

# Temporal Dynamics of Solid-State Thermally Activated Delayed Fluorescence: Disorder or Ultraslow Solvation?

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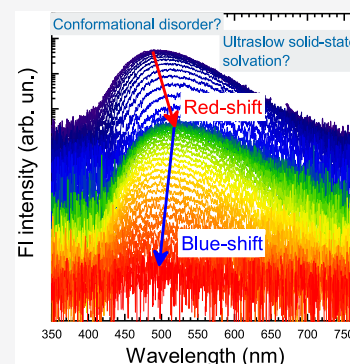


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Supporting Information

**ABSTRACT:** Time-resolved emission spectra of thermally activated delayed fluorescence (TADF) compounds in solid hosts demonstrate significant temporal shifts. To explain the shifts, two possible mechanisms were suggested, namely, slow solid-state solvation and conformational disorder. Here we employ solid hosts with controllable polarity for analysis of the temporal dynamics of TADF. We show that temporal fluorescence shifts are independent of the dielectric constant of the solid film; however, these shifts evidently depend on the structural parameters of both the host and the TADF dopant. A  $\leq 50\%$  smaller emission peak shift was observed in more rigid polymer host polystyrene than in poly(methyl methacrylate). The obtained results imply that both the host and the dopant should be as rigid as possible to minimize fluorescence instability.



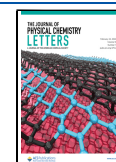
The most efficient organic light-emitting diode (OLED) devices are made of emissive thermally activated delayed fluorescence (TADF) compounds.<sup>1–3</sup> The high emission yield of TADF compounds is achieved by minimizing the singlet–triplet energy gaps ( $\Delta E_{ST}$ ) and enabling thermally activated reverse intersystem crossing of triplet states.<sup>4,5</sup> Among the approaches that are suggested to enable TADF, like the multiresonance effect<sup>6</sup> or hot-exciton pathway,<sup>7</sup> the most popular is based on constructing TADF compounds from electron-donating (D) and electron-accepting (A) fragments.<sup>8</sup> In this case, a lower  $\Delta E_{ST}$  is achieved by minimizing the electron exchange energy, which in fact depends on the HOMO–LUMO overlap.<sup>9</sup> Therefore, building the molecular compounds of D and A units decouples HOMO and LUMO and enables triplet upconversion.<sup>10</sup> The efficient separation of frontier molecular orbitals could be achieved either by twisting D and A units toward orthogonality<sup>11</sup> or by keeping the molecular structure rather flat but with extended conjugation in the HOMO and LUMO.<sup>12,13</sup> Despite the chosen design strategy, TADF compounds are still composed of singly bonded molecular units with high twisting flexibility and pronounced charge-transfer character. TADF decay processes in solutions are rather straightforward; however, in the solid state, they are accompanied by temporal spectral shifts and increased delayed fluorescence lifetimes,<sup>14–19</sup> which is disadvantageous for OLEDs, as it accelerates efficiency roll-off at large luminance values.<sup>20</sup> To date, two possible mechanisms have been proposed to explain the temporal dynamics of solid-state TADF. The first relies on conformational disorder<sup>14</sup> and suggests that the solid surrounding locks

the TADF structures in a variety of D–A twisting conformations, having different singlet–triplet energy gaps<sup>14–19</sup> and, therefore, different radiative decay rates ( $k_r$ ) and emission wavelengths. In the case of prompt fluorescence (PF), initial decay occurs from the conformer states with the largest  $k_r$  and thus the largest singlet energy. In contrast, the later decay is associated with conformers with a lower  $k_r$  and a smaller singlet energy, hence enabling the red-shift of prompt fluorescence (PF). In the case of delayed fluorescence (DF), the trend is the opposite as the molecular conformations with the lowest radiative decay rate (stronger charge-transfer character) will have the smallest  $\Delta E_{ST}$  and therefore the most rapid DF, enabling its blue-shift. Moreover, DF properties tend to be strongly dependent on the TADF dopant concentration due to the selective concentration quenching of long-lived triplet species.<sup>21</sup> Solid-state solvation<sup>22–25</sup> was suggested as the second mechanism responsible for temporal DF shifts. From this point of view, the stabilization of excited TADF molecules is ultraslow due to the hindered dipole orientation of the host matrix, enabling temporal shifts of emission up to milliseconds. It is noteworthy that larger emission shifts are typically found in more polar solid hosts providing stronger dipole–dipole interaction

