Si cantilever, equipped with a magnetic tip and having a resonance frequency $f_0 \approx 75$ kHz and elastic constant $k \approx 2.8$ N/m, was used. The tip, coated by a ferromagnetic Co/Cr film, was characterized by nominal low moment $\mu \approx 0.3 \times 10^{-13}$ emu and measured coercivity $H_{c,tip} \approx 550 \div 600$ G. The magnetic imaging was done by scanning in non-contact regime and by mapping line by line the frequency shift $df = f - f_0$ of the resonating cantilever, due to the stray field coming out from the sample (f is the oscillation frequency measured during tip-sample interaction and f_0 is the force free resonant frequency of the cantilever). MFM maps were obtained by scanning at constant tip-sample heights, between $110 \div 180$ nm, and the attractive/repulsive tip-sample interaction was mapped using color contrast. MFM spatial resolution is determined by the lift height used during the scanning and limited by the large magnetic volume of the tip.

Scanning Tunneling Microscopy and Spectroscopy. STM/STS experiments were performed at T = 1.5 K by using a cryogenic UHV Unisoku USM-1300 system, equipped with RHK electronics. Tunneling spectroscopy was performed by using a standard lock-in technique with an alternating current modulation of 0.2 mV at 373 Hz. The conductance maps were acquired by scanning the tip over the sample surface at high voltage (20 mV), acquiring the lock-in signal at the Fermi energy in each location. Conductance maps reveal therefore, vortices by tracking the difference of the electronic density of states inside and outside the vortex cores. Topography was always acquired simultaneously to check the location where the spectroscopic information was recorded. All differential conductance spectra dI/dV were taken with the same tunneling parameters, with the junction stabilized at V = 10 mV, I = 100 pA.

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Author Contributions

C.D.G. performed low temperature MFM experiments, carried out MFM and STM/STS data analysis and wrote the manuscript. F.B. performed low temperature MFM experiments and contributed to the manuscript preparation. M.I. proposed the research project, carried out the STM/STS experiments and contributed to the manuscript preparation. G.K. proposed the research project, prepared some of the samples used for this study, characterized the transport properties of Nb. A.S. performed low temperature MFM experiments. S.A.M. carried out the STM/STS experiments, fabricated and characterized some of the magnetic materials used in this study and fabricated the Pb films. D.A.D. contribute to the data analysis. V.Y. fabricated the Nb films. V.N. supervised the preparation of magnetic materials and Nb/Py bilayers. A.M.C. contributed to the manuscript preparation. All the authors discussed the results and contributed to the manuscript preparation.

Additional Information

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