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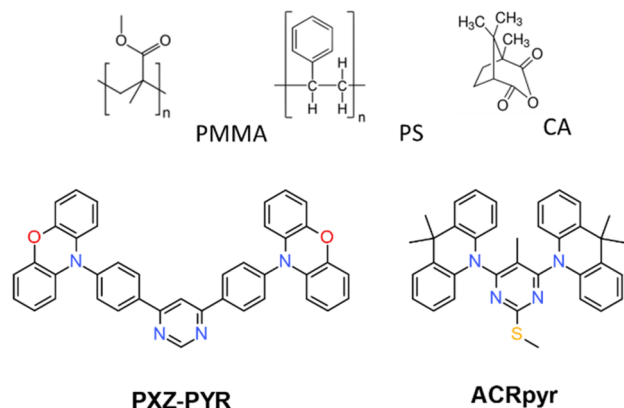
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between the dopant and host.<sup>24</sup> On the contrary, there are experimental and numerical reports stating that slow solid-state solvation (SSS) is actually ultrafast and, thus, cannot explain the temporal dynamics of TADF in the solid state.<sup>14,26</sup> Therefore, knowing the actual mechanism is crucial, as the increase in emission yield is usually achieved by embedding TADF dopants in polar hosts, fostering the decrease in  $\Delta E_{\text{ST}}$ .<sup>25</sup>

To date, most of the studies analyzing the temporal behavior of solid-state TADF are performed in different hosts, based on either polymer or small-molecule compounds.<sup>15,22,24</sup> However, various host materials differ in more than one parameter. For example, although DPEPO {bis[2-(diphenylphosphino)phenyl]ether oxide} and PYD2 [2,6-bis(9H-carbazol-9-yl)pyridine] are two widely used hosts with different dielectric constants ( $\epsilon$ ),<sup>14</sup> a simple comparison of emission properties in those hosts cannot discern the polarity<sup>15</sup> from disorder effects.<sup>24</sup> This is because compounds have completely different molecular structures, implying their different rigidities. DPEPO is thought to be a less rigid host than PYD2.<sup>14</sup> In such a case, the less restrictive environment of DPEPO could enable greater conformational disorder causing larger temporal shifts of DF as compared to the shifts imposed by polarity. Therefore, it is important to discern the impact of polarity and rigidity and analyze both processes independently.

In this paper, we present a brief approach to the spectroscopic analysis of individual solvation and disorder effects in the solid-state TADF. Two polymer hosts with different molecular structures, polystyrene (PS) and poly(methyl methacrylate) (PMMA), were selected and later selectively doped with camphoric anhydride (CA) having a large ground-state dipole moment<sup>26–29</sup> (see Figure 1). This

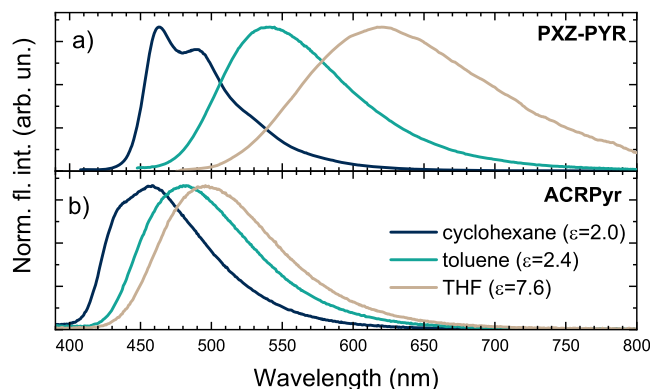


**Figure 1.** Molecular structures of PMMA, PS, CA, PXZ-PYR, and ACRPyr.

allowed broad-range tuning of the  $\epsilon$  values of PS and PMMA hosts.<sup>26</sup> The molecular alignment of CA is more rapid than that of the polymer host,<sup>24,26</sup> though both should contribute to the solvation. More pronounced interaction between CA and polar units in the polymer backbone is expected at larger CA doping loads through the dipolar communication, later leading to altered dipolar interactions between the excited TADF dopant and the ground-state host. Such subtle variations should also affect the PF shifts. In this study, PXZ-PYR<sup>30</sup> and ACRPyr<sup>31</sup> compounds possessing more and less flexible molecular structures, respectively, were selected as TADF emitters. The comparison of emission properties of these emitters in the same host but featuring different  $\epsilon$  values

allowed us to visualize the polarity effects; meanwhile, the disorder effects were assessed by comparing the temporal dynamics of TADF compounds dispersed in hosts with the same polarity but a different molecular structure.

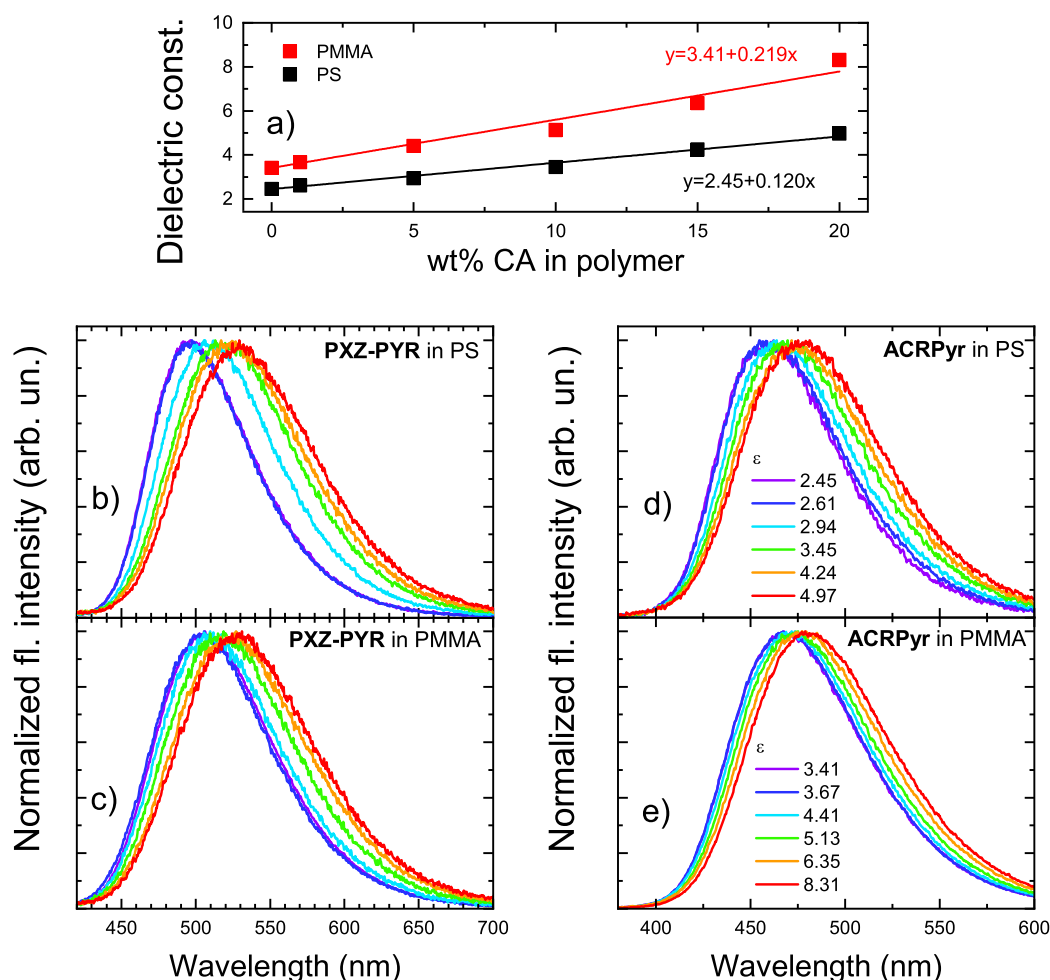
At first, a solvatochromic study of the compound solutions was performed (see Figure 2). Typical behavior for charge-



**Figure 2.** Fluorescence spectra of PXZ-PYR and ACRPyr in dilute cyclohexane, toluene, and THF solutions.

transfer (CT) compounds was observed, namely, the red-shift of the emission peak and an increase in the full width at half-maximum (fwhm).<sup>32</sup> Remarkably stronger solvatochromic behavior was observed for PXZ-PYR with an  $\sim 680$  meV red-shift, compared to an  $\sim 220$  meV shift for ACRPyr when utilizing a solvent with increasing polarity [from cyclohexane to tetrahydrofuran (THF)]. To assess the controllable solvation in the solid state, camphoric anhydride was employed. A high ground-state dipole moment with a small CA molecule ( $\sim 6$  D<sup>27</sup>) enabled the selective tuning of the polarity of polymer films when doped with CA at different loads<sup>26–29</sup> (see Figure 3a). As one can see, the dielectric constant of polymer films can be tuned in a broad range of 2.45–4.97 for PS (same as in the report by Madigan et al.<sup>27</sup>) and 3.41–8.31 for PMMA with a linear  $\epsilon$  dependence on CA doping load in the range of 0–20 wt % (see also Table S1). CA was transparent up to nearly 260 nm, being optically inactive in the wavelength range of PXZ-PYR and ACRPyr (see Figure S1). No traces of any aggregate species were observed in the absorption spectra of PXZ-PYR and ACRPyr in PS and PMMA hosts doped with CA (see Figure S2).

Fluorescence spectra of PXZ-PYR and ACRPyr in PS and PMMA films with different CA doping loads are shown in Figure 3b–e. Like solutions, typical positive solvatochromic behavior was observed for emission spectra (see Table S2 for details). Emission peak shifts of 150 and 110 meV for PXZ-PYR and 100 and 70 meV for ACRPyr in PS and PMMA films, respectively, were observed. fwhm values of emission spectra followed the same trend, systematically increasing for 15% or 7% for PXZ-PYR and 12% or 2% for ACRPyr in PS and PMMA films. The solvatochromic emission shift of PXZ-PYR was like that of 2PXZ-OXD,<sup>28</sup> a TADF dopant with a similar molecular structure ( $\sim 150$  meV for the CA doping range of 0–20 wt %). SSS behavior was weaker than in solutions due to the more restrictive solid surrounding.<sup>28</sup> As in solutions, ACRPyr showed a smaller shift of the emission peak with an increase in  $\epsilon$ , most likely due to the lower excited-state dipole moment.<sup>14</sup> This was also evident from Lippert–Mataga plots of PXZ-PYR and ACRPyr in PS and PMMA films,



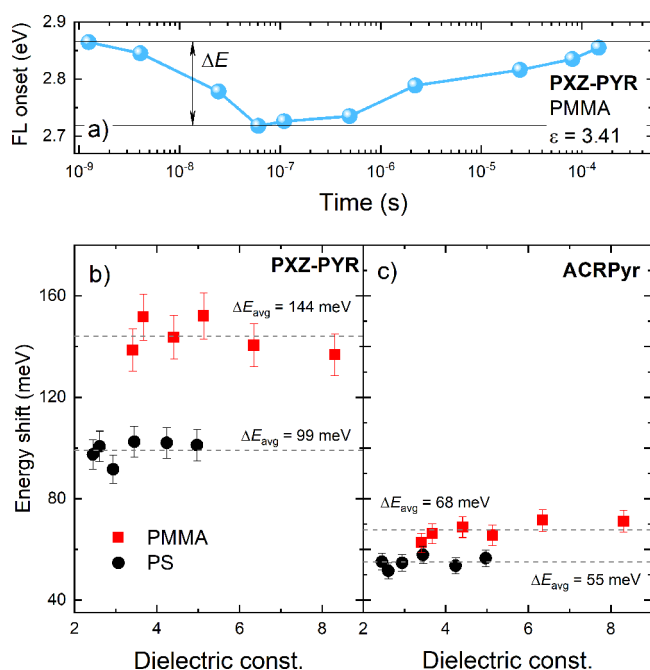
**Figure 3.** (a) Dielectric constants of PS (black) and PMMA (red) films with different CA doping loads. Color lines are linear fits. Fluorescence spectra of PXZ-PYR and ACRPyR in PS (b and d, respectively) and PMMA (c and e, respectively) films with different dielectric constants (different CA doping loads). The concentration of the TADF dopant was set at 1 wt %.

relating the Stokes shift and the change in the dipole moment upon photoexcitation (see Figure S3). Such a positive SSS course implies stronger stabilization of the excited state in a more polar surrounding. Therefore, if dipole–dipole interactions between the host and dopant are the driving force of slow temporal emission dynamics, increasing the dielectric constant of a polymer film should alter the temporal fluorescence shifts.<sup>22,24</sup>

To test this assumption, time-resolved fluorescence spectra of PXZ-PYR and ACRPyR were measured in PS and PMMA films with additional CA doping (see Figure S4) and energy shifts were estimated (see Figure 4). The magnitude of the energy shift ( $\Delta E$ ) was described as the difference in the spectral onsets of the initial PF spectrum (at a 1.25 ns delay) and the latest PF spectrum (at an  $\sim 100$  ns delay)<sup>16</sup> (see Figures S4 and S5). No evident changes in  $\Delta E$  were found upon the increase in the level of CA doping (see Figure 4b,c), though it can be suggested that solvation may not be the driving force of temporal PL shifts.  $\Delta E$  values were nicely scattered around the average energy shift value ( $\Delta E_{\text{avg}}$ ). Although  $\Delta E_{\text{avg}}$  was independent of the dielectric constant of the solid host, a clear correlation between the magnitude of the energy shift and the structural properties of the polymer and TADF dopant was observed. Such behavior, on the contrary, can be explained by a conformational disorder concept.<sup>14–18,31</sup>

$\Delta E$  clearly was lower for ACRPyR than for PXZ-PYR (144 meV vs 68 meV and 99 meV vs 55 meV in PMMA and PS, respectively). The main difference in molecular structure between PXZ-PYR and ACRPyR is the rotational flexibility of the donor units.<sup>16</sup> Acridine in ACRPyR is sterically constrained by the 5-methyl unit with limited ability to rotate along the C–N axis.<sup>31</sup> In the solid state, such a limited distribution of D–A twisting angles leads to a low dispersion of singlet-state energies, singlet–triplet energy gaps, and, therefore, small PF and DF peak shifts. In the case of PXZ-PYR, the phenoxazine donor unit is confined by only hydrogens in the spacing phenyl unit, leading to evidently greater rotational flexibility.<sup>30</sup> Here, flexible PXZ-PYR has a wider distribution of  $S_1$  energies, leading to more pronounced temporal emission peak shifts. Moreover,  $\Delta E$  was evidently smaller in PS films for both compounds, particularly for PXZ-PYR (144 meV vs 99 meV and 68 meV vs 55 meV for PXZ-PYR and ACRPyR, respectively). The conformational disorder (described by the magnitude of  $\Delta E$ ) is also controlled by the host parameters.<sup>14</sup> A more rigid host has a more tightly packed environment with a reduced D–A twisting angle distribution. Therefore, in this case, PS could be regarded as more rigid polymer host, probably due to the presence of a rigid phenyl unit rather than a flexible ester fragment in PMMA; however, the exact nature is still unclear and needs more in-depth analysis. The





**Figure 4.** (a) Temporal dynamics of PXZ-PYR in a PMMA film with a 0 wt % CA doping load.  $\Delta E$  shows the energy shift of PF.<sup>16</sup> Energy shifts for (b) PXZ-PYR and (c) PXZ-PYR in PS (black) and PMMA (red) films with different  $\epsilon$  values. Error bars are also included.

combination of a rigid molecular structure and a rigid polymer host implied minimized conformational disorder for ACRPYR in PS with a  $\Delta E$  of only 55 meV. Further reduction of spectral instability probably could be achieved by selecting even more rigid hosts, like well-known OLED host PYD2.<sup>14</sup>

In conclusion, seeking to reveal the solvation role in the temporal emission dynamics, we performed the photophysical analysis of TADF compounds doped in two different polymer hosts with selectively tuned polarity. No evident dependence of the temporal shifts of prompt and delayed fluorescence on the dielectric constant of the polymer film was found. If solid-state solvation is the driving force of temporal emission dynamics, somewhat different shifts should have been observed in a more polar surrounding. However, the observed behavior was in accordance with a conformational disorder mechanism. Explicitly, a more sterically restricted molecular structure of the ACRPYR dopant favored smaller emission shifts, which were further diminished by employing a more rigid polymer host. Such enhanced spectral stability of TADF compounds is highly preferable for rapid triplet upconversion. For further proof of this mechanism, in-depth solvation studies in a more diverse solid surrounding are necessary. We believe our findings will improve our better understanding of the design concepts of TADF compounds and foster efficiency roll-off optimization in OLED devices.

## EXPERIMENTAL METHODS

Synthesis and characterization of PXZ-PYR and ACRPYR were described previously.<sup>30,33,34</sup> Photophysical properties of PXZ-PYR and ACRPYR were analyzed in PS and PMMA (Sigma-Aldrich) films at a concentration of 1 wt %. Solid-state films were prepared by dissolving each material at the appropriate ratio in toluene solutions and then wet-casting the solutions on quartz substrates in air. The dielectric constant of the polymer host was altered by adding various amounts of ( $\pm$ )-camphoric

anhydride (CA)<sup>28,29</sup> (Fine Synthesis Ltd.). Comparable pore radii of both polymers allowed us to compare SSS effects after doping with CA.<sup>26</sup> The dielectric permittivity was estimated using the CELIV technique.<sup>35</sup> Linearly increasing the voltage pulse was applied to neat and doped ITO/PS/Al (thickness of PS  $d = 200$  nm) and ITO/PMMA/Al (thickness of PMMA  $d = 300$  nm) sandwich structures. The thickness of the polymer film was estimated via a Dimension Icon atomic force microscope (Bruker). Initial current density step  $j_0$  was measured at a voltage rise speed  $A$  of 1 V/10 ms. The dielectric permittivity of polymer films was estimated according to equation  $\epsilon = j_0 d / A \epsilon_0$ . Time-integrated and time-resolved fluorescence spectra were recorded using a nanosecond YAG:Nd<sup>3+</sup> laser NT 242 (Ekspla,  $\tau = 6$  ns, pulse energy of 200  $\mu$ J, repetition rate of 1 kHz) and time-gated iCCD camera New iStar DH340T (Andor). Fluorescence decay transients were obtained by exponentially increasing the delay and integration time.<sup>36</sup> The emission intensity was obtained by integrating all emission spectra. Solid-state samples were mounted in a closed cycle He cryostat (Cryo Industries 204N) for all measurements seeking oxygen-free conditions.

## ASSOCIATED CONTENT

### Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acs.jpclett.1c03810>.

CA absorption spectrum, absorption spectra of PXZ-PYR and ACRPYR at different CA doping loads, Lipper–Mataga plots, and time-resolved PL spectra (PDF)

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## Notes

The authors declare no competing financial interest.

